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Imaging charge density fluctuations in graphene using Coulomb blockade spectroscopy

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Using scanning tunneling microscopy (STM), we have imaged local charge density fluctuations in monolayer graphene. By placing a small gold nanoparticle on the end of the STM tip, a charge sensor is created. By raster scanning the tip over the surface and using Coulomb blockade spectroscopy, we map the local potential on the graphene. We observe a series of electron and hole doped puddles with a characteristic length scale of ∼20 nm. Theoretical calculations for the correlation length of the puddles based on the number of impurities are in agreement with our measurements.

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Graphene is the two-dimensional form of carbon consisting of a single layer of carbon atoms arranged in a hexagonal lattice. This simple geometry is a repository of exceptional electronic, mechanical, and chemical properties. These properties emerge from the linear band structure of graphene. The charge carriers in graphene behave as massless Dirac fermions and exhibit very high mobility.^{[1,2](#page-4-0)} This makes graphene very promising for device applications.[3](#page-4-0) Efforts in this direction have yielded spectacular results from using graphene as a nanopore template for $DNA⁴$ $DNA⁴$ $DNA⁴$ to producing 30-in. monolayer graphene films on a copper substrate, detaching them, and fabricating transparent electrodes.^{[5](#page-4-0)} However, one of the challenges in order to make graphene devices is controlling the electrostatic environment. In particular, it has been shown that graphene on silicon dioxide $(SiO₂)$ tends to form a series of electron and hole puddles near the Dirac point. $6-8$ These puddles reduce the mobility of the graphene. One of the major sources of the puddles is the presence of random charged impurities on the graphene layer and in between the graphene and $SiO₂$. Recent calculations have shown that the puddles may also be induced by strain due to the rippling of the graphene sheet. 9 Hence a complete understanding of puddles is vital for progress in graphene applications.

In this paper, we use a nanoparticle on the end of a scanning tunneling microscope (STM) tip as a sensitive charge sensor to spatially map variations in the electron charge density. Our technique allows a direct quantitative measurement of the charge density fluctuations with a spatial resolution of several nanometers, leading to an estimation of the number of impurities. Previous studies of electron and hole puddles on graphene lacked the spatial and/or energy resolution to fully resolve the puddles and obtain information about their characteristic distribution. Charge density variations on a length scale of ∼150 nm have been measured with a single electron transistor (SET) .⁶ The size of the SET and distance from the graphene limited the spatial resolution of these measurements. Previous STM measurements of the electron and hole puddles lacked the energy resolution of our current technique because they relied on a shift in the local density of states.^{7,8} In these measurements, we use the sharp peaks in *dI/dV* measurements due to Coulomb blockade to significantly improve the energy resolution.

Graphene was first isolated by exfoliating it from threedimensional (3D) graphite and attaching it onto a $SiO₂$ substrate.^{[10](#page-4-0)} There are many alternative ways to make graphene, such as epitaxy on silicon carbide substrates 11 or chemical vapor deposition growth on copper¹² and nickel.^{[13](#page-4-0)} However, the mechanical exfoliation technique still provides the highestquality devices. Here, we have studied exfoliated graphene on a SiO2 substrate using a STM operating in ultrahigh vacuum at 4.6 K. Previous STM studies on exfoliated graphene have been instrumental in correlating its structure with the electronic properties[.7,8,14](#page-4-0) In this paper, we report density fluctuations in graphene using Coulomb blockade spectroscopy measurements. Coulomb blockade is the suppression of electron tunneling when the charging energy $e^2/2C$, where *C* is the capacitance of the nanoparticle, is unavailable to the electrons at low temperature and voltages, i.e., $k_B T, eV \ll e^2/2C$.^{[15](#page-4-0)} With a nanoparticle at the end of the STM tip, the system exhibits the two junctions which are necessary to produce Coulomb blockade. The tunneling gap between the nanoparticle and the surface acts as one junction, while the barrier between the tip and nanoparticle is the other. The high spatial resolution of the STM and the low temperature allows a spatially resolved measurement of Coulomb blockade. As Coulomb blockade is sensitive to not only changes in capacitance but also the electrostatic environment of the nanoparticle, we are able to detect charge fluctuations on graphene using Coulomb blockade spectroscopy.

Graphene flakes were exfoliated from graphite on a Si substrate with 300 nm of thermally grown oxide. Ti*/*Au electrodes were then deposited on the graphene through a shadow mask.^{[16](#page-4-0)} This technique does not require the use of poly(methyl methacrylate) (PMMA) as a resist. Therefore it yields clean graphene devices that need no additional cleaning procedures prior to imaging with STM. The graphene sample is transferred to the ultrahigh vacuum ($p \le 10^{-11}$ mbar) STM and cooled to 4.6 K. Electrochemically etched tungsten tips were used for imaging and spectroscopy. The following procedure was used to attach a nanoparticle to the tip. First the tip was cleaned by a series of voltage pulses on a Au surface. Typical pulses were 8 V for 10 ms. After this cleaning procedure, the density of states of the tip was checked by performing spectroscopy on the Au surface. If a flat density of states was observed, then a nanoparticle would be attached.

FIG. 1. (Color online) (a) Schematic of the graphene sample with gold contacts on a $SiO₂$ substrate. The nanoparticle at the tip apex is shown in the closeup sketch. (b) Optical microscope image of the monolayer graphene flake and gold electrodes. (c) Large-scale STM image of the graphene surface. ($V_t = -0.5$ V, $I_t = 100$ pA, 100 nm \times 100 nm). The scale bar is 20 nm. (d) STM image showing the atomic resolution of the hexagonal lattice of monolayer graphene $(V_t = -0.5 \text{ V}, I_t = 100 \text{ pA}, 7 \text{ nm} \times 7 \text{ nm})$. The scale bar is 1.4 nm.

This was done by scanning the Au surface which had many small Au clusters on it. After scanning for a few minutes with $V_t = -0.5$ V and $I_t = 100$ pA, the resolution of the surface was observed to increase. At this point a Au nanoparticle would be attached to the tip apex. This was confirmed by the sharp peaks in the density of states discussed below.

Figure $1(a)$ shows a schematic diagram of the measurement setup. The bias voltage is applied to the STM tip and the sample is grounded. The particle on the tip apex is a Au nanoparticle attached after the tip conditioning on the Au surface. Because of the cleanliness of the samples, we are able to obtain largearea images of the graphene surface which are free of defects or residue from sample processing. Figure $1(b)$ shows the optical microscope image of the monolayer graphene flake with gold contacts on the $SiO₂$ substrate. Figure 1(c) shows a typical $100 \text{ nm} \times 100 \text{ nm}$ area of the graphene surface. The sample is smooth, with a slowly varying height presumably due to the underlying $SiO₂$ substrate.^{[17](#page-4-0)} There may also be some intrinsic corrugations remaining in the graphene.^{[18](#page-4-0)} There are no sharp features that arise from the presence of contaminants on the surface or defects. Atomic resolution topography of the sample is shown in Fig. $1(d)$ showing the hexagonal lattice of carbon atoms as expected for monolayer graphene along with height variation due to the uneven SiO_2 substrate.^{[7,8,14,19,20](#page-4-0)}

We have measured the local density of states (LDOS) as a function of energy. For this measurement, the tip is held at a fixed location and the height is stabilized with a tunneling current of $I_t = 100 \text{ pA}$ and tip voltage $V_t = -0.5 \text{ V}$. Then the feedback loop is turned off and the voltage between the tip and substrate is ramped. The differential conductance as a function of voltage is acquired. An ac modulation voltage of 5 mV rms at 570 Hz is applied to the tip and the resulting current is detected with a lock-in amplifier. Figure [2\(a\)](#page-2-0) shows a typical *dI/dV* curve acquired with a nanoparticle at the end of the tip. There are a series of sharp peaks that are equally spaced in energy. This is in contrast to a measurement on graphene without the nanoparticle, which shows a smoothly varying density of states. $7,14$ The peaks arise due to the addition of individual electrons to the nanoparticle. The set point current and voltage sets the overall resistance between the tip and graphene. Using a small set point current ensures that the resistance between the nanoparticle and graphene, R_{sub} , is always much greater than the resistance between the nanoparticle and tip, R_{tip} . Because of this strong asymmetry in the tunnel barriers, the barrier between the nanoparticle and graphene is always the ratelimiting barrier. Therefore, we obtain a Coulomb staircase with the spacing of the peaks determined by the capacitance between the nanoparticle and graphene. 21

We have measured the dI/dV curves for different back gate voltages on the monolayer graphene sample. Once again the curves are characterized by sharp peaks which vary as a function of gate voltage. Figure [2\(b\)](#page-2-0) plots the *dI/dV* curves as a function of gate voltage and tip voltage from a different location on the same sample as Fig. $2(a)$. It shows the characteristics of a Coulomb blockade diamond plot, providing further evidence of the presence of a small nanoparticle on the tip. In the region around zero tip voltage, there are bright red regions showing that there is no flow of current, indicating Coulomb blockade. As the tip voltage increases a series of peaks appear. Between each peak the number of electrons on the nanoparticle changes by one. The gate acts to induce charge on the nanoparticle and therefore the peaks change in energy. Only one set of peaks is strong, and this is due to the asymmetry of the tunnel barriers. This strong peak corresponds to lining up the energy of a state on the nanoparticle with the energy of the graphene. Therefore, as the voltage between the tip and graphene is increased, we see a series of peaks separated by e/C_{sub} , where C_{sub} is the capacitance between the nanoparticle and graphene. The strong peaks slope toward decreasing tip voltage as the gate voltage is increased. This is the opposite direction expected for a normal gate electrode which would tend to induce negative charges as the gate voltage increases. However, the effect of the gate on the nanoparticle is screened by the graphene. A positive voltage on the gate induces electrons on the graphene which then tend to shift the energy levels on the nanoparticle. Effectively, the graphene layer acts to reverse the sign of the gate voltage.

As the tip is moved to different positions of the graphene, the energies where peaks occur shift. This is seen in Fig. [2\(c\),](#page-2-0) which shows the spectroscopy results from two locations on the graphene surface separated by a distance of 10 nm. The entire *dI/dV* curve has been shifted along the tip voltage axis. All of the peaks still have the same spacing, showing that the capacitance between the nanoparticle and graphene remains unchanged. However, the energy at which the peaks occur has changed. This is due to the presence of an offset charge on the nanoparticle. Below we will show that this offset arises due to variations in the potential on the surface of the

FIG. 2. (Color online) (a) dI/dV point spectroscopy showing peaks due to the addition of individual electrons to the nanoparticle. (b) *dI/dV* spectroscopy as a function of tip voltage and gate voltage, showing a series of Coulomb blockade diamonds. This demonstrates that the nanoparticle on the end of the tip acts as a quantum dot. This data set was taken at a different location from the curve shown in (a). (c) *dI/dV* curves taken at two different points on the graphene surface. Each shows a series of peaks but they are horizontally offset due to the different potential on the graphene surface. (d) Distribution of peak spacings for a 100 nm \times 100 nm area of the graphene film. The peaks are clustered around the same value, showing that there is a single quantum dot located on the end of the tip. Set point parameters are $V_t = -0.5$ V, $I_t = 100 \text{ pA for (a)–(c), and } V_g = 0 \text{ V for (a), (c), and (d).}$

graphene. Figure $2(d)$ shows the distribution of peak spacings over a 100 nm \times 100 nm area. As the peak spacings cluster around a single value, this shows that we are using the same nanoparticle for all of our measurements. On the other hand, the distribution of the energies of the peaks over the same area shows a uniform distribution. When a new nanoparticle is attached to the tip, the peak spacing changes but otherwise the results are unchanged. We have observed peak spacings ranging from 0.150 to 0.250 V. This gives values of C_{sub} of 1.1– 0.6 aF. Assuming, that the nanoparticle is spherically shaped, we can calculate its size using the formula for the capacitance of a sphere above a plane:

$$
C \approx 2\pi\epsilon r \ln\left[\frac{r+h}{h}\right],
$$

where r is the radius of the sphere and h is the separation of the sphere from the plane.^{[22](#page-4-0)} This formula is valid when $r \gg h$. Using a value of $h = 0.5$ nm, we find the radius of the nanoparticles to range from 5 to 7 nm.

Figure [3](#page-3-0) shows the local density of states over a 100 nm \times 100 nm area of the sample. The set point parameters are $V_t = -0.5$ V and $I_t = 100$ pA. Six images from a movie of the density of states as a function of tip voltage are shown. 23 The bright regions represent the locations where electrons are added to the nanoparticle. They are peaks in the *dI/dV* curve. Following one of the bright lines corresponds to moving along a path of constant induced charge on the nanoparticle. As the voltage between the tip and graphene is changed, the location where electrons are added to the nanoparticle also changes. Figures $3(a)$ – $3(f)$ shows the evolution of the rings as the tip voltage is changed. Notice that the circled ring grows as the voltage goes from 0.47 V to 0.37 V. When the tip voltage has reached 0.30 V [Fig. $3(f)$], an additional electron has been added to the nanoparticle and the image is equivalent to Fig. $3(a)$ at 0.47 V. The spatial variation in the density of

FIG. 3. (Color online) Local density of states maps taken at a series of tip voltages as indicated for each map. Each map is 100 nm \times 100 nm. Set point parameters are $V_t = -0.5$ V, $I_t = 100$ pA, $V_g =$ 0 V. A ring is circled to show how it changes with energy. They start out circular (a), (b) and slowly become asymmetric (c)–(e). Then a second ring forms (f) at the same location once the charge on the nanoparticle changes by one.

states map implies that the electrostatic environment of the nanoparticle is changing as a function of position. Therefore, these maps give information about the local potential or charge density fluctuations in graphene as discussed below.

To obtain a map of the local potential or equivalently the charge density, we can analyze the energy at which the peaks occur as a function of position. Since the peaks are arising due to Coulomb blockade, we expect that the spacing in energy between peaks will always remain constant at e/C_{sub} . This is the reason that Figs. $3(a)$ and $3(f)$ look identical; they are separated in energy by e/C_{sub} . On the other hand, the energy at which the peaks occur gives information about the amount of charge which must be induced on the nanoparticle in order for it to conduct. If the offset charge on the nanoparticle is $e/2$, then we expect a series of peaks at ne/C_{sub} , where *n* is the number of electrons induced on the nanoparticle. As the offset charge changes, the energies of the peaks will shift such that they occur at $(\Delta Q + ne)/C_{sub}$, where ΔQ is the offset charge relative to *e/*2. In both cases, the peaks are spaced by e/C_{sub} . We have used the data in Fig. 3 to measure the offset charge at each point on the graphene surface. This offset charge can be converted to a potential difference ΔV by dividing by C_{sub} . The results are shown in Fig. 4. There are regions of positively and negatively charged puddles on the surface of the graphene. This map shows a typical size scale of electron and hole puddles of 20 nm. The regions of positive potential correspond to areas of the graphene flake where the rings in Fig. 3 are shrinking as the tip voltage is decreased. Likewise, rings that grow with decreasing tip voltage, such as the ring in the pink ellipse in Fig. 3, are areas of negative potential on the graphene.

The potential difference can be used to estimate a local electron density using the band structure of graphene. Using the linear dispersion relation of graphene, we find that fluctuations in charge density are related to fluctuations in potential

FIG. 4. (Color online) Potential fluctuations in the same 100 nm \times 100 nm region of the flake as in Fig. 3. Red areas have positive potential while blue areas are negative.

by $\Delta n = (\Delta V)^2 / \pi (\hbar v_F)^2$, where $v_F = 1.1 \times 10^6$ m/s is the Fermi velocity. We find that the full width at half maximum (FWHM) of our distribution of charge densities is $\Delta n =$ 3×10^{11} cm⁻². This is approximately a factor of 6 higher than found in previous SET measurements of electron and hole puddles on graphene.⁶ However, our significantly improved spatial resolution of 1–2 nm leads to the higher variation because we are now sensitive to charge fluctuations on the nanometer scale as opposed to the previous measurements which were limited to ∼150 nm resolution. With the lower spatial resolution, the charge fluctuations tend to be averaged away. Our value for the charge fluctuation is similar to previous STM measurements of electron and hole puddles, δ however, this technique has the ability to resolve much smaller changes in charge density because of the high-energy sensitivity of the Coulomb blockade spectroscopy, 5 meV, which is improved by a factor of 10 as compared to 50 meV reported in the SET measurements.

By performing an autocorrelation of the image in Fig. 4, we can find the correlation length of the puddles. The results are shown in Fig. $5(a)$. It shows a FWHM of the distribution of 20 nm. Theoretical calculations have shown that scattering from random charged impurities causes electron and hole puddles and their correlation length varies with the concentration of these impurities. 24 We can estimate

FIG. 5. (a) Correlation length for the puddles shown in Fig. 4. (b) Resistivity as a function of gate voltage for the graphene flake measured at 4.6 K.

the number of impurities in our sample from the electrical conductivity at high gate voltages. $2⁵$ The conductivity is given by $\sigma = 20e\epsilon V_g/(hn_i t)$, where n_i is the impurity density, ϵ is the dielectric constant of $SiO₂$, *t* is the oxide thickness, and V_g is the gate voltage. Figure $5(b)$ shows the resistivity as a function of gate voltage at 4.6 K. From this plot, we can estimate the density of impurities as $n_i \approx 1 \times 10^{11}$ cm⁻². This impurity concentration gives a theoretical correlation length for the puddles of 12 nm from Fig. $2(b)$ in Ref. 24 where the calculations are carried out to get the ground-state carrier density using Thomas-Fermi-Dirac (TF) theory.²⁴ The global nature of the transport measurement compared to our local measure of the charge density fluctuations may explain the small discrepancy between the theoretical puddle size and the size determined from our images.

Thus we have shown that a nanoparticle at the STM tip apex can act as a sensitive charge sensor to detect the charge density fluctuations in graphene. Using spatially resolved spectroscopy measurements we have mapped these

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- ¹K. S. Novoselov, A. K. Geim, S. V. Morozov, D. Jiang, M. I. Katsnelson, I. V. Grigorieva, S. V. Dubonos, and A. A. Firsov, [Nature \(London\)](http://dx.doi.org/10.1038/nature04233) **438**, 197 (2005).
- 2A. H. Castro Neto , F. Guinea, N. M. R. Peres, K. S. Novoselov, and A. K. Geim, [Rev. Mod. Phys.](http://dx.doi.org/10.1103/RevModPhys.81.109) **81**, 109 (2009).
- 3A. K. Geim, Science **324**[, 1530 \(2009\).](http://dx.doi.org/10.1126/science.1158877)
- 4G. F. Schneider, S. W. Kowalczyk, V. E. Calado, G. Pandraud, H. W. Zandbergen, L. M. K. Vandersypen, and C. Dekker, [Nano](http://dx.doi.org/10.1021/nl102069z) Lett. **10**[, 3163 \(2010\).](http://dx.doi.org/10.1021/nl102069z)
- 5S. Bae, H. Kim, Y. Lee, X. Xu, J. Park, Y. Zheng, J. Balakrishnan, T. Lei, H. Kim, Y. Song, Y. Kim, K. Kim, B. Ozyilmaz, J. Ahn, B. Hong, and S. Iijima, [Nat. Nanotechnol.](http://dx.doi.org/10.1038/nnano.2010.132) **5**[, 574 \(2010\).](http://dx.doi.org/10.1038/nnano.2010.132)
- 6J. Martin, N. Akerman, G. Ulbricht, T. Lohmann, J. H. Smet, K. von Klitzing, and A. Yacoby, [Nat. Phys.](http://dx.doi.org/10.1038/nphys781) **4**, 144 [\(2008\).](http://dx.doi.org/10.1038/nphys781)
- ⁷A. Deshpande, W. Bao, F. Miao, C. N. Lau, and B. J. LeRoy, *[Phys.](http://dx.doi.org/10.1103/PhysRevB.79.205411)* Rev. B **79**[, 205411 \(2009\).](http://dx.doi.org/10.1103/PhysRevB.79.205411)
- 8Y. Zhang, V. W. Brar, C. Girit, A. Zettl, and M. F. Crommie, [Nat.](http://dx.doi.org/10.1038/nphys1365) Phys. **5**[, 722 \(2009\).](http://dx.doi.org/10.1038/nphys1365)
- 9M. Gibertini, A. Tomadin, M. Polini, A. Fasolino, and M. I. Katsnelson, Phys. Rev. B **81**[, 125437 \(2010\).](http://dx.doi.org/10.1103/PhysRevB.81.125437)
- 10K. S. Novoselov, A. K. Geim, S. V. Morozov, D. Jiang, Y. Zhang, S. V. Dubonos, I. V. Grigorieva, and A. A. Firsov, [Science](http://dx.doi.org/10.1126/science.1102896) **306**, 666 [\(2004\).](http://dx.doi.org/10.1126/science.1102896)
- 11C. Berger, Z. Song, X. Li, X. Wu, N. Brown, C. Naud, D. Mayou, T. Li, J. Hass, A. N. Marchenkov, E. H. Conrad, P. N. First, and W. A. de Heer, Science **312**[, 1191 \(2006\).](http://dx.doi.org/10.1126/science.1125925)

fluctuations and found a characteristic puddle size of 20 nm. A comparison with theoretical calculations shows that our graphene samples have an impurity density on the order of 1011 cm−2, which gives rise to these puddles. In order for graphene devices to reach their full potential, the number of impurities must be significantly reduced. This technique of using a gold nanoparticle at the tip apex for spatially resolved measurements will enable us to detect the reduction in the number of impurities and consequently test the quality of the flakes for device fabrication.

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- $12X$. Li, W. Cai, J. An, S. Kim, J. Nah, D. Yang, R. Piner, A. Velamakanni, I. Jung, E. Tutuc, S. K. Banerjee, L. Colombo, and R. Ruoff, Science **324**[, 1312 \(2009\).](http://dx.doi.org/10.1126/science.1171245)
- 13K. Kim, Y. Zhao, H. Jang, S. Lee, J. Kim, K. Kim, J. Ahn, P. Kim, J. Choi, and B. Hong, [Nature \(London\)](http://dx.doi.org/10.1038/nature07719) **457**, 706 (2009).
- 14Y. Zhang, V. W. Brar, F. Wang, C. Girit, Y. Yayon, M. Panlasigui, A. Zettl, and M. F. Crommie, Nat. Phys. **4**[, 627 \(2008\).](http://dx.doi.org/10.1038/nphys1022)
- 15M. H. Devoret, D. Esteve, H. Grabert, G. L. Ingold, H. Pothier, and C. Urbina, [Phys. Rev. Lett.](http://dx.doi.org/10.1103/PhysRevLett.64.1824) **64**, 1824 (1990).
- 16W. Bao, G. Liu, Z. Zhao, H. Zhang, D. Yan, A. Deshpande, B. J. LeRoy, and C. N. Lau, Nano Res. **3**[, 98 \(2010\).](http://dx.doi.org/10.1007/s12274-010-1013-5)
- 17W. G. Cullen, M. Yamamoto, K. M. Burson, J. H. Chen, C. Jang, L. Li, M. S. Fuhrer, and E. D. Williams, [Phys. Rev. Lett.](http://dx.doi.org/10.1103/PhysRevLett.105.215504) **105**, [215504 \(2010\).](http://dx.doi.org/10.1103/PhysRevLett.105.215504)
- ¹⁸V. Geringer, M. Liebmann, T. Echtermeyer, S. Runte, M. Schmidt, R. Ruckamp, M. C. Lemme, and M. Morgenstern, [Phys. Rev. Lett.](http://dx.doi.org/10.1103/PhysRevLett.102.076102) **102**[, 076102 \(2009\).](http://dx.doi.org/10.1103/PhysRevLett.102.076102)
- 19M. Ishigami, J. H. Chen, W. G. Cullen, M. S. Fuhrer, and E. D. Williams, Nano Lett. **7**[, 1643 \(2007\).](http://dx.doi.org/10.1021/nl070613a)
- 20E. Stolyarova, K. T. Rim, S. Ryu, J. Maultzsch, P. Kim, L. E. Brus, T. F. Heinz, M. S. Hybertsen, and G. W. Flynn, [Proc. Natl. Acad.](http://dx.doi.org/10.1073/pnas.0703337104) Sci. USA **104**[, 9209 \(2007\).](http://dx.doi.org/10.1073/pnas.0703337104)
- 21A. E. Hanna and M. Tinkham, Phys. Rev. B **44**[, 5919 \(1991\).](http://dx.doi.org/10.1103/PhysRevB.44.5919)
- 22C. Wasshuber, *Computation Single Electronics* (Springer, Berlin, 2001).
- 23See supplemental material at [\[http://link.aps.org/supplemental/](http://link.aps.org/supplemental/10.1103/PhysRevB.83.155409) [10.1103/PhysRevB.83.155409\]](http://link.aps.org/supplemental/10.1103/PhysRevB.83.155409) for the STM movies for Fig. 3.
- 24E. Rossi and S. Das Sarma, [Phys. Rev. Lett.](http://dx.doi.org/10.1103/PhysRevLett.101.166803) **101**, 166803 [\(2008\).](http://dx.doi.org/10.1103/PhysRevLett.101.166803)
- 25S. Adam, E. H. Hwang, V. M. Galitski, and S. Das Sarma , [Proc.](http://dx.doi.org/10.1073/pnas.0704772104) [Natl. Acad. Sci. USA](http://dx.doi.org/10.1073/pnas.0704772104) **104**, 18392 (2007).