

Electronic structures, magnetism, and phonon spectra in the metallic cubic perovskite BaOsO₃Myung-Chul Jung¹ and Kwan-Woo Lee^{1,2,*}¹*Department of Applied Physics, Graduate School, Korea University, Sejong 339-700, Korea*²*Department of Display and Semiconductor Physics, Korea University, Sejong 339-700, Korea*

(Received 6 March 2014; revised manuscript received 26 June 2014; published 17 July 2014)

By using *ab initio* calculations, we investigated a cubic perovskite BaOsO₃ and a few related compounds that have been synthesized recently and formally have a metallic d^4 configuration. In BaOsO₃, which shows obvious three-dimensional fermiology, a nonmagnetism is induced by a large spin-orbit coupling (SOC), which is precisely equal to an exchange splitting ~ 0.4 eV of the t_{2g} manifold. However, the inclusion of on-site Coulomb repulsion as small as $U^c \approx 1.2$ eV, only 1/3 of the t_{2g} bandwidth, leads to the emergence of a spin-ordered moment, indicating that this system is on the verge of magnetism. In contrast to BaOsO₃, our calculations suggest that the ground state of an orthorhombic CaOsO₃ is a magnetically ordered state due to the reduction of the strength of SOC (about a half of that of BaOsO₃) driven by the structure distortion, although the magnetization energy is only a few tenths of meV. Furthermore, in the cubic BaOsO₃ and BaRuO₃, our full-phonon calculations show several unstable modes, requiring further research.

DOI: [10.1103/PhysRevB.90.045120](https://doi.org/10.1103/PhysRevB.90.045120)

PACS number(s): 71.20.Be, 71.20.Dg, 75.50.Cc, 71.18.+y

I. INTRODUCTION

Transition-metal oxides show abundant phenomena involved in the interplay among charge, spin, orbital, and lattice degrees of freedom. Recently, the effects of spin-orbit coupling (SOC) have been intensively investigated in the t_{2g} manifold of an \mathcal{MO}_6 octahedral structure ($\mathcal{M} = 5d$ transition metals) [1–11]. In the atomic limit, a large SOC leads to transforming the t_{2g} manifold into an effective angular momentum of $|\mathcal{L}| = 1$ [1].

A conventional SOC picture says that a d^4 system is a trivial nonmagnetic insulator in the large-SOC limit, which may be suitable for a nonmagnetic (NM) and nonmetallic NaIrO₃ [12]. However, two interesting studies have recently appeared. Khaliullin suggested a van Vleck-type Mott insulating state due to excitations between a singlet $J = 0$ state and triplet $J = 1$ states [10]. Meetei *et al.* proposed unusual Mott insulators of a charge-disproportionated ferromagnetic (FM) $J = 1$ state as well as a FM $J = 2$ state, as including effects of correlation [11]. One may expect different physical phenomena for a metallic $5d^4$ system such as BaOsO₃, which will be focused on in this research.

Several decades ago, Chamberland and Sarkozy synthesized a body-centered cubic KSbO₃-type and a $6H$ hexagonal phase [13,14]. Although no detailed information of the crystal structures and the physical properties is available, the $6H$ hexagonal phase, which is a semiconductor with an activation energy of 0.39 eV, follows the Curie–Weiss behavior above 100 K (no measurement has been done below this temperature). The observed effective moment is $2.81\mu_B$, which is very close to the spin-only value for $S = 1$, indicating negligible effects of SOC for such a large structure distortion [14]. Very recently, Shi *et al.* synthesized a distortion-free cubic perovskite phase by using a technique of high temperature (~ 2000 K) and high pressure (17 GPa) [15]. The resistivity is metallic but indicates a small upturn at 50 K, which may be due to the polycrystallinity of the sample. The heat-capacity

measurement shows a metallic behavior but can be fit well by a linear combination of the Debye and the Einstein models, suggesting a complicated behavior in phonon modes around 300 K [15]. Shi *et al.* also synthesized two orthorhombic, isovalent perovskites CaOsO₃ and SrOsO₃ [15]. CaOsO₃ shows complicated electronic properties, whereas SrOsO₃ has metallic characteristics in both the resistivity and the heat-capacity measurements. The resistivity data of CaOsO₃ is semiconductor like, but $\ln \rho(T)$ does not follow typical models of T^{-1} or $T^{-1/4}$ forms. Also, the heat capacity contains itinerant electronic character even at very low T . It was claimed that CaOsO₃ is near a Mott insulating state [15]. For the magnetic properties, these three osmates commonly follow the Curie–Weiss susceptibility, though no order moment has been observed. At low T , the susceptibility measurements of CaOsO₃ and SrOsO₃ show larger enhancement than that of BaOsO₃, indicating more magnetic tendencies in these two orthorhombic osmates.

Through first-principles calculations, we address the electronic structures and magnetic properties of cubic BaOsO₃, which are substantially affected by SOC, structure distortion, and correlation effects. Cubic BaOsO₃ is compared and contrasted to the orthorhombic CaOsO₃ to investigate the interplay between magnetism, SOC, and structure distortion. In addition, we perform full-phonon calculations to inspect the stability of the cubic phases of isovalent BaOsO₃ and BaRuO₃, which can be synthesized only by a technique of very high pressure and temperature [15,16].

II. CALCULATION METHOD

Our calculations were carried out by using the experiment lattice parameter $a = 4.02573$ Å for BaOsO₃ [15] with two all-electron full-potential codes, FPLO [17] and WIEN2K [18]. Both SOC and correlations have been considered within extensions of the Perdew–Burke–Erznerhof generalized gradient approximation (PBE-GGA) functional [19]. When necessary, we optimized the lattice or internal parameters until the forces were smaller than 1 meV/a.u. (see below). The Brillouin zone was sampled with a very dense k mesh of $24 \times 24 \times 24$ to

*mckwan@korea.ac.kr

check the convergence carefully for this metallic system. In WIEN2K, the basis size was determined by $R_{mt}K_{max} = 7$ and the augmented plane wave (APW) radii of Ba 2.50, Os 2.08, O 1.7 (in units of a.u.).

We also performed linear-response phonon calculations by using QUANTUM ESPRESSO [20] with the PBE-GGA ultrasoft-pseudopotential [19] and the fully relativistic pseudopotential for GGA + SOC. These calculations were carried out with a $4 \times 4 \times 4$ q -mesh, a $24 \times 24 \times 24$ k -mesh, an energy cutoff of 45 Ry, and a charge cutoff of 450 Ry.

III. RESULTS

A. Electronic structure of BaOsO₃

First, we address the electronic structure of nonmagnetic (NM) BaOsO₃, since the ground state is NM when considering SOC. We discuss this in Sec. III C in detail.

The top of Fig. 1 displays the NM band structure of BaOsO₃, excluding SOC. The O p states and the Os t_{2g} manifold extend over the range of -9 to 0.5 eV, relative to the Fermi energy E_F . As clearly visible in the corresponding DOS of Fig. 2(b), E_F pinpoints at the $2/3$ filling of the t_{2g} manifold, which is consistent with the d^4 configuration. The orbital-projected DOS of Fig. 2(a) shows strong hybridization between Os $5d$ and O $2p$ states: roughly $pd\sigma$ states at -9 to -5 eV (0.5 eV to 6 eV), $pd\pi$ states at -7.2 to -3 eV (-3 to 1 eV), and pure O p_π states at -4.2 to -2.8 eV. Here, the numbers inside the parentheses are for each antibonding states, usually denoted the e_g and t_{2g} manifolds, respectively. These bands below 1 eV were fit by the tight-binding Wannier-function technique implemented in FPLO. We obtained two important parameters: $pd\pi$ hopping $t_\pi = 1.32$ eV and direct oxygen-oxygen hopping $t'_\pi = 0.14$ eV [21]. The t'_π value is similar to that of BaRuO₃, but the value of t_π is about 10% larger than that of BaRuO₃ [22], reflecting the wider extension of the $5d$ orbital than the $4d$ orbital. Remarkably, in the band

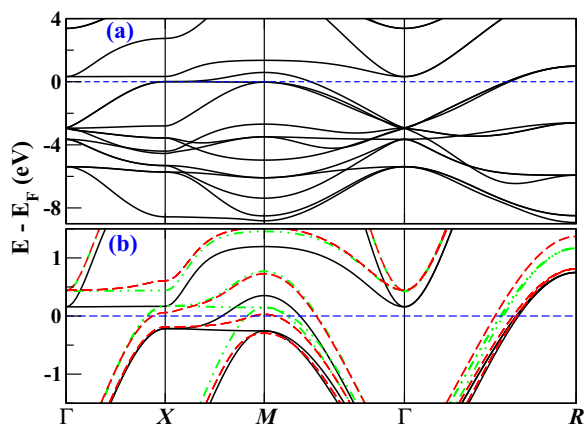


FIG. 1. (Color online) (a) GGA NM band structure of BaOsO₃ in the regime including O $2p$ and Os $5d$ orbitals. The bands above 0.5 eV are the Os e_g manifold. (b) Blowup GGA FM band structure (black solid lines are for spin up and green dot-dashed lines are for spin down), which is overlapped by the GGA + SOC one (red dashed lines). The horizontal dashed lines denote the Fermi energy E_F , which is set to zero.

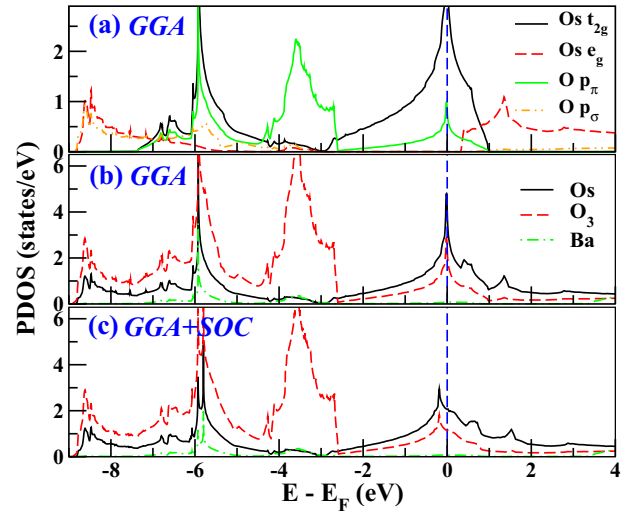


FIG. 2. (Color online) Orbital-projected densities of states (PDOSs) per atom in (a) GGA, and (b) atom PDOSs in GGA and in (c) GGA + SOC of NM BaOsO₃. The corresponding FM DOS is not shown here, since this is very similar to that of NM, except for the positions of E_F at 0.25 eV and -0.15 eV for the spin-up and spin-down channels, respectively, relative to E_F of NM. For NM, the single-spin DOS at E_F is $N(E_F) = 3.38$ states per eV.

structure, there are three nearly dispersionless bands around the M point, in addition to a flat band just above E_F along the Γ - X line, which often appears due to a lack of $dd\delta$ hopping in conventional perovskites [23]. The former three bands lead to sharp peaks at -6 eV, -3.5 eV, and E_F in the DOS of Fig. 2(b), whereas the latter conduction band does not. The van Hove singularity at E_F results from the quite flat band with the character of $pd\pi$ antibonding between Os d_{xz} and O p_z orbitals along the X - M line, which shows a large band mass of $m^* \approx -70$. This van Hove singularity enhances the magnetic instability (see below).

To investigate the effects of SOC, we carried out GGA + SOC calculations. The blowup band structure near E_F denoted by (red) dashed lines in the bottom of Fig. 1 shows a strength of SOC of $\xi \approx 0.4$ eV, as measured near E_F , where SOC affects considerably. This value is largely reduced from the atomic value ~ 1 eV due to the strong p - d hybridization. Including SOC, the flat $pd\pi$ antibonding band becomes more dispersive and shifts down, leading to making the van Hove singularity blunted below E_F , as shown in the PDOS in Fig. 2(c). This system becomes very three dimensional. As a result, the total DOS $N(E_F)$ becomes only a half of that of the GGA NM, suggesting that the magnetic instability is significantly diminished by SOC, as discussed below.

B. Fermiology

The NM Fermi surfaces (FSs) colored with the Fermi velocity are displayed in Fig. 3. FSs of both GGA and GGA + SOC consist of three pieces: two hole-like pieces and a Γ -centered spherical electron-like piece. Neglecting SOC, FSs have more flat regimes in various directions, implying the possibility of a charge or spin fluctuation. The FSs have much lower Fermi velocity in the $\langle 100 \rangle$ directions, reflecting the

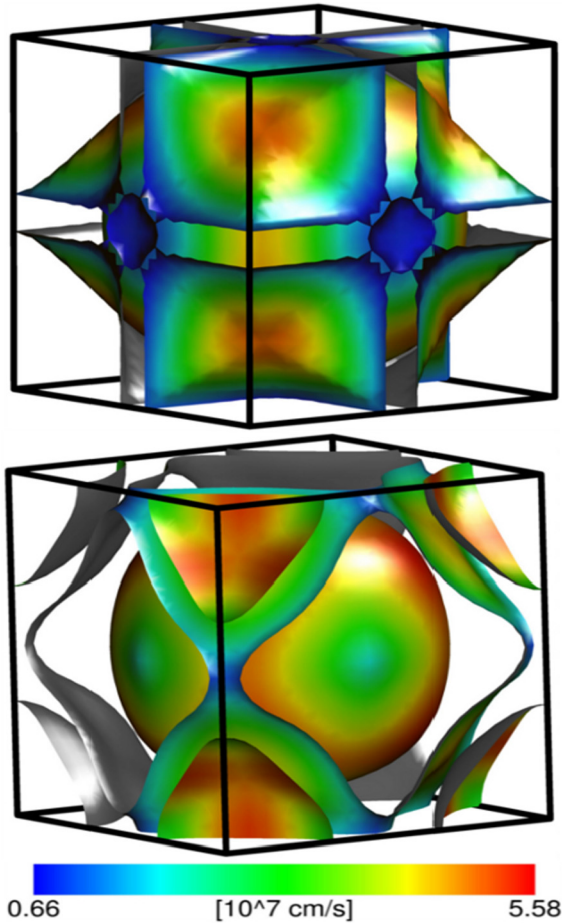


FIG. 3. (Color online) NM Fermi surfaces (FSs) in both GGA (top panel) and GGA + SOC (bottom panel) colored with Fermi velocity, which show three-dimensional character. In the GGA case the R -centered hole FS is not shown for better visualization, since this FS is very similar to that of GGA + SOC, except for a little larger size. The R and M points are at each corner and midway of each side, respectively.

fact that bands at E_F are quite dispersionless in the direction. However, our trials to obtain antiferromagnetic states of three typical types are always converged to NM or FM states in both GGA and GGA + SOC. This would rule out the possibility of a spin fluctuation.

Considering the effects of SOC, the most significant change occurs in the open FS surrounding the Γ -centered sphere, resulting in being an M -centered sandglass-like shape. In the FS, most flat parts disappear, but some survive in faces toward the R point. The R -centered FS is shrunk and becomes nearly spherical with a radius of $\sim 2/3(\pi/a)$, containing ~ 0.31 holes. The Γ -centered spherical FS becomes more isotropic, although nodules appear in the $\langle 100 \rangle$ directions. The radius of the spherical FS decreases by $\sim 10\%$ to compensate for the reduction of the hole-like FSs and attains a radius of $0.9\pi/a$, containing about 0.76 electrons.

Transport properties can be studied with the Fermi velocity and plasma frequency. For the GGA case, the root-mean-square of the Fermi velocity is $v_F^{\text{rms}} = 2.71 \times 10^7$ cm/s, a typical value for a metal. The plasma energy $\hbar\Omega_{p,ii}$, which

is proportional to $v_F^{\text{rms}} \sqrt{N(E_F)}$, is 4.55 eV. The inclusion of SOC increases v_F^{rms} to 3.63×10^7 cm/s, which is consistent with the fact that the bands around E_F become more dispersive. So, the magnitude of the plasma energies is similar to the case neglecting SOC.

C. Magnetic tendencies

Within GGA, FM with a total moment of $0.85\mu_B$ is energetically favored over NM, but the difference in energy is only 11 meV. We performed fixed spin-moment calculations in GGA to investigate the stability of the FM state. We obtained $I = 0.79$ eV, leading to $IN(E_F) \approx 1.2$ with $N(E_F) = 1.53$ states/eV-spin for FM. This value, above unity of the Stoner criterion, is close to that of the cubic FM BaRuO₃ with the similar moment [22].

The bottom of Fig. 1 displays the enlarged FM band structure in GGA, showing that the exchange splitting of the t_{2g} manifold is about 0.4 eV. Remarkably, this value is identical to the strength of SOC, resulting in a transition from FM to NM in this metallic d^4 system. Whenever including SOC [24], our trials always converge to NM, which is consistent with the experiment.

Although the strength of correlation is still unclear in this system, the linear specific coefficient ratio of $\gamma_{\text{expt}}/\gamma_0$ can be a guide to determine whether correlation is considered. From $N(E_F)$ obtained from GGA + SOC, $\gamma_0 = 7.68$ (in units of $\text{mJ mol}^{-1} \text{K}^{-2}$), which is consistent with a previous report [15]. Compared with the experiment value of $\gamma_{\text{expt}} = 16.8$ [15], the ratio of $\gamma_{\text{expt}}/\gamma_0 \approx 2$ implies a moderate correlation strength in BaOsO₃. So, using the GGA + U + SOC approach, we applied the effective on-site Coulomb repulsion $U_{\text{eff}} = U - J$ to the Os ion to inspect whether this system is close to a magnetic ordering. Here, J is Hund's exchange integral. The change in the total spin moment is described with varying strength of U_{eff} in Fig. 4. At U_{eff} as small as 0.6 eV, a magnetic moment emerges. Assuming a typical $J \approx 0.6$ eV for $5d$ systems, the critical on-site Coulomb repulsion U^c is only 1.2 eV, which

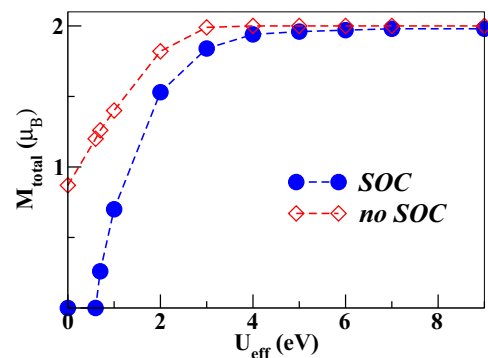


FIG. 4. (Color online) Total spin moment versus strength of the effective on-site Coulomb repulsion U_{eff} in GGA + U + SOC. For comparison, results of GGA + U are also given. In GGA + U + SOC, a small moment appears at $U_{\text{eff}}^c = 0.6$ eV and is saturated to $\sim 2\mu_B$ above $U_{\text{eff}} \approx 4$ eV. For almost all range of U_{eff} studied here, the Os orbital moment is negligible, but $0.1\mu_B$ to $0.2\mu_B$ appears above $U_{\text{eff}} = 6$ eV due to an enhanced orbital polarization in the large- U regime.

is obviously within the range of a reasonable U value for heavy transition metals. This fact suggests that this system is in the vicinity of a spin-ordered state, which is consistent with the Curie–Weiss behavior in the susceptibility [15]. Also, as U_{eff} increases, the moment rapidly increases and reaches the saturated value of $\sim 2\mu_B$ at $U_{\text{eff}} \approx 4$ eV.

D. Interplay among structure distortion, spin-orbit coupling, and magnetism

A structure distortion of the \mathcal{MO}_6 octahedron from the ideal one reduces the strength of SOC. So, one may expect a magnetic ordering to emerge in the orthorhombic perovskite CaOsO_3 with relatively small structure distortion [15], since the NM ground state of the cubic BaOsO_3 is purely due to effects of SOC. In GGA, FM with the moment of $0.82\mu_B/\text{Os}$ has lower energy by 6 meV/f.u. than NM, which is consistent with our fixed spin-moment calculations. The obtained Stoner parameter $I = 0.76$ eV is a little smaller than in BaOsO_3 , but $IN(E_F)$ is almost identical to that of BaOsO_3 . This indicates similar magnetic tendencies in both CaOsO_3 and BaOsO_3 , when ignoring SOC.

Figure 5 shows the blowup PDOS in both GGA and GGA + SOC, which indicate the exchange splitting of the t_{2g} manifold of about 0.3 and 0.1 eV for GGA and GGA + SOC, respectively. This reduction of 0.2 eV is almost identical to the strength of SOC in CaOsO_3 , which is only a half of that of BaOsO_3 . Consistent with this reduction, the inclusion of SOC reduces the moment by 50% and leads to a little magnetization energy, by only a few tenths of meV, although FM remains favored. A Stoner parameter in GGA + SOC can be estimated by $I \sim 1/N(E_F) = 0.35$, which is a half of the GGA value. Our results suggest that the unusual magnetism, observed in the experiment, in the orthorhombic CaOsO_3 results from emerging magnetic ordering induced by reducing the strength of SOC.

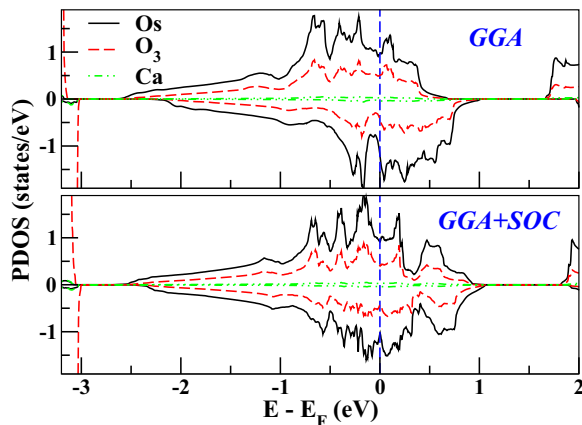


FIG. 5. (Color online) Enlarged PDOSs in GGA (top panel) and in GGA + SOC (bottom panel) of FM CaOsO_3 in the region of the partially filled t_{2g} manifold, which is 25% narrower than in BaOsO_3 . $N(E_F)$ is 4.40 (3.71) and 3.16 (2.82) in NM and FM, respectively, in units of states/eV per formula unit. The values inside parentheses are obtained from GGA + SOC calculations.

E. Phonon instability

Through first-principles calculations, Rahman *et al.* suggested that two types of Jahn–Teller distortions in a cubic and isovalent BaFeO_3 , which are related to an antiferroic distortion of either Os or O ions along the c axis, lead to a magnetic phase transition [25]. In the cubic BaOsO_3 , our calculations show that these distortions are energetically unfavored, which is consistent with the phonon dispersion (see below). As mentioned in the introduction, however, the measurement of the specific heat suggested an unusual behavior [15]. Thus, we carried out linear-response phonon calculations for our GGA optimized and a few compressed volumes, using both GGA and GGA + SOC.

Within GGA, we obtained several imaginary frequencies throughout all regimes (not shown here). Even for the inclusion of SOC, most of these unstable modes still survive, as given in the top of Fig. 6. As often observed in cubic perovskites [26], there are unstable modes, corresponding to the tilting of oxygens (triplet R_{25} , $36i$ cm^{-1}) and the rotation of in-plane

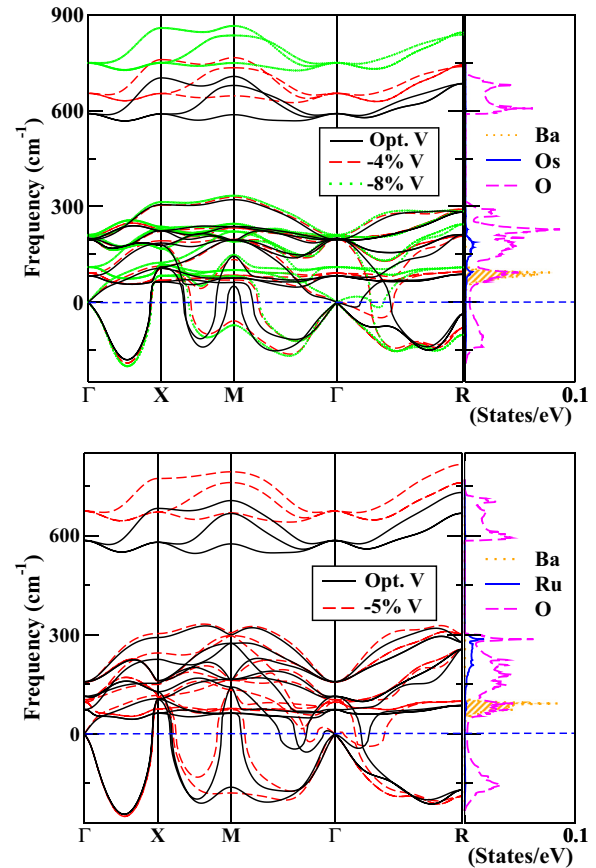


FIG. 6. (Color online) Top: Cubic BaOsO_3 phonon dispersion curve and atom-projected phonon DOSs, obtained from GGA + SOC by using our optimized volume and the 4% and 8% reduced volumes from the optimized value. Bottom: The same plots of cubic BaRuO_3 in GGA with our optimized and 5%-reduced volumes. The latter corresponds to the experimental value. SOC is excluded, since the strength of SOC is small [22]. In both plots, each atom-projected phonon DOS is given for the optimized volume. Negative values in the phonon dispersion curves correspond to imaginary phonon frequencies.

oxygens (singlet M_3) [27]. The latter appears at $\sim 60i$ cm $^{-1}$ in the compressed volume. Structure distortions involving these modes have been intensively investigated in several perovskites by Amisi *et al.* and He *et al.* [26,28], who suggest the gadolinium orthoferrite GdFeO $_3$ -type structural distortion as the stablest one. However, in BaOsO $_3$ the GdFeO $_3$ -type distortion is energetically unfavored. The most unstable modes of doublet 180i cm $^{-1}$ appear midway of the Γ - X line and are involved in the transverse acoustic mode, in which a displacement of one of the planar oxygens along the [110] direction is dominant. There are a few additional unstable modes along the M - Γ and the Γ - R lines. The latter is similar to the instability of the transverse acoustic mode observed in the possible charge-density-wave (CDW) Y $_3$ Co [29]. However, consistent with the strong three dimensionality in BaOsO $_3$, no indication of CDW is observed in our calculations. We also performed these calculations with two compressed volumes, which are 4% and 8% smaller than the optimized value, to investigate effects of pressure on the instability. The former corresponds to the experimental value. As shown in the top of Fig. 6, most of the unstable modes remain nearly unchanged, indicating that volume contraction does not play an important role in the instability. The complicated phonon instability seems to be of interest, requiring further studies in both theoretical and experimental viewpoints, although this is not an issue covered in this research.

For comparison, phonon calculations were carried out in the cubic BaRuO $_3$, which is cubic even at temperature as low as 10 K [16]. The bottom of Fig. 6 displays our results, which show very similar behavior to that of BaOsO $_3$. It is unclear yet why these phonon spectra show serious instabilities, although both the cubic BaOsO $_3$ and BaRuO $_3$ phases are experimentally stable. One possible scenario is that quantum fluctuations would stabilize the cubic phase in both systems, as discussed for some cubic perovskites [30,31]. This is true only when the difference in energy between the cubic phase and a distorted phase is very small. Also, the instability may be related to the unusual behavior in the specific heat and the fact that

these cubic systems can be synthesized only by a technique of extreme conditions.

IV. DISCUSSION AND CONCLUSION

Comparing the experimental linear specific-heat coefficient $\gamma_{\text{expt}} = 18$ mJ mol $^{-1}$ K $^{-2}$ of CaOsO $_3$ with our theoretical values, there is an enhancement of a factor of two to three, implying a moderate correlation. Within the GGA + SOC + U approach, applying U_{eff} to Os ions in CaOsO $_3$ leads to an insulating state at $U_{\text{eff}}^c \approx 6.8$ eV via a half-metallic state at $U_{\text{eff}} \approx 4$ eV. It may imply that this system is near an insulating state, resulting in the atypical electrical properties of CaOsO $_3$.

In conclusion, we investigated the effects of SOC on the magnetism of a few metallic $5d^4$ systems synthesized recently. The cubic BaOsO $_3$ shows a transition from a Stoner-type FM to NM incurred by SOC, of which the strength is identical to the exchange splitting of the t_{2g} manifold. However, at small $U^c \approx 1.2$ eV in GGA + SOC + U calculations, a small magnetic moment revives, suggesting that BaOsO $_3$ is an incipient magnet. On the other hand, our results indicate that the orthorhombic and isovalent CaOsO $_3$ is FM, since the structure distortion leads to reducing the strength of SOC. We anticipate this fact to be true for the isostructural and isovalent SrOsO $_3$. These results are consistent with a spin-ordered state observed very recently in a distorted double perovskite Sr $_2$ YIrO $_6$ with $5d^4$ Ir $^{5+}$ ions [32]. Our findings indicate that these systems are good examples to investigate the interplay between magnetism, lattice, SOC, and correlation. Further research is required to clarify these issues from both the theoretical and experimental viewpoints.

ACKNOWLEDGMENTS

We acknowledge P. Giannozzi and C.-J. Kang for useful discussions on phonon instabilities, and W. E. Pickett for fruitful communications in the early stage of this research. This research was supported by NRF-2013R1A1A2A10008946.

-
- [1] K.-W. Lee and W. E. Pickett, *Europhys. Lett.* **80**, 37008 (2007), and references therein.
 - [2] T. Dodds, T.-P. Choy, and Y. B. Kim, *Phys. Rev. B* **84**, 104439 (2011).
 - [3] G. Chen and L. Balents, *Phys. Rev. B* **84**, 094420 (2011).
 - [4] K.-W. Lee and W. E. Pickett, *Phys. Rev. B* **77**, 115101 (2008).
 - [5] M.-C. Jung, Y.-J. Song, K.-W. Lee, and W. E. Pickett, *Phys. Rev. B* **87**, 115119 (2013).
 - [6] H. Matsuura and K. Miyake, *J. Phys. Soc. Jpn.* **82**, 073703 (2013).
 - [7] B. J. Kim, H. Jin, S. J. Moon, J.-Y. Kim, B.-G. Park, C. S. Leem, J. Yu, T. W. Noh, C. Kim, S.-J. Oh, J.-H. Park, V. Durairaj, G. Cao, and E. Rotenberg, *Phys. Rev. Lett.* **101**, 076402 (2008).
 - [8] B. J. Kim, H. Ohsumi, T. Komesu, S. Sakai, T. Morita, H. Takagi, and T. Arima, *Science* **323**, 1329 (2009).
 - [9] X. Liu, V. M. Katukuri, L. Hozoi, W.-G. Yin, M. P. M. Dean, M. H. Upton, J. Kim, D. Casa, A. Said, T. Gog, T. F. Qi, G. Cao, A. M. Tsvelik, J. van den Brink, and J. P. Hill, *Phys. Rev. Lett.* **109**, 157401 (2012).
 - [10] G. Khaliullin, *Phys. Rev. Lett.* **111**, 197201 (2013).
 - [11] O. N. Meetei, W. S. Cole, M. Randeria, and N. Trivedi, *arXiv:1311.2823* (2013).
 - [12] M. Bremholm, S. E. Dutton, P. W. Stephens, and R. J. Cava, *J. Solid State Chem.* **184**, 601 (2011).
 - [13] R. F. Sarkozy and B. L. Chamberland, *Mat. Res. Bull.* **8**, 1351 (1973).
 - [14] B. L. Chamberland, *Mat. Res. Bull.* **13**, 1273 (1978).
 - [15] Y. Shi, Y. Guo, Y. Shirako, W. Yi, X. Wang, A. A. Belik, Y. Matsushita, H. L. Feng, Y. Tsujimoto, M. Arai, N. Wang, M. Akaogi, and K. Yamaura, *J. Am. Chem. Soc.* **135**, 16507 (2013).
 - [16] C.-Q. Jin, J.-S. Zhou, J. B. Goodenough, Q. Q. Liu, J. G. Zhao, L. X. Yang, Y. Yu, R. C. Yu, T. Katsura, A. Shatskiy, and E. Ito, *Proc. Natl. Acad. Sci. USA* **105**, 7115 (2008).
 - [17] K. Koepnick and H. Eschrig, *Phys. Rev. B* **59**, 1743 (1999).

- [18] K. Schwarz and P. Blaha, *Comput. Mater. Sci.* **28**, 259 (2003).
- [19] J. P. Perdew, K. Burke, and M. Ernzerhof, *Phys. Rev. Lett.* **77**, 3865 (1996).
- [20] P. Giannozzi, S. Baroni, N. Bonini, M. Calandra, R. Car, C. Cavazzoni, D. Ceresoli, G. L. Chiarotti, M. Cococcioni, I. Dabo, A. Dal Corso, S. Fabris, G. Fratesi, S. de Gironcoli, R. Gebauer, U. Gerstmann, C. Gougoussis, A. Kokalj, M. Lazzeri, L. Martin-Samos, N. Marzari, F. Mauri, R. Mazzarello, S. Paolini, A. Pasquarello, L. Paulatto, C. Sbraccia, S. Scandolo, G. Sclauzero, A. P. Seitsonen, A. Smogunov, P. Umari, and R. M. Wentzcovitch, *J. Phys.: Condens. Matter* **21**, 395502 (2009).
- [21] I. I. Mazin and D. J. Singh, *Phys. Rev. B* **56**, 2556 (1997).
- [22] Y.-J. Song and K.-W. Lee, *J. Korean Phys. Soc.* **62**, 1869 (2013).
- [23] H. Rosner, R. Weht, M. D. Johannes, W. E. Pickett, and E. Tosatti, *Phys. Rev. Lett.* **88**, 027001 (2001).
- [24] Our LSDA + SOC calculations show the same results, indicating that these results are insensitive to a selection of the exchange-correlation functional.
- [25] G. Rahman, J. M. Morbec, R. Ferradás, and V. M. García-Suárez, [arXiv:1310.3502](https://arxiv.org/abs/1310.3502) (2013).
- [26] S. Amisi, E. Bousquet, K. Katcho, and Ph. Ghosez, *Phys. Rev. B* **85**, 064112 (2012).
- [27] R. A. Cowley, *Phys. Rev.* **134**, A981 (1964).
- [28] J. He and C. Franchini, *Phys. Rev. B* **89**, 045104 (2014).
- [29] A. Podlesnyak, G. Ehlers, H. Cao, M. Matsuda, M. Frontzek, O. Zaharko, V. A. Kazantsev, A. F. Gubkin, and N. V. Baranov, *Phys. Rev. B* **88**, 024117 (2013).
- [30] W. Zhong and D. Vanderbilt, *Phys. Rev. Lett.* **74**, 2587 (1995); *Phys. Rev. B* **53**, 5047 (1996).
- [31] R. Kagimura, M. Suewattana, and D. J. Singh, *Phys. Rev. B* **78**, 012103 (2008).
- [32] G. Cao, T. F. Qi, L. Li, J. Terzic, S. J. Yuan, L. E. DeLong, G. Murthy, and R. K. Kaul, *Phys. Rev. Lett.* **112**, 056402 (2014).