Possible phase separation and weak localization in the absence of a charge-density wave in single-phase 1T-VS₂

A. Gauzzi,^{*} A. Sellam, G. Rousse, Y. Klein, D. Taverna, P. Giura, M. Calandra, and G. Loupias *IMPMC*, *UPMC-Sorbonne Universités*, *CNRS*, *MNHN*, *IRD*, *4*, *place Jussieu*, 75005 Paris, France

F. Gozzo[†]

Swiss Light Source, Paul Scherrer Institut, 5232 Villigen, Switzerland

E. Gilioli and F. Bolzoni

Istituto dei materiali per elettronica e magnetismo, Consiglio Nazionale delle Ricerche, Area delle Scienze, 43100 Parma, Italy

G. Allodi and R. De Renzi Dipartimento di fisica, Università di Parma, 43100 Parma, Italy

G. L. Calestani

Dipartimento di chimica - GIAF, Università di Parma, 43100 Parma, Italy

P. Roy

Synchrotron Soleil, L'Orme des Mérisiers, Saint-Aubin - 91192 Gif-sur-Yvette, France (Received 16 February 2014; revised manuscript received 9 May 2014; published 23 June 2014)

We report on a systematic study of the structural, magnetic, and transport properties of high-purity 1T-VS₂ powder samples prepared under high pressure. The results differ notably from those previously obtained by deintercalating Li from LiVS₂. First, no charge-density wave (CDW) is found by transmission electron microscopy down to 94 K, though *ab initio* phonon calculations unveil a latent CDW instability driven by an acoustic phonon softening at the wave vector $\mathbf{q}_{\text{CDW}} \approx (0.21, 0.21, 0)$ previously reported in deintercalated samples. A further indication of latent lattice instability is given by an anomalous expansion of the V-S bond distance at low temperature. Second, infrared optical absorption and electrical resistivity measurements give evidence of nonmetallic properties, consistent with the observation of no CDW phase. On the other hand, magnetic susceptibility and NMR data suggest the coexistence of localized moments with metallic carriers, in agreement with *ab initio* band structure calculations. This discrepancy is reconciled by a picture of electron localization induced by disorder or electronic correlations leading to a phase separation of metallic and nonmetallic domains in the nm scale. We conclude that 1T-VS₂ is at the verge of a CDW transition and suggest that residual electronic doping in Li deintercalated samples stabilizes a uniform CDW phase with metallic properties.

DOI: 10.1103/PhysRevB.89.235125

PACS number(s): 71.45.Lr, 72.15.Rn, 61.50.-f, 63.20.D-

I. INTRODUCTION

There has recently been a renewed interest in the interplay between charge-density wave (CDW) and superconducting states in layered dichalcogenides MX_2 (M = transition metal; X = S,Se), such as TiSe₂, TaSe₂, or TaS₂. The CDW mechanism remains controversial and various scenarios beyond the classic mechanism of Peierls instability [1], such as phonon softening [2], Coulomb repulsion [3–5], or exciton condensation [6], have been invoked. Clarifying this controversy may help to explore the possibility of unconventional, e.g., exciton-mediated, superconductivity in this system, as suggested by Ginzburg for metal-dielectric bilayers [7].

In order to address this issue, we carried out a systematic study of the structural, magnetic, and transport properties of pure VS₂, a model d^1 system with quasi-two-dimensional

properties. Its 1T (or CdI₂-type) structure (see Fig. 1) is made of layers of VS₆ octahedra separated by a van der Waals gap and is described by the $P\overline{3}m1$ symmetry. As compared to the isoelectronic and isostructural compound 1T-TaS₂, which has been extensively studied, VS₂ is simpler owing to the absence of superconductivity and to the lower Zvalue of V, which leads to a small spin-orbit coupling. Hence VS_2 is a model system to study the stability conditions of the CDW phase. Despite these favorable characteristics, the literature on VS_2 is limited because of its metastability [8] and V-rich $V_{1+x}S_2$ phases, where interstitial V atoms are located between the layers, are obtained at ambient pressure [9,10]. The pure (x = 0) phase has been hitherto synthesized only by deintercalating Li from LiVS₂ [8,11]. Early magnetic [8] and NMR [12] studies on such deintercalated samples point at a CDW phase below 305 K; however, this phase was observed only recently by transmission electron microscopy [11], which shows an incommensurate in-plane propagation vector, $\mathbf{q}_{\text{CDW}} = (0.21, 0.21, 0)$. Similar to 1T-TaS₂ at high temperature or to 2H-TaS₂ at low temperature, the incommensurate CDW phase of VS_2 is concomitant to metallic properties. Indeed,

^{*}andrea.gauzzi@upmc.fr

[†]Present address: Excelsus Structural Solutions sprl, 1150 Bruxelles, Belgium.



FIG. 1. (Color online) Synchrotron x-ray diffractogram measured on VS₂ powders at 5 K at the Materials Science beamline of the Swiss Light Source (SLS) using a wavelength $\lambda = 0.49575$ Å. Red and black lines represent observed and calculated profile after Rietveld refinement, respectively, while the blue line is their difference. Green ticks represent the calculated position of the Bragg peaks (see also Table I). Inset: the 1*T* structure of VS₂ described by the $P\bar{3}m1$ space group.

only commensurate CDW phases are insulating, as in $TiSe_2$ and TaS_2 at low temperature. Our main result is that no CDW metallic phase is found in high-purity VS₂ samples synthesized under high pressure.

The paper is organized as follows. In Sec. II, we report on the sample preparation under high pressure and on the experimental methods. In Sec. III, we report on the main experimental results, the salient results being the absence of the CDW phase concomitant to electron localization. Section IV is devoted to the discussion of the results and to the conclusions.

II. EXPERIMENT

A. High-pressure synthesis

We reproducibly synthesized pure VS₂ powders under high pressure using a multianvil apparatus. We first prepared a 1:2.05 mixture of vanadium metal and sulfur powders within a high-pressure capsule made of thin Pt foil. A 5% sulfur excess was used to ensure the full oxidation of V. The capsule was subsequently kept at 5 GPa and 700 °C for 2 h. Preliminary measurements using a commercial x-ray diffractometer equipped with a Cu K_{α} source indicate that the as-prepared powders are single phase within the sensitivity limit of the apparatus. The external part of the capsule in contact with the Pt foil exhibits the presence of PtS₂ and was removed. The single-phase stoichiometric properties of the powders were confirmed by a synchrotron x-ray diffraction study described below. Our strategy of using high-pressure synthesis has been motivated by two early studies showing that, the excess of vanadium is reduced to x = 0.18 at 0.2 GPa [13] and to x = 0.11 at 2 GPa [14]. An extrapolation of these results to higher pressures suggests that the x = 0 phase should be stabilized at 5 GPa, which is confirmed by our present finding.

B. Structural study

The as-prepared powders have been studied by means of synchrotron x-ray powder diffraction using a wavelength $\lambda = 0.49575$ Å at the Materials Science beamline of the Swiss Light Source at the Paul Scherrer Institut. The characteristics of the beamline are described in detail elsewhere [15]. Salient feature is the use of the high-resolution and fast Mythen microstrip detector enabling the simultaneous detection of the diffracted intensity over 120° in 2ϑ in the Debye-Scherrer (transmission) geometry. The substantially reduced data acquisition time and the parallel detection allowed us to perform an accurate structural study as a function of temperature in a wide 5-270 K range. In order to avoid preferential orientation effects, which are expected considering the layered structure of VS_2 , the powders were placed in a spinning glass capillary mounted on the sample holder. The data were collected during both the cooling down and the heating up of the samples in order to exclude thermal hysteresis effects. The structural properties were further investigated using a JEOL 2100F high resolution transmission electron microscope (HRTEM) reaching 1.9 Å resolution and equipped with a 200 keV field-emission electron gun. The measurements were carried out on individual $\sim 1 \ \mu m$ size crystallites at room and liquid nitrogen temperatures. In the latter case, the temperature of the sample holder was ≈ 94 K. No alteration of the structure induced by the electron beam was detected.

C. Magnetic and transport properties

The dc magnetization of the as-prepared powders was measured in the 5–300 K range using a Quantum Design SQUID magnetometer equipped with a 5 T NbTi superconducting magnet. Both zero-field-cooling and field-cooling curves were taken at 10 Oe. Magnetization curves were taken also as a function of field up to 5 T at fixed temperatures in the 5–300 K range. In order to probe selectively the magnetic behavior of the V ions, ⁵¹V NMR experiments were performed by employing a home-built spectrometer [16] and a variable-field superconducting magnet. Spectra were recorded in field sweep mode at a fixed frequency of 78.3 MHz by employing a standard 90°- τ -90° spin-echo pulse sequence with a delay, τ , of 10 μ s and a pulse duration of 2.4 μ s. Spin lattice relaxation was measured by the saturation recovery method.

The frequency-dependent optical absorption, A_{ω} , was measured at the AILES beamline of the SOLEIL synchrotron facility on a pellet obtained by intimately mixing 0.5% weight of VS₂ powders with CsI powders. The characteristics of this beamline are described in detail elsewhere [17]; in summary, we used a pulse tube closed cycle cryostat by Cryomech to vary the temperature within the 4–300 K range and a commercial Bruker IFS125/HR spectrometer to measure the transmission in the far infrared 150–650 cm⁻¹ range. A spectral resolution of 1 cm⁻¹ was achieved using a 6 μ m beam splitter combined with a commercial Infrared Lab

TABLE I. Refined structure of VS₂ in the trigonal *P*-3*m*1 space group at 5 K. Refined lattice parameters are a = b = 3.23055(1) Å and c = 5.70915(2) Å. Numbers in parentheses indicate statistical uncertainty. Atomic coordinates *x*, *y*, and *z* are in reduced lattice units.

Atom	Wyckoff pos.	Site symmetry	x	у	Z	$B_{\rm iso} \; (\times 10^{-4} \; {\rm \AA}^2)$
V	1 <i>a</i>	т	0	0	0	(*)
S	2d	1	1/3	2/3	0.255 03(12)	0.326(11)
(*) D 0	$020(18)$ Anisotropia θ	$n = 10^{4}$ for the	Vatam			

(*) $B_{eq} = 0.920(18)$. Anisotropic β_{ij} parameters (×10⁴) for the V atom: $\beta_{11} = \beta_{22} = 236.4(4.3); \beta_{33} = 22.4(2.3); \beta_{12} = -118.2(2.1); \beta_{13} = 0; \beta_{23} = 0$

Reliability factors with all nonexcluded points (not corrected for background):

 $R_p = 2.53; R_{wp} = 4.52; R_{exp} = 3.40; \chi^2 = 1.76$

bolometer detector. The absorption of VS₂ was obtained by subtracting the absorption of a pure CsI pellet from the absorption of the VS₂-CsI pellet; the absorbance was then determined using the usual relation. The study of the transport properties was complemented by a dc electrical resistivity measurement down to 2 K on a sintered sample using a commercial Quantum Design Physical Property Measurement System.

D. Band structure calculations

First principles calculations were performed using density functional theory (DFT) in the linear response [18] using the Quantum-Espresso code [19] within the local density approximation (LDA) [20]. We used norm-conserving [21] and ultrasoft [22] pseudopotentials for S and V, respectively, and included semicore states in the V pseudopotential. Energy cutoff values were 60 Ry for the kinetic energy expansion and 600 Ry for the charge density. The Brillouin zone integration was performed over a $24 \times 24 \times 12$ electron-momentum grid and a $4 \times 4 \times 2$ phonon-momentum grid. The Fermi surface Hermitian-Gaussian smearing used in the simulation was 0.01 Ry.

III. RESULTS

A. Structural properties

Rietveld refinements of the x-ray diffractograms measured on the as-prepared VS₂ powders confirm the previously reported 1T structure in the whole 5-270 K range and indicate that the samples are single phase, with the exception of $\leq 0.3\%$ vol of PtS₂ impurity formed by the reaction of S with the Pt capsule. For all temperatures, good refinements with reliability factors $R_p \approx 2.5$ were obtained in the $P\overline{3}m1$ symmetry of the 1T structure by assuming no interstitial vanadium atoms (x = 0). The result of the refinement is reported in Fig. 1 and Table I for the 5 K data. The small thermal parameters indicate a good degree of crystallinity and a limited disorder. In the above symmetry, all the V-S bond distances in the VS_6 octahedron are equal and the only structural distortion allowed is a tilt of the octahedron axis with respect to the octahedron plane. We find a small tilt of $3.90(4)^\circ$, which indicates that the octahedra are almost regular. We also verified the possibility of interstitial V atoms at the 1b site (0.0.1/2). For the 5 K data, this refinement yielded a site occupancy factor x = 0.05, though the reliability factor was found to improve only slightly from $R_p = 2.53$ to 2.20. We thus conclude that the excess x, if any, is 0.05 or less.

Figure 2 summarizes the temperature dependence of the structural parameters. Whilst the behavior of the *a* axis is normal, the *c* axis exhibits an anomalous upturn at $T_{II} \approx 120$ K, which leads to a broad maximum at $T_{III} \approx 50$ K. This anomaly is associated with a V-shaped behavior of the V-S distance, *d*, showing a sudden linear increase with decreasing temperature at T_{II} [see Fig. 2(b)]. By extrapolating the high-temperature behavior of *d* down to low temperatures, it is found that the anomaly corresponds to an expansion of *d* by ~0.01 Å. Notable is the fact that the *a*- (*c*-) axis parameter of the Li



FIG. 2. (Color online) Temperature dependence of the lattice parameters, *a* and *c* (top panel), and of the unit cell volume, *V*, and the V-S distance, *d* (bottom), of VS₂ obtained from the Rietveld refinement of the powder diffraction data described in the text. The anomalies of the *c* axis and of the V-S distance at T_{II} and T_{III} are discussed in the text. T_I indicates the temperature at which the electrical resistivity of Fig. 8 exhibits a minimum.



FIG. 3. (Color online) Transmission electron microscopy diffraction pattern taken at 94 K on a grain of VS₂ marked by a circle in the inset. The (*hk*0) Bragg peaks and the in-plane $P\overline{3}m1$ unit cell are shown. The absence of satellite peaks rules out the presence of long-range structural modulations.

deintercalated samples [8]. Further evidence of a structural difference between the two types of samples is given by TEM. Contrary to a previous report on deintercalated crystals [11], the present TEM diffraction patterns exhibit only the main spots of the hexagonal lattice and no trace of satellite peaks are found either at room temperature or at 94 K (see Fig. 3). In conclusion, no CDW phase or any other long-range structural modulations are detected, although the anomalous behavior of the V-S distance points at an incipient structural instability.

B. Magnetic properties

1. Magnetic susceptibility measurements

Figure 4 shows the temperature dependence of the dc magnetic susceptibility, $\chi(T)$, measured at 10 Oe. The ZFC and FC curves are smooth and display no hysteresis; thus no magnetic orderings or other phase transitions occur, in agreement with the x-ray diffraction results. In Li deintercalated VS_2 [11] and in the isostructural and isoelectronic compounds VSe₂ [23] and TaS₂ [24], the CDW transition manifests itself as an abrupt drop of χ . The absence of this feature in our samples confirms the TEM evidence of no structural modulation. The $\chi(T)$ curve is well described by a conventional Curie-like term, $\chi(T) = C/(T + \vartheta)$, in addition to a constant paramagnetic term, χ_0 , with the following fitting parameters: $\chi_0 = 2.62 \pm 0.03 \times 10^{-4} \text{ emu mol}^{-1}$, a Curie constant, $C = 6.29 \pm 0.02 \times 10^{-2} \text{ emu mol}^{-1}$ K, and a vanishing Weiss constant, $\vartheta = -0.46 \pm 0.01$ K. These values are similar to those found in the isostructural and isoelectronic compound VSe_2 in the CDW phase [23].

Assuming a band picture and that the above χ_0 value entirely arises from the Pauli contribution, we estimate a sizable density of states at the Fermi level, $g(E_F) = \chi_0/\mu_B^2 \approx$



FIG. 4. (Color online) Top panel (a): temperature dependence of the magnetic susceptibility, $\chi(T)$, of a representative VS₂ sample measured at 10 Oe in both, zero-field- and field-cooling (ZFC, FC) mode. The two ZFC and FC curves are identical within the experimental error. The continuous line represents a Curie-Weiss fit of the ZFC data, as described in the text. Inset: the same data are plotted as ($\chi - \chi_0$)*T* vs *T* in order to put into evidence the deviation of the data from the ideal Curie-Weiss dependence. The anomalies at T_{II} , T_{III} , and T_{IV} are concomitant to the anomalies of Figs. 2 and 8 and discussed in the text. Bottom panel (b): temperature dependence of the inverse magnetic susceptibility, from which the constant term χ_0 has been subtracted, which gives evidence of the overall Curie-Weiss behavior of the data with negligible Weiss constant, $\vartheta \approx 0$.

8.1 states $eV^{-1} cell^{-1}$. However, the NMR data reported below indicate a significant orbital contribution; hence we believe that the above $g(E_F)$ value is overestimated and a more realistic estimate is given in the section devoted to the NMR results. Li deintercalated VS₂ samples exhibit two times larger χ_0 values $\approx 6.0 \times 10^{-4}$ emu/mol [8,11], so our samples are expected to be less metallic than the latter ones, in agreement with the optical absorption data presented below. The measured Curie constant, *C*, corresponds to an effective moment $\mu = 0.79 \mu_B$ per V ion much smaller than the value $\mu = 1.73\mu_B$ expected for $S = 1/2 V^{4+}$ ions within a ionic picture. This reduction suggests a two-band scenario, where a majority of electrons are delocalized, whilst a minority $\approx 21\%$ form localized moments.

A departure from the ideal Curie behavior is noted in the $(\chi - \chi_0)T$ vs *T* plot of Fig. 4; three anomalies are seen at T_{II} , T_{III} , and T_{IV} . The first two ones are concomitant to the structural anomalies mentioned above and shown in Fig. 2. The *V*-shaped feature at $T_{II} \approx 120$ K mimics a similar feature in

the temperature dependence of the V-S bond distance, whilst the maximum at $T_{III} \approx 50$ K follows the maximum of the *c*-axis lattice parameter in Fig. 2. These anomalies are also found in the NMR response and in the electrical resistivity (see below), so they should reflect subtle changes of the electronic structure.

2. NMR measurements

In the whole 1.6–300 K range studied, we measured a strong ⁵¹V NMR signal. Note that this signal does not arise from the localized moments of the V ions because the nuclei of paramagnetic V⁴⁺ ions probe a large fluctuating hyperfine field $B_{hf} \approx 10$ T [25–27]. This field causes exceedingly fast nuclear relaxations that cannot be detected, unless a strong exchange interaction narrows the resonance line. This is not the case here, considering the negligible exchange energy measured, $J/k_B \sim \vartheta \lesssim 1$ K. We estimated the amplitude and correlation time of the magnetic fluctuations at the nucleus to be $\delta\omega = {}^{51}\gamma B_{hf} \approx 7 \times 10^8$ s⁻¹ and $\tau = \hbar/J \approx 10^{-11}$ s, respectively, where ${}^{51}\gamma/2\pi = 11.19$ MHz/T is the nuclear gyromagnetic ratio of ${}^{51}V$. From these values, one obtains a transverse relaxation time $T_2 \approx (\delta\omega^2\tau)^{-1} \approx 0.2 \ \mu$ s much shorter than the instrumental dead time, ~10 \ \mus [28]. Thus the nuclear resonance of the V⁴⁺ ions is not detectable and the observed signal originates from nonmagnetic V.

In Fig. 5 we show two representative ${}^{51}V$ field-sweep spectra at 150 and 1.6 K. They are characterized by two sharp and inhomogeneously broadened peaks and by a broad shoulder. Since the sample is single phase, this indicates that the nonmagnetic V ions probe two distinct electronic environments, which support an electronic phase separation picture in the local scale. Both peaks exhibit a sizable positive Knight shift of the resonance line with respect to the reference



FIG. 5. (Color online) ⁵¹V field-sweep NMR of VS₂ at 1.6 and 150 K, recorded at 78.3 MHz. The reference field, B_{ref} , is indicated by the vertical red line. For clarity, the 1.6 K spectrum has been scaled by a factor of 2 after correction for the nuclear susceptibility $\propto 1/k_BT$. The continuous line is a fit of the two sharp peaks indicated by the broken line. The small peak at 6.927 T produced by the ⁶³Cu signal of the copper coil has been excluded from the fit.

field $B_{\rm ref} = 6.9959$ T. Upon cooling, the two-peak structure is better resolved due to a transfer of spectral weight from the main peak at higher field to the minor peak. The spectra do not exhibit the characteristic quadrupolar pattern of I = 7/2nuclear spins expected for powder samples of ${}^{51}V$. This pattern would display a sharp central line and 2I - 1 satellite lines shifted by the quadrupolar field and broadened by the random orientation of the electric field gradient (EFG), which gives rise to peculiar powder singularities. We believe that the quadrupolar structure is not resolved because the high fields used cause a large magnetic broadening of a spatially inhomogeneous distribution of Knight shifts. This explanation is supported by a former estimate of the quadrupolar frequency, v_Q . Specifically, for VS₂ [12] and VSe₂ [29], a value $2\pi \nu_Q/^{51} \gamma \approx 0.033$ T (in field units) was found. Since v_0 is proportional to the EFG at the nucleus, the broad shoulders are attributed to the quadrupole satellite transitions in the presence of magnetic and possibly EFG inhomogeneities, which would smear the powder singularities.

The two sharp peaks are fitted well in the whole 1.6–300 K range studied, which enables us to plot the Knight shifts ⁵¹K of the two peaks and the shift ⁵¹ \overline{K} of the center of gravity of the spectrum as a function of temperature [see Fig. 6(a)]. For both peaks, ⁵¹K increases upon cooling down to $T_{III} \approx 50$ K, where an anomaly in the susceptibility data is also observed, then it levels off. Below 100 K, the average shift ⁵¹ \overline{K} exhibits a more marked temperature dependence than that of the single peaks owing to the aforementioned transfer of spectral weight. The plot also shows the temperature dependence of the NMR signal amplitude, ⁵¹A, integrated over the spectrum and corrected for the nuclear susceptibility $\propto 1/k_BT$. Note that ⁵¹A is nearly T independent at high temperature, whilst it drops by $\approx 50\%$ below $T_{IV} \approx 20$ K, where the susceptibility also decreases. Considering that ⁵¹A is directly proportional to the number



FIG. 6. (Color online) (a) Temperature dependence of the Knight shifts ${}^{51}K$ of the two peaks (open triangles), the center of gravity of the spectra ${}^{51}\overline{K}$ (filled triangles) and the integrated spectral amplitude multiplied by temperature ${}^{51}A$ (squares). (b) Temperature dependence of the spin-lattice relaxation rate T_1^{-1} and best fit to the Korringa law $T_1^{-1} \propto T$ (dashed line). Inset: the same as before for the Korringa product $(TT_1)^{-1}$.

of ${}^{51}V$ nuclei detected in the resonance, this loss of signal indicates that, for a sizable fraction of the nuclei, the spin-spin relaxation time, T_2 , is significantly reduced. This prevents detection if this time becomes shorter than the dead time of the NMR receiver. It follows that the correlation time of the fluctuations probed by these nuclei increases dramatically below T_{IV} . The nature of such slow fluctuations remains to be established; they may be either magnetic or electric, e.g., due to slow charge motion, which would produce random EFG modulations. The ${}^{51}K$ data provide further evidence that the NMR signal does not arise from localized V moments. Namely, the orbital contribution to K and to the susceptibility χ is usually negligible for a magnetic ion; thus K is proportional to χ via the hyperfine coupling constant and K should follow the Curie law, which is not observed here. Moreover, in a magnetic ion, a *negative* hyperfine field at the nucleus is expected [25], in contrast with our observation of *positive K*.

The temperature dependence of the spin-lattice relaxation was measured for the main peak. The recovery of the nuclear magnetization M(t) to thermal equilibrium was found to obey the following multiexponential law, which is appropriate for the magnetic relaxation of the quadrupole-resolved central line of a I = 7/2 nucleus following its selective saturation [30]:

$$\frac{M(t)}{M(t=\infty)} = 1 - \frac{4}{21}e^{-2Wt} - \frac{2}{11}e^{-12Wt} - \frac{20}{91}e^{-30Wt} - \frac{175}{429}e^{-56Wt}, \quad (1)$$

where *W* is the transition probability between two adjacent nuclear Zeeman levels. In Fig. 6(b), the spin-lattice relaxation rate $T_1^{-1} \equiv 2W$, determined by the analysis of M(t) using Eq. (1), is plotted as a function of temperature. In the 10–150 K range, the data follow the Korringa law $T_1^{-1} \propto T$, characteristic of a nonmagnetic metal with a temperature-independent spin susceptibility. The validity of this law in the above range is also apparent in the $(T T_1)^{-1}$ vs T plot, which shows a nearly constant value $\approx 0.17 \text{ K}^{-1}\text{s}^{-1}$. At lower temperature below T_{IV} , a large deviation from this simple behavior is found.

In a metal, K and T_1^{-1} probe the static and dynamic electronic spin susceptibilities, respectively. The two quantities are then predicted to scale with each other according to the universal Korringa relation:

$$K^2 = R(TT_1)^{-1}, (2)$$

where *R* is the Korringa ratio. For a simple *s*-band metal with negligible electronic correlations, *R* is expected to be equal to the universal value $S_0 = (\hbar/4\pi k_B)(\gamma_e/\gamma_n)^2$, where γ_e and γ_n are the electronic and nuclear gyromagnetic ratios [28]. For a *d*-band metal, a similar result holds, with $R = \kappa S_0$ and $2 \le \kappa \le 5$ [31]. From Fig. 6, it is clear that the above relation cannot be followed by our temperature-dependent ⁵¹*K* and temperature-independent $(TT_1)^{-1}$ data. Even restricting ourselves to the 10 K $\le T \le 50$ K range, where both quantities are approximately constant, anomalously high values $R/S_0 = 10$ -100 are obtained. This is explained by comparing our data with those by Tsuda *et al.* [12], who found one order of magnitude smaller Knight shifts but a comparable $(TT_1)^{-1} \approx 0.4$ K⁻¹s⁻¹ at T < 100 K, implying a spin susceptibility smaller by a factor of 1.5 only in our sample. The difference between the T_1 values reported by Tsuda *et al.* and ours is not significant, for the recovery law of a quadrupolar nucleus depends upon the experimental conditions, e.g., the width of the irradiated band and the length of the saturation pulse train [32,33]. This may lead to deviations from Eq. (1), hence an uncertainty on T_1 . Tsuda *et al.* showed that ${}^{51}K$ contains two contributions with opposite sign which nearly cancel each other: a positive orbital one, ${}^{51}K_{orb}$, and a negative one, ${}^{51}K_d$, proportional to the spin susceptibility of the *d*-wave conduction band. The discrepancy between Tsuda's ⁵¹K values and ours are then reconciled by assuming a similar value of the K_d term (and a similar Korringa product) for the two samples but a much larger orbital contribution in our sample, which explains the larger Korringa ratios R. A large orbital shift ${}^{51}K_{\rm orb}$ in *d*-wave metals originates from the presence of low-lying excited multiplets via the van Vleck mechanism [34,35] and thus depends upon the crystal field splitting. Since the latter is sensitive to the local structural parameters, we believe that the anomaly of ${}^{51}K$ and ${}^{51}A$ at T_{III} reflects the corresponding anomaly of the c/a ratio and of the V-S bond distance.

To conclude this section, we endeavor to estimate the Pauli contribution χ_d to the magnetic susceptibility from the T_1^{-1} data and from the values of the Korringa and hyperfine constants determined experimentally in previous reports. According to band calculations [36] and ARPES data [11], we should assume a full *d*-wave character of the conduction band. The Korringa ratio was determined to be $R = 5.0 \times 10^{-6}$ s K in both VS₂ and VSe₂ [12,29]. According to Eq. (2), we obtain $K_d = -0.09\%$ for our sample. Next, we consider the proportionality relation ${}^{51}K_d = \mathcal{A}_{iso}\chi_d/N_A$, where \mathcal{A}_{iso} is the isotropic core-polarization term in the hyperfine coupling of vanadium and N_A is the Avogadro number. From the electron spin resonance of the V^{3+} ion [37,38], we take $A_{iso} = -85 \text{ kOe}/\mu_B$ to obtain $\chi_d = 6 \times 10^{-5} \text{ emu/mol}$, which corresponds to a density of states $g(E_F) \approx 2.0$ states $eV^{-1}cell^{-1}$. We finally obtain $\chi_{orb} = \chi_0 - \chi_d \approx$ 2×10^{-4} emu/mol, comparable with the values reported in the related compound $LiVO_2$ [39].

C. Transport properties

1. Infrared optical conductivity

The optical absorption spectra, A_{ω} , measured in the infrared 140–650 cm⁻¹ range is shown for two representative temperatures, 5 and 300 K, in Fig. 7. Salient features are as follows: (i) the absence of a Drude peak; (ii) a sizable and monotonic increase of A_{ω} with ω ; (iii) the absence of phonon features at high temperature and one broad phonon feature at $\omega_{ph} \approx 390 \,\mathrm{cm}^{-1}$ at low temperature; (iv) the spectra change little with temperature in the whole 5–300 K range. Specifically, besides the appearance of the above phonon feature at low temperature, the only noticeable change is a small increase of the absorption with temperature at low frequency. Considering that, under the present experimental conditions of low absorption, A_{ω} is simply proportional to the real part of the optical conductivity, σ_{ω} , features (i) and (ii) give evidence of nonmetallic transport, whilst (iv) suggests a modest charge localization. According to a symmetry analysis of the 1T structure, feature (iii) is



FIG. 7. (Color online) Temperature dependence of the optical absorption of pure VS₂ powders as a function of frequency in the 4–300 K range. The absence of a Drude peak and a weak increase of absorption with temperature indicate a nonmetallic behavior. Note the absence of phonon peaks, except for a weak absorption structure at \approx 390 cm⁻¹ (\approx 49 meV) suggestive of a highly screened phonon mode, in agreement with the prediction of a E_u infrared-active mode at 405.8 cm⁻¹ (see text).

consistent with the expectation of two infrared-active phonon modes with A_{2u} and E_u symmetry. It is then plausible that only one of these modes is detected at low temperature; indeed, the measured frequency, ω_{ph} , is in good agreement with the calculated value of 405.8 cm⁻¹ for the E_u mode (see section on phonon calculations). The fact that the phonon feature at ω_{ph} is visible only at low temperature and is broadened indicates a partial screening of the infrared active dipoles, in agreement with the observation of Korringa behavior in the NMR data and of a significant density of states extracted from the measured χ_0 value and calculated *ab initio* (see below). On the other hand, the absence of a Drude peak indicates that the electron density responsible for the screening is not sufficient to ensure a metallic transport, possibly because of disorder or electron-electron correlations leading to charge localization. Conversely, a picture of conventional metal and semimetal accounts for the infrared response of other transition metal dichalcogenides, such as 2H-TaS₂ [40] and 1T-TiSe₂ [41], respectively, which both exhibit well-developed Drude peaks.

A picture of charge localization has been previously proposed also for the isostructural and isoelectronic d^1 compound 1T-TaS₂ [3,5], which indeed displays a qualitatively similar optical response in the so-called nearly commensurate CDW (NCCDW) phase at room temperature [42]. On the other hand, in the commensurate CDW phase at low temperature, the optical spectra of 1T-TaS₂ [42] and also of 1T-TiSe₂ [41] display a large number of sharp phonon features, consistent with the large number of infrared phonon modes expected for the CDW superstructure. The absence of such additional phonon features in our samples is a further strong argument against the existence of a CDW phase. It would be interesting to see whether the Li deintercalated samples display additional infrared phonon features in the CDW phase.



FIG. 8. (Color online) Temperature dependence of the dc electrical resistivity, ρ , and of its derivative. Note the characteristic semimetallic behavior characterized by a crossover from a positive to a negative slope of ρ at T_I , and two abrupt changes of the slope at T_{II} and T_{IV} concomitant to anomalies of the lattice parameters and of the magnetic susceptibility (see Figs. 2 and 4).

2. Electrical resistivity

In Fig. 8, we report the electrical resistivity curve $\rho(T)$ measured in the 2–325 K range on one representative VS_2 sample. Consistent with the optical conductivity results, the overall temperature dependence of ρ exhibits a modest variation within the 5.5–7.2 m Ω cm range. Specifically, at low temperature, one notes a weak increase which confirms the above evidence for charge localization. At high temperature, the weakly negative resistivity coefficient above $T_I \approx 250$ K is explained by a small increase of thermally activated carriers, which is characteristic of a small energy gap at E_F . A comparison between the present resistivity data and those obtained on Li deintercalated samples [11] further confirms that the latter samples are different from the present ones. First, we find no anomaly of $\rho(T)$ at $T_{CDW} = 305$ K, which supports the conclusion on the absence of CDW transition. Second, the marked metallic behavior of the above samples below \approx 260 K, with a sizable residual resistivity ratio, $RRR \sim 10$, is completely different from the nonmetallic behavior reported here. On the other hand, the present data display two anomalies at $T_{II} \approx 120$ K and $T_{IV} \approx 20$ K concomitant to the anomalies in the dependence of the V-S distance and of the magnetic susceptibility with temperature shown in Fig. 3 and Fig. 4, respectively. In Fig. 8, these anomalies appear as inflexion points of the $\rho(T)$ curve and are better seen in the derivative of the curve. The inflexion point at T_{II} corresponds to a slowing down of the resistivity increase upon cooling, which is interpreted as a weakening of the localization below T_{II} . The opposite seems to occur at T_{IV} . In order to provide a full picture of the transport properties of the present high-pressure VS_2 phase and to confirm the proposed scenario of charge localization, a single crystal study would be desirable.

D. Electronic structure and lattice stability

We first performed a full structural optimization of the crystal structure by varying both internal coordinates and unit cell parameters. We found that the LDA result



FIG. 9. (Color online) Left: electronic band structure and density of states of VS_2 in the local density approximation. The size of the circles on a given band is proportional to the vanadium 3*d* component of the band. Right: phonon dispersion along the (1,1,0) reciprocal space direction. The label q_{CDW} indicates the propagation vector of the CDW instability reported in Ref. [11].

substantially underestimates the experimental volume. Namely, we obtained $a_{\text{LDA}} = 3.102$ Å, $c_{\text{LDA}}/a_{\text{LDA}} = 1.786$, and a *z* coordinate of 0.262 (in reduced coordinates) for the *S* atom. Since a compressed volume tends to weaken the tendency towards CDW formation (it is recalled that the CDW phase disappears under hydrostatic pressure in all metallic transition metal dichalcogenides), here we used the experimental lattice parameters and optimized only the internal coordinate of the *S* atom in order to investigate the stability of the structure towards CDW formation.

1. Electronic band structure

The electronic structure of VS₂ shown in Fig. 9 turns out to be very weakly dependent on the volume used in the calculations. The overall shape of the band structure closely recalls that of TiSe₂ [2] but with a different position of the Fermi level and a smaller hybridization between the chalcogen *p* states and the transition metal 3*d* states. According to the band structure, VS₂ should be a metal with a density of states at the Fermi level $g(E_F) = 3.2$ states $eV^{-1}cell^{-1}$. This value is in agreement with the χ_0 value obtained from susceptibility measurements, considering that only one-third of this value arises from the Pauli term, as suggested by the analysis of the NMR data. As to the band characteristics at the Fermi level, the *d* electrons form electron pockets at the *M* and *L* symmetry points and an additional small electron pocket is present along the *K*-*A* direction.

2. Phonon dispersion

An insight into the occurrence of a CDW transition is obtained from the harmonic phonon dispersion calculation reported in Fig. 9. Within the harmonic approximation, an imaginary phonon frequency (here plotted as negative) is the signature of a second order structural instability, such as a CDW. In VS₂, we do indeed find that the harmonic phonon dispersion shows an instability of a transverse acoustic phonon at the same wave vector $\mathbf{q}_{\text{CDW}} = (0.21, 0.21, 0)$ of the CDW phase found experimentally in Ref. [11]. The caveat is that the calculation turns out to be extremely sensitive to the values of lattice parameters (much more than in the case of other dichalcogenides) and the unstable phonon mode is only slightly imaginary. This suggests that the structure can be stabilized by anharmonic effects. This is what indeed occurs in the parent compound 2H-NbS₂ [43], where harmonic calculations overestimate the tendency towards CDW formation and anharmonic effects stabilize the lattice without CDW. The lack of anharmonic effects in the present calculations and the high sensitivity of the DFT phonon dispersion on the lattice parameters do not allow us to draw more definitive conclusions. It is nevertheless safe to conclude that VS_2 is at the verge of a CDW instability, which accounts for the contrasting results about the occurrence of a CDW phase in Li deintercalated samples and in the present ones. Finally, the calculations predict two infrared-active E_u modes at 200.9 cm⁻¹ and 405.8 cm⁻¹ and two Raman-active ones at 246.3 cm⁻¹ and 355.3 cm⁻¹ with E_g and A_{1g} symmetry, respectively. The prediction of the 405.8 cm⁻¹ mode is consistent with the observation of a highly screened phonon mode at a similar frequency in the optical absorption, as discussed above.

IV. DISCUSSION AND CONCLUSIONS

In conclusion, we systematically studied the structural, magnetic, and transport properties of single-phase 1T-VS₂ samples prepared under high pressure. Contrary to previous reports on Li deintercalated samples, we found that 1T-VS₂ is nonmetallic and displays no long-range structural modulations, such as CDW. This difference is attributed to

the different synthesis route employed which does not require any chemical methods. Specifically, we envisage that residual Li atoms and iodine or acetonitrile molecules introduced by the Li deintercalation method may alter the doping level, which would be sufficient to stabilize the metallic and CDW phase owing to the semimetallic character of the band structure. The doping effect inherent in the deintercalation process is supported by the observation of structural modulations in chemically exfoliated MoS₂ and WS₂ by means of Li intercalation [44]. This scenario would also explain the different lattice parameters of the present VS₂ samples with respect to those of the Li deintercalated ones. A further support of this scenario is given by the enhancement of metallic properties in the related layered compounds AuVS₂ [45] and $Ag_{0.75}VS_2$ [46], where the nominal valence of V is close to 3+ instead of 4+.

Within a rigid band picture, the evidence of nonmetallic conductivity would be at odds with the Korringa-like behavior of the spin-lattice relaxation time, the sizable Pauli susceptibility, $\chi_0 \sim 10^{-4}$ emu mol⁻¹, and the prediction of metallic properties by band structure calculations. This discrepancy indicates that the above picture is not appropriate and localization effects due to disorder or electronic correlations should be invoked. An enhancement of the Pauli susceptibility induced by these effects in quasi two dimensions would also account for the comparable χ_0 values reported in other 1*T*-VS₂-related compounds, regardless of their metallic or insulating properties. Examples are the misfit layer system $La_{1,17-x}Sr_xVS_{3,17}$ [47,48], which displays no appreciable change of χ_0 across the doping-induced metal-insulator transition, the MV_2S_4 family (M=Ti, Cr, Fe, Ni, or Cu) [49,50], where both metallic and insulating properties are found, and the insulator $Sr_3V_5S_{11}[51]$.

In support of this scenario, it is recalled that a Mott state has been proposed by several groups for 1T-TaS₂, which

- R. E. Peierls, *Quantum Theory of Solids* (Clarendon, Oxford, 1955).
- [2] M. Calandra and F. Mauri, Phys. Rev. Lett. 106, 196406 (2011).
- [3] P. Fazekas and E. Tosatti, Physica B+C 99, 183 (1980).
- [4] K. Rossnagel and N. V. Smith, Phys. Rev. B 73, 073106 (2006).
- [5] B. Sipos, A. Kusmartseva, A. Akrap, H. Berger, L. Forro, and E. Tutis, Nat. Mater. 7, 960 (2008).
- [6] H. Cercellier, C. Monney, F. Clerc, C. Battaglia, L. Despont, M. G. Garnier, H. Beck, P. Aebi, L. Patthey, H. Berger, and L. Forró, Phys. Rev. Lett. 99, 146403 (2007).
- [7] V. L. Ginzburg, Sov. Phys. Usp. 13, 335 (1970).
- [8] D. W. Murphy, C. Cros, F. J. Di Salvo, and J. V. Waszczak, Inorg. Chem. 16, 3027 (1977).
- [9] H. Katsuta, R. B. McLellan, and K. Suzuki, J. Phys. Chem. Solids 40, 1089 (1979).
- [10] P. Poddar and A. K. Rastogi, J. Phys.: Condens. Matter 14, 2677 (2002).
- [11] M. Mulazzi, A. Chainani, N. Katayama, R. Eguchi, M. Matsunami, H. Ohashi, Y. Senba, M. Nohara, M. Uchida, H. Takagi, and S. Shin, Phys. Rev. B 82, 075130 (2010).

displays similar magnetic and transport properties in the aforementioned NCCDW phase [3-5,52]. In this phase, the direct STM observation of a nm-size domain structure [53], in excellent agreement with calculations [54], in conjunction with high-pressure resistivity measurements suggests a picture of insulating and commensurate CDW regions separated by metallic and weakly CDW-distorted ones [5]. A phase separation scenario may then apply to the present case as well, for our magnetic susceptibility data unveil the existence of both localized moments and metallic carriers, whilst the NMR data show that the metallic V sites are inhomogeneous. The existence of nm-size domains would indeed explain the absence of satellite diffraction peaks in our TEM pictures, while a slightly different doping level in Li deintercalated samples would be sufficient to stabilize a uniform metallic/CDW phase, in agreement with the prediction of a latent CDW instability by phonon calculations. The confirmation of a phase separation scenario for the present 1T-VS₂ samples awaits further studies by means of local probes, such as scanning tunneling microscopy/spectroscopy.

ACKNOWLEDGMENTS

The authors acknowledge L. Cario and M. Marezio for stimulating discussions and R. Lobo and B. Léridon for making available their PPMS system. M.C. gratefully acknowledges financial support of the Graphene Flagship and of the French National ANR funds under reference ANR-11-IDEX-0004-02, ANR-11-BS04-0019, and ANR-13-IS10-0003-01. Computer facilities were provided by CINES, CCRT, and IDRIS (Project No. x2014091202).

- [12] T. Tsuda, H. Yasuoka, Y. Kitaoka, and F. D. Salvo, J. Magn. Magn. Mater. **31–34**, 1101 (1983).
- [13] M. Yokoyama, M. Yoshimura, M. Wakihara, S. Somiya, and M. Taniguchi, J. Solid State Chem. 60, 182 (1985).
- [14] M. Nakano-Onoda, S. Yamaoka, K. Yukino, K. Kato, and I. Kawada, J. Less-Common Met. 44, 341 (1976).
- [15] B. Patterson, R. Abela, H. Auderset, Q. Chen, F. Fauth, F. Gozzo, G. Ingold, H. Kühne, M. Lange, D. Maden, D. Meister, P. Pattison, T. Schmidt, B. Schmitt, C. Schulze-Briese, M. Shi, M. Stampanoni, and P. Willmott, Nucl. Instrum. Methods, Phys. Res. A 540, 42 (2005).
- [16] G. Allodi, A. Banderini, R. De Renzi, and C. Vignali, Rev. Sci. Instrum. 76, 083911 (2005).
- [17] P. Roy, M. Rouzières, Z. Qi, and O. Chubar, Infrared Phys. Technol. 49, 139 (2006).
- [18] S. Baroni, S. de Gironcoli, A. Dal Corso, and P. Giannozzi, Rev. Mod. Phys. 73, 515 (2001).
- [19] P. Giannozzi, S. Baroni, N. Bonini, M. Calandra, R. Car, C. Cavazzoni, D. Ceresoli, G. L. Chiarotti, M. Cococcioni, I. Dabo, A. D. Corso, S. de Gironcoli, S. Fabris, G. Fratesi, R. Gebauer, U. Gerstmann, C. Gougoussis, A. Kokalj, M. Lazzeri, L. Martin-Samos, N. Marzari, F. Mauri,

- R. Mazzarello, S. Paolini, A. Pasquarello, L. Paulatto, C. Sbraccia, S. Scandolo, G. Sclauzero, A. P. Seitsonen,
- A. Smogunov, P. Umari, and R. M. Wentzcovitch, J. Phys.: Condens. Matter **21**, 395502 (2009).
- [20] J. P. Perdew and A. Zunger, Phys. Rev. B 23, 5048 (1981).
- [21] N. Troullier and J. L. Martins, Phys. Rev. B 43, 1993 (1991).
- [22] D. Vanderbilt, Phys. Rev. B 41, 7892 (1990).
- [23] F. J. DiSalvo and J. V. Waszczak, Phys. Rev. B 23, 457 (1981).
- [24] F. J. DiSalvo and J. V. Waszczak, Phys. Rev. B 22, 4241 (1980).
- [25] A. J. Freeman and R. E. Watson, in *Magnetism*, edited by G. T. Rado (Academic Press, New York, 1965), Vol. IIA.
- [26] T. Ohama, H. Yasuoka, M. Isobe, and Y. Ueda, Phys. Rev. B 59, 3299 (1999).
- [27] J. Kikuchi, H. Yasuoka, Y. Kokubo, and Y. Ueda, J. Phys. Soc. Jpn. 63, 3577 (1994).
- [28] A. Abragam, *The Principles of Nuclear Magnetism*, International Series of Monographs on Physics (Clarendon Press, New York, 1961).
- [29] T. Tsuda, Y. Kitaoka, and H. Yasuoka, Physica B+C 105, 414 (1981).
- [30] A. Narath, Phys. Rev. 162, 320 (1967).
- [31] Y. Yafet and V. Jaccarino, Phys. Rev. 133, A1630 (1964).
- [32] E. R. Andrew and D. P. Tunstall, Proc. Phys. Soc. 78, 1 (1961).
- [33] T. Rega, J. Phys.: Condens. Matter 3, 1871 (1991).
- [34] A. M. Clogston, V. Jaccarino, and Y. Yafet, Phys. Rev. 134, A650 (1964).
- [35] R. E. Walstedt, *The NMR Probe of High-T_c Materials* (Springer, Berlin, 2008), p. 228.
- [36] H. Myron, Physica B+C 99, 243 (1980).
- [37] P. Christmann, B. K. Meyer, J. Kreissl, R. Schwarz, and K. W. Benz, Phys. Rev. B 53, 3634 (1996).
- [38] R. N. Schwartz, M. Ziari, and S. Trivedi, Phys. Rev. B 49, 5274 (1994).

- [39] T. Jin-no, Y. Shimizu, M. Itoh, S. Niitaka, and H. Takagi, Phys. Rev. B 87, 075135 (2013).
- [40] W. Z. Hu, G. Li, J. Yan, H. H. Wen, G. Wu, X. H. Chen, and N. L. Wang, Phys. Rev. B 76, 045103 (2007).
- [41] G. Li, W. Z. Hu, D. Qian, D. Hsieh, M. Z. Hasan, E. Morosan, R. J. Cava, and N. L. Wang, Phys. Rev. Lett. 99, 027404 (2007).
- [42] L. V. Gasparov, K. G. Brown, A. C. Wint, D. B. Tanner, H. Berger, G. Margaritondo, R. Gaál, and L. Forró, Phys. Rev. B 66, 094301 (2002).
- [43] M. Leroux, M. Le Tacon, M. Calandra, L. Cario, M.-A. Méasson, P. Diener, E. Borrissenko, A. Bosak, and P. Rodière, Phys. Rev. B 86, 155125 (2012).
- [44] G. Eda, T. Fujita, H. Yamaguchi, D. Voiry, M. Chen, and M. Chhowalla, ACS Nano 6, 7311 (2012).
- [45] A. Gauzzi, E. Gilioli, F. Licci, M. Marezio, G. Calestani, C. Franchini, and S. Massidda, Phys. Rev. B 66, 085106 (2002).
- [46] M. N. Ali, H. Ji, D. Hirai, M. Fuccillo, and R. Cava, J. Solid State Chem. 202, 77 (2013).
- [47] Y. Yasui, T. Nishikawa, Y. Kobayashi, M. Sato, T. Nishioka, and M. Kontani, J. Phys. Soc. Jpn. 64, 3890 (1995).
- [48] L. Cario, J. Rouxel, A. Meerschaut, Y. Moelo, B. Corraze, and O. Chauvet, J. Phys.: Condens. Matter 11, 2887 (1999).
- [49] A. V. Powell, D. C. Colgan, and P. Vaqueiro, J. Mater. Chem. 9, 485 (1999).
- [50] Y. Klein, H. Moutaabbid, A. Soyer, M. D'Astuto, G. Rousse, J. Vigneron, A. Etcheberry, and A. Gauzzi, J. Solid State Chem. 184, 2333 (2011).
- [51] Y. Klein, H. Moutaabbid, N. Boubiche, A. Soyer, G. Rousse, and A. Gauzzi, J. Phys.: Condens. Matter 25, 435604 (2013).
- [52] W. Geertsma, C. Haas, R. Huisman, and F. Jellinek, Solid State Commun. 10, 75 (1972).
- [53] B. Burk, R. E. Thomson, A. Zettl, and J. Clarke, Phys. Rev. Lett. 66, 3040 (1991).
- [54] K. Nakanishi and H. Shiba, J. Phys. Soc. Jpn. 53, 1103 (1984).