Strong anisotropy in the electromagnetic properties of Na₂Ti₂ X_2O (X = As, Sb) crystals

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eceived 14 July 2015, Tevised manuscript Teceived 5 October 2015, published 24 October 2015)

Na₂Ti₂X₂O (X = As, Sb) crystals have been grown from the flux method. X-ray diffraction characterization revealed an anti-K₂NiF₄-type layered structure (tetragonal, space group *I4/mmm*) for both compounds. Magnetic susceptibility ($\chi(T)$) and electrical resistivity ($\rho(T)$) measurements revealed major kinks at ~115 K (T_{s1}) and ~320 K (T_{s2}) for Na₂Ti₂Sb₂O and Na₂Ti₂As₂O, respectively, signifying possibly the opening of density wave gaps. Both Na₂Ti₂Sb₂O and Na₂Ti₂As₂O showed remarkably strong anisotropy in their electromagnetic transport properties, and values of γ_{ρ} (ρ_{c}/ρ_{ab}) even reached ~140 and ~430, respectively, being much larger than that of iron pnictide BaFe₂As₂ ($\gamma_{\rho} \sim 2$ -5). The γ_{ρ} of Na₂Ti₂Sb₂O changed slightly with cooling, though a small drop at T_{s1} occurred. In contrast, the γ_{ρ} of Na₂Ti₂As₂O changed strikingly by exhibiting not only a small change at T_{s2} but also a sudden decrease of ~50 K, reduced nearly 1/3. Specific heat measurement indicated that Na₂Ti₂Sb₂O was only partially gapped with $\gamma_1 = 4.1$ mJ mol⁻¹ K⁻², though a long-range order was established at Ts_1 , while Na₂Ti₂As₂O was fully gapped. The remarkably strong electromagnetic anisotropy revealed in Na₂Ti₂X₂O suggests the crucial role of the TiO₂X₄ layer for the transport properties of layered titanium oxypnictides.

DOI: 10.1103/PhysRevB.88.144513

PACS number(s): 74.70.Xa, 72.15.-v, 74.25.F-, 75.30.Gw

I. INTRODUCTION

To date, high- T_c superconductivity only appears in layered materials such as cuprates and iron arsenides, in which the strongly correlated electrons in two-dimensional (2D) electron system dominate the transport properties and hence lead to a notable feature of strong electronic anisotropy. Electronic anisotropy imposed in the layered structure can reach a remarkably high degree, with the out-of-plane *c*-axis resistivity $\rho_{\rm c}(T)$ being orders of magnitude larger than in-plane resistivity $\rho_{ab}(T)$ in the normal state.^{1,2} Superconductivity can even survive in cuprates when the 2D CuO2 plane is essentially decoupled,³⁻⁶ thus suggestive of a pivotal role of the 2D CuO₂ layer in giving rise to high- T_c superconductivity. Furthermore, unconventional superconductivity, including particularly high- T_c superconductivity, usually appears neighboring other orders of ground states, like charge-density wave (CDW),^{7,8} spin-density wave (SDW),^{9,10} or magnetism (anti- or ferromagnetism).¹¹ The presence of competing phenomena also offers important clues for understanding the superconductivity. The superconductivity in iron arsenides in proximity to magnetism, for instance, has been widely believed to gain energy from the spin fluctuations to have a high T_c .^{12–14} Thus, a system with both features of a 2D electron system imposed in a layered structure and SDW, CDW, or magnetism would be of interest in serving as an excellent platform for exploration of unusual superconductivity. In this regard, layered titanium oxypnictides with the formula $LnTi_2X_2O$ $(Ln = Ba, Na_2, (SrF)_2, (SmO)_2; X = As, Sb)$ naturally deserve more attention, because they are promising in regard to meeting both requirements.¹⁵⁻²⁴ Recently, superconductivity with a T_c of ~1.2 K was discovered in BaTi₂Sb₂O.²¹ The superconductivity appears after a precursory phase transition at \sim 50 K. However, conclusions on the nature of the phase transition and its correlation with the superconductivity have not come to a consensus. Associated with the magnetic susceptibility and resistivity anomalies, a structural distortion in the Ti₂Sb₂O layer, rather than long-range magnetic ordering, was detected by a very early powder neutron diffraction measurement on Na₂Ti₂Sb₂O.²⁵ Density functional theory (DFT) calculations indicated a density wave (DW) instability due to the nesting of the nearly 2D Fermi surface.¹⁶ Recently, the arguments have been converged to the types of DW states, i.e., CDW or SDW.²⁶ Subedi performed DFT calculations on BaTi₂Sb₂O, and the results suggested a CDW instability.²⁷ Being consistent with this calculation, a nuclear magnetic resonance and nuclear quadrupole resonance measurements on BaTi₂Sb₂O did not reveal any signature of magnetic order at low temperature and indicated a possible commensurate CDW ordering.²⁸ However, more recent DFT calculations on Na₂Ti₂X₂O indicated SDW instabilities.^{29–31}

Na₂Ti₂Sb₂O and Na₂Ti₂As₂O exhibit phase transitions at different temperatures. For Na2Ti2Sb2O it is ~115 K (T_{s1}) ,^{18,25} while for Na₂Ti₂As₂O it takes place at a much higher temperature of ~320 K (T_{s2}) .^{15,18} Na₂Ti₂X₂O has a similar structure as that of La₂CuO₄, which crystallizes in a K₂NiF₄type layered structure,¹¹ but the Ti₂O square lattice, formed by edge-shared TiO₂X₄ octahedra, with Ti³⁺(3d¹) being coordinated by two oxide anions and four pnictide anions, adopts an anticonfiguration to the CuO₂ square lattice in LaCu₂O₄.²⁵ In this anti-K₂NiF₄-type layered structure (space group I4/mmm), the edge-shared TiO₂X₄ octahedra are alternatively stacked and interspersed by double Na⁺ layers along the c axis. In the electronic structure of $Na_2Ti_2X_2O_1$, the Ti 3d orbitals, being hybridized with the X p orbitals in the layer, are mainly involved in the multiband structure at the Fermi level, and the TiO_2X_4 layers are believed to dominate the transport properties;²⁷ thus, an electronic anisotropy would be expected. Considering the many similarities between $Na_2Ti_2X_2O$ and iron-based superconductors, study on electronic anisotropy of them would be helpful for further understanding the role of the 2D electron system for the transport properties. Moreover, aiming at unveiling the nature of the phase transition in Na₂Ti₂ X_2O , electromagnetic data on a single crystal are also desirable. Though crystal growth of NaTi₂Sb₂O was already reported,²⁵ detailed electromagnetic characterizations have so far been absent. We have systematically investigated Na₂Ti₂Sb₂O and Na₂Ti₂As₂O single crystals by means of x-ray diffraction (XRD), magnetic susceptibility, isothermal magnetization, electrical resistivity, and specific heat measurements. Electromagnetic anisotropy of the crystals is our primary focus.

II. EXPERIMENTAL DETAILS

Na2Ti2As2O and Na2Ti2Sb2O single crystals were grown using the flux method. NaAs and NaSb flux were prepared beforehand by separately reacting the stoichiometric mixture of the Na (2N) chunk and the As (4N) or Sb (4N) powder in evacuated and sealed quartz tubes. Mixture of Na and As was slowly heated to 400 °C, while Na and Sb was heated to 350 °C, and both were reacted at the temperatures for 12 h. High-purity starting materials of Ti (4N) and TiO₂ powder (5N) were mixed together with flux in a molar ratio of 3:1:20. They were put into Al₂O₃ ampoules, and the assembly was sealed into Ta tubes under argon gas. The Ta tubes were then sealed in evacuated quartz tubes and afterward heated in a furnace. For Na₂Ti₂As₂O, the reaction was at 1050 °C for 5 h, followed by a slow cooling to $750 \,^{\circ}$ C at a rate of $3 \,^{\circ}$ C/h. For Na₂Ti₂Sb₂O, the reaction was at 1120 °C for 5 h; it was cooled to $820 \,^{\circ}$ C at a rate of $3 \,^{\circ}$ C/h. Both tubes were then cooled quickly to 600 °C, following the slow cooling process, and inverted and spun immediately in a centrifugal separator at this temperature to remove excess flux. After separating the flux, platelike single crystals with a typical dimension of $\sim 3 \times 3 \times 0.1 \text{ mm}^3$ were left.

Phase identification of the crystals was performed at room temperature on a Shimadzu XRD-7000 x-ray diffractometer using Cu K α 1(λ = 1.540 596 Å). XRD data were collected in a 2θ range of 5–80° with a scan step interval of 0.02°. The anisotropic resistivity was measured using a generalization of the Montgomery method.³² The typical dimensions of crystals used to determine the resistivity were $\sim 1.2 \times 1.0 \times 0.07 \text{ mm}^3$ for $Na_2Ti_2Sb_2O$ and $1.7 \times 1.8 \times 0.06$ mm³ for $Na_2Ti_2As_2O$, with the shortest being the lattice constant c. The measurement was conducted by cooling samples in a physical properties measurement system from Quantum Design (QD). Magnetic susceptibility (χ) for both the $H \parallel ab$ plane and the $H \parallel c$ axis was measured between 2 and 400 K under zero-field-cooling (ZFC) and field-cooling (FC) conditions in a superconducting quantum interference device-vibrating sample magnetometer from QD. A magnetic field (H) of 50 kOe was employed. Specific heat (C_p) was measured on the same apparatus by a time-relaxation method between 2 and 300 K. About 3.6-mg crystals of Na₂Ti₂Sb₂O and 2.0 mg of Na₂Ti₂As₂O were selected for use.

III. RESULTS AND DISCUSSION

Typical single-crystal XRD patterns for Na₂Ti₂As₂O and Na₂Ti₂Sb₂O are presented in Fig. 1. A preferred orientation toward $[0 \ 0 \ 2n]$ (*n* is an integer) is observed, though very small peaks probably arising from other orientations or some unidentified impurities are also detected. This indicates that the *c* axis of the crystals synthesized in this paper is basically perpendicular to the crystal plane (*ab* plane). In addition, the good formation of the expected tetragonal



FIG. 1. Room temperature XRD patterns of the $Na_2Ti_2X_2O$ (X = As, Sb) crystals with the main Bragg reflection being indexed, showing main *c*-axis orientation with very sharp peaks.

crystal structure (space group I4/mmm) was confirmed by indexing the peaks using the tetragonal primitive unit cell. The calculated lattice constant for Na₂Ti₂As₂O is c = 15.2835 Å, and for Na₂Ti₂Sb₂O, it is c = 16.5847 Å; these are roughly consistent with the values reported previously.^{18,20} The XRD characterization thus suggests the reliability of the crystal quality.

The $\chi(T)$ measured for $H || c (\chi_c)$ and $H || ab (\chi_{ab})$ of Na₂Ti₂Sb₂O and Na₂Ti₂As₂O crystals, respectively, are depicted separately in Figs. 2(a) and 3(a). Even at a first glance, common features between the two compounds are strikingly displayed. First, χ_c is much larger than χ_{ab} . Second, the ZFC and FC data in both directions show no difference within measured temperature range.



FIG. 2. (Color online) (a) The magnetic susceptibility χ vs *T* of Na₂Ti₂Sb₂O measured for $H \parallel c (\chi_c)$ and $H \parallel ab (\chi_{ab})$ in 50 kOe. (b) The anisotropic resistivity ρ vs temperature *T* measured on a single crystal in a zero magnetic field.



FIG. 3. (Color online) (a) The anisotropic magnetic susceptibility χ vs *T* of Na₂Ti₂As₂O measured in 50 kOe. (b) The anisotropic electrical resistivity ρ vs temperature *T* measured in a zero magnetic field.

Third, sudden drops of $\chi(T)$ for Na₂Ti₂Sb₂O and Na₂Ti₂As₂O in both directions starting at T_{s1} (~115 K) and T_{s2} (~320 K) are strikingly displayed, being in agreement to previously reported phase transition temperatures for them. The drop of χ_{ab} for Na₂Ti₂Sb₂O is apparently much sharper than that for Na₂Ti₂As₂O. Moreover, χ_c of the two crystals decreases monotonically below T_{s1} and T_{s2} , while χ_{ab} first shows a decrease to a local minimum of ~60 and ~150 K for Na₂Ti₂Sb₂O and Na₂Ti₂As₂O, respectively, and then starts to rise. The rise became steep at ~20 and ~60 K for Na₂Ti₂Sb₂O and Na₂Ti₂As₂O, respectively, with Curie-Weiss-like tails on χ_{ab} at low temperature. The evaluated effective moments for



FIG. 4. (Color online) Temperature-dependent heat capacity C_p of Na₂Ti₂Sb₂O. The upper left inset indicates an expansion of the data in the vicinity of the peak. The lower right inset indicates the lowest temperature limit of the data in the plot of C_p/T vs T^2 .

the tails are $\sim 0.04 \ \mu_B/f.u.$ and $0.17 \ \mu_B/f.u.$ for Na₂Ti₂Sb₂O and Na₂Ti₂As₂O, respectively.

Finally, in the overall profiles of $\chi(T)$ for both Na₂Ti₂Sb₂O and Na₂Ti₂As₂O, magnetic anisotropies with respect to the two directions of $H \parallel ab$ and $H \parallel c$ are apparently gradually weakened below T_{s1} and T_{s2} , thus indicating a close relation between the DW-type gap formation and the magnetic anisotropy.

The main panels of Figs. 2(b) and 3(b) present the temperature dependence of resistivity $\rho(T)$ for Na₂Ti₂Sb₂O and Na₂Ti₂As₂O, respectively, measured for two resistivity components ρ_c and ρ_{ab} on a single sample. The common features between them are as follows: (1) ρ_c and ρ_{ab} have a similar profile, (2) ρ_c is much larger than ρ_{ab} over the measured temperature range, and (3) clear metal-insulator transitions (MITs) at T_{s1} and T_{s2} for Na₂Ti₂Sb₂O and Na₂Ti₂As₂O are visible. Nevertheless, some differences are also clearly exhibited. First, for Na₂Ti₂Sb₂O, after the very sharp MIT at T_{s1} , an insulator-metal transition appears at nearly the same temperature at T_{s1} on both ρ_c and ρ_{ab} , and the metallic conductions then show slight upturns below ~ 20 K. The slight upturn is possibly induced by a weak electron localization effect. For Na2Ti2As2O, two shoulderlike features appear on both ρ_c and ρ_{ab} . The first part is between T_{s2} and 400 K before the MIT due to a poorly metallic conduction, and the second part is between ~ 60 and ~ 150 K due to a poorly insulating conduction. Below ~50 K, the upturns of both ρ_c and ρ_{ab} are more prominent than those observed for Na2Ti2Sb2O. These characterized temperatures for $\rho(T)$ coincide well with those derived from $\chi(T)$, indicating an intimate relationship between magnetic and electronic correlations. In contrast to Na₂Ti₂Sb₂O being primarily metallic except for the very weak low-temperature upturn, Na₂Ti₂As₂O remains insulating after the MIT.

Second, seen from the plotted $\rho_c/\rho_{ab}(\gamma_\rho)$ ratio as a function of temperature by the inset in Fig. 3(a) for Na₂Ti₂Sb₂O, a slightly drop of γ_{ρ} that occurred at ~115 K could be observed, from ~140 to ~135; γ_{ρ} then increases slightly with cooling. The values of γ_{ρ} for Na₂Ti₂Sb₂O are quite large compared with the typical value for iron pncitides, $\sim 2-5$ for BaFe₂As₂, indicating a highly anisotropic electronic nature of Na₂Ti₂Sb₂O. For Na₂Ti₂As₂O, γ_{ρ} is higher than 400 above 300 K, indicative of a remarkably anisotropic electronic structure. A rapid drop below 50 K from \sim 380 at 50 K to \sim 264 at 2 K, rather than at T_{s2} , is strikingly displayed, and the change at T_{s2} is hard to identify due to the data quality. Nevertheless, the γ_{ρ} values are remarkably larger than those of Na₂Ti₂Sb₂O. The sharp drop of γ_{ρ} below 50 K for Na₂Ti₂As₂O implies that the band structure is reconstructed again. It seems that the compound experiences another phase transition below 50 K. Subtle features seem to be present in the magnetization and specific heat measurement near 40 K, but this issue needs further exploration.

Figures 4 and 5 show the specific heat data of Na₂Ti₂Sb₂O and Na₂Ti₂As₂O, respectively. For Na₂Ti₂Sb₂O, a lambdalike kink at T_{s1} with a sharp peak is visible, indicating the establishment of a long-range order due to the DW-type gap opening. For Na₂Ti₂As₂O, we failed to detect the possible long-range order at T_{s2} because of technical reasons. Over the measured temperature range from 2 to 300 K, no clear anomaly signifying transitions is visible in $C_p(T)$. As shown in the insets



FIG. 5. (Color online) Temperature dependent heat capacity C_p of Na₂Ti₂As₂O. The inset shows the C_p/T vs T^2 plot of the data measured in both 0- and 70-kOe magnetic fields.

in Fig. 3(a) and 3(b), the low-temperature portions of the data acquired from Na₂Ti₂Sb₂O and Na₂Ti₂As₂O below 20 K were preliminary analyzed using the approximated Debye model $C_p/T = \gamma + 2.4\pi^4 r N_A k_B (1/T_D^3)T^2$, where γ is a constant term, r is the number of atoms per formula unit, k_B is the Boltzmann constant, and T_D represents the Debye temperature. The analysis gave rise to a γ value of 4.1 mJ mol⁻¹ K⁻² and a T_D of 239 K for Na₂Ti₂Sb₂O, as well as to a γ value of 0 mJ mol⁻¹ K⁻² and a T_D of 218 K for Na₂Ti₂As₂O. These data indicate that finite densities of states are present at the Fermi level for Na₂Ti₂Sb₂O while Na₂Ti₂As₂O is fully gapped. In the presence of a magnetic field of 70 kOe, the results change slightly; values of γ of 4.07 mJ mol⁻¹ K⁻² and 0 mJ mol⁻¹ K⁻² for Na₂Ti₂Sb₂O and Na₂Ti₂As₂O, respectively, indicate small magnetic contributions to the low-temperature $C_p(T)$.

The present paper on single-crystal samples revealed remarkably strong electrical anisotropy for Na₂Ti₂ X_2O . In particular, the Na₂Ti₂As₂O compound seems to possess multiple phase transitions and has more complex properties. Provided that the phase transitions in Na₂Ti₂ X_2O are of the SDW type promoted by the nesting between electron and hole pockets, analogous to that in iron pnictides,^{9,10} it would be of high possibility in producing exotic behaviors, including superconductivity through suppressing the SDW via carrier doping, or applying pressure, as did in iron pnictides.^{33–36} Based on our present results, it is hard to distinguish the exact type of DW. Measurements using more sensitive experimental probes on single crystals, like muon spin rotation or neutron diffraction, would hopefully give more in-depth insights. Moreover, the differences between Na₂Ti₂Sb₂O and Na₂Ti₂As₂O, including the different onset temperatures of phase transitions and electromagnetic behaviors, are possibly ascribed to the different energy levels of Sb and As *p* orbitals, because the only difference in their electronic structures is the hybridization states between the *p* orbitals of *X* and the 3*d* orbitals of Ti.^{21,24} Recalling that the SDW temperature of *Ln*FeAsO is nearly robust to the change of rare earth elements *Ln* from La to Sm across a series of other *Ln*,³⁷ whereas it is more fragile to element replacement on an Fe or As site,^{33–35} the different phase transition temperatures between Na₂Ti₂Sb₂O and Na₂Ti₂As₂O indicate the important role of *X* in Na₂Ti₂X₂O as that of Fe or As in iron arsenides.

IV. CONCLUSIONS

Electromagnetic characterizations on Na2Ti2Sb2O and Na₂Ti₂As₂O crystals revealed remarkably strong anisotropy, with the electrical resistivity anisotropy with respect to $H \| c$ and $H \| ab$ directions even reaching the degrees of ~ 140 and ~430 for Na₂Ti₂Sb₂O and Na₂Ti₂As₂O, respectively, which are much higher than those for iron pnictides. Phase transition temperatures derived from the electromagnetic measurements coincide well with each other for Na2Ti2Sb2O and Na₂Ti₂As₂O, respectively. The specific heat measurements revealed that Na₂Ti₂Sb₂O is partially gapped and Na₂Ti₂As₂O is fully gapped, and the phase transitions in them are likely less magnetic. However, as a possibly analogue to that in iron pnictides, the applied magnetic field of 70 kOe is too small to shift a DW anomaly. It needs a further check using higher magnetic fields. Unraveling the nature of the phase transition would be a crucial gradient toward understanding the unusual superconductivity, as well as to acquiring generic knowledge of the correlation between the superconductivity and a DW state in both layered titanium oxypnictide and iron pnictides. Study of this class of material thus supports continuous efforts.

ACKNOWLEDGMENTS

This paper was supported in part by the Ministry of Science and Technology of China (973 Projects No. 2011CB921701 and No. 2011CBA00110), National Natural Science Foundation of China (Grants No. 11274367 and No. 11120101003), and Chinese Academy of Sciences.

- ¹T. Ito, H. Takagi, T. Ishibashi, and S. Uchida, Nature **350**, 596 (1991).
- ²K. Takenaka, K. Mizuhashi, H. Takagi, and S. Uchida, Phys. Rev. B **50**, 6534 (1994).
- ³D. N. Basov and T. Timusk, Rev. Mod. Phys. 77, 721 (2005).
- ⁴A. Gozar, G. Logvenov, L. F. Kourkoutis, A. T. Bollinger, L. A. Giannuzzi, D. A. Muller, and I. Bozovic, Nature (London) **455**, 782 (2008).
- ⁵A. A. Schafgans, C. C. Homes, G. D. Gu, S. Komiya, Y. Ando, and D. N. Basov, Phys. Rev. B 82, 100505(R) (2010).
- ⁶M. R. Norman, Science **332**, 196 (2011).

- ⁷R. M. Fleming, L. W. ter Haar, and F. J. DiSalvo, Phys. Rev. B **35**, 5388 (1987).
- ⁸E. Morosan, H. W. Zandbergen, B. S. Dennis, J. W. G. Bos, Y. Onose, T. Klimczuk, A. P. Ramirez, N. P. Ong, and R. J. Cava, Nat. Phys. **2**, 544 (2006).
- ⁹J. Dong, H. J. Zhang, G. Xu, Z. Li, G. Li, W. Z. Hu, D. Wu, G. F. Chen, X. Dai, J. L. Luo, Z. Fang, and N. L. Wang, Europhys. Lett. **83**, 27006 (2008).
- ¹⁰C. de la Cruz, Q. Huang, J. W. Lynn, J. Li, W. Ratcliff II, J. L. Zarestky, H. A. Mook, G. F. Chen, J. L. Luo, N. L. Wang, and P. C. Dai, Nature (London) **453**, 899 (2008).

- ¹¹J. G. Bednorz and K. A. Müller, Z. Phys. B 64, 189 (1986).
- ¹²I. I. Mazin, D. J. Singh, M. D. Johannes, and M. H. Du, Phys. Rev. Lett. **101**, 057003 (2008).
- ¹³K. Kuroki, S. Onari, R. Arita, H. Usui, Y. Tanaka, H. Kontani, and H. Aoki, Phys. Rev. Lett. **101**, 087004 (2008).
- ¹⁴V. Cvetkovic and Z. Tesanovic, Europhys. Lett. **85**, 37002 (2009).
- ¹⁵E. A. Axtell III, T. Ozawa, S. M. Kauzlarich, and R. R. P. Singh, J. Solid State Chem. **134**, 423 (1997).
- ¹⁶W. E. Pickett, Phys. Rev. B **58**, 4335 (1998).
- ¹⁷T. C. Ozawa, S. M. Kauzlarich, M. Bieringer, and J. E. Greedan, Chem. Mater. **13**, 1804 (2001).
- ¹⁸R. H. Liu, D. Tan, Y. A. Song, Q. J. Li, Y. J. Yan, J. J. Ying, Y. L. Xie, X. F. Wang, and X. H. Chen, Phys. Rev. B **80**, 144516 (2009).
- ¹⁹X. F. Wang, Y. J. Yan, J. J. Ying, J. Q. Li, M. Zhang, N. Xu, and X. H. Chen, J. Phys. Condens. Matter **22**, 075702 (2010).
- ²⁰R. H. Liu, Y. A. Song, Q. J. Li, J. J. Ying, Y. J. Yan, Y. He, and X. H. Chen, Chem. Mater. **22**, 1503 (2010).
- ²¹T. Yajima, K. Nakano, F. Takeiri, T. Ono, Y. Hosokoshi, Y. Matsushita, J. Heister, Y. Kobayashi, and H. Kageyama, J. Phys. Soc. Jpn. 81, 103706 (2012).
- ²²P. Doan, M. Gooch, Z. Tang, B. Lorenz, A. Moeller, J. Tapp, P. C. W. Chu, and A. M. Guloy, J. Am. Chem. Soc. **134**, 16520 (2012).
- ²³H.-F. Zhai, W.-H. Jiao, Y.-L. Sun, J.-K. Bao, H. Jiang, X.-J. Yang, Z.-T. Tang, Q. Tao, X.-F. Xu, Y.-K. Li, C. Cao, J.-H. Dai, Z.-A. Xu, and G. H. Cao, Phys. Rev. B 87, 100502(R) (2013).

- ²⁴T. Yajima, K. Nakano, F. Takeiri, J. Heister, T. Yamamoto, Y. Kobayashi, and H. Kageyama, J. Phys. Soc. Jpn. 82, 013703 (2013).
- ²⁵T. C. Ozawa, T. Pantoja, E. A. Axtell III, S. M. Kauzlarich, J. E. Greedan, M. Bieringer, and J. W. Richardson, Jr., J. Solid State Chem. **153**, 275 (2000).
- ²⁶Y. Huang, H. P. Wang, W. D. Wang, Y. G. Shi, and N. L. Wang, Phys. Rev. B 87, 100507(R) (2013).
- ²⁷A. Subedi, Phys. Rev. B 87, 054506 (2013).
- ²⁸S. Kitagawa, K. Ishida, K. Nakano, T. Yajima, and H. Kageyama, Phys. Rev. B **87**, 060510(R) (2013).
- ²⁹H. Jiang, Y. L. Sun, J. H. Dai, G. H. Cao, and C. Cao, arXiv:1207.6705.
- ³⁰D. J. Singh, New J. Phys. 14, 123003 (2012).
- ³¹X. W. Yan and Z. Y. Lu, J. Phys.: Condens. Matter **25**, 365501 (2013).
- ³²H. C. Montgomery, J. Appl. Phys. **42**, 2971 (1971).
- ³³M. Rotter, M. Tegel, and D. Johrendt, Phys. Rev. Lett. **101**, 107006 (2008).
- ³⁴A. S. Sefat, R. Jin, M. A. McGuire, B. C. Sales, D. J. Singh, and D. Mandrus, Phys. Rev. Lett. **101**, 117004 (2008).
- ³⁵Z. Ren, Q. Tao, S. Jiang, C. M. Feng, C. Wang, J. H. Dai, G. H. Cao, and Z. A. Xu, Phys. Rev. Lett. **102**, 137002 (2009).
- ³⁶H. Okada, K. Igawa, H. Takahashi, Y. Kamihara, M. Hirano, H. Hosono, K. Matsubayashi, and Y. Uwatoko, J. Phys. Soc. Jpn. 77, 113712 (2008).
- ³⁷K. Ishida, Y. Nakai, and H. Hosono, J. Phys. Soc. Jpn. **78**, 062001 (2009).