Slow dynamics of interacting antiferromagnetic nanoparticles

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We study magnetic relaxation dynamics, memory and aging effects in interacting polydisperse antiferromagnetic NiO nanoparticles by solving a master equation using a two-state model. We investigate the effects of interactions using dipolar, nearest-neighbour short-range (NNSR) and long-range mean-field (LRMF) interactions. The magnetic relaxation of the nanoparticles in a time-dependent magnetic field has been studied using LRMF interaction. The size-dependent effects are suppressed in the ac-susceptibility, as the frequency is increased. We find that the memory dip, that quantifies the memory effect is about the same as that of non-interacting nanoparticles for the NNSR case. There is a stronger memory-dip for LRMF, and a weaker memory-dip for the dipolar interactions. We have also shown a memory effect in the Zero-field-cooled magnetization for the dipolar case, a signature of glassy behavior, from Monte-Carlo studies.

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I. INTRODUCTION

Magnetism in nanoparticles has received enormous attention in recent years due to its technological¹⁻³ as well as fundamental research aspects. $4-32$ Amid many studies related to magnetic nanoparticles, the important concerns were related to the relaxation behavior of the assemblies of nanoparticles which has been addressed in the recent times.^{[7–12](#page-11-0)} The dynamics of an assembly of nanoparticles at low temperatures has gained a lot of attention over the last few years. In a dilute system of nanoparticles, the interparticle interaction is very small as compared to the anisotropy energy of the individual particles. These isolated particles follow the dynamics in accordance with the Néel-Brown model, 33 and the system is known as superparamagnetic. The giant spin moment of the nanoparticles thermally fluctuates between their easy directions at high temperatures. As the temperature is lowered toward a blocking temperature, the relaxation time becomes equal to the measuring time and the superspin moments freeze along one of their easy directions. As the role of interparticle interaction becomes significant, the nanoparticles do not behave like individual particles; rather, their dynamics is governed by the collective behavior of the particles, like in a spin glass. $7-12$

In our recent work, 34 we studied the slow dynamics of NiO nanoparticles distributed sparsely so that particle-particle interactions could be neglected. However, in reality, interparticle interactions play a major role in describing various interesting phenomena observed experimentally in a collection of nanoparticles. This leads us to include interactions among the particles in our study. If the interparticle interaction is too small, the dynamics of an assembly of nanoparticles is a result of the individual nanoparticle dynamics. Increasing the particle-particle interaction, the behavior of the system becomes more complicated, even if we consider each nanoparticle as a single giant-spin. The dynamics of the assembly of nanoparticles involves various degrees of freedom related to the individual particles coupled to each other with interparticle interactions. In this paper, we will discuss various interactions comparatively.

The main type of magnetic interactions that can play an important role in the assemblies of nanoparticles are: (i) longrange interactions (for example, dipolar interactions) among the particles, and (ii) short-range exchange-interactions arising due to the surface spins of the particles which are in close contact. The dipolar interaction is a peculiar kind of interaction which may favour ferromagnetic or antiferromagnetic alignment of the magnetic moments depending upon the geometry of the system. This interaction among the nanoparticles may give rise to a collective behavior which may lead to a dynamics similar to that of a spin-glass. A resemblance from spin glass system owes to a random distribution of easy axes, which causes disorder and frustration of the magnetic interactions. The complex interplay between the disorder and frustration determines the state of the system and its dynamical properties. These systems are widely known as superspin glasses. This superspin glass phase has been characterised by observations of a critical slowing-down, 13 a divergence in the non-linear susceptibility, $\frac{16}{16}$ $\frac{16}{16}$ $\frac{16}{16}$ and aging and relaxation effects in the low-frequency ac susceptibility.^{[17](#page-11-0)} The Monte Carlo simulations on the system of an assembly of nanoparticles show aging 35 and magnetic relaxation behavior 36 like in a spin glass. For an interacting assembly of nanoparticles, the aging and memory effects are the two aspects that have been studied extensively in recent years, however, mostly in the case of ferromagnetic particles[.7–9,16,18–29](#page-11-0)

In this paper, we investigate the effects of interactions on a collection of a few antiferromagnetic nanoparticles, which has received relatively lesser attention. Recently, we have shown that for NiO nanoparticles, a combined effect of the surface-roughness effect and finite-size effects in the core magnetization leads to size-dependent fluctuations in net magnetic-moment.³⁷ These size-dependent fluctuations in the magnetization lead to a dynamics which is qualitatively different from that of ferromagnetic nanoparticles. We will study the dynamics of the system by analytically solving the master equation for a two-state model. We will invoke various interactions and study the relaxation phenomena under these interactions comparatively. We will show the effect of the sizedependent magnetization fluctuations on the time-dependent properties of an interacting assembly of nanoparticles. We will also compare the dynamics with that of the noninteracting case. We will discuss the memory effects in the FC as well as the ZFC magnetization protocols. The organisation of this paper

is as follows. In Section II, we discuss the two-state model. We discuss the ZFC and FC magnetizations in Sec. [III,](#page-3-0) and the ac-susceptibility study in Sec. [IV.](#page-4-0) In Section [V,](#page-6-0) we show the memory effect investigations. The aging effect has been presented in Sec. [VI.](#page-9-0) Finally, we summarize in Sec. [VII.](#page-10-0)

II. RELAXATION PHENOMENA AND VARIOUS INTERACTIONS

We assume a nanoparticle to be a giant spin in which the exchange interactions are so strong that all the spins in the particle show a coherent rotation in unison. Hence, the dynamics of a system of nanoparticles can be easily understood as the dynamics of a giant spin-moment. Here, we are dealing with antiferromagnetic nanoparticles, which display size dependent fluctuations in the magnetization 37 as shown in the inset of the figure Fig. 1. However, the net magnetic moment of ferromagnetic nanoparticles shows a linear dependence on the volume of the particles. 23 As we study the dynamics of antiferromagnetic nanoparticles, we might expect the role of these size-dependent fluctuations in the magnetization to be manifested in the time-dependent properties of these antiferromagnetic nanoparticles. We use a simple model where the energy of each particle *i* is contributed by the anisotropy energy and the Zeeman energy. Now, to incorporate interactions in the present model, we add an interaction energy term V_{ij} in the Hamiltonian. Thus, the Hamiltonian is written as:

$$
\mathcal{H} = -\sum_{i} \mathcal{K}V_i - \sum_{i} \vec{\mu}_i \cdot \vec{H} + \sum_{i} \sum_{j \neq i} V_{ij}, \qquad (1)
$$

and the energy of a particle is given by

$$
E_i = -\mathcal{K}V_i - \mu_i H + \sum_{j \neq i} \mathcal{V}_{ij},\tag{2}
$$

where $\vec{\mu}_i$ is the magnetic moment of the *i*th particle, $\mu_i =$ $|\vec{\mu}_i|$, K is the anisotropy constant, and *H* is the applied field.

FIG. 1. A lognormal distribution of nanoparticles with sizes ranging from $1.3a_0-5.3a_0$, where $a_0 = 4.17\text{Å}$. The width of the distribution is 0*.*6. The inset displays the magnetic moment vs the particle size for the same size range. The magnetic moment has a non-monotonic and oscillatory dependence on *R*. [37](#page-11-0)

Because of the interaction term V_{ij} , the energy of the *i*th particle depends on the state of all the other particles. We assume each nanoparticle to be an Ising spin, and the magnetic moment of each particle to be $\vec{\mu}_i = \mu_i s_i \hat{z}$, with $s_i = \pm 1$. The various interactions used in the present study are:

(1) A nearest-neighbour short-range (NNSR) type interaction, which can be written as:

$$
\mathcal{V}_{ij}^{\text{sr}} = J^{\text{sr}} \sum_{\langle i,j \rangle} \mu_i \mu_j,\tag{3}
$$

where the summation is taken over the nearest-neighbour sites. Here $J^{\rm sr}$ is the NNSR interaction strength.

(2) A long-Range, mean-field (LRMF) type interaction, 38 which is given as:

$$
\mathcal{V}_{ij}^{\text{lr}} = \frac{J^{\text{lr}}}{2N} \left(\sum_{i=1}^{N} \mu_i \right)^2
$$

=
$$
\frac{J^{\text{lr}}}{2N} \sum_{i=1}^{N} \mu_i^2 + \frac{J^{\text{lr}}}{N} \sum_{j \neq i} \mu_i \mu_j,
$$
 (4)

where N is the total number of nanoparticles and J^{lr} is interaction parameter. The above equation Eq. (4) is scaled by *N* to ensure that the total interaction-energy is proportional to *N* and not N^2 .

(3) A long-range dipolar interaction, with the particle *j* separated by a distance r_{ij} from particle *i*, given as:

$$
\mathcal{V}_{ij}^{\text{dipolar}} = \alpha \sum_{j \neq i} \left[\frac{\vec{\mu}_i \vec{\mu}_j}{r_{ij}^3} - \frac{3(\vec{\mu}_i \cdot \vec{r}_{ij})(\vec{\mu}_j \cdot \vec{r}_{ij})}{r_{ij}^5} \right],\tag{5}
$$

where $\alpha = \mu_0 \mu_B^2 / 4\pi r_0^2$ characterises the strength of the dipolar interaction and r_0 is the lattice parameter of the lattice, where nanoparticles are placed. Variations in r_0 may result the nanoparticles to be getting closer or away from each other and hence, increasing or decreasing the effects of dipolar interactions.

Let us consider a model case of NNSR. Using the mean field approximation, we can find a mean field felt by the ith particle due to interactions with its neighbours. We can replace Eq. (3) by its mean-field form as:

$$
\mathcal{V}_{ij}^{\text{MF}} = \mu_i \sum_{j \neq i} \langle \mu_j \rangle,
$$

= $\mu_i H_i^{\text{MF}},$ (6)

where H_i^{MF} is the mean field on the *i*th spin from all the nearest neighbor spins. Thus, in this framework, each particle can be viewed under the influence of a local magnetic-field:

$$
H_i^{\text{eff}} = H + H_i^{\text{MF}}.\tag{7}
$$

The effect of interactions on a nanoparticle can be incorporated by replacing the external field by an effective field which may lead to a mean-field equation of state and can be solved selfconsistently. Hereafter, in all the cases of interactions, we will define an effective field H_i^{eff} on a nanoparticle *i* as the sum

of the external field H and a locally-changing interaction-field i.e.,

$$
H_i^{\text{eff}} = H + \frac{1}{\mu_i} \sum_{j \neq i} \mathcal{V}_{ij}.
$$
 (8)

The above equation shows that the role of the interactions is to modify the energy barrier, which is solely due to the anisotropy contributions of each particle in the non-interacting case. For strong interactions, their effects become dominant and the individual energy-barriers can no longer be considered to be the only relevant energy scale. In this case, the relaxation is governed by a co-operative phenomenon of the system. The energy landscape with a complex hierarchy of local minima is similar to that of a spin glass. We should note that in contrast with the static energy barrier distribution arising only from the anisotropy contribution, the reversal of one particle moment may change the energy barriers of the assembly, even in the weak-interaction limit. Therefore, the energy-barrier distribution gets modified as the magnetization relaxes.

By defining an effective field given by Eq. (8), we can map an interacting assembly of nanoparticles as a collection of noninteracting nanoparticles, where each particle experiences an effective field H_i^{eff} which gets modified as the temperature changes. In the absence of the magnetic field, the superparamagnetic relaxation-time for the thermal activation over the energy barrier KV_i is given by $\tau_i = \tau_0 \exp(\frac{CV_i}{k_B T})$, where τ_0 , the microscopic time, is of the order of 10⁻⁹ s. The anisotropy constant K has a typical value of about $4 \times$ 10^{-1} J cm⁻³ for NiO. (Ref. [39\)](#page-11-0).

The occupation probabilities with the magnetic moments parallel and antiparallel to the magnetic field direction are denoted by $p_1(t)$ and $p_2(t) = 1 - p_1(t)$, respectively. The magnetic moment of each particle is supposed to occupy one of the two available states with energies $-\mathcal{K}V_i + \mu_i^s H_i^{\text{eff}}$ or $-CV_i - \mu_i^s H_i^{\text{eff}}$, where μ_i^s is the saturation magnetic-moment of the *i*th particle. These probabilities satisfy a master equation, which is given by,

$$
\frac{d}{dt}p_1(t) = -\frac{1}{\tau_i}p_1(t) + \frac{1}{2\tau_i} \left[1 + \frac{\mu_i^s H_i^{\text{eff}}}{T} \right].
$$
 (9)

The magnetic moment of each particle of volume V_i can be written as $\mu_i = [2p_1 - 1]\mu_i^s$, as

$$
\frac{d}{dt}\mu_i = -\frac{\mu_i}{\tau_i} + \frac{1}{2\tau_i} \left[1 + \frac{\left(\mu_i^s\right)^2 H_i^{\text{eff}}}{T} \right].\tag{10}
$$

We can define magnetization of each particle as $M_i \equiv \mu_i / V_i$. Because H_i^{eff} also contains a summation over all the other μ_j 's interacting with the particle, the right-hand side of the above equation can be written as the sum of the interaction term with jth particle and the term containing the magnetic field. Thus, we can write Eq. (10) as:

$$
\frac{d}{dt}M_i = A(i,j)M_j - a_{0i}.\tag{11}
$$

Here $A(i, j)$ represents the interaction term which is given below for each case separately, and $a_{0i} = -(\mu_i^s)^2 H/V_i \tau_i T$.

1. For NNSR interaction, $A(i, j)$ is given as:

$$
A(i,j) = \begin{cases} -\frac{1}{\tau_i}, & \text{if } i = j; \\ \frac{J^{\text{st}}(\mu_i^s)^2}{V_i \tau_i T} \delta_{\vec{r}, \vec{r} + \epsilon}, & \text{if } i \neq j, \end{cases}
$$

where \vec{r} is the position vector of any site and ϵ is the nearestneighbour distance.

2. For LRMF interaction, we can write $A(i, j)$ as

$$
A(i,j) = \begin{cases} -\frac{1}{\tau_i} + \frac{J^{\text{tr}}}{2N} \frac{(\mu_i^s)^2}{V_i \tau_i T}, & \text{if } i = j; \\ \frac{J^{\text{tr}}}{N} \frac{(\mu_i^s)^2}{V_i \tau_i T}, & \text{if } i \neq j. \end{cases}
$$

3. For dipolar interaction we can write,

$$
A(i,j) = \begin{cases} -\frac{1}{\tau_i}, & \text{if } i = j; \\ \frac{\alpha(\mu_i^s)^2}{V_i \tau_i T} \frac{(r_{ij}^2 - 3z_{ij}^2)}{r_{ij}^s}, & \text{if } i \neq j. \end{cases}
$$

There are N equations of the form given by Eq. (11) for the system of *N* particles. We can write these N equations in a matrix form as:

$$
\frac{d}{dt}\hat{\mathcal{M}} = \mathcal{A}\hat{\mathcal{M}} - \hat{a}_0,\tag{12}
$$

where A is a $N \times N$ matrix whose elements are given by *A*(*i*, *j*), defined above, and $\hat{\mathcal{M}}$ and \hat{a}_0 are vectors of length *N*. The elements of $\hat{\mathcal{M}}$ are the M_i 's, and those of \hat{a}_0 are the a_{0i} 's. Eq. (12) can be solved for the various interactions defined above. The formal solution of the matrix Eq. (12) takes the form:

$$
\hat{\mathcal{M}}(t) = e^{At}\hat{C} + \hat{\nu},\tag{13}
$$

where $\hat{v} = A^{-1}\hat{a}_0$, a time-independent solution of Eq. (13) and $\hat{C} = \hat{\mathcal{M}}(0) - \hat{v}$, $\hat{\mathcal{M}}(0)$ being the value of $\hat{\mathcal{M}}$ at $t = 0$. Thus we can write,

$$
\hat{\mathcal{M}}(t) = e^{\mathcal{A}t}\hat{\mathcal{M}}(0) + (1 - e^{\mathcal{A}t})\hat{\nu}.
$$
 (14)

The matrix A in Eq. (14) is a general matrix of order $N \times N$. To evaluate e^{At} , we use the MATLAB function $expm$ which implements a diagonal Pade approximation of exponential of ´ the matrix with a scaling and squaring technique. $41,42$

Padé approximation to an exponential of a matrix β is written as:

$$
e^{\mathcal{B}} \approx R_{pq}(\mathcal{B}) = [D_{pq}(\mathcal{B})]^{-1} N_{pq}(\mathcal{B}), \tag{15}
$$

where the numerator term is given as,

$$
N_{pq}(\mathcal{B}) = \sum_{j=0}^{p} \frac{(p+q-j)!p!}{(p+q)!j!(p-j)!} \mathcal{B}^{j},\tag{16}
$$

and the denominator term is written as

$$
D_{pq}(\mathcal{B}) = \sum_{j=0}^{q} \frac{(p+q-j)!q!}{(p+q)!j!(q-j)!}(-\mathcal{B})^{j}.
$$
 (17)

Diagonal Padé approximants can be calculated by taking $q = p$ in Eqs. (16) and (17). The index *p* depends upon the norm of the matrix B. Using a backward-error analysis, p can be calculated. 42 Unfortunately, the Padé approximants are accurate only near the origin. This problem can be overcome by the scaling and squaring technique which exploits,

$$
e^{\mathcal{B}} = \left(e^{\mathcal{B}/2^k} \right)^{2^k}.
$$
 (18)

Firstly, a sufficiently large *k* is chosen such that $\mathcal{B}/2^k$ is close to the zero matrix. Then, a diagonal Pade approximant is ´ used to calculate $exp(B/2^k)$. Finally, the result is squared k times to obtain the approximation to e^{B} . We can use Padé approximation with a scaling and squaring technique to calculate e^{At} in Eq. [\(14\)](#page-2-0). However, the norm of the matrix $A, ||A|| \approx 10^9$, is too large in all three cases of interactions in Eqs. [\(3\)](#page-1-0), [\(4\)](#page-1-0), and [\(5\)](#page-1-0), for $N = 1000$. The scaling and squaring method comes very handy to overcome this large value of the norm.^{[42](#page-11-0)} In this case, the degree of the approximant is $p = 13$. The total magnetic moment of the system of nanoparticles with volume distribution $P(V_i)$ is given by

$$
\bar{M} = \frac{1}{N} \sum_{i=1}^{N} M_i = \int M_i P(V_i) dV_i.
$$
 (19)

The size-distribution plays a significant role in the overall dynamics of the system of nanoparticles governed by Eq. (19). The exponential dependence of τ on the particle size V_i assures that even a weak polydispersity may lead to a broad distribution of relaxation times, which gives rise to an interesting slow dynamics. For a dc measurement, if relaxation time coincides with the measurement time scale τ_m , we can define⁴³ a critical volume V_B as $\mathcal{K}V_B = k_B T_B \ln(\tau_m/\tau_0)$, where T_B is referred as blocking temperature. The critical volume V_B has a strong linear dependence on T_B and weakly logarithmic dependence on the observation time scale τ_m . If the volume of the particle V_i in a polydisperse system is less than V_B , the super spin would have undergone many rotations within the measurement time scale with an average magnetic moment zero. These particles are termed as *superparamagnetic* particles. On the other hand if $V_i > V_B$, the super spins can not completely rotate within the measurement time window and show *blocked or frozen* behavior. However, the particles having volume $V_i \simeq V_B$ are in *dynamically active* regime. The systems of magnetic nanoparticles are in general polydisperse. The shape and size of the particles are not well-known but the particle-size distribution is often found to be lognormal.^{[44](#page-11-0)} We consider the system consisting of lognormally-distributed and the volume V_i of each particle is obtained from a lognormal distribution

$$
P(V_i) = \frac{1}{\sigma V_i \sqrt{2\pi}} exp\left[\frac{-(ln(V_i) - \nu)^2}{2\sigma^2}\right],
$$
 (20)

where $v = ln(\bar{V})$, \bar{V} is the mean size and σ the width of the distribution. The distribution consists of 1000 particles of sizes between $R = 1.3a_0$ and $R = 5.3a_0$, where $a_0(=4.17\text{Å})$ is the lattice parameter of NiO. 39 The total number of particles is deliberately chosen to be small in order to simplify the numerical calculation, as well as to get some insight into the role of size-dependent fluctuations in the magnetization in the present study. In all the cases, we use $\sigma = 0.6$ and $\nu = 3.57$. For the dipolar case, the nanoparticles are arranged in a $10 \times 10 \times 10$ simple-cubic box. We can solve Eqs. [\(14\)](#page-2-0) and (19) for any heating/cooling process using any of the interactions discussed above. A recipe to solve these equations for a zero-field-cooled (ZFC) protocol is as follows. The system is cooled from a very high temperature to the lowest temperature in the absence of any magnetic field. The absence of the field during cooling causes a complete demagnetization of the nanoparticles. This condition is the same as $p_1(0) = 1/2$, or $\mathcal{M}(0) = 0$ in Eq. [\(14\)](#page-2-0). Now, a constant field is applied and the system is heated to a high temperature. At each temperature change, we evolve the system using Eq. [\(14\)](#page-2-0). Suppose we increase the temperature from *T* to $T + dT$. Then, the magnetization at $T + dT$ is given as:

$$
\hat{\mathcal{M}}_{\text{ZFC}}(t, T + dT) = e^{\mathcal{A}(T + dT)t} \hat{\mathcal{M}}_{\text{ZFC}}(0, T + dT)
$$

$$
+ (1 - e^{\mathcal{A}(T + dT)t}) \hat{\nu}(T + dT), \tag{21}
$$

where the initial value of the magnetization vector at $T + dT$ is the same as the magnetization vector at *T* , relaxed for the wait-time t_w . Thus, we can write:

$$
\hat{\mathcal{M}}_{\text{ZFC}}(0, T + dT) = \hat{\mathcal{M}}_{\text{ZFC}}(t_w, T). \tag{22}
$$

Because the elements of the matrix *A* and the vector $\hat{ν}$ depend on *T*, we have shown the functional dependence of \hat{M} on *T*, which varies during the process. In the above expression, t_w is the wait-time at each temperature change. The magnetization is averaged over the volume distribution $P(V_i)$ as:

$$
\bar{M}_{\text{ZFC}} = \int M_{\text{ZFC}}(t, T; V_i) P(V_i) dV_i, \tag{23}
$$

where $M_{\text{ZFC}}(t, T; V_i)$ is the *i*th element of $\mathcal{\hat{M}}_{\text{ZFC}}(t, T - dT)$, given by Eq. (21) . We can define the heating rate of the process as the total time elapsed at each temperature change. Thus, a heating rate of $10^{12} \tau_0$ per temperature-unit corresponds to a heating process in which the system is relaxed for $t = 10^{12} \tau_0$ at each temperature-step.

Henceforth in this paper, volume V_i is used in units of the average volume \bar{V} , which is 193 a_0^3 in our case. Also, the average anisotropic-energy $K\bar{V}$ is taken as the unit of energy. By setting $k_B = 1$, we can use $\mathcal{K}\overline{V}$ as a unit of temperature *T* and the field H . Hereafter, we use a dimensionless quantity $h =$ $\mu_B H/Ka_0^3$ as the unit of the field, e.g., $h = 0.01$ is equivalent to a magnetic field of 300 Gauss.

III. ZFC AND FC MAGNETIZATIONS

We begin our study with the most fundamental protocols, i.e., the study of zero-field-cooled (ZFC) and field-cooled (FC) magnetizations. In the ZFC process, the system is first demagnetized at a very high temperature and then cooled down to a low temperature in a zero magnetic-field. A small magnetic-field is then applied and the magnetization is calculated as a function of the temperature. During this heating process, the evolution of magnetization is given by Eqs. (21) and (22) . On the other hand in the FC protocol, the system is cooled in the presence of a small magnetic-field from higher temperatures to a low temperature. For a very high temperature *T*high, we can take the state of the system to be paramagnetic, which guaranties us to assume $\mathcal{M}_{FC}(0, T_{\text{high}}) = 0$ at $t = 0$. However, if the process would have been started at a low temperature, the state of the system at $t = 0$ will be hard to estimate. For a decrease of the temperature from T to $T - dT$, we can write:

$$
\hat{\mathcal{M}}_{\text{FC}}(t, T - dT) = e^{\mathcal{A}(T - dT)t} \hat{\mathcal{M}}_{\text{FC}}(0, T - dT) \n+ (1 - e^{\mathcal{A}(T - dT)t}) \hat{v}(T - dT), \tag{24}
$$

where

$$
\hat{\mathcal{M}}_{\text{FC}}(0, T - dT) = \hat{\mathcal{M}}_{\text{FC}}(t_w, T). \tag{25}
$$

The magnetization is averaged over the volume distribution $P(V_i)$, given as,

$$
\bar{M}_{\rm FC} = \int M_{\rm FC}(t, T; V_i) P(V_i) dV_i, \qquad (26)
$$

where $M_{\text{FC}}(t,T;V_i)$ is the *i*th element of $\mathcal{\hat{M}}_{\text{FC}}(t,T)$, given by Eq. (24). In the present study, we calculate the ZFC and FC magnetization for various temperature, using long-range and short-range interactions. For all these cases, we use a constant temperature decrease/increase of $\Delta T = 0.004$ for every $10^{12} \tau_0$. In Fig. 2 (a), (b), and (c), we have shown ZFC and FC magnetization against temperature plots for NNSR, LRMF and dipolar interactions. Because the interaction plays a role in decreasing the net magnetization, 40 we have invoked an antiferromagnetic interaction in the case of LRMF and NNSR interactions. We find that on increasing the temperature, the magnetization M_{ZFC} first increases, attains a maximum at a blocking temperature, and then starts decreasing.

In all the cases, we find that as we increase the interaction strength, the peak of the ZFC magnetization shifts toward lower temperatures. In the case of LRMF, we find a dramatic smoothness in the ZFC magnetization as the interaction is increased. The smoothness of the ZFC magnetization indicates a lowering of the effects of size-dependent fluctuations in the magnetization. The reason for the smoothness may be the onset of a collective behavior of the nanoparticles with the increase in the interaction parameter. An increase in the interaction strength is equivalent to bringing nanoparticles closer to each other. As the nanoparticles come closer to each other, their dynamics loses their individuality and the net effect of other nanoparticles leads to a collective behavior. The same behavior can be seen in all the other cases. For the sake of comparison, we have also shown the ZFC magnetization for the non-interacting case as well as for the interacting cases together in Fig. [3.](#page-5-0) We find that in all the cases in Fig. 2, the FC magnetization M_{FC} coincides with M_{ZFC} at higher temperatures, but departs from the ZFC curve at lower temperatures that are well above the blocking temperature. On further lowering the temperature, the FC magnetization tends to a constant value. The blocking temperature shows a substantial dependence on the heating rate. For an infinitelyslow heating-rate, T_B approaches zero and the ZFC curve shows a similar behavior as the FC curve. We also find that for a very weak interaction, the FC magnetization never decreases as the temperature is lowered. But, the increase in the interaction leads to a flatness in the FC magnetization below the blocking temperature which can be seen in all the cases shown in FIG. 2. This flatness in the FC-curves again shows the signature of the cooperative phenomenon of the nanoparticles, arising due to the interactions. We see that the FC and ZFC magnetizations in the noninteracting case are greater than those in the interacting cases. We also

FIG. 2. (Color online) ZFC and FC curves for the (a) NNSR, (b) LRMF and (c) dipolar cases. For each case, there is a constant temperature decrease/increase of $\Delta T = 0.004$ for every $10^{12} \tau_0$ and a field $h = 0.01$.

find that the blocking temperature for the noninteracting case $(T_B = 0.04)$ is much lower than the corresponding interacting one $(T_B = 0.05)$.

IV. AC-SUSCEPTIBILITY

The relaxation phenomena of a magnetic system are often investigated in the presence of an oscillating magnetic-field.^{[7,13,35,45,46](#page-11-0)} It is worthwhile to use an oscillatory magneticfield in our study as the frequency of oscillations of the

FIG. 3. (Color online) A comparison of the ZFC curves for all the interacting cases. Here, again a constant temperature decrease/increase of $\Delta T = 0.004$ for every 10¹2 τ_0 and a field of *h* = 0.01. The values of interaction parameters are $\alpha = 2.5 \times 10^{-6}$, $J^{\text{sr}} = 2.5 \times 10^{-6}$ and $J^{\text{lr}} = 4.5 \times 10^{-8}$ for dipolar, NNSR, and LRMF, respectively.

magnetic field may compare with the inverse of the relaxation time of the nanoparticles. For a time varying magnetic field $h(t) = h_0 e^{-i\omega t}$, the analog of Eq. [\(12\)](#page-2-0) is given by:

$$
\frac{d}{dt}\hat{\mathcal{M}} = \mathcal{A}\hat{\mathcal{M}} - \hat{a}(t),\tag{27}
$$

where $\hat{a}(t) = \hat{a}_0 e^{-i\omega t}$, and ω is the frequency of oscillation of the magnetic field. The above Eq. (27) can be easily solved by multiplying e^{-At} on both sides of the equation. The solution is given by,

$$
\hat{\mathcal{M}} = e^{\mathcal{A}t}\hat{C} - \int_0^t e^{\mathcal{A}(t-s)}\hat{a}(s)ds,\tag{28}
$$

where $\hat{C} = \hat{\mathcal{M}}_0$. Using $\hat{a}(s) = \hat{a}_0 e^{-i\omega s}$ in the above equation, we get,

$$
\hat{\mathcal{M}} = e^{\mathcal{A}t} \hat{\mathcal{M}}_0 - \int_0^t e^{\mathcal{A}(t-s)} \hat{a}_0 e^{-i\omega s} ds.
$$
 (29)

Carrying out the integral in the above equation, we have,

$$
\hat{\mathcal{M}} = e^{\mathcal{A}t} \hat{\mathcal{M}}_0 + (\mathcal{A} + i\omega \mathcal{I})^{-1} \hat{a}_0 e^{-i\omega t} \n- e^{\mathcal{A}t} (\mathcal{A} + i\omega \mathcal{I})^{-1} \hat{a}_0,
$$
\n(30)

where $\mathcal I$ is the identity matrix of order $N \times N$. Equation (30) involves the matrix exponential term e^{At} , which can be evaluated using diagonal Padé approximants with a scaling and squaring technique from MATLAB. This procedure has been discussed in Sec. [II.](#page-1-0) Here, using matrix exponential function *expm*, we can calculate e^{At} in Eq. (30). From Eq. [\(19\)](#page-3-0), we can easily calculate $\overline{M}(t)$ by averaging the above value of the magnetization over a size-distribution. The response of the time-varying field $h(t) = h_0 e^{-i\omega t}$ can be expressed as the sum of an in-phase component, χ' and an out-of-phase component, χ'' , of the susceptibility $\chi(\omega) = \chi' - i\chi''$. The in-phase component with the applied field, χ' , is the lossless component, whereas the out-of-phase component with the applied field, χ'' , is the lossy component. It is obvious that if the change of the external field is very fast as compared to the relaxation time of the particles τ , ($\omega \gg \tau^{-1}$), then the particles cannot follow the field variation; hence, the magnetization gradually decreases to zero as *ω* increases. Magnetization is maximum at the resonance frequency $\omega = \tau^{-1}$ and decreases gradually as ω increases or decreases from the point $\omega =$ τ^{-1} . If the frequency is much less than reorientation of the magnetization (i.e. $\omega \ll \tau^{-1}$, then the magnetization is always in equilibrium over the time-scale of the measurement. The dynamic susceptibility can be defined as:

$$
\chi(t) = M(t)/h(t). \tag{31}
$$

The susceptibility $\chi_{av}(\omega)$ averaged over a time period T can be written as:

$$
\chi_{av}(\omega) \equiv \chi'(\omega) - i\chi''(\omega) = \frac{1}{T} \int_0^T \chi(t)dt.
$$
 (32)

Now using $M(t) = M_R + iM_I$, where M_R and M_I are the real and imaginary parts of the magnetization, we can separate *χ* and χ'' , respectively, as:

$$
\chi'(\omega) = \frac{1}{h_0 T} \int_0^T [M_R \sin(\omega t) + M_I \cos(\omega t)] dt, \qquad (33)
$$

and

$$
\chi''(\omega) = \frac{1}{h_0 T} \int_0^T [M_R \cos(\omega t) - M_I \sin(\omega t)] dt. \quad (34)
$$

Using the above set of equations, we can calculate the real and imaginary parts of the ac-susceptibility. For the ac-susceptibility calculations, the system is demagnetized at the lowest temperature and an ac-magnetic-field of amplitude $h_0 = 0.01$ and frequency $\omega/2\pi$ is applied. The system is then heated upto a high temperature. At each temperature change, we evolve the system for $T = 10^{12} \tau_0$ and calculate *χ'* and *χ* using the above dynamical equations. However, for high frequencies, if we calculate the susceptibility starting from a very high temperature, we find no change in the behavior of the curves as compared to those calculated starting from the lowest temperature. But, for very low frequencies, the two curves calculated from opposite root depart from each other at low temperatures.

It has been experimentally observed that ac-susceptibility measurements are sensitive to the interaction effects 13 and confirmed using Monte Carlo simulations.^{[35](#page-11-0)} We also compare the effect of interactions by considering the noninteracting and interacting cases individually. We find that noninteracting case, which is well described by the individual relaxation of particles differs qualitatively from the interacting case, where the interactions among the particles make the dynamics complex. The real and imaginary parts of ac-susceptibility for the noninteracting case as well as the interacting case using LRMF interactions is shown in Fig. [4.](#page-6-0) For both the cases, the frequency of the ac-field is fixed to $\omega/2\pi = 1/500$. We find that because the interaction, the in-phase part as well as the out-of-phase part show reduction in the net value of the susceptibility. The effect of size-dependent fluctuations can be observed in the in-phase and out-of-phase components of the averaged susceptibility as ripples. It is interesting to see that the ripples are smoothened in the interacting cases. The correlated behavior of nanoparticles is responsible for the averaging of

FIG. 4. (Color online) A comparison of the (a) real, and (b) imaginary components of the ac-susceptibility for the noninteracting and interacting cases for both the cases, the frequency of the ac-field is $ω/2π = 1/500$.

the ripples in this case. The effects of interaction can also be seen as the shifting of the peaks of χ' and χ'' curves toward lower temperatures.

We have also performed a detailed study of the frequency dependence of the real and imaginary parts of the susceptibility. In Figs. $5(a)$ and $5(b)$ we have shown a frequency dependence of the susceptibility for the interacting case by plotting χ' and χ'' versus temperature at various frequencies. We see that the increase in the frequency causes a lowering in the magnitude of χ' and χ'' . This can be easily understood by the competition between the frequency of applied field *ω* and the relaxation time of each particle (τ^{-1}) . By increasing the frequency, the particles which were contributing to χ' and χ'' , will find difficulty to follow the field variation. This lowers the magnitude of χ' and χ'' . Again, we find that the ripples in the susceptibility components due to size-dependent fluctuations get suppressed with an increase in the frequency. We also see that the peak in the χ' and χ'' curves slightly shifts toward higher temperatures as the frequency is increased. This behavior has been earlier reported by experiments.³² We can see that the peak in the ac-susceptibility corresponds to $\omega \approx \tau^{-1}$. Thus, the increase in the frequency causes bigger

FIG. 5. (Color online) (a) Real and (b) imaginary parts of the susceptibility for the interacting case (LRMF) are plotted against the temperature for various frequencies of the applied ac-magnetic-field.

size particles, whose relaxation time is larger, to contribute to the peak of the ac-susceptibility. Hence, peak of the ac-susceptibility shifts toward the higher temperature as the frequency of applied field increases.

V. MEMORY EFFECTS

A. FC memory-effect

Sun *et al.*^{[19](#page-11-0)} have shown a memory-effect phenomenon in the dc-magnetization by a series of measurements on a permalloy $Ni₈₁Fe₁₉$ nanoparticle sample. Later, this memory effect was also been reported by Sasaki *et al.*[23](#page-11-0) and Tsoi *et al.*[24](#page-11-0) for the non-interacting or weakly-interacting superparamagnetic system of ferritin nanoparticles and $Fe₂O₃$ nanoparticles, respectively. All of the studies were concerned only to the ferromagnetic nanoparticles. Experiments on NiO nanoparticles by Bisht and Rajeev 47 also confirm a weak memory effect in these particles. Chakraverty *et al.*[48](#page-12-0) have investigated the effect of polydispersity and interactions among the particles in an assembly of nickel ferrite nanoparticles embedded in a host non-magnetic $SiO₂$ matrix. They found that either tuning the interparticle interaction or tailoring the particle-size distribution in a nano-sized magnetic system leads to important

FIG. 6. (Color online) FC memory-effects for various interacting cases and the noninteracting case with a stop of 10^{14} τ_0 at $T =$ 0*.*04. The uppermost set of curves shows the noninteracting case, while the dipolar and the NNSR cases follow them toward the lower magnetization. The LRMF case is shown at the bottom.

applications in memory devices. In our recent study, 34 we had also shown the memory effects for an assembly of noninteracting NiO antiferromagnetic nanoparticles by the analytical solution of a two-state model. The protocol for the memory effect is as follows. Firstly, we cool the system from a very high temperature to $T_{base} = 0.005$ with a probing field of $h = 0.01$ switched on. The system is again heated from *T*base to get the reference curves, which are shown as "Ref." in Fig. 6. The dynamics of the system in this process is the same as that given by Eqs. (24) , (25) , and (26) during the cooling process, and Eqs. (21) , (22) , and (23) , during the heating process. During these processes, we allow $10^{12} \tau_0$ to ellapse for every temperature step of $\Delta T = 0.001$. We again cool the system from T_H to T_{base} but this time with a stop of 10^{14} τ_0 at $T = 0.04$. The field is cut off during the stop. At the stop, the magnetization is given by,

$$
\hat{\mathcal{M}}_{\text{cool}}(t, T_{\text{stop}}) = e^{\mathcal{A}(T_{\text{stop}})t} \hat{\mathcal{M}}_{\text{cool}}(0, T_{\text{stop}}),
$$
(35)

where

$$
\hat{\mathcal{M}}_{\text{cool}}(0, T_{\text{stop}}) = \hat{\mathcal{M}}_{\text{cool}}(t_w, T_{\text{stop}}),
$$
\n(36)

and

$$
\bar{M}_{\text{cool}} = \int M_{\text{cool}}(t, T; V_i) P(V_i) dV_i. \tag{37}
$$

After the pause, the field is again applied and the system is again cooled upto the base temperature T_{base}. The recovered magnetization can be given as:

$$
\hat{\mathcal{M}}_{\text{cool}}(t, T_{\text{stop}} - dT) = e^{\mathcal{A}(T_{\text{stop}} - dT)t} \hat{\mathcal{M}}_{\text{cool}}(0, T_{\text{stop}} - dT)
$$

$$
+ (1 - e^{\mathcal{A}(T_{\text{stop}} - dT)t}) \hat{\nu}(T_{\text{stop}} - dT),
$$
(38)

where

$$
\hat{\mathcal{M}}_{\text{cool}}(0, T_{\text{stop}} - dT) = \hat{\mathcal{M}}_{\text{cool}}(t_w, T_{\text{stop}}). \tag{39}
$$

The process is shown as the "cooling" curves in Fig. 6. Finally, we heat the system at the same rate as that of cooling without any stop, which is shown as the "heating" curves in Fig. 6. The heating process follows Eqs. (21) , (22) , and (23) . However,

TABLE I. The memory fraction, defined by Eq. (42), has been given for the various interactions.

Interaction	MF
Non-interacting	0.24
Dipolar	0.18
NNSR	0.22
LRMF	0.302

the transition from the cooling process to the heating process at *T*base is given by,

$$
\hat{\mathcal{M}}_{\text{heat}}(t, T_{\text{base}} + dT) = e^{\mathcal{A}(T_{\text{base}} + dT)t} \hat{\mathcal{M}}_{\text{heat}}(0, T_{\text{base}} + dT) + (1 - e^{\mathcal{A}(T_{\text{base}} + dT)t}) \hat{\nu}(T_{\text{base}} + dT),
$$
\n(40)

where

$$
\hat{\mathcal{M}}_{\text{heat}}(0, T_{\text{base}} + dT) = \hat{\mathcal{M}}_{\text{heat}}(t_w, T_{\text{base}}). \tag{41}
$$

As has been discussed earlier, the interactions among the particles play an important role in the slow dynamics, we study the memory effect using the above protocol for various long-range and short-range interactions. Because the memoryeffect phenomenon in the FC magnetization protocol arises because polydispersity in the system of nanoparticles, we find a memory dip in all the cases. Yet, we can find out how strong the memory effect is under various interactions. We compare the relative strength of the memory effects for various interactions by introducing a parameter, the Memory Fraction (MF), at the stop during the memory measurement, defined as:

Memory Fraction (MF) =
$$
\frac{\Delta M}{M^{\text{ref}}} = \frac{M - M^{\text{ref}}}{M^{\text{ref}}}
$$
 (42)

The calculated values of the MF for the no interaction, dipolar, NNSR and LRMF cases are given in Table I. In all the cases, we fix the waiting-time at the stop to be $10^{14} \tau_0$. The trend is a bit surprising. As we know, the memory effect in non-interacting antiferromagnetic nanoparticles is more than that for ferromagnetic particles of the same size-distribution when the distribution consists of smaller sizes. 34 Here, we find that invoking various interactions do change the net memory-dip in antiferromagnetic nanoparticles. As compared to the non-interacting case, we see a reduction in the memory dip for dipolar interactions, an almost similar dip for the NNSR case, and an increased memory dip for the LRMF case. Because we are employing a simple model in the present study, we find the best effects of the interactions in the case of LRMF. We find that as the interaction parameter increases, the effect of the size-dependent fluctuations decreases, and thus, the memory dip decreases in the dipolar interactions case. On further increasing the interaction parameter, a collective dynamics becomes responsible for the enhanced memory-dip in the NNSR and LRMF cases.

In our earlier work, 34 we had analysed the role of polydispersity in the memory effect for the non-interacting case. As there was no interaction, the dynamics of the individual particles was responsible for this peculiar effect. However, in the case of interacting antiferromagnetic nanoparticles, we can see that the size-dependent magnetization-fluctuations and interactions among the nanoparticles do modify the memory dip. This memory effect in an interacting system of nanoparticles can be understood in terms of the droplet picture.[23](#page-11-0) Droplet theory earlier, proposed for spin-glass systems, is very helpful in getting some insight into spin configurations in these interacting systems of nanoparticles. According to this model for a spin-glass, if the system is rapidly quenched in a field *h* to a temperature *T* below the critical temperature, spin-glass domains, or clusters called droplets, which are in local equilibrium with respect to *T* , and *h* grow in size. In this picture, a small temperature-change causes substantial changes in the equilibrium state. At any time *t*, droplets of various sizes exist. Thus, the system is analogous to the non-interacting case where various particle-sizes can be replaced by various clusters grown at temperature *T* and field h . In droplet theory, $49,50$ the dynamics of the droplets is considered to be a thermally-activated process where the energy barrier to form a droplet scales with *L*, the size of the droplet. This droplet picture is relevant to the present study. The thermally-activated dynamics of droplets shows a resemblance to the two-state model of the superparamagnet. We see that because the mean-field interaction with all the other particles, the individual magnetic moment can no longer remains dissociated from the other spins. Hence, an effective moment replaces the individual moment. Because the averaging is done due to the presence of other spins, the net effect on the dynamics is modified according to the interactions between them. As the memory dip ΔM depends strongly on the wait-time parameter and the distribution of sizes the parameter $\Delta M/M$ becomes very effective in the qualitative description of the FC memory-effect.

B. ZFC memory effect

Sasaki et al.^{[23](#page-11-0)} suggested a ZFC memory-effect protocol to confirm whether the observed memory effect is because of glassy behaviour or not. In this method, we first cool the system rapidly in a zero-field from a high temperature to the stop temperature, which is well below the blocking/freezing temperature. The system is left to relax for a wait-time t_w . The rapid cooling is then resumed down to the lowest temperature where a small magnetic field is applied, and the magnetization is calculated during the heating process. A reference ZFCcurve can also be obtained without any stop during the cooling process. If system exhibits memory effect, a memory dip should be observed around the stop temperature during the heating process. Because this dip is substantially smaller, it is better to see the behavior of the difference between the aged and the normal ZFC magnetizations as a function of the temperature. In the present context of study, we have also investigated the ZFC memory-effect. The system, during sudden cooling under a very small field of 0*.*000001 (almost zero-field), is put on hold for a stop of 10^4 s at $T = 0.04$. During heating the system, we find a very small dip at the stop temperature. This very small dip is a drawback of our simplified model. Because we did not incorporate randomness in our model, the simple two-state model could not capture the ZFC memory-dip. In order to demonstrate the memory effect, we perform Monte Carlo simulations^{[51](#page-12-0)} in a $5 \times 5 \times 5$ simulation box of a simple cubic lattice with 125 particles of different sizes. Again, the volume V_i of each particle is obtained from a lognormal distribution as defined in Eq. [\(20\)](#page-3-0). In the Monte Carlo method, during each Monte Carlo step (MCS), we select a particle *i* from 1000 random sequences of 125 particles with either 'up' or 'down' superspin orientations. The attempted flip of the orientation is accepted with a probability of $\exp(-E_i/k_B T)$, where E_i is the energy of the *i*th particle given by Eq. [\(1\)](#page-1-0). A Monte Carlo steps is the time-unit in our simulation. Whenever a flip attempt is successful, the magnetic moment of the particle *i* is updated according to the magnetic-moment versus particle-size plot shown in the inset of Fig. [1.](#page-1-0) One Monte Carlo step is equivalent to the intrinsic time-step τ_0 of the superspins. All the processes discussed below using the Monte Carlo method are under fast cooling and heating rates. Though very fast cooling and heating rates are seemingly far from reality, a qualitative picture of the interacting nanoparticles evolves by this method.

In the present study, we focus on the dipolar-interaction term and take the interaction parameter to be $\alpha = 5 \times 10^{-7}$. The variation of the interaction parameter α has the same effect as that by changing the lattice constant of the lattice on which the particles are placed. We repeat the same protocol for the ZFC memory-effect as discussed above. We first cool the system rapidly in a zero-field from a high temperature $T = 0.6$ to the stop temperature $T_s = 0.1$ (well below the blocking/freezing temperature) and let the system relax for 50000 MCS. The rapid cooling is then resumed down to the lowest temperature where a magnetic field of $h = 0.01$ is applied and the magnetization is calculated during the heating process where the temperature is changed in steps of $\Delta T =$ $0.01K\overline{V}/k_B$ for every 100 MCS. A reference ZFC-curve is also obtained without any stop during the cooling process. We have plotted the difference between the aged and the normal ZFC magnetizations as a function of the temperature for the interacting case with dipolar interactions among the particles in Fig. 7. For the non-interacting case we pointed out no ZFC memory-effect for a noninteracting assembly of NiO nanoparticles; 34 however, for the present case we see

FIG. 7. (Color online) ZFC memory-effect using Monte Carlo simulations. The difference between the aged and the normal (reference) ZFC magnetization is shown. The memory dip around the stop temperature $T_s = 0.1$ can be seen. In the inset, we have shown the aged and the reference ZFC-magnetizations together.

a dip appearing in $\Delta M = M_{\text{ref}} - M$ just at the temperature $T_s = 0.1$, where the stop and wait has been made during the ZFC rapid cooling process. The ZFC memory-effect can be explained by droplet theory.²³ It has been reported earlier that below a critical temperature, droplets (clusters) are growing as the time elapses. This time-dependent growth of the clusters responsible for the unusual memory-effect in the interacting assembly of nanoparticles. As we allow more time to elapse at T_s , the clusters grow proportionally to the wait-time. When the cooling is resumed, this equilibrated clusters freeze at lower temperatures. During the heating process, just near T_s , the frozen clusters rearrange themselves, and a dip can be observed in the ΔM vs. *T* curve.

VI. AGING EFFECTS

The magnetization relaxation of nanoparticles can also be studied by analyzing the wait-time dependence of magnetization. In one such protocol, the system is cooled from a high temperature to the lowest temperature in the presence of a constant magnetic-field. During the cooling process, the system is halted at a temperature lower than the blocking temperature. The system is paused for a wait-time t_w before switching off the magnetic field. The relaxation of magnetization show the t_w dependence. This phenomenon is known as aging effect. Aging effects in polydispersed assemblies of interacting nanoparticles have been been receiving a great deal of attention. The aging effects in interacting nanoparticles has been indicated a glassy behavior in these systems. The aging-effect has been well-studied in the spinglass systems, $52-54$ as well as in interacting assemblies of ferromagnetic nanoparticles.^{9,18,20,23,24} The aging effect or wait-time dependence in the thermoremanant magnetization (TRM) protocol has also been observed in noninteracting assemblies of nanoparticles. Most of the aging effect studies are centered around ferromagnetic nanoparticles. However, recent experiments on antiferromagnetic NiO nanoparticles show aging effect in these systems also. 47 In our earlier work, 34 we had shown the aging effect for a collection of a few noninteracting NiO nanoparticles. We find that in the noninteracting case, the wait-time dependence was due to an ill-defined initial state of the system just before switching off the field. When the interaction among the particles comes into the picture, we should be able to observe the effects arising due to the collective behavior of the nanoparticles. Most of the aging experiments are performed by the measurement of the time-dependent ZFC and TRM magnetizations. The methodology in a TRM protocol is as follows. The system is cooled in a field to a base temperature T_{base} below the blocking temperature T_B . During the cooling process, the magnetization is given by Eqs. (24) , (25) , and (26) . After a waiting-time t_w , the magnetic field is switched off and we find the relaxation in the magnetization. The magnetization in this case is given by,

$$
\hat{\mathcal{M}}_{\text{TRM}}(t, T_{\text{stop}}) = e^{\mathcal{A}(T_{\text{stop}})t} \hat{\mathcal{M}}_{\text{TRM}}(0, T_{\text{stop}}), \tag{43}
$$

where

$$
\hat{\mathcal{M}}_{\text{TRM}}(0, T_{\text{stop}}) = \hat{\mathcal{M}}_{\text{TRM}}(t_{\text{w}}, T_{\text{stop}}),
$$
\n(44)

and

$$
\bar{M}_{\text{TRM}} = \int M_{\text{TRM}}(t, T; V_i) P(V_i) dV_i. \tag{45}
$$

When the TRM magnetization is plotted against the time, it is found that the relaxation in the magnetization depends upon the waiting-time *t*w.

In this section we will study the aging effect for a polydispersed system of interacting NiO nanoparticles in the TRM protocol. Our investigation is carried out by cooling the system in the presence of magnetic field of $h = 0.01$ upto the base temperature $T_{base} = 0.024$ and then cutting the field off after a wait-time t_w to let the system relax. In a previous work, 34 we had discussed the aging effects for various size-distributions and made a comparative study with the ferromagnetic particles case. However, in this section, we perform a comparative study of the various interactions in NiO nanoparticles, as discussed in Sec. [II.](#page-1-0) In Fig. 8, we plot the thermoremanant magnetization against the time on a logarithmic scale for the noninteracting as well as the interacting cases: dipolar, NNSR and LRMF. Before switching off the field, we wait for $t_w = 10^{12} \tau_0$. We see that the decay in the magnetization is fastest in the non-interacting case, which shows a two-step relaxation.^{[34](#page-11-0)} In the non-interacting case, the dynamics of an individual particle is not correlated with that of others in its surroundings. The role of size-dependent fluctuations can be seen as ripples in the relaxation of the magnetization, but in the interacting cases, we see a smoother curve which shows the averaging of the fluctuations in the magnetization because the mean field arising from the neighboring particles. Because the interactions the magnetization persists for a longer time as compared to the noninteracting case. Hence, we can say that due to the interactions, the dynamics of the nanoparticles gets slower. We have also plotted the magnetization versus logarithmic-time for various wait-times $t_w = 10^{12} \tau_0$, $10^{13} \tau_0$, and 10^{14} τ_0 in Fig. [9.](#page-10-0) We see a wait-time dependence in all the cases, though weak. In the inset of these curves, we have shown the corresponding relaxation-rate for each case, defined as $-\frac{\partial M}{\partial \log(t/\tau_0)}$. Multiple peaks in all the cases reflect the distinction between the antiferromagnetic nanoparticles and

FIG. 8. (Color online) A comparison of the aging effect for four cases: No interaction, Dipolar, NNSR, and LRMF. In each case, the wait-time is $t_w = 10^{12} \tau_0$.

FIG. 9. (Color online) The aging effect for waiting-times $t_w = 10^{12} \tau_0$, $10^{13} \tau_0$, and $10^{14} \tau_0$ has been shown for: (a) no interaction, (b) dipolar, (c) NNSR, and d) LRMF interactions. In the interacting cases (b), (c) and (d), we see a smooth decay of the magnetization with time. In the inset of each figure, the relaxation rate is plotted against $log(t/\tau_0)$.

the ferromagnetic nanoparticles: An effect of size-dependent magnetization-fluctuations. In a nutshell, we conclude that the role of size-dependent fluctuations in the magnetization dynamics has been controlled a little bit due to the interactions. The decay of the thermoremanant magnetization gets slower as the interaction is increased.

VII. CONCLUSIONS

We have studied the slow dynamics of an interacting assembly of antiferromangetic nanoparticles by solving a two-state model analytically. A collection of a few interacting antiferromagnetic nanoparticles has been comparatively studied using various long-range and short-range interactions. We find that the ZFC magnetization shows ripples in all the cases, which starts decreasing as the interaction parameter increases. Because of the interactions, the dynamics of the system is governed not only by a broad distribution of particle relaxation-times arising from the polydispersity, but also by a collective behavior amongst the particles themselves. We have also studied the dynamics of the system using an ac magnetic-field. We find that the real and imaginary components of the ac-susceptibility show ripples, which are due to the size-dependent magnetization-fluctuations. The effects of these size-dependent fluctuations are displayed in the non-interacting as well as interacting cases; however, we find that the ripples are smoothened in the interacting cases. Because of the interaction, we also find a shifting of the peaks of the χ' and χ'' curves toward lower temperatures. We have also calculated the frequency dependence of χ' and χ'' , and found that the frequency causes a lowering in the magnitudes of *χ*' and *χ*". The frequency dependence of *χ*' and *χ*" shows that the particles become less responsive to the magnetic field as the frequency is increased. We have shown the memory-effect in the field cooled magnetization protocol for an interacting polydispersed assembly of nanoparticles. We have studied the effects of interactions on the memory dip by varying the interaction parameter. As compared with noninteracting case, we find a reduced memory-dip for dipolar interactions, an almost same memory-dip for NNSR interactions, and an increased memory-dip for LRMF interactions. Because of the limitations in our model, we find the best effect of the interactions only in the case of LRMF. We find that as the interaction parameter increases, the effect of the size-dependent fluctuations decreases, and thus, the memory dip decreases in the dipolar interactions cases. On further increasing the interaction parameter, a collective dynamic becomes responsible for the enhanced memory dip in the NNSR case and LRMF cases. The memory effect and a weak aging effect have also been discussed for all the cases. For the noninteracting case, the size-dependent fluctuations play a role in the obtaining of a high memory fraction. As the interactions are incorporated, the memory fraction decreases, but for the strong interaction cases, the memory fraction increases. As the present model could not incorporate randomness and disorder, we have performed a Monte Carlo study considering the dipolar interactions among the particles. We find a memory dip in the ZFC magnetization, which indicates the importance of the interactions among the particles. The ZFC memoryeffect is attributed only to the collective dynamics of the nanoparticles. The system of interacting nanoparticles under various interactions also shows a wait-time dependence. The decay of thermoremanant magnetization slows down as the interaction among the nanoparticles increases. We have done a comparative study of the aging effects in the no interaction, dipolar, NNSR, and LRMF cases.

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