Magnetization and spin-flip Raman scattering in $Cd_{1-x}Cr_xSe$ **and** $Cd_{1-x}V_xSe$

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Spin-flip Raman scattering (SFRS) from donor-bound electrons in II-VI diluted magnetic semiconductors (DMSs) containing 3*d* transition-metal ions, e.g., Cd1−*^x*Cr*x*Se and Cd1−*^x*V*x*Se, allows one to explore the exchange interaction between the conduction band or donor electrons with the 3*d* magnetic ions, which is the so-called *s*-*d* exchange interaction. The *s*-*d* exchange energy, being the SFRS Raman shift after subtracting the intrinsic Zeeman splitting characterized by *g*[∗] = 0*.*52 for CdSe, together with the magnetization measurements of those DMSs, allow one to deduce the *s*-*d* exchange constant $N_0\alpha$. For Cd_{1−*x*}Cr_{*x*}Se and Cd_{1−*x*}V_{*x*}Se, the $N_0\alpha$ values are 210 ± 13 and 269 ± 10 meV, respectively.

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I. INTRODUCTION

II-VI semiconductors in which 3*d* transition-metal ions (3*d* TMIs) are incorporated resulting in diluted magnetic semiconductors (II-VI DMSs) have been the focus of intense research thanks to the coexistence of semiconductor and magnetic phenomena, resulting from the so-called *sp*-*d* exchange interaction between the band carriers and localized electrons of the magnetic ions.^{[1](#page-6-0)} Theoretical investigations^{[2,3](#page-6-0)} predict that the *p*-*d* exchange interaction in II-VI DMSs with 3*d* transition-metal ions having less than a half-filled 3*d* shell (e.g., Sc, Ti, V, and Cr) are ferromagnetic, which is in striking contrast to the antiferromagnetic character of those having a half-filled (e.g., Mn) or more than a half-filled (e.g., Co, Fe, Ni, and Cu) 3*d* shell. This ferromagnetic *p*-*d* exchange interaction has been confirmed experimentally in several Cr-based $DMSs^{4-7}$ on the basis of magnetoreflectivity measurements. Mac *et al.*[8](#page-6-0) also report a ferromagnetic *pd* exchange interaction in Cd_{1−*x*}V_{*x*}S as deduced from the Zeeman effect of the excitonic reflectivity and luminescence. The distinct varieties of magnetism displayed by TMIs continue to provide a strong motivation for synthesizing and studying such DMSs. The magnetism displayed by a given TMI is intimately related to its localized magnetic moment influenced by its internal electronic level structure as a constituent of the nonmagnetic host lattice. As a result, Mnand Co-based DMSs show paramagnetism which follow $B_{5/2}$ and $B_{3/2}$ Brillouin functions, respectively;^{1,9} Fe-based DMSs, however, exhibit the van-Vleck-type paramagnetism.⁹⁻¹¹ For V-based DMSs, e.g., Cd1−*^x*V*x*Te, the magnetization follows a *B*3*/*² Brillouin function. However, Cr-based DMSs dis-play a paramagnetism^{[4,12–15](#page-6-0)} that is intermediate between the Brillouin type and van Vleck type. In V- and Cr-based DMSs, the ferromagnetic *sp*-*d* exchange interaction may also mediate the ferromagnetic coupling between the magnetic ions via *d*-*d* exchange interaction. Indeed, room-temperature ferromagnetism of $Zn_{1-x}Cr_xTe$ film with $x = 0.2$ has been reported $16,17$ and has received notice in connection with potential spintronic¹⁸ applications.

Due to the small concentrations in which V and Cr can be incorporated in DMSs, with correspondingly small excitonic Zeeman splittings, not all of the excitonic Zeeman lines are resolved in magnetoreflectivity measurements. As a result, the *sp-d* exchange interaction constants $N_0\alpha$ and N_0 β cannot be deduced separately in those measurements, but rather only the difference $(N_0\alpha - N_0\beta)$ can be evaluated.^{[4–7](#page-6-0)} The value $(N_0\alpha - N_0\beta)$ was found to be negative for all of the Cr-based DMSs studied so far, in contrast to the positive values for Mn-, Co-, and Fe-based DMSs.⁹ With the assumption that $N_0\alpha$ for Cr-based DMSs is about 0.2 eV, which is reasonable and indeed is the case for Mn-, Co-, and Fe-based DMSs reported so far, N_0 β is deduced to be positive for Cr-based DMSs. All of the experimental evidence thus points to a ferromagnetic *p*-*d* exchange interaction in Cr-based II-VI DMSs. A direct measurement of $N_0\alpha$ is thus important. Resonantly enhanced excitonic spin-flip Raman scattering (SFRS) offers an opportunity to determine the magnitude *and* sign of $N_0\alpha$ accurately, even for DMSs with very low TMI concentrations, as demonstrated in zinc blende $Cd_{1-x}V_xTe^{19}$ $Cd_{1-x}V_xTe^{19}$ $Cd_{1-x}V_xTe^{19}$ In the present study, we report and discuss SFRS in $Cd_{1-x}Cr_xSe$ and $Cd_{1-x}V_xSe$ that have a wurtzite structure.

II. EXPERIMENT

 $Cd_{1-x}Cr_{x}Se$ and $Cd_{1-x}V_{x}Se$ crystals were grown using the modified vertical Bridgman technique with nominal Cr and V concentrations of $\leq 2\%$ due to their low solubility. The crystallographic direction of the **cˆ** axis of these crystals was determined from their x-ray Laue pictures. Wavelength-modulated reflectivity (WMR) spectra were obtained at low temperatures in the zero magnetic field. The technique for obtaining WMR is discussed in detail in Ref. [20.](#page-6-0) Photoluminescence (PL) and Raman spectra were recorded with a photon-counting detection system incorporating a cooled photomultiplier, following

FIG. 1. (Color online) Wavelength-modulated reflectivity spectra: (a) $Cd_{1-x}Cr_xSe$ and (b) $Cd_{1-x}V_xSe$. The two signatures correspond to exciton *A* (lower energy) and exciton *B* (higher energy), respectively.

the spectral analysis with a double-grating monochromator. A third grating operated in tandem is often necessary for a more rigorous rejection of the parasitic radiation close to the incident laser energy. An optical cryostat incorporating a superconducting magnet was employed for applying magnetic fields up to 6 T and for achieving temperatures as low as 1.8 K. Raman spectra were excited with a tunable Coherent CR699 ring dye laser using DCM [2-[2-4-(dimethylamino) phenyl]-6 methyl-4H-pyran-4-ylidene]-propanedinitrile as the dye. The magnetization was measured in magnetic fields up to 7 T, employing a Quantum Design MPMS XL7 superconducting quantum interference magnetometer.

III. RESULTS AND DISCUSSION

A. WMR spectra

WMR spectra of $Cd_{1-x}Cr_xSe$ and $Cd_{1-x}V_xSe$ ternary alloys as a function of x are shown in Figs. 1(a) and 1(b), respectively, where *A* and *B* excitonic signatures are clearly observed. The excitonic features for the three Cd1−*^x*Cr*x*Se specimens shift with increasing *x* by 2.5, 5.2, and 7 meV with respect to the corresponding features for CdSe, while those for the available $Cd_{1-x}V_xSe$ specimens shift by 2.7 to 4.2 meV with respect to that of CdSe. Both Figs. $1(a)$ and $1(b)$ indicate that $Cd_{1-x}Cr_xSe$ and $Cd_{1-x}V_xSe$ ternary alloys with good optical quality have resulted in the crystal growth.

B. Magnetization and its relation with *s***-***d* **exchange energy**

Villeret *et al.*^{[21](#page-6-0)} have systematically investigated the lowlying electronic levels of transition-metal ions (TMIs) with the $3d^n$ electronic configuration, $n = 1$ to 9, in II-VI DMSs with either the zinc-blende or the wurtzite structure. The electronic configurations for Cr^{3+} , Cr^{2+} , and Cr^{+} are $3d^{3}$, $3d^4$, and $3d^5$, respectively, with the corresponding lowest terms ⁴*F* (*L* = 3, *S* = 3/2), ⁵*D* (*L* = 2, *S* = 2), and ⁶*S*_{5/2} $(S = 5/2, L = 0)$. The energy level schemes for Cr^{3+} , Cr^{2+} , and Cr^+ in II-VI DMSs when subjected to the crystal field, spin-orbit interaction, and Jahn-Teller effect (for Cr^{3+} and Cr^{2+} only) have been described in detail in our previous

FIG. 2. (Color online) Magnetization of Cd1−*^x*Cr*x*Se, after subtracting the diamagnetism of the host, as a function of *B* at several temperatures. The full circles are experimental data and the solid lines are the fitting of the magnetization with B_2 Brillouin function, and *x* ∼ 0*.*003.

paper.^{[22](#page-6-0)} Since the ground state of Cr^{2+} in the II-VI DMSs, e.g., in ZnTe and CdTe, is a singlet, 23.24 yet with magnetic excited states close enough for it to acquire a magnetic moment by mixing, the magnetization of Cr^{2+} exhibits properties intermediate between a van-Vleck-type and a Brillouin-type paramagnetism. However, consistent with the quenching of the orbital moments, the magnetization of Cr^{2+} in Cr-based $DMSs^{14,15,25}$ $DMSs^{14,15,25}$ $DMSs^{14,15,25}$ can be approximately fitted with a B_2 Brillouin function. The *x* values of the three $Cd_{1-x}Cr_xSe$ specimens used to obtained Fig. $1(a)$, deduced by fits to the magnetization with a *B*² Brillouin function, are approximately 0.001, 0.002, and 0.003. In Fig. 2, the magnetization of Cd1−*^x*Cr*x*Se specimens, measured in Fig. 1(a), as a function of *B*, with $x \sim 0.003$. At low temperatures (e.g., 2 K), the magnetization of $Cd_{1-x}Cr_xSe$ saturates at high *B*, in contrast to the absence of saturation expected for van Vleck paramagnetism^{10,11} at lowest $T(2 K)$ and highest B (6 T). However, the approach to saturation as a function of *B* at low temperatures is more gradual than that for the B_2 Brillouin function; in other words, the magnetism of Cr-based DMSs is between van-Vleck-type and Brillouin-type paramagnetism. To the extent that the fits for magnetization with B_2 are not exact at 2 and 5 K, the *x* values deduced should be considered as nominal.

Besides the Cr^{2+} (see Refs. [4,12–15,24–26\)](#page-6-0) charge state, Cr^+ and Cr^{3+} have also been reported in II-VI DMSs in microwave electron paramagnetic resonance (EPR) . $27-36$ While the observed signature of SFRS in Cd1−*^x*Cr*x*Se is clearly related to Cr^{2+} , we did not observe Raman-EPR features associated with Cr^+ . However, Raman-EPR signatures of Cr^+ in $Zn_{1-x}Cr_xTe$ and $Cd_{1-x}Cr_xTe$ were reported by us,²² but not for the spin-flip Raman scattering (SFRS) from donor-bound electrons. The reason may be related to crystal-growth conditions which determine whether the host semiconductor is a *p* or *n* type; it is well known that as-grown CdSe is an *n* type due to Se vacancies, while CdTe is a *p* type due to Cd vacancies. In $Zn_{1-x}Cr_xTe$ and $Cd_{1-x}Cr_xTe$, a small fraction of Cr^{2+} transforms into Cr^{+} acceptors by capturing an additional electron from the "anonymous" donors, and thus displays Raman-EPR of Cr^+ , but leaving insufficient neutral donors for observing spin-flip Raman scattering. However, in $Cd_{1-x}Cr_xSe$, it appears that no Cr^+ occurs as the absence of Raman-EPR of Cr⁺ shows, whereas the observation of SFRS is evidence for the presence of neutral donors. It is not clear why no EPR and Raman-EPR of Cr^{2+} , Cr^{+} , and Cr^{3+} in Cd1−*^x*Cr*x*Se have been reported.

Vanadium in II-VI DMSs also occurs in V^{3+} , V^{2+} , and V^+ charge states, as observed in EPR $37-40$ and optical measurements.^{41–43} It was also found that V^{3+} can be photo converted into V^{2+} with the light of an appropriate photon energy. The electronic configurations for V^{3+} , V^{2+} , and V^{+} are $3d^2$, $3d^3$, and $3d^4$, respectively, with the corresponding lowest terms ³F ($L = 3$, $S = 1$), ⁴F ($L = 3$, $S = 3/2$), and ⁵D $(L = 2, S = 2)$. V^{2+} (V^+) has the same electronic configuration as Cr^{3+} (Cr^{2+}), and thus can be expected to display a similar internal energy-level structure. When subjected to the crystal field and spin-orbit interaction, V^{3+} is expected to display an isotropic paramagnetic behavior described by a *B*¹ Brillouin function.^{[19](#page-6-0)} V²⁺ (Cr³⁺) and V⁺ (Cr²⁺) are subjected to a static Jahn-Teller distortion, and the magnetization of V^{2+} (Cr^{3+}) can be well described by $B_{3/2}$, while that of V^+ (Cr^{2+}) can be approximately described by*B*2. Since the magnetization of Cd1−*^x*V*x*Se after subtracting the diamagnetic contribution of the host, as shown in Fig. 3, can be quite precisely fitted

FIG. 3. (Color online) Magnetization of Cd_{1−*x*}V_{*x*}Se, after subtracting the diamagnetism of the host, as a function of *B* at several temperatures. The full circles are experimental data and the solid lines are the fitting of the magnetization with $B_{3/2}$ Brillouin function.

with a $B_{3/2}$ Brillouin function, we conclude that a significant majority of the vanadium ions in $Cd_{1-x}V_{x}Se$ is in the V^{2+} charge state. The *x* value thus deduced with a $B_{3/2}$ fit in Fig. 3 yields the same number (8.6×10^{-4}) as deduced from the data at four different temperatures. Even though the vanadium concentration incorporated in CdSe is very small, it is sufficient to observe SFRS from donor-bound electrons in Cd_{1−*x*}V_{*x*}Se under resonant conditions.

The Raman shift of SFRS from donor-bound electrons in DMSs is given by $19,44$

$$
\hbar\omega_{\text{SFRS}} = g^* \mu_B B - x \alpha N_0 \langle\!\langle S_z \rangle\!\rangle, \tag{1}
$$

where *g*[∗] is the intrinsic *g* factor of the host crystals (CdSe in this case), x is the concentration of magnetic ions (e.g., Cr), $N_0\alpha$ is the *s*-*d* exchange constant between the delocalized electrons in the conduction band or donor-bound electrons in large effective mass orbits, on the one hand, and the localized magnetic ions, on the other, and $\langle S_z \rangle$ is the thermal and spatial average of the magnetic ion spin projection along the direction of the external magnetic field. $\langle S_z \rangle$ can be expressed in terms of macroscopic magnetization *M*[∗] *^m* (Ref. [9\)](#page-6-0) as

$$
M_m^* = -\frac{g_S \mu_B N_A}{W(x)} x \langle\langle S_z \rangle\rangle, \tag{2}
$$

where N_A is the Avogadro's number, $W(x)$ is the molar weight of the DMS, and g_S is the *g* factor of the magnetic ion. M_m^* can be deduced from the measured magnetization *Mm* after subtracting the diamagnetic contribution of the host, $M_m^* =$ $M_m - \chi_{dia} H$. Equation (1) thus becomes

$$
\hbar\omega_{\text{SFRS}} = g^* \mu_B B + \alpha N_0 \frac{W(x)}{g_S \mu_B N_A} M_m^*.
$$
 (3)

FIG. 4. (Color online) First- and second-order Stokes and anti-Stokes spin-flip Raman scattering (SFRS) from donor-bound electrons in CdSe recorded at 8 K and 6 T.

For $Cd_{1-x}Cr_xSe$, $g^* = 0.52$, as shown in the inset of Fig. [4,](#page-2-0) and $g_S = 1.887$ (see Ref. [25\)](#page-6-0). The second term in Eq. [\(3\)](#page-2-0) is called the *s*-*d* exchange energy, which is linear with respect to *M*[∗] *^m* and its linear least-squares fit yields the *s*-*d* exchange constant, $N_0\alpha$.

C. Spin-flip Raman scattering (SFRS) in CdSe, Cd1−*^x***Cr***x***Se,** and $Cd_{1-x}V_xSe$

Raman spectra were recorded in the backscattering geometry and a right-hand laboratory coordinate system (**h***,***k***,***v**) defined by the magnetic field **B** along the horizontal (**h**) direction as well as the direction of propagation of the incident light by \bar{k} . Strong resonant Raman scattering, with the incident photon energy close to the excitonic band gap, is observed in the $\bar{\mathbf{k}}(\mathbf{h}\mathbf{h})\mathbf{k}$ configuration with the mutually orthogonal \mathbf{k} , $\hat{\mathbf{c}}$, and **B**. The Raman spectrum of CdSe recorded at $T = 8$ K and $B = 6$ T under an excitonic resonant condition is shown in Fig. [4.](#page-2-0) Stokes (S) and anti-Stokes (AS) Raman lines were observed with the Raman shift linear in magnetic field and independent of temperature over the range 2–10 K. Figure [4](#page-2-0) also shows the second order of the Stokes (2S) and anti-Stokes (2AS) Raman features. The Raman lines are attributed to SFRS from donor-bound electrons, as discussed below. The SFRS Raman shift in an intrinsic CdSe is given by

$$
\hbar \omega_{\text{SFRS}} = g^* \mu_B B, \tag{4}
$$

where g^* is the intrinsic *g* factor, μ_B is the Bohr magneton, and *B* is the magnetic induction. The linear least-squares fit of the *B* dependence of the SFRS Raman shift, as shown in the inset of Fig. [4,](#page-2-0) yields $g^* = 0.52$, which is identical to that obtained by Henry *et al.*[45](#page-7-0)

The PL spectrum of the same CdSe sample displays a strong peak (*E*PL) at 1.82025 eV, whereas the *A*-exciton signature in the WMR measurements in Fig. $1(a)$ occurs at 1.8238 eV. We assign the former to an exciton bound to a shallow donor and the latter to the free *A* exciton; these values are consistent with those reported by Yu *et al.*[46](#page-7-0) at 1.819 and 1.8231, respectively. Figure $5(a)$ shows the resonance profile of Stokes and anti-Stokes components of SFRS from donorbound electrons in CdSe. Since the incident photon energy for resonance enhancement is around E_{PL} , it is reasonable to attribute the microscopic process of the spin-flip Raman transitions of donor-bound electrons as mediated via exciton *A* bound to a shallow neutral donor. The differential scattering cross section⁴⁷ can be written as

$$
\frac{d\sigma}{d\Omega} \sim f^2 \bigg(\frac{e^2}{mc^2}\bigg)^2 \frac{\omega_s}{\omega_L} \frac{(\hbar \omega_L)^2}{(E_i - \hbar \omega_L)^2 + (\Gamma/2)^2},\tag{5}
$$

where *f* is the oscillator strength for an intermediate state at energy E_i , Γ is a phenomenological broadening parameter, and $\hbar\omega_L$ is the incident photon energy. As pointed out by Heiman *et al.*, [48](#page-7-0) the bound exciton contributes much more than the free excitons since the oscillator strength for bound excitons is much larger, and the contribution from bound excitons *B* and *C* is neglected due to the energy range. Thus, only bound exciton *A* dominates in the above Raman process, which is consistent with the resonance enhancement occurring near E_{PL} associated with donor-bound exciton A in Fig. $5(a)$.

FIG. 5. (Color online) The intensities of the Stokes and the anti-Stokes Raman lines in $Cd_{1-r}Cr_rSe$ as a function of the incident photon energy, i.e., the resonance profiles, recorded at $T = 5$ K and $B = 6$ T. E_{PL} is the peak position of the photoluminescence (PL) spectrum. FWHM is the full width at half maximum.

Figure [6](#page-4-0) shows the Raman spectrum in $Cd_{1-x}Cr_{x}Se$ with $x = 0.002$, recorded at [6](#page-4-0) T and 5 K. The inset of Fig. 6 shows the *x* dependence of the Raman signature. The Raman shift of SFRS from donor-bound electrons in CdSe is 1.5 cm⁻¹. whereas the Raman signatures in $Cd_{1-x}Cr_xSe$ with $x = 0.002$ and 0.003 in Fig. [6](#page-4-0) occur at 5.7 and 9.7 cm⁻¹, respectively. The Raman signature is also temperature and magnetic field dependent, as shown in Fig. [7.](#page-4-0) The intensity of the Stokes and the anti-Stokes Raman components in Cd1−*^x*Cr*x*Se, recorded at 6 T and 5 K, as a function of the incident and scattered photon energy, i.e., the resonance profile, are shown in Figs. 5(b) and $5(c)$ for $x = 0.002$ and 0.003, respectively. As in the case of CdSe shown in Fig. $5(a)$, the resonant enhancement of the Raman signature in both Cd1−*^x*Cr*x*Se specimens occurs when the incident photon energy approaches the corresponding photoluminescence energy (E_{PL}) of the donor-bound exciton *A*. The *x*, *T*, and *B* dependence, together with the resonant profile of the Raman signature in Cd1−*^x*Cr*x*Se, are clearly characteristics of Raman shifts due to the spin flip of shallow neutral donor-bound electrons. The solid curves in Fig. 5 are fits to the data using Eq. (5) . The full width at half maximum (FWHM) of the resonant profile for Cd1−*^x*Cr*x*Se increases with x . As shown in Eq. (5) , the scattering cross section is directly related to the intermediate state [energy E_i in Eq. (5)], i.e., one of the Zeeman states of an exciton *A* bound to a neutral donor. For CdSe, the Zeeman components of the bound exciton merge into a single broad band with a relatively small FWHM of 0.8 meV. With the increase of *x*, the separation of

FIG. 6. (Color online) Raman spectrum of SFRS from donorbound electrons in Cd_{1−*x*}Cr_{*x*}Se recorded at *B* = 6 T and $T = 5$ K, displaying the Stokes (S) and anti-Stokes (AS) Raman lines. The inset shows the *x* dependence of the Stokes Raman shift.

the Zeeman components in Cd1−*^x*Cr*x*Se increase, resulting in the broader band for $x = 0.002$ with FWHM of 2.4 meV, and eventually evolves into three separate peaks for $x = 0.003$.

Figure 7 shows the *s*-*d* exchange energy, as well as M_m^* , for $Cd_{1-x}Cr_xSe$ as a function of the magnetic field at several temperatures. The *s*-*d* exchange energy versus magnetization

FIG. 7. (Color online) The *s*-*d* exchange energy (full circles), as well as magnetization (solid lines), after subtracting the host diamagnetism vs *B* plotted at $T = 2, 5, 10,$ and 20 K for Cd_{1−*x*}Cr_{*x*}Se $(x = 0.003)$.

FIG. 8. (Color online) The *s*-*d* exchange energy vs magnetization at *T* = 2 K for Cd_{1−*x*}Cr_{*x*}Se (*x* = 0.003); the solid line is the linear least-squares fit.

at 2 K is displayed in Fig. 8 for $x = 0.003$, and for $x = 0.002$ in the inset. The linear least-squares fit yields $N_0\alpha$ to be 215 \pm 14 and 205 ± 12 meV, for $x = 0.003$ and 0.002, respectively. The average $N_0\alpha$ for Cd_{1−*x*}Cr_{*x*}Se is thus 210 ± 13 meV, which is their value of $N_0\alpha$, 220 ± 20 meV, for Cd_{1−*x*}Cr_{*x*}S.

Figure [9](#page-5-0) shows the Raman spectrum of SFRS in $Cd_{1-x}V_{x}Se$, recorded at 6 T and 5 K, and the inset shows its *x* dependence. The Raman shift of SFRS in $Cd_{1-x}V_xSe$ $(x = 8.6 \times 10^{-4})$ in Fig. [9](#page-5-0) is 3.7 cm⁻¹ at 6 T and 5 K, with 2.1 cm−¹ attributed to the *s*-*d* exchange interaction and 1.5 cm−¹ attributed to the intrinsic Zeeman splitting of CdSe. The PL spectra of Cd_{1−*x*}V_{*x*}Se ($x = 8.6 \times 10^{-4}$) in Fig. [10\(a\)](#page-5-0) shows donor-bound excitonic features at 1.8230 eV, which is close to the incident photon energies for resonance enhancement in Fig. [10\(b\)](#page-5-0) for the Stokes at 1.8237 eV and for the anti-Stokes at 1.8232 eV. Thus the stronger resonance peaks are attributed to the SFRS mediated via an exciton *A* bound to a shallow neutral donor. The incident photon energies for the weaker resonance peaks in Fig. [10\(b\)](#page-5-0) for the Stokes at 1.8283 eV and for the anti-Stokes at 1.8276 eV are close to the *A* exciton of $Cd_{1-x}V_x$ $Cd_{1-x}V_x$ $Cd_{1-x}V_x$ Se ($x = 8.6 \times 10^{-4}$) in Fig. 1 at 1.8280 eV, thus we assign the weak features in Fig. $10(b)$ as SFRS mediated via a free exciton *A*. For the detailed microscopic mechanisms of the above Raman processes, we refer to Ref. [19.](#page-6-0)

Figure [11](#page-5-0) shows the *s*-*d* exchange energy, as well as the magnetization, for $Cd_{1-x}V_xSe$ as a function of *B* at several temperatures. Both the *s*-*d* exchange energy and magnetization show saturation at high *B* and low *T*, which is characteristic of the Brillouin-type paramagnetism. For Cd_{1−*x*}V_{*x*}Se, we assume $g_S = 2$ (Ref. [19\)](#page-6-0) in Eq. [\(3\)](#page-2-0) since the exact value for V in CdSe is not available at the present time. The *s*-*d* exchange

0.4

squares, and triangles), as well as magnetization (solid lines) after subtracting the host diamagnetism vs *B* plotted at $T = 2, 5, 10,$ and 20 K for Cd_{1−*x*}V_{*x*}Se (*x* = 0.003).

FIG. 9. (Color online) Raman spectrum of SFRS from donorbound electrons in $Cd_{1-x}V_x$ Se recorded at *B* = 6 T and *T* = 5 K. The Stokes (S) and anti-Stokes (AS) Raman lines are shown. The inset shows the *x* dependence of the Stokes Raman shift.

energy versus magnetization at 2 K is plotted in Fig. 12, and the linear least-squares fit yields the *s*-*d* exchange constant

FIG. 11. (Color online) The *s*-*d* exchange energy (full circles,

FIG. 10. (Color online) (a) Photoluminescence (PL) spectrum of Cd1−*^x*V*x*Se showing a strong donor-bound *A*-exciton feature. (b) The intensities of the Stokes and the anti-Stokes Raman lines as a function of the incident photon energy, i.e., the resonance profiles, recorded at $T = 5$ K and $B = 6$ T.

FIG. 12. (Color online) The *s*-*d* exchange energy vs magnetization at $T = 2$ K for Cd_{1−*x*}V_{*x*}Se ($x = 0.003$); the solid line is the linear least-squares fit.

IV. CONCLUDING REMARKS

The accurate determination of $N_0\alpha$ in Cr-based II-VI DMSs plays an important role in deducing $N_0\beta$ and thus confirming their ferromagnetic character of the *p*-*d* exchange interaction.^{4–7} The SFRS investigation has yielded $N_0\alpha$ in $Cd_{1-x}Cr_xSe$ and $Cd_{1-x}V_xSe$ (with very small *x*), respectively. Compared to the Mn-based III-V DMSs,^{[49](#page-7-0)} i.e., $Ga_{1-x}Mn_xAs$ and In1−*^x*Mn*x*As, which display ferromagnetism only up to 170 K, $\text{Zn}_{1-x}\text{Cr}_x\text{Te}$ ($x = 0.2$) shows it up to room temperature.16,17 As an illustrative example for a potential spintronic application of such a semiconductor, we cite the strong resonant tunneling effect exploited in Ga_{1−*x*}Mn_{*x*}As as a magnetic tunnel transistor.⁵⁰

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