Orbital-selective magnetism in the spin-ladder iron selenides $Ba_{1-x}K_xFe_2Se_3$

J. M. Caron,¹ J. R. Neilson,^{1[,*](#page-3-0)} D. C. Miller,¹ K. Arpino,¹ A. Llobet,² and T. M. McQueen^{1,[†](#page-3-0)}

¹*Institute for Quantum Matter, Department of Chemistry, and Department of Physics and Astronomy, The Johns Hopkins University,*

Baltimore, Maryland 21030, USA

²*Lujan Neutron Scattering Center, Los Alamos National Laboratory, Mail Stop H805, Los Alamos, New Mexico 87545, USA*

(Received 16 February 2012; revised manuscript received 22 April 2012; published 14 May 2012)

Here we show that the $2.80(8)\mu_B$ Fe⁻¹ block antiferromagnetic order of BaFe₂Se₃ transforms into stripe antiferromagnetic order in KFe₂Se₃ with a decrease in moment to 2.1(1) μ_B Fe⁻¹. This reduction is larger than expected from the change in electron count from Ba^{2+} to K^+ , and occurs with the loss of the displacements of Fe atoms from ideal positions in the ladders, as found by neutron pair distribution function analysis. Intermediate compositions remain insulating, and magnetic susceptibility measurements show a suppression of magnetic order and probable formation of a spin glass. Together, these results imply an orbital-dependent selection of magnetic versus bonded behavior, driven by relative bandwidths and fillings.

DOI: [10.1103/PhysRevB.85.180405](http://dx.doi.org/10.1103/PhysRevB.85.180405) PACS number(s): 74*.*70*.*Xa, 75*.*50*.*Ee, 61*.*05*.*F−, 74*.*78*.*−w

The origin of high-temperature superconductivity remains controversial.^{[1](#page-3-0)} In the case of iron-based superconductors, even the origin of magnetic and metallic ground states of nonsuperconducting parent compounds is contested. Theories range from the multiband character and nesting of the Fermi surface^{2,3} and that iron pnictides are Hund's metals,^{[4](#page-3-0)} to proposals that the compounds are more directly related to the cuprates[.5,6](#page-3-0) Further complication arises from the varied behavior of different iron compounds, including whether magnetic order competes^{[7,8](#page-3-0)} or coexists^{[9](#page-3-0)[–11](#page-4-0)} with superconductivity, as well as the presence or absence of structural distortions, 12 and their relation to magnetic order.^{[13](#page-4-0)} Orbital selection is critical to understanding the nature of these magnetic and structural states in iron-based superconductors, independent of the choice of theory[.14–16](#page-4-0)

The spin-ladder compounds $AFe₂X₃$ ($A = K$, Rb, Cs, or Ba and $X =$ chalcogenide) are structurally related to the iron superconductors, $17-19$ and are built of double chains ("ladders") of edge-sharing [Fe*X*4] tetrahedra. The *A* cations separate the ladders to form the three-dimensional structure [Fig. $1(a)$]. Compared to the layered iron-based superconductors, the $AFe₂X₃$ compounds have significantly decreased bandwidths from the reduced dimensionality [Fig. [1\(b\)\]](#page-1-0). Consequently, these compounds provide a unique opportunity to reveal the interplay between structure and magnetism in $[FeX₄]$ -based materials without the complication of metallic behavior. Here we report a systematic study of charge doping of the spin-ladder compound Ba_{1−*x*}K_{*x*}Fe₂Se₃ using resistivity, magnetic susceptibility, synchrotron powder x-ray diffraction (PXRD), and neutron powder diffraction (NPD) with pairdistribution function analysis (PDF). The data intimate a relationship between local Fe-Fe displacements and magnetic order, driven by a shift in electronic state from unpaired and magnetic to paired and bonded as a function of electron count. Further, these results imply an orbitally dependent preference for magnetism, suggesting that such relationships are pertinent to iron-based superconductivity.

Samples were made as previously reported. $20,21$ All samples were prepared and handled inside an argon-filled glovebox. NPD data were collected at temperatures between 5 and 300 K on polycrystalline $KF_{2}Se_3$ loaded in a vanadium can using the time-of flight High Intensity Powder Diffractometer (HIPD) at the Lujan Center, Los Alamos Neutron Science Center, Los Alamos National Laboratory. During shipment and loading, a portion decomposed into FeSe and $KSe₂$ (volatile). Thus, a FeSe impurity phase (*<*10 wt %), not seen in PXRD data of the same sample batch, was included in the neutron refinements. PDF analysis was performed on the total neutron scattering data and $G(r)$ were extracted with $Q_{\text{max}} = 29 \text{ Å}^{-1}$ using PDFGETN.^{[22](#page-4-0)} Refinements to the PDF were performed using PDFGUI 23 after defining instrumental resolution parameters from data collected on polycrystalline Si ($Q_{broad} = 0.041$, $Q_{\text{damp}} = 0.015$. PXRD data were collected at 300 K on beamline 11-BM at the Advanced Photon Source. Rietveld analyses were performed using FULLPROF,^{[24](#page-4-0)} and GSAS/EXPGUI.^{[25](#page-4-0)} The magnetic structure was solved by representational analysis as implemented by SARA*h* and BASIREPS/FULLPROF. [24](#page-4-0)

Electrical resistivity and dc magnetization measurements were performed between 1.8 and 300 K using a Quantum Design Physical Properties Measurement System. For resistivity, platinum wires were attached to sintered polycrystalline pellets using silver paste and dried in a desiccator. Magnetization measurements were carried out at $\mu_0 H = 1$ and 2 T, and with $\chi \approx \Delta M / \Delta H = [M_{2T} - M_{1T}]/[1 \text{ T}].$

Resistivity data, normalized to ρ_{300K} , are shown in Fig. [2\(a\).](#page-1-0) While substitution of Ba by K formally oxidizes iron from 2+ to 2.5+, all samples are insulating, with the resistivities increasing exponentially with decreasing temperature. The temperature dependence exhibits one-dimensional variable range hopping (VRH) behavior, given by $ln(\rho/\rho_{300 \text{ K}}) \propto$ $(A^2T'_0/T)^{1/2}$, where *T* is the temperature, *A* is a constant, and \tilde{T}_0 is the density of states in the localization length.²⁶ For most samples, $\log(\rho/\rho_{300 \text{ K}})$ vs $T^{-1/2}$ is linear over the entire measurement range, but for $x = 0.3, 0.9$, and 1.0, there are two discrete regions of linearity, with crossovers at $T_{\text{cross}} = 85(5)$, 70(5), and 170(5) K, respectively.

To understand if the crossover behavior is related to magnetic order, as in orthorhombic iron arsenides, $27,28$ the magnetic susceptibilities of $Ba_{1-x}K_xFe_2Se_3$ are shown in Fig. $2(b)$. For $x = 0$, we previously showed that there is a change in slope at $T_N = 256$ K, corresponding to the form-ation of magnetic order.^{[20](#page-4-0)} On increasing x , the temperature

FIG. 1. (Color online) Schematic of KF_2Se_3 (space group *Cmcm*) showing (a) stacking of $[Fe₂Se₃]$ ladders and (b) the perspective perpendicular to the plane of the ladders.

of the slope changes quickly, and drops and broadens. This observation is not surprising given that iron-deficient BaFe_{1.79(2)}Se₃ samples (nominally $Fe^{2.23+}$) have spin-glass behavior at $T < 50$ K,^{[29](#page-4-0)} and imply that at intermediate *x*, Ba1−*^x*K*x*Fe2Se3 also exhibits spin-glass behavior. Long-range antiferromagnetic order reappears for $x \geqslant 0.9$; the apparent diffuse nature of the transition in magnetization is attributed to crystalline anisotropy[.30](#page-4-0)

From these data, there is no obvious relation between the resistivity crossovers and magnetic susceptibility: For $x = 0.3$, $T_{\text{cross}} = 85(5)$ K, but spin-glass formation does not occur until $T < 50$ K, whereas for $x = 0.9$, $T_{cross} = 70(5)$ K, but magnetic order sets in at a significantly higher temperature, $T_N \approx 250$ K. This suggests that the resistivity crossovers

FIG. 2. (Color online) (a) Normalized electrical resistivity on polycrystalline pellets of Ba_{1−*x*}K_{*x*}Fe₂Se₃ highlighting onedimensional variable range hopping behavior. (b) The temperature dependences of the magnetic susceptibilities illustrate the initial suppression of magnetic order (indicated by arrows) with increasing *x*, followed by a reappearance of antiferromagnetic order.

FIG. 3. (Color online) (a) NPD data (black circles) of KF_2Se_3 and Rietveld refinement of the best magnetic structure. (b) Temperature-dependent NPD data illustrate the growth of the magnetic Bragg peaks on cooling. The magnetic moment saturates at $2.1(1)\mu_B$ Fe⁻¹. (c) Schematic of the stripe magnetic structure of KFe₂Se₃ compared to the block magnetic structure of BaFe₂Se₃ (Ref. [20\)](#page-4-0).

arise from some other effect. One possibility is changes in iron deficiency, but our PXRD Rietveld refinements show no deviation from the nominal stoichiometries from loss of K or Fe. More likely is that all samples do have the resistivity crossovers, but that for some the crossovers occur outside our measurement range. Further studies on single-crystal specimens are needed to identify whether the crossover behavior is present in all specimens, and whether they are magnetic in origin.

To investigate the apparent suppression and reappearance of long-range magnetic order, the magnetic structure for $x = 1$ (KFe₂Se₃) was solved using NPD. The best fit to data at $T = 5$ K is shown in Fig. 3(a). Magnetic Bragg peaks appear on cooling [Fig. 3(b)] and are indexed as satellites to the *Cmcm* unit cell^{[17](#page-4-0)} with a magnetic propagation vector of $\vec{k} = \left(\frac{1}{2}, \frac{1}{2}, 0\right)$. Representational analysis describes four irreducible representations, each spanned by 12 basis vectors corresponding to each unique Fe site. Trial magnetic structures were tested by Rietveld refinement to the data at 5 K, assuming all basis vectors point along the same crystallographic axis and are multiplied by a Fourier coefficient with the same magnitude. Only one configuration (Table I) describes the

TABLE I. Crystal structure parameters of KF_2Se_3 at 5 K. Space group: *Cmcm*, $a = 9.30(7)$ Å, $b = 11.41(8)$ Å, $c = 5.60(4)$ Å. The magnetic structure is described by a propagation vector of $\vec{k} = (\frac{1}{2}, \frac{1}{2}, 0)$ with $\Gamma_{\text{irrep}} = \Psi_3 - \Psi_6 + \Psi_9 - \Psi_{12}$ and a Fourier coefficient giving a moment of 2.1(1) μ _B Fe⁻¹.

Site	x	v	Z.	$Uiso$ (Å ²)
K	0.5	0.1667	0.25	0.013
Fe	0.352(2)	0.5	0	0.003
Se1	$_{0}$	0.127(2)	0.25	0.004
Se ₂	0.212(2)	0.379(2)	0.25	0.004

FIG. 4. (Color online) (a) PDFs of KF_2Se_3 at $T = 5$, 100, and 300 K (black circles) with fits to the crystallographic *Cmcm* structure. (b) Representations of a single ladder illustrate the changes in local structure and magnetism with substitution of the *A* site in $AFe₂Se₃$.

observed reflections ($R_{\text{mag}} = 15.4\%$ vs 28.9% for the next best, 90◦ bank). The Fourier coefficient was refined as a function of temperature [Fig. $3(b)$] and reveals a transition to a long-range magnetic state at $T_N \approx 250$ K, with the moment saturating at $2.1(1)\mu_B$ Fe⁻¹. The magnetic structure of KF_2Se_3 [Fig. [3\(c\)\]](#page-1-0) is of stripe antiferromagnetic order, with spins oriented in chain, parallel to the chain direction. The stripe order is analogous to the magnetic structure observed in many parent pnictides such as $BaFe₂As₂$ and $LaFeAsO.^{31,32}$ $LaFeAsO.^{31,32}$ $LaFeAsO.^{31,32}$ This contrasts with the block antiferromagnetic order that we found in $BaFe₂Se₃$ with spins oriented perpendicular to the chain direction. 20 The block order has no direct analogs among known iron superconductors, but is closely related to the magnetic order of A_x Fe_{2−*y*}Se₂.^{[9](#page-3-0)}

In BaFe₂Se₃, local Fe-Fe displacements and magnetic order are coupled. $20,33$ To determine if similar displacements are present in KF_2Se_3 , PDF analysis of KF_2Se_3 was carried out at $T = 5$, 100, and 300 K. At all temperatures, the PDFs are well described by the average *Cmcm* crystallographic structure [Fig. 4(a)], with a single Fe-Fe distance (2.83 Å) along the chain direction. The poorest fit is to the $T = 300$ K PDF, and puts an upper bound on local Fe-Fe distances from 2.75 to 2.87 Å ($\Delta \leqslant 0.12$ Å). This is substantially less than in BaFe₂Se₃,^{[20](#page-4-0)} where Fe-Fe distances of 2.62 and 2.83 Å $(\Delta = 0.21 \text{ Å})$ are found [Fig. 4(b)]. These data show that the large local Fe-Fe displacements seen in $BaFe₂Se₃$ are much reduced or absent in KF_2Se_3 .

The properties of Ba_{1−*x*}K_{*x*}Fe₂Se₃ are summarized in Fig. 5. For $x > 0$, block antiferromagnetic order (AFM-B) is suppressed, indicated by the rapid decrease in T_{mag} . This is followed by the appearance of stripe antiferromagnetic order (AFM-S) for $x > 0.9$. At intermediate *x*, long-range magnetic order is likely replaced by a spin-glass state. Structural parameters from a Rietveld refinement of PXRD data using the *Pnma* space group generates long and short Fe-Fe distances [Fig. $5(b)$, solid diamonds and triangles, respectively]. The difference between the long and short Fe-Fe separations remains nearly constant with $0 < x \le 0.5$, while the average

FIG. 5. (Color online) Phase diagram of Ba_{1−*x*}K_{*x*}Fe₂Se₃. (a) As *x* increases, block antiferromagnetic order (AFM-B) is suppressed, but for $x \ge 0.9$, stripe antiferromagnetic order (AFM-S) appears. (b) The average Fe-Fe distance at 300 K steadily increases (open circles), while the Fe-Fe displacements, determined by Rietveld (long, solid diamonds; short, solid triangles) and PDF analysis of KFe2Se3 (solid squares), vanish. Concomitantly, a linear increase in $[(k_B T_0)^{-1}]$ is consistent with the addition of hopping carriers. Above $x > 0.7$, a downturn indicates either a reduction in the number of, or an increase in localization of, carriers. (c) Orbital selective magnetism in Ba_{1−*x*}K_{*x*}Fe₂Se₃: BaFe₂Se₃ has four half-filled *d* bands, three narrow and one dispersed. While on-site repulsion (*U*) gaps the narrow bands, the dispersed band is gapped by a density wave. In KF_2Se_3 , depopulation reduces the effect of *U* on one of the narrow bands, resulting in its switch to nonmagnetic behavior.

Fe-Fe distance [Fig. $5(b)$, open red circles] steadily increases. For $x > 0.5$, the magnitude of the average Fe-Fe displacements is reduced. Local Fe-Fe displacements, determined by PDF analysis [solid squares, Fig. $5(b)$], confirm distinct short and long distances for $x = 0$ that are dramatically reduced or vanish when $x = 1$. Further support for the loss of Fe-Fe displacements comes from the change in space group from *Pnma* to *Cmcm* around $x = 0.9$. Fe-Fe displacements are allowed in *Pnma*, but not in *Cmcm*. Together, these data imply a large change in local Fe-Fe distances and exchange interactions across the series.

These structural changes also track with changes in onedimensional VRH seen in resistivity. For *x <* 0.7, we observe a linear increase in $[(k_B T_0)^{-1}]$ (with T_0 defined as $A^2 T_0$ and fit to the high-*T* regions). The value of $[(k_B T_0)^{-1}]$ is proportional to the product of the localized density of states and J. M. CARON *et al.* PHYSICAL REVIEW B **85**, 180405(R) (2012)

the localization length. 26 This increase is consistent with the addition of carriers on doping, assuming a constant localization length. For $x \ge 0.9$, the trend in $[(k_B T_0)^{-1}]$ with *x* inverts, indicating either a reduction in the density of states, or increased localization, of carriers. Such an inflection implies a nontrivial rearrangement of the electronic structure.

Explanation of these data requires consideration of the multiband electronic structure, derived from the five 3*d* orbitals near the Fermi level. The compounds studied here are one dimensional and all are insulating. Therefore, all bands that would cross E_F must be gapped by some mechanism. The local bonding geometry and electron count are comparable to the two-dimensional iron superconductors. As such, the bands that must be gapped are similar, in this case, four partly filled metal d -orbital-derived bands.^{[29,34](#page-4-0)}

Gap formation can arise from on-site repulsion (Hubbard *U*) or by the formation of nonmagnetic order, such as a charge density wave (CDW)[.35–37](#page-4-0) Localization by a Hubbard *U* to form magnetic states is favored for narrow bands near half filling. For the empirically observed $2.80(8)\mu_{\rm B}$ Fe⁻¹ observed in BaFe₂Se₃, this implies that U gaps three bands, while a density wave or some more complex order opens a gap in the fourth. Such a density wave, if it produces a CDW, would then explain the two distinct Fe-Fe distances along the chain in BaFe₂Se₃. This distortion in turn causes variations in nearest-neighbor (NN) exchange, resulting in the block magnetic structure.

In KFe2Se3, there are 0.5*e*[−] per Fe fewer electrons, which, in the simplest case, results in quarter filling of one of the three bands originally split by U [Fig. $5(c)$]. If U was still responsible for the splitting of this third band, then a magnetic moment of $2.5\mu_B$ Fe⁻¹ would be observed. However, NPD shows only 2.1(1) μ B Fe⁻¹. This implies that one of the three magnetic bands in $BaFe₂Se₃$ is instead gapped in a nonmagnetic fashion in KF_2Se_3 (the effect of *U* is reduced away from half filling). The switch to nonmagnetic behavior then either produces a second CDW with periodicity and/or phase different than the first, or some more complex density wave (valley density wave, $38 \text{ bond-order wave}, \frac{39}{3}$ etc.). In either case, there is no longer a single CDW with a periodicity of 2*d*_{Fe−Fe}, resulting in a reduction or elimination of the Fe-Fe displacements, as observed for KF_2Se_3 .

Such a change also explains the suppression and reappearance of magnetic order due to a variation in the local exchange from Fe-Fe displacements. Such disorder reduces spin-freezing temperatures, even when the individual exchange interactions remain strong.[40](#page-4-0) The crystal structures of $Ba_{1-x}K_xFe_2Se_3$ also support this model; depopulation of the magnetic band for $x \le 0.5$ leaves the $2d_{\text{Fe}-\text{Fe}}$ CDW intact, retaining the long and short Fe-Fe distances [Fig. [5\(b\)\]](#page-2-0). For $x > 0.5$, the shrinking magnitude of the Fe-Fe displacements indicates the switch to more complex nonmagnetic density wave behavior.

This switch from three to two bands gapped by *U* also means that the orbitals contributing to the magnetic moment changes (as each band has a different set of orbital contributions), implying an orbital selection of magnetism. This change explains the switch in moment direction, since the shape of the electron density no longer contains contributions from the third band. Further, this model explains the inflection of $[(k_B T_0)^{-1}]$: Incremental increases in charge doping beyond $x > 0.7$ do not provide additional carriers; instead, the added carriers are localized by a density wave.

In conclusion, the competition of bandwidth and Hubbard *U* for each *d* band in $Ba_{1-x}K_xFe_2Se_3$ brings an orbital selectivity to the magnetic state. The change to antiferromagnetic stripe order is a natural consequence of the more uniform nearest-neighbor exchange interactions, arising from a change in underlying density wave(s). The insulating behavior results from different bands being gapped by different mechanisms; some are magnetic, some are not. Our results imply that small changes in band filling have a large effect on the nature of each *d* band in iron-based materials. Further work is needed to identify the specific change in orbital contributions in $Ba_{1-x}K_xFe_2Se_3$. We speculate that this sensitivity, combined with interactions between bands, can give rise to exotic spin-singlet states and may explain the high-temperature superconductivity in these systems.

This research is supported by the US Department of Energy, Office of Basic Energy Sciences, Division of Materials Sciences and Engineering under Award No. DE-FG02-08ER46544. This work has benefited from the use of HIPD at the Lujan Center at Los Alamos Neutron Science Center, funded by DOE Office of Basic Energy Sciences. Los Alamos National Laboratory is operated by Los Alamos National Security LLC under DOE Contract No. DE-AC52- 06NA25396. This research has also benefited from the use of the Advanced Photon Source at Argonne National Laboratory supported by the US Department of Energy, Office of Science, Office of Basic Energy Sciences, under Contract No. DE-AC02-06CH11357.

- 2V. Cvetkovic and Z. Tesanovic, [Phys. Rev. B](http://dx.doi.org/10.1103/PhysRevB.80.024512) **80**, 024512 [\(2009\).](http://dx.doi.org/10.1103/PhysRevB.80.024512)
- 3V. Cvetkovic and Z. Tesanovic, [Europhys. Lett.](http://dx.doi.org/10.1209/0295-5075/85/37002) **85**, 37002 [\(2009\).](http://dx.doi.org/10.1209/0295-5075/85/37002)
- 4Z. P. Yin, K. Haule, and G. Kotliar, Nat. Mater. **10**[, 932 \(2011\).](http://dx.doi.org/10.1038/nmat3120)
- 5P. Anderson, Science **235**[, 1196 \(1987\).](http://dx.doi.org/10.1126/science.235.4793.1196)
- 6F. Wang and D. H. Lee, Science **332**[, 200 \(2011\).](http://dx.doi.org/10.1126/science.1200182)
- 7J. Bacsa, A. Y. Ganin, Y. Takabayashi, K. E. Christensen,
- K. Prassides, M. J. Rosseinsky, and J. B. Claridge, [Chem. Sci.](http://dx.doi.org/10.1039/c1sc00070e) **2**[, 1054 \(2011\).](http://dx.doi.org/10.1039/c1sc00070e)
- 8W. Li, H. Ding, P. Deng, K. Chang, C. Song, K. He, L. Wang, X. Ma, J.-P. Hu, X. Chen, and Q.-K. Xue, Nat. Phys. **7**, 1 (2011).
- $9W.$ Bao, Q.-Z. Huang, G.-F. Chen, D.-M. Wang, J.-B. He, and Y.-M. Qiu, [Chin. Phys. Lett.](http://dx.doi.org/10.1088/0256-307X/28/8/086104) **28**, 086104 (2011).

^{*} jneilso2@jhu.edu

[†] mcqueen@jhu.edu

¹I. Mazin, Physics **4**[, 26 \(2011\).](http://dx.doi.org/10.1103/Physics.4.26)

- 10D. H. Ryan, W. N. Rowan-Weetaluktuk, J. M. Cadogan, R. Hu, W. E. Straszheim, S. L. Budko, and P. C. Canfield, [Phys. Rev. B](http://dx.doi.org/10.1103/PhysRevB.83.104526) **83**, [104526 \(2011\).](http://dx.doi.org/10.1103/PhysRevB.83.104526)
- 11L. Li, Z. R. Yang, Z. T. Zhang, W. Tong, C. J. Zhang, S. Tan, and Y. H. Zhang, Phys. Rev. B **84**[, 174501 \(2011\).](http://dx.doi.org/10.1103/PhysRevB.84.174501)
- 12T. McQueen, A. Williams, P. W. Stephens, J. Tao, Y. Zhu, V. Ksenofontov, F. Casper, C. Felser, and R. J. Cava, [Phys. Rev. Lett.](http://dx.doi.org/10.1103/PhysRevLett.103.057002) **103**[, 057002 \(2009\).](http://dx.doi.org/10.1103/PhysRevLett.103.057002)
- 13C. A. Balseiro, P. Schlottmann, and F. Yndurain, [Phys. Rev. B](http://dx.doi.org/10.1103/PhysRevB.21.5267) **21**, [5267 \(1980\).](http://dx.doi.org/10.1103/PhysRevB.21.5267)
- 14W. Lv and P. Phillips, Phys. Rev. B **84**[, 174512 \(2011\).](http://dx.doi.org/10.1103/PhysRevB.84.174512)
- 15C. C. Chen, J. Maciejko, A. P. Sorini, B. Moritz, R. R. P. Singh, and T. P. Devereaux, Phys. Rev. B **82**[, 100504\(R\) \(2010\).](http://dx.doi.org/10.1103/PhysRevB.82.100504)
- 16I. Eremin and A. V. Chubukov, Phys. Rev. B **81**[, 024511 \(2010\).](http://dx.doi.org/10.1103/PhysRevB.81.024511)
- 17K. O. Klepp, W. Sparlinek, and H. Boller, [J. Alloys Compd.](http://dx.doi.org/10.1016/0925-8388(95)02087-X) **238**, 1 [\(1996\).](http://dx.doi.org/10.1016/0925-8388(95)02087-X)
- 18M. Reissner, W. Steiner, J. Wernisch, and H. Boller, in *Proceedings of the 28th International Conference on the Applications of the Mössbauer Effect (ICAME 2005)*, edited by P.-E. Lippens, J.-C. Jumas, and J.-M. R. Genin (Springer, Berlin, 2007), p. 1301. ´
- 19H. Steinfink, *Mixed Valence States of Iron in the Ba-Fe-S System* (American Chemical Society, Washington, DC, 1980).
- 20J. M. Caron, J. R. Neilson, D. C. Miller, A. Llobet, and T. M. McQueen, Phys. Rev. B **84**[, 180409 \(2011\).](http://dx.doi.org/10.1103/PhysRevB.84.180409)
- ²¹The two end members, $BaFe₂Se₃$ and $KFe₂Se₃$, were made from the reaction of the elements by first prereacting polished low-oxygen Fe pieces (Alfa 99.99%, low oxygen) and Se beads (Alfa 99.999%) in a 2:3 ratio in evacuated quartz tubes at 850 ◦C for 40 h and then heated to 1050 °C over 6 h before furnace cooling. This was then intimately mixed with 1% excess stoichiometry of small pieces of Ba or K metal, placed in alumina crucibles in evacuated quartz tubes, and heated at 20 ◦C*/*h to 450 ◦C and held for 14 h. The prereacted powders were then ground, pressed into pellets, and reheated in alumina crucibles in evacuated quartz tubes at 765° C (Ba) and 700 °C (K) for 12 h at a time, followed by quenching in water (K) only). Regrindings and reheatings were performed until each sample was phase pure as determined by laboratory x-ray diffraction (six to ten repetitions). Intermediate compositions $Ba_{1-x}K_xFe_2Se_3$, where $x = 0.1, 0.3, 0.55, 0.7,$ and 0.9, were prepared by mixing the appropriate ratio of the end members and heating once, as pellets, in alumina crucibles in evacuated quartz tubes at 700 ◦C followed by quenching in water.
- $22P$. F. Peterson, M. Gutmann, T. Proffen, and S. [J.](http://dx.doi.org/10.1107/S0021889800007123) L. Billinge, J. [Appl. Crystallogr.](http://dx.doi.org/10.1107/S0021889800007123) **33**, 1192 (2000).
- 23C. L. Farrow, P. Juhas, J. W. Liu, D. Bryndin, E. S. Bozin, J. Bloch, T. Proffen, and S. J. L. Billinge,[J. Phys. Condens. Matter](http://dx.doi.org/10.1088/0953-8984/19/33/335219) **19**, 335219 [\(2007\).](http://dx.doi.org/10.1088/0953-8984/19/33/335219)
- ²⁴J. Rodríguez-Carvajal, *Physica B* **192**[, 55 \(1993\).](http://dx.doi.org/10.1016/0921-4526(93)90108-I)
- 25B. H. Toby, [J. Appl. Crystallogr.](http://dx.doi.org/10.1107/S0021889801002242) **34**, 210 (2001).
- 26P. A. Lee, [Phys. Rev. Lett.](http://dx.doi.org/10.1103/PhysRevLett.53.2042) **53**, 2042 (1984).
- 27T.-M. Chuang, M. P. Allan, J. Lee, Y. Xie, N. Ni, S. L. Bud'ko, G. S. Boebinger, P. C. Canfield, and J. C. Davis, [Science](http://dx.doi.org/10.1126/science.1181083) **327**, 181 [\(2010\).](http://dx.doi.org/10.1126/science.1181083)
- 28J.-H. Chu, J. G. Analytis, K. De Greve, P. L. McMahon, Z. Islam, Y. Yamamoto, and I. R. Fisher, Science **329**[, 824 \(2010\).](http://dx.doi.org/10.1126/science.1190482)
- 29B. Saparov, S. Calder, B. Sipos, H. Cao, S. Chi, D. J. Singh, A. D. Christianson, M. D. Lumsden, and A. S. Sefat, [Phys. Rev. B](http://dx.doi.org/10.1103/PhysRevB.84.245132) **84**, [245132 \(2011\).](http://dx.doi.org/10.1103/PhysRevB.84.245132)
- 30H. Lei, H. Ryu, A. I. Frenkel, and C. Petrovic, [Phys. Rev. B](http://dx.doi.org/10.1103/PhysRevB.84.214511) **84**, [214511 \(2011\).](http://dx.doi.org/10.1103/PhysRevB.84.214511)
- 31Q. Huang, Y. Qiu, W. Bao, M. A. Green, J. W. Lynn, Y. C. Gasparovic, T. Wu, G. Wu, and X. H. Chen, [Phys. Rev. Lett.](http://dx.doi.org/10.1103/PhysRevLett.101.257003) **101**, [257003 \(2008\).](http://dx.doi.org/10.1103/PhysRevLett.101.257003)
- 32C. De La Cruz, Q. Huang, J. W. Lynn, J. Li, W. Ratcliff, J. L. Zarestky, H. A. Mook, G. F. Chen, J. L. Luo, N. L. Wang, and P. Dai, [Nature \(London\)](http://dx.doi.org/10.1038/nature07057) **453**, 899 (2008).
- 33Y. Nambu, K. Ohgushi, S. Suzuki, F. Du, M. Avdeev, Y. Uwatoko, K. Munakata, H. Fukazawa, S. Chi, Y. Ueda, and T. J. Sato, [Phys.](http://dx.doi.org/10.1103/PhysRevB.85.064413) Rev. B **85**[, 064413 \(2012\).](http://dx.doi.org/10.1103/PhysRevB.85.064413)
- 34A. Subedi, L. Zhang, D. J. Singh, and M. H. Du, [Phys. Rev. B](http://dx.doi.org/10.1103/PhysRevB.78.134514) **78**, [134514 \(2008\).](http://dx.doi.org/10.1103/PhysRevB.78.134514)
- 35M. Nakamura, Phys. Rev. B **61**[, 16377 \(2000\).](http://dx.doi.org/10.1103/PhysRevB.61.16377)
- 36M. H. Whangbo, in *Crystal Chemistry and Properties of Materials with Quasi-One-Dimensional Structures*, edited by J. Rouxel (Reidel, Dordrecht, 1986), pp. 27–85.
- 37T. M. McQueen, D. M. Ho, C. J. Cahua, R. J. Cava, R. A. J. Pascal, and Z. G. Soos, [Chem. Phys. Lett.](http://dx.doi.org/10.1016/j.cplett.2009.04.085) **475**, 44 (2009).
- 38J. Kang and Z. Tesanovic, [Phys. Rev. B](http://dx.doi.org/10.1103/PhysRevB.83.020505) **83**, 020505(R) [\(2011\).](http://dx.doi.org/10.1103/PhysRevB.83.020505)
- 39M. Kumar and Z. G. Soos, [Phys. Rev. B](http://dx.doi.org/10.1103/PhysRevB.82.155144) **82**, 155144 [\(2010\).](http://dx.doi.org/10.1103/PhysRevB.82.155144)
- 40J. R. Neilson, D. E. Morse, B. C. Melot, D. P. Shoemaker, J. A. Kurzman, and R. Seshadri, [Phys. Rev. B](http://dx.doi.org/10.1103/PhysRevB.83.094418) **83**, 094418 [\(2011\).](http://dx.doi.org/10.1103/PhysRevB.83.094418)