

Crystallography and physical properties of BaCo_2As_2 , $\text{Ba}_{0.94}\text{K}_{0.06}\text{Co}_2\text{As}_2$, and $\text{Ba}_{0.78}\text{K}_{0.22}\text{Co}_2\text{As}_2$ V. K. Anand,^{1,*} D. G. Quirinale,¹ Y. Lee,¹ B. N. Harmon,¹ Y. Furukawa,¹ V. V. Ogloblichev,² A. Huq,³ D. L. Abernathy,⁴ P. W. Stephens,⁵ R. J. McQueeney,^{1,†} A. Kreyssig,¹ A. I. Goldman,¹ and D. C. Johnston^{1,‡}¹Ames Laboratory and Department of Physics and Astronomy, Iowa State University, Ames, Iowa 50011, USA²Institute of Metal Physics, Ural Division of Russian Academy of Sciences, Ekaterinburg 620990, Russia³Spallation Neutron Source, Oak Ridge National Laboratory, Oak Ridge, Tennessee 37890, USA⁴Quantum Condensed Matter Division, Oak Ridge National Laboratory, Oak Ridge, Tennessee 37831, USA⁵Department of Physics and Astronomy, SUNY at Stony Brook, Stony Brook, New York 11794, USA

(Received 27 June 2014; revised manuscript received 15 August 2014; published 28 August 2014)

The crystallographic and physical properties of polycrystalline and single-crystal samples of BaCo_2As_2 and K-doped $\text{Ba}_{1-x}\text{K}_x\text{Co}_2\text{As}_2$ ($x = 0.06, 0.22$) are investigated by x-ray and neutron powder diffraction, magnetic susceptibility χ , magnetization, heat capacity C_p , ^{75}As nuclear magnetic resonance (NMR), and electrical resistivity ρ measurements versus temperature T . The crystals were grown using both Sn flux and CoAs self-flux, where the Sn-grown crystals contain 1.6–2.0 mol% Sn. All samples crystallize in the tetragonal ThCr_2Si_2 -type structure (space group $I4/mmm$). For BaCo_2As_2 , powder neutron diffraction data show that the c -axis lattice parameter exhibits anomalous negative thermal expansion from 10 K to 300 K, whereas the a axis lattice parameter and the unit cell volume show normal positive thermal expansion over this T range. No transitions in BaCo_2As_2 were found in this T range from any of the measurements. Below 40–50 K, we find $\rho \propto T^2$, indicating a Fermi liquid ground state. A large density of states at the Fermi energy $\mathcal{D}(E_F) \approx 18$ states/(eV f.u.) for both spin directions is found from low- T $C_p(T)$ measurements, whereas the band-structure calculations give $\mathcal{D}(E_F) = 8.23$ states/(eV f.u.). The enhancement of the former value above the latter is inferred to arise from electron-electron correlations and the electron-phonon interaction. The derived intrinsic $\chi(T)$ monotonically increases with decreasing T , with anisotropy $\chi_{ab} > \chi_c$. The ^{75}As -NMR shift data versus T have the same T dependence as the derived $\chi(T)$ data, demonstrating that the derived $\chi(T)$ data are intrinsic. The observed ^{75}As nuclear spin dynamics rule out the presence of Néel-type antiferromagnetic electronic spin fluctuations, but are consistent with the presence of ferromagnetic and/or stripe-type antiferromagnetic spin fluctuations. The crystals of $\text{Ba}_{0.78}\text{K}_{0.22}\text{Co}_2\text{As}_2$ were grown in Sn flux and show properties very similar to those of undoped BaCo_2As_2 . On the other hand, the crystals from two batches of $\text{Ba}_{0.94}\text{K}_{0.06}\text{Co}_2\text{As}_2$ grown in CoAs self-flux show evidence of weak ferromagnetism at $T \lesssim 10$ K with small ordered moments at 1.8 K of ≈ 0.007 and $0.03\mu_B$ per formula unit, respectively.

DOI: [10.1103/PhysRevB.90.064517](https://doi.org/10.1103/PhysRevB.90.064517)

PACS number(s): 74.70.Xa, 75.20.En, 65.40.Ba, 76.60.–k

I. INTRODUCTION

The discovery of high-temperature superconductivity in the 122-type iron arsenides AFe_2As_2 ($A = \text{Ca}, \text{Sr}, \text{Ba}, \text{Eu}$) at temperatures up to $T_c = 38$ K upon chemical substitution or application of pressure motivated much research on this class of iron-based superconductors [1–13]. Extensive investigations have shown that the superconductivity can be achieved by partial chemical substitutions at the A, Fe, or As sites. In a search for unusual behaviors in other pure 122-type compounds that could potentially serve as parent compounds for high- T_c superconductivity or exhibit other novel ground states, we studied, e.g., BaMn_2As_2 , APd_2As_2 ($A = \text{Ca}, \text{Sr}, \text{and Ba}$) and $\text{ACu}_{2-x}\text{As}_2$ ($A = \text{Ca}, \text{Sr}$). Antiferromagnetic (AFM) order was discovered in insulating BaMn_2As_2 below a Néel temperature $T_N = 625$ K arising from long-range ordering of the Mn spin $S = 5/2$ local magnetic moments [14–17].

On substituting K for Ba to form $\text{Ba}_{1-x}\text{K}_x\text{Mn}_2\text{As}_2$, metallic hole-doped samples were produced for $x = 1.6\%$ to 40%, where the doped holes evidently interact only weakly with the Mn local moments [18,19]. At K-doping levels $x = 19\%$ and 26%, Bao *et al.* observed weak ferromagnetism (FM) [20], which was later confirmed and studied in detail at a higher doping level of 40% [21]. At the latter doping level, half-metallic FM below a Curie temperature of ≈ 100 K was inferred to arise from the itinerant doped-hole carriers that coexist with the long-range AFM ordering of the Mn moments below $T_N = 480$ K, where the ordered moments of the FM and AFM substructures are perpendicular to each other [21].

The compounds SrCu_2As_2 (Ref. [22]) and $\text{CaCu}_{1.7}\text{As}_2$ exhibit sp -band metallic behavior with a phase transition at 55 K in $\text{CaCu}_{1.7}\text{As}_2$ (Refs. [23,24]) that may arise from ordering of the Cu vacancies [24]. Bulk s -wave superconductivity below $T_c = 1.27$ and 0.92 K occurs in CaPd_2As_2 and SrPd_2As_2 , respectively [25], and below $T_c = 1.69$ and 3.04 K in the related compounds CaPd_2Ge_2 (Ref. [26]) and SrPd_2Ge_2 [27], respectively.

Much work has been done to study and understand the superconducting $\text{A}(\text{Fe}_{2-x}\text{Co}_x)_2\text{As}_2$ systems with $x \ll 1$ [9,10,28–32], but substantial research has now also been done on the $x = 1$ metallic but nonsuperconducting endpoint compounds ACo_2As_2 ($A = \text{Ca}, \text{Sr}, \text{Ba}$). These three

*Present address: Helmholtz-Zentrum Berlin für Materialien und Energie, Hahn-Meitner Platz 1, D-14109 Berlin, Germany.

†Present address: Oak Ridge National Laboratory, Oak Ridge, Tennessee 37831, USA.

‡johnston@ameslab.gov

compounds crystallize in the same body-centered tetragonal ThCr_2Si_2 -type structure (space group $I4/mmm$) as the AFe_2As_2 compounds do [33]. A-type AFM order was observed in CaCo_2As_2 below a sample-dependent Néel temperature 52 K–76 K [34–37], in which the Co magnetic moments in an ab plane align ferromagnetically along the c axis and have AFM alignment between adjacent Co planes. Both the A-type AFM structure and the positive Weiss temperature in the Curie-Weiss law indicate that the dominant interactions in CaCo_2As_2 are FM. The Co saturation moment from magnetization M versus applied magnetic field H isotherm measurements is only $\approx 0.3\mu_{\text{B}}/\text{Co}$, and a self-consistent modeling of the M data in terms of the local-moment picture was not obtained [37], so the magnetism in CaCo_2As_2 appears to be itinerant in nature. Furthermore, a significant vacancy concentration of $\approx 7\%$ on the Co sites was inferred from refinement of x-ray and neutron diffraction data as well as from wavelength-dispersive x-ray chemical analysis, so the composition of “ CaCo_2As_2 ” is actually $\text{CaCo}_{1.86}\text{As}_2$ [36,37]. The presence of this large concentration of disordered vacancies on the Co site causes the ^{75}As nuclear magnetic resonance (NMR) spectrum to be very weak and diffuse [38]. The large range of T_{N} values given above that has been observed in different studies of CaCo_2As_2 may be due to a variable sample-dependent concentration of these vacancies.

The SrCo_2As_2 and BaCo_2As_2 compounds are enigmatic. On the one hand, the large magnitudes of the magnetic susceptibility χ for both materials suggest within an itinerant magnetism picture that they should both be unstable to itinerant FM [39,40]. On the other hand, the T dependence of χ , the H dependence of M , and other measurements on SrCo_2As_2 and BaCo_2As_2 show no evidence for any tendency towards FM ordering or indeed for any type of static magnetic ordering above 2 K [39,40]. In fact, the $\chi(T)$ for SrCo_2As_2 exhibits a broad maximum at 115 K [40], which is characteristic of short-range dynamic AFM correlations in a low-dimensional spin system [41]. Furthermore, inelastic neutron scattering measurements on SrCo_2As_2 single crystals [42] revealed strong AFM fluctuations at the same “stripe” propagation vector as observed [9] in the paramagnetic and AFM states of the AFe_2As_2 parent compounds. This result suggests that SrCo_2As_2 is a plausible candidate for a high- T_{c} superconductor, so the lack of superconductivity in this material is a conundrum at present.

Investigations of the physical properties of BaCo_2As_2 self-flux-grown single crystals by Sefat *et al.* [39] revealed metallic behavior for both the ab -plane and c -axis electrical resistivity ρ versus temperature T from 2 K to 300 K. The low- T ab plane $\rho(T)$ followed a T^2 dependence, indicating a Fermi liquid ground state. The $\chi(T)$ showed a large magnitude with a weak broad maximum at ~ 150 K followed by a weak upturn at lower T . No magnetic ordering was observed above 1.8 K. The low- T heat capacity $C_{\text{p}}(T)$ yielded a large Sommerfeld electronic heat capacity coefficient γ , indicating a large density of states at the Fermi energy $\mathcal{D}^{\nu}(E_{\text{F}}) = 17.4$ states/eV f.u. for both spin directions, where f.u. stands for formula unit. Electronic structure calculations using the local density approximation (LDA) also gave a large $\mathcal{D}_{\text{band}}(E_{\text{F}}) = 8.5$ states/eV f.u. for both spin directions [39]. Within an itinerant magnetism model the authors suggest that BaCo_2As_2 should be ferromagnetic, but that

long-range FM order is suppressed by quantum fluctuations due to proximity to a magnetic quantum critical point [39].

Two angle-resolved photoemission spectroscopy (ARPES) studies reveal that the Fermi surface of BaCo_2As_2 exhibits no obvious significant Fermi surface nesting [43,44], in agreement with band-structure calculations [39,43,44]. This situation contrasts with the clear nesting between the hole and electron pockets in the paramagnetic (PM) states of the AFe_2As_2 compounds that results in long-range itinerant spin-density wave AFM at temperatures $T_{\text{N}} \lesssim 200$ K at the “stripe” wave vector spanning the electron and hole pockets [9]. On the other hand, as noted above it was discovered that AFM fluctuations occur in the closely related compound SrCo_2As_2 at the *same* stripe wave vector [42] even though there is no obvious Fermi surface nesting at that wave vector as deduced from ARPES data and band-structure calculations for this compound [40].

In view of the above results on CaCo_2As_2 and especially on SrCo_2As_2 , it is of great interest to confirm and extend the previous [39] physical property measurements on BaCo_2As_2 . Furthermore, if BaCo_2As_2 is indeed situated close to a magnetic quantum critical point as suggested [39], one might expect the physical properties to change appreciably with doping. We therefore further investigated the physical properties of undoped BaCo_2As_2 and studied the effects of partial K substitution for Ba in $\text{Ba}_{1-x}\text{K}_x\text{Co}_2\text{As}_2$ ($x = 0.06, 0.22$). A polycrystalline sample of BaCo_2As_2 was synthesized for neutron diffraction measurements. Single crystals were synthesized using both Sn flux and CoAs self-flux and their properties were investigated by powder x-ray and neutron diffraction, $\rho(T)$, $\chi(T)$, $M(H, T)$, $C_{\text{p}}(T)$, and ^{75}As NMR measurements. We also report electronic structure calculations of BaCo_2As_2 and compare the results with the experimental data.

Our measurements indicate that BaCo_2As_2 shows no phase transitions of any kind from 10 K to 300 K. However, the c -axis lattice parameter shows an anomalous negative thermal expansion over this temperature range, similar to but a factor of two smaller than our previous findings [40] for SrCo_2As_2 . We confirmed the T^2 dependence of the ab plane $\rho(T)$ at low T and the Sommerfeld coefficient in the low- T $C_{\text{p}}(T)$ found by Sefat *et al.* [39]. The $M(H)$ isotherms for our crystals grown out of Sn flux show that $M \propto H$ for temperatures from 2 K to 300 K, with no evidence for saturation that might arise from FM correlations. The anisotropy of χ is $\approx 20\%$ for these crystals. The temperature dependences of χ for crystals grown with both Sn flux and CoAs self-flux are similar to those reported in Fig. 1 of Ref. [39]. However, our $\chi(T)$ data, which are reproducible for different crystals, disagree with those in Fig. 1 of Ref. [39] in both the magnitude of χ and the nature of the anisotropy. Our calculations of the density of states versus energy $\mathcal{D}(E)$ show a sharp peak at the Fermi energy E_{F} that arises from the previously identified $d_{x^2-y^2}$ flat band at or near E_{F} in SrCo_2As_2 [44]. A large $\mathcal{D}(E_{\text{F}})$ is also evident from the experimental Sommerfeld electronic heat capacity coefficient.

For a K-doped crystal grown in Sn flux with composition $\text{Ba}_{0.78}\text{K}_{0.22}\text{Co}_2\text{As}_2$, we did not observe any significant difference in the properties from those of pure BaCo_2As_2 . This result does not support the hypothesis [39] that BaCo_2As_2 is close to a magnetic quantum critical point. On the other hand, the magnetic measurements on two batches of CoAs

self-flux-grown crystals of $\text{Ba}_{1-x}\text{K}_x\text{Co}_2\text{As}_2$ with $x \approx 0.06$ reveal the presence of weak FM below 20 K with small ordered (saturation) moments of $\approx 0.007\mu_B$ and $0.03\mu_B$ per formula unit, respectively.

The experimental and theoretical details and crystallographic studies are presented in Secs. II and III, respectively. The physical properties are described in Secs. IV–VII for $x = 0, 0.054, 0.065, \text{ and } 0.22$, respectively. The electronic density of states calculation for BaCo_2As_2 ($x = 0$) is presented in Sec. IV C. A summary is given in Sec. VIII.

II. METHODS

A polycrystalline sample of BaCo_2As_2 was synthesized by solid-state reaction for the powder neutron diffraction measurements. The purity and sources of the elements used were Ba (99.99%) from Sigma Aldrich and Co (99.998%) and As (99.999 99%) from Alfa Aesar. Small pieces of Ba were mixed with Co and As powder in the stoichiometric ratio and pressed into a pellet that was then sealed in a tantalum tube under high-purity Ar gas. The sealed tantalum tube was sealed inside an evacuated quartz tube and heated from room temperature to 590°C in 15 h, held for 15 h, heated to 620°C in 6 h, held for 15 h, heated to 750°C in 5 h, held for 10 h, heated to 780°C in 1 h, held for 3 h, heated to 800°C in 1/2 h, held there for 2 h, and then furnace cooled. After this prereaction the sample was ground thoroughly and again pelleted and sealed in a tantalum tube under Ar that was then sealed in an evacuated quartz tube. The sample was heated to 1000°C in 16 h, held there for 2 h, and then heated to 1100°C in 2 h, held for 48 h, and then quenched into liquid nitrogen.

Single crystals of $\text{Ba}_{1-x}\text{K}_x\text{Co}_2\text{As}_2$ ($x = 0, 0.065, 0.22$) were grown from solution using both Sn and prereacted CoAs as the flux. The sources and purities of the elements used in addition to those given above were K (99.95%) and Sn (99.999%) from Alfa Aesar. The chemical compositions of the crystals were determined using wavelength-dispersive x-ray spectroscopy (WDS) analysis with a JEOL electron probe microanalyzer and energy dispersive x-ray spectroscopy (EDX) analysis using a JEOL scanning electron microscope. WDS analysis of self-flux-grown BaCo_2As_2 crystals revealed the desired 1:2:2 stoichiometry.

The Sn-flux growth of BaCo_2As_2 was carried out with stoichiometric BaCo_2As_2 and Sn in a 1:15 molar ratio placed in an alumina crucible that was sealed inside a partially argon-filled ($\approx 1/3$ atm pressure) silica tube that was then prereacted at 650°C for 12 h, heated to 900°C at a rate of $100^\circ\text{C}/\text{h}$, held for 5 h, then heated to 1100°C at a rate of $50^\circ\text{C}/\text{h}$, where it was held for 30 h, then slowly cooled to 600°C at a rate of $4^\circ\text{C}/\text{h}$. Thin shiny platelike BaCo_2As_2 crystals of typical size $3 \times 2 \times 0.2 \text{ mm}^3$ were obtained by centrifuging the flux at the latter temperature. WDS analysis of a Sn-flux-grown BaCo_2As_2 crystal revealed the presence of 1.6 mol% Sn in the crystal.

For the self-flux growth of BaCo_2As_2 crystals, Ba and prereacted CoAs were taken in 1:4 molar ratio and placed in an alumina crucible that was sealed in a partially argon-filled ($\approx 1/4$ atm pressure) quartz tube. Crystal growth was carried out by heating to 1200°C at $60^\circ\text{C}/\text{h}$, holding for 5 h, heating to 1300°C at $50^\circ\text{C}/\text{h}$, holding for 1 h, and then slowly cooling

to 1180°C at $2^\circ\text{C}/\text{h}$. Shiny platelike BaCo_2As_2 crystals of typical size $4 \times 2.5 \times 1 \text{ mm}^3$ were obtained upon decanting the flux with a centrifuge at 1180°C .

In order to grow $\text{Ba}_{1-x}\text{K}_x\text{Co}_2\text{As}_2$ crystals using Sn flux we started with stoichiometric $\text{Ba}_{0.5}\text{K}_{0.5}\text{Co}_2\text{As}_2$ and Sn in a 1:15 molar ratio placed in an alumina crucible that was sealed under high-purity Ar in a tantalum tube that was then sealed in a quartz tube with $\approx 1/3$ atm Ar pressure. After prereaction at 650°C for 12 h it was heated to 1150°C following the same steps as above, where it was held for 35 h and then slowly cooled to 600°C at a rate of $3^\circ\text{C}/\text{h}$. Decanting the flux at this temperature using a centrifuge yielded platelike crystals of typical size $2.5 \times 1.5 \times 0.1 \text{ mm}^3$, which turned out to be $\text{Ba}_{0.78(1)}\text{K}_{0.22(1)}\text{Co}_2\text{As}_2$ from EDX analyses. This potassium concentration is much smaller than in the nominal starting ratio. The presence of ≈ 2.0 mol% Sn was inferred from EDX analysis on a $\text{Ba}_{0.78}\text{K}_{0.22}\text{Co}_2\text{As}_2$ crystal. Sn is also incorporated into the structure of FeAs-based $\text{Ba}_{1-x}\text{K}_x\text{Fe}_2\text{As}_2$ ($x = 0, 0.45$) crystals grown in Sn flux [45].

Self-flux growths of $\text{Ba}_{1-x}\text{K}_x\text{Co}_2\text{As}_2$ crystals with $x > 0$ were carried out using nominal starting compositions $x = 0.25, 0.5, 0.75$ and prereacted CoAs, again in a 1:4 molar ratio, placed in an alumina crucible, sealed in a tantalum tube under 1 atm Ar gas that was then sealed into a partially Ar-filled ($\approx 1/4$ atm pressure) quartz tube. Crystals were obtained by heating to 1260°C following the above steps of self-flux growth, holding for 5 h and slow cooling to 1160°C at $2^\circ\text{C}/\text{h}$, at which point the excess flux was removed by centrifuging. Shiny platelike crystals of typical size $3 \times 2 \times 0.5 \text{ mm}^3$ were obtained. In contrast to Sn-grown crystals where 22% K doping was achieved, the K concentrations obtained from the EDX analyses of the self-flux-grown K-doped BaCo_2As_2 crystals were much smaller. We obtained $\text{Ba}_{0.935(5)}\text{K}_{0.065(5)}\text{Co}_2\text{As}_2$, $\text{Ba}_{0.955(3)}\text{K}_{0.045(3)}\text{Co}_2\text{As}_2$, and $\text{Ba}_{0.946(5)}\text{K}_{0.054(5)}\text{Co}_2\text{As}_2$ for the starting growth concentrations $x = 0.25, 0.5, 0.75$, respectively. Since the difference in K contents of these three compositions are very small, we measured the physical properties of crystals from the two batches with compositions $\text{Ba}_{0.935}\text{K}_{0.065}\text{Co}_2\text{As}_2$ and $\text{Ba}_{0.946}\text{K}_{0.054}\text{Co}_2\text{As}_2$.

The crystal structures of crushed crystals of BaCo_2As_2 , $\text{Ba}_{0.94}\text{K}_{0.06}\text{Co}_2\text{As}_2$, and $\text{Ba}_{0.78}\text{K}_{0.22}\text{Co}_2\text{As}_2$ were determined using laboratory-based powder x-ray diffraction (XRD) with $\text{Cu } K_\alpha$ radiation on a Rigaku Geigerflex x-ray diffractometer. The magnetic measurements on the crystals were carried out using a Quantum Design, Inc., superconducting quantum interference device magnetic properties measurement system (MPMS). The contribution from the sample holder was subtracted from the measured magnetic moment except for the measurements in low magnetic fields ($H < 0.1$ T). Some $M(H)$ plots presented in this paper have M units of $\text{G cm}^3/\text{mol}$ and others have M units of $\mu_B/\text{f.u.}$ The conversion is $1 \mu_B/\text{f.u.} = 5585 \text{ G cm}^3/\text{mol}$.

The heat capacity and electrical resistivity measurements were carried out using a Quantum Design, Inc., physical properties measurement system (PPMS). The heat capacity was measured by the relaxation method and the resistivity was measured by the standard four-probe ac technique using the heat capacity and ac transport options of the PPMS, respectively. A ^3He attachment to the PPMS was used to measure the heat capacity down to 0.45 K.

High-resolution x-ray powder diffraction patterns were collected at ambient temperature for crushed single crystals of BaCo_2As_2 at the beamline X16C at the National Synchrotron Light Source. The samples were loaded into a 1-mm-diameter glass capillary along with Si powder which served as a standard for the refinement of the lattice parameters. The x-ray wavelength, 0.6995 \AA , was chosen using a Si(111) double monochromator. The powder diffraction pattern was collected in the $5\text{--}45^\circ$ 2θ range with a constant step size of 0.005° and a linearly varying counting time of 1–3 s/point. The incident beam intensity was monitored with an ion chamber and the diffracted radiation was measured with a NaI(Tl) scintillation detector. The axial and in-plane resolution of the diffractometer were set by slits and a Ge(111) analyzer crystal, respectively.

Neutron powder diffraction data for a polycrystalline sample of BaCo_2As_2 were collected at the Spallation Neutron Source (POWGEN) at Oak Ridge National Laboratory [46]. The sample was loaded into a vanadium sample can in the presence of He exchange gas to maintain thermal contact to the sample at low temperatures. Data were collected as the sample temperature was increased from 10 K to 300 K in 15 K intervals using a central wavelength of 1.066 \AA .

Nuclear magnetic resonance measurements were carried out for a CoAs self-flux-grown BaCo_2As_2 single crystal on the ^{75}As nucleus ($I = 3/2$, $\gamma/2\pi = 7.2919 \text{ MHz/T}$) using a homemade phase-coherent spin-echo pulse spectrometer. ^{75}As -NMR spectra were obtained by sweeping the magnetic field H at a fixed frequency $f = 54.4 \text{ MHz}$. The magnetic field was applied parallel to either the crystal c axis or the ab plane. The origin of the Knight shift $K = 0$ of the ^{75}As nucleus was determined by ^{75}As -NMR measurements of GaAs. The ^{75}As nuclear spin-lattice relaxation rate ($1/T_1$) was measured with a saturation recovery method.

Electronic structure calculations were carried out for BaCo_2As_2 using the full-potential linearized augmented plane-wave (FP-LAPW) method [47] with the LDA [48], where we employed $R_{\text{MT}}k_{\text{max}} = 9.0$ with muffin tin (MT) radii $R_{\text{MT}} = 2.1 \text{ a.u.}$ for all atoms in order to obtain self-consistent charge density. 828 \mathbf{k} points were selected in the irreducible Brillouin zone and calculations were iterated to reach the total energy convergence criterion which was 0.01 mRy/cell . The body-centered tetragonal lattice parameters $a = b = 3.958 \text{ \AA}$, $c = 12.670 \text{ \AA}$ from Ref. [33] were used. The As atom was relaxed to get the theoretical c -axis position $z_{\text{As}} = 0.3441$. This value is close to the experimental value $z_{\text{As}} = 0.3509$ obtained in the following section.

III. CRYSTALLOGRAPHY

A. Powder x-ray diffraction measurements on a laboratory-based diffractometer

The crystal structures of the compounds were determined from the analysis of room-temperature powder XRD data collected on the crushed $\text{Ba}_{1-x}\text{K}_x\text{Co}_2\text{As}_2$ ($x = 0, 0.054, 0.065, 0.22$) single crystals. The XRD data were analyzed by Rietveld structural refinement using the software FULLPROF [49] confirming the single-phase nature and ThCr_2Si_2 -type body-centered tetragonal structure ($I4/mmm$) of the compounds. The XRD patterns and Rietveld refinement

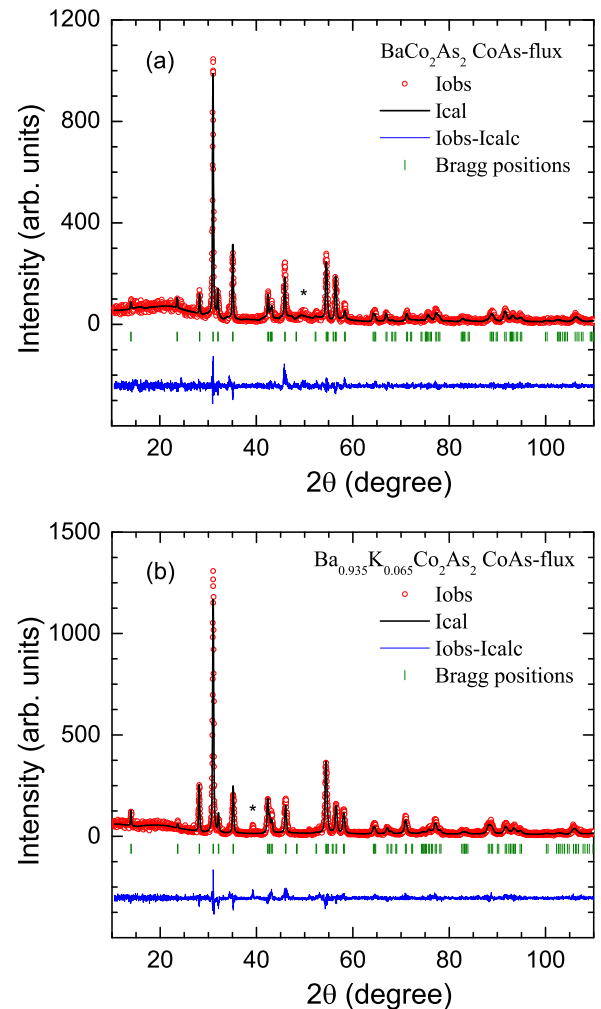


FIG. 1. (Color online) Powder XRD patterns of (a) BaCo_2As_2 and (b) $\text{Ba}_{0.935}\text{K}_{0.065}\text{Co}_2\text{As}_2$ recorded at room temperature using a laboratory-based diffractometer. The solid lines through the experimental points are the Rietveld refinement profiles calculated for the ThCr_2Si_2 -type body-centered tetragonal structure (space group $I4/mmm$). The short vertical bars mark the fitted Bragg peak positions. The lowermost curves represent the difference between the experimental and calculated intensities. The unindexed peaks marked with stars correspond to peaks from residual CoAs flux on the surface of the samples.

profiles for two representative compositions $x = 0$ and 0.065 are shown in Fig. 1. A few weak unindexed peaks are also observed in the XRD patterns due to the residual CoAs flux on the surface of the crystals. During the refinement the thermal parameters B were kept fixed to $B \equiv 0$ and the fractional occupancies of atoms were fixed to the stoichiometric values of unity as within the error bars the lattice parameters and the As c -axis position parameter z_{As} were insensitive to small changes in B and occupancies of atoms. For K-doped compounds the fractional occupancies of Ba and K were kept fixed to the values as determined from the EDX composition analysis. The crystallographic and refinement parameters obtained are listed in Table I. Our crystal data for BaCo_2As_2 ($x = 0$) are in good agreement with the literature values [33,39] that are also listed in Table I.

TABLE I. Crystal and Rietveld refinement parameters obtained from powder x-ray diffraction data for crushed $\text{Ba}_{1-x}\text{K}_x\text{Co}_2\text{As}_2$ ($x = 0, 0.054, 0.065, 0.22$) crystals and powder neutron diffraction data for a polycrystalline sample of BaCo_2As_2 ($x = 0$) crystallizing in the ThCr_2Si_2 -type body-centered tetragonal structure (space group $I4/mmm$). The atomic coordinates of Ba/K, Co, and As atoms are $(0, 0, 0)$, $(0, 1/2, 1/4)$, and $(0, 0, z_{\text{As}})$, respectively. Also included are literature data for a polycrystalline sample with $x = 0$ from Ref. [33] and for crushed single crystals with $x = 0$ from Ref. [39].

Composition	$x = 0$ Ref. [33]	$x = 0$ Ref. [39]	$x = 0^{\text{a}}$ CoAs flux	$x = 0^{\text{b}}$	$x = 0^{\text{a}}$ Sn flux	$x = 0.054^{\text{a}}$ CoAs flux	$x = 0.065^{\text{a}}$ CoAs flux	$x = 0.22^{\text{a}}$ Sn flux
Crystal data								
a (Å)	3.958(5)	3.9537(1)	3.9522(4)	3.957 43(2)	3.9416(3)	3.9427(6)	3.9422(4)	3.9191(6)
c (Å)	12.67(2)	12.6524(6)	12.646(2)	12.669 56(9)	12.7073(8)	12.694(2)	12.696(2)	12.818(2)
c/a	3.201(1)	3.200 14(7)	3.200(1)	3.201 46(4)	3.224(1)	3.220(1)	3.220(2)	3.271(2)
V_{cell} (Å ³)	198.5(8)	197.78(2)	197.53(4)	198.421(2)	197.43(2)	197.32(5)	197.31(3)	196.88(6)
z_{As}	0.361		0.3509(3)	0.350 87(3)	0.3510(2)	0.3516(4)	0.3517(2)	0.3514(4)
Refinement								
quality								
χ^2			1.36	1.713	1.43	2.38	1.48	2.17
R_{p} (%)			14.3	10.10	14.6	17.2	14.8	17.5
R_{wp} (%)			19.0	5.01	19.9	23.2	19.8	23.1

^aFrom laboratory-based powder x-ray diffraction data.

^bFrom synchrotron-based x-ray and neutron powder diffraction data that were refined together.

B. Powder diffraction measurements on synchrotron-based x-ray and neutron diffractometers

Rietveld corefinements of the ambient-temperature x-ray and neutron powder diffraction patterns for BaCo_2As_2 were accomplished using the General Structure Analysis System package (GSAS) [50] and the graphical user interface (EXPGUI) [51]. The neutron and x-ray powder data and refinements are shown in Fig. 2 and the results of the corefinement are given in Table I. For the x-ray data, the strongest Si lines at low Q were excluded from the refinements. The principal impurity phase found in the sample of crushed single crystals grown in CoAs self-flux is adventitious CoAs which comprises approximately 8% of the sample by weight. A few very-low-intensity diffraction peaks in the x-ray powder pattern could not be identified conclusively with other impurity phases.

The refinements confirmed that BaCo_2As_2 crystallizes in the ThCr_2Si_2 -type body-centered tetragonal structure (space group $I4/mmm$). The c/a ratio of ≈ 3.2 and the interlayer As-As distance $d_{\text{As-As}} = (1 - 2z_{\text{As}})c \approx 3.78$ Å confirm that BaCo_2As_2 has an uncollapsed tetragonal structure at ambient pressure, in contrast to $\text{CaCo}_{1.86}\text{As}_2$ [36,37], which has a collapsed tetragonal structure arising from dimerization of the As atoms along the c axis [22]. The x-ray and neutron refinements (together and separately) show that, within experimental error, all atomic sites are fully occupied, again in contrast to the Co deficiency observed for the Ca-based compound [36,37].

The temperature dependences of the a and c lattice parameters, as well as of the unit cell volume and z_{As} determined from refinements of the neutron powder diffraction data, are plotted in Fig. 3. Similar to what was found for the closely related compound SrCo_2As_2 , the c -axis thermal expansion coefficient α_c is negative from 10 K to 300 K, whereas α_a (the a -axis thermal expansion coefficient) is positive over this same temperature range. However, the change in the c lattice parameter over this temperature range is roughly half of that noted previously for SrCo_2As_2 , whereas the changes

in the a lattice parameters are comparable. Nevertheless, as noted previously [40] negative thermal expansion coefficients in the paramagnetic state are unusual. The unit cell volume of BaCo_2As_2 smoothly decreases as temperature is lowered from 300 K to 10 K. The c/a ratio correspondingly increases by about 0.7% on cooling, from 3.202 at 300 K to 3.225 at 10 K. Finally, the arsenic positional parameter z_{As} increases with decreasing temperature over the measured range by roughly the same percentage as the c lattice constant increases.

IV. PHYSICAL PROPERTIES OF BaCo_2As_2

A. Electrical resistivity

The T dependences of the in-plane electrical resistivities ρ of BaCo_2As_2 crystals grown using CoAs and Sn flux are shown in Fig. 4. The $\rho(T)$ data for both crystals exhibit metallic character. The residual resistivity ratio is $\text{RRR} \equiv \rho(300 \text{ K})/\rho(1.8 \text{ K}) = 21.8$ for the CoAs-grown crystal and 6.1 for the Sn-grown crystal. The large RRR of the CoAs-grown crystal indicates good crystal quality. The reason for the lower RRR of the Sn-grown crystal is likely due to the incorporation of Sn into the bulk of the crystal with a concentration of 1.6 mol% as described in Sec. II. No signature of any phase transition is evident in the $\rho(T)$ data for either crystal except for a rapid drop in ρ of the Sn-grown crystal at $T \approx 3.5$ K due to the superconductivity of a small amount of adventitious Sn metal from the Sn flux.

As shown in the insets of Figs. 4(a) and 4(b) the $\rho(T)$ data exhibit a T^2 temperature dependence below 40–50 K. We analyzed the low- T data below 40–50 K by

$$\rho(T) = \rho_0 + AT^2, \quad (1)$$

where A is a constant. The straight lines in the insets of Figs. 4(a) and 4(b) are the fits of $\rho(T)$ by Eq. (1) for $1.8 \text{ K} \leq T \leq 50 \text{ K}$ in (a) and for $4.0 \text{ K} \leq T \leq 40 \text{ K}$ in (b). The T^2 dependencies of ρ indicate a Fermi-liquid ground state in both crystals of BaCo_2As_2 . The Kadowaki-Woods

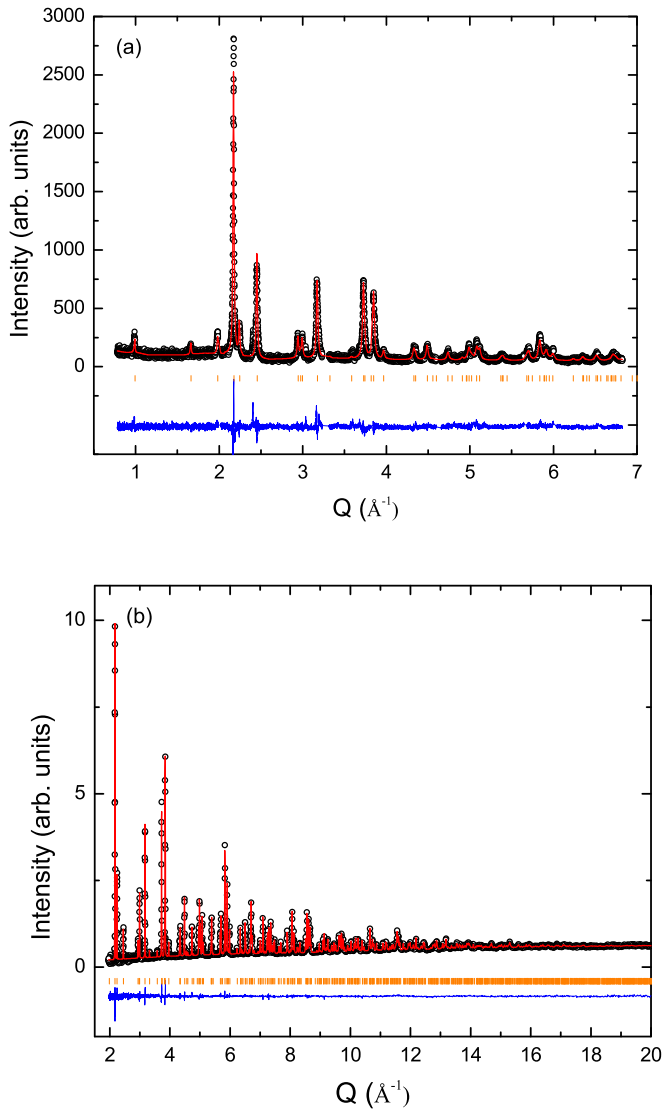


FIG. 2. (Color online) (a) Synchrotron x-ray powder diffraction and (b) neutron powder diffraction patterns for crushed single crystals and a polycrystalline sample, respectively, of BaCo₂As₂ at 300 K. Gaps in the x-ray data correspond to regions where strong lines associated with the Si internal standard are found. Shown are the measured data (open circles), the calculated profile fits from Rietveld corefinement of the x-ray and neutron data (red lines), and the residuals (blue curve below fit). The vertical orange tick marks correspond to the positions of the Bragg peaks of BaCo₂As₂.

ratio $R_{\text{KW}} = A/\gamma^2$ for each crystal is given in Table II. The value of R_{KW} for the CoAs-flux-grown crystal is smaller than but comparable to the universal value of $R_{\text{KW}} = A/\gamma^2 = 1.0 \times 10^{-5} \mu\Omega \text{ cm}/(\text{mJ}/\text{mol K})^2$ for strongly correlated and heavy fermion systems [52].

The $\rho(T)$ data were further fitted over the full temperature range of the measurements by

$$\rho(T) = \rho_1 + \rho(T = \Theta_{\text{R}})\rho_{\text{BG}}(T/\Theta_{\text{R}}), \quad (2a)$$

within the framework of the Bloch-Grüneisen (BG) model that describes the electrical resistivity $\rho_{\text{BG}}(T)$ due to scattering of conduction electrons by longitudinal acoustic lattice vibrations

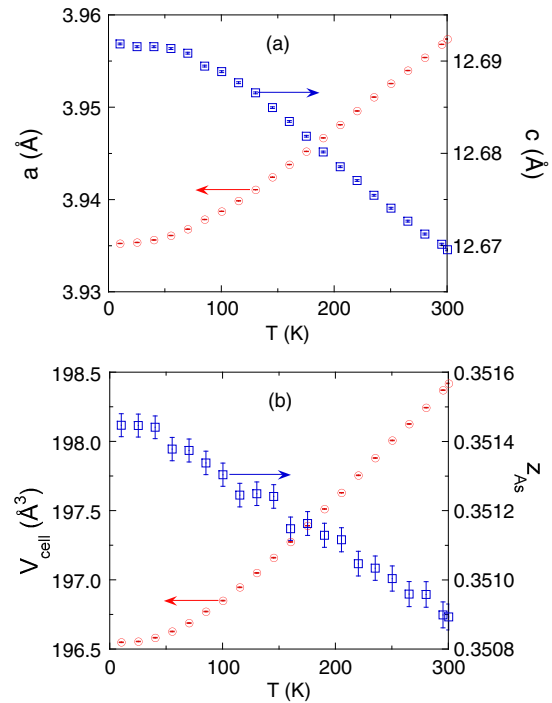


FIG. 3. (Color online) The temperature dependence of (a) the lattice parameters a and c and (b) the unit cell volume, V_{cell} , and the As c -axis positional parameter, z_{As} , for BaCo₂As₂ determined from refinements of the neutron powder diffraction data. Error bars are plotted for each quantity.

and is given by [53]

$$\rho_{\text{BG}}(T/\Theta_{\text{R}}) = 4\mathcal{R} \left(\frac{T}{\Theta_{\text{R}}} \right)^5 \int_0^{\Theta_{\text{R}}/T} \frac{x^5}{(e^x - 1)(1 - e^{-x})} dx, \quad (2b)$$

where Θ_{R} is the Debye temperature obtained from the resistivity data, \mathcal{R} is a material-dependent prefactor, and

$$\rho(T/\Theta_{\text{R}} = 1) \approx 0.946464\mathcal{R} \quad (2c)$$

is obtained from Eq. (2b). Least-squares fits of the $\rho(T)$ data by Eqs. (2) for $1.8 \text{ K} \leq T \leq 300 \text{ K}$ and for $4.0 \text{ K} \leq T \leq 300 \text{ K}$ for CoAs- and Sn-flux-grown BaCo₂As₂ crystals are shown by the red curves in Figs. 4(a) and 4(b), respectively, where we used the analytic Padé approximant function [54] of T/Θ_{R} in place of the integral on the right side of Eq. (2b). The fitting parameters are listed in Table II together with those of the other compounds studied in this paper. The value of \mathcal{R} was obtained from the fitted value of $\rho(\Theta_{\text{R}})$ using Eq. (2c).

B. Heat capacity

The T dependencies of the heat capacity C_p of CoAs flux-grown and Sn flux-grown BaCo₂As₂ crystals are shown in Figs. 5(a) and 5(b), respectively. No anomalies are observed between 0.45 K and 300 K that would reflect the occurrence of phase transitions. The same value $C_p(T = 300 \text{ K}) \approx 130 \text{ J}/\text{mol K}$ is observed for both the CoAs-flux and Sn-flux-grown crystals, which is close to the expected classical Dulong-Petit value of $C_V = 3nR = 15R = 124.7 \text{ J}/\text{mol K}$ at

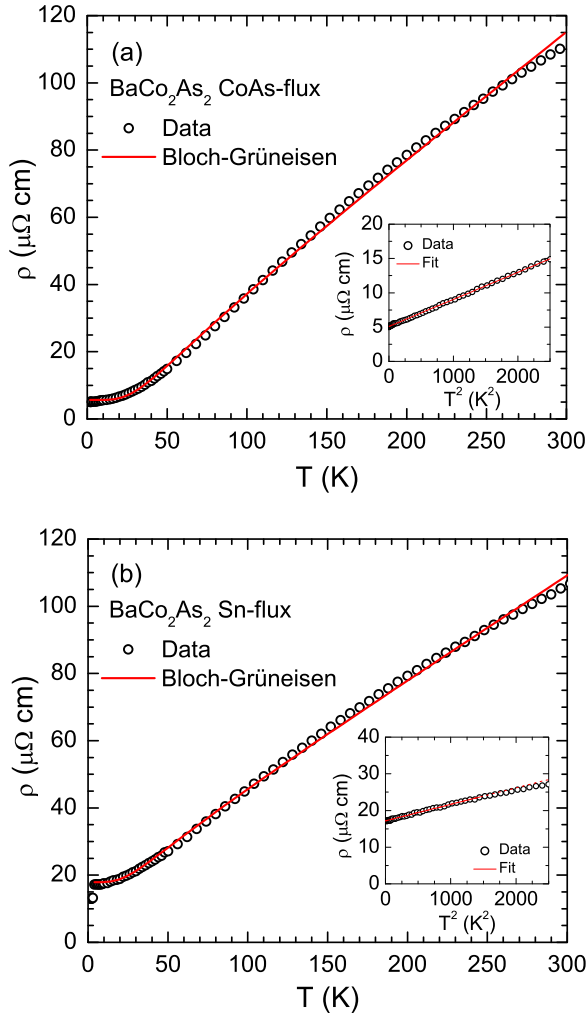


FIG. 4. (Color online) In-plane electrical resistivity ρ of BaCo_2As_2 single crystals grown using (a) CoAs flux and (b) Sn flux as a function of temperature T measured in zero magnetic field. The red solid curves are fits by the Bloch-Grüneisen model. Insets: ρ vs T^2 plots below 50 K. The straight lines are the fits by $\rho = \rho_0 + AT^2$ for $1.8 \text{ K} \leq T \leq 50 \text{ K}$ in (a) and for $4.0 \text{ K} \leq T \leq 40 \text{ K}$ in (b).

constant volume, where $n = 5$ is the number of atoms per f.u. and R is the molar gas constant [55,56].

The low- T $C_p(T)$ data below 10 K plotted as C_p/T versus T^2 are shown in the insets of Figs. 5(a) and 5(b) for CoAs-flux- and Sn-flux-grown crystals, respectively. These data were

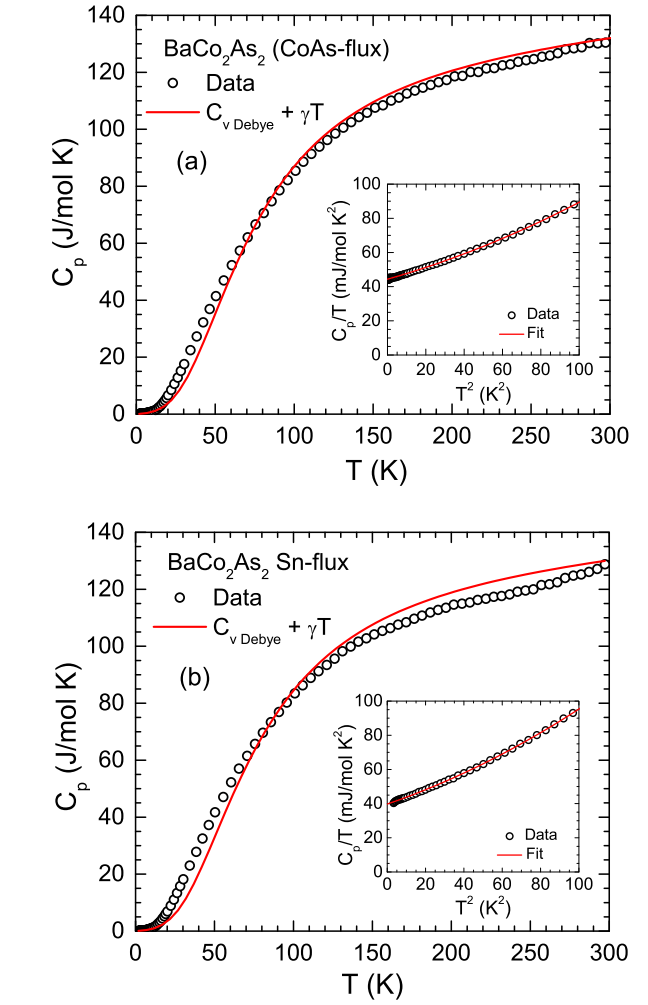


FIG. 5. (Color online) Heat capacity C_p of BaCo_2As_2 single crystals grown using (a) CoAs flux and (b) Sn flux as a function of temperature T measured in zero magnetic field for $1.8 \text{ K} \leq T \leq 300 \text{ K}$. The solid curve is the sum of the contributions from the Debye lattice heat capacity $C_{V\text{Debye}}(T)$ and electronic heat capacity γT according to Eq. (6a). Insets: C_p/T versus T^2 plots below 10 K. The red curves represent the fits of the C_p/T data by Eq. (3) for $0.45 \text{ K} \leq T \leq 10 \text{ K}$ in (a) and $1.8 \text{ K} \leq T \leq 10 \text{ K}$ in (b).

fitted by

$$\frac{C_p(T)}{T} = \gamma + \beta T^2 + \delta T^4, \quad (3)$$

TABLE II. Parameters obtained from fits of the resistivities ρ within the ab plane of the listed single crystals by Eqs. (1) and (2). The quantity \mathcal{R} is obtained from the fitted value of $\rho(\Theta_R)$ using Eq. (2c). The units of the Kadowaki-Woods ratio $R_{\text{KW}} = A/\gamma^2$ are $10^{-5} \mu\Omega \text{ cm}/(\text{mJ/mol K})^2$.

Compound	ρ_0 ($\mu\Omega \text{ cm}$)	A ($10^{-3} \frac{\mu\Omega \text{ cm}}{\text{K}^2}$)	R_{KW}	ρ_1 ($\mu\Omega \text{ cm}$)	Θ_R (K)	$\rho(\Theta_R)$ ($\mu\Omega \text{ cm}$)	\mathcal{R} ($\mu\Omega \text{ cm}$)
BaCo_2As_2 (CoAs flux)	5.10(2)	3.94(1)	0.20	5.6(2)	174(5)	61(2)	64.4
BaCo_2As_2 (Sn flux)	17.18(3)	4.50(5)	0.28	17.9(3)	140(6)	41(2)	43.3
$\text{Ba}_{0.946}\text{K}_{0.054}\text{Co}_2\text{As}_2$ (CoAs flux)	9.94(1)	3.31(1)	0.17	10.4(1)	181(4)	54.6(1)	57.7
$\text{Ba}_{0.935}\text{K}_{0.065}\text{Co}_2\text{As}_2$ (CoAs flux)	11.81(2)	3.72(1)	0.17	12.2(2)	164(3)	49.3(9)	52.1
$\text{Ba}_{0.78}\text{K}_{0.22}\text{Co}_2\text{As}_2$ (Sn flux)	22.07(3)	6.06(5)	0.69	22.8(2)	122(5)	38(2)	40.1

TABLE III. The linear specific heat coefficients γ and the coefficients β and δ to the T^3 and T^5 terms in the low- T heat capacity, respectively, and the density of states at the Fermi energy $\mathcal{D}(E_F)$ for both spin directions for $\text{Ba}_{1-x}\text{K}_x\text{Co}_2\text{As}_2$ ($x = 0, 0.05, 0.065, 0.22$) single crystals. The Debye temperatures Θ_D obtained for $0.45 \text{ K} \leq T \leq 10 \text{ K}$ (low- T) and $0.45 \text{ K} \leq T \leq 300 \text{ K}$ (all T) fits to heat capacity data and the Debye temperatures Θ_R obtained from fitting electrical resistivity data by the Bloch-Grüneisen model for $1.8 \text{ K} \leq T \leq 300 \text{ K}$ are also listed.

Compound (crystal-growth flux)	γ (mJ/mol K ²)	β (mJ/mol K ⁴)	δ ($\mu\text{J/mol K}^6$)	$\mathcal{D}^\nu(E_F)$ (states/eV f.u.)	Θ_D (K) (low T)	Θ_D (K) (all T)	Θ_R (K)
BaCo_2As_2 (CoAs flux)	44.39(3)	0.319(3)	1.32(4)	18.82(2)	312(2)	301(3)	174(5)
BaCo_2As_2 (Sn flux)	39.75(7)	0.386(4)	1.70(5)	16.85(3)	293(2)	311(4)	140(6)
$\text{Ba}_{0.946}\text{K}_{0.054}\text{Co}_2\text{As}_2$ (CoAs flux)	43.5(3)	0.32(2)	1.36(2)	18.4(2)	312(7)	310(4)	181(4)
$\text{Ba}_{0.935}\text{K}_{0.065}\text{Co}_2\text{As}_2$ (CoAs flux)	47.2(2)	0.271(7)	1.62(7)	20.0(1)	330(3)	304(3)	164(3)
$\text{Ba}_{0.78}\text{K}_{0.22}\text{Co}_2\text{As}_2$ (Sn flux)	29.7(3)	0.378(9)	1.44(8)	12.6(2)	295(3)		122(5)

where γT is the Sommerfeld electronic heat capacity contribution and $\beta T^3 + \delta T^5$ is the low- T lattice contribution. The γ , β , and δ parameters obtained from fits of the data in the inset of Fig. 5(a) for $0.45 \text{ K} \leq T \leq 10 \text{ K}$ and the inset of Fig. 5(b) for $1.8 \text{ K} \leq T \leq 10 \text{ K}$ by Eq. (3) are listed in Table III. The coefficients γ , β , and δ depend somewhat on the flux used for growth of the crystals and on the details of the associated heat treatment. The fits are shown as the solid red curves in the respective insets.

The densities of states at the Fermi energy $\mathcal{D}(E_F)$ estimated from the γ values using the relation [55]

$$\gamma = \frac{\pi^2 k_B^2}{3} \mathcal{D}^\nu(E_F) \quad (4)$$

are $\mathcal{D}^\nu(E_F) = 18.82(2)$ and $16.85(3)$ states/eV f.u. for both spin directions for the CoAs-flux-grown and Sn-flux-grown crystals, respectively. These γ and $\mathcal{D}^\nu(E_F)$ values for BaCo_2As_2 are very large compared to those of the 122-type iron arsenide compounds [57–59]. The value $\gamma = 41.0 \text{ mJ/mol K}^2$ obtained for BaCo_2As_2 by Sefat *et al.* [39] is intermediate between our two values in Table III. Large γ values have also been observed in $\text{CaCo}_{1.86}\text{As}_2$ (Ref. [37]) and SrCo_2As_2 [40].

The Debye temperature Θ_D was estimated from β using the relation [55]

$$\Theta_D = \left(\frac{12\pi^4 n R}{5\beta} \right)^{1/3}, \quad (5)$$

yielding $\Theta_D = 312(2) \text{ K}$ and $293(3) \text{ K}$ for CoAs-flux-grown and Sn-flux-grown crystals, respectively.

An analysis of the $C_p(T)$ data over the whole measured T range gives another estimate of Θ_D . In this case, the $C_p(T)$ data were fitted by

$$C_p(T) = \gamma T + n C_{V \text{ Debye}}(T), \quad (6a)$$

where [56]

$$C_{V \text{ Debye}}(T) = 9R \left(\frac{T}{\Theta_D} \right)^3 \int_0^{\Theta_D/T} \frac{x^4 e^x}{(e^x - 1)^2} dx \quad (6b)$$

is the Debye lattice heat capacity at constant volume due to acoustic phonons. The least-squares fits of the $C_p(T)$ data for $1.8 \text{ K} \leq T \leq 300 \text{ K}$ by Eqs. (6) are shown by the red curves in Figs. 5(a) and 5(b), where we used the analytic Padé approximant fitting function [54] for the Debye lattice heat capacity integral in Eq. (6b) and the above values of γ that were

kept fixed, with Θ_D the only adjustable parameter. The values $\Theta_D = 301(3) \text{ K}$ and $311(4) \text{ K}$ were found for CoAs-flux- and Sn-flux-grown crystals, respectively. Even though these two Θ_D values are nearly the same, the systematic deviations of the fits in Fig. 5(b) from the data for the Sn-grown crystal are larger than those in Fig. 5(a) for the CoAs-grown crystal, suggesting a significant difference between the phonon densities of states versus energy for crystals grown using the two fluxes.

The parameters obtained from the heat capacity fits for BaCo_2As_2 are summarized in Table III together with those of the other compounds studied below. The Debye temperatures Θ_R obtained from fitting the $\rho(T)$ data by the Bloch-Grüneisen model are also listed in Table III for comparison with the respective Θ_D values, where the Θ_R values are seen to be roughly a factor of two smaller than the respective Θ_D values. These large differences suggest that the Bloch-Grüneisen theory for the resistivity arising from electron scattering by acoustic phonons is not appropriate for modeling the high- T $\rho(T)$ data for these compounds.

C. Electronic structure calculations

The electronic density of states (DOS) for BaCo_2As_2 as a function of energy E relative to the Fermi energy E_F is shown in Fig. 6. A similar DOS versus energy has been previously reported for BaCo_2As_2 by Sefat *et al.* [39]. The major contribution to the DOS at E_F comes from the Co $3d$ band that is reflected as a sharp peak at E_F , which was also present in Fig. 4 of Ref. [39]. The sharp peak in the DOS at E_F is due to a Co $d_{x^2-y^2}$ flat band as in SrCo_2As_2 [40], where the x and y axes are defined with respect to the Co square lattice. The calculated DOS at E_F is

$$\mathcal{D}_{\text{band}}(E_F) = 8.23 \frac{\text{states}}{\text{eV f.u.}} \quad (7)$$

for both spin directions, which is comparable to but somewhat smaller than the value $\mathcal{D}_{\text{band}}(E_F) = 8.5 \text{ states/(eV f.u.)}$ found from LDA band calculations in Ref. [39]. The atom-decomposed contributions to $\mathcal{D}_{\text{band}}(E_F)$ are Ba, 0.24 states/(eV f.u.); Co, 5.92 states/(eV f.u.); and As, 0.48 states/(eV f.u.) for both spin directions. As usual, due to the interstitial DOS which is also accounted for in calculation of the total DOS, the total DOS is larger than the sum of the atom-decomposed partial contributions to $\mathcal{D}_{\text{band}}(E_F)$.

The bare density of states obtained from the DOS calculation is smaller than the value $\mathcal{D}(E_F) = 18.82(2) \text{ states/eV f.u.}$

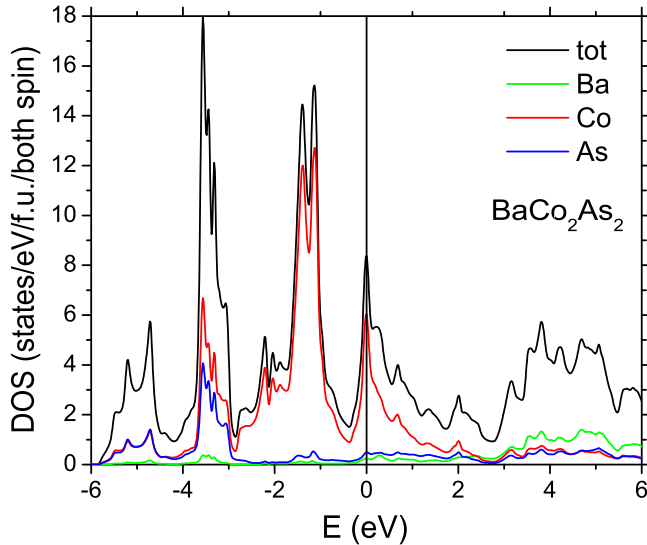


FIG. 6. (Color online) The density of states (DOS) for BaCo_2As_2 as a function of energy E relative to the Fermi energy $E_F = 0$ calculated using FP-LAPW. The atom-decomposed DOS of Ba, Co, and As atoms are also shown.

for both spin directions in Table III obtained from the γ value for the CoAs-grown crystal, evidently due to the presence of many-body electron-phonon and electron-electron (m^*/m_{band}) enhancement effects. We obtain the combined enhancement as

$$(1 + \lambda_{\text{el-ph}}) \frac{m^*}{m_{\text{band}}} = \frac{D^\nu(E_F)}{D_{\text{band}}(E_F)} = 2.29, \quad (8)$$

where $\lambda_{\text{el-ph}}$ is the electron-phonon coupling constant. This value is close to the enhancement value of 2.2 obtained for BaCo_2As_2 in Ref. [39]. Xu *et al.* found that their ARPES band dispersion data for BaCo_2As_2 could be reproduced by band theory with only a modest renormalization $m^*/m_{\text{band}} \approx 1.4$ [43], which together with Eq. (8) yields the value $\lambda_{\text{el-ph}} \approx 0.6$. This electron-phonon coupling constant is similar to the value $\lambda_{\text{el-ph}} = 0.76$ obtained from *ab initio* calculations for the isostructural and closely related compound BaNi_2As_2 [60].

D. Magnetization and magnetic susceptibility

The T dependences of the zero-field-cooled magnetic susceptibility $\chi \equiv M/H$ of CoAs-flux-grown as well as Sn-flux-grown BaCo_2As_2 single crystals measured in a magnetic field $H = 3.0$ T applied along the c axis ($\chi_c, H \parallel c$) and in the ab plane ($\chi_{ab}, H \perp c$) are shown in Fig. 7. These data indicate that BaCo_2As_2 remains paramagnetic without any magnetic phase transitions above 1.8 K, as also deduced previously [39]. A weak T dependence is observed with an upturn in χ below ~ 50 K. The χ of the Sn-flux-grown crystal is smaller than that of CoAs-flux-grown crystal. The χ of both the CoAs-flux-grown and Sn-flux-grown crystals are anisotropic over the entire 1.8–350 K temperature range of measurement with $\chi_{ab} > \chi_c$. This anisotropy with $\chi_{ab} > \chi_c$ was reproducible on several batches of crystals. The same sign of the anisotropy in χ is commonly observed in undoped and doped FeAs-based compounds [9].

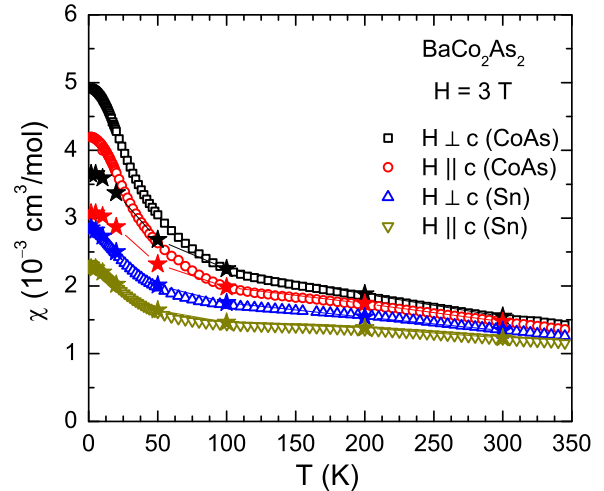


FIG. 7. (Color online) Zero-field-cooled magnetic susceptibility $\chi \equiv M/H$ of CoAs-flux-grown and Sn-flux-grown BaCo_2As_2 single crystals versus temperature T in the T range 1.8–350 K measured in a magnetic field $H = 3.0$ T applied along the c axis ($\chi_c, H \parallel c$) and in the ab plane ($\chi_{ab}, H \perp c$). The stars of respective colors represent the χ obtained from fitting $M(H)$ isotherm data by Eq. (9). The solid lines connecting the stars are guides to the eye.

The $M(H)$ isotherms obtained for BaCo_2As_2 at eight temperatures between 1.8 and 300 K for both $H \perp c$ (M_{ab}) and $H \parallel c$ (M_c) are shown in Figs. 8 and 9 for CoAs-flux-grown and Sn-flux-grown crystals, respectively. While the $M(H)$ isotherms for the Sn-flux-grown crystals are almost linear in H even at the lowest measured $T = 1.8$ K, the $M(H)$ isotherms of CoAs-flux-grown crystals exhibit nonlinearity at low H at $T \lesssim 20$ K, although the $M(H)$ isotherms for $T \geq 100$ K are almost linear in H . The nonlinearity in $M(H)$ was reproducibly found in another batch of BaCo_2As_2 crystals grown out of CoAs self-flux.

This nonlinearity in the $M(H)$ isotherms in Fig. 8 of CoAs flux-grown crystals at $T \lesssim 20$ K might be considered to be due to the presence of a small amount of FM impurity. However, the FM impurity cannot be CoAs from the flux, because CoAs is a Pauli paramagnetic metal [61]; another potential impurity is CoAs_2 but this compound is a diamagnetic semiconductor [62]. Furthermore, the nonlinearity in the $M(H)$ curves for $H < 1$ T does not rise as fast with increasing field as expected from a ferromagnetic impurity which has a spontaneous ordered moment at $H = 0$ (see Figs. 15, 16, and 21 below). A more likely scenario is that the low-field nonlinearity in the $M(H)$ isotherms of the CoAs flux-grown crystals at $T \lesssim 20$ K is due to the presence of saturable paramagnetic defects. In this case the absence of such nonlinearities in the $M(H)$ data for the Sn flux-grown crystals would be associated with absence of paramagnetic defects in these crystals. This inference is consistent with the different crystal growth temperature profiles for the FeAs- and Sn-grown crystals because the FeAs-grown crystals were quenched from a high temperature of 1180 °C, whereas the Sn-grown crystals were quenched from the much lower temperature of 600 °C, as discussed in Sec. II. The lower quenching temperature would tend to anneal out paramagnetic defects present at higher temperatures.

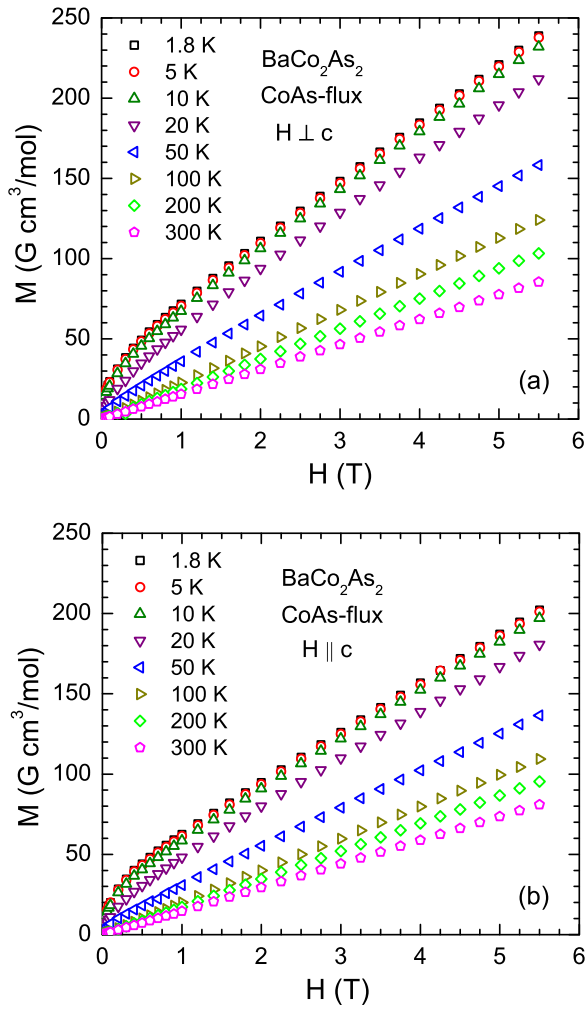


FIG. 8. (Color online) Isothermal magnetization M of a CoAs-flux-grown BaCo_2As_2 single crystal as a function of applied magnetic field H measured at the indicated temperatures for H applied (a) in the ab plane ($M_{ab}, H \perp c$) and (b) along the c axis ($M_c, H \parallel c$).

The intrinsic susceptibility χ of a CoAs-flux-grown BaCo_2As_2 crystal is the slope of the high-field linear part of an $M(H)$ isotherm where the paramagnetic defects are saturated and is obtained from a fit of the $M(H)$ data for $H \geq 2$ T by

$$M(H) = M_s + \chi H, \quad (9)$$

where M_s is the paramagnetic defect saturation magnetization. The resulting χ data versus temperature are shown as stars in Fig. 7.

For the CoAs-flux-grown crystal the intrinsic χ values obtained from the slopes of the high-field fits of the $M(H)$ data in Fig. 8 by Eq. (9) decrease slowly with increasing T for $T \geq 100$ K with negligible saturation magnetization values $M_s^{ab} = 0.0(5)$ G cm³/mol and $M_s^c = 0.0(4)$ G cm³/mol. For this reason these $\chi(T)$ values, shown as the stars in Fig. 7, agree with the values of $M(T)/H$ data in Fig. 7 in this T range. On the other hand, the intrinsic χ values for $T \lesssim 20$ K are significantly smaller than obtained from the $M(T)/H$ measurements as seen in Fig. 7, a difference that arises from the low-field nonlinearity of the $M(H)$ isotherms in Fig. 8 in

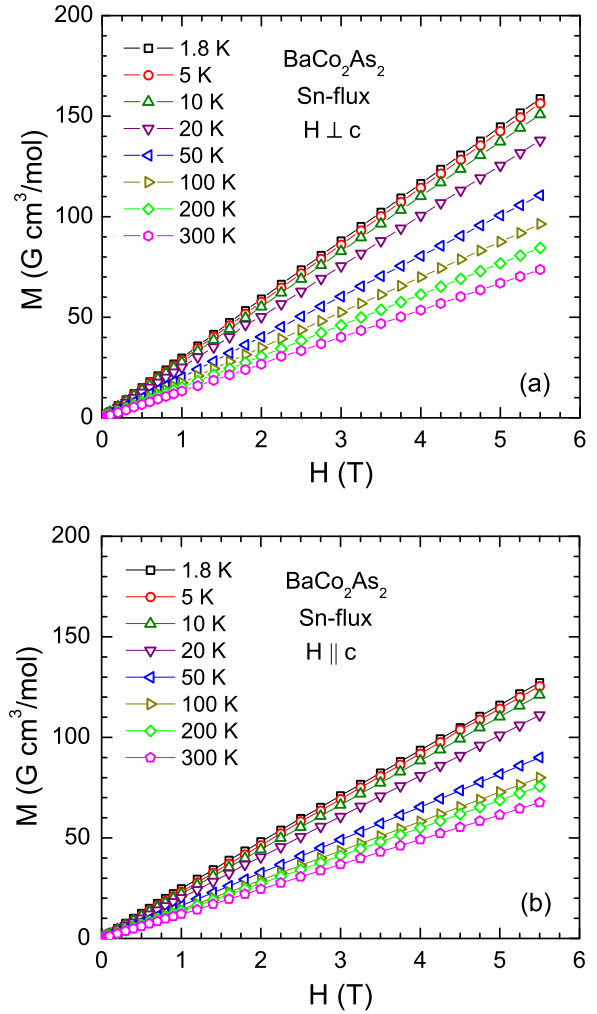


FIG. 9. (Color online) Isothermal magnetization M of a Sn-flux-grown BaCo_2As_2 single crystal as a function of applied magnetic field H measured at the indicated temperatures for H applied (a) in the ab plane ($M_{ab}, H \perp c$) and (b) along the c axis ($M_c, H \parallel c$).

this temperature range. We conclude that intrinsic upturns in the $\chi(T)$ data occur below ~ 100 K in Fig. 7. This conclusion is confirmed by the NMR shift measurements of the T -dependent spin susceptibility of a CoAs self-flux-grown crystal in the following section.

By fitting the $M(H)$ data for the CoAs-flux-grown crystal at $T = 1.8$ K and $H > 2$ T in Fig. 8 by Eq. (9), one obtains a saturation magnetization of the paramagnetic defects for either field direction as $M_s \approx 40$ G cm³/mol $\approx 0.0072 \mu_B/\text{f.u.}$, which corresponds to the saturation magnetization of 0.72 mol% of spin- $\frac{1}{2}$ defects with spectroscopic splitting factor $g = 2$.

For the Sn-flux-grown crystals, good agreement is seen in Fig. 7 between the $\chi \equiv M(T)/H$ data and the χ values obtained from the high-field slope of the $M(H)$ isotherm data in Fig. 9 for the entire 1.8 K to 300 K temperature range, indicating that the M_s values are negligible. Thus, there are no observable FM impurities or saturable paramagnetic defects in Sn-flux-grown BaCo_2As_2 crystals.

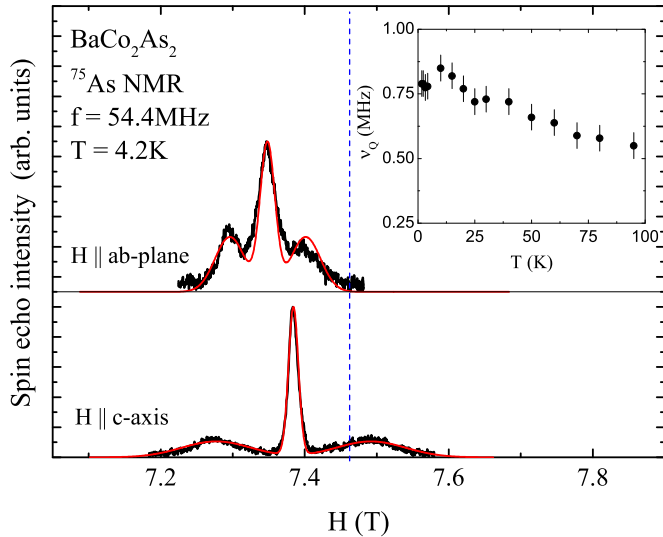


FIG. 10. (Color online) Field-swept ^{75}As -NMR spectra at temperature $T = 4.2$ K for magnetic fields perpendicular (top panel) and parallel (bottom panel) to the c axis. The black and red lines are observed and simulated spectra, respectively. The vertical dotted line corresponds to the zero-shift ($K = 0$) position. Inset: ^{75}As nuclear quadrupole frequency ν_Q versus T .

E. NMR spectroscopy

1. ^{75}As NMR spectra

Figure 10 shows field-swept ^{75}As -NMR spectra of a CoAs-flux-grown BaCo_2As_2 crystal at $T = 4.2$ K for magnetic fields $H \parallel c$ axis and $H \parallel ab$ plane. The spectra exhibit features typical of a nuclear spin $I = 3/2$ with Zeeman and quadrupolar interactions, which result in a sharp central transition and two satellite peaks split by the quadrupolar interaction of the As nucleus with the local electric field gradient (EFG). The observed quadrupole-split NMR spectra were reproduced by the nuclear spin Hamiltonian

$$\mathcal{H} = -\gamma \hbar \vec{I} \cdot \vec{H}_{\text{eff}} + \frac{h\nu_Q}{6} [3I_z^2 - I(I+1)], \quad (10)$$

where H_{eff} is the effective field at the As site (summation of external field H and the hyperfine field H_{hf}), h is Planck's constant, $\hbar = h/(2\pi)$, and ν_Q is the nuclear quadrupole frequency. Here the asymmetry parameter of the EFG is assumed to be zero because of the tetragonal symmetry of the compound. The red curves in Fig. 10 show simulated spectra calculated from Hamiltonian (10). The principle axis of the EFG at the As site in tetragonal BaCo_2As_2 is along the crystal c axis as in SrCo_2As_2 (Ref. [40]) and also as in members of the AFe_2As_2 family with tetragonal symmetry [63–67] since the As site has a local fourfold rotational symmetry about the c axis. The satellite linewidth, which reflects the distribution of EFG due to defects or lattice distortion, is relatively large. To reproduce the linewidth, one needs to introduce $\sim 63\%$ distributions of the quadrupole frequency ($\Delta\nu_Q = 0.5$ MHz) at $T = 4.2$ K, as shown by red curves in the figure. This is much larger than the $\sim 7\%$ distributions of $\nu_Q = 8.45$ MHz at $T = 4.2$ K for the case of ^{75}As -NMR in SrCo_2As_2 [40]. This indicates that the local As environment in BaCo_2As_2 has a

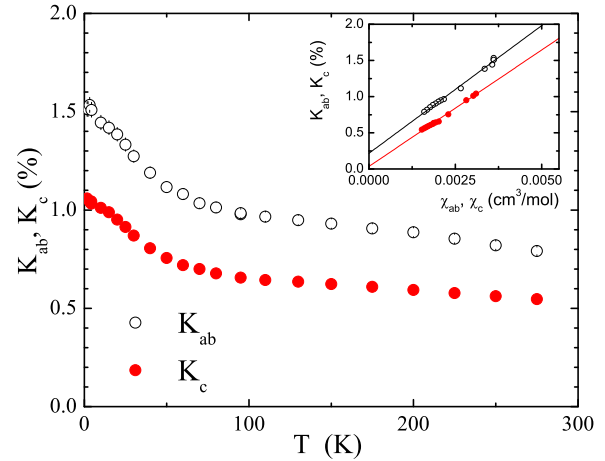


FIG. 11. (Color online) (a) T -dependence of ^{75}As -NMR shifts K_{ab} and K_c for BaCo_2As_2 . Inset: K versus magnetic susceptibility χ plots for the corresponding ab and c components of K and χ with T as an implicit parameter, where we used the χ data shown by stars in Fig. 7 below 150 K and the observed χ for $T \geq 150$ K. The solid lines are linear fits for the two field directions.

lower degree of homogeneity than in SrCo_2As_2 . The spectra for BaCo_2As_2 for both H directions did not show any obvious magnetic broadenings at any temperature between 1.6 K and 275 K, which indicates that static magnetic ordering does not occur in this compound above 1.6 K.

The inset of Fig. 10 shows the T dependence of the nuclear quadrupole frequency $\nu_Q = eQV_{ZZ}/2h$, where e is the fundamental charge, Q is the quadrupole moment of the As nucleus, and V_{ZZ} is the EFG at the As site. The ν_Q increases by about 45% from 0.55 MHz at 95 K to 0.8 MHz at 1.6 K. This behavior contrasts sharply with that observed in SrCo_2As_2 , where ν_Q decreases from 10 MHz at 250 K to 8.45 MHz at 1.6 K. An increase of ν_Q by $\sim 30\%$ on lowering T from 2 MHz at 300 K to 2.6 MHz at ~ 210 K was observed in SrFe_2As_2 [63]. Since V_{ZZ} arises from hybridization between the As- $4p$ and Co- $3d$ orbitals with an additional contribution from the noncubic part of the spatial distribution of surrounding ions, the small ν_Q in BaCo_2As_2 compared with that in SrCo_2As_2 indicates a weaker hybridization between the orbitals in BaCo_2As_2 . This can be qualitatively interpreted when one takes into account the difference of the c -axis lattice constant of 12.67 Å in BaCo_2As_2 and 11.80 Å in SrCo_2As_2 .

Figure 11 shows the T dependence of the NMR shifts K_{ab} and K_c for H parallel to the ab plane and to the c axis, respectively. We emphasize that these results are intrinsic to the crystal and are not sensitive to the presence of the paramagnetic defects in CoAs-grown crystals detected in the above $\chi(T)$ measurements. With decreasing T , both K_{ab} and K_c increase slowly down to ~ 50 K and show a significant additional increase below that temperature, similar to the corresponding intrinsic χ_{ab} and χ_c data for a CoAs flux-grown crystal of BaCo_2As_2 (the stars in Fig. 7). The NMR shift consists of the T -dependent spin shift $K_{\text{spin}}(T)$ and the T -independent orbital (chemical) shift K_{chem} : $K(T) = K_{\text{spin}}(T) + K_{\text{chem}}$, where $K_{\text{spin}}(T)$ is proportional to the intrinsic spin susceptibility $\chi_{\text{spin}}(T)$ via the hyperfine coupling constant A , $K_{\text{spin}}(T) =$

$\frac{A\chi_{\text{spin}}(T)}{N_A}$. Here N_A is Avogadro's number. As shown in the inset of Fig. 11, both K_{ab} and K_c indeed vary linearly with the corresponding intrinsic χ , where we used χ data shown by stars in Fig. 7 below 150 K and the observed χ for $T \geq 150$ K. From the respective slopes, the hyperfine coupling constants are estimated to be $A_{ab} = (40 \pm 3)$ kOe/ μ_B and $A_c = (36 \pm 2)$ kOe/ μ_B , respectively. These values are respectively smaller than $A_{ab} = 65.9$ kOe/ μ_B and $A_c = 45.0$ kOe/ μ_B for SrCo₂As₂ [40]. Since the hyperfine coupling is mainly due to hybridization between the 4s and 4p orbitals of the As atoms with the 3d orbitals of the Co atoms, the small hyperfine couplings indicate a weak hybridization between these orbitals in BaCo₂As₂. This is consistent with the small ν_Q for ⁷⁵As in BaCo₂As₂ because the ν_Q is also affected by the strength of the hybridization between the As-4p and Co-3d orbitals. As pointed out above, a weak hybridization between the orbitals can be qualitatively explained by taking into account the difference between the *c*-axis lattice constants in BaCo₂As₂ and SrCo₂As₂.

The agreement between the respective T dependences for both field directions of the intrinsic $\chi(T)$ in Fig. 7 of a CoAs-grown BaCo₂As₂ crystal with those of the NMR $K(T)$ of a similar crystal demonstrates that the former T dependences are the intrinsic T -dependences of the spin susceptibilities of BaCo₂As₂. These T dependencies are different from those of pure and doped BaFe₂As₂ crystals above their long-range AFM spin-density wave ordering temperatures [9], where the $\chi(T)$ data for both field directions decrease with decreasing T instead of increasing with decreasing T as in BaCo₂As₂.

K_{chem} is estimated to be $(0.22 \pm 0.02)\%$ and $(0.04 \pm 0.02)\%$ for H parallel to the *ab* plane and *c* axis, respectively. The $K_{\text{chem},c}$ is similar to $K_{\text{chem},c} \sim 0$ in SrCo₂As₂, and $K_{\text{chem},ab}$ is larger than $K_{\text{chem},ab} \sim 0$ in SrCo₂As₂. On the other hand, $K_{\text{chem},ab} = 0.22\%$ is close to $K_{\text{chem}} \sim 0.2\%$ reported in BaFe₂As₂ (Ref. [64]) and $K_{\text{chem}} \sim 0.22\%$ in BaFe_{1.84}Co_{0.16}As₂ [68]. Kitagawa *et al.* commented that the value of $K_{\text{chem}} \sim 0.2\%$ seemed rather large [64]. At present the origin of the compound- and orientation-dependent K_{chem} is not clear.

2. ⁷⁵As nuclear spin dynamics

The ⁷⁵As nuclear spin-lattice relaxation rate $1/T_1$ at each T is determined by fitting the nuclear magnetization M versus time t dependence of the central line after saturation using the double-exponential function

$$1 - \frac{M(t)}{M(\infty)} = 0.1 \exp(-t/T_1) + 0.9 \exp(-6t/T_1), \quad (11)$$

as expected for the central line of the spectrum of the ⁷⁵As ($I = 3/2$) nucleus, where $M(t)$ and $M(\infty)$ are the nuclear magnetization at time t after saturation and the equilibrium nuclear magnetization at time $t \rightarrow \infty$, respectively. Figure 12(a) shows $1/T_1$ versus T for magnetic fields parallel and perpendicular to the *c* axis at $H = 7.4$ T. For both field orientations, the $1/T_1$ values decrease monotonically with decreasing T , show slight increases around 20 K and then decrease again at low temperatures. $1/T_1$ was also measured at a low magnetic field $H = 1.64$ T as shown in Fig. 12(a), where the $1/T_1$ data are limited to temperatures below ~ 20 K

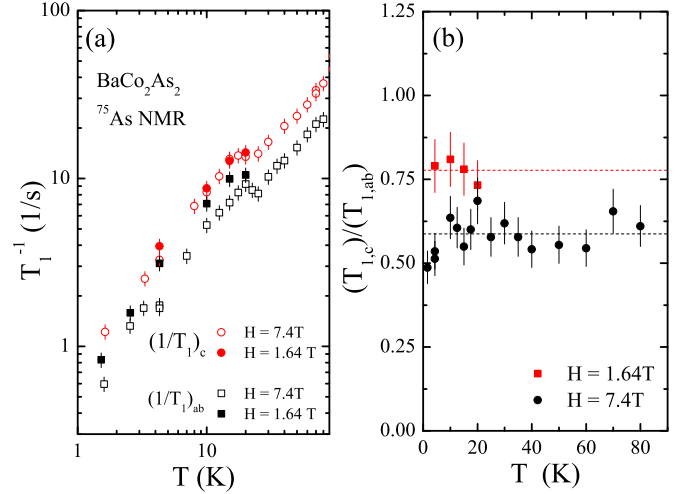


FIG. 12. (Color online) (a) ⁷⁵As nuclear spin-lattice relaxation rate $1/T_1$ versus temperature T for BaCo₂As₂ for magnetic field directions $H \parallel c$ axis and $H \parallel ab$ plane under two magnetic fields $H = 1.64$ and 7.4 T. (b) T -dependences of the ratio $r \equiv T_{1,c}/T_{1,ab}$ for $H = 1.64$ and 7.4 T.

due to poor signal intensity at higher temperatures at the low magnetic field. As can be seen in the figure, $1/T_{1,c}$ is independent of H within our experimental uncertainty, but $1/T_{1,ab}$ increases with decreasing H .

Based on these results, we now discuss the relationship between the anisotropy of our $1/T_1$ data and the anisotropy of the Co spin fluctuations. We closely follow the procedure previously developed for Fe pnictides [63,69,70], and especially for the recent ⁷⁵As-NMR measurements of SrCo₂As₂ [40]. Defining the amplitude of the spin fluctuation spectral density as $S_\alpha(\mathbf{q}, \omega_N)$ ($\alpha = ab, c$), where \mathbf{q} is the wave vector of a spin fluctuation in terms of Co square lattice notation, one obtains [40,69]

$$\frac{1}{T_{1,ab}} = \begin{cases} |A_{ab}S_{ab}|^2 + |A_cS_c|^2 & [\mathbf{q} = (0,0)], \\ 2|A_{ab}S_{ab}|^2 & [\mathbf{q} = (\pi,0), (0,\pi)], \\ |DS_{ab}|^2 & [\mathbf{q} = (\pi,\pi)], \\ 2|DS_{ab}|^2 & [\mathbf{q} = (\pi,\pi)], \end{cases} \quad (12)$$

where $\mathbf{q} = (0,0)$ is the *ab*-plane component of the ferromagnetic wave vector, $\mathbf{q} = (\pi,0)$ or $(0,\pi)$ is the stripe wave vector as observed in the AFe₂As₂ and SrCo₂As₂ compounds, and $\mathbf{q} = (\pi,\pi)$ is the alternately termed Néel-, checkerboard-, or G-type AFM wave vector. Then defining the ratio

$$r \equiv \frac{1/T_{1,ab}}{1/T_{1,c}}, \quad (13)$$

Eqs. (12) give

$$r = \begin{cases} 0.5 + 0.5\left(\frac{A_cS_c}{A_{ab}S_{ab}}\right)^2 & [\mathbf{q} = (0,0)], \\ 0.5 + \left(\frac{S_{ab}}{S_c}\right)^2 & [\mathbf{q} = (\pi,0), (0,\pi)], \\ 0.5 & [\mathbf{q} = (\pi,\pi)]. \end{cases} \quad (14)$$

As plotted in Fig. 12(b), the r value is almost constant versus T with the magnetic-field-dependent values of $r \approx 0.78$ and 0.59 for $H = 1.64$ and 7.4 T, respectively. We first analyze the T_1 data with the first of Eqs. (14). Using the ratio of the hyperfine fields $A_c/A_{ab} = 36/40 = 0.9$ determined from the K - χ plots in the inset of Fig. 11, the values of $r \approx 0.78$ and 0.59 provide anisotropic spin fluctuations with $\mathcal{S}_c = 0.83\mathcal{S}_{ab}$ and $\mathcal{S}_c = 0.47\mathcal{S}_{ab}$ for $H = 1.64$ and 7.4 T, respectively. This means that ferromagnetic Co spin fluctuations in the ab plane are stronger than along the c axis. Anisotropic ferromagnetic spin fluctuations were also pointed out in SrCo_2As_2 [40], where \mathcal{S}_c is larger than \mathcal{S}_{ab} in contrast to our result for BaCo_2As_2 . As for the magnetic field dependence, Eqs. (12) show that the H -independent $1/T_{1,c}$ along the c axis indicates that \mathcal{S}_{ab} is not affected by the application of H ; thus, \mathcal{S}_c can be concluded to be suppressed with H . Néel-type spin fluctuations can be ruled out because, according to Eqs. (12), r must be independent of H , in conflict with our measurements which show an H -dependent r .

Our $1/T_1$ data can also be analyzed with the second of Eqs. (14) in terms of stripe-type spin fluctuations with $\mathbf{q} = (\pi, 0)$ or $(0, \pi)$. Such spin fluctuations/correlations have been revealed in SrCo_2As_2 from inelastic neutron scattering measurements [42]. Using Eqs. (14), the values of $r \approx 0.78$ and 0.59 can be reproduced with anisotropic stripe-type AFM spin fluctuations with $\mathcal{S}_{ab} = 0.52\mathcal{S}_c$ and $\mathcal{S}_{ab} = 0.28\mathcal{S}_c$ for $H = 1.64$ and 7.4 T, respectively. From Eqs. (12), \mathcal{S}_c is independent of H from the $1/T_{1,c}$ data, so we conclude that \mathcal{S}_{ab} is suppressed by H for the stripe-type AFM spin fluctuations. An anisotropy in stripe-type spin fluctuations is also observed in various Fe-based superconductors near T_N in the paramagnetic state, but \mathcal{S}_{ab} is always larger than \mathcal{S}_c [63,71], in contrast to our results.

Although it is clear that the spin fluctuations are anisotropic and H dependent in BaCo_2As_2 , one cannot uniquely determine the nature of the spin fluctuations from the NMR data alone. Indeed, spin fluctuations may occur with peaks at both the FM and stripe-type AFM wave vectors in the \mathbf{q} -dependent static susceptibility $\chi(\mathbf{q})$ obtained from LDA calculations such as reported for SrCo_2As_2 in Ref. [42].

V. PHYSICAL PROPERTIES OF $\text{Ba}_{0.946}\text{K}_{0.054}\text{Co}_2\text{As}_2$

Single crystals of $\text{Ba}_{0.946}\text{K}_{0.054}\text{Co}_2\text{As}_2$ were grown in CoAs self-flux. The $\chi(T) \equiv M(T)/H$ data obtained for a single crystal at low fields $H \leq 0.05$ T are shown in Fig. 13. Figure 13 shows that at $H = 0.01$ T a sharp increase in χ occurs below 18 K with a peak near 10 K in $\chi(T)$ measured on warming after zero-field-cooling (ZFC) for both $H \parallel c$ and $H \perp c$. An irreversibility is observed between the ZFC and field-cooled (FC) χ data below 12.5 K. Such behaviors are typically observed in systems with ferromagnetic (FM) correlations. An increase in H causes the peak in $\chi(T)$ to broaden and χ becomes almost T independent below 8 K for $H \geq 0.05$ T. No superconductivity is observed at $T \geq 1.8$ K. The anisotropy of $\chi(T)$ measured in a much larger field $H = 3.0$ T is presented in Fig. 14. An anisotropy $\chi_{ab} > \chi_c$ is observed over the entire T range of the measurement.

The $M(H)$ isotherm data for a $\text{Ba}_{0.946}\text{K}_{0.054}\text{Co}_2\text{As}_2$ single crystal at different temperatures are shown in Fig. 15. For

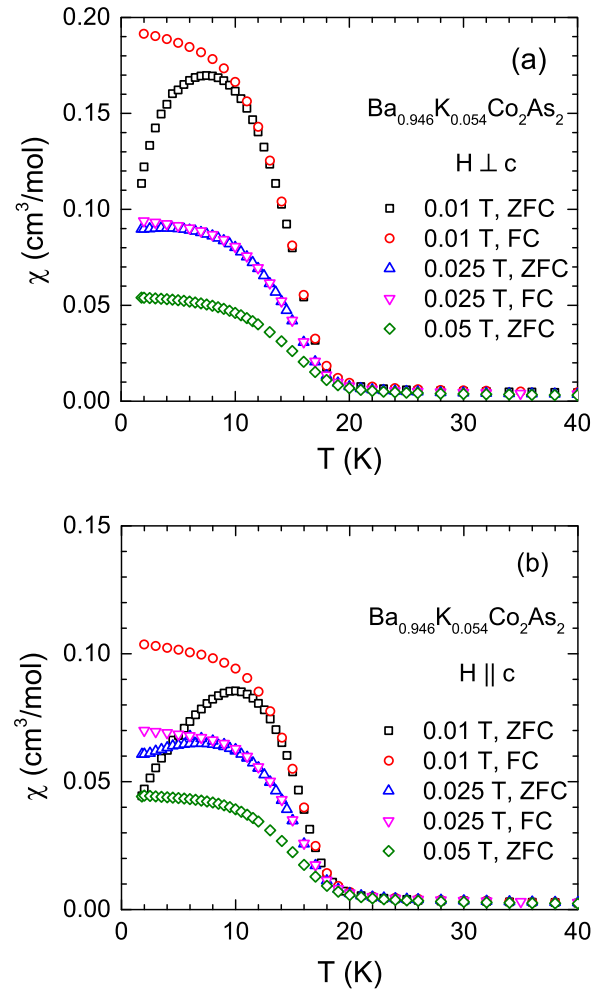


FIG. 13. (Color online) Zero-field-cooled (ZFC) and field-cooled (FC) magnetic susceptibility χ of a $\text{Ba}_{0.946}\text{K}_{0.054}\text{Co}_2\text{As}_2$ single crystal grown in CoAs flux as a function of temperature T in the temperature range 1.8–40 K measured in low magnetic fields H applied (a) in the ab plane ($\chi_{ab}, H \perp c$) and (b) along the c axis ($\chi_c, H \parallel c$).

$T > 50$ K, the $M(H)$ isotherms are almost linear in H with anisotropy $M_{ab} > M_c$, consistent with the $\chi(T)$ data in Fig. 14. The appearance of a spontaneous FM moment at low H is clearly seen in the $M(H)$ isotherms at temperatures $T \leq 10$ K, which are below the temperature of ~ 18 K at which a FM-like increase is observed in $\chi(T)$ in Fig. 13. The saturation (ordered) moment is obtained by fitting the high-field linear $M(H)$ data at 1.8 K in Fig. 15 by Eq. (9), yielding $M_s \approx 0.007\mu_B/\text{f.u.}$ for both $H \perp c$ and $H \parallel c$. Interestingly, this is the same M_s as found above from Fig. 8 for the paramagnetic defects in a CoAs-flux-grown crystal of BaCo_2As_2 .

Figure 16 shows the hysteresis of $M(H)$ at $T = 1.8$ K on sweeping the field between ± 0.10 T. The hysteresis is typical of a (weak) ferromagnet where here the remanent magnetization is about 0.002 – $0.003 \mu_B/\text{f.u.}$ for the two field directions and the respective coercive fields are about 0.01 T.

The $C_p(T)$ data of a $\text{Ba}_{0.946}\text{K}_{0.054}\text{Co}_2\text{As}_2$ crystal are shown in Fig. 17. The low- T $C_p(T)$ data do not show any anomaly at or below the onset of the FM-like increase in the low-field χ

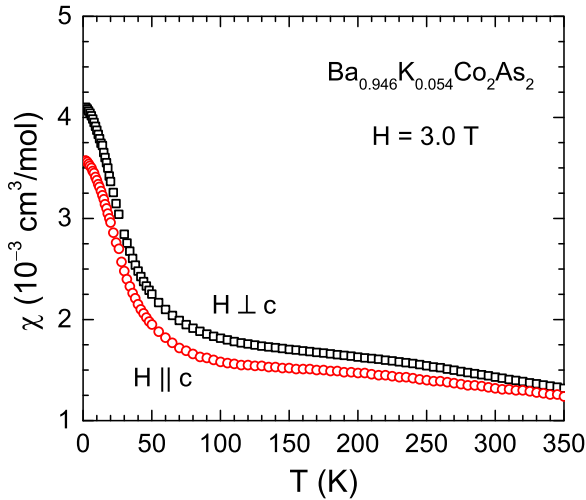


FIG. 14. (Color online) Zero-field-cooled magnetic susceptibility χ of a $\text{Ba}_{0.946}\text{K}_{0.054}\text{Co}_2\text{As}_2$ single crystal grown in CoAs flux as a function of temperature T in the temperature range 1.8–300 K measured in a magnetic field $H = 3.0$ T applied along the c axis ($\chi_c, H \parallel c$) and in the ab plane ($\chi_{ab}, H \perp c$).

at ~ 18 K. This is likely due to the very small ordered moment. The low- T $C_p(T)$ data for $1.8 \text{ K} \leq T \leq 10 \text{ K}$ shown in the inset of Fig. 17 were analyzed by Eq. (3), yielding the γ , β , and δ values listed in Table III. The red curve in the inset of Fig. 17 shows the fit of $C_p(T)/T$. The values of $\mathcal{D}(E_F)$ obtained from γ and Θ_D obtained from β are listed. Another estimate of $\Theta_D = 310(3)$ K is obtained from a fit of the $C_p(T)$ data for $1.8 \text{ K} \leq T \leq 300 \text{ K}$ by Eqs. (6), which is shown by the red curve in Fig. 17, where we used the analytic Padé approximant fitting function [54] for the Debye lattice heat capacity integral.

The in-plane $\rho(T)$ data for a $\text{Ba}_{0.946}\text{K}_{0.054}\text{Co}_2\text{As}_2$ crystal are shown in Fig. 18. Like the $C_p(T)$ data, the $\rho(T)$ data do not show any anomaly at or below the FM-like onset temperature ~ 18 K observed in the $\chi(T)$ data. The $\rho(T)$ is metallic with $\rho(T = 1.8 \text{ K}) = 10.0 \mu\Omega \text{ cm}$ and $\text{RRR} = 10.2$. The low value of ρ_0 and high value of RRR indicate the high quality of the crystal. The $\rho(T)$ data for $1.8 \text{ K} \leq T \leq 50 \text{ K}$ were fitted by Eq. (1) yielding $\rho_0 = 9.94(1) \mu\Omega \text{ cm}$ and $A = 3.31(1) \times 10^{-3} \mu\Omega \text{ cm/K}^2$. The red curve in the inset of Fig. 18 shows the fit of ρ vs T^2 data with these parameters. The T^2 dependence of ρ at low T indicates that a Fermi liquid ground state persists in the K-doped crystals. The Kadowaki-Woods ratio $R_{\text{KW}} = A/\gamma^2 = 0.17 \times 10^{-5} \mu\Omega \text{ cm}/(\text{mJ/mol K})^2$. The $\rho(T)$ data were also fitted by Eqs. (2) for $1.8 \text{ K} \leq T \leq 300 \text{ K}$, where again we used the analytic Padé approximant function [54], which is shown by the red curve in Fig. 18, yielding $\Theta_R = 181(4)$ K. The parameters of the various fits to the $\rho(T)$ data are summarized in Table II.

VI. PHYSICAL PROPERTIES OF $\text{Ba}_{0.935}\text{K}_{0.065}\text{Co}_2\text{As}_2$

The $\chi(T)$ for a CoAs flux-grown crystal of $\text{Ba}_{0.935}\text{K}_{0.065}\text{Co}_2\text{As}_2$ was measured at different values of H . Here again no superconductivity is observed at $T \geq 1.8$ K. At $H = 0.01$ T the $\chi(T) \equiv M(T)/H$ data shown

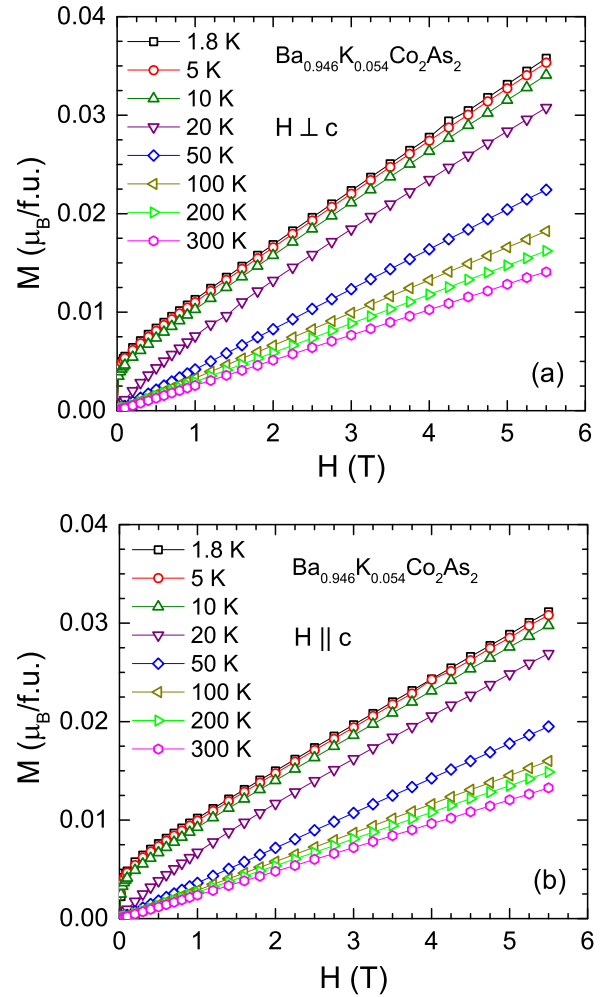


FIG. 15. (Color online) Isothermal magnetization M of a $\text{Ba}_{0.946}\text{K}_{0.054}\text{Co}_2\text{As}_2$ single crystal grown in CoAs flux as a function of applied magnetic field H measured at the indicated temperatures for H applied (a) in the ab plane ($M_{ab}, H \perp c$) and (b) along the c axis ($M_c, H \parallel c$).

in Fig. 19 exhibit a rapid FM-like increase below 19 K with an irreversibility between ZFC and FC χ below 11 K and a peak in the ZFC χ at 9 K for both $H \parallel c$ and $H \perp c$. With increasing H the peak becomes broader and χ becomes almost independent of T below 8 K for $H = 0.05$ T. Although the $\chi(T)$ for $H \parallel c$ and $H \perp c$ exhibit similar behaviors at low H , they have somewhat different magnitudes. On the other hand, from the $\chi(T)$ data measured at $H = 3.0$ T that are shown in Fig. 20 it is evident that there is almost no anisotropy in χ at this field over the entire T range of measurement.

$M(H)$ isotherms for a $\text{Ba}_{0.935}\text{K}_{0.065}\text{Co}_2\text{As}_2$ single crystal at eight temperatures with $H \perp c$ and $H \parallel c$ are shown in Figs. 21(a) and 21(b), respectively. While for $T > 50$ K the $M(H)$ isotherms are almost linear in H , the low- T $M(H)$ isotherms exhibit a FM-like rapid increase of M with H for $H < 0.1$ T. Fitting the high-field data by Eq. (9), the saturation moments are similar for $H \perp c$ and $H \parallel c$, with the values $M_s = 0.028$ and $0.029 \mu_B/\text{f.u.}$, respectively. These M_s values are about a factor of four larger than obtained from Fig. 15 for

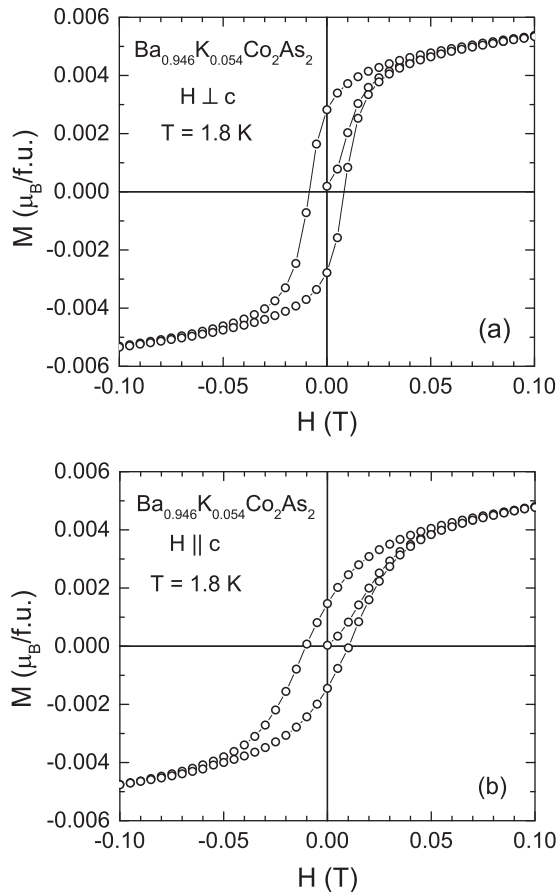


FIG. 16. Isothermal magnetization M of a $\text{Ba}_{0.946}\text{K}_{0.054}\text{Co}_2\text{As}_2$ single crystal (self-flux grown) as a function of applied magnetic field H measured at 1.8 K for H applied (a) in the ab plane ($M_{ab}, H \perp c$) and (b) along the c axis ($M_c, H \parallel c$).

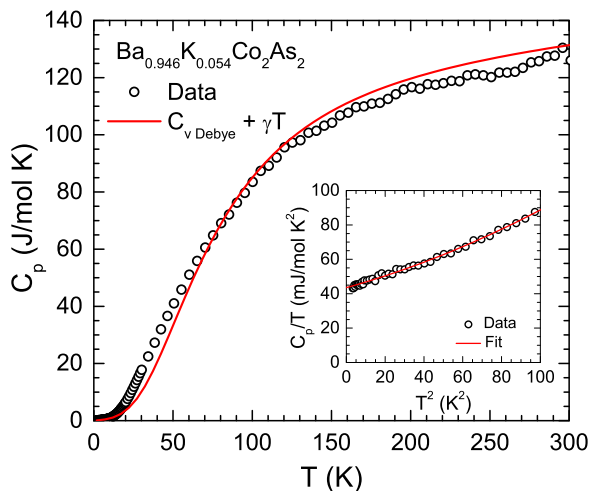


FIG. 17. (Color online) Heat capacity C_p of a $\text{Ba}_{0.946}\text{K}_{0.054}\text{Co}_2\text{As}_2$ single crystal grown in CoAs flux as a function of temperature T measured in zero magnetic field. The solid curve is a fit by the sum of the contributions from the Debye lattice heat capacity $C_{v, \text{Debye}}(T)$ and the electronic heat capacity γT according to Eq. (6a). Inset: C_p/T versus T^2 below 10 K; the red curve represents the fit of the data by Eq. (3) for $1.8 \text{ K} \leq T \leq 10 \text{ K}$.

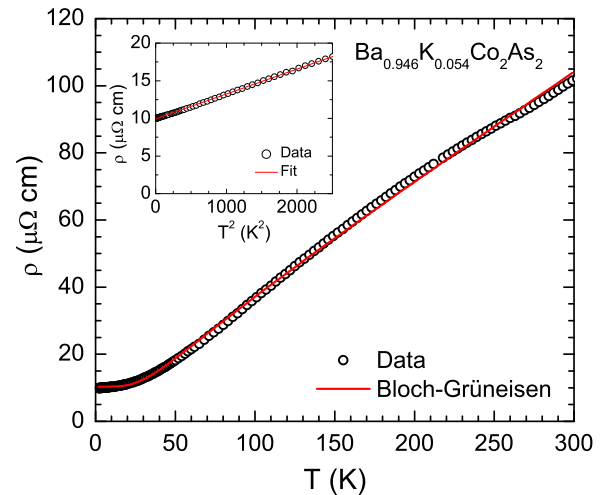


FIG. 18. (Color online) In-plane electrical resistivity ρ of a $\text{Ba}_{0.946}\text{K}_{0.054}\text{Co}_2\text{As}_2$ single crystal grown in CoAs flux as a function of temperature T measured in zero magnetic field. The red solid curve is a fit by the Bloch-Grüneisen model. (Inset) ρ vs T^2 below 50 K. The straight red line is a fit by $\rho = \rho_0 + AT^2$ for $1.8 \text{ K} \leq T \leq 50 \text{ K}$.

a self-flux-grown crystal of $\text{Ba}_{0.946}\text{K}_{0.054}\text{Co}_2\text{As}_2$ with a similar composition.

The $C_p(T)$ data of a $\text{Ba}_{0.935}\text{K}_{0.065}\text{Co}_2\text{As}_2$ crystal are shown in Fig. 22. No anomaly is seen in the low- T $C_p(T)$ data which would indicate a (weak) ferromagnetic transition, likely due to the very small ordered moment. An analysis of the low- T $C_p(T)$ data for $1.8 \text{ K} \leq T \leq 10 \text{ K}$ shown in the inset of Fig. 22 by Eq. (3) gives the fitting parameters γ , β , and δ listed in Table III. The value of $\mathcal{D}(E_F)$ obtained from γ and Θ_D obtained from β are listed in Table III. A fit of the $C_p(T)$ data for $1.8 \text{ K} \leq T \leq 300 \text{ K}$ by Eqs. (6) using the analytic Padé approximant fitting function [54] for the Debye lattice heat capacity integral gives the value of $\Theta_D = 304(3)$ listed in Table III. The fit is shown by the red curve in Fig. 22.

The in-plane $\rho(T)$ data of a $\text{Ba}_{0.935}\text{K}_{0.065}\text{Co}_2\text{As}_2$ crystal are shown in Fig. 23. A metallic behavior is inferred from the T dependence of ρ without any anomaly in the low- T data. The residual resistivity is $\rho(T = 1.8 \text{ K}) = 11.8 \mu\Omega \text{ cm}$ and $\text{RRR} \approx 9$, demonstrating the high quality of the crystal. An analysis of the $\rho(T)$ data for $1.8 \text{ K} \leq T \leq 50 \text{ K}$ by Eq. (1) gives $\rho_0 = 11.81(2) \mu\Omega \text{ cm}$ and $A = 3.72 \times 10^{-3} \mu\Omega \text{ cm/K}^2$. The fit is shown by the red curve in the inset of Fig. 23. The Kadowaki-Woods ratio $R_{\text{KW}} = A/\gamma^2 = 0.17 \times 10^{-5} \mu\Omega \text{ cm}/(\text{mJ/mol K}^2)$. The $\rho(T)$ data were fitted by Eqs. (2) for $1.8 \text{ K} \leq T \leq 300 \text{ K}$ using the analytic Padé approximant function [54], as shown by red curve in Fig. 23. The fit parameters are summarized in Table II.

VII. PHYSICAL PROPERTIES OF $\text{Ba}_{0.78}\text{K}_{0.22}\text{Co}_2\text{As}_2$

The $\text{Ba}_{0.78}\text{K}_{0.22}\text{Co}_2\text{As}_2$ crystals discussed in this section were grown in Sn flux.

The $\chi(T) \equiv M(T)/H$ data of a $\text{Ba}_{0.78}\text{K}_{0.22}\text{Co}_2\text{As}_2$ crystal measured in $H = 3.0 \text{ T}$ are shown in Fig. 24. Here again no superconductivity is observed at $T \geq 1.8 \text{ K}$. A comparison of the χ of $\text{Ba}_{0.78}\text{K}_{0.22}\text{Co}_2\text{As}_2$ with that of Sn-flux-grown

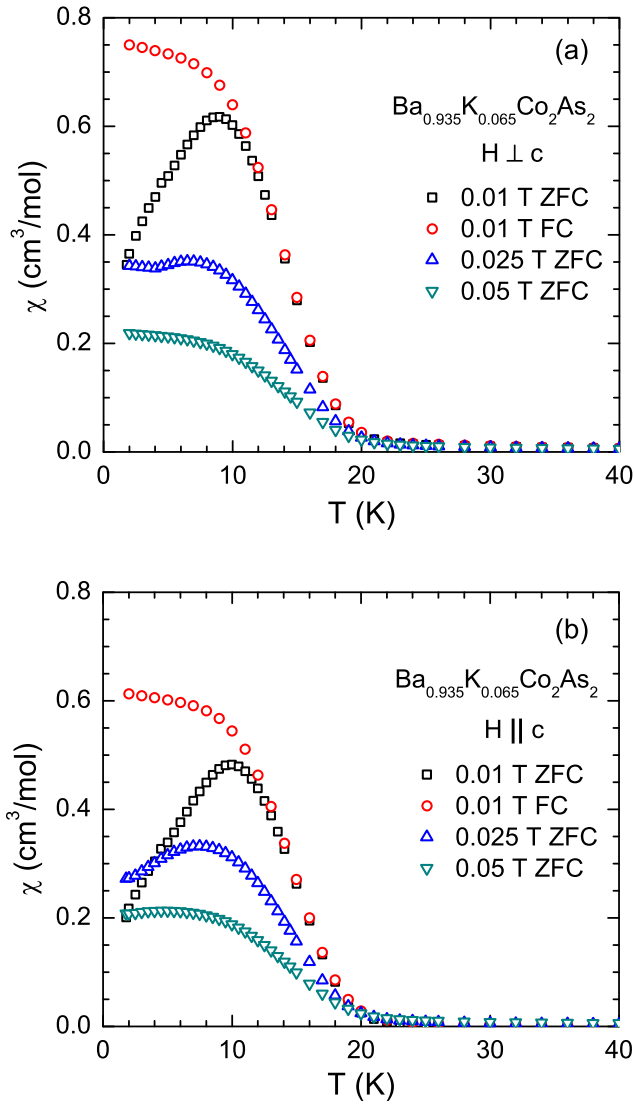


FIG. 19. (Color online) Zero-field-cooled magnetic susceptibility χ of a $\text{Ba}_{0.935}\text{K}_{0.065}\text{Co}_2\text{As}_2$ single crystal grown in CoAs flux as a function of temperature T in the temperature range 1.8–40 K measured in low magnetic fields H applied in the ab plane ($\chi_{ab}, H \perp c$) and (b) along the c axis ($\chi_c, H \parallel c$).

BaCo_2As_2 is shown in Fig. 24(a). It is seen that the magnitude of χ decreases with K doping. An expanded view of $\chi(T)$ of $\text{Ba}_{0.78}\text{K}_{0.22}\text{Co}_2\text{As}_2$ is shown in Fig. 24(b), which shows a broad peak in χ near 100 K. A similar feature has been observed in the T dependence of χ of SrCo_2As_2 [40].

The $M(H)$ isotherms of a $\text{Ba}_{0.78}\text{K}_{0.22}\text{Co}_2\text{As}_2$ crystal at eight temperatures are shown in Fig. 25. These are similar to those of a Sn-flux-grown BaCo_2As_2 crystal shown in Fig. 9. In particular, the $M(H)$ isotherms are almost linear in H at all temperatures. The high-field $M(H)$ data were fitted by Eq. (9) to obtain values of χ . These are plotted as stars in Fig. 24(b) and are seen to be in good agreement with the M/H data. The magnetic behavior of this $\text{Ba}_{0.78}\text{K}_{0.22}\text{Co}_2\text{As}_2$ crystal is therefore much different from those of $\text{Ba}_{0.946}\text{K}_{0.054}\text{Co}_2\text{As}_2$ and $\text{Ba}_{0.935}\text{K}_{0.065}\text{Co}_2\text{As}_2$ grown out of CoAs flux discussed

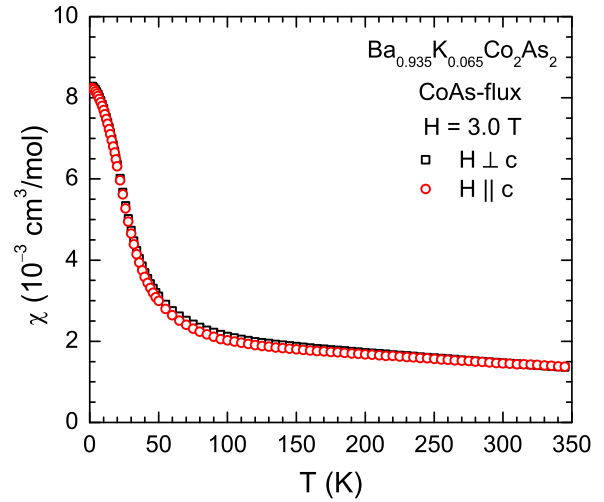


FIG. 20. (Color online) Zero-field-cooled magnetic susceptibility χ of a $\text{Ba}_{0.935}\text{K}_{0.065}\text{Co}_2\text{As}_2$ single crystal grown in CoAs flux as a function of temperature T in the temperature range 1.8–300 K measured in a magnetic field $H = 3.0$ T applied along the c axis ($\chi_c, H \parallel c$) and in the ab plane ($\chi_{ab}, H \perp c$).

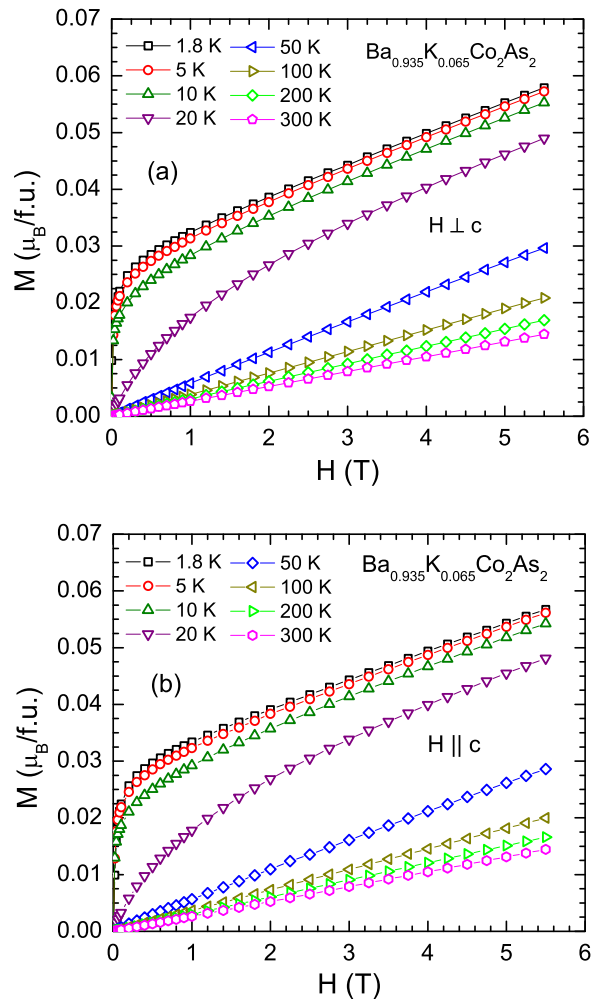


FIG. 21. (Color online) Magnetization M versus field H of a $\text{Ba}_{0.935}\text{K}_{0.065}\text{Co}_2\text{As}_2$ crystal grown in CoAs flux and measured at the indicated temperatures for H applied (a) in the ab plane ($M_{ab}, H \perp c$) and (b) along the c axis ($M_c, H \parallel c$).

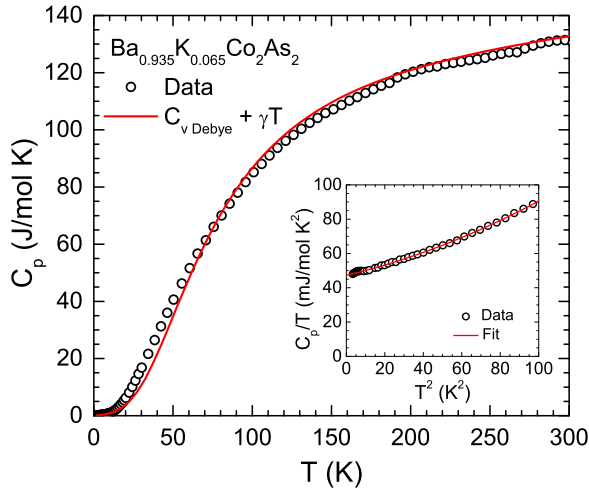


FIG. 22. (Color online) Heat capacity C_p of a $\text{Ba}_{0.935}\text{K}_{0.065}\text{Co}_2\text{As}_2$ crystal grown in CoAs flux as a function of temperature T measured in zero magnetic field. The solid curve is a fit of the C_p/T data by the sum of the contributions from the Debye lattice heat capacity $C_{v,\text{Debye}}(T)$ and electronic heat capacity γT according to Eqs. (6). Inset: C_p/T vs T^2 plot below 10 K; the red curve represents the fit of the C_p/T data by Eq. (3) for $1.8 \text{ K} \leq T \leq 10 \text{ K}$.

above. In particular, no FM-like behaviors are observed in the $M(H)$ isotherms in Fig. 25.

The $C_p(T)$ data of a $\text{Ba}_{0.78}\text{K}_{0.22}\text{Co}_2\text{As}_2$ crystal are shown in Fig. 26. No anomaly is observed in $C_p(T)$ except for an extrinsic weak kink near 3.8 K due to the superconductivity of a very small amount of Sn flux attached to the crystal. A fit of the low- T $C_p(T)/T$ versus T^2 data in the inset of Fig. 22 for $4.0 \text{ K} \leq T \leq 10 \text{ K}$ by Eq. (3) gives the γ , β , and δ values listed in Table III. The fit is shown as the red curve in the inset of Fig. 22. The values of $D(E_F)$ obtained from γ and Θ_D obtained from β are also listed in Table III.

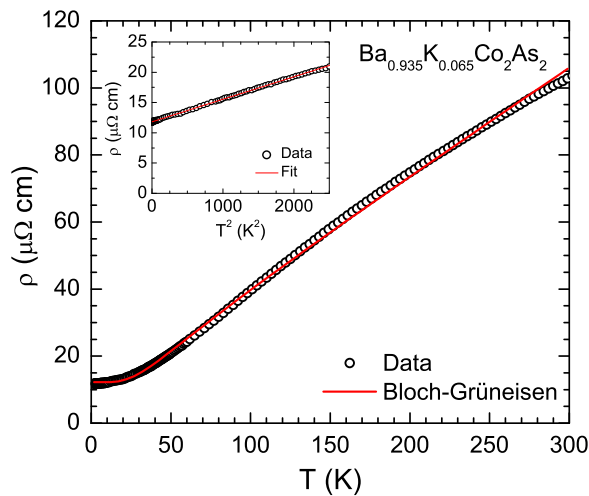


FIG. 23. (Color online) In-plane electrical resistivity ρ of a $\text{Ba}_{0.935}\text{K}_{0.065}\text{Co}_2\text{As}_2$ crystal grown in CoAs flux as a function of temperature T measured in zero magnetic field. The red solid curve is the fit by the Bloch-Grüneisen model. Inset: ρ vs T^2 plots below 50 K; the solid line is the fit by $\rho = \rho_0 + AT^2$ for $1.8 \text{ K} \leq T \leq 50 \text{ K}$.

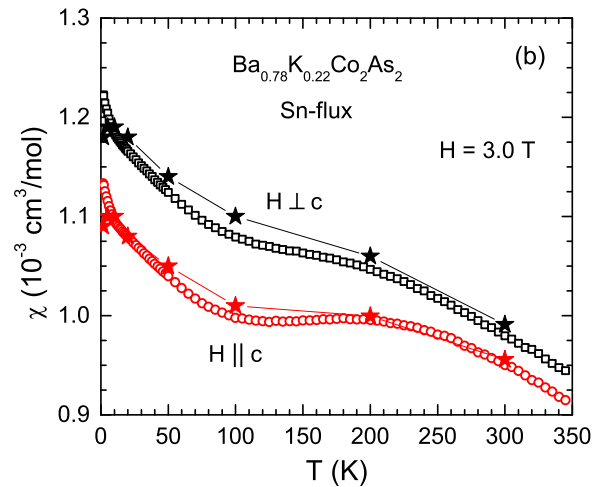
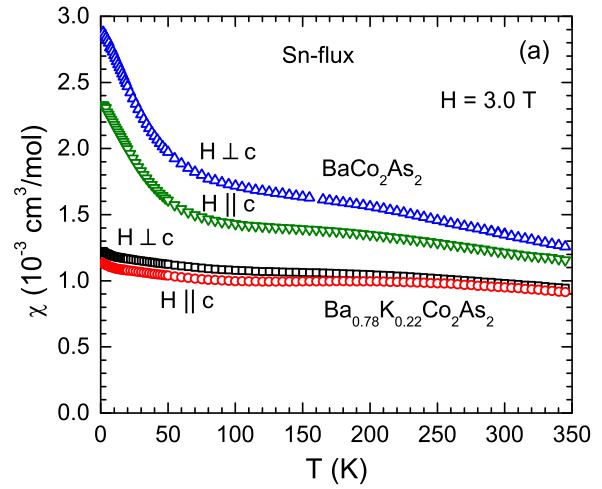


FIG. 24. (Color online) (a) Zero-field-cooled magnetic susceptibility $\chi \equiv M/H$ of a $\text{Ba}_{0.78}\text{K}_{0.22}\text{Co}_2\text{As}_2$ single crystal grown in Sn flux versus temperature T in the T range 1.8 K–300 K measured in a magnetic field $H = 3.0 \text{ T}$ applied along the c axis ($\chi_c, H \parallel c$) and in the ab plane ($\chi_{ab}, H \perp c$). Also shown are the $\chi(T)$ data of a Sn-flux-grown BaCo_2As_2 crystal from Fig. 7. (b) The $\chi(T)$ of $\text{Ba}_{0.78}\text{K}_{0.22}\text{Co}_2\text{As}_2$ on an expanded scale. The stars of respective colors represent the χ obtained from fitting $M(H)$ isotherm data by Eq. (9). The solid lines connecting the stars are guides to the eye.

The in-plane $\rho(T)$ data of a $\text{Ba}_{0.78}\text{K}_{0.22}\text{Co}_2\text{As}_2$ crystal in $H = 0$ are shown in Fig. 27. A metallic behavior is demonstrated by the magnitude and T dependence of ρ . A sharp drop in ρ below $\sim 3.8 \text{ K}$ is due to the superconductivity of a small amount of Sn flux attached to the crystal. No other anomaly is seen in the $\rho(T)$ data. The residual resistivity $\rho(T = 3.8 \text{ K}) = 22.3 \mu\Omega \text{ cm}$ and RRR = 5.4. A fit of the ρ versus T^2 data in the inset of Fig. 27 for $10 \text{ K} \leq T \leq 30 \text{ K}$ by Eq. (1) gives $\rho_0 = 22.07(3) \mu\Omega \text{ cm}$ and $A = 6.06(5) \times 10^{-3} \mu\Omega \text{ cm}/\text{K}^2$. The fit is shown as the red straight line in the inset. The Kadowaki-Woods ratio $R_{\text{KW}} = A/\gamma^2 = 0.69 \times 10^{-5} \mu\Omega \text{ cm}/(\text{mJ}/\text{mol K}^2)$. The $\rho(T)$ data were further fitted by the Bloch-Grüneisen model in Eqs. (2) for $4.0 \text{ K} \leq T \leq 300 \text{ K}$ using the analytic Padé approximant function [54], as shown by the red curve in Fig. 27. A fit of the $\rho(T)$ data

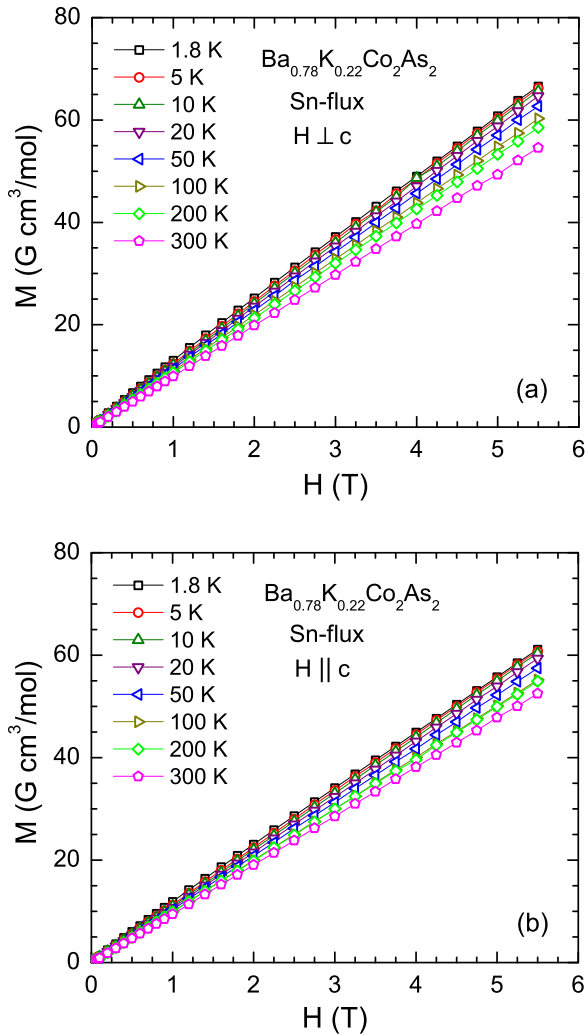


FIG. 25. (Color online) Isothermal magnetization M of a $\text{Ba}_{0.78}\text{K}_{0.22}\text{Co}_2\text{As}_2$ crystal grown in Sn flux as a function of applied magnetic field H measured at the indicated temperatures for H applied (a) in the ab plane ($M_{ab}, H \perp c$) and (b) along the c axis ($M_c, H \parallel c$).

between 10 and 30 K by $\rho(T) = \rho_0 + AT^2$ is shown in the inset of Fig. 27. The fit parameters are summarized in Table II.

VIII. SUMMARY

The powder x-ray and neutron diffraction studies of BaCo_2As_2 , $\text{Ba}_{0.94}\text{K}_{0.06}\text{Co}_2\text{As}_2$ and $\text{Ba}_{0.78}\text{K}_{0.22}\text{Co}_2\text{As}_2$ at room temperature showed that the three compositions crystallize in the same body-centered tetragonal ThCr_2Si_2 -type structure. Additional neutron powder diffraction data for BaCo_2As_2 were obtained down to 10 K, which revealed no evidence for temperature-induced lattice distortions. On the other hand, these data revealed that the tetragonal c axis lattice parameter shows negative thermal expansion over this T range, whereas the in-plane a axis parameter and the unit cell volume show normal expansion on increasing the temperature. The T -dependent behaviors of a and c are similar to those found previously for SrCo_2As_2 [40].

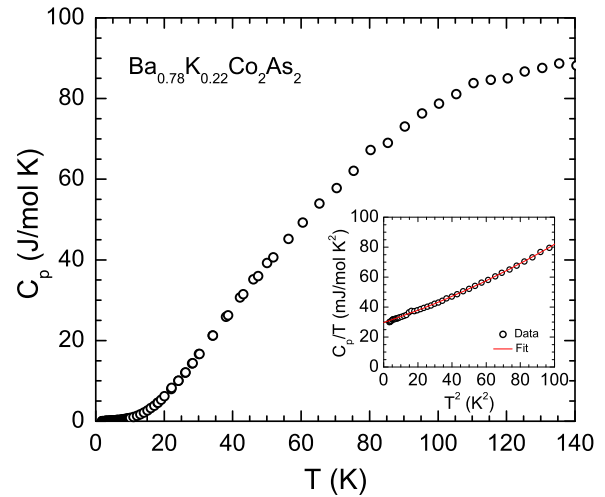


FIG. 26. (Color online) Heat capacity C_p of a $\text{Ba}_{0.78}\text{K}_{0.22}\text{Co}_2\text{As}_2$ crystal grown in Sn flux as a function of temperature T measured in zero magnetic field. Inset: C_p/T versus T^2 below 10 K; the red curve represents the fit of C_p/T data by Eq. (3) for $4 \text{ K} \leq T \leq 10 \text{ K}$.

The electrical resistivity data for all three compounds exhibit metallic character and a T^2 temperature dependence at low temperatures indicative of a Fermi-liquid ground state. The heat capacity and electrical resistivity data show no evidence for any phase transitions between 2 K and 300 K for BaCo_2As_2 . Our magnetization and magnetic susceptibility data for crystals from two batches of $\text{Ba}_{0.94}\text{K}_{0.06}\text{Co}_2\text{As}_2$ crystals grown in CoAs self-flux showed weak ferromagnetism below ~ 10 K with small saturation moments of ≈ 0.007 and $0.03 \mu_B/\text{f.u.}$, respectively, whereas a crystal of $\text{Ba}_{0.78}\text{K}_{0.22}\text{Co}_2\text{As}_2$ grown in Sn flux showed no such behavior. This nonmonotonic behavior of the weak ferromagnetism versus K content may be associated with the different fluxes used to grow the crystals

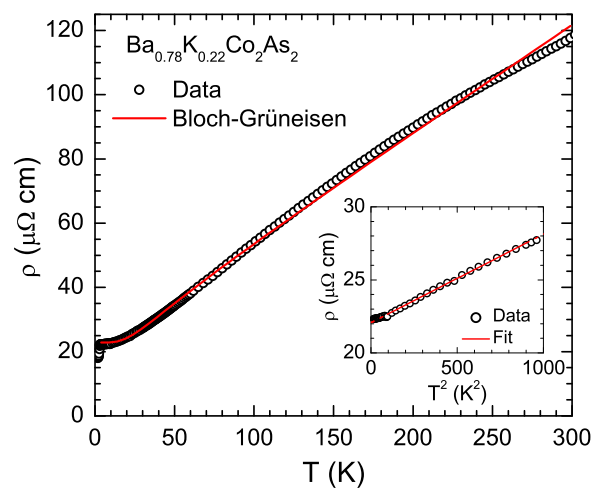


FIG. 27. (Color online) In-plane electrical resistivity ρ of a $\text{Ba}_{0.78}\text{K}_{0.22}\text{Co}_2\text{As}_2$ single crystal grown in Sn flux as a function of temperature T measured in zero magnetic field. The red solid curve is the fit by the Bloch-Grüneisen model. Inset: ρ vs T^2 for $T < 32$ K; the straight red line is a fit of the data by $\rho = \rho_0 + AT^2$ for $10.0 \text{ K} \leq T \leq 30 \text{ K}$.

with the 6% and 22% potassium concentrations. No evidence for bulk superconductivity was found for any of the crystals above 1.8 K.

The low-temperature heat capacity data for the three compounds show similar and rather large Sommerfeld electronic heat capacity coefficients, indicating large densities of states at the Fermi energy $\mathcal{D}(E_F)$. We infer that the factor of two enhancement of this $\mathcal{D}(E_F)$ above the band-structure value arises from a combination of many-body electron-electron effects that renormalize the band structure (decreasing the band widths) and the effect of the electron-phonon interaction. For BaCo_2As_2 , the former enhancement was previously estimated to be $m^*/m_{\text{band}} \approx 1.4$ [43], which we use to compute the latter contribution to be ≈ 1.6 that in turn yields the electron-phonon coupling constant $\lambda_{\text{el-ph}} \approx 0.6$. This value of $\lambda_{\text{el-ph}}$ is similar to the value of 0.76 obtained from *ab initio* calculations for BaNi_2As_2 [60].

Our band-structure value for the density of states at the Fermi energy for BaCo_2As_2 , $\mathcal{D}_{\text{band}}(E_F) = 8.23$ states/(eV f.u.) for both spin directions, is similar to the value of 8.5 states/(eV f.u.) for both spin directions found in Ref. [39]. These values are sufficiently large that on the basis of an itinerant magnetism picture one expects BaCo_2As_2 to be an itinerant ferromagnet [39], which is not observed. To explain the lack of ferromagnetism, Sefat *et al.* suggested that BaCo_2As_2 is close to a quantum critical point where quantum critical fluctuations destabilize long-range ferromagnetic order [39].

The T dependencies of our intrinsic χ and ^{75}As -NMR shift data for CoAs-flux-grown single crystals of BaCo_2As_2 are the same, indicating that the T dependencies of χ_{ab} and χ_c are intrinsic to BaCo_2As_2 . These T dependencies are similar to those reported previously for $\chi(T)$ of a BaCo_2As_2 crystal by Sefat *et al.* [39], but their magnitude of χ is much larger than ours and they reported an anisotropy with $\chi_{ab} < \chi_c$ that is opposite to ours. As discussed in Ref. 63 of Ref. [40], these differences between our $\chi(T)$ results and those of Ref. [39] likely result from the presence of ferromagnetic impurities in their crystal combined with their low measurement field of 0.1 T.

We find no clear evidence from our $M(H)$ isotherms or our $\chi(T)$ data that BaCo_2As_2 is near a ferromagnetic quantum critical point as suggested in Ref. [39]. The ^{75}As -NMR dynamics we observe can be explained by stripe-type AFM fluctuations, FM fluctuations, or a combination of these. If a material is near a quantum critical point, one might expect that the properties are sensitive to doping. However, our crystals of K-doped $\text{Ba}_{0.78}\text{K}_{0.22}\text{Co}_2\text{As}_2$ grown in Sn flux exhibit magnetic properties similar to those of undoped BaCo_2As_2 . On the other hand, two batches of $\text{Ba}_{0.94}\text{K}_{0.06}\text{Co}_2\text{As}_2$ crystals grown in CoAs self-flux show weak ferromagnetism with small ordered moments of ≈ 0.007 and $0.03\mu_B/\text{f.u.}$, respectively. We speculate that these differences between the magnetic behaviors of the $\text{Ba}_{0.94}\text{K}_{0.06}\text{Co}_2\text{As}_2$ and $\text{Ba}_{0.78}\text{K}_{0.22}\text{Co}_2\text{As}_2$ crystals are due to magnetic defect level differences arising

from growing the former crystals in CoAs flux and quenching from 1180 °C, and growing the latter crystals in Sn flux and quenching from the much lower temperature of 600 °C. Additionally, as pointed out in Sec. II, the latter crystals contain about 2.0 mol% Sn within the structure.

Interestingly, this conundrum between FM and AFM interactions and ordering was present in the early days of the iron-arsenide high- T_c field in 2008, where the band-structure density of states $\mathcal{D}_{\text{band}}(E_F)$ calculated for the parent LaFeAsO compound was approximately large enough to lead to itinerant ferromagnetism [72], whereas itinerant antiferromagnetism (a spin-density wave) was observed instead [9], indicating a competition between these magnetic structures. A related observation is that SrCo_2As_2 shows no long-range magnetic ordering but exhibits a broad maximum in $\chi(T)$ at about 115 K typically associated with low-dimensional dynamic AFM correlations [40]. Furthermore, from inelastic neutron scattering measurements SrCo_2As_2 shows rather strong AFM fluctuations at the same stripe wave vector at which the FeAs-based materials exhibit AFM fluctuations and long-range AFM ordering [42], whereas the calculated $\mathcal{D}_{\text{band}}(E_F)$ suggests incipient ferromagnetism [40]. On the other hand, the isostructural compound $\text{CaCo}_{1.86}\text{As}_2$ exhibits dominant FM correlations [34,35,37], and the magnetically ordered state consists of ferromagnetically aligned layers of Co spins in the ab plane that are antiferromagnetically aligned from layer to layer along the c axis to form an A-type AFM structure [36]. It will be interesting to resolve the puzzle of how these conflicting interactions and magnetic structures compete and compromise in the ACo_2As_2 compounds.

Note added. Two papers recently appeared that are related to our work. ^{75}As -NMR data on BaCo_2As_2 were obtained in Ref. [73]. Electronic structure calculations for BaCo_2As_2 were also reported [74].

ACKNOWLEDGMENTS

We thank Abhishek Pandey for experimental assistance and for many helpful discussions. The research at Ames Laboratory was supported by the U.S. Department of Energy, Office of Basic Energy Sciences, Division of Materials Sciences and Engineering. Ames Laboratory is operated for the U.S. Department of Energy by Iowa State University under Contract No. DE-AC02-07CH11358. V.V.O. thanks the Ames Laboratory—USDOE for providing the opportunity to be a visiting scientist at the Laboratory and also thanks the Russian Foundation for Basic Research (Grant No. 12-02-31814) for support. Use of the National Synchrotron Light Source, Brookhaven National Laboratory, was supported by the U.S. Department of Energy, Office of Basic Energy Sciences, under Contract No. DE-AC02-98CH10886. Research at the Spallation Neutron Source at Oak Ridge National Laboratory was sponsored by the Scientific User Facilities Division, Office of Basic Energy Sciences, U.S. Department of Energy.

- [1] M. Rotter, M. Tegel, and D. Johrendt, *Phys. Rev. Lett.* **101**, 107006 (2008).
 [2] G. F. Chen, Z. Li, G. Li, W.-Z. Hu, J. Dong, J. Zhou, X.-D. Zhang, P. Zheng, N.-L. Wang, and J.-L. Luo, *Chin. Phys. Lett.* **25**, 3403 (2008).

- [3] K. Sasmal, B. Lv, B. Lorenz, A. M. Guloy, F. Chen, Y.-Y. Xue, and C. W. Chu, *Phys. Rev. Lett.* **101**, 107007 (2008).
 [4] G. Wu, H. Chen, T. Wu, Y. L. Xie, Y. J. Yan, R. H. Liu, X. F. Wang, J. J. Ying, and X. H. Chen, *J. Phys.: Condens. Matter* **20**, 422201 (2008).

- [5] A. S. Sefat, R. Jin, M. A. McGuire, B. C. Sales, D. J. Singh, and D. Mandrus, *Phys. Rev. Lett.* **101**, 117004 (2008).
- [6] M. S. Torikachvili, S. L. Bud'ko, N. Ni, and P. C. Canfield, *Phys. Rev. Lett.* **101**, 057006 (2008).
- [7] K. Ishida, Y. Nakai, and H. Hosono, *J. Phys. Soc. Jpn.* **78**, 062001 (2009).
- [8] P. L. Alireza, Y. T. C. Ko, J. Gillett, C. M. Petrone, J. M. Cole, G. G. Lonzarich, and S. E. Sebastian, *J. Phys.: Condens. Matter* **21**, 012208 (2009).
- [9] D. C. Johnston, *Adv. Phys.* **59**, 803 (2010).
- [10] P. C. Canfield and S. L. Bud'ko, *Annu. Rev. Condens. Matter Phys.* **1**, 27 (2010).
- [11] D. Mandrus, A. S. Sefat, M. A. McGuire, and B. C. Sales, *Chem. Mater.* **22**, 715 (2010).
- [12] A. Damascelli, Z. Hussain, and Z.-X. Shen, *Rev. Mod. Phys.* **75**, 473 (2003).
- [13] P. A. Lee, N. Nagaosa, and X.-G. Wen, *Rev. Mod. Phys.* **78**, 17 (2006).
- [14] Y. Singh, A. Ellern, and D. C. Johnston, *Phys. Rev. B* **79**, 094519 (2009).
- [15] J. An, A. S. Sefat, D. J. Singh, and M. H. Du, *Phys. Rev. B* **79**, 075120 (2009).
- [16] Y. Singh, M. A. Green, Q. Huang, A. Kreyssig, R. J. McQueeney, D. C. Johnston, and A. I. Goldman, *Phys. Rev. B* **80**, 100403 (2009).
- [17] D. C. Johnston, R. J. McQueeney, B. Lake, A. Honecker, M. E. Zhitomirsky, R. Nath, Y. Furukawa, V. P. Antropov, and Y. Singh, *Phys. Rev. B* **84**, 094445 (2011).
- [18] A. Pandey, R. S. Dhaka, J. Lamsal, Y. Lee, V. K. Anand, A. Kreyssig, T. W. Heitmann, R. J. McQueeney, A. I. Goldman, B. N. Harmon, A. Kaminski, and D. C. Johnston, *Phys. Rev. Lett.* **108**, 087005 (2012).
- [19] J. Lamsal, G. S. Tucker, T. W. Heitmann, A. Kreyssig, A. Jesche, A. Pandey, W. Tian, R. J. McQueeney, D. C. Johnston, and A. I. Goldman, *Phys. Rev. B* **87**, 144418 (2013).
- [20] J.-K. Bao, H. Jiang, Y.-L. Sun, W.-H. Jiao, C.-Y. Shen, H.-J. Guo, Y. Chen, C.-M. Feng, H.-Q. Yuan, Z.-A. Xu, G.-H. Cao, R. Sasaki, T. Tanaka, K. Matsubayashi, and Y. Uwatoko, *Phys. Rev. B* **85**, 144523 (2012).
- [21] A. Pandey, B. G. Ueland, S. Yeninas, A. Kreyssig, A. Sapkota, Y. Zhao, J. S. Helton, J. W. Lynn, R. J. McQueeney, Y. Furukawa, A. I. Goldman, and D. C. Johnston, *Phys. Rev. Lett.* **111**, 047001 (2013).
- [22] V. K. Anand, P. K. Perera, A. Pandey, R. J. Goetsch, A. Kreyssig, and D. C. Johnston, *Phys. Rev. B* **85**, 214523 (2012).
- [23] B. Cheng, B. F. Hu, R. Y. Chen, G. Xu, P. Zheng, J. L. Luo, and N. L. Wang, *Phys. Rev. B* **86**, 134503 (2012).
- [24] V. K. Anand and D. C. Johnston, *Phys. Rev. B* **86**, 214501 (2012).
- [25] V. K. Anand, H. Kim, M. A. Tanatar, R. Prozorov, and D. C. Johnston, *Phys. Rev. B* **87**, 224510 (2013).
- [26] V. K. Anand, H. Kim, M. A. Tanatar, R. Prozorov, and D. C. Johnston, *J. Phys.: Condens. Matter* (to be published); [arXiv:1403.4595](https://arxiv.org/abs/1403.4595).
- [27] H. Fujii and A. Sato, *Phys. Rev. B* **79**, 224522 (2009).
- [28] A. Leithe-Jasper, W. Schnelle, C. Geibel, and H. Rosner, *Phys. Rev. Lett.* **101**, 207004 (2008).
- [29] S. Nandi, M. G. Kim, A. Kreyssig, R. M. Fernandes, D. K. Pratt, A. Thaler, N. Ni, S. L. Bud'ko, P. C. Canfield, J. Schmalian, R. J. McQueeney, and A. I. Goldman, *Phys. Rev. Lett.* **104**, 057006 (2010).
- [30] R. Hu, S. L. Bud'ko, W. E. Straszheim, and P. C. Canfield, *Phys. Rev. B* **83**, 094520 (2011).
- [31] J. Zhao, D.-X. Yao, S. Li, T. Hong, Y. Chen, S. Chang, W. Ratchiff, II, J. W. Lynn, H. A. Mook, G. F. Chen, J. L. Luo, N. L. Wang, E. W. Carlson, J. Hu, and P. Dai, *Phys. Rev. Lett.* **101**, 167203 (2008).
- [32] R. A. Ewings, T. G. Perring, J. Gillett, S. D. Das, S. E. Sebastian, A. E. Taylor, T. Guidi, and A. T. Boothroyd, *Phys. Rev. B* **83**, 214519 (2011).
- [33] M. Pfisterer and G. Nagorsen, *Z. Naturforsch.* **35b**, 703 (1980); **38b**, 811 (1983).
- [34] B. Cheng, B. F. Hu, R. H. Yuan, T. Dong, A. F. Fang, Z. G. Chen, G. Xu, Y. G. Shi, P. Zheng, J. L. Luo, and N. L. Wang, *Phys. Rev. B* **85**, 144426 (2012).
- [35] J. J. Ying, Y. J. Yan, A. F. Wang, Z. J. Xiang, P. Cheng, G. J. Ye, and X. H. Chen, *Phys. Rev. B* **85**, 214414 (2012).
- [36] D. G. Quirinale, V. K. Anand, M. G. Kim, A. Pandey, A. Huq, P. W. Stephens, T. W. Heitmann, A. Kreyssig, R. J. McQueeney, D. C. Johnston, and A. I. Goldman, *Phys. Rev. B* **88**, 174420 (2013).
- [37] V. K. Anand, R. S. Dhaka, Y. Lee, B. N. Harmon, A. Kaminski, and D. C. Johnston, *Phys. Rev. B* **89**, 214409 (2014).
- [38] Y. Furukawa (private communication).
- [39] A. S. Sefat, D. J. Singh, R. Jin, M. A. McGuire, B. C. Sales, and D. Mandrus, *Phys. Rev. B* **79**, 024512 (2009).
- [40] A. Pandey, D. G. Quirinale, W. Jayasekara, A. Sapkota, M. G. Kim, R. S. Dhaka, Y. Lee, T. W. Heitmann, P. W. Stephens, V. Ogloblichev, A. Kreyssig, R. J. McQueeney, A. I. Goldman, A. Kaminski, B. N. Harmon, Y. Furukawa, and D. C. Johnston, *Phys. Rev. B* **88**, 014526 (2013).
- [41] D. C. Johnston, in *Handbook of Magnetic Materials*, edited by K. H. J. Buschow (Elsevier Science, Amsterdam, 1997), Vol. 10, Chap. 1, pp. 1–237.
- [42] W. Jayasekara, Y. Lee, A. Pandey, G. S. Tucker, A. Sapkota, J. Lamsal, S. Calder, D. L. Abernathy, J. L. Niedziela, B. N. Harmon, A. Kreyssig, D. Vaknin, D. C. Johnston, A. I. Goldman, and R. J. McQueeney, *Phys. Rev. Lett.* **111**, 157001 (2013).
- [43] N. Xu, P. Richard, A. van Roekeghem, P. Zhang, H. Miao, W.-L. Zhang, T. Qian, M. Ferrero, A. S. Sefat, S. Biermann, and H. Ding, *Phys. Rev. X* **3**, 011006 (2013).
- [44] R. S. Dhaka, Y. Lee, V. K. Anand, D. C. Johnston, B. N. Harmon, and A. Kaminski, *Phys. Rev. B* **87**, 214516 (2013).
- [45] N. Ni, S. L. Bud'ko, A. Kreyssig, S. Nandi, G. E. Rustan, A. I. Goldman, S. Gupta, J. D. Corbett, A. Kracher, and P. C. Canfield, *Phys. Rev. B* **78**, 014507 (2008).
- [46] A. Huq, J. P. Hodges, O. Gourdon, and L. Heroux, *Z. Kristallogr. Proc.* **1**, 127 (2011).
- [47] P. Blaha, K. Schwarz, G. K. H. Madsen, D. Kvasnick, and J. Luitz, WIEN2K, an Augmented Plane Wave + Local Orbitals Program for Calculation Crystal Properties (Technical Universität Wien, Austria, 2001).
- [48] J. P. Perdew and Y. Wang, *Phys. Rev. B* **45**, 13244 (1992).
- [49] J. Rodríguez-Carvajal, *Physica B* **192**, 55 (1993); see also www.ill.eu/sites/fullprof/.
- [50] A. C. Larson and R. B. Von Dreele, Tech. Rep. LAUR 86-748, Los Alamos National Laboratory, 2004.
- [51] B. H. Toby, *J. Appl. Cryst.* **34**, 210 (2001).
- [52] K. Kadowaki and S. B. Woods, *Solid State Commun.* **58**, 507 (1986).

- [53] F. J. Blatt, *Physics of Electronic Conduction in Solids* (McGraw-Hill, New York, 1968).
- [54] R. J. Goetsch, V. K. Anand, A. Pandey, and D. C. Johnston, *Phys. Rev. B* **85**, 054517 (2012).
- [55] C. Kittel, *Introduction to Solid State Physics*, 8th ed. (Wiley, New York, 2005).
- [56] E. S. R. Gopal, *Specific Heats at Low Temperatures* (Plenum, New York, 1966).
- [57] G. F. Chen, Z. Li, J. Dong, G. Li, W. Z. Hu, X. D. Zhang, X. H. Song, P. Zheng, N. L. Wang, and J. L. Luo, *Phys. Rev. B* **78**, 224512 (2008).
- [58] F. Ronning, T. Klimczuk, E. D. Bauer, H. Volz, and J. D. Thompson, *J. Phys.: Condens. Matter* **20**, 322201 (2008).
- [59] C. R. Rotundu, B. Freelon, T. R. Forrest, S. D. Wilson, P. N. Valdivia, G. Pinuellas, A. Kim, J.-W. Kim, Z. Islam, E. Bourret-Courchesne, N. E. Phillips, and R. J. Birgeneau, *Phys. Rev. B* **82**, 144525 (2010).
- [60] A. Subedi and D. J. Singh, *Phys. Rev. B* **78**, 132511 (2008).
- [61] B. Saparov, J. E. Mitchell, and A. S. Sefat, *Supercond. Sci. Technol.* **25**, 084016 (2012).
- [62] T. Siegrist and F. Hulliger, *J. Solid State Chem.* **63**, 23 (1986).
- [63] K. Kitagawa, N. Katayama, K. Ohgushi, and M. Takigawa, *J. Phys. Soc. Jpn.* **78**, 063706 (2009).
- [64] K. Kitagawa, N. Katayama, K. Ohgushi, M. Yoshida, and M. Takigawa, *J. Phys. Soc. Jpn.* **77**, 114709 (2008).
- [65] S.-H. Baek, N. J. Curro, T. Klimczuk, E. D. Bauer, F. Ronning, and J. D. Thompson, *Phys. Rev. B* **79**, 052504 (2009).
- [66] F. L. Ning, K. Ahilan, T. Imai, A. S. Sefat, R. Jin, M. A. McGuire, B. C. Sales, and D. Mandrus, *Phys. Rev. B* **79**, 140506(R) (2009).
- [67] R. R. Urbano, E. L. Green, W. G. Moulton, A. P. Reyes, P. L. Kuhns, E. M. Bittar, C. Adriano, T. M. Garitezi, L. Bufaiçal, and P. G. Pagliuso, *Phys. Rev. Lett.* **105**, 107001 (2010).
- [68] F. L. Ning, K. Ahilan, T. Imai, A. S. Sefat, M. A. McGuire, B. C. Sales, D. Mandrus, P. Cheng, B. Shen, and H.-H. Wen, *Phys. Rev. Lett.* **104**, 037001 (2010).
- [69] S. Kitagawa, Y. Nakai, T. Iye, K. Ishida, Y. Kamihara, M. Hirano, and H. Hosono, *Phys. Rev. B* **81**, 212502 (2010).
- [70] M. Hirano, Y. Yamada, T. Saito, R. Nagashima, T. Konishi, T. Toriyama, Y. Ohta, H. Fukazawa, Y. Kohori, Y. Furukawa, K. Kihou, C.-H. Lee, A. Iyo, and H. Eisaki, *J. Phys. Soc. Jpn.* **81**, 054704 (2012).
- [71] Y. Nakai, S. Kitagawa, T. Iye, K. Ishida, Y. Kamihara, M. Hirano, and H. Hosono, *Phys. Rev. B* **85**, 134408 (2012).
- [72] D. J. Singh and M.-H. Du, *Phys. Rev. Lett.* **100**, 237003 (2008).
- [73] K. Ahilan, T. Imai, A. S. Sefat, and F. L. Ning, *Phys. Rev. B* **90**, 014520 (2014).
- [74] A. van Roekeghem, T. Ayrál, J. M. Tomczak, M. Casula, N. Xu, H. Ding, M. Ferrero, O. Parcollet, H. Jiang, and S. Biermann, [arXiv:1408.3136](https://arxiv.org/abs/1408.3136).