Dependence of spin-pumping spin Hall effect measurements on layer thicknesses and stacking order

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Voltages generated from inverse spin Hall and anisotropic magnetoresistance effects via spin pumping in ferromagnetic (F)/nonmagnetic (N) bilayers are investigated by means of a broadband ferromagnetic resonance approach. Varying the nonmagnetic layer thickness enables the determination of the spin diffusion length in Pd of 5.5 \pm 0.5 nm. We also observe a systematic change of the voltage line shape when reversing the stacking order of the F/N bilayer, which is qualitatively consistent with expectations from spin Hall effects. However, even after independent calibration of the precession angle, systematic quantitative discrepancies in analyzing the data with spin Hall effects remain.

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I. INTRODUCTION

Spin Hall effects, which occur in nonmagnetic metals due to spin-orbit coupling, give rise to the interconversion of spin and charge currents. $]^{-4}$ The transverse nature of these effects opens interesting possibilities for magnetic memory and logic devices. This is due to the possibility of generating large net spin currents in thin-film geometries, 5.6 and connecting magnetization dynamics in ferromagnetic insulators with charge currents.^{[7](#page-6-0)} Recent theoretical^{8–11} and experimental^{[5,6,12–20](#page-6-0)} work has identified relevant materials for spintronic applications based on spin Hall effects. Two experimental approaches have been developed to quantitatively investigate spin Hall effects: nonlocal injection inlateral spin valve structures, 2^{1-24} and electrically detected ferromagnetic resonance (FMR) in ferromagnetic (F)/nonmagnetic metallic (N) bilayers, $14,15,17,18$ which is of interest in the present article. Large discrepancies exist among experiments to quantify the magnitude of the spin Hall angle, which describes the conversion efficiency between spin and charge currents.[4](#page-6-0) Electrically detected FMR avoids complications of complex current flow patterns that appear in lateral spin valves that utilize metals with short spin diffusion lengths. However, the analysis of the FMR line shape and, thus, the quantification of the spin Hall angle and conductivity remains unclear.

The difficulty comes from a superposition of different contributing mechanisms, such as anisotropic magnetoresistance (AMR), anomalous Hall effect (AHE), spin pumping-inverse spin Hall effect (SHE), and spin transfer torque (STT). Each can give rise to a voltage as a function of applied magnetic field, and the voltages can yield symmetric or antisymmetric Lorentzian signatures depending on the measurement geometry (which includes the sample geometry, as well as applied field direction, and excitation field orientation). Another issue comes from the ability to unambiguously determine the relative phase between the electric rf currents in the sample and the magnetization dynamics, which may be both magnetic field and current driven. 24 In some cases this can obscure the electrically detected FMR line shape. The issue of phase mixing is complex, as it appears to vary drastically with the sample geometry and frequency. Additional complications may arise from a nonuniformity of the microwave power along a coplanar waveguide, 26 which can lead to systematic error in the analysis of the line shape. Therefore the choice of a suitable geometry that addresses the effect of interest is crucial.

We use a geometry that limits both the STT effect and the phase difference between the electric and magnetic dynamic fields. We report a study of the electric detection of inverse spin Hall effects from spin pumping in permalloy ($Py =$ $Ni₈₀Fe₂₀)/N$ bilayers (N = Pd, Pt, Au) for different metal thicknesses and stack orders of the F/N bilayer. The paper is organized as follows. After discussing general experimental details, we present the metal thickness dependence of the line shape and show how it enables the determination of the spin diffusion length of the metal. Next, we focus on the change of the line shape when reversing the stacking order of the bilayers. Finally, we show how calibration of the precession cone angle resolves some inconsistencies in the experimental data.

II. EXPERIMENTAL DETAILS

Figure $1(a)$ shows a schematic of the spin pumping inverse SHE experiment. It consists of 20-*μ*m-wide Py/Pt bilayers integrated with a 3-mm-long, 30-*μ*m-wide, and 150 nm-thick coplanar wave-guide (CPW) transmission line made of Au. The bilayer, which is deposited directly on the substrate (undoped GaAs or intrinsic Si with a 300-nm-thick layer of thermally grown $SiO₂$), is separated from the CPW by an 80-nm-thick spacer layer of $SiO₂$ or MgO. Four electrical leads for the electrically detected FMR are connected to the Py/Pt bilayer, as shown in Fig. $1(b)$. In order to minimize inductively coupled currents in the sample, it is essential to keep these electrical contacts between the central line and the ground plate within the CPW, since samples with contacts away from the CPW have shown complex variation of the phase of the electrically detected FMR with frequency. The sample is then wire bonded into a printed circuit board with a matching CPW and SMA connectors and mounted in an electromagnet, allowing a rotation in the plane of the sample. The relative angle between the transmission line and the applied field must be kept between 5◦ and 70◦, (If it is 0◦, the spin Hall voltage along the line vanishes, and if it is too large, the excitation field becomes parallel to the equilibrium direction and is inefficient.) The FMR is performed in transmission via either

FIG. 1. (Color online) (a) Schematic of the spin pumping spin Hall effect experiment showing the respective polarity (rf input, external field, and dc voltage contacts) used for the measurement. (b) Layout of voltage contacts used for the measurements (sketch not up to scale).

a vector network analyzer (Anritsu 37147C) or an amplitude modulation of the microwave source (Agilent 8257D) and lock-in detection using a diode. We work at constant frequency and sweep the magnetic field through the FMR. The frequency range of the measurement is kept between 3 and 11 GHz, as higher frequencies are no longer transmitted through the wire bonds. We also amplify the microwave power, while staying in the linear response regime, in order to measure a SHE signal in the tens of μ V.

The *rf* current flowing in the waveguide generates an rf magnetic field that excites magnetization dynamics in the permalloy layer. At resonance, the magnetization precession of the Py layer creates a spin accumulation at the F/N interface via the spin pumping mechanism. 27 The spin accumulation diffuses away from the interface, forming a gradient within the thickness of the N layer. This leads, via the spin-orbit interaction, to a charge imbalance that is perpendicular to both the spin polarization P and the direction of the diffusion of the spin current 3

$$
\boldsymbol{J}_c = e\gamma D(\nabla \times \boldsymbol{P}).\tag{1}
$$

Here, *e* is the electron charge, *D* is the electron diffusion coefficient, and γ is the spin Hall angle. Thus the sign of the spin Hall voltage measured along the N layer depends on the spin diffusion direction and, therefore, must change with the relative order of the bilayer stack. The spin Hall voltages expected to have a symmetric Lorentzian line shape, following the evolution of spin pumping via the FMR. This spin Hall voltage is measured separately via either lock-in amplification or with a dc voltmeter.

III. THICKNESS DEPENDENCE

Figure $2(a)$ shows dc voltages measured at 7 GHz, for GaAs/Pd (t_{Pd}) / Py bilayers with various thicknesses t_{Pd} of Pd, while keeping the Py thickness constant at 15 nm. For a singlelayer of Py $(t_{\text{Pd}} = 0)$, where no SHE is expected, the measured voltage exhibits a purely antisymmetric line shape. As the thickness of the N-layer increases, the line shape of the dc voltage acquires a symmetric component that saturates at $t_{\text{Pd}} = 15$ nm. Note that there is a difference in overall amplitude of the signal, which may originate from different rf transmission of the wire bonds for different samples. 25 Therefore a direct comparison of the voltage amplitude between each sample is not straightforward.

The antisymmetric component of the single layer linewidth is attributed to a homodyne AMR effect, $28,29$ which originates from an rf electrical current generated in the sample from either capacitive coupling to rf electric fields in the waveguide or inductive coupling to rf magnetic fields. The magnitude of this homodyne AMR voltage is given by $14,15$

$$
V_{\text{AMR}} = I_{\text{FN}} \Delta R_{\text{AMR}} \frac{\sin 2\theta}{2} \frac{\sin 2\alpha}{2} \cos \phi_0. \tag{2}
$$

 I_{FN} corresponds to the part of the microwave current flowing in the bilayer. To a first approximation, I_{FN} is assumed proportional to the current in the CPW, according to the ratio of the CPW and the bilayer dc resistance. ΔR_{AMR} is the bilayer absolute dc magnetoresistance when changing the magnetization by 90[○] with respect to the current direction. The angles θ , α , and ϕ_0 correspond to the largest value of the precession cone angle, the angle of the applied field with respect to the central line, and the phase difference between the microwave field and the magnetization dynamics, respectively.

The geometry of our sample design, where the voltage contacts are close to the signal line of the CPW, minimizes inductive coupling and results mainly in rf currents in the sample due to capacitive coupling. In other words, the rf currents in the sample and the CPW are in phase with each other, resulting in an antisymmetric Lorentzian voltage signal due to the continuously varying phase ϕ_0 of the magnetization dynamic response with respect to the excitation field (i.e., inphase far below resonance, $\pi/2$ at resonance, and out-of-phase far above resonance).

In contrast, the inverse SHE contribution is independent of the phase of the magnetization dynamics and gives rise to a purely symmetric Lorentzian voltage contribution given $bv^{14,15}$

$$
V_{\text{SHE}} = \frac{\gamma e L E f g_{\text{mix}} \lambda_s \sin \alpha \sin^2 \theta}{\frac{t_N}{\rho_N} + \frac{t_F}{\rho_F}} \tanh\left(\frac{t_N}{2\lambda_s}\right),\tag{3}
$$

where, L, E, f, λ_s and $\rho_N(\rho_F)$ correspond to the length of the bilayer, a correction factor for the ellipticity of the magnetization precession, $2⁹$ the frequency, the spin diffusion length in the normal metal, and the normal metal (ferromagnet) resistivity. The effective spin mixing conductance g_{mix} determines the flow rate of spin pumped into the metal, taking into account the backflow of spin into the ferromagnet. The term $tanh(t_N/2\lambda_s)$ describes the decay of the spin accumulation in the metal away from the F/N interface.

FIG. 2. (Color online) (a) AMR-SHE spectra for (GaAs/*x*Pd/15Py) (*x* = 0, 5, 10, 15 nm) measured at 7 GHz. (b) Thickness dependence of the weight *W* of the antisymmetric component for Pd. (c) Estimation of the spin diffusion length for Pd for all measured frequency.

In the linear response regime, both the inverse SHE and the AMR components have the same power dependence (proportional to the square of the microwave field), so that

the resultant dc voltage is a sum of a symmetric (V_S) and an antisymmetric (V_A) Lorentzian: $V_{dc} = W V_A + (1 - W) V_S$, where *W* represents the weight of the antisymmetric component. When $W = 1$ ($W = 0$), the signal line shape is fully antisymmetric (symmetric).

Figure 2(b) summarizes the metal thickness dependence of the dc voltage line shape measured for $\alpha = 40^{\circ}$, by plotting the weight *W* versus Pd thickness. Only four frequencies $(4, 6, 8, ...)$ and 10 GHz) are shown for the sake of clarity. Starting from a single layer of Py $(t_{\text{Pd}} = 0)$, the line shape of the dc voltage remains antisymmetric $(W = 1)$ throughout the frequency range. This shows, in particular, that for a single ferromagnetic layer, no additional phase difference occurs between the rf current and the rf field, unlike what was suggested recently.^{17,25} As the thickness of Pd increases, the weight *W* decreases until it reaches a minimum value ($t_{\text{Pd}} \sim 15$ nm) where it remains mostly constant. Note that this behavior is qualitatively similar independent of the measurement frequency. For $t_{\text{Pd}} > \lambda_s$, the spin diffusion length, no SHE occurs above λ_s as the spin accumulation has fully relaxed; therefore *W* reaches a plateau. The decrease of the weight with the Pd thickness appears more pronounced at higher frequency. This observation is related to the ellipticity of the magnetization precession, which reduces the spin pumping effect, and thus, lowers the relative amplitude of inverse SHE versus AMR at lower frequency.^{[30](#page-7-0)}

Using Eq. [\(2\),](#page-1-0) the precession angle θ can be directly estimated from the AMR voltage contribution assuming that the precession angle is uniform along the whole length and that the correlation between the rf field h_{rf} from the CPW and the microwave current flowing in the bilayer is well established. However, it was recently shown that these assumptions do not hold, 26 and, as will be shown, these assumptions can lead to large discrepancies in the estimates of the spin Hall angles when reversing the bilayer stacking order. In the limit of small precession angle $θ = h_{rf} cos(α)/ΔH$ ($ΔH$ being the half line width at half maximum), the ratio of the maxima, $V_{\text{SHE}}/V_{\text{AMR}}$, can be written as

$$
\frac{V_{\text{SHE}}}{V_{\text{AMR}}} = \frac{\gamma e \mathcal{E} f g_{\text{mix}} \lambda_s}{R_{\text{CPW}} I_{\text{CPW}} \frac{\Delta R_F}{R_F}} \frac{h_{rf}}{\Delta H} \frac{\rho_F L}{t_F} \tanh\left(\frac{t_N}{2\lambda_s}\right),\qquad(4)
$$

where R_{CPW} and I_{CPW} correspond to the resistance and microwave current in the CPW, and $\Delta R_F/R_F$ is the relative change of magnetoresistance for the ferromagnet (typically $~\sim$ 0.01 for Py). Equation (4) shows that the metal thickness dependence of $V_{\text{SHE}}/V_{\text{AMR}}$ is only contained in the distribution of the spin accumulation tanh($t_N/2\lambda_s$), and that above a certain thickness ($t_N > \sim 3\lambda_s$), the ratio $V_{\text{SHE}}/V_{\text{AMR}}$ becomes independent of the metal thickness. In order to fit the thickness dependence of the weight *W*, which is related to $V_{\text{SHE}}/V_{\text{AMR}}$ by the relation $W = 1/(1 + |V_{\text{SHE}}/V_{\text{AMR}}|)$, we use a function of the form $1/[1 + \text{Atanh}(t_N/2\lambda_s)]$, where *A* is a constant independent of the metal thickness. In this way, we can directly extract λ_s from the fitting, and verify, as shown in Fig. 2(c), that λ_s remains constant for all measured frequencies. This alternative technique yields a value of $\lambda_s = 5.5 \pm 0.5$ nm for Pd, which is somewhat smaller than previously reported values measured at lower temperatures^{[16,](#page-6-0)[31](#page-7-0)} and comparable to other values measured for Pd at room temperature. $32,33$

FIG. 3. (Color online) (a) AMR-SHE spectra for a GaAs/15Py/15Pd sample at 4 and 7 GHz, 40◦ applied field angle and 5 dBm. (b) Measurements under identical conditions for the reverse stack order (GaAs/15Pd/15Py). Frequency dependence of the weight *W* of the antisymmetric component of the AMR-SHE voltage for both stacking orders with Pd(c) and Pt (d).

IV. STACKING ORDER DEPENDENCE

We now look at the change of the dc voltage upon reversing the stacking order of the F/N bilayer. Figures $3(a)$ and 3(b) show dc voltages measured for GaAs/15Py/15Pd and GaAs/15Pd/15Py bilayers, respectively, at 4 and 7 GHz, 40◦ applied field angle, and 5 dBm rf power. Starting from the F/N configuration shown in Fig. $3(a)$ with the same polarity as in Fig. $1(a)$ for the dc contacts, the input port of microwave, and the direction of the applied field, both the anti-symmetric (V_A) and symmetric (V_S) components of the voltage are positive. We keep the same measurement polarity for the reverse configuration (N/F) in Fig. $3(b)$ and observe that only the symmetric component has changed sign. The antisymmetric component, which is understood as a heterodyne AMR effect, keeps the same sign in both configurations, as expected, since it only depends on the relative polarity of the microwave current, the dc contacts, and the field orientation, which are identical in both measurements. The change of sign of the symmetric component is consistent with the cross-product relation of the inverse SHE [see Eq. (1)]. In the F/N configuration, where the Py is underneath the metal, the pure spin current pumped into the Pt diffuses upward and, according to Eq. [\(1\),](#page-1-0) gives rise to a positive inverse spin Hall voltage, whereas in the N/F configuration the pure spin current diffuses downward, and the resultant SHE voltage is negative.

However, in addition to the line shape change given by the sign reversal of the symmetric component, we also notice that the relative amplitudes of the symmetric and antisymmetric components are not identical. The amplitude of V_S with respect to *VA* in the F/N configuration appears larger than in the N/F configuration. This change of the line shape with the stacking order is summarized in Fig. $3(c)$, by showing the weight of the antisymmetric amplitude *W* for the whole frequency range. The value of *W* remains higher in the N/F than in the F/N configuration over the whole frequency range, meaning that the line shape is more antisymmetric for N/F versus F/N. The difference in *W* between F/N and N/F, which is ∼20% at 3 GHz, increases at higher frequency, where it is ∼30%. We also observed similar trends with Pt, which is also shown in Fig. $3(c)$, and where we used the same thicknesses (15 nm Py, 15 nm Pt, and 150 nm of Au for the CPW). The value of *W* in the case of Pt appears slightly lower than in the case of Pd (meaning that the relative amplitude of the symmetric component is larger in the case of Pt). This is expected, as the spin Hall angle is known to be larger for Pt than for $Pd^{14–16}$ $Pd^{14–16}$ $Pd^{14–16}$ However, the difference between the two stacking orders is similar for Pd.

FIG. 4. (Color online) Frequency dependence of *W* for a GaAs/15Py/15Pd (solid symbols) and a GaAs/15Pd/15Py (open symbols). The line shape remains unchanged with applied field angle (a) and rf output power (b).

To further investigate the change in the line shape when changing the stacking order of the F/N bilayer, we verify in Fig. [4](#page-3-0) that the line shape remains unchanged with applied field angle and microwave power, as is expected from the expression of*V*SHE/*V*AMR [see Eq.[\(4\)\]](#page-2-0). We also measured identical devices on a different substrate $Si/SiO₂$ (not shown) and found the same trend, $W(N/F) > W(F/N)$, with a slightly reduced difference between the two configurations. Along with these changes in the line shape between the two configurations, which have been observed systematically on several different samples, we also observed a systematic reduction of the absolute dc AMR as well as M_s in the F/N configuration [see inset Figs. $5(a)$ and $5(b)$]. However, this reduction of the dc AMR, which may be due to a difference in the growth of the Py, whether it is deposited first on the substrate or on the metal layer, does not account for the difference of a factor of two, when estimating the spin Hall angle from the two configurations.

A possible origin of this difference in the line shape could be the additional Oersted field coming from rf currents passing through the metal layer. In the F/N configuration, where both the CPW and the metal sit on top of the Py layer, the two rf fields h_{CPW} and h_N add up, whereas in the N/F configuration, where the Py layer is in between the CPW and the metal layer, the two fields oppose each other. Therefore, according to the expression of $V_{\text{SHE}}/V_{\text{AMR}}$ [see Eq. [\(4\)\]](#page-2-0), the ratio of SHE and AMR amplitude at a given microwave power (I_{CPW}) ought to be larger in the F/N configuration where the net resulting rf field that the Pysees is larger than in the N/F configuration. As it was shown recently, this change of the microwave field in the Py layer between the two stacking orders could also be related to a difference in microwave screening from eddy currents, 34 although the thicknesses of the layers in our samples are much smaller than the skin depth. It appears, therefore, essential to obtain an independent calibration for the microwave fields in the Py.

V. CALIBRATION OF PRECESSION ANGLE

We adopt the AMR-FMR method of Costache *et al.*[35](#page-7-0) to determine the precession angle in our samples independent of assumptions about the rf magnetic fields. The method consists in measuring the reduction of the AMR due to the opening angle of the magnetization at resonance. Similar to the SHE-AMR measurement, we excite the FMR via an amplitude modulation of the microwave power and measure the AMR-FMR with lock-in amplification. The carrier frequency of the modulation, which we vary between 1–50 kHz, does not matter for the measurement. We apply the magnetic field along the central line ($\alpha = 0$), where neither SHE nor heterodyne AMR voltages occur, and apply a constant dc current ($I_{dc} = 10$ to 20 mA) throughout the entire measurement. Prior to the measurement, we allow at least 10 min for the system (sample $+$ circuit board) to reach thermal equilibrium. Figures $5(a)$ and $5(b)$ show AMR-FMR measurements for $SiO₂/15Py/15Pd$ and $SiO₂/15Pd/15Py$ samples, respectively, at 6 GHz for three output powers (5, 8, and 11 dBm). The AMR-FMR measurements, which are sensitive to a voltage change due to modulation of the magnetization dynamics, have a Lorentzian line shape at resonance. Away from the FMR, the absence of magnetization dynamics results in no voltage change. As the

FIG. 5. (Color online) AMR-FMR measurements at 6 GHz and rf output power 5, 8, and 11 dBm for (a) $SiO₂/15Pd/15Py$ (open symbols) and (b) $SiO₂/15Py/15Pd$ (solid symbols). Insets in (a) and (b) show the corresponding dc magnetoresistance with the field applied perpendicular to the long direction (easy axis) of the samples. (c) Frequency dependence of the precession angle at resonance.

field approaches the resonant condition, the angle between the magnetization and the direction of the current starts to increase as the precession angle opens up, resulting in a change of the lock-in voltage equal to: $\Delta V = \Delta R_{\text{AMR}} I_{\text{dc}} \sin^2 \theta$, where ΔR_{AMR} is the maximum change of AMR at 90° angle. The values of the precession angle at resonance *θ*res follow an increase of $\sqrt{2}$ as the power is doubled ($\theta_{res} = h_{rf}/\Delta H$ and *h*_{rf} ∼ \sqrt{P}_{rf}), which confirms the linear response regime of our measurement.

Note that the amplitude of ΔV differs between the F/N [see Fig. $5(a)$] and N/F [see Fig. $5(b)$] configurations, although both were measured under identical conditions of microwave power and dc current. Part of the difference in the amplitude of ΔV comes from a reduction of the AMR effect in the F/N configuration, as shown in the insets of Figs. $5(a)$ and $5(b)$. Furthermore, the difference in the value of θ_{res} between the two configurations also illustrates the difference of microwave transmission. As the samples are connected via wire bonds to the circuit board, the microwave transmission is not only different from sample to sample, but is also frequency dependent and drops continuously throughout the frequency range, especially beyond 11 GHz. The frequency dependence of *θ*re*^s*, as shown in Fig. $5(c)$, indicates that the $SiO₂/15Pd/15Py$ sample has a minimum microwave transmission at 8 GHz, whereas the power transmission remains fairly constant between 7 and 10 GHz for the $SiO₂/15Py/15Pd$ sample. This different behavior may reflect differences of impedance mismatch for different samples, which may result in varying standing-wave patterns for the microwaves in the samples. 26

Considering possible spatial in homogeneities of the microwave field along the sample, calibrating the cone angle with the AMR-FMR method has the advantage that its dependence on the cone angle is identical to that obtained from the SHE voltage [see Eq. (3)], namely, both are proportional to $\sin^2\theta$. In other words, an average value of $\sin^2\theta$ determined via the AMR-FMR measurement contains any nonuniformities of the rf field along the line and also across the wire due to the skin effect.

Finally, only the amplitude of the symmetric component of the SHE-AMR signal is needed for the estimation of the spin Hall angle γ , as we can directly implement the determination of $\sin^2\theta$ via the above calibration. All other parameters of Eq. [\(3\)](#page-1-0) are known except the spin-mixing conductance, whose estimation remains a topic of debate due to complications, such as proximity effects and spin backflow at the interface.^{[33,36,37](#page-7-0)} We use $\lambda_s(Pd) = 5.5$ nm for Pd, as found from the thickness dependence of the line shape.

Although we did not study the thickness dependence of the line shape for Pt or Au, we used λ_s (Pt) = 4 nm and λ_s (Au) $=$ 35 nm as reported in the literature^{16,[31,38](#page-7-0)} and assuming that $\lambda_s(\text{Pt}) < \lambda_s(\text{Pd})$. However, we note that the actual value of the spin diffusion length in Pt is controversial, $39,40$ with reported values ranging from 1–10 nm. Nevertheless, the quantitative analysis of the spin Hall angle based on Eq. [\(3\)](#page-1-0) is only weakly dependent on the actual value of the spin diffusion length, as discussed in Refs. [4](#page-6-0) and [15.](#page-6-0)

The resistivities measured for each sample have a thickness dependence ranging from 42 to 17 $\mu\Omega$ cm between 2.5 and 60 nm for Pd, and are reproducible for identical thicknesses. Other values for our films are ρ (Pt, 15 nm) = 25 $\mu \Omega$ cm, ρ (Au, 30) $n(m) = 5 \mu \Omega \text{ cm}$, and ρ (Py, 15 nm) = 40 $\mu \Omega \text{ cm}$. The frequency dependence of the FMR linewidth presented in Fig. $6(a)$ shows additional damping due to spin pumping for all samples. One can see that the slope of this frequency dependence of the FMR linewidth, which is proportional to the total magnetic damping, is greater for Pt than for Pd, which is also greater than for Au. This increase of the additional damping between Au, Pd, and Pt is in accordance with the spin pumping effect and spin back flow theory that was recently established.³⁹ From the measurement of this additional damping, we estimate the effective spin mixing conductance g_{mix} that takes into account the spin backflow for each metal following the formalism of Tserkovniak *et al.*[27](#page-6-0) As can be observed in Fig. 6(a), we estimate that the spin mixing conductance from the broadening

FIG. 6. (Color online) (a) Frequency dependence of the FMR linewidth showing the additional damping due to spin pumping. (b) Estimation of the product spin mixing conductance \times spin Hall angle for Pt, Pd, and Au following the independent calibration of the cone angle.

of the linewidth still has ∼20% uncertainty, as additional effects, such as magnetic proximity, 36 could also contribute to the FMR linewidth. Therefore we summarize in Fig. 6(b), for each frequency and stacking configuration for Pd, Pt, and Au, the measured products *g*mix*γ* , for which, according to Eq. [\(3\),](#page-1-0) we have better accuracy. We also report in Table I the spin Hall angles for each metal for the corresponding value of

TABLE I. Spin mixing conductance and spin Hall angle reported for the various metal/Py samples measured, with their corresponding resistivity and spin diffusion length.

				Metal ρ ($\mu\Omega$ cm) λ_s (nm) α (10 ⁻²) g_{mix} (10 ¹⁹ m ⁻²) γ (%)	
P_V	40	5 ⁷	0.8 ± 0.1		
Au	5	35	0.9 ± 0.2	0.9 ± 0.2	0.25 ± 0.1
Pd	20		5.5 $1.1 + 0.2$	2.3 ± 0.4	1.2 ± 0.3
Pr	25.			$4\quad 1.2 \pm 0.3 \quad 3.0 \pm 0.6$	2.7 ± 0.5

the spin mixing conductances. The values are $\gamma = 0.012 \pm 1$ 0.002 and $g_{mix} = 2.3 \pm 0.4 \, 10^{19} \, \text{m}^{-2}$ for Pd, $\gamma = 0.027 \pm 1.0$ 0.005 and $g_{\text{mix}} = 3.0 \pm 0.6 \cdot 10^{19} \text{ m}^{-2}$ for Pt, and $\gamma =$ 0.0025 ± 0.0008 and $g_{mix} = 0.9 \pm 0.2$ 10¹⁹ m⁻² for Au.

The AMR-FMR calibration of the cone angle reduced the difference in the γ estimates for the two stacking order configurations. However, our refined analysis systematically shows a 20% larger γ value in the N/F configuration for Pd and Pt. This remaining difference in the estimate of γ is no longer related to a change of rf field, as this problem was addressed by the independent calibration of $\sin^2\theta$. Another possible way to account for this difference could be related to a phase offset of the magnetization dynamics dependent of the stacking order, as was recently demonstrated by Bailey *et al.* via x-ray magnetic circular dichroism.^{[40](#page-7-0)} This change of phase could directly affect the asymmetry of the SHE-AMR line shape.

VI. CONCLUSION

We have demonstrated that the symmetric component of the SHE-AMR line shape initially increases with normal metal thickness and then becomes thickness independent. The line shape change with metal thickness is explained within a previously used framework that takes into account both the homodyne AMR and the inverse SHE. This enables direct determination of the spin diffusion length of the normal metal. Furthermore, we showed that the symmetric component of the SHE-AMR voltage changes sign when the stacking order of the bilayer is reversed, in agreement with the cross product relation of the inverse SHE. In addition, there is a systematic change of the SHE-AMR line shape upon reversal of the normal metal/ferromagnet stacking order. The relative amplitude of the symmetric component was always found to be larger in the case of the substrate/F/N configuration. To a large extent, this systematic difference in the line shape between F/N and N/F can be understood as a difference in the rf magnetic excitation field $h_{\rm rf}$ in the permalloy. This can be accounted for by using an approach based on the change of magnetoresistance at the ferromagnetic resonance that enables the independent estimate of the precession cone angle for each frequency. The final estimate of the spin Hall angle using this independent calibration of the precession angle is therefore improved.

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