Magnetic phase diagram in the Co-rich side of the $L\text{Co}_{1-x}\text{Fe}_x\text{AsO}$ **(** $L = \text{La}, \text{Sm}$ **) system**

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The magnetic phase diagram has been mapped out via the measurements of electronic resistivity, magnetization, and specific heat in the cobalt-based layered $LCo_{1-x}Fe_xAsO (L = La, Sm)$ compounds. The ferromagnetic (FM) transition at ∼63 K for LaCoAsO is rapidly suppressed upon Fe doping, and ultimately disappears around *x* = 0.3 in the LaCo1−*^x*Fe*x*AsO system. When La is replaced by magnetic rare-earth element Sm, the 3*d* electrons first undergo a FM transition at *Tc* ∼ 75 K, followed by an antiferromagnetic (AFM) transition at a lower temperature T_{N1} ~ 45 K. With partial Fe doping on the Co site, both FM (*T_c*) and AFM (*T_{N1}*) transition temperatures are significantly suppressed, and finally approach 0 K at *x* = 0*.*3 and 0.2, respectively. Meanwhile, a third magnetic transition at $T_{N2} \sim 5.6$ K for SmCoAsO, associated with the AFM order of the Sm³⁺ 4*f* moments, is uncovered and T_{N2} is found to be almost robust against the small Fe doping. These results suggest that the $4f$ electrons of Sm3⁺ have an important effect on the magnetic behavior of 3*d* electrons in the 1111 type Co-based *L*Co1−*^x*Fe*x*AsO systems. In contrast, the magnetism of the *f* electrons is relatively unaffected by the variation of the 3*d* electrons. The rich magnetic phase diagram in the Co-rich side of the *L*Co_{1−*x*}Fe_xAsO system, therefore, is established.

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I. INTRODUCTION

Correlated electron systems $LTPO$ ($L =$ rare-earth element, $T =$ transition-metal element, $P =$ pnictogen element) have attracted great attention due to their various electronic and magnetic properties, such as high transition temper-ature superconductivity,^{[1](#page-5-0)} itinerant ferromagnetism,^{[2](#page-5-0)[,3](#page-6-0)} giant magnetoresistance, 4 spin density wave (SDW), 5 and structural instability.⁶ For example, iso-structural LaOMnAs is an AFM semiconductor, $\frac{7}{1}$ and LaONiAs shows superconductivity be-low 3 K.^{[8](#page-6-0)} In the case of $T = Co$, *LCoAsO* was reported to be an itinerant ferromagnet with the Curie temperature T_c between 60 and 80 K for La and Sm, respectively.^{[2](#page-5-0)[,9](#page-6-0)} Among them, the compound LaFeAsO, which is a parent compound of wellknown iron-based superconductors, exhibits a spin-density wave antiferromagnetic transition at about 150 K.¹ When Fe is partially replaced by Co atoms, the AFM order from Fe ions is suppressed and then superconductivity emerges, and the compound exhibits a good metallic behavior down to the superconducting transition temperature.^{10,11} Similar results have also been reported for Co-doped CeFeAsO,^{[12](#page-6-0)} PrFeAsO, 13,14 NdFeAsO, 15 15 15 and SmFeAsO 11 systems. Thus, magnetism is closely related to superconductivity in these iron-based high-temperature superconductors.

On the other hand, the *L*CoAsO compounds (also referred to as Co-1111 system) with the same space group as ZrCuSiAs exhibit rich magnetic properties at low temperature. LaCoAsO is reported to be an itinerant ferromagnet with two-dimensional ferromagnetic spin fluctuations. $2,16$ $2,16$ When La is substituted by other magnetic rare-earth elements, *L*CoAsO (*L* = Nd, Sm, and Gd $)^{9,17-21}$ undergoes multiple magnetic phase transitions as the temperature decreases. Furthermore, the AFM order due to the magnetic sublattice of *L* ions at very low temperature can be also observed in those compounds, almost irrelevant to the doping at *T* or *P* sites. For example, in the case of $SmCoAsO,^{20}$ $SmCoAsO,^{20}$ $SmCoAsO,^{20}$ a ferromagnetic (FM) transition occurs around T_c of 75 K, followed by a FM-AFM transition from the magnetic coupling between the CoAs layers around 45 K, and finally another AFM order from Sm ion forms at 5.6 K as recently reported by other groups.^{18,22,23} Indeed, several groups^{[9,24](#page-6-0)} have suggested that the Ruderman-Kittel-Kasuya-Yoshida (RKKY) interaction may play a role in the FM-AFM transition. It is ascribed to the interaction between the localized magnetic moments of lanthanide 4*f* electrons and the ferromagnetic ordered magnetic moments of cobalt 3*d* itinerant electrons. The neutron diffraction experiments $15,17$ and specific heat measurements $18,22,23$ have detected the localized magnetic moment of *L* 4*f* electrons in those parent compounds.

Up to now, main studies about Co-containing 1111 system focus on these low Co concentrations[10–15](#page-6-0) and *L*Co*P*O parent compounds.^{2,[4,9,17–20](#page-6-0)} There are few reports on the study of chemical doping in *LCoAsO*,^{[25](#page-6-0)} and the magnetic phase diagram on the Co-rich side of *L*CoAsO is less known. In this paper, we report our detailed study of the magnetic properties of Fe-doped *L*Co1−*^x*Fe*x*AsO (*L* = La, Sm) system on the Co-rich side. To study the interplay between 4*f* electrons and 3*d* electrons, the LaCo_{1−*x*}Fe_{*x*}AsO system is employed as a comparison. We performed powder x-ray diffraction, electrical resistivity, and magnetization measurements, as well as the first-principles calculations. The results of these measurements and calculations indicate that the FM order is quickly suppressed by Fe doping in the LaCo_{1−*x*}Fe_{*x*}AsO system, and finally disappears at about $x = 0.3$. In the case of SmCo1−*^x*Fe*x*AsO, the FM and AFM transitions of the 3*d* electrons are gradually suppressed and then disappear at $x = 0.3$ and 0.2, respectively. However, the AFM order at low temperature due to Sm^{3+} is robust and T_{N2} slightly varies with increasing Fe content. A rich magnetic phase diagram for the $x \le 0.3$ *L*Co_{1−*x*}Fe_{*x*}AsO system is therefore established.

II. EXPERIMENTAL

The polycrystalline samples of $LCo_{1-x}Fe_xAsO$ ($L = La$, Sm) were synthesized by two-step solid state reaction methods

in vacuum, similar to our previous reports.¹¹ The pellets of $LCo_{1−x}Fe_xAsO (x = 0, 0.05, 0.1, 0.2, 0.3)$ were annealed in an evacuated quartz tube at 1423 K for 40 h and furnace-cooled to room temperature.

Crystal structure measurement was performed by powder x-ray diffraction (XRD) at room temperature using a D/MaxrA diffractometer with Cu K_{α} radiation and a graphite monochromator. Lattice parameters were calculated by leastsquares fitting using at least 20 XRD peaks. The electrical resistivity was measured by the four-terminal method. The temperature dependence of dc magnetization was measured on a Quantum Design Magnetic Property Measurement System (MPMS-5). The measurement of specific heat was performed on on a Quantum Design Physical Property Measurement System (PPMS-9).

The magnetic properties of LaCo1−*^x*Fe*x*As were calculated using the plane-wave basis pseudopotential method implemented in the Quantum ESPRESSO package. The exchangecorrelation interactions were modeled with the Perdew, Burke, and Enzerhoff flavor of generalized gradient approximation.²⁶ To model the dilute substitutional iron doping effect, a virtual crystal approximation (VCA) was employed to treat Fe and Co sites.

III. RESULTS AND DISCUSSION

A. Magnetic properties in LaCo1−*^x***Fe***x***AsO**

Figure $1(a)$ shows the powder XRD patterns of the typical LaCo_{1−*x*}Fe_{*x*}AsO samples and Fig. 1(b) shows the variations of lattice parameters with respect to the Fe content (*x*). Where the main diffraction peaks of those samples can be well indexed based on a tetragonal cell of ZrCuSiAs-type structure, weak peaks exist due to impurity phase CoAs. The content of impurity phase CoAs estimated by Rietveld fitting is less than 5%. It is worth noting that CoAs has been reported to be nonmagnetic from 4.[2](#page-5-0) to 300 K.² The a axis decreases slightly with increasing Fe content, and the *c* axis increases accordingly, resulting in the increase of the cell volume, since the ionic radius of a tetrahedrally coordinated $Fe²⁺$ ion is larger than that of $Co²⁺$. The systematic increase in the *c* axis indicates successful substitution of Co by Fe. Similar variations of lattice constants were also observed in the NdFe1−*^x*Co*x*AsO in previous reports[.25](#page-6-0)

Figure [2](#page-2-0) shows the temperature dependence of the electric resistivity ρ and magnetic susceptibility χ of LaCo_{1−*x*}Fe_{*x*}AsO samples. The inset shows magnetic susceptibility vs temperature between 100 and 300 K. In Fig. $2(a)$, for LaCoAsO, the resistivity falls monotonically with decreasing temperature from 300 K, and a resistivity hump can be clearly identified at about 63 K, which is related with the FM transition temperature *Tc*. As Fe content increases to 0.1, this hump shifts to about 35 K, and then for $x = 0.2$, the anomaly in resistivity is not observed. Actually, the anomaly around T_c becomes more obvious in the derivative of ρ shown in Fig. [2\(b\),](#page-2-0) where T_c decreases with increasing Fe content and shifts to below 2 K at $x = 0.3$. Meanwhile, the resistivity value gradually increases with the Fe doping levels, which can be attributed to the less itinerant nature of the Fe 3*d* electron than that of Co. The magnetic susceptibility for the LaCo1−*^x*Fe*x*AsO

FIG. 1. (Color online) Structural characterization of LaCo1−*^x*Fe*x*AsO samples. (a) Powder XRD patterns of representative LaCo1−*^x*Fe*x*AsO samples. The asterisked peak positions designate the impurity phase of CoAs. (b) Lattice parameters as functions of Fe content.

under $H = 1$ kOe in the zero field cooled (ZFC) configuration was plotted in Fig. $2(c)$. For the parent LaCoAsO sample, the magnetic susceptibility increases dramatically below T_c of 63 K, suggesting that the Co sublattice forms FM order in the CoAs layer. A similar magnetic behavior has been reported in the literature.^{[2](#page-5-0)} As Co is partially replaced by Fe, the FM transition temperature T_c is sharply suppressed and shifts to a lower temperature. For $x = 0.3$, the formation of long-range FM order cannot be identified in $\chi(T)$ down to 2 K. Furthermore, the magnetic susceptibility value drops to several orders of magnitude of LaCoAsO. On the other hand, it can be seen from the inset that the magnetic susceptibility curve above 100 K exhibits the Curie-Weiss behavior for LaCoAsO. As Fe content increases, the susceptibility gradually becomes less *T* dependent and finally remains constant for $x = 0.3$, indicating that Fe doping strongly reduces the moment of 3*d* electrons in Co-based 1111 compounds.

Figure [3](#page-2-0) shows the *M*-*H* loop curves at several temperatures for LaCo_{1−*x*}Fe_{*x*}AsO. For $x \le 0.2$, the *M*-*H* curves are nearly linear above T_c , indicating that those compounds are paramagnetic at these temperatures. Below T_c , these curves deviate from linearity and become slightly S shaped, suggesting the emergence of FM order. Further decreasing

FIG. 2. (Color online) (a) Temperature dependence of resistivity *ρ* for the LaCo_{1−*x*}Fe_{*x*}AsO (*x* = 0, 0.1, 0.2, 0.3) samples. (b) The derivative of resistivity near the ferromagnetic phase transition. (c) Temperature dependence of magnetic susceptibility *χ* under a magnetic field of 1000 Oe in the ZFC configuration.

temperature to 3 K, the molar magnetization sharply increases and then saturates with the increase of magnetic field, and the small finite hysteresis can be distinguished (the data are not shown here). These results suggest that the ground state of these samples is FM. For $x = 0.3$, the *M*-*H* curve always shows the linear behaviors above 3 K, indicating that this sample remains paramagnetic. The magnetic moment M_0 estimated by extrapolating the M - H curves to $T = 0$ K is 0.35 μ _B per Co for LaCoAsO, which is very close to the value reported previously.² With increasing Fe content, M_0 quickly decreases, consistent with the fact that the FM transition temperature T_c shifts to lower temperature.

Figure 4 shows the Local Density Approximation (LDA) calculation results of LaCo1−*^x*Fe*x*As system. The ground state of the system was determined by comparing the total energy of three possible long-range magnetic orderings, i.e., ferromagnetic (FM), checkerboard antiferromagnetic (CB-AFM), and stripe-like antiferromagnetic (SDW-AFM), as well as the

FIG. 3. (Color online) Field dependence of magnetization at various temperatures for the LaCo_{1−*x*}Fe_{*x*}AsO ($x = 0, 0.1, 0.2, 0.3$) samples.

nonmagnetic (NM) configurations. As shown in Fig. 4(a), the ground state is clearly FM at the cobalt side $(x = 0.0)$, where the CB-AFM order cannot be stabilized over the whole range we considered $(0.0 \le x \le 0.5)$. As *x* increases, the energy of FM configuration quickly rises, suggesting the iron

FIG. 4. (Color online) LDA results of LaOCo_{1−*x*}Fe_{*x*}As properties. (a) Magnetic configuration energy (per Fe atom) with respect to NM total energy. (b) NM lattice constants variation with respect to *x*. (c) Arsenic height z_{As} and transition-metal magnetic moment m_T ; the dashed blue line is the actual LDA results for FM phases, while the solid blue line suggests the actual scenario with phase transition taken into consideration.

doping will suppress the formation of FM long-range order. At *x* ∼ 0*.*25, the SDW-AFM order takes over and becomes the ground state. However, one should keep in mind that the disorder effect is not fully taken care of in VCA, and a disordered dopant pattern is detrimental to the formation of AFM long-range order. Furthermore, the magnetic coupling strength is suppressed with increasing *x* from $x = 0.2$ to 0.5, as suggested by the increasing configuration energy of both FM and SDW-AFM orders. With these considerations, we conclude that the actual ground state should be paramagnetic with local magnetic fluctuations.

We then compare the LDA lattice structure variation in Fig. [4\(b\).](#page-2-0) As the experimental measurements were performed at room temperature when the long-range magnetic order was not yet formed, we compare the NM lattice constants. It could be seen that between $x \in [0.0, 0.3]$, the LDA lattice constants show the same trend as the experimental results but with much larger variation. Beyond $x = 0.3$, the lattice constants show much less variation. For the FM phase of LaCo_{1−*x*}Fe_{*x*}AsO, we also examine its magnetic moment per transition metal m_T and the arsenic height z_{As} [Fig. [4\(c\)\]](#page-2-0). At $x = 0.0$, the LaOCoAs compound has a small moment of 0.6 μ_B/Co , indicating a weak FM ground state with low Curie temperature. As the doping level *x* increases, the moment almost linearly decreases to $x = 0.3 \mu_B/T$ at $x = 0.3$ at the VCA level. As discussed above, the disorder effect will further suppress the moment. Meanwhile, z_{As} also increases almost linearly with respect to *x* from ~1.18 Å to ~1.24 Å. The variation of z_{As} is consistent with the change of the superexchange via arsenic, which becomes less FM and more AFM with increasing *x*.

B. Magnetic properties in SmCo1−*^x***Fe***x***AsO**

Figure 5(a) shows the powder XRD patterns of SmCo_{1−*x*}Fe_{*x*}AsO samples and Fig. 5(b) shows the variations of lattice parameters with Fe content (x) . All those samples are single phase since no extra peak is observed. Similar to the case of LaCo1−*^x*Fe*x*AsO, Fe doping causes slight decrease in the *a* axis, while the *c* axis monotonously increases.

Figure [6](#page-4-0) shows the temperature dependence of the electric resistivity and magnetic susceptibility of the SmCo1−*^x*Fe*x*AsO samples. In Fig. $6(a)$, for the undoped parent compound $SmCoAsO₁²⁰$ $SmCoAsO₁²⁰$ $SmCoAsO₁²⁰$ the resistivity monotonically decreases with decreasing temperature from 300 K, followed by a distinguishable kink around 45 K which can be associated with the FM-AFM transition temperature (defined as T_{N1}). As Fe content increases to 0.1, this kink becomes more pronounced and moves to lower temperatures, and no anomaly is observed below T_{N1} . For $x = 0.2$, the resistivity anomaly related with T_{N1} disappears, but another tiny kink can be identified around 5.1 K, which can be attributed to the AFM transition (T_{N2}) due to the magnetic sublattice of Sm ions. As *x* increases to 0.3, T_{N2} in the resistivity becomes more remarkable [shown in inset of Fig. $6(a)$]. To identify the magnetic ordering transition more clearly, the derivative of resistivity below 150 K is plotted in Fig. $6(b)$, which shows a maximum around T_c , a minimum near T_{N1} , and a peak at T_{N2} . With increasing Fe content, the maximum around T_c shifts to lower temperatures, and the minimum near T_{N1} even becomes more pronounced for $x = 0.1$ and disappears as $x = 0.2$. The

FIG. 5. (Color online) Structural characterization of SmCo1−*^x*Fe*x*AsO samples. (a) Powder XRD patterns of representative SmCo1−*^x*Fe*x*AsO samples. (b) Lattice parameters as functions of Fe content.

magnetic susceptibility data are shown in Fig. $6(c)$; it can be seen that SmCoAsO shows a sharp peak around 45 K. This peak is ascribed to the FM to AFM transition of the cobalt sublattice, which has been reported for $L = Nd$, Sm, and Gd.⁹ As Fe substitutes Co, T_{N1} is gradually suppressed and shifts to lower temperatures. Meanwhile, the intensity of the peaks becomes weaker, consistent with the resistivity data. At $x = 0.2$, it is noted that a tiny hump around 6.5 K is observed, which may not be explained by the AFM transition T_{N1} according to our *M-H* data (see Fig. [8\)](#page-4-0). The detailed discussion is beyond the scope of the current work and will be given in the future. When *x* increases to 0.3, the magnetization sharply drops and a tiny peak is detected at 5.5 K. Considering the specific heat data in Fig. [7,](#page-4-0) this transition (T_{N2}) is attributed to the AFM ordering of the Sm^{3+} sublattice.^{[22,23,27](#page-6-0)} Similar results can be found in previous papers.^{15,17,27} For $x < 0.3$ samples, such a low temperature peak in the magnetization curves is not observable due to the magnetic ordering of 3*d* electrons.

To further study the magnetic phase transition, the zero field specific heat versus temperature curves for those samples are summarized in Fig. [7.](#page-4-0) For all the samples, no anomaly in the curves is found around T_c associated with the FM transition, which is also the case in $NdCoAsO.¹⁷$ A small broad peak related to the AFM transition T_{N1} can be observed at 45 K for SmCoAsO and then shifts to 22 K for $x = 0.1$ (the data are not shown here). However, it is worth noticing that another clear

FIG. 6. (Color online) (a) Temperature dependence of resistivity ρ for the SmCo_{1−*x*}Fe_{*x*}AsO (*x* = 0, 0.05, 0.1, 0.2, 0.3) samples. (b) Derivative of resistivity below 150 K. The data are normalized to $(d\rho/dT)_{T=150 \text{ K}}$. (c) Temperature dependence of magnetic susceptibility *χ* under a magnetic field of 1000 Oe in the ZFC configuration. The inset shows the enlarged resistivity for $x = 0.2$ and 0.3 at low temperature.

broad peak from the AFM ordering of the Sm^{3+} sublattice can be observed at 5.5 and 3.5 K, respectively. As *x* increases to 0.2, no extra peak is detected except for the large anomaly near T_{N2} of 5 K, which then shifts to 5.5 K for $x = 0.3$, consistent with the magnetic susceptibility data shown in Fig. 6(c). Similar results are observed in the case of $SmFeAsO²⁷, SmCoAsO¹⁸$ $SmFeAsO²⁷, SmCoAsO¹⁸$ $SmFeAsO²⁷, SmCoAsO¹⁸$ and SmCoPO.²³ However, the hump in SmCoAsO related to the Sm AFM ordering becomes sharper and higher with increasing Fe content, which suggests that the increase of the *c* lattice weakens the coupling of 3*d* electron and 4*f* electron. Low-temperature specific heat measurements also confirm that the peak associated with T_{N2} is robust as $x \le 0.3$.

Figure 8 shows the field dependence of the magnetization at various temperatures for all the samples. As reported in previous works[,18,20](#page-6-0) SmCoAsO undergoes three magnetic phase transitions at T_c , T_{N1} , and T_{N2} , respectively. The magnetization sharply increases and saturates with increasing field between T_c and T_{N1} . To study the magnetic structure at different temperatures, these data are collected at temperatures where the PM or FM state dominates in Fig. $6(b)$ and at 2 K. Obviously, for $x \le 0.1$, the *M-H* curve is linear at 2 K and above T_c , and the S shape of FM behavior is only observed between T_c and T_{N1} . As x increases to 0.2, the linear feature

FIG. 7. (Color online) Specific heat of $\text{SmCo}_{1-x}\text{Fe}_x\text{AsO}$ ($x = 0$, 0.05, 0.1, 0.2, 0.3) samples at zero field below 100 K.

FIG. 8. (Color online) Magnetic field dependence of magnetization at several different temperatures for the $\text{SmCo}_{1-x}\text{Fe}_x\text{AsO}$ ($x=0$, 0.05 0.1, 0.2, 0.3) samples.

FIG. 9. (Color online) Magnetic phase diagram for the SmCo_{1−*x*}Fe_{*x*}AsO (solid symbols) and LaCo_{1−*x*}Fe_{*x*}AsO (open symbols) system, respectively. The transition temperatures were determined from the measurements of magnetic susceptibility, magnetization, and specific heat. T_{N2} is taken from resistivity and specific heat data.

is observed only above T_c , and the M - H curve displays FM behavior below 2 K, implying that the AFM transition T_{N1} approaches zero. This feature is different from the case of NdCo_{1−*x*}Fe_{*x*}AsO,^{[25](#page-6-0)} where both FM and AFM order are not observed around 0.2. At $x = 0.3$, the *M*-*H* curves goes back to the linear behavior at 2 K, indicating that the FM order from the Co sublattice is completely destroyed. On the other hand, with increasing Fe content, both the saturation moment and the transition temperature T_c gradually decrease, implying that the 3*d* electron magnetism becomes weaker.

Based on the above data, the magnetic phase diagram of *L*Co1−*^x*Fe*x*AsO is established in Fig. 9. In LaCo1−*^x*Fe*x*AsO, the T_c associated with FM transition is sharply suppressed and disappears around $x = 0.3$. At the same time, in the SmCo_{1−*x*}Fe_{*x*}AsO system, the T_{N1} from the FM to AFM transition of the Co sublattice is shifted to lower temperatures with increasing Fe content and such transition has not been observed at $x = 0.2$. Meanwhile, at the T_c , the FM order gradually decreases and is completely suppressed at $x = 0.3$, similar to the case of LaCo1−*^x*Fe*x*AsO. Here we note that for the Co-parent compounds, T_c increases slightly when La is replaced by Sm. But a new FM-AFM transition of the 3*d* electrons is induced in the latter case. This manifests an interesting interplay between the 3*d* electrons and the local Sm^{3+} moments.

Thus, in the 1111-type Co-based *L*Co_{1−*x*}Fe_{*x*}AsO systems, 4*f* electrons of rare-earth elements have an important effect on the magnetic behavior of 3*d* electrons. Whereas the antiferromagnetic transition temperature of Sm moments T_{N2} almost does not change within the whole doping regime.

This implies that the AFM ordering of the Sm 4*f* electrons is robust against Fe and Co substitution within the CoAs layer. Therefore, the microscopic origin of the *f*-electron AFM order of this system should be mainly due to the superexchange interactions between the f local moments.^{[28](#page-6-0)} These superexchange interactions are bridged by two kinds of *f* -*p* orbital hybridizations: one via the Sm-O path and another the Sm-As path. 28 Meanwhile, the RKKY interaction mediated by the charge carries within the CoAs layer may not play a crucial role in the *f* -electron magnetism because the RKKY interaction would be explicitly dependent on the variations of 3*d* electrons. Because the radius of the Sm^{3+} ion is smaller than that of La^{3+} , one may speculate that the FM-AFM transition is not related to the magnetic 4*f* electrons, but rather due to the enhancement of three dimensionality as the lattice parameter c decreases. To clarify this possibility, we also performed LDA calculations on LaCoAsO but used the lattice parameters of SmCoAsO. We found that the FM state of Co *d* electrons is robust against the decreasing lattice constant. Experimentally, similar results were always observed in NdCoAsO under pressure.²⁹ Therefore, we suggest that the FM-AFM transition at T_{N1} of the 3*d* electrons should be due to their coupling (polarization) to the 4*f* moments along the *z* direction.

IV. CONCLUSION

In the 1111 Co-based $LCo_{1-x}Fe_xAsO (L = La, Sm)$ system, a series of *L*Co1−*^x*Fe*x*AsO samples were synthesized and their transport and magnetic properties investigated. A rich magnetic phase diagram of the *L*Co_{1−*x*}Fe_{*x*}AsO systems is then established. The FM order is observed in both $LCoAsO (L =$ La,Sm) systems, and is completely destroyed with increasing Fe doping content to 0.3. Meanwhile, in SmCo1−*^x*Fe*x*AsO, T_{N1} is suppressed to below 2 K as $x = 0.2$, but the AFM order of the rare-earth element Sm ion survives in the whole doping regime $x \le 0.3$. This also indicates that the disorder effect induced by Fe and Co doping is very weak. Based on these results, it is concluded that in $LCO_{1-x}Fe_xAsO$ systems, while the magnetic properties of the 4*f* electrons of rare-earth elements are robust against the variations of 3*d* electrons, they do play a significant role in the magnetic behaviors of the 3*d* electrons. These materials therefore provide a prototypical testing ground for exploring the interplay between 4*f* and 3*d* electrons in transition-metal compounds.

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¹Y. Kamihara, T. Watanabe, M. Hirano, and H. Hosono, [J. Am.](http://dx.doi.org/10.1021/ja800073m) Chem. Soc. **130**[, 3296 \(2008\).](http://dx.doi.org/10.1021/ja800073m)

²H. Yanagi, R. Kawamura, T. Kamiya, Y, Kamihara, M. Hirano, T. Nakamura, H. Osawa, and H. Hosono, [Phys. Rev. B](http://dx.doi.org/10.1103/PhysRevB.77.224431) **77**, 224431 [\(2008\).](http://dx.doi.org/10.1103/PhysRevB.77.224431)

- 3H. Ohta and K. Yoshimura, Phys. Rev. B **79**[, 184407 \(2009\).](http://dx.doi.org/10.1103/PhysRevB.79.184407)
- 4H. Ohta, C. Michioka, and K. Yoshimura, [Phys. Rev. B](http://dx.doi.org/10.1103/PhysRevB.84.134411) **84**, 134411 [\(2011\).](http://dx.doi.org/10.1103/PhysRevB.84.134411)
- 5C. de la Cruz, Q. Huang, J. W. Lynn, J. Li, W. Ratcliff II, H. A. Mook, G. F. Chen, J. L. Luo, N. L. Wang, and P. C. Dai, [Nature](http://dx.doi.org/10.1038/nature07057) **453**[, 899 \(2008\).](http://dx.doi.org/10.1038/nature07057)
- ⁶M. Rotter, M. Tegel, and D. Johrendt, I. Schellenberg, W. Hermes, and R. Pttgen, Phys. Rev. B **78**[, 020503\(R\) \(2008\).](http://dx.doi.org/10.1103/PhysRevB.78.020503)
- ${}^{7}K$. Kayanuma, H. Hiramatsu, T. Kamiya, M. Hirano, and H. Hosono, J. Appl. Phys. **105**[, 073903 \(2009\).](http://dx.doi.org/10.1063/1.3093685)
- 8Z. Li, G. Chen, J. Dong, G. Li, W. Hu, D. Wu, S. Su, P. Zheng, T. Xiang, N. Wang, and J. Luo, Phys. Rev. B **78**[, 060504\(R\) \(2008\).](http://dx.doi.org/10.1103/PhysRevB.78.060504)
- 9H. Ohta and K. Yoshimura, Phys. Rev. B **80**[, 184409 \(2009\).](http://dx.doi.org/10.1103/PhysRevB.80.184409)
- ¹⁰A. S. Sefat, A. Huq, M. A. McGuire, R. Y. Jin, B. C. Sales, D. Mandrus, L. M. D. Cranswick, P. W. Stephens, and K. H. Stone, Phys. Rev. B **78**[, 104505 \(2008\).](http://dx.doi.org/10.1103/PhysRevB.78.104505)
- 11C. Wang, Y. K. Li, Z. W. Zhu, S. Jiang, X. Lin, Y. K. Luo, S. Chi, L. J. Li, Z. Ren, M. He, H. Chen, Y. T. Wang, Q. Tao, G. H. Cao, and Z. A. Xu, Phys. Rev. B **79**[, 054521 \(2009\).](http://dx.doi.org/10.1103/PhysRevB.79.054521)
- ¹²J. Prakash, S. J. Singh, S. Patnaik, and A. K. Ganguli, [Solid State](http://dx.doi.org/10.1016/j.ssc.2008.11.028) Commun. **149**[, 181 \(2009\).](http://dx.doi.org/10.1016/j.ssc.2008.11.028)
- 13X. Lin, H. J. Guo, C. Y. Shen, Y. K. Luo, Q. Tao, G. H. Cao, and Z. A. Xu, Phys. Rev. B **83**[, 014503 \(2011\).](http://dx.doi.org/10.1103/PhysRevB.83.014503)
- 14P. Shirage, K. Miyazawa, H. Kito, H. Eisaki, and A. Iyo, [Physica C](http://dx.doi.org/10.1016/j.physc.2008.12.012) **469**[, 898 \(2009\).](http://dx.doi.org/10.1016/j.physc.2008.12.012)
- 15A. Marcinkova, D. A. M. Grist, I. Margiolaki, T. C. Hansen, S. Margadonna, and J. G. Bos, Phys. Rev. B **81**[, 064511 \(2010\).](http://dx.doi.org/10.1103/PhysRevB.81.064511)
- 16H. Ohta, C. Michioka, and K. Yoshimura, [J. Phys. Soc. Jpn.](http://dx.doi.org/10.1143/JPSJ.79.054703) **79**, [054703 \(2010\).](http://dx.doi.org/10.1143/JPSJ.79.054703)
- ¹⁷M. A. McGuire, D. J. Gout, V. O. Garlea, A. S. Sefat, B. C. Sales, and D. Mandrus, Phys. Rev. B **81**[, 104405 \(2010\).](http://dx.doi.org/10.1103/PhysRevB.81.104405)
- 18V. P. S. Awana, I. Nowik, P. Anand, K. Yamaura, M. E. Takayama, and I. Felner, Phys. Rev. B **81**[, 212501 \(2010\).](http://dx.doi.org/10.1103/PhysRevB.81.212501)
- 19H. Ohta, C. Michioka, A. Matsuo, K. Kindo, and K. Yoshimura, Phys. Rev. B **82**[, 054421 \(2010\).](http://dx.doi.org/10.1103/PhysRevB.82.054421)
- 20Y. K. Li, J. Tong, H. Han, L. Zhang, Q. Tao, G. H. Cao, and Z. A. Xu, [Sci. China Phys. Mech. Astron.](http://dx.doi.org/10.1007/s11433-010-4001-6) **53**, 1194 (2010).
- 21J. Sugiyama, M. Mnsson, O. Ofer, K. Kamazawa, M. Harada, D. Andreica, A. Amato, J. H. Brewer, E. J. Ansaldo, H. Ohta, C. Michioka, and K. Yoshimura, Phys. Rev. B **84**[, 184421 \(2011\).](http://dx.doi.org/10.1103/PhysRevB.84.184421)
- 22A. Pal, M. Tropeana, S. D. Kaushik, M. Hussain, H. Kishan, and V. P. S. Awana, J. Appl. Phys. **109**[, 07E121 \(2011\).](http://dx.doi.org/10.1063/1.3551743)
- 23A. Pal, S. S. Mehdi, M. Husain, B. Gahtori, and V. P. S. Awana, [J.](http://dx.doi.org/10.1063/1.3662151) Appl. Phys. **110**[, 103913 \(2011\).](http://dx.doi.org/10.1063/1.3662151)
- 24J. Sugiyama, M. Mnsson, O. Ofer, K. Kamazawa, M. Harada, D. Andreica, A. Amato, J. H. Brewer, E. J. Ansaldo, H. Ohta, C. Michioka, and K. Yoshimura, Phys. Rev. B **84**[, 184421 \(2011\).](http://dx.doi.org/10.1103/PhysRevB.84.184421)
- 25Michael A. McGuire, Athena S. Sefat, Brian C. Sales, and David Mandrus, Phys. Rev. B **82**[, 092404 \(2010\).](http://dx.doi.org/10.1103/PhysRevB.82.092404)
- 26J. P. Perdew, K. Burke, and M. Enzerhoff, [Phys. Rev. Lett.](http://dx.doi.org/10.1103/PhysRevLett.77.3865) **77**, 3865 [\(1996\).](http://dx.doi.org/10.1103/PhysRevLett.77.3865)
- 27S. Riggs, C. Tarantini, J. Jaroszynski, A. Gurevich, A. Palenzona and M. Putti, T. Duc Nguyen, and M. Affronte, [Phys. Rev. B](http://dx.doi.org/10.1103/PhysRevB.80.214404) **80**, [214404 \(2009\).](http://dx.doi.org/10.1103/PhysRevB.80.214404)
- 28J. Dai, J.-X. Zhu, and Q. Si, Phys. Rev. B **80**[, 020505 \(2009\).](http://dx.doi.org/10.1103/PhysRevB.80.020505)
- 29W. Uhoya, G. MTsoi, Y. K. Vohra, M. A. McGuire, A. S. Sefat, B. C. Sales, D. Mandrus, and S. T. Weir, [J. Phys. Condens. Matter](http://dx.doi.org/10.1088/0953-8984/22/18/185702) **22**[, 185702 \(2010\).](http://dx.doi.org/10.1088/0953-8984/22/18/185702)