# Phase diagram of $Ba_{1-x}K_xFe_2As_2$

S. Avci, <sup>1,\*</sup> O. Chmaissem, <sup>1,2</sup> D. Y. Chung, <sup>1</sup> S. Rosenkranz, <sup>1</sup> E. A. Goremychkin, <sup>1,3</sup> J. P. Castellan, <sup>1</sup> I. S. Todorov, <sup>1</sup> J. A. Schlueter, <sup>1</sup> H. Claus, <sup>1</sup> A. Daoud-Aladine, <sup>3</sup> D. D. Khalyavin, <sup>3</sup> M. G. Kanatzidis, <sup>1,4</sup> and R. Osborn <sup>1</sup> 

\*\*Materials Science Division, Argonne National Laboratory, Argonne, Illinois 60439-4845, USA

\*\*Physics Department, Northern Illinois University, DeKalb, Illinois 60115, USA

\*\*ISIS Pulsed Neutron and Muon Source, Rutherford Appleton Laboratory, Chilton, Didcot OX11 0QX, United Kingdom

\*\*Department of Chemistry, Northwestern University, Evanston, Illinois 60208-3113, USA

(Received 16 March 2012; revised manuscript received 20 April 2012; published 7 May 2012)

We report the results of a systematic investigation of the phase diagram of the iron-based superconductor  $Ba_{1-x}K_xFe_2As_2$  from x=0 to x=1.0 using high-resolution neutron and x-ray diffraction and magnetization measurements. The polycrystalline samples were prepared with an estimated compositional variation of  $\Delta x \lesssim 0.01$ , allowing a more precise estimate of the phase boundaries than reported so far. At room temperature,  $Ba_{1-x}K_xFe_2As_2$  crystallizes in a tetragonal structure with the space group symmetry of I4/mmm, but at low doping, the samples undergo a coincident first-order structural and magnetic phase transition to an orthorhombic (O) structure with the space group Fmmm and a striped antiferromagnet (AF) with the space group  $F_cmm'm'$ . The transition temperature falls from a maximum of 139 K in the undoped compound to 0 K at x=0.252, with a critical exponent as a function of doping of 0.25(2) and 0.12(1) for the structural and magnetic order parameters, respectively. The onset of superconductivity occurs at a critical concentration of x=0.130(3), and the superconducting transition temperature grows linearly with x until it crosses the AF/O phase boundary. Below this concentration, there is microscopic phase coexistence of the AF/O and superconducting order parameters, although a slight suppression of the AF/O order is evidence that the phases are competing. At higher doping, superconductivity has a maximum  $T_c$  of 38 K at x=0.4 that falls to 3 K at x=1.0. We discuss reasons for the suppression of the spin density wave order and the electron-hole asymmetry in the phase diagram.

DOI: 10.1103/PhysRevB.85.184507 PACS number(s): 74.70.Xa, 61.50.Ks, 61.05.fm, 74.20.Mn

#### I. INTRODUCTION

There is now an extensive body of research into the origin of superconductivity in the iron-based superconductors that demonstrates the importance of the subtle interplay of their electronic properties with crystalline structure.<sup>1,2</sup> These compounds contain a common structural motif, namely a square planar net of iron atoms tetrahedrally coordinated with pnictogens or chalcogens producing  $Fe_2X_2$  layers, in which X = As or Se/Te in the highest  $T_c$  compounds. They are separated by buffer layers comprising, for example, rare earth oxides in the so-called 1111 systems, such as those based on LaFeAsO, or alkaline earths in the so-called 122 systems, such as those based on BaFe<sub>2</sub>As<sub>2</sub>. Like the cuprate superconductors, the buffer layers can act as charge reservoirs, controlling the carrier concentration and inducing superconductivity in the iron planes by the introduction of aliovalent dopants (e.g., LaFeAsO<sub>1-x</sub>F<sub>x</sub><sup>3,4</sup> and  $Ba_{1-x}K_xFe_2As_2^{5,6}$ ), but it is also possible to dope the pnictogen or chalcogen sites (e.g.,  $BaFe_2As_{2-x}P_x^{7,8}$ ) or the iron planes by substituting other transition metal ions (e.g.,  $BaFe_{2-x}Co_xAs_2$  and  $BaFe_{2-x}Ni_xAs_2^{9,10}$ ).

In this article, we report on a systematic investigation of the phase diagram of  $Ba_{1-x}K_xFe_2As_2$  from x=0 to x=1.0 using high resolution neutron and x-ray diffraction combined with bulk characterization. In spite of the diversity of doping strategies employed to modify the superconducting properties of the iron-based superconductors, their phase diagrams show remarkable similarities. There is typically an undoped "parent" compound that is antiferromagnetic rather than superconducting. These are fully compensated metals, whose electronic properties are dominated by multiple

iron-derived d-bands near the Fermi level with approximately equal concentrations of hole and electron carriers. 11 The Fermi surfaces consist of quasi-two-dimensional cylinders, with two or three hole pockets at the Brillouin zone centers and two electron pockets at the M-points on the zone boundaries, i.e., along the direction of the nearest-neighbor iron–iron bonds. 12 All Fermi surface pockets have similar radii in the undoped compounds, making their electronic structure particularly susceptible to magnetic instabilities resulting from a nesting of the disconnected hole and electron Fermi surfaces. 13 Since the nesting wavevector corresponds to the antiferromagnetic wavevector observed by neutron diffraction, 14-16 it is plausible that the magnetism can be explained by a purely itinerant model of spin density waves (SDWs). Indeed, ab initio density functional theory predicts the correct magnetic structure, 17 although there is ongoing debate about the strength of electron correlations. 18,19

The magnetic structure breaks the tetragonal symmetry with an in-plane wavevector of either  $(0,\pi)$  or  $(\pi,0)$  in the unfolded Brillouin zone with one iron atom per unit cell. With finite magnetoelastic coupling, this would induce an orthorhombic structural transition, which is usually observed to occur at the same temperature as the magnetic order in the parent compounds.  $^{20,21}$  However, the addition of both hole and electron charge carriers through chemical substitution suppresses both transitions. A variety of scenarios are possible in Ginzburg-Landau treatments of magnetoelastic coupling.  $^{22,23}$  The two-phase transitions could be first or second order and occur simultaneously or separately. In most iron-based compounds, the two transition temperatures split with doping,  $^{24}$  and there is a report of a split transition in Ba<sub>1-x</sub>  $K_x$  Fe<sub>2</sub>As<sub>2</sub>.  $^{25}$  However, this is inconsistent with our

previously reported neutron diffraction data, which show unambiguously that they are coincident and first order for all x before they are both suppressed at  $x \leq 0.3$ . Unusually, the two order parameters, magnetic and structural, are proportional to each other, apparently indicating biquadratic coupling that is usually only observed at a tetracritical point,  $^{26,27}$  not over an extended range of compositions. Possible explanations for this observation are discussed in this paper's conclusions.

Superconductivity emerges before the complete suppression of the antiferromagnetic/orthorhombic (AF/O) phase and coexists at low-doping levels. The nature of the competition between AF/O order and superconductivity is a central question in understanding iron-based superconductivity. Earlier reports based on local probes found that in  $Ba_{1-x}K_xFe_2As_2$ , the coexistence region is characterized by a mesoscopic phase separation into AF/O and superconducting droplets. However, our previously reported diffraction data are only consistent with a microscopic phase coexistence, a conclusion since supported by muon spin rotation ( $\mu$ SR) experiments, suggesting that the earlier reports may be due to compositional fluctuations within the samples.

One of the main reasons for studying  $Ba_{1-x}K_xFe_2As_2$  is that superconductivity extends to much higher hole-doping levels, with 0.5 holes per Fe atom, than in the electron-doped superconductors produced by transition metal substitutions. In the case of  $BaFe_{2-x}Co_xAs_2$ , superconductivity vanishes at only 0.12 electrons per Fe atom. <sup>10</sup> Furthermore, the maximum T<sub>c</sub> with hole doping is 38 K, significantly higher than the maximum  $T_c$  of  $\sim 25$  K obtained with electron doping. This electron-hole asymmetry in the phase diagram has been attributed to enhanced Fermi surface nesting in the hole-doped compounds, consistent with angle-resolved photoemission spectroscopy (ARPES) data and band structure calculations.<sup>33</sup> This explanation is also supported by the evolution of resonant spin excitations, which become incommensurate due to the mismatch in hole and electron Fermi surface volumes when  $T_{\rm c}$  starts to fall.<sup>34</sup> On the other hand, there is also a strong correlation between  $T_c$  and internal structural parameters such as the Fe-As-Fe bond angles. <sup>35,36</sup> These are known to influence the band structure and the degree of moment localization, but their role in optimizing superconductivity and the implications for the gap symmetry is a matter of debate.<sup>37,38</sup>

There have been two previous reports of the doping dependence of this series in addition to our own brief report, which are all in qualitative agreement. <sup>6,36,39</sup> At room temperature, all members of the  $Ba_{1-x}K_xFe_2As_2$  series crystallize in a tetragonal structure with the space group symmetry of I4/mmm (Fig. 1). Low-doped samples also exhibit a low-temperature phase transition to an orthorhombic structure with space group Fmmm.<sup>5</sup> Superconducting samples at higher doping have a maximum T<sub>c</sub> of 38 K and remain tetragonal at all measured temperatures down to 1.7 K. However, there are significant discrepancies in the published reports concerning the critical dopant concentrations defining the onset of superconductivity and the suppression of the AF/O phase, with the latter varying from  $x = \sim 0.3^{36}$  to  $x = \sim 0.4^{39}$  As already mentioned, there have also been disagreements about the nature of the competition among the three ordered phases at low doping. We believe that these discrepancies are due to uncertainties in the actual composition of the synthesized samples, since it is well

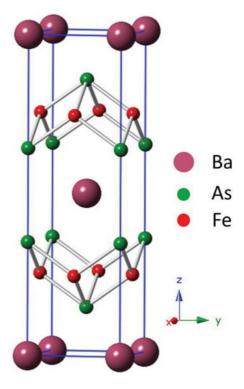


FIG. 1. (Color online) Structure of  $BaFe_2As_2$ , which crystallizes in a tetragonal  $ThCr_2Si_2$ -type structure with the space group symmetry of I4/mmm. Potassium substitutes onto the barium sites.

known that potassium is particularly volatile. Controlling the inhomogeneity to within acceptable limits in order to improve the accuracy of the various phase boundaries has been a key goal of this work, and we estimate that we have been able to make samples in which  $\Delta x < 0.01$ . We performed neutron and x-ray diffraction studies of the magnetic and structural order using high-resolution powder diffractometers (HRPDs) so that the systematic variation of the lattice parameters and internal structural parameters can be used to estimate the degree of uncertainty in the average composition and its variation within the samples.

In this article, we present the results of Rietveld refinements for the entire series and use this analysis, along with bulk measurements, to produce a comprehensive magnetic and structural phase diagram that provides insight into the nature of the phase competition that underlies iron-based superconductivity. Our results show that there is a steeper decrease in  $T_{\rm c}$  and hence a narrower region of phase coexistence of the AF/O order with superconductivity than previously reported. After a description of our experimental results, we combine our findings with results reported in the literature on the electron-doped Ba(Fe<sub>1-x</sub>Co<sub>x</sub>)<sub>2</sub>As<sub>2</sub> series in order to elucidate the origin of the electron-hole asymmetry in the phase diagram.

## II. EXPERIMENTAL DETAILS

The synthesis of homogeneous, single-phase  $Ba_{1-x}K_xFe_2As_2$  samples is known to be particularly delicate due to unfavorable kinetics, high vapor pressures, and a significant difference in the chemical reactivity of K and Ba

with FeAs that may stabilize other binary by-products. For this work, the synthesis and properties of our samples were optimized by the systematic examination of all reasonable combinations of reaction parameters, e.g., purity of the starting materials, reaction containers, temperature, and duration of heating. Our final samples were produced according to the following procedure: Handling of all materials was performed in a nitrogen-filled glove box. Raw materials (BaAs, KAs, and Fe<sub>2</sub>As) were prepared by heating elemental mixtures at 400°C, 600°C, and 850°C, respectively. The stoichiometric mixture of these starting precursors for a desired composition was thoroughly ground to ensure uniformity and homogeneity and subsequently annealed at 1050°C in a Nb tube sealed in a quartz tube. The closed metal tubes are needed to eliminate any chemical loss that may otherwise result from the evaporation of K and As. A large number of high-quality samples were synthesized, covering the full phase diagram  $0 \le x \le 1$ . Special emphasis was given to the  $0.1 \le x \le 0.25$  range, in which the potassium content was incremented in very small amounts with  $\Delta x = 0.025$ . The deliberate synthesis of samples with finely tuned K content was necessary to carefully investigate the rapid suppression of magnetism with increasing K and to elucidate the nature of phase coexistence with superconductivity within the same sample. Samples with coarse K increments would otherwise lead to inconclusive results.

Initial characterization of the samples was performed by x-ray diffraction, magnetization measurements, and inductively coupled plasma (ICP) elemental analysis. For select samples, neutron and x-ray powder diffraction experiments were performed on the HRPD (x = 0, 0.1, 0.125, 0.15, 0.175, 0.2, 0.21, 0.24, 0.3, 0.5, and 1) and WISH diffractometers (x = 0.22 and 0.25) at the ISIS Pulsed Neutron and Muon Source (Rutherford Appleton Laboratory) and on beamline 11-BM (x = 0.28) at the Advanced Photon Source (Argonne

National Laboratory). The resolution  $\Delta d/d$  at 2 Å is 0.001 for HRPD and 0.002 for Wish. For diffraction experiments, the samples were sealed under vacuum for shipment and reopened just before the measurements in a helium environment to prevent air exposure. The nuclear and magnetic structures, together with interatomic bond lengths and bond angles, were determined by the Rietveld refinement technique using the comprehensive General Structure Analysis System software suite<sup>40</sup> and the associated graphical user interface EXPGUI.<sup>41</sup> Traces of no more than 0.5–3% by weight of FeAs and Fe<sub>2</sub>As impurity phases were observed in some of the samples, but they were too small to affect the analysis.

Our neutron diffraction results show that the synthesis methods reduced the compositional uncertainty significantly, which we estimate to be  $\Delta x \lesssim 0.01$ . In previously reported phase diagrams, it is not clear whether the potassium contents were nominal or actual measured values. Because of its volatility, it is always necessary to add excess potassium, so the eventual stoichiometry is largely governed by the Ba/Fe ratio. Thus, any perceived discrepancy with other work, such as Ref. 39, is probably due to differences in the handling and control of the volatile potassium and arsenic constituents. In this article, we used a number of methods to characterize the sample compositions, including direct measurements of the stoichiometry from ICP analysis. To produce the final compositions used in our phase diagrams, we started with the nominal x values determined from the starting Ba/Fe ratio and then smoothed the variation in the a-axis lattice parameter from  $0.1 \le x \le 0.25$  using a power law function (Table I). In most cases, the agreement with the nominal value was better than 0.005, with just two samples requiring a significant shift of 0.02 and 0.03, respectively. In all cases, the adjustments also improved the consistency of other measurements, such as the variations in transition temperatures and order parameters.

TABLE I. Structural and magnetic phase diagram of  $Ba_{1-x}K_xFe_2As_2$ . The nominal value of x represents the starting composition given by the Ba/Fe ratio. The fitted value is determined by smoothing the variation in the a-axis lattice parameter from  $0.1 \le x \le 0.25$  using a power law function. The first column is the value of x used in the text and figures.  $T_N$  and  $T_S$  are determined as described in the text. The magnetic moments and orthorhombic order parameters,  $\delta = (a - b)/(a + b)$ , are determined from the low-temperature Rietveld refinements.

x	x (fitted)	x (nominal)	x (ICP)	$T_{\rm c}$ (K)	T <sub>N</sub> magn. (K)	T <sub>N</sub> neutron (K)	$T_{\rm s}\left({\rm K}\right)$	Magnetic moment ( $\mu_B$ )	$\delta \times 10^3$
0		0	0		139(1)	139.0(1)	138.17(6)	0.756(36)	3.92(4)
0.1	0.097	0.1	0.094(2)		136(1)	136.5(3)	136.02(8)	0.741(21)	3.68(3)
0.125	0.126	0.125	0.114(2)		130(2)		128.29(6)	0.697(29)	3.49(4)
0.15	0.15	0.15	0.139(1)	4	122(2)	122.1(3)	122.09(7)	0.702(21)	3.35(4)
0.175	0.172	0.175	0.159(2)	10	113(2)	113.9(1)	112.1(6)	0.683(24)	3.14(5)
0.2	0.202	0.2	0.184(2)	17	100(2)	102.0(1)	102.00(2)	0.652(46)	2.76(7)
0.21	0.209	0.24		18		96.0(1)	96.0(3)	0.610(32)	2.59(3)
0.22	0.225	0.22		23.5		93.97(1)	93.93(1)	0.550(22)	2.20(9)
0.24	0.237	0.26		26		79.9(1)	80.0(2)	0.572(29)	2.00(3)
0.25	0.249	0.24		28.5		74.9(1)	74.8(8)	0.456(22)	1.43(8)
0.28		0.28		34					
0.3		0.3	0.312(4)	36					
0.4		0.4		38					
0.5		0.5	0.476(8)	34					
0.7		0.7	0.675(3)	20					
0.9		0.9	0.892(1)	7					
1		1	1.00(2)	3					

The first column of Table I shows the value of x that we used in the text and figure labels, derived by rounding the fitted values to the nearest 0.005, although the fitted values were used in the plots and in numerical analysis of the doping dependence — e.g., the fit of  $T_c$  vs x. For  $x \ge 0.28$ , where the precise composition is not so critical, we used the nominal values.

We were also able to monitor fluctuations in composition within a single sample, because HRPD has sufficiently high resolution that it is sensitive to distributions of the lattice parameter and internal strains caused by compositional gradients and even particle size broadening. The diffraction peaks from the (220) reflection in Fig. 1 of Ref. 6 show that there is no change in the linewidths and lineshapes of the diffraction peaks from the undoped compound up to x = 0.24, consistent with a high degree of compositional homogeneity ( $\Delta x \lesssim 0.01$ ). The regular spacing between the peak positions is in agreement with the fixed steps in x.

### III. RESULTS

### A. Structural phase diagram

In the undoped  $BaFe_2As_2$  compound, there is a structural phase transition at 139 K from the tetragonal  $ThCr_2Si_2$ -type structure with the space group symmetry I4/mmm to an orthorhombic  $\beta$ – $SrRh_2As_2$ -type structure with the space group Fmmm.<sup>5</sup> The structure of both the tetragonal and the orthorhombic phases of  $Ba_{1-x}K_xFe_2As_2$  can be described as a stack of edge-sharing  $Fe_2As_2$  layers separated by layers of (Ba,K) ions (Fig. 1). The (Ba,K) ions occupy crystallographic positions that are tetrahedrally coordinated with four arsenic anions.

With potassium doping, the tetragonal to orthorhombic structural transition temperature  $T_s$  decreases until it is fully suppressed for x>0.25. In Fig. 2, we show the lattice parameters as a function of temperature for x=0,0.1,0.21, and 0.3. Below x=0.3, a significant orthorhombic splitting of the basal plane a and b lattice parameters is observed. The evolution of the orthorhombic order parameter defined by  $\delta=(a-b)/(a+b)$  is discussed later. Although the transitions appear to be continuous, we observed small but sharp volume anomalies at all structural phase transitions (Fig. 3 in Ref. 6), showing that they are weakly first order in character over the entire phase diagram.

The structural transition temperatures in Table I were determined by fitting a power law,  $\delta \propto (T_s - T)^{\beta}/T_s$  close

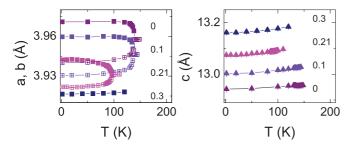


FIG. 2. (Color online) Variation of lattice constants a-c with temperature for x=0, 0.1, 0.21, and 0.3. The lattice constants, a and b, in the orthorhombic phase are divided by  $\sqrt{2}$ .

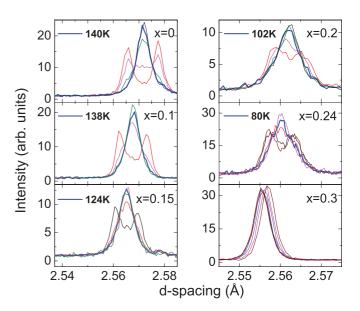


FIG. 3. (Color online) Temperature dependence of the (110) peak in the vicinity of structural transition temperature  $T_s$ . The bold curve shows the peak at the estimated  $T_s$ . For  $0 \le x \le 0.24$ , the other curves show the peak at intervals of 2 K around  $T_s$ . For x = 0.3, the peak is shown in 20 K intervals between 1.7 and 120 K.

to the transition temperature, yielding exponents mostly in the range  $\beta \approx 0.13$  to 0.2. Apart from x = 0, where  $\beta = 0.129(3)$ , in reasonable agreement with Ref. 27, the exponents are modified by compositional fluctuations and so are not reliable estimates of the critical behavior. The exponents at x = 0.21 and 0.24 were anomalously high ( $\beta = 0.25$  and 0.30, respectively), which could reflect a slightly greater degree of compositional variation within those samples. As a check on these values of  $T_s$ , we used the peak profiles close to the transition temperature Figure 3 shows the temperature dependence of the tetragonal (110) Bragg peak for x = 0, 0.1, 0.15, 0.2, 0.24, and 0.3 in the vicinity of the structural transition. This reflection splits into two (022) and (202) orthorhombic peaks below  $T_s$ . Close to the transition, the two peaks cannot be resolved, but we can determine  $T_s$  from the temperature dependence of the full width at half maximum (FWHM) as they merge. In the high-temperature phase, FWHM  $\sim 0.0037(3)$  Å is independent of the composition. These two methods of determining  $T_{\rm s}$ agreed within the errors.

Figure 4 shows the evolution of the lattice parameters and unit cell volume at 1.7 K as a function of x. Barium substitution by the smaller potassium cations reduces the in-plane a and b lattice parameters while significantly lengthening the out-of-plane c-axis. However, the c-axis enhancement is not large enough to fully compensate for the shrinking basal plane axes, and the unit cell volume gradually decreases in magnitude upon increasing the K content until  $x \approx 0.5$ . The reason for the nonmonotonic behavior of the unit cell volume at high dopant levels is not understood and requires further investigation.

The behavior of the Fe-Fe and Fe-As interatomic distances (at 1.7 K) are presented in Fig. 5. The Fe-Fe distances mimic the in-plane lattice parameters both as a function of K content and of temperature. The similarity in behavior occurs because Fe atoms occupy special rigid positions along the long edges

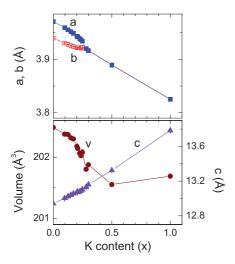


FIG. 4. (Color online) Variation of lattice constants and volume in  $Ba_{1-x}K_xFe_2As_2$  with x at 1.7 K. Solid lines are guides for the eye. The lattice constants a and b and the volume in the orthorhombic phase are divided by  $\sqrt{2}$  and 2, respectively, from those for the *Fmmm* space group.

of the lattice. Six As-Fe-As bond angles can be identified in each FeAs<sub>4</sub> tetrahedron. In the tetragonal I4/mmm structure, these angles can be grouped into two independent angles: two equivalent angles,  $\alpha_1$ , and four equivalent smaller ones,  $\alpha_2$ . In the orthorhombic Fmmm structure below  $T_N$ , the angle  $\alpha_1$  remains unaffected, but the angle  $\alpha_2$  splits into two pairs of equivalent angles  $\alpha_2'$  and  $\alpha_2'$  with an angular separation of  $\sim 0.5-0.6^\circ$ . A sketch showing the different angles is displayed in Fig. 6. As shown in the figure, the angle  $\alpha_1$  increases

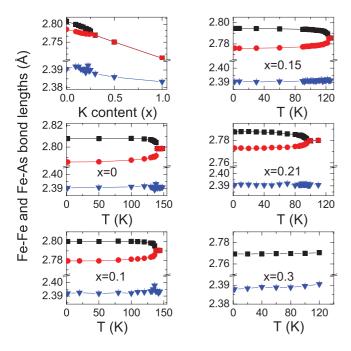


FIG. 5. (Color online) Variation of Fe-Fe and Fe-As bond lengths with x at 1.7 K and with temperature for different K substitutions. Blue triangles represent the Fe-As bonds. Black squares and red circles represent the Fe-Fe bond lengths merging at  $T_{\rm s}$ . Solid lines are guides for the eye.

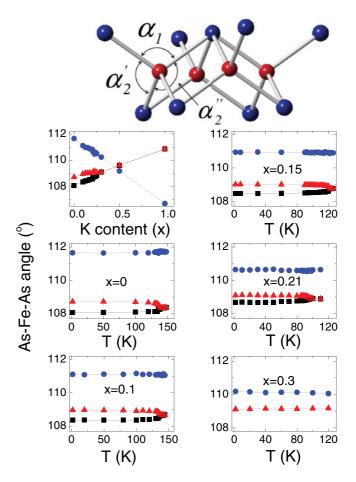


FIG. 6. (Color online) Variation of As-Fe-As bond angles with x at 1.7 K and with temperature for different K substitutions. The top panel shows  $\alpha_1$ ,  $\alpha_2'$ , and  $\alpha_2''$  in an orthorhombic setting. Blue circles represent  $\alpha_1$ ; red triangles and black squares represent  $\alpha_2'$  and  $\alpha_2''$  merging into one  $\alpha_2$  at  $T_s$ . Solid lines are guides for the eye.

linearly and continuously throughout the whole phase diagram. However, starting from BaFe<sub>2</sub>As<sub>2</sub>, the angles  $\alpha_2'$  and  $\alpha_2'$  increase with increasing K until reaching a critical composition below  $x\approx 0.3$ , beyond which the structural transitions are suppressed and the two angles merge into  $\alpha_2$ , which continues to increase with higher K contents. The refined values of the lattice parameters, bond lengths, and bond angles at 1.7 K are shown in Table II.

### B. Magnetic phase diagram

Neutron powder diffraction data reveal the presence of weak magnetic Bragg reflections that appear below the structural phase transition for all of the orthorhombic samples. The magnetic peaks shown in Fig. 7, located at 2.45 and 3.43 Å, were indexed as 121 and 103, in agreement with the widely reported antiferromagnetic SDW ground state.  $^{15,16,39,43}$  As with the structural transitions, the antiferromagnetic transition (Néel) temperatures as a function of doping were determined by power law fits to temperature variation of the magnetic moment (see Table I). These coincide with the orthorhombic transition for all values of x. The structural and magnetic orders are identified in the same measurements: the first from

TABLE II. Results of Rietveld refinements for  $Ba_{1-x}K_xFe_2As_2$  from neutron powder diffraction data collected on HRPD at 1.7 K. For x < 0.3, the space group is

Fmmm $a=b;$	Finnin, in which $a \neq b$ ; there are two inequia $a = b$ ; there is one Fe-Fe bond distance and	$\neq b$ ; there are $= -e$ -Fe bond di	two inequivaler istance and two	Finally, in which $a \neq b$ ; there are two inequivalent Fe-Fe band distances and three inequivalent As-Fe-As bond angles. For $x \ge 0.3$ , the space group is $I4/mmm$ , in which $a = b$ ; there is one Fe-Fe band distance and two inequivalent As-Fe-As bond angles.	ances and thre e-As bond ang	te inequivalent gles.	As-Fe-As bor	nd angles. For	$x \geqslant 0.3$ , the spa	ice group is I4/n	4/mmm, in which
x	a (Å)	b (Å)	c (Å)	Volume (ų)	z As	Fe-As (Å)	Fe-Fe (Å)	Fe-Fe (Å)	As-Fe-As (°)	As-Fe-As (°)	As-Fe-As (°)
0	5.6157(2)	5.5718(2)	12.9424(4)	404.970(45)	0.35375(3)	2.3905(1)	2.8078(1)	2.7859(1)	108.071(3)	108.718(3)	111.648(2)
0.1	5.5997(1)	5.5587(1)	13.0031(4)	404.755(36)	0.35405(3)	2.3919(1)	2.7998(1)	2.7793(1)	108.356(7)	108.961(7)	111.110(15)
0.125	5.5940(2)	5.5551(2)	13.0243(5)	404.745(52)	0.35405(3)	2.3918(2)	2.7970(1)	2.7775(1)	108.438(8)	109.011(8)	110.977(16)
0.15	5.5890(2)	5.5517(2)	13.0404(5)	404.634(46)	0.35399(3)	2.3905(1)	2.7945(1)	2.7758(1)	108.464(3)	109.013(3)	110.947(3)
0.175	5.5842(2)	5.5492(2)	13.0563(6)	404.598(55)	0.35386(3)	2.3901(2)	2.7921(1)	2.7746(1)	108.522(9)	109.037(9)	110.864(17)
0.2	5.5767(4)	5.5460(4)	13.0736(9)	404.353(82)	0.35372(5)	2.3885(1)	2.7883(2)	2.7730(2)	108.578(5)	109.030(5)	110.815(4)
0.21	5.5750(1)	5.5462(2)	13.0749(4)	404.288(36)	0.35404(2)	2.39073(4)	2.7875(1)	2.7731(1)	108.678(2)	109.102(2)	110.640(2)
0.22	5.5706(5)	5.5461(4)	13.0803(12)	404.126(112)	0.3543(7)	2.3925(6)	2.7855(3)	2.7733(3)	108.798(21)	109.155(21)	110.47(4)
0.24	5.5672(2)	5.5449(2)	13.0888(4)	404.051(39)	0.35394(3)	2.3894(2)	2.7836(1)	2.7724(1)	108.750(7)	109.078(7)	110.592(14)
0.25	5.5636(4)	5.5476(4)	13.1027(10)	404.157(90)	0.35398(6)	2.3900(5)	2.7810(2)	2.7732(2)	108.846(17)	109.073(16)	110.50(3)
0.3	3.9165(2)	3.9165(2)	13.1614(5)	201.877(23)	0.35383(4)	2.3878(3)	2.7694(2)	2.7694(2)	109.090(28)		110.237(14)
0.5	3.8893(2)		13.3242(6)	201.554(20)	0.35376(4)	2.3859(3)	2.7501(1)		109.615(11)		109.185(21)
_	3.8251(2)		13.7846(5)	201.691(40)	0.35314(4)	2,3833(4)	2.7047(1)		110.855(12)		106.708(23)

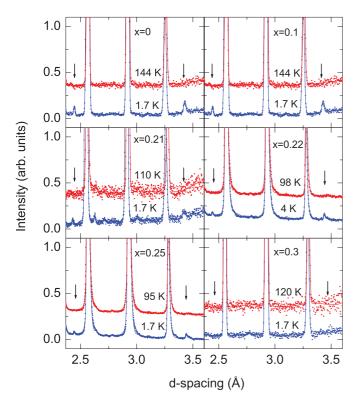


FIG. 7. (Color online) Neutron diffraction at 1.7 K with the magnetic Bragg peaks at a d-spacing of 2.45 and 3.43 Å indicated by the arrows. They are absent above  $T_{\rm N}$  for x=0,0.1,0.21,0.22, and 0.25. At x=0.3, no magnetic peaks are observed.

the splitting of the nuclear Bragg peaks and the second by the intensity of the magnetic Bragg peaks. This means that the conclusion that the two transitions are coincident does not depend on the accuracy of the thermometry. As shown in Ref. 6, the two order parameters determined in this way are directly proportional to each other at all temperatures over the entire phase diagram, an unusual result that we explain in more detail in the discussion.

As a further check on our data, the values of  $T_{\rm N}$  determined by neutron diffraction are in excellent agreement with those determined by peaks in the temperature derivative of the magnetization (Fig. 8 and Table I). These magnetization peaks decrease in magnitude because of the progressive attenuation of the magnetic signal due to increasing K content until they are no longer detected for  $x \geqslant 0.21$ .

A full analysis of the magnetic structure was performed using the allowed subgroup magnetic symmetries of *Fmmm*. All possible models were tested, but only the magnetic space group  $F_cmm'm'$  resulted in a proper fit to the data. Removal of the time reversal symmetry from two of the mirror planes resulted in an antiferromagnetic arrangement of the magnetic moments with a magnetic wavevector  $\mathbf{Q}=(1,0,1)$ ; that is, the Fe magnetic moments are antiferromagnetically coupled in the x and z directions and ferromagnetically coupled along the y axis. This model is consistent with results previously reported for the parent BaFe<sub>2</sub>As<sub>2</sub> material.  $^{15,16,44}$ 

Rietveld refinements of both the atomic and the magnetic structures were performed simultaneously as a function of temperature and doping, allowing the magnetic moment to be

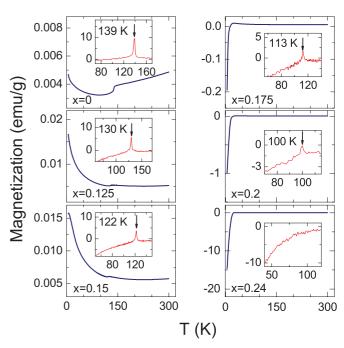


FIG. 8. (Color online) SQUID magnetization measurements for  $x=0,\,0.125,\,0.15,\,0.175,\,0.2,\,$  and 0.24 in a 2-kG applied magnetic field. Insets are the first derivatives of the magnetization curves  ${\rm d}M/{\rm d}T$  ( $10^{-5}$ ) used to determine the Néel temperatures given by the arrows. For x>0.2, the magnetization anomaly at  $T_{\rm N}$  is too weak to be detected.

defined in absolute units by normalization of the intensity of the magnetic Bragg peaks to the structural Bragg peaks. The neutron data displayed in Fig. 9 were collected at 1.7 K and normalized to the sample mass and exposure time (measured in beam pulses). The figure qualitatively shows that the intensities of the magnetic (121) and (103) Bragg reflections remain roughly unchanged for the x=0 and 0.1 samples, followed by a monotonic decrease upon increasing the K content until they nearly vanish at x=0.24. The refinements show that the magnetic moment drops from  $\mu=0.75\mu_{\rm B}$  for the parent BaFe<sub>2</sub>As<sub>2</sub> material to  $0.46\mu_{\rm B}$  for  $x\approx0.25$  (see Table II). No magnetic peaks are observed beyond this value.

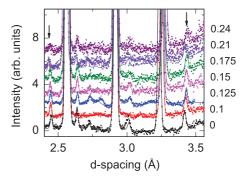


FIG. 9. (Color online) Dependence of the neutron diffraction intensity for  $Ba_{1-x}K_xFe_2As_2$  at 1.7 K with x. The magnetic Bragg peaks are shown by the arrows. The solid lines represent the calculated intensity of the Rietveld refinement.

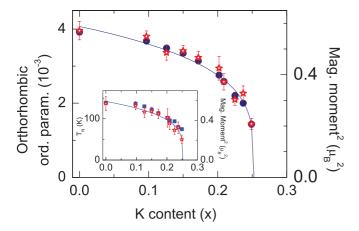


FIG. 10. (Color online) Dependence of the square of the magnetic moment  $\mu^2$  (red stars) and the orthorhombic order parameter  $\delta = (a-b)/(a+b)$  (blue circles) at 1.7 K on the potassium concentration x. The inset shows a comparison of  $T_{\rm N}$  and  $\mu^2$  vs x, showing that  $T_{\rm N} \propto \mu^2$ . Solid lines in the main panel and the inset are a fit to  $(1-x/x_{\rm c})^{2\beta}$ , with  $\beta=0.125$ .

The doping dependence of the two order parameters making up the AF/O phase is shown in Fig. 10. We compare  $\delta$  and  $\mu^2$  vs x, showing that they are directly proportional over the entire range of the AF/O order. We have not been able to measure any samples between  $0.25 \leqslant x \leqslant 0.28$ , but a power law fit to  $\delta$  and  $\mu^2$  close to  $x_c$ , i.e., to  $(1-x/x_c)^{2\beta}$ , gives a critical concentration of 0.252 with an exponent of  $\beta=0.125(1)$  for the structural and magnetic order parameters. The inset to Fig. 10 shows that  $T_N$  is also proportional to  $\mu^2$ . In a mean-field model,  $T_N$  scales as  $J\mu^2$ , where J is the effective interionic exchange interaction, so this result seems to indicate that J is approximately independent of x over this range.

#### C. Superconductivity and phase coexistence

Bulk superconductivity is observed for all samples with  $x \ge$ 0.15. Zero-field magnetization data show that superconducting transition temperatures peak at  $\sim$ 38 K for x = 0.4 before slowly decreasing to 3 K for the end member KFe<sub>2</sub>As<sub>2</sub> (Fig. 11). By comparing the samples' magnetic moment with that of a Sn-powder sample of similar volume, we estimate that these samples are bulk superconductors with a volume fraction of at least 80%. The uncertainty is due to variations in the demagnetization factor between samples. All superconducting transitions are well defined and sharp, even close to the critical concentration, where  $T_c$  is varying rapidly with x. Even low levels of compositional inhomogeneity associated with the uneven distribution of Ba/K ions would be revealed in magnetic susceptibility measurements by a broad or steppedlike transition from the normal state to the superconducting state, so this is further evidence of the sample quality. The increase in  $T_c$  at low doping is approximately linear with dopant concentration up to  $x \approx 0.25$ , so we estimated the critical concentration for the onset of superconductivity using linear regression to be  $x_c = 0.130(3)$ .

In Ref. 6, we discussed the behavior of the order parameters below  $T_c$ . We observed a small reduction in both the magnetic

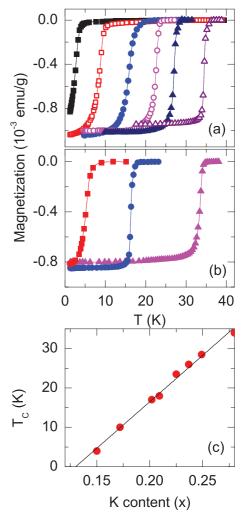


FIG. 11. (Color online) SQUID magnetization measurements (zero field cooled) in the 0.1-G magnetic field for (a) x=0.15 (solid squares), 0.175 (open squares), 0.21 (solid circles), 0.22 (open circles), 0.25 (solid triangles), and 0.3 (open triangles) and (b) x=0.5 (solid triangles), 0.7 (solid circles), and 0.9 (solid squares), showing well-defined superconducting transitions. Magnetization values are normalized to the mass of the samples. (c) Superconducting transition temperatures (onset  $T_{\rm c}$ ) of the underdoped compounds. Solid line represents the linear regression showing that the critical concentration for superconductivity is 0.130(3).

and the structural order parameters of  $\sim$ 5% at x=0.21 and 0.24, without seeing evidence of additional phases. In a scenario in which the sample divides into separate mesoscopic regions of the AF/O phase and the superconducting phase, this implies that 95% of the sample remains in the AF/O phase and only 5% becomes superconducting. This is inconsistent with the magnetization measurements showing bulk superconductivity. While it is not possible to rule out the presence of other phases, the results indicate that there is microscopic phase coexistence of magnetism and superconductivity, with the reduction in the AF/O order parameters being due to competition with the superconducting order parameter. This competition has been discussed extensively by Fernandes and Schmalian, 45 who show that there should be an additional

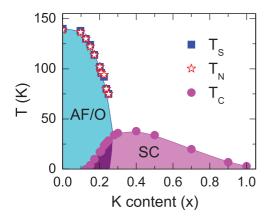


FIG. 12. (Color online) Phase diagram of  $Ba_{1-x}K_xFe_2As_2$  with the superconducting critical temperatures  $T_c$  (circles), the Néel temperatures  $T_N$  (stars), and the structural transition temperatures  $T_S$  (squares).

phase boundary, with a positive slope vs x, between the coexistence region and the region of the purely superconducting phase. We have not yet identified any anomalies corresponding to the phase line below  $T_{\rm c}$ , so we assume that it rises steeply with x.

#### IV. DISCUSSION

The overall phase diagram of  $Ba_{1-x}K_xFe_2As_2$  is shown in Fig. 12. We first discuss the nature of the SDW order and orthorhombic order. Unlike the electron-doped compounds, where the two transitions split within increased doping, the two transitions are coincident and first order in  $Ba_{1-x}K_xFe_2As_2$ . We reported the first-order character of the transition by the observation of volume anomalies at  $T_s$ . Similar volume anomalies were observed by Tegel et al.46 in unsubstituted SrFe<sub>2</sub>As<sub>2</sub> and EuFe<sub>2</sub>As<sub>2</sub>, but because of the small magnitude of these anomalies, the authors suggested that the structural phase transition may be second order. However, other authors reported first-order transitions in polycrystalline SrFe<sub>2</sub>As<sub>2</sub><sup>20</sup> and single crystals of CaFe<sub>2</sub>As<sub>2</sub><sup>21</sup> and BaFe<sub>2</sub>As<sub>2</sub>.<sup>44</sup> In the latter reference, a first-order–like hysteresis was obtained for the intensities of the (101) Bragg peak when measured on cooling and warming. However, no such hysteresis was observed by Wilson et al.<sup>27</sup> when examining their BaFe<sub>2</sub>As<sub>2</sub> single crystal. The systematic observation of volume anomalies across the phase diagram is unambiguous evidence that, at least in this system, all transitions are first order, although weakly first order with extremely small hysteresis.

Phenomenological theory of magnetoelastic coupling predicts the possibility of simultaneous first-order transitions driven by a linear–quadratic term in the Ginzburg-Landau expansion<sup>22,23</sup> (see also Ref. 47). This is the lowest-order term allowed by symmetry. If the magnetic transition were to occur at a higher temperature than the structural transition (in the absence of any competition), the magnetic order would drive the structural order in a simultaneous first-order transition. The converse would produce two split transitions, as seen in most iron-based compounds.<sup>24</sup> There is a report

of a split transition in  $Ba_{1-x}K_xFe_2As_2$  based on nuclear magnetic resonance results,<sup>25</sup> but this is a local probe, which cannot necessarily identify compositional fluctuations. As we already discussed, the neutron measurements, which represent true averages over the bulk, are quite unambiguous that the two transitions are simultaneous, although there could be some rounding of the transitions at higher doping from small compositional fluctuations.

The two order parameters are directly proportional to each other as a function of temperature, which seems to indicate an unusual biquadratic coupling in the Ginzburg-Landau expansion rather than a linear-quadratic coupling. This is usually only observed at a tetracritical point<sup>26,27</sup> where two phase boundaries intersect, whereas our observations extend over a range of compositions. There are a number of possible reasons for this. The most intriguing and exotic idea is that the AF/O order parameters are both secondary to another order parameter and directly driven by it. This would be the case in, for example, valley density wave theory, in which a mother density wave drives both the magnetic and the chargedensity wave orders.<sup>48</sup> A second explanation is provided by the theoretical work of Nevidomskyy, 49 which uses a microscopic Kugel-Khomskii model to produce a biquadratic spin-orbital term in the free energy. A subtle, but ultimately more conventional, explanation is that the coupling is linear quadratic after all, but the proximity to a first-order transition produces a temperature dependence that is approximately equivalent to biquadratic coupling to first order.<sup>29</sup> This is in the context of a theory in which the Ising-nematic order, produced by an itinerant model of Fermi surface nesting, drives the structural transition. Support for this explanation is provided by Fig. 10, where the doping dependence of the magnetic and structural order parameters at low temperature indicates linear-quadratic coupling. Whatever the eventual explanation, it is clear that this result is key to understanding the nature of the normal state and the role of nematic order in the eventual superconductivity.

The strong coupling between AF/O order parameters persists into the regime of phase coexistence with superconductivity. We already argued that  $Ba_{1-x}K_xFe_2As_2$  is characterized by microscopic phase coexistence because mesoscopic phase separation would result in a significant decrease in the volume fraction of the AF/O phase below  $T_c$ . The consensus in favor of microscopic phase coexistence has existed for some time for electron-doped superconductors, such as  $Ba(Fe_{1-x}Co_x)_2As_2$ , <sup>24</sup> where the phase boundary below  $T_{\rm c}$  to a nonmagnetic, purely superconducting region has also been identified. Theoretically, this behavior is consistent with unconventional  $s^{\pm}$  pairing of the Cooper pairs, suggesting that itinerant long-range magnetism and superconductivity may coexist and compete for the same electrons.<sup>45</sup> However, the idea of microscopic phase coexistence was more controversial in  $Ba_{1-x}K_xFe_2As_2$  because of local probe measurements that seemed to indicate a phase separation into mesoscopic regions of magnetism and superconductivity. 30,31 Because the most recent  $\mu$ SR data are also consistent with microscopic phase coexistence,<sup>32</sup> it appears that the earlier reports may have been due to compositional fluctuations close to the phase boundaries and that microscopic phase coexistence has now been confirmed.

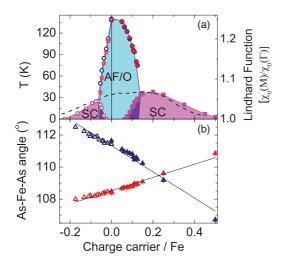


FIG. 13. (Color online) (a) Magnetic and structural phase diagram of electron-doped Ba(Fe<sub>1-x</sub>Co<sub>x</sub>)<sub>2</sub>As<sub>2</sub> and hole-doped  $Ba_{1-x}K_xFe_2As_2$  with  $T_c$  (squares),  $T_N$  (stars), and  $T_s$  (circles). The x-axis is normalized to the charge carrier per iron atom. Data for the electron-doped side where the transition temperatures are represented by open symbols are taken from Ref. 50. The error bars for  $T_{\rm N}$  and  $T_{\rm s}$ values in the hole-doped side are within the symbols. The dashed line enveloping the superconducting dome represents the Lindhard function taken from Ref. 33. (b) Charge carrier dependence of the As-Fe-As bond angles for both electron and hole doping. Solid triangles represent the results of our neutron diffraction study at 1.7 K for the hole-doped  $Ba_{1-x}K_xFe_2As_2$ . At this temperature, one of the As-Fe-As angles splits due to orthorhombic distortion below x =0.3. Therefore, we took the average of these two splitting angles. The As-Fe-As bond angle data for the electron doped side are taken from Ref. 51. Solid lines are visual guides.

Finally, we discuss the electron-hole asymmetry in the phase diagram, shown in Fig. 13, where we have added data from the literature<sup>50,51</sup> to allow a comparison with the more commonly studied electron-doped superconductors. In this phase diagram, the x-axis is normalized to the number of charge carriers per Fe atom. Neupane et al. recently suggested that this asymmetry is due to differences in the effective masses of the hole and electron pockets.<sup>33</sup> This is justified by ARPES data that show hole doping can be well described within a rigid band approximation.<sup>52</sup> An *ab initio* calculation of the Lindhard function of the noninteracting susceptibility at the Fermi surface nesting wavevector shows exactly this asymmetry, with a peak at  $x \approx 0.4$ , where the maximum  $T_c$ occurs. Our recent inelastic neutron scattering measurements of the resonant spin excitations that are also sensitive to Fermi surface nesting showed a similar correlation between the strength of superconductivity and the mismatch in the hole and electron Fermi surface volumes<sup>34</sup> that is responsible for the fall of the Lindhard function at high x. An overall envelope may be drawn (dashed line in Fig. 13) to encompass both the hole and the electron superconducting domes of the phase diagram. If anything, the Lindhard function underestimates the asymmetry, predicting a larger superconducting dome on the electron-doped side. We attribute this behavior to the iron arsenide layers remaining intact in the potassium-substituted series, whereas Co substitution for Fe disturbs the contiguity of the FeAs<sub>4</sub> tetrahedra and interferes with superconductivity in these layers.

Interestingly, the maximum overall  $T_c$  also correlates with the perfect tetrahedral angle of ~109.5°, as demonstrated in the Fig. 13(b). In the plot, average <As-Fe-As> bond angles for our K-substituted series have been extracted from the Rietveld refinements. The As-Fe-As bond angles for  $Ba(Fe_{1-x}Co_x)_2As_2$  are extracted from the literature.<sup>51</sup> The continuity of the bond angles across the electron- and holedoped sides of the phase diagram is remarkable, and the crossing of the two independent angles at  $x \approx 0.4$  to yield a perfect tetrahedron and maximum  $T_c$  is clear. This has been remarked upon before in other systems. 35,53 It is possible that these two apparently distinct explanations for the maximum  $T_c$ are two sides of the same coin. In a theoretical analysis of the 1111 compounds, <sup>38</sup> it has been suggested that the pnictogen height is important in controlling the energies of different orbital contributions to the d-bands and so affects the strength of the interband scattering that produces superconductivity.

We now turn our attention to the SDW region of the phase diagram. While it is clear that SDW order has to be suppressed in order to allow superconductivity to develop, it is not immediately clear what is responsible for the suppression. Both the strength of magnetic interactions and the superconductivity, at least in an itinerant model, depend on the same Lindhard function<sup>54</sup>: the former on the peak in the susceptibility at the magnetic wavevector and the latter on an integral over the Fermi surfaces. It would seem therefore that the magnetic transition temperature should also peak at  $x \approx 0.4$ . One intriguing reason it would peak at x = 0 is because magnetic order is more sensitive to disorder-induced suppression of the peak susceptibility, whereas superconductivity is more robust. There is some support for this idea from the observation that isoelectronic doping produces a similar suppression of magnetic order to the one seen with hole doping.<sup>51</sup> On the other hand, Kimber et al. succeeded in rendering the parent BaFe<sub>2</sub>As<sub>2</sub> material to exhibit zero resistance at 30.5 K by the application of significant external pressures up to 5.5 GPa<sup>55</sup> (i.e., without introducing disorder), but they remarked that superconductivity needs to be confirmed by other bulk measurement techniques. Interestingly, the authors also correlate the induced  $T_c$  with approaching a perfect tetrahedron angle of 109.5°, similar to our observations for  $x \approx 0.4$ .

### V. CONCLUSIONS

We have synthesized high-quality samples covering the full phase diagram of  $Ba_{1-x}K_xFe_2As_2$ . Using high-resolution neutron powder diffraction and superconducting quantum interference device (SQUID) magnetization measurements, we investigated the effects of potassium substitution on superconductivity, structural transformation, and magnetic ordering. Our measurements allowed the construction of a detailed magnetic and structural phase diagram, which displays a narrower phase coexistence region than that given in previous reports. Moreover, neutron diffraction and the SQUID magnetization data confirmed that magnetic and structural transitions are coincident with first-order transitions. Additionally, we determined the effects of temperature and

substitution on the various internal atomic and structural parameters. Our results confirm the importance of obtaining precise structural parameters across the whole phase diagram as a way of providing insight into the nature of the phase competition that underlies iron-based superconductivity.

#### ACKNOWLEDGMENTS

The work was at Argonne National Laboratory was supported by the US Department of Energy, Office of Science, Materials Science and Engineering Division, under Contract No. DE-AC02-06CH11357.

- \*savci@anl.gov
- <sup>1</sup>G. Stewart, Rev. Mod. Phys. **83**, 1589 (2011).
- <sup>2</sup>D. C. Johnston, Adv. Phys. **59**, 803 (2010).
- <sup>3</sup>Y. Kamihara, T. Watanabe, M. Hirano, and H. Hosono, J. Am. Chem. Soc. **130**, 3296 (2008).
- <sup>4</sup>H. Luetkens, H. H. Klauss, M. Kraken, F. J. Litterst, T. Dellmann, R. Klingeler, C. Hess, R. Khasanov, A. Amato, C. Baines, M. Kosmala, O. J. Schumann, M. Braden, J. E. Hamann-Borrero, N. Leps, A. Kondrat, G. Behr, J. Werner, and B. B. Uuml Chner, Nat. Mater. 8, 305 (2009).
- <sup>5</sup>M. Rotter, M. Tegel, and D. Johrendt, Phys. Rev. Lett. **101**, 107006 (2008).
- <sup>6</sup>S. Avci, O. Chmaissem, E. A. Goremychkin, S. Rosenkranz, J.-P. Castellan, D. Y. Chung, I. S. Todorov, J. A. Schlueter, H. Claus, M. G. Kanatzidis, A. Daoud-Aladine, D. Khalyavin, and R. Osborn, Phys. Rev. B **83**, 172503 (2011).
- <sup>7</sup>S. Jiang, H. Xing, G. Xuan, C. Wang, Z. Ren, C. Feng, J. Dai, Z. Xu, and G. Cao, J. Phys. Condens. Matter **21**, 382203 (2009).
- <sup>8</sup>S. Kasahara, T. Shibauchi, K. Hashimoto, K. Ikada, S. Tonegawa, R. Okazaki, H. Shishido, H. Ikeda, H. Takeya, K. Hirata, T. Terashima, and Y. Matsuda, Phys. Rev. B 81, 184519 (2010).
- <sup>9</sup>S. Nandi, M. G. Kim, A. Kreyssig, R. M. Fernandes, D. K. Pratt, A. Thaler, N. Ni, S. L. Bud'ko, P. C. Canfield, J. Schmalian, R. J. McQueeney, and A. I. Goldman, Phys. Rev. Lett. **104**, 057006 (2010).
- <sup>10</sup>P. Canfield and S. Bud'ko, Annu. Rev. Condens. Matter Phys. 1, 27 (2010).
- <sup>11</sup>D. J. Singh and M.-H. Du, Phys. Rev. Lett. **100**, 237003 (2008).
- <sup>12</sup>H. Ding, P. Richard, K. Nakayama, K. Sugawara, T. Arakane, Y. Sekiba, A. Takayama, S. Souma, T. Sato, T. Takahashi, Z. Wang, X. Dai, Z. Fang, G. F. Chen, J.-L. Luo, and N.-L. Wang, Europhys. Lett. 83, 47001 (2008).
- <sup>13</sup>A. V. Chubukov, D. V. Efremov, and I. Eremin, Phys. Rev. B 78, 134512 (2008).
- <sup>14</sup>C. de la Cruz, Q. Huang, J. Lynn, J. Li, W. Ratcliff II, J. Zarestky, H. Mook, G. F. Chen, J.-L. Luo, N.-L. Wang, and P. Dai, Nature 453, 899 (2008).
- <sup>15</sup>Q. Huang, Y. Qiu, W. Bao, M. A. Green, J. W. Lynn, Y. C. Gasparovic, T. Wu, G. Wu, and X. H. Chen, Phys. Rev. Lett. 101, 257003 (2008).
- <sup>16</sup>Y. Su, P. Link, A. Schneidewind, T. Wolf, P. Adelmann, Y. Xiao, M. Meven, R. Mittal, M. Rotter, D. Johrendt, T. Brueckel, and M. Loewenhaupt, Phys. Rev. B 79, 064504 (2009).
- <sup>17</sup>T. Yildirim, Phys. Rev. Lett. **101**, 057010 (2008).
- <sup>18</sup>Q. Si and E. Abrahams, Phys. Rev. Lett. **101**, 076401 (2008).
- <sup>19</sup>Q. Si, E. Abrahams, J. Dai, and J.-X. Zhu, New J. Phys. **11**, 045001 (2009).
- <sup>20</sup> A. Jesche, N. Caroca-Canales, H. Rosner, H. Borrmann, A. Ormeci, D. Kasinathan, H. H. Klauss, H. Luetkens, R. Khasanov, A. Amato, A. Hoser, K. Kaneko, C. Krellner, and C. Geibel, Phys. Rev. B 78, 180504 (2008).

- <sup>21</sup>A. I. Goldman, D. N. Argyriou, B. Ouladdiaf, T. Chatterji, A. Kreyssig, S. Nandi, N. Ni, S. L. Bud'ko, P. C. Canfield, and R. J. McQueeney, Phys. Rev. B 78, 100506 (2008).
- <sup>22</sup>V. Barzykin and L. P. Gor'kov, Phys. Rev. B **79**, 134510 (2009).
- <sup>23</sup> A. Cano, M. Civelli, I. Eremin, and I. Paul, Phys. Rev. B 82, 020408 (2010)
- <sup>24</sup>D. K. Pratt, W. Tian, A. Kreyssig, J. L. Zarestky, S. Nandi, N. Ni, S. L. Bud'ko, P. C. Canfield, A. I. Goldman, and R. J. McQueeney, Phys. Rev. Lett. **103**, 087001 (2009).
- <sup>25</sup>R. R. Urbano, E. L. Green, W. G. Moulton, A. P. Reyes, P. L. Kuhns, E. M. Bittar, C. Adriano, T. M. Garitezi, L. Bufaiçal, and P. G. Pagliuso, Phys. Rev. Lett. 105, 107001 (2010).
- <sup>26</sup>P. Wong, P. M. Horn, R. G. Birgeneau, and G. Shirane, Phys. Rev. B 27, 428 (1983).
- <sup>27</sup>S. D. Wilson, Z. Yamani, C. R. Rotundu, B. Freelon, E. Bourret-Courchesne, and R. J. Birgeneau, Phys. Rev. B 79, 184519 (2009).
- <sup>28</sup>R. M. Fernandes, D. K. Pratt, W. Tian, J. Zarestky, A. Kreyssig, S. Nandi, M. G. Kim, A. Thaler, N. Ni, P. C. Canfield, R. J. Mcqueeney, J. Schmalian, and A. I. Goldman, Phys. Rev. B 81, 140501 (2010).
- <sup>29</sup>R. M. Fernandes, A. V. Chubukov, J. Knolle, I. Eremin, and J. Schmalian, Phys. Rev. B **85**, 024534 (2012).
- <sup>30</sup>M.-H. Julien, H. Mayaffre, M. Horvatic, C. Berthier, X. D. Zhang, W. Wu, G. F. Chen, N.-L. Wang, and J. L. Luo, Europhys. Lett. 87, 37001 (2009).
- <sup>31</sup>J. T. Park, D. S. Inosov, C. Niedermayer, G. L. Sun, D. Haug, N. B. Christensen, R. Dinnebier, A. V. Boris, A. J. Drew, L. Schulz, T. Shapoval, U. Wolff, V. Neu, X. Yang, C. T. Lin, B. Keimer, and V. Hinkov, Phys. Rev. Lett. 102, 117006 (2009).
- <sup>32</sup>E. Wiesenmayer, H. Luetkens, G. Pascua, R. Khasanov, A. Amato, H. Potts, B. Banusch, H.-H. Klauss, and D. Johrendt, Phys. Rev. Lett. 107, 237001 (2011).
- <sup>33</sup>M. Neupane, P. Richard, Y.-M. Xu, K. Nakayama, T. Sato, T. Takahashi, A. V. Federov, G. Xu, X. Dai, Z. Fang, Z. Wang, G.-F. Chen, N.-L. Wang, H.-H. Wen, and H. Ding, Phys. Rev. B 83, 094522 (2011).
- <sup>34</sup>J.-P. Castellan, S. Rosenkranz, E. A. Goremychkin, D. Y. Chung, I. S. Todorov, M. G. Kanatzidis, I. Eremin, J. Knolle, A. V. Chubukov, S. Maiti, M. R. Norman, F. Weber, H. Claus, T. Guidi, R. A. Bewley, and R. Osborn, Phys. Rev. Lett. 107, 177003 (2011).
- <sup>35</sup>J. Zhao, Q. Huang, C. de la Cruz, S. Li, J. Lynn, Y. Chen, M. A. Green, G. F. Chen, G. Li, Z. Li, J.-L. Luo, N.-L. Wang, and P. Dai, Nat. Mater. 7, 953 (2008).
- <sup>36</sup>M. Rotter, M. Pangerl, M. Tegel, and D. Johrendt, Angew. Chem. Int. Ed. 47, 7949 (2008).
- <sup>37</sup>T. Saito, S. Onari, and H. Kontani, Phys. Rev. B **82**, 144510 (2010).
- <sup>38</sup>K. Kuroki, H. Usui, S. Onari, R. Arita, and H. Aoki, Phys. Rev. B 79, 224511 (2009).
- <sup>39</sup>H. Chen, Y. Ren, Y. Qiu, W. Bao, R. H. Liu, G. Wu, T. Wu, Y. L. Xie, X. F. Wang, Q. Huang, and X. H. Chen, Europhys. Lett. **85**, 17006 (2009).

- <sup>40</sup>A. C. Larson and R. B. Von Dreele, *General Structure Analysis System (GSAS)*, Los Alamos National Laboratory Report LAUR, 86–748 (Los Alamos, NM, 2004).
- <sup>41</sup>B. H. Toby, J. Appl. Crystallogr. **34**, 210 (2001).
- <sup>42</sup>W. I. F. David, Phys. B **180**, 567 (1992).
- <sup>43</sup>M. Rotter, M. Tegel, D. Johrendt, I. Schellenberg, W. Hermes, and R. Pöttgen, Phys. Rev. B **78**, 020503 (2008).
- <sup>44</sup>M. Kofu, Y. Qiu, W. Bao, S.-H. Lee, S. Chang, T. Wu, G. Wu, and X. Chen, New J. Phys. 11, 055001 (2009).
- <sup>45</sup>R. M. Fernandes and J. Schmalian, Phys. Rev. B **82**, 014521 (2010).
- <sup>46</sup>M. Tegel, M. Rotter, V. Weiß, F. Schappacher, R. Pöttgen, and D. Johrendt, J. Phys. Condens. Matter 20, 452201 (2008).
- <sup>47</sup>D. Bergman and B. Halperin, Phys. Rev. B **13**, 2145 (1976).
- <sup>48</sup>V. Cvetkovic and Z. Tešanović, Phys. Rev. B **80**, 024512 (2009).
- <sup>49</sup>A. H. Nevidomskyy, e-print arXiv:1104.1747.

- <sup>50</sup>J.-H. Chu, J. G. Analytis, C. Kucharczyk, and I. R. Fisher, Phys. Rev. B **79**, 014506 (2009).
- <sup>51</sup>S. Drotziger, P. Schweiss, K. Grube, T. Wolf, P. Adelmann, C. Meingast, and H. V. Löhneysen, J. Phys. Soc. Jpn. **79**, 124705 (2010)
- <sup>52</sup>T. Sato, K. Nakayama, Y. Sekiba, P. Richard, Y.-M. Xu, S. Souma, T. Takahashi, G. F. Chen, J. L. Luo, N.-L. Wang, and H. Ding, Phys. Rev. Lett. **103**, 047002 (2009).
- <sup>53</sup>C.-H. Lee, A. Iyo, H. Eisaki, H. Kito, M. T. Fernandez-Diaz, T. Ito, K. Kihou, H. Matsuhata, M. Braden, and K. Yamada, J. Phys. Soc. Jpn. 77, 083704 (2008).
- <sup>54</sup>D. Singh, M.-H. Du, L. Zhang, A. Subedi, and J. An, Phys. C 469, 886 (2009).
- <sup>55</sup>S. A. J. Kimber, A. Kreyssig, Y. Z. Zhang, H. O. Jeschke, R. Valenté, F. Yokaichiya, E. Colombier, J. Yan, T. C. Hansen, T. Chatterji, R. McQueeney, P. Canfield, A. I. Goldman, and D. Argyriou, Nat. Mater. 8, 471 (2009).