Ferromagnetic spinel CuCr₂Se₄ studied by Raman spectroscopy and lattice dynamics calculations

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The lattice dynamics of the ferromagnetic spinel $CuCr₂Se₄$ ($T_C=430$ K) was studied experimentally by measuring the Raman spectra and theoretically by calculation of zone center phonon frequencies within a shell model. All Raman allowed modes *A*1*g*+*Eg*+ 3*F*2*g*- were identified and assigned to specific atomic motions. The relative intensity of the Raman lines varies strongly with excitation photon energy between 1.58 and 2.71 eV but no significant phonon anomalies are observed near T_{C} .

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I. INTRODUCTION

In the recent years there is a revived interest in the properties of ferromagnetic chromium chalcogenide spinels MCr_2X_4 (*M* = divalent element and *X* = S, Se, Te). These materils are prospective candidates for spin-based electronic (spintronic) applications as the strong interaction between the electronic and spin subsystems results in drastic changes in the electronic transport and optical properties near the Curie temperature, T_{C} .^{[1–](#page-4-0)[3](#page-4-1)} CuCr₂Se₄ (T_{C} =430 K) attracts particular attention as it is ferromagnetic at room temperature and some theoretical models predict that with suitable doping it may become half-metallic—being an excellent metal for one spin channel and excellent insulator for the other spin channel[.4](#page-4-2) While the structural, magnetic, and electric properties of $CuCr₂Se₄$ have been subject of numerous studies, there are to our knowledge no reports on its lattice dynamics and possible effects of spin-phonon coupling. In this work we present and analyze the experimental Raman spectra of $CuCr₂Se₄$ single crystals in close comparison with predictions of lattice dynamics calculations within a shell modes. The Raman mode intensities exhibit resonant behavior for excitation photon energies between 1.58 and 2.71 eV but no significant anomalies in phonon parameters are observed near the magnetic transition at $T_{\rm C}$.

II. SAMPLES AND METHODS

Single crystals of $CuCr₂Se₄$ were grown using the chemical vapor transport technique following a procedure similar to that described in Ref. [5.](#page-4-3) Briefly, polycrystalline $CuCr₂Se₄$ was used as the precursor with iodine gas as transport agent placed in a sealed quartz tube (length: 15 cm; diameter: 1.5 cm). The source temperature was maintained at 870 \degree C while the crystal growth at the other end of the tube occurred at 770 °C. The reaction was carried out for a period of 80 h, after which the tube was cooled down and broken to collect the single plateletlike crystals with (111) (111) (111) surfaces (see Fig. 1) deposited in the cooler zone.

The Raman spectra were measured with five laser lines (784, 633, 515, 488, and 458 nm) using a T64000 spectrometer equipped with microscope and heating stage. In temperature-dependent studies the laser power at the probe spot $(3-4 \mu m)$ diameter) was kept below 0.1 mW to assure negligible local laser heating.

III. RESULTS AND DISCUSSION

The structure of MCr_2X_4 is described by the $Fd\overline{3}m$ (No. 227) space group. The primitive cell contains 14 atoms and the reduction in the 42-dimensional representation Γ at \vec{k} $= 0$ into irreducible representations gives⁶

$$
\Gamma = A_{1g} + E_g + F_{1g} + 3F_{2g} + 2A_u + 2E_u + 5F_{1u} + 2F_{2u}.
$$

The A_{1g} , E_g , and the three F_{2g} modes are Raman active, four of the five F_{1u} modes are IR active and one is an acoustic mode. The F_{1g} , $2A_{2u}$, $2E_u$, and $2F_{2u}$ modes are silent. Simple calculations for backward scattering from a (111) surface (Table \overline{I} \overline{I} \overline{I}) show that the scattered intensity should not depend on the rotation of the crystal surface around the propagation

FIG. 1. (Color online) $CuCr₂Se₄$ surface topography. The laser focus spot is also seen.

Mode $0^{\circ} < \beta < 90^{\circ}$ Parallel $\vec{e}_i \parallel \vec{e}_s \beta = 0^\circ$ Crossed $\vec{e_i} \perp \vec{e_s} \beta = 90^\circ$ *A*_{1*g*} $a^2 \cos^2 \beta$ a^2 0
 E_c $b^2(\sin^2 \beta + \cos^{\beta})$ b^2 b^2 E_g $b^2(\sin^2 \beta + \cos^{\beta})$ ab² *b*² *b*² F_{2g} $d^2(1-\frac{2}{3}\sin^2\beta)$ d^2 $\frac{2}{3}$ $\frac{2}{3}d^2$

TABLE I. Polarization selection rules for backward scattering from a (111) surface.

direction of the incident light but only on the angle β between the incident \vec{e}_i and scattered \vec{e}_s polarizations.

A. Raman spectra

Figure [2](#page-1-1) shows the Raman spectra of $CuCr₂Se₄$ obtained with 633-nm excitation at room temperature with parallel $(\vec{e}_i \mid \vec{e}_s)$ and crossed $(\vec{e}_i \perp \vec{e}_s)$ scattering configurations. Based on the selection rules of Table [I](#page-1-0) the symmetries of the Raman lines at 108 cm⁻¹(F_{2g}), 144 cm⁻¹(E_g), 170 cm⁻¹(F_{2g}), 220 cm⁻¹(F_{2g}), and 227 cm⁻¹(A_{1g}), are unambiguously determined. The Raman mode frequencies of $CuCr₂Se₄$ are compared to those of isostructural $ZnCr_2Se_4$,^{[7](#page-4-5)} CdCr₂Se₄,^{[8](#page-4-6)} and $HgCr_2Se_4$ (Ref. [9](#page-4-7)) in Table [II.](#page-1-2)

The spectra obtained with 784 nm (1.58 eV), 633 nm (1.96 eV) , 515 nm (2.41 eV) , 488 nm (2.54 eV) , and 458 nm (2.71 eV) , normalized to the intensity of the E_g mode, are compared in Fig. [3.](#page-2-0) The most obvious change with increasing excitation photon energy is the increase in the relative intensity of the F_{2g} modes near 170 and 220 cm⁻¹. Similar resonant behavior has been reported for $CdCr_2Se_4$.^{[8](#page-4-6)}

The temperature-dependent Raman spectra between 300 and 460 K as obtained with 633-nm excitation are shown in Fig. [4.](#page-2-1) The variations in the phonon parameters do not exhibit detectable anomalies near the magnetic transition that may be considered as manifestation of strong spin-phonon

FIG. 2. (Color online) Raman spectra of $CuCr₂Se₄$ with parallel (HH) and crossed (HV) scattering configurations.

TABLE II. Comparison of Raman mode frequencies $(in cm^{-1})$ of CuCr₂Se₄, ZnCr₂Se₄, CdCr₂Se₄, and HgCr₂Se₄ at 300 K.

Mode	CuCr ₂ Se ₄ (This work)	ZnCr ₂ Se ₄ (Ref. 7)	CdCr ₂ Se ₄ (Ref. 8)	HgCr ₂ Se ₄ (Ref. 9)
A_{1g}	227	240	242	236
E_{g}	144	152	156	153
F_{2g}	108	112	85	60
F_{2g}	170	182	169	164
F_{2g}	220	230	226	207

coupling. Here again the results are similar to those for $CdCr_2Se_4$ and $HgCr_2Se_4.$ ^{[8](#page-4-6)[,9](#page-4-7)}

B. Lattice dynamics calculations

As it is clear from the experimental results, the polarized Raman measurements allow complete discrimination of the Raman-active modes of different symmetry. In the *Eg* and A_{1g} vibrations only Se atoms are involved. The displacement vectors of the Se atoms in these modes are determined uniquely on the basis of symmetry considerations (see Table **[III](#page-2-2)**). The corresponding displacement patterns are shown in Fig. [5.](#page-2-3) The *A*1*^g* mode is a symmetric radial breathing of the CuSe₄ tetrahedra while the E_g mode is a result of Se displacement tangential to the Cu-Se bonds. In F_{2g} modes, however, a strong admixture of Cu and Se vibrations could be expected due to the comparable atomic masses of copper and selenium Thus, lattice dynamics calculations are necessary in order to obtain detailed information for the atomic displacements in these modes.

Lattice dynamical calculations for isostructural MCr_2X_4 chalcogenides $(M=Zn, Cd,Hg;X=S,Se)$ reported before are based on valence force fields $10,11$ $10,11$ or a rigid-ion model $(RIM).¹²$ $(RIM).¹²$ $(RIM).¹²$ The valence force constants are of limited transferability in compounds with significant ionic bonding. As noted in Ref. [11,](#page-4-9) the character of the various phonons as relating to eigenvectors and potential energy distribution differs to a much larger extent than expected for isostructural compounds. The rigid-ion model proposed in Ref. [12](#page-4-10) is based upon pairwise cation-anion and anion-anion shortrange potentials. The potential parameters, however, are mapped onto a set of effective stretching and bending force constants. The reported effective force constants do not display a clear trend over the series of different M atoms and their transferability in the case of $M = Cu$ is not evident. Moreover, RIM does not account for the atomic polarizabilities and is unable to reproduce other important lattice properties as the high-frequency dielectric constant, the LO-TO splitting, as well as the off-diagonal elastic constants.

For the reasons outlined above we performed lattice dynamics calculations for $CuCr₂Se₄$ in the frame of the shell model (SM), which is the minimal extension over RIM capable to account for the ionic polarizability and to predict correct LO-TO splitting. In order to reduce the number of adjustable model parameters some physical simplifications were adopted. First, a valence shell was considered for the Se

FIG. 3. (Color online) Raman spectra of $CuCr₂Se₄$ with 784 nm (1.58 eV) , 633 nm (1.96 eV) , 515 nm (2.41 eV) , 488 nm (2.54 eV) , and 458 nm $(2.71$ eV) excitations.

atoms only while Cu and Cr were treated as rigid ions. Second, the van der Waals attractive interaction was considered to act only between Se shells while it was neglected for the Cu (Cr) core—Se shell pairs. These assumptions are justified by the much higher polarizability of the Se−2 ion compared to that of the transition-metal ions. The rigid-ion approximation for transition metals is a common approximation in the shell-model calculations on transition-metal oxides.¹³ The short-range interatomic interactions were modeled by a Buckingham potential: $V(r) = A \exp(-r/\rho) - C/r^6$, where a nonzero van der Waals constant *C* was retained for the Se shell—Se shell pairs only.

Even with the above simplifications the number of adjustable parameters exceeds the number of observables for $CuCr₂Se₄$, the lattice constant *a*, the Se fractional coordinate *u*, and the five Raman-mode frequencies. For this reason, the

FIG. 4. (Color online) Temperature-dependent Raman spectra of CuCr₂Se₄ near the ferromagnetic transition $(T_{\rm C} = 430 \text{ K})$. The inset shows the variations with *T* of the A_{1g} frequency.

TABLE III. Symmetry-adapted displacement vectors of the Se(u , u , u) atom in the A_{1g} and E_g modes. The displacement vectors for the other atoms in the unit cell can be obtained by application of the corresponding symmetry transformation matrices (Ref. [6](#page-4-4)).

Mode symmetry	Normalized displacement vectors
A_{1g}	$\vec{e} = (1,1,1)/\sqrt{3}$
E_{ϱ}	$\vec{e}_1 = (2, -1, -1)/\sqrt{6}$ $\vec{e}_2 = (0,+1,-1)/\sqrt{2}$

 A , ρ , and C parameters for the Se-Se interaction, the Se core and shell charges *X* and *Y*, and the Se core-shell spring constant *k* were adjusted initially by fitting of the lattice constant, static and optical dielectric constants, the LO and TO phonon frequencies, and the elastic modulus in ZnSe. As the next step these parameters were transferred to $CuCr₂Se₄$ and the parameters of the Cu-Se and Cr-Se core-shell interactions were optimized in order to fit the lattice parameter, the Se fractional coordinate u and the frequencies of the Ramanactive modes. The list of as obtained SM parameters is shown in Table [IV.](#page-3-0)

A comparison between the calculated and the experimental lattice parameters and Raman-mode frequencies is shown in Table [V.](#page-3-1) The calculated atomic displacements for the three F_{2g} modes are depicted in Fig. [6.](#page-3-2) The lowest-frequency mode at 108 cm−1 consists of an almost pure Cu translation within a practically static Se cage. This assignment corroborates the fact that the mode frequency scales with the mass *M* of the *A* atom $(A = Cu, Cd, Hg)$ as $M^{-1/2}$ (see Table [II](#page-1-2)). As an additional piece of evidence, we performed phonon calculations by substituting Cu with Cd and Hg and keeping the same values for the interatomic potentials. We obtained the frequencies of the lowest F_{2g} mode at 83 cm⁻¹ for $A = Cd$ and 63 cm−1 for *A*=Hg in perfect agreement with the experimental values. The medium-frequency mode at 173 cm⁻¹ can be described as Cu translation against the Se cage, which moves in the opposite direction. The highest-frequency vibration at 200 cm−1 is a superposition of an in-phase stretching of the Cu-Se bonds and Se-Cu-Se angle bending.

C. Resonant behavior of Raman scattering

The Raman scattering cross-section $S(h\omega)$ in nontransparent materials is related to the combined electronic density of states. As a rule $S(\hbar \omega)$ is proportional to the $\left|\frac{d\epsilon(\hbar \omega)}{dh \omega}\right|^2$ and has maxima near interband gaps. The experimentally observed dependence of the Raman intensity on the excitation photon energy $I(h\omega)$, however, is additionally modified by the spectral properties of the Raman setup and the absorption and

FIG. 5. (Color online) Tetrahedral displacements of the A_{1g} and *Eg* modes.

Atom	Core charge X	Shell charge Y	Core-shell spring constant k (eV/\AA^2)
Cu	1.00		
Cr	2.30		
Se	2.19	-3.59	69.7
Atomic pair	A (eV)	ρ $\rm (\AA)$	C $(eV \times \AA^6)$
Cu core—Se shell	6827.4	0.2364	θ
Cr core—Se shell	13817.5	0.2513	$\overline{0}$
Se shell—Se shell	1482.2	0.3440	136.3

TABLE IV. Adjusted shell-model parameters for $CuCr₂Se₄$.

reflection losses, which occur for the incident and scattered light. The detail analysis of resonant Raman scattering in closely related $CdCr_2Se_4$ has shown that $S(h\omega)$ has a maximum near 2 eV .^{[8](#page-4-6)} It has also been shown in that study that the intensity of the F_{2g} Raman lines, if normalized to the E_g intensity, strongly increases with photon energy between $1.\overline{5}$ and 2.8 eV thus demonstrating the dependence of scattering cross section on the phonon symmetry. As it follows from Fig. [3,](#page-2-0) similar enhancement of the relative intensity of the F_{2g} modes is observed for CuCr₂Se₄. Figure [7](#page-4-12) shows in more detail the variation with $\hbar \omega$ of the normalized intensity for the 170 cm⁻¹(F_{2g}), 220 cm⁻¹(F_{2g}), and 227 cm⁻¹(A_{1g}) modes.

For tentative explanation of the resonant enhancement of the F_{2g} intensity let us consider the *ab initio* electronic-structure calculations^{14,[15](#page-4-14)} for CuCr₂Se₄. Following these calculation the electronic density of states above the Fermi level is dominated by the Cr *d* states while those below the Fermi level by the Cu *d* and the Se *p* states. An optical transition between Cu and Cr *d* bands can be disregarded due to the large spatial separation between the Cu and Cr ions. Therefore, a possible candidate for an optical transition could be an electron transfer between the Se *p* and Cr *d* orbitals. Transition energies of 3.2 eV between Se $p_{1/2}$ and Cr $d_{3/2}$ states, and 2.84 eV between Se $p_{3/2}$ and Cr $d_{5/2}$ states could be expected on the basis of calculated centers of gravity of the corresponding bands.¹⁵ The latter value is close the experi-

FIG. 6. (Color online) Atomic displacement of the Raman modes of $CuCr₂Se₄$.

mentally observed maximum at 2.75 eV of the optical conductivity of $CuCr₂Se₄$ reported by Ohgushi *et al.*^{[16](#page-4-15)}

The lattice dynamics calculation can help us understand why A_{1g} and F_{2g} modes exhibit stronger coupling to Se *p*-Cr d transitions than the E_g . The coupling of a Ramanactive phonon to an optical transition is mainly due to two mechanisms: (i) modulation of the dipolar transition moment and (ii) modulation of the transition energy by the atomic displacements. As a rule the modulation of the transition energy has stronger resonant behavior and gives a dominant contribution to the Raman scattering intensity under resonant conditions. We will consider the optical $p \rightarrow d$ transitions in a simplified tight-binding picture assuming that the transition energy is roughly equal to the difference between the on-site energies of the Se *p* and Cr *d* orbitals. Since the on-site energy is a scalar quantity, it transforms according to the permutational representation of the given atomic position. The decomposition of the permutational representations for the $16d$ (Cr) and $32e$ (Se) positions gives

$$
\Gamma(\text{Se}) = A_{1g} + A_{2u} + F_{2g} + F_{1u},\tag{1}
$$

$$
\Gamma(\text{Cr}) = A_{1g} + F_{2g}.\tag{2}
$$

Therefore, among Raman-active modes only A_{1g} and F_{2g} modulate the Se and Cr orbital energies and, respectively, the

TABLE V. Cell parameters, frequencies $(in cm⁻¹)$, and atomic motions.

Cell parameters			Expt.	Calc
Lattice constant, $a(A)$			10.337	10.363
Se position, u			0.382	0.383
	Frequency			
Mode	Expt.	Calc	Mode description in terms of $CuSe4$ vibrations	
F_{2g}	108	108	Cu translation	
E_{g}	144	156	Se-Cu-Se bending	
\mathcal{F}_{2g}	170	173	Cu translation against Se	
F_{2g}	220	200	Cu-Se stretching+Se-Cu-Se bending	
A_{1g}	227	218	Symmetric Cu-Se stretching	

FIG. 7. (Color online) Variation in the normalized Raman intensity of the 170 cm⁻¹(F_{2g}), 220 cm⁻¹(F_{2g}), and 227 cm⁻¹(A_{1g}) modes with excitation photon energy.

energy of the $p \rightarrow d$ transitions. This fact can be understood by inspecting the atomic displacement patterns in Fig. [6.](#page-3-2) In the A_{1g} and F_{2g} modes the Se displacements have identical radial components relative to the Cr atom and, therefore, give identical variations in the Madelung potential at the Cr site. For the E_g mode, however, the radial components of the Se displacements are of alternating signs leading to cancellation of the Madelung potential variations at the Cr site. By following our lattice dynamics calculations it is easy to note that the resonant enhancement of the F_{2g} modes increases with the increase in the radial character of the vibration. Interestingly, the A_{1g} mode shows an enhanced normalized intensity also with 784 nm (1.58 eV) excitation, which is in the energy range of the Cr d - d transitions.¹⁶

IV. CONCLUSIONS

The Raman spectra of the ferromagnetic spinel $CuCr₂Se₄$ $(T_{\rm C} = 430 \, \text{K})$ was studied experimentally and analyzed theoretically by calculation of lattice dynamics. The normalized Raman scattering intensity for A_{1g} and F_{2g} exhibits resonant behavior similar to that known for the isostructural $CdCr₂Se₄$ but no significant phonon anomalies are observed near T_{C} .

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