

Hydrostatic pressure study of the paramagnetic-ferromagnetic phase transition in (Ga,Mn)As

M. Gryglas-Borysiewicz,¹ A. Kwiatkowski,¹ M. Baj,¹ D. Wasik,¹ J. Przybytek,¹ and J. Sadowski^{2,3}

¹*Institute of Experimental Physics, Faculty of Physics, University of Warsaw, Hoża 69, 00-681 Warsaw, Poland*

²*Institute of Physics, PAS, Al. Lotników 32/46, 02-668 Warsaw, Poland*

³*MAX-Lab, Lund University, 221 00 Lund, Sweden*

(Received 6 April 2010; revised manuscript received 30 August 2010; published 21 October 2010)

The effect of hydrostatic pressure on the paramagnetic-ferromagnetic phase transition has been studied in (Ga,Mn)As. The variation in the Curie temperature (T_C) with pressure was monitored by two transport methods: (1) measurement of zero-field resistivity versus temperature $\rho(T)$ and (2) dependence on temperature of the Hall voltage hysteresis loop. Two specimens of different resistivity characteristics were examined. The measured pressure-induced changes in T_C were relatively small (on the order of 1 K/GPa) for both samples, however they were opposite for the two.

DOI: [10.1103/PhysRevB.82.153204](https://doi.org/10.1103/PhysRevB.82.153204)

PACS number(s): 75.50.Pp, 75.47.-m, 75.60.-d, 75.40.-s

(Ga,Mn)As is one of the most intensively investigated diluted magnetic semiconductors during last decades. The understanding of physical phenomena governing its magnetic properties is crucial for increasing Curie temperature (T_C) and thus for possible application of this material in spintronic devices. The origin of ferromagnetism in (Ga,Mn)As was quantitatively explained within the p - d Zener model assuming magnetic interaction between the localized magnetic moments of Mn^{2+} ions mediated by holes in the valence band.¹⁻³ This model, in the case of semiconductors, where the carrier density is smaller than the magnetic ion concentration is equivalent to the Ruderman-Kittel-Kasuya-Yosida (RKKY) approach employed in the diluted magnetic metals.⁴ Within this picture the ferromagnetic ordering temperature, T_C depends, in particular, on the local p - d exchange interaction and a free hole concentration. It was demonstrated that indeed an increase in the hole concentration in a field effect transistor structure led to an enhancement of the ferromagnetic state.⁵ On the other hand it was found that the exchange energy scales with the lattice constant as,¹ $N_0\beta \sim a_0^{-3}$, and therefore an external hydrostatic pressure could influence the exchange coupling. Although the studies of (In,Mn)Sb diluted magnetic semiconductor under hydrostatic pressure provided an evidence for an increase in carrier-mediated magnetic coupling,^{6,7} giving rise to higher Curie temperature, the effect of hydrostatic pressure on (Ga,Mn)As semiconductor is not as clear.⁶ Therefore additional study was performed in order to clarify the role of external hydrostatic pressure in the paramagnetic-ferromagnetic phase transition in (Ga,Mn)As. p -type $Ga_{1-x}Mn_xAs$ layers were grown by molecular-beam epitaxy (MBE) on (100) GaAs substrate. In our studies two different samples were used: A777 and A963. The former sample had 20-nm-thick layer of (Ga,Mn)As with Mn content $x=7\%$. After the MBE growth this sample was capped with amorphous As and annealed in the MBE growth chamber at the temperature of 210 °C (controlled by the IR pyrometer) for two hours (see Ref. 8 for details). The Curie temperature determined from superconducting quantum interference device (SQUID) magnetometry was close to 85 K (Fig. 1, squares). The latter sample had a (Ga,Mn)As layer of 50 nm and $x=6\%$. This sample was not annealed after the MBE growth. The Curie temperature for

A963 sample was approximately 50 K (Fig. 1, diamonds). Since the amount of the magnetic material of our samples was very small, the SQUID signal was also very small. Precise measurements reveal also the presence of a very small high-temperature component in the magnetization. It was checked that this contribution did not originate from (Ga,Mn)As layer as it was present even after the layer was removed.

The effect of pressure on the paramagnetic-ferromagnetic phase transition in (Ga,Mn)As was studied by transport measurements under hydrostatic pressure up to 1 GPa at temperatures range 20–300 K. Standard Hall bars were prepared lithographically and six ohmic contacts were made with indium. The samples were placed in a beryllium-copper self-clamping high-pressure cell. A mixture of kerosene and oil was used as a pressure transmitting medium.

To monitor pressure-induced changes in Curie temperature we adopted two procedures: (1) the study of zero-field resistivity $\rho(T)$; the position of maximum in $d\rho/dT$ was reported to coincide with the Curie temperature⁹ and (2) Hall

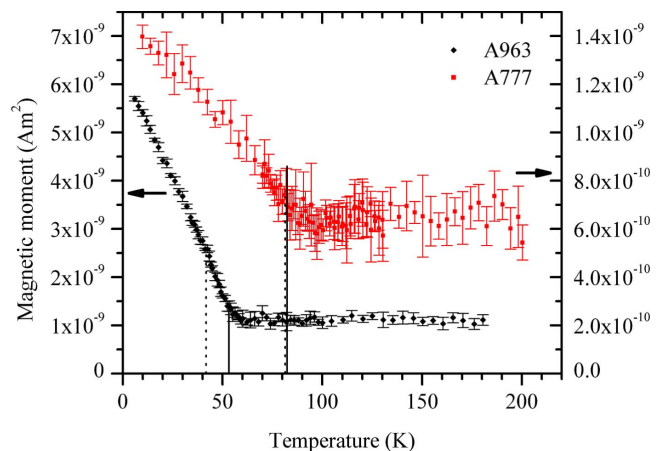


FIG. 1. (Color online) Temperature dependence of magnetic moments measured by SQUID at $B=2$ mT at ambient pressure for both samples (diamonds—A963 and squares—A777). The dotted and solid vertical lines depict the values of T_C extracted from the maximum of $d\rho/dT$ and from the Hall voltage hysteresis, respectively (see text).

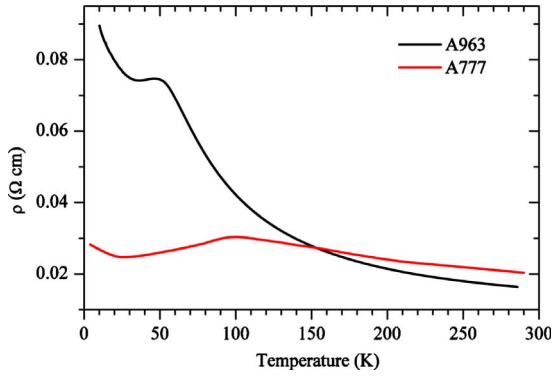


FIG. 2. (Color online) Resistivity as a function of temperature $\rho(T)$ measured for the two samples at ambient pressure, before pressure cycles.

resistivity measurements in a magnetic field perpendicular to the sample surface. In ferromagnetic phase the Hall resistivity is a sum of the ordinary Hall effect due to the Lorentz force and the anomalous Hall effect originating from the asymmetric scattering in the presence of magnetization,¹⁰

$$R_{Hall} = \frac{R_0 B}{d} + \frac{R_S \mu_0 M}{d}, \quad (1)$$

where R_0 is the ordinary Hall coefficient, B and M the magnetic field and the magnetization perpendicular to the layer, respectively, R_S the anomalous Hall coefficient, and d the thickness of the (Ga,Mn)As layer. For $T < T_C$ hysteresis loop in M and thus in $R_{Hall}(B)$ is observed. The studies of the evolution of the hysteresis provide the method to monitor T_C .

The results of the measurements of zero-field resistance versus temperature performed at ambient pressure are shown in Fig. 2. The behavior of $\rho(T)$ of the samples is quite different: A777 has rather metallic character while A963 is clearly of semiconducting type. Characteristic rounded cusps are observed at about 100 and 50 K for the sample A777 and A963, respectively, which are believed to be related to ferromagnetic-paramagnetic phase transition.¹¹ Figure 3 shows the evolution of the resistivity $\rho(T)$ dependence on

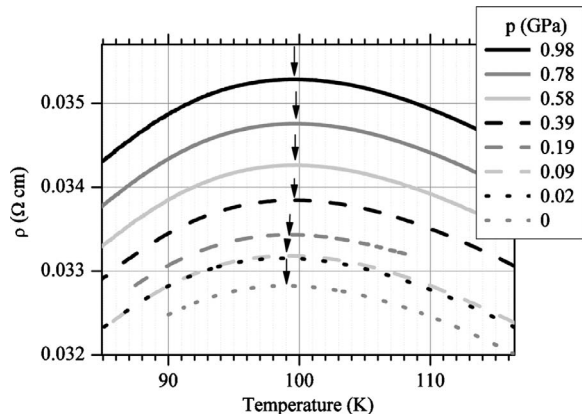


FIG. 3. Resistivity evolution as a function of pressure for the A777 sample. Arrows show the evolution of the position and magnitude of the maximum in $\rho(T)$.

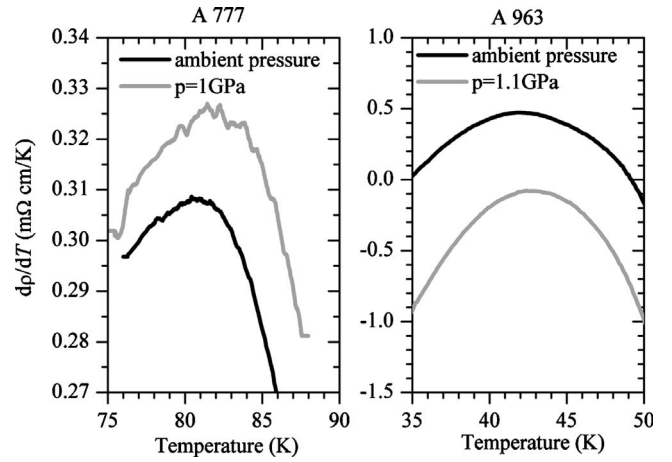


FIG. 4. Temperature derivative of the resistivity, $d\rho/dT$ calculated for both samples from the experimental data measured at ambient pressure and high pressure.

pressure. A clear shift toward lower temperatures is observed. At the same time the value of the resistivity increased by about 7%. Figure 4 shows temperature derivative of resistivity $d\rho/dT$, calculated numerically and presented close to maxima, which can be identified with T_C .⁹ For both samples the maxima shift under pressure toward higher temperatures, although the shift is relatively small.

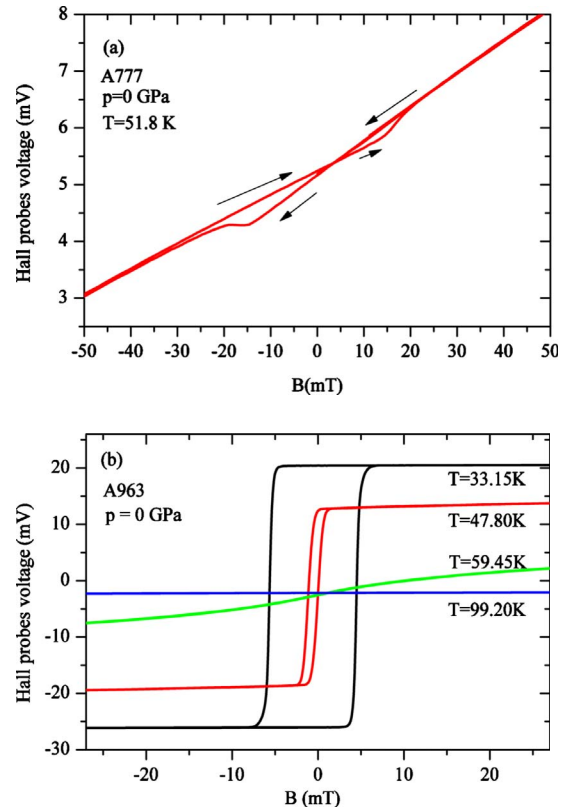


FIG. 5. (Color online) Runs of the Hall voltage versus magnetic field revealing hysteresis loops for both our samples: (a) A777 and (b) A963. Magnetic field was perpendicular to the sample plane. For the sample A963 the shape of the hysteresis loop indicates that an easy axis was perpendicular to the sample plane.

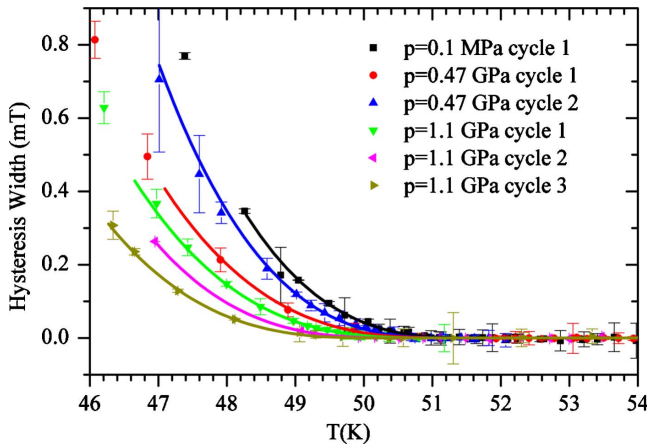


FIG. 6. (Color online) Temperature dependence of the width of the hysteresis loop measured for the sample A963 in a series of experiments made at various pressures.

The results of the second method are shown in Fig. 5, which presents hysteresis loops of Hall voltage, measured for $T < T_C$. Remarkably, the shapes of hysteresis loops are very different for both samples, which indicates different magnetic easy axis for the samples studied (for A777—in the sample plane and for A963—perpendicular to plane). This is consistent with observations that for samples with lower hole concentrations easy axis can be perpendicular to plane.¹² Figure 6 presents the measured dependence of the width of the hysteresis loop on temperature for a few series of experiments made at various pressures for sample A963. One can see that hysteresis appears at temperatures close to 50 K. The results obtained for nominally the same values of pressure but during different experimental cycles differ slightly, which will be addressed later, but an overall pressure-induced decrease in T_C measured with this method is clearly visible.

The summary of our experimental results is shown in Fig. 7. Referring to the absolute value of T_C , for sample A777 both methods give consistent values of T_C which agree very well with the SQUID data of Fig. 1. For the sample A963 the two methods are inconsistent—the values of T_C differ by more than 10 K. The results obtained from the hysteresis analysis agree with the SQUID data while the others do not. Most probably this is due to a diverging background in $\rho(T)$ dependence observed at low temperatures for A963 sample which shifts the maximum of $d\rho/dT$ toward lower temperatures. This also suggests that for samples that show strongly temperature-dependent semiconducting type $\rho(T)$, the method proposed in Ref. 9 cannot be reliably applied. Referring to the pressure trends observed, for the A777 the application of hydrostatic pressure leads to the increase in Curie temperature. Both methods used reveal the similar trend, with a pressure coefficient of $dT_C/dp = +(2-3)$ K/GPa. This value is much higher (approximately four times) than the one found for (In,Mn)Sb.⁷ However, for the sample A963 measurements of the hysteresis loop revealed an opposite pressure-induced shift ($dT_C/dp = -1.0 \pm 0.2$ K/GPa). As mentioned above, the second method is not credible enough to draw any conclusions, due to high impact of the diverging

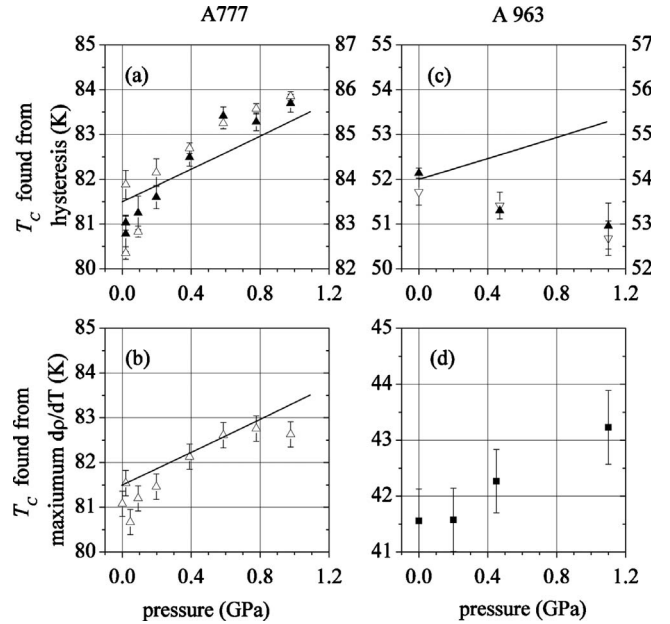


FIG. 7. Pressure dependence of Curie temperature deduced [(a) and (c)] from the hysteresis loops detected by Hall resistivity and [(b) and (d)] from the maximum of $d\rho/dT$. (a) and (b) refer to the sample A777 and (c) and (d) to the sample A963. The points give the values averaged over different pressure cycles. The two sets of points visible in Hall resistivity results (a) and (c) correspond to different Hall probes. Although there is a difference in the absolute value of the T_C , the observed pressure trends are identical. The solid lines show the predictions of the model, described in the text, assuming that the free hole concentration changes due to volume changes only.

background. To sum up, our experimental results showed that pressure-induced variation in Curie temperature is small and strongly depends on the metallic/semiconductor character of a sample.

Within RKKY interaction the temperature of ferromagnetic-paramagnetic phase transition is given by¹

$$T_C = x_{eff} S(S+1) \frac{(N_0\beta)^2}{N_0} \frac{A_F m^* k_F}{12k_B \pi^2 \hbar^2} - T_{AF}, \quad (2)$$

where x_{eff} is an effective manganese concentration, S is the spin of Mn ion ($=5/2$), N_0 is a concentration of the cation sites, $N_0\beta$ is the p - d exchange energy, k_F is the Fermi wave vector, m^* is a valence-band density of states effective mass, k_B is the Boltzmann constant, A_F is a constant describing carrier-carrier interactions, and T_{AF} is a parameter describing antiferromagnetic superexchange interaction—for (Ga,Mn)As we put $T_{AF} \approx 0$.

External hydrostatic pressure can influence the following parameters from Eq. (2): (1) *lattice cell volume and thus* N_0 . In our estimations the pressure-induced change in N_0 was calculated according to the compressibility coefficients for GaAs,¹³ (2) *Fermi vector, k_F* . It is proportional to $p^{1/3}$ (where p stands for the hole concentration). We assumed that p changes due to volume changes only. (3) *Exchange energy*. According to Ref. 1 it should vary with a lattice constant as $N_0\beta \sim a_0^{-3}$ (i.e., β can be treated as pressure independent).

(4) *Heavy- and light-hole effective masses.* To our knowledge it is not completely clear how these parameters change with pressure. However we adopted $\frac{1}{m_{hh}^*} \frac{dm_{hh}^*}{dp} = 0.074 \text{ GPa}^{-1}$ and $\frac{1}{m_{lh}^*} \frac{dm_{lh}^*}{dp} = -0.001 \text{ GPa}^{-1}$ after Ref. 14.

The pressure dependence of T_C predicted by Eq. (2) is given by the solid lines in Fig. 7. The model reasonably well describes the pressure variation in T_C for the sample A777 but is inconsistent with the experimental data obtained for the sample A963 from the analysis of the hysteresis loop. We believe that this striking difference in pressure shift of T_C for both samples is related to different sample properties. Both samples have substantially different Curie temperatures and different easy axes but what seems the most important for the understanding of the pressure shifts of their T_C is the qualitatively different $\rho(T)$ dependences. Sample A777 is semi-metallic while A963 is semiconducting. This means that in A963 the degree of localization of holes is much higher. According to the generalized alloy theory¹⁵ the difference in the localization of holes may lead to both: different apparent values and different pressure dependences of p - d exchange energy $N_0\beta^{(app)}$ (and thus the Curie temperature). In particular, for samples close to metal-insulator transition, a decrease in the T_C is expected when the localization is increased. To discuss the localization of Mn acceptor we recall experimental studies,^{16,17} showing that for Mn acceptor in Ga compounds (GaN, GaP, GaAs, and GaSb) the smaller is the distance between the neighboring atoms, the more localized is the Mn acceptor. Under hydrostatic pressure we expect a similar trend—although the anion remains the same, the interatomic distances are reduced and in consequence localization increases. Such an interpretation is supported by the experimental observation of a strong increase in sample re-

sistivity as a function of pressure (on the order of 30% for the pressure range used for A963, in contrast to 7% for A777).

There are two points that should be mentioned. Our experiments reveal some pressure-induced irreversible changes in sample properties (Fig. 6). Such observations of sample degradation are common in the pressure studies, when the lattice mismatch between the subsequent layers is increased by pressure [this is the case for (Ga,Mn)As on GaAs] and often they impede to make credible conclusions. However, due to tedious, several measurement cycles, the overall pressure-induced trend is clearly demonstrated. Another point is that in the model discussed the free hole concentration was assumed to change due to volume changes only. This does not necessarily have to be true. If in contrast one assumes that the observed pressure changes in the resistivity are entirely due to hole concentration (which is unrealistic), Eq. (2) would predict the decrease in T_C for both samples with the rate of a few kelvins per 1 GPa. This means that there is a need of complementary studies of the influence of pressure on the free hole concentration in GaMnAs.

In summary, we have observed that in two (Ga,Mn)As samples differing in the degree of hole localization the pressure-induced variation in the Curie temperature was opposite. This finding is explained by a different dependence of T_C upon increasing p - d hybridization depending on whether the sample is far or close to the metal-to-insulator transition.¹⁵

We would like to thank T. Dietl and A. Wołoś for valuable discussions. J.S. acknowledges support from the FunDMS Advanced Grant of the European Research Council within the “Ideas” Seventh Framework Programme of the EC.

¹T. Dietl, H. Ohno, and F. Matsukura, *Phys. Rev. B* **63**, 195205 (2001).

²T. Jungwirth, J. König, J. Sinova, J. Kučera, and A. H. MacDonald, *Phys. Rev. B* **66**, 012402 (2002).

³T. Jungwirth, J. Sinova, J. Mašek, J. Kučera, and A. H. MacDonald, *Rev. Mod. Phys.* **78**, 809 (2006).

⁴T. Dietl, *Semicond. Sci. Technol.* **17**, 377 (2002).

⁵H. Ohno, D. Chiba, F. Matsukura, T. Omiya, E. Abe, T. Dietl, Y. Ohno, and K. Ohtani, *Nature (London)* **408**, 944 (2000).

⁶M. Csontos, G. Mihály, B. Jankó, T. Wojtowicz, W. L. Lim, X. Liu, and J. K. Furdyna, *Phys. Status Solidi C* **1**, 3571 (2004).

⁷M. Csontos, G. Mihály, B. Jankó, T. Wojtowicz, X. Liu, and J. K. Furdyna, *Nature Mater.* **4**, 447 (2005).

⁸M. Adell, L. Ilver, J. Kanski, V. Stanciu, P. Svedlindh, J. Sadowski, J. Z. Domagala, F. Terki, C. Hernandez, and S. Charar, *Appl. Phys. Lett.* **86**, 112501 (2005).

⁹V. Novák, K. Olejník, J. Wunderlich, M. Cukr, K. Výborný, A. W. Rushforth, K. W. Edmonds, R. P. Campion, B. L. Gallagher, J. Sinova, and T. Jungwirth, *Phys. Rev. Lett.* **101**,

077201 (2008).

¹⁰C. M. Hurd, *The Hall Effect in Metals and Alloys* (Plenum Press, New York, 1972).

¹¹F. Matsukura, H. Ohno, A. Shen, and Y. Sugawara, *Phys. Rev. B* **57**, R2037 (1998).

¹²M. Sawicki, F. Matsukura, T. Dietl, G. M. Schott, C. Ruester, G. Schmidt, L. W. Molenkamp, and G. Karczewski, *J. Supercond.* **16**, 7 (2003).

¹³O. Madelung, *Semiconductors: Data Handbook* (Springer, New York, 2004).

¹⁴S. Adachi, *GaAs and Related Materials: Bulk Semiconducting and Superlattice* (World Scientific, Singapore, 1994).

¹⁵T. Dietl, *Phys. Rev. B* **77**, 085208 (2008).

¹⁶A. Wołoś, M. Piersa, G. Strzelecka, K. P. Korona, A. Hruban, and M. Kaminska, *Phys. Status Solidi C* **6**, 2769 (2009).

¹⁷B. Stepanek, P. Hubik, J. J. Mares, J. Kristofik, V. Sestakova, L. Pekarek, and J. Sestak, *Semicond. Sci. Technol.* **9**, 1138 (1994).