

Observation of a phase transition at 55 K in single-crystal $\text{CaCu}_{1.7}\text{As}_2$

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We present the structural, magnetic, thermal and *ab*-plane electronic transport properties of single crystals of $\text{CaCu}_{1.7}\text{As}_2$ grown by the self-flux technique that were investigated by powder x-ray diffraction, magnetic susceptibility χ , isothermal magnetization M , specific heat C_p , and electrical resistivity ρ measurements as a function of temperature T and magnetic field H . X-ray diffraction analysis of crushed crystals at room temperature confirm the collapsed tetragonal ThCr_2Si_2 -type structure with $\sim 15\%$ vacancies on the Cu sites as previously reported, corresponding to the composition $\text{CaCu}_{1.7}\text{As}_2$. The $\chi(T)$ data are diamagnetic, anisotropic, and nearly independent of T . The χ is larger in the *ab* plane than along the *c* axis, as also observed previously for SrCu_2As_2 and for pure and doped BaFe_2As_2 . The $C_p(T)$ and $\rho(T)$ data indicate metallic *sp*-band character. In contrast to the $\chi(T)$ and $C_p(T)$ data that do not show any evidence for phase transitions below 300 K, the $\rho(T)$ data exhibit a sharp decrease on cooling below a temperature $T_i = 54\text{--}56$ K, depending on the crystal. The $\rho(T)$ data show no hysteresis on warming and cooling through T_i and the transition thus appears to be second order. The phase transition may arise from spatial ordering of the vacancies on the Cu sublattice. The T_i is found to be independent of H for $H \leq 8$ T. A positive magnetoresistance is observed below T_i that increases with decreasing T and attains a value in $H = 8.0$ T of 8.7% at $T = 1.8$ K.

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I. INTRODUCTION

The discoveries of high- T_c superconductivity in iron pnictides and chalcogenides motivated many efforts to identify the mechanism^{1–12} and the relationships of these materials to the high- T_c cuprates.^{10–15} In the latter compounds the copper has a $\text{Cu}^{+2} 3d^9$ electronic configuration and carries a local magnetic moment with spin $S = 1/2$ which is retained even in the superconducting regime of the phase diagram. Here we are concerned with the so-called 122-type subclass of the iron arsenide superconductors with the body-centered tetragonal (bct) ThCr_2Si_2 structure with space group $I4/mmm$. If a 122-type Cu-arsenide compound having localized $S = 1/2$ Cu^{+2} moments could be synthesized, such a compound would bridge the gap between the iron arsenide and cuprate families of high- T_c superconductors. With this in mind, we previously reported the physical properties of SrCu_2As_2 and SrCu_2Sb_2 that instead turned out to be nonmagnetic *sp*-band metals,¹⁶ consistent with the theoretical prediction for SrCu_2As_2 by Singh.¹⁷ His electronic structure calculations for BaCu_2As_2 and SrCu_2As_2 indicated that the Cu *3d* bands in these compounds are narrow and lie about 3 eV below the Fermi energy E_F and therefore the Cu *3d* orbitals give very small contributions to the density of states at E_F .¹⁷ From a systematic study of the interlayer *X-X* distance d_{X-X} ($A = \text{Ca}, \text{Sr}, \text{Ba}$; $X = \text{P}, \text{As}$) and the *c/a* ratio for AM_2X_2 *3d* transition metal *M* compounds with the ThCr_2Si_2 structure, we concluded that SrCu_2As_2 crystallizes in a collapsed tetragonal (cT) structure.¹⁶

Recent activity in the Fe-based-superconductor field has focused on the remarkable physical properties of the class of $A_{1-x}\text{Fe}_{2-y}\text{Se}_2$ ($A = \text{alkali metal}$) compounds that are similar to the layered ThCr_2Si_2 -type materials except with substantial numbers of vacancies on the A and Fe sublattices that can become spatially ordered.^{18,19} Depending on the Fe vacancy concentration, superconductivity at temperatures up to ~ 30 K and/or large-moment antiferromagnetism with very high Néel temperatures up to ~ 600 K can be stabilized. Thus, the

influence of transition metal site vacancies on the physical properties of such 122-type compounds is of great current interest.

Here we report powder x-ray diffraction, magnetic susceptibility χ , isothermal magnetization M , specific heat C_p , and *ab*-plane electrical resistivity ρ measurements as a function of temperature T and magnetic field H on single crystals of $\text{CaCu}_{1.7}\text{As}_2$ which was previously reported by Pilchowski and Mewis to form in the bct ThCr_2Si_2 structure with a large ($\sim 15\%$) concentration of vacancies on the Cu site.²⁰ To our knowledge, there are no previous studies of the physical properties of this compound including possible temperature-induced Cu vacancy ordering transitions. Like SrCu_2As_2 , we find that $\text{CaCu}_{1.7}\text{As}_2$ exhibits an anisotropic *T*-independent diamagnetic behavior indicating that the Cu atoms have a nonmagnetic $3d^{10} \text{Cu}^{+1}$ electronic configuration. However, the $\rho(T)$ data for $\text{CaCu}_{1.7}\text{As}_2$ exhibit a phase transition of unknown origin at a transition temperature $T_i = 54\text{--}56$ K, depending on the crystal, as revealed by a sharp well-defined decrease on cooling below T_i . In view of the large ($\sim 15\%$) disordered vacancy concentration on the Cu sites found by Pilchowski and Mewis at room temperature (RT) and confirmed by us, this phase transition may reflect the occurrence of Cu vacancy ordering at T_i . This possibility could be checked via low-*T* x-ray- and/or neutron-diffraction measurements.

II. EXPERIMENTAL DETAILS

Single crystals of $\text{CaCu}_{1.7}\text{As}_2$ were grown using prereacted CuAs self-flux starting with the high-purity elements Ca (99.98%), Cu (99.999%) and As (99.99999%) from Alfa Aesar. Ca and CuAs were taken in a 1 : 4 molar ratio and placed in an alumina crucible which was then sealed inside an evacuated quartz tube. The sample was heated to 1100 °C at a rate of 60 °C/h, held there for 12 h and then cooled

to 800 °C at a rate of 2.5 °C/h at which time the flux was decanted using a centrifuge. Shiny plate-like crystals of typical size $2.5 \times 2 \times 0.3 \text{ mm}^3$ were obtained using this procedure. All crystals for which measurements are reported here were obtained from the same growth batch.

The structure of the crystals was determined by powder x-ray diffraction (XRD) using Cu K_α radiation on a Rigaku Geigerflex x-ray diffractometer. The chemical composition was determined by wavelength dispersive x-ray spectroscopy (WDS) analysis using a JEOL JXA-8200 electron probe microanalyzer.

The magnetization M measurements were carried out using a Quantum Design, Inc. superconducting-quantum-interference-device-based magnetic properties measurement system in applied magnetic fields H up to 5.5 T. A crystal was mounted on a 0.5-mm-diameter horizontal rotatable high-purity quartz rod that was inserted in holes in a clear plastic straw attached to the sample hang-down rod. The sample was attached to the quartz rod using a small amount of GE 7031 varnish. The contribution to the measured magnetization due to the quartz rod and varnish was measured separately and corrected for.

The $C_p(T)$ and $\rho(T, H)$ measurements were carried out using a Quantum Design, Inc. physical properties measurement system using the heat capacity and ac transport options, respectively, at fields up to 8 T. The subscript “p” in C_p refers to measurements at constant pressure. The C_p was measured using a relaxation method. The ρ was measured in the ab plane using a standard four-probe ac technique with 25- μm -diameter Pt leads attached to the sample with EPO-TEK P1011 silver epoxy that was cured in air at 110 °C for 1 h. We did not cut the as-grown rectangular-shaped crystals #1 and #2 that we used for the resistivity measurements because such cutting can potentially introduce microcracks in the crystals and/or exfoliation of the $\text{Cu}_{1.7}\text{As}$ layers. The accuracy of the measurements due to uncertainties in the geometric factor is estimated to be $\sim 10\%$.

III. RESULTS AND DISCUSSION

A. Crystallography

Powder XRD data were collected on crushed $\text{CaCu}_{1.7}\text{As}_2$ single crystals at room temperature (RT) and analyzed by Rietveld refinement using the FULLPROF software.²¹ Figure 1 shows the XRD data and the Rietveld fit profile. The refinement confirmed the single-phase nature of the crystals and the ThCr_2Si_2 -type body-centered tetragonal structure with space group $I4/mmm$ found previously by Pilchowski and Mewis from a single-crystal structure refinement.²⁰ Our crystallographic and refinement parameters are listed in Tables I and II and compared with those of Pilchowski and Mewis.²⁰ The lattice parameters a and c and the z coordinate of the As atoms z_{As} that we obtained are in good agreement with the literature values.²⁰

While refining the XRD powder pattern for $\text{CaCu}_{1.7}\text{As}_2$ we noticed that the lattice parameters and the As c axis position parameter z_{As} were insensitive to changes in the thermal parameters B within the error bars, so we kept B fixed to $B \equiv 0$. On the other hand, the refinement quality and the calculated line intensities were very sensitive to the fractional

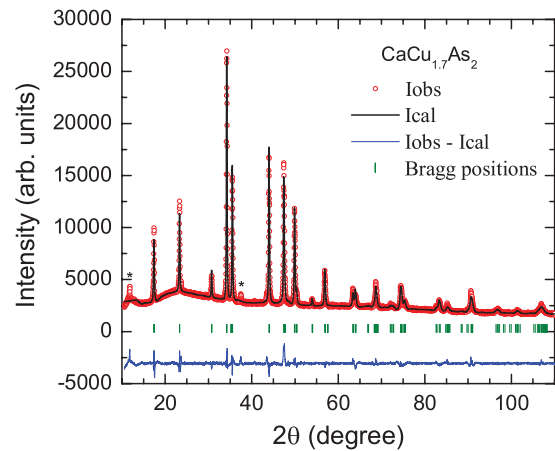


FIG. 1. (Color online) Powder x-ray diffraction pattern of $\text{CaCu}_{1.7}\text{As}_2$ recorded at room temperature. The solid line through the experimental points is the Rietveld refinement profile calculated for the body-centered tetragonal ThCr_2Si_2 -type structure (space group $I4/mmm$). The short vertical bars mark the fitted Bragg-peak positions. The lowermost curve represents the difference between the experimental and calculated intensities. The unindexed peaks marked with stars correspond to peaks from small amounts of flux that could not be removed from the crystals before crushing them for the XRD measurements.

occupancy of the 4d sites by Cu. Therefore we allowed the occupancy of this site by Cu to vary during the refinement, while the occupancies of the Ca and As positions were kept fixed at the stoichiometric values of unity. As shown in Table II, the refined values of the Cu-site occupancy obtained both by us and by Pilchowski and Mewis show a large vacancy concentration on the Cu sites of $\approx 15\%$, corresponding to an approximate composition of $\text{CaCu}_{1.7}\text{As}_2$. However, there are no indications of superstructure reflections in the XRD patterns that would indicate ordering of the Cu vacancies at room temperature, and Pilchowski and Mewis also did not report such reflections. Thus we assume that the Cu vacancies are randomly distributed on the Cu sites at room temperature.

Our WDS analysis at about ten points on the ab plane of a $\text{CaCu}_{1.7}\text{As}_2$ crystal gave the average atomic ratios as

TABLE I. Crystallographic and Rietveld refinement parameters obtained from powder XRD data of crushed $\text{CaCu}_{1.7}\text{As}_2$ crystals. Also included are data from Ref. 20.

Structure	ThCr_2Si_2 -type	
Space group	$I4/mmm$	
Formula units/unit cell	$Z = 2$	
Lattice parameters (RT)	This work	Ref. 20
a (Å)	4.1148(2)	4.129(1)
c (Å)	10.1914(4)	10.251(1)
c/a	2.4768(2)	2.482(2)
V_{cell} (Å ³)	172.55(1)	174.8(1)
Refinement quality		
χ^2	5.63	
R_p (%)	2.98	
R_{wp} (%)	4.44	

TABLE II. Atomic coordinates obtained from the Rietveld refinements of powder XRD data of crushed $\text{CaCu}_{1.7}\text{As}_2$ crystals. Also included are data from Ref. 20.

Atom	Wyckoff position	x	y	z	Fractional occupancy (%)	Ref.
Ca	2a	0	0	0	100	This work
Ca	2a	0	0	0	100	20
Cu	4d	0	1/2	1/4	82.5(5)	This work
Cu	4d	0	1/2	1/4	87.5(8)	20
As	4e	0	0	0.3779(2)	100	This work
As	4e	0	0	0.3799(2)	100	20

Ca : Cu : As = 21.4(2) : 36.5(3) : 42.1(2), corresponding to the stoichiometry $\text{Ca}_{1.02(2)}\text{Cu}_{1.73(2)}\text{As}_2$ assuming the occupancy of the As site to be 100%. This result confirms complete occupancy of the Ca site and again indicates a large vacancy concentration on the Cu sites.

For the molar heat capacity, magnetization, and magnetic susceptibility data presented below, a “mole” is defined as a mole of $\text{CaCu}_{1.7}\text{As}_2$ formula units (f.u.).

From the crystal data in Tables I and II we obtain the interlayer As–As distance as $d_{\text{As–As}} = (1 - 2z_{\text{As}})c = 2.49$ Å. This value of $d_{\text{As–As}}$ for $\text{CaCu}_{1.7}\text{As}_2$ is close to the covalent (single) bond distance 2.38 Å for As.²² Furthermore, the values of c/a and $d_{\text{As–As}}$ fall in the respective ranges for the collapsed tetragonal structure compounds shown in Fig. 22 of Ref. 16. This shows that, like SrCu_2As_2 , $\text{CaCu}_{1.7}\text{As}_2$ also has a cT structure. A consequence of the formation of the cT structure is an unusual oxidation state of $\text{As}^{-2} \equiv [\text{As–As}]^{-4}$, which together with Ca in the Ca^{+2} oxidation state indicates that the Cu in $\text{CaCu}_{1.7}\text{As}_2$ has an oxidation state of $\approx +1.2$. However, the nonmagnetic nature of this compound deduced from our magnetic measurements presented below in Sec. III C suggests instead a filled Cu 3d shell, a Cu electronic configuration $3d^{10}$, and formal oxidation states Cu^{+1} and $\text{As}^{-1.85}$. The latter value suggests the presence of hole conduction on the As sublattice.

B. Heat capacity

Figure 2 shows C_p versus T of a $\text{CaCu}_{1.7}\text{As}_2$ crystal from 1.8 to 300 K. No obvious anomaly that might be associated with the occurrence of a phase transition is observed over this T range. The value of $C_p(300 \text{ K})$ is $\approx 116 \text{ J mol}^{-1} \text{ K}^{-1}$, which is close to the classical Dulong-Petit prediction of the lattice heat capacity $C_V = 3nR = 14.1R = 117.2 \text{ J mol}^{-1} \text{ K}^{-1}$ at constant volume, where n is the number of atoms per formula unit ($n = 4.7$ here) and R is the molar gas constant.^{23,24}

A conventional linear fit of $C_p(T)/T = \gamma + \beta T^2$ in the temperature range $1.8 \text{ K} \leq T \leq 3.5 \text{ K}$ is shown in the inset of Fig. 2 and gives the electronic Sommerfeld specific heat coefficient $\gamma = 2.0(2) \text{ mJ mol}^{-1} \text{ K}^{-2}$ and the lattice heat capacity coefficient $\beta = 0.33(2) \text{ mJ mol}^{-1} \text{ K}^{-4}$. The density of states at the Fermi energy $\mathcal{D}(E_F)$ for both spin directions is estimated from γ using the single-band relation²³ $\gamma = (\pi^2 k_B^2 / 3) \mathcal{D}(E_F)$, from which we obtain $\mathcal{D}(E_F) = 0.85(9) \text{ states}/(\text{eV f.u.})$ for both spin directions. The Debye temperature Θ_D is estimated

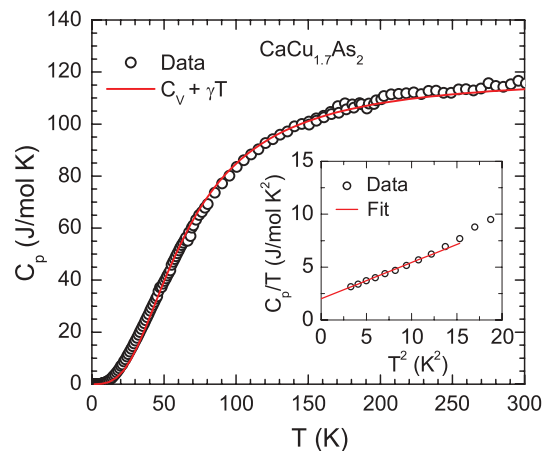


FIG. 2. (Color online) Heat capacity C_p of a $\text{CaCu}_{1.7}\text{As}_2$ single crystal as a function of temperature T measured in zero magnetic field. The red solid curve is the fitted sum of the contributions from the Debye lattice heat capacity $C_{V\text{Debye}}(T)$ and predetermined electronic heat capacity γT according to Eq. (1). Inset: C_p/T versus T^2 for $T \leq 4.5 \text{ K}$. The straight red line is a fit of the data by $C_p/T = \gamma + \beta T^2$ for $1.8 \text{ K} \leq T \leq 3.5 \text{ K}$.

from β using the relation²³ $\Theta_D = [12\pi^4 nR / (5\beta)]^{1/3}$, giving $\Theta_D = 303(6) \text{ K}$.

The entire $C_p(T)$ data set from 1.8 to 300 K was fitted by

$$C_p(T) = \gamma T + nC_{V\text{Debye}}(T), \quad (1)$$

where γ was fixed to the value $\gamma = 2.0 \text{ mJ mol}^{-1} \text{ K}^{-2}$ obtained above and the Debye heat capacity $C_{V\text{Debye}}(T)$ describes the heat capacity due to acoustic phonons at constant volume V and is given by²⁴

$$C_{V\text{Debye}}(T) = 9R \left(\frac{T}{\Theta_D} \right)^3 \int_0^{\Theta_D/T} \frac{x^4 e^x}{(e^x - 1)^2} dx. \quad (2)$$

The fit is shown as the solid red curve in Fig. 2. In the fit, we used our analytic Padé approximant function given in Ref. 25 that accurately represents $C_{V\text{Debye}}(T)$ and obtained $\Theta_D = 265(1) \text{ K}$, which is somewhat smaller than the value $\Theta_D = 303(6) \text{ K}$ obtained from fitting the low- T $C_p(T)$ data above. The difference between these two values reflects the T dependence of Θ_D .^{24,25}

C. Magnetization and magnetic susceptibility

Figure 3 shows the zero-field-cooled magnetic susceptibility $\chi \equiv M/H$ of a $\text{CaCu}_{1.7}\text{As}_2$ crystal as a function of T from 1.8 to 350 K for $H = 3.0 \text{ T}$ applied along the c axis (χ_c , $H \parallel c$) and in the ab plane (χ_{ab} , $H \perp c$). The $\chi(T)$ data for both directions of H are diamagnetic and nearly independent of T . The χ_c is significantly more negative than χ_{ab} . The same type of χ anisotropy was previously observed for SrCu_2As_2 ,¹⁶ BaFe_2As_2 ,²⁶ SrFe_2As_2 ,²⁷ and other doped and undoped FeAs-based compounds.¹⁰

Curie-like upturns occur in $\chi(T)$ at low T in Fig. 3 that are likely due at least in part to the presence of small amounts of saturable paramagnetic (PM) impurities in the $\text{CaCu}_{1.7}\text{As}_2$ crystal. Our analysis of $M(H)$ isotherms in the Appendix allows us to approximately correct for such contributions. The

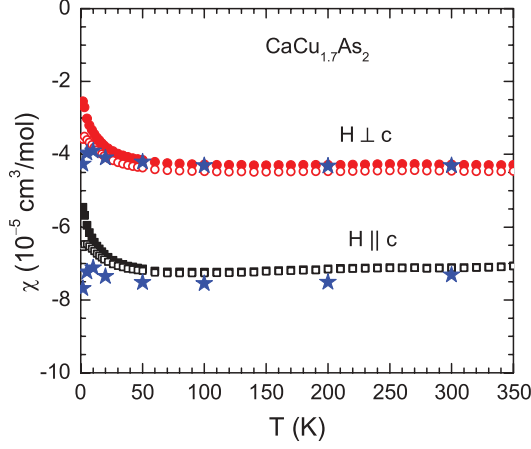


FIG. 3. (Color online) Zero-field-cooled magnetic susceptibility χ of a $\text{CaCu}_{1.7}\text{As}_2$ single crystal versus temperature T in the temperature range 1.8–350 K measured in a magnetic field $H = 3.0$ T applied along the c axis (χ_c , $H \parallel c$) and in the ab plane (χ_{ab} , $H \perp c$) (solid symbols). The open symbols represent the intrinsic susceptibility of $\text{CaCu}_{1.7}\text{As}_2$ after correcting for the ferromagnetic and paramagnetic impurity contributions as described in the Appendix. The filled blue stars represent the intrinsic χ obtained from fitting $M(H)$ isotherm data by Eq. (A2) and are more reliable.

intrinsic susceptibilities after corrections for paramagnetic and ferromagnetic (FM) impurity contributions are shown as open symbols in Fig. 3. The corrected susceptibilities still show small residual upturns at low T which, in view of the intrinsic T -independent susceptibilities obtained at eight temperatures from analysis of the $M(H)$ isotherms (shown in Fig. 3 as filled blue stars), are due to inaccuracies in correcting for the impurity contributions to the M/H versus T data.

The intrinsic χ consists of different contributions given by

$$\chi = \chi_{\text{core}} + \chi_{\text{VV}} + \chi_{\text{L}} + \chi_{\text{P}}. \quad (3)$$

The first three terms are orbital susceptibilities. χ_{core} is the diamagnetic core susceptibility, χ_{VV} is the paramagnetic Van Vleck susceptibility, and χ_{L} is the diamagnetic Landau susceptibility of the conduction electrons. The last term χ_{P} is the paramagnetic Pauli spin susceptibility. The $\chi_{\text{core}} = -1.53 \times 10^{-4} \text{ cm}^3/\text{mol}$ is estimated using the atomic diamagnetic susceptibilities.²⁸ χ_{P} is estimated from $\chi_{\text{P}} = \mu_{\text{B}}^2 \mathcal{D}(E_{\text{F}})$ (Ref. 29), giving $\chi_{\text{P}} = 2.7(3) \times 10^{-5} \text{ cm}^3/\text{mol}$ using $\mathcal{D}(E_{\text{F}}) = 0.85(9)$ states/(eV f.u.) for both spin directions obtained above in Sec. III B. The χ_{L} is obtained from $\chi_{\text{L}} = -\frac{1}{3}(m_e/m^*)^2 \chi_{\text{P}}$,^{29,30} which gives $\chi_{\text{L}} = -0.9 \times 10^{-5} \text{ cm}^3/\text{mol}$ assuming that the effective mass m^* equals the free electron mass m_e . The angle and temperature average of the anisotropic χ in Fig. 3 over the T range 30 to 350 K is $\langle \chi \rangle = [2\langle \chi_{ab} \rangle + \langle \chi_c \rangle]/3 = -5.3 \times 10^{-5} \text{ cm}^3/\text{mol}$. We can now estimate $\langle \chi_{\text{VV}} \rangle$ using the above estimated values of χ_{core} , χ_{P} , and χ_{L} yielding the powder-averaged $\langle \chi_{\text{VV}} \rangle = 8.2 \times 10^{-5} \text{ cm}^3/\text{mol}$ from Eq. (3), which is a physically realistic value. The T -independent anisotropic Van Vleck contributions are $\chi_{\text{VV}}^c = 6.3 \times 10^{-5} \text{ cm}^3/\text{mol}$ and $\chi_{\text{VV}}^{ab} = 9.1 \times 10^{-5} \text{ cm}^3/\text{mol}$ for $H \parallel c$ and $H \perp c$, respectively.

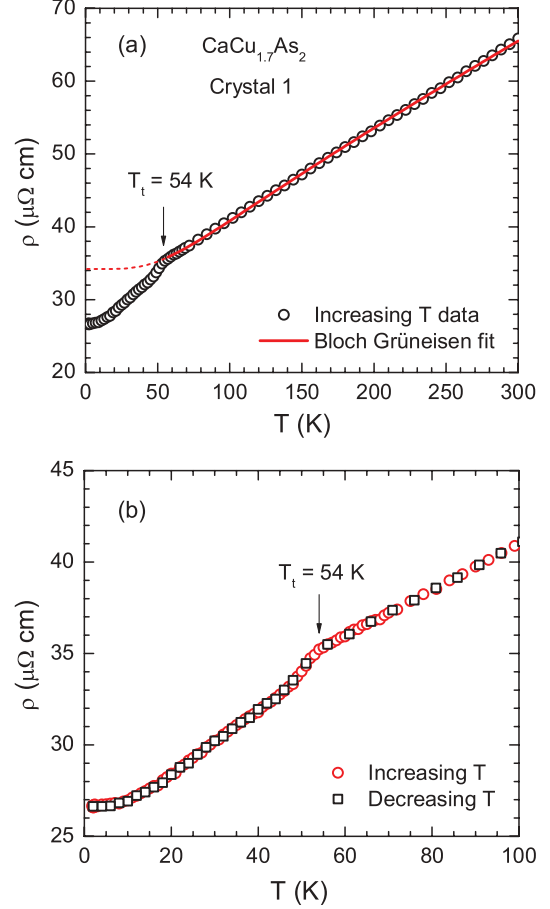


FIG. 4. (Color online) (a) In-plane electrical resistivity ρ of $\text{CaCu}_{1.7}\text{As}_2$ single crystal #1 as a function of temperature T measured in zero magnetic field (open circles) showing a transition at $T_t = 54$ K. A fit by the Bloch-Grüneisen model from 57 to 300 K is shown by the solid red curve, and an extrapolation of the fit to $T = 0$ is shown as the dashed red curve. (b) Expanded plot of ρ versus T for heating and cooling cycles through T_t at $T < 100$ K.

D. Electrical resistivity

Figure 4 shows $\rho(T)$ of $\text{CaCu}_{1.7}\text{As}_2$ crystal #1 from 1.8 to 300 K in $H = 0$. The residual resistivity at $T = 1.8$ K is $\rho_0 = 26.7(1) \mu\Omega \text{ cm}$ and the residual resistivity ratio is $\text{RRR} \equiv \rho(300 \text{ K})/\rho_0 \approx 2.5$. The $\rho(T)$ data exhibit metallic behavior with an almost linear T dependence of ρ above 55 K. A sharp decrease is observed in $\rho(T)$ upon cooling below a transition temperature $T_t = 54$ K. This behavior is reproduced without hysteresis upon heating and cooling through T_t , as shown in Fig. 4(b), suggesting a second-order transition. The transition anomaly at 54 K was reproduced in a $\rho(T)$ measurement on another crystal #2 for which we found $T_t = 56$ K, as shown in Fig. 5. As noted in previous sections, no evidence for the transition was observed in our $\chi(T)$ measurements, suggesting that the transition is not magnetic in nature. The transition may be associated with spatial ordering of the Cu vacancies discussed above.

We also measured $\rho(T < 100)$ K in high magnetic fields $H \leq 8.0$ T as shown in Fig. 6(a). The transition temperature T_t is found to be independent of H over this field range. However, a magnetoresistance (MR) $\Delta\rho/\rho(0) = [\rho(H) - \rho(0)]/\rho(0)$ is

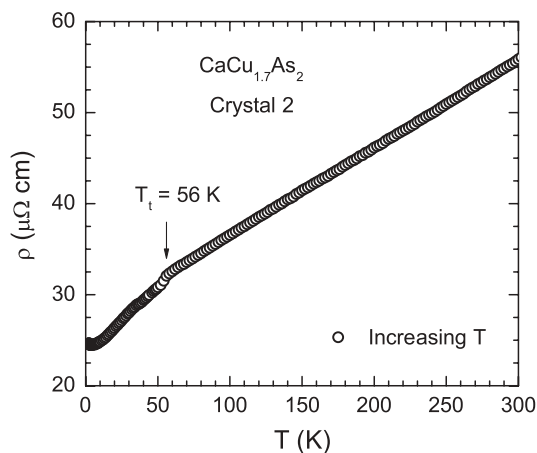


FIG. 5. In-plane electrical resistivity ρ of $\text{CaCu}_{1.7}\text{As}_2$ single crystal #2 versus temperature T in zero magnetic field showing the same shape of transition at $T_t = 56$ K as seen for crystal #1 at 54 K in Fig. 4.

observed at temperatures $T < T_t$ and is plotted versus H for $T = 1.8$ and 10 K in Fig. 6(b). The MR is positive up to 8.0 T with $\text{MR} \approx 8.7\%$ at 1.8 K and $H = 8.0$ T. The reason that measurable MR values are only observed at $T < T_t$ is not clear.

Our detailed $\rho(T, H)$ measurements on crystal #1 indicate that the transition at $T_t = 54$ K is not due to microcracks in the sample because the data on heating and cooling the same sample in Figs. 4 and 6(a) reproducibly show the same sharp transition temperature with the same shape of the anomaly, and the transition occurs reproducibly between samples as seen in Fig. 5. Furthermore, we can see no mechanism by which the magnetoresistance that is observed only below T_t in Fig. 6 could arise from microcracks.

The transition in $\text{CaCu}_{1.7}\text{As}_2$ seen in $\rho(T)$ at 54–56 K in Figs. 4 and 5 might potentially be due to an extrinsic phase transition in residual Cu-As flux attached to the crystals. Three binary phases exist in the Cu-As binary phase diagram with compositions of approximately Cu_3As , Cu_5As_2 , and Cu_2As . Previous ρ , χ , and nuclear quadrupole resonance measurements versus T on these phases^{31,32} showed no phase transitions at any temperature near our phase-transition temperature $T_t = 54$ –56 K. Moreover, the SrCu_2As_2 crystals for which we reported the properties in Ref. 16 were grown using the same CuAs self-flux, and these crystals showed no evidence for a transition at $T \approx 55$ K in our $\rho(T)$ measurements on them. Thus we rule out this extrinsic cause of the transition we see in $\text{CaCu}_{1.7}\text{As}_2$ at T_t .

The lack of an obvious heat capacity anomaly at ~ 55 K in Fig. 2 might be construed as evidence that the second-order phase transition at $T_t = 54$ –56 K indicated by the resistivity data in Figs. 4–6 is not a bulk effect. However, an observable change in the heat capacity is only expected if the temperature derivative dS/dT of the entropy S changes sufficiently strongly at T_t . In particular, the heat capacity of a material is $C_p = dQ/dT = T dS/dT$, where dQ is the increment of heat absorbed and $dS = dQ/T$ is the incremental change in the total entropy of the system. Bulk phase transitions can occur that involve very little change in the slope of the entropy versus

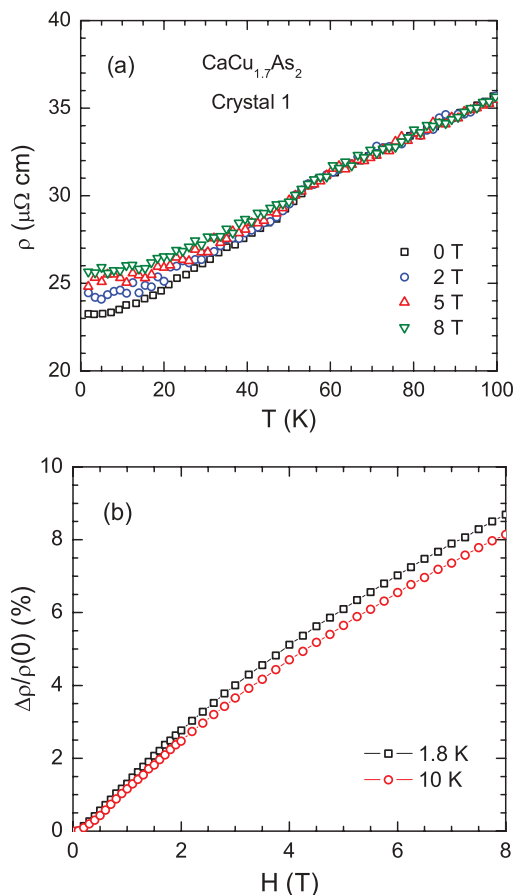


FIG. 6. (Color online) (a) In-plane electrical resistivity ρ of a $\text{CaCu}_{1.7}\text{As}_2$ single crystal versus temperature T measured in the indicated magnetic fields H . The $\rho(T)$ data in this figure were measured after remounting the leads, resulting in a slight change in the absolute values of ρ between this figure and Fig. 4. (b) Magnetoresistance $[\rho(H) - \rho(0)]/\rho(0)$ versus H at $T = 1.8$ and 10 K.

temperature of the system. For example, the bulk second-order antiferromagnetic phase transition at the Néel temperature $T_N \sim 300$ K of the layered cuprate La_2CuO_4 has not been observed by heat-capacity measurements because the change in dS/dT at T_N is too small.³³ Thus the lack of an observable heat-capacity anomaly at T_t in Fig. 2 does not rule out a bulk phase transition at that temperature.

In the Bloch-Grüneisen (BG) model, the resistivity arises from scattering of conduction electrons by longitudinal acoustic lattice vibrations, given by³⁴

$$\rho_{\text{BG}}(T) = 4\mathcal{R}(\Theta_{\text{R}}) \left(\frac{T}{\Theta_{\text{R}}} \right)^5 \int_0^{\Theta_{\text{R}}/T} \frac{x^5}{(e^x - 1)(1 - e^{-x})} dx, \quad (4)$$

where Θ_{R} is the Debye temperature obtained from fitting resistivity data. For polyatomic systems the prefactor $\mathcal{R}(\Theta_{\text{R}})$ is given by^{16,25}

$$\mathcal{R}(\Theta_{\text{R}}) = \frac{\hbar}{e^2} \left[\frac{\pi^3 (3\pi^2)^{1/3} \hbar^2}{4n_{\text{cell}}^{2/3} a^* k_{\text{B}} \Theta_{\text{R}}} \left(\frac{1}{M} \right)_{\text{ave}} \right], \quad (5)$$

where \hbar is Planck's constant divided by 2π , k_{B} is Boltzmann's constant, e is the fundamental electric charge, n_{cell} is the

number of conduction (valence) electrons per atom, $(1/M)_{\text{ave}}$ is the average inverse mass of the atoms in the compound, and $a^* = [V_{\text{cell}}/nZ]^{1/3}$ is the equivalent lattice parameter of a primitive cubic unit cell containing one atom, Z being the number of formula units per unit cell and n the number of atoms per f.u.

As discussed in Refs. 25 and 34, it is usually not possible to obtain an accurate fit to $\rho(T)$ data by the BG model with the single adjustable parameter Θ_{R} . Therefore we allowed the prefactor in Eq. (4) to vary independently and fit our in-plane $\rho(T > T_{\text{t}})$ data for $\text{CaCu}_{1.7}\text{As}_2$ in Fig. 4(a) by

$$\rho(T) = \rho'_0 + \rho(\Theta_{\text{R}})f(T/\Theta_{\text{R}}), \quad (6)$$

where ρ'_0 is the residual resistivity extrapolated from $T > T_{\text{t}}$ and from Eq. (4) one obtains^{16,25}

$$f(y) = \frac{\rho_{\text{BG}}(T)}{\rho_{\text{BG}}(T = \Theta_{\text{R}})} = 4.226259y^5 \int_0^{1/y} \frac{x^5}{(e^x - 1)(1 - e^{-x})} dx, \quad (7)$$

where $y = T/\Theta_{\text{R}}$ and

$$\rho_{\text{BG}}(T = \Theta_{\text{R}}) = 0.9464635\mathcal{R}(\Theta_{\text{R}}). \quad (8)$$

A fit of $\rho(T)$ data by Eqs. (6) and (7) thus has three independent adjustable parameters ρ'_0 , $\rho(\Theta_{\text{R}})$, and Θ_{R} .

A good fit of the $\rho(T)$ data in Fig. 4(a) for $57 \text{ K} \leq T \leq 300 \text{ K}$ was obtained, as shown by the solid red curve in Fig. 4(a) where we used an accurate analytic Padé approximant function of y in place of Eq. (7), as given in Ref. 25. The parameters obtained from the fit are $\rho'_0 = 34.2(1) \mu\Omega \text{ cm}$, $\rho(\Theta_{\text{R}}) = 33.7(6) \mu\Omega \text{ cm}$, and $\Theta_{\text{R}} = 320(6) \text{ K}$. The $\mathcal{R}(\Theta_{\text{R}})$ calculated from the value of $\rho(\Theta_{\text{R}})$ using Eq. (8) is $\mathcal{R}(\Theta_{\text{R}}) = 35.6 \mu\Omega \text{ cm}$. In order to compare the resistivity at, for example, room temperature, with the value predicted by the BG theory, one needs an estimate of the conduction carrier concentration n_{cell} in Eq. (5). Such an estimate is not currently available.

IV. CONCLUSIONS

We have successfully grown single crystals of $\text{CaCu}_{1.7}\text{As}_2$ and investigated their crystallographic, magnetic, thermal, and electronic transport properties. Rietveld refinements of powder XRD data for crushed crystals and WDS chemical analyses of single-crystal surfaces indicate the presence of $\approx 15\%$ vacancies on the Cu sites, consistent with literature data. No superconductivity was observed above 1.8 K. Our crystallographic and refinement parameters are listed in Tables I and II and a summary of the parameters obtained from our various physical property measurements on $\text{CaCu}_{1.7}\text{As}_2$ is given in Table III. The $\chi(T)$ data reveal a nearly T -independent anisotropic diamagnetic behavior, indicating that the Cu atoms in $\text{CaCu}_{1.7}\text{As}_2$ are in the Cu^{+1} oxidation state with a nonmagnetic $3d^{10}$ electronic configuration, as expected from the collapsed-tetragonal crystal structure of this compound. The formal oxidation state of the As, which participates in As-As interlayer bonding, is then $\text{As}^{-1.85}$, which suggests hole conduction on the As sublattice. The $C_{\text{p}}(T)$ and $\rho(T)$ data reveal metallic behavior. A small density of states at the Fermi level is found, consistent with $\text{CaCu}_{1.7}\text{As}_2$ being an

TABLE III. Values of parameters obtained from analyses of heat capacity, magnetic susceptibility, and electrical resistivity measurements of $\text{CaCu}_{1.7}\text{As}_2$. The notation $\langle \dots \rangle$ denotes a temperature and/or angular average of the enclosed quantity.

Property	Value
Heat capacity	
γ	2.0(2) mJ mol ⁻¹ K ⁻²
β	0.33(2) mJ mol ⁻¹ K ⁻⁴
$\mathcal{D}(E_{\text{F}})$	0.85(9) states/(eV f.u.) (both spin directions)
Θ_{D}	303(6) K (from low T)
Θ_{D}	265(1) K (from all T)
Susceptibility	
$\langle \chi \rangle$	$-5.3 \times 10^{-5} \text{ cm}^3/\text{mol}$
χ_{core}	$-1.53 \times 10^{-4} \text{ cm}^3/\text{mol}$
χ_{P}	$2.7(3) \times 10^{-5} \text{ cm}^3/\text{mol}$
χ_{L}	$-0.9 \times 10^{-5} \text{ cm}^3/\text{mol}$
$\langle \chi_{\text{VV}} \rangle$	$8.2 \times 10^{-5} \text{ cm}^3/\text{mol}$
χ_{VV}^{ab}	$9.1 \times 10^{-5} \text{ cm}^3/\text{mol}$
χ_{VV}^c	$6.3 \times 10^{-5} \text{ cm}^3/\text{mol}$
Resistivity	
ρ_0	26.7(1) $\mu\Omega \text{ cm}$
RRR	≈ 2.5
ρ'_0	34.2(1) $\mu\Omega \text{ cm}$
$\rho(\Theta_{\text{R}})$	33.7(6) $\mu\Omega \text{ cm}$
$\mathcal{R}(\Theta_{\text{R}})$	35.6 $\mu\Omega \text{ cm}$
Θ_{R}	320(6) K

sp -band metal. The overall $C_{\text{p}}(T)$ and $\rho(T > T_{\text{t}})$ behaviors are well-described by the Debye model and the Bloch-Grüneisen model, respectively. However, the $\rho(T)$ of $\text{CaCu}_{1.7}\text{As}_2$ exhibits a transition of unknown origin at $T_{\text{t}} = 54\text{--}56 \text{ K}$ without any thermal hysteresis, suggesting that the transition is second order. A significant positive magnetoresistance develops below this transition temperature. This transition may arise from spatial ordering of the Cu vacancies on cooling below T_{t} . High-resolution x -ray- and/or neutron-diffraction measurements at low T could test this hypothesis.

Note added. Cheng *et al.*³⁵ reported in this journal the observation of a transition in “ CaCu_2As_2 ” single crystals at 50 K from $\rho(T)$ measurements. As in the present paper, their $\chi(T)$ data for this compound showed no evidence of the transition. These authors did not mention the large concentration of vacancies on the Cu sites previously reported in Ref. 20 and confirmed by us. Cheng *et al.* noted that the shape of the transition in $\rho(T)$ is similar to those observed for $\text{CaFe}_2(\text{As}_{1-x}\text{P}_x)_2$ (Ref. 36) and $\text{Ca}_{1-x}\text{R}_x\text{Fe}_2\text{As}_2$ ($R = \text{lanthanide}$, Ref. 37) arising from transitions from tetragonal to collapsed-tetragonal (cT) structures upon cooling below room temperature. However, as we have discussed herein and previously,¹⁶ $\text{CaCu}_{1.7}\text{As}_2$ as well as SrCu_2As_2 and BaCu_2As_2 are already in the cT phase at room temperature.

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APPENDIX: MAGNETIZATION ISOTHERMS AND ANALYSES

Figure 7 shows $M(H)$ isotherms for a $\text{CaCu}_{1.7}\text{As}_2$ crystal measured at eight temperatures between 1.8 and 300 K with H applied along the c axis (M_c , $H \parallel c$) and in the ab plane (M_{ab} , $H \perp c$). The $M(H)$ data exhibit anisotropic diamagnetic behavior with $M_{ab}(H) > M_c(H)$, consistent with the above $\chi(T)$ data. The $M(H)$ isotherms are almost linear in H for $T \geq 50$ K. For $T \leq 20$ K the $M(H)$ isotherms exhibit slight nonlinearity which we attribute to the presence of saturable paramagnetic impurities in the sample.

In order to estimate the ferromagnetic impurity contribution we fit the $M(H)$ data at $T \geq 50$ K for $H \geq 2$ T by

$$M(H) = M_s + \chi H, \quad (\text{A1})$$

where M_s is the FM impurity saturation magnetization and χ is the susceptibility. Within the error bars T -independent anisotropic values of M_s for $H \parallel c$ and $H \perp c$ were obtained as listed in Table IV. The M_s values indicate that only a

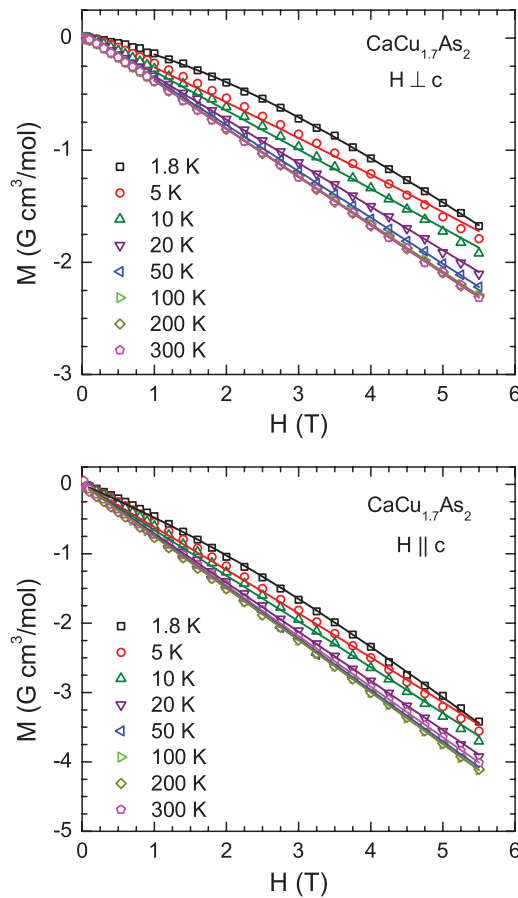


FIG. 7. (Color online) Isothermal magnetization M of a $\text{CaCu}_{1.7}\text{As}_2$ single crystal as a function of applied magnetic field H measured at the indicated temperatures for H applied (a) in the ab plane (M_{ab} , $H \perp c$) and, (b) along the c axis (M_c , $H \parallel c$). The solid curves are fits by Eq. (A2) with $0 \leq H \leq 5.5$ T for $H \parallel c$ and with $1.0 \leq H \leq 5.5$ T for $H \perp c$.

TABLE IV. Parameters obtained for $\text{CaCu}_{1.7}\text{As}_2$ from fitting $M(H)$ isotherms at 1.8 K by Eqs. (A1) and (A2), where $\theta_{\text{imp}} \equiv 0$ and $S_{\text{imp}} \equiv 1/2$. Here M_s is the saturation magnetization of ferromagnetic impurities, χ is the intrinsic susceptibility, and f_{imp} is the molar fraction of paramagnetic impurities.

Field direction	M_s ($\text{G cm}^3/\text{mol}$)	χ ($10^{-5} \text{ cm}^3/\text{mol}$)	f_{imp} (10^{-5})
$H \perp c$	0.05(1)	-4.27(3)	5.8(1)
$H \parallel c$	0.00(2)	-7.68(7)	7.2(3)

trace amount of ferromagnetic impurities is present. The M_s value of $0.05 \text{ G cm}^3/\text{mol}$ corresponds to the saturation magnetization of 4 molar ppm of Fe metal impurities. However, since $\text{CaCu}_{1.7}\text{As}_2$ is a diamagnetic compound with a small magnitude of χ , even such trace amounts of ferromagnetic impurities are observable in χ and M measurements.

In order to estimate the paramagnetic impurity contribution to M we fitted the $M(H)$ data for each field direction by

$$M(T, H) = M_s + \chi H + f_{\text{imp}} M_{S_{\text{imp}}} B_{S_{\text{imp}}}(x), \quad (\text{A2})$$

where χ is the intrinsic susceptibility of the compound, f_{imp} is the molar fraction of PM impurities, $M_{S_{\text{imp}}} = N_A g_{\text{imp}} \mu_B S_{\text{imp}}$ is the PM impurity saturation magnetization, N_A is Avogadro's number, μ_B is the Bohr magneton, and g_{imp} and S_{imp} are the spectroscopic splitting factor (g factor) and the spin of the impurities, respectively. Our unconventional definition of the Brillouin function $B_{S_{\text{imp}}}$ is given by³⁸

$$B_{S_{\text{imp}}}(x) = \frac{1}{2S_{\text{imp}}} \left\{ (2S_{\text{imp}} + 1) \coth \left[(2S_{\text{imp}} + 1) \frac{x}{2} \right] - \coth \left(\frac{x}{2} \right) \right\}, \quad (\text{A3a})$$

where

$$x \equiv \frac{g_{\text{imp}} \mu_B H}{k_B (T - \theta_{\text{imp}})}, \quad (\text{A3b})$$

and we have included a Weiss temperature θ_{imp} in the argument of $B_{S_{\text{imp}}}(x)$ in order to take account of interactions between the paramagnetic impurities in an average mean-field way. In particular, a Taylor series expansion of Eq. (A3a) for $x \ll 1$ yields a Curie-Weiss law for the impurities [i.e., $\chi_{\text{imp}} = C_{\text{imp}}/(T - \theta_{\text{imp}})$].

In order to reduce the number of fitting parameters we set the impurity g factor to the fixed value $g_{\text{imp}} = 2$. The M_s^{ab} and M_s^c saturation magnetization values for $H \perp c$ and $H \parallel c$, respectively, were set to the values obtained above from linear fits of the high-field data at $H \geq 2$ T at temperatures $T \geq 50$ K. Since M_s^c is zero the $M(H)$ data for $H \parallel c$ were fitted in the whole range $0 \leq H \leq 5.5$ T. However, since M_s^{ab} is nonzero the $M(H)$ data for $H \perp c$ were fitted only in the range $1.0 \leq H \leq 5.5$ T. The S_{imp} was found to be close to $1/2$, so in the final fits we set $S_{\text{imp}} = 1/2$. Furthermore, we found that allowing θ_{imp} to vary during a fit does not make a significant change in the quality of fit and also gives θ_{imp} close to zero. Therefore in the final fits we set $\theta_{\text{imp}} \equiv 0$ for both field directions. The final fits of the $M(H)$ data in Fig. 7

by Eq. (A2) are shown by solid curves in Fig. 7. The fitting parameters obtained for the 1.8 K $M(H)$ isotherms are listed in Table IV. The intrinsic χ values obtained are plotted in Fig. 3 as filled blue stars. These data demonstrate that the intrinsic χ is independent of T over the measured temperature range $1.8 \text{ K} \leq T \leq 300 \text{ K}$. We subtracted the ferromagnetic

and paramagnetic impurity contributions from the measured $M(T)$ and obtained the intrinsic χ which is shown by open symbols in Fig. 3. The presence of a weak residual upturn in the corrected $\chi \equiv M/H$ versus T data below 25 K indicates that the paramagnetic impurity contribution has not been completely accounted for in these measurements.

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