

Measurement of Gilbert damping parameters in nanoscale CPP-GMR spin valves

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In situ device-level measurement of thermal mag-noise spectral linewidths in 60 nm diameter current-perpendicular-to-plane-giant magnetoresistive spin-valve stacks of IrMn/ref/Cu/free, with reference and free layer of similar CoFe/CoFeGe alloy, are used to simultaneously determine the intrinsic Gilbert damping for both magnetic layers. It is shown that careful alignment at a “magic angle” between free- and reference-layer static equilibrium magnetization can allow direct measurement of the broadband intrinsic thermal spectra in the virtual absence of spin-torque effects which otherwise grossly distort the spectral line shapes and require linewidth extrapolations to zero current (which are nonetheless also shown to agree well with the direct method). The experimental magic-angle spectra are shown to be in good qualitative and quantitative agreement with both macrospin calculations and micromagnetic eigenmode analysis. Despite similar composition and thickness, it is repeatedly found that the IrMn exchange pinned reference layer *has an order of magnitude larger* intrinsic Gilbert damping ($\alpha \approx 0.1$) than that of the free layer ($\alpha \approx 0.01$). It is argued that the large reference-layer damping results from strong, off-resonant coupling to lossy modes of an IrMn/ref couple, rather than commonly invoked two-magnon processes.

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I. INTRODUCTION

Spin-torque phenomena, in tunneling magnetoresistive (TMR) or giant-magnetoresistive (GMR) film stacks lithographically patterned into ~ 100 nm nanopillars and driven with dc electrical currents perpendicular to the plane (CPP) of the films have in recent years been the topic of numerous theoretical and experimental papers, both for their physics as well as potential applications for magnetic memory elements, microwave oscillators, and magnetic field sensors and/or magnetic recording heads.¹ In all cases, the electrical current density at which spin-torque instability or oscillation occurs in the constituent magnetic film layers is closely related to the magnetic damping of these ferromagnetic (FM) films.

This paper considers the electrical measurement of thermal mag-noise spectra to determine intrinsic damping at the *device level* in CPP-GMR spin-valve (SV) stacks of sub-100 nm dimensions (intended for read-head applications), which allows simultaneous *R-H* and transport characterization on the same device. Compared to traditional ferromagnetic resonance (FMR) linewidth measurements at the bulk film level, the device-level approach naturally includes finite-size and spin-pumping² effects characteristic of actual devices, as well as provides immunity to inhomogeneous and/or two-magnon linewidth broadening not relevant to nanoscale devices. Complementary to spin-torque FMR using ac excitation currents,³ broadband thermal excitation naturally excites all modes of the system (with larger, more quantitatively modeled signal amplitudes) and allows simultaneous damping measurement in both reference and free FM layers of the spin valve, which will be shown to lead to some interesting and unexpected conclusions. However, spin torques at finite dc currents can substantially alter the absolute linewidth, and so it is necessary to account for or eliminate this effect in order to determine the intrinsic damping.

II. PRELIMINARIES AND MAGIC ANGLES

Figure 1(a) illustrates the basic film stack structure of a prospective CPP-GMR SV read sensor, which apart from the Cu spacer between free layer (FL) and reference layer (RL), is identical in form to well-known, present-day TMR sensors. In addition to the unidirectional exchange coupling between the IrMn and the pinned layer (PL), the usual “synthetic-antiferromagnet (AF)” structure PL/Ru/RL is meant to increase magnetostatic stability and immunity to field-induced rotation of the PL-RL couple, as well as strongly reduce its net demagnetizing field on the FL which otherwise can rotate in response to signal fields. However, for simplicity in interpreting and modeling the spectral and transport data of Sec. III, the present experiment restricts attention to devices with a single RL directly exchange coupled to IrMn, as shown in Fig. 1(b).

The simplest practical model for describing the physics of the device of Fig. 1(b) is a macrospin model that treats the RL unit magnetization \hat{m}_{RL} as fixed, with only the FL magnetization $\hat{m}_{FL}(t) \leftrightarrow \hat{m}(t)$ as possibly dynamic in time. As was

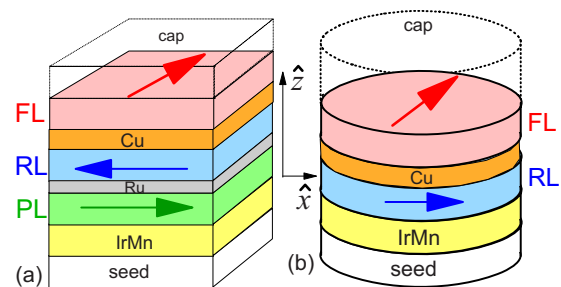


FIG. 1. (Color online) (a) Cartoon of prospective CPP-GMR spin-valve sensor stack, analogous to that used for contemporary TMR read head. (b) Cartoon of simplified spin-valve stack used for present experiments, patterned into ~ 60 nm circular pillars using e-beam lithography.

described previously,⁴ the *linearized* Gilbert equations for small deviations $\mathbf{m}' = (m'_{y'}, m'_{z'})$ about equilibrium $\hat{\mathbf{m}}_0 \leftrightarrow \hat{\mathbf{x}}'$ can be expressed in the primed coordinates \mathbf{m}' as a two-dimensional (2D) tensor/matrix equation,⁵

$$(\vec{D} + \vec{G}) \cdot \frac{d\mathbf{m}'}{dt} + \vec{H}' \cdot \mathbf{m}' = \mathbf{h}'(t) \equiv p \frac{\partial \mathbf{m}'}{\partial \hat{\mathbf{m}}} \cdot \mathbf{h}(t),$$

$$\vec{D} \equiv \alpha \frac{p}{\gamma} \begin{pmatrix} 1 & 0 \\ 0 & 1 \end{pmatrix}, \quad \vec{G} \equiv \frac{p}{\gamma} \begin{pmatrix} 0 & -1 \\ 1 & 0 \end{pmatrix}, \quad p \equiv \frac{(M_s V)_{\text{FL}}}{\Delta m},$$

$$\vec{H}' \equiv (\mathbf{H}^{\text{eff}} \cdot \hat{\mathbf{m}}_0) \begin{pmatrix} 1 & 0 \\ 0 & 1 \end{pmatrix} - \frac{\partial \mathbf{m}'}{\partial \hat{\mathbf{m}}} \cdot \frac{\partial \mathbf{H}^{\text{eff}}}{\partial \hat{\mathbf{m}}} \cdot \frac{\partial \hat{\mathbf{m}}}{\partial \mathbf{m}'}. \quad (1)$$

In Eq. (1), $\partial \mathbf{H}^{\text{eff}} / \partial \hat{\mathbf{m}}$ is a three-dimensional (3D) Cartesian tensor, $\partial \hat{\mathbf{m}} / \partial \mathbf{m}'$ is a 3×2 transformation matrix between 3D unprimed and 2D primed vectors (with $\partial \mathbf{m}' / \partial \hat{\mathbf{m}}$ its transpose) which depends only on $\hat{\mathbf{m}}_0$, and $\mathbf{h}(t)$ is a 3D perturbation field supposed as the origin of the deviations $\mathbf{m}'(t)$. The magnetic moment Δm is an arbitrary *fixed* value but $\Delta m \rightarrow (M_s V)_{\text{FL}}$ is a natural choice for Sec. II. Using an explicit Slonczewski-type⁶ expression for the spin-torque contribution, $\mathbf{H}^{\text{eff}}(\hat{\mathbf{m}})$ is taken to be

$$\mathbf{H}^{\text{eff}} = \frac{-1}{\Delta m} \frac{\partial E}{\partial \hat{\mathbf{m}}} - \eta(\cos \theta) H_{\text{ST}} (\hat{\mathbf{m}}_{\text{RL}} \times \hat{\mathbf{m}}_{\text{FL}}),$$

$$\cos \theta \equiv \hat{\mathbf{m}}_{\text{RL}} \cdot \hat{\mathbf{m}}_{\text{FL}},$$

and

$$H_{\text{ST}} \equiv (\hbar/2e) P_{\text{eff}} I_e / \Delta m \quad (2)$$

for any free-energy function $E(\hat{\mathbf{m}})$. A positive electron current I_e implies electron flow from the RL to the FL. P_{eff} is the net spin polarization of the current inside the Cu spacer. Oersted-field contributions to \mathbf{H}^{eff} will be neglected here. The function $\eta(\cos \theta)$ is discussed below.

With $\mathbf{h}(t)=0$ in Eq. (1), nontrivial solutions $\mathbf{m}'(t) = \mathbf{m}' e^{-st}$ require s satisfy $\det[\vec{H} - s(\vec{D} + \vec{G})] = 0$. The value $I_e \equiv I_e^{\text{crit}}$ when $\text{Re } s = 0$ defines the critical onset of spin-torque instability. Using Eq. (1), the general criticality condition is expressible as

$$\underbrace{\alpha(H'_{y'y'} + H'_{z'z'})}_{I_e \text{ independent}} + \underbrace{H'_{y'z'} - H'_{z'y'}}_{\propto I_e} = 0, \quad (3a)$$

$$\frac{H'_{y'z'} - H'_{z'y'}}{H_{\text{ST}}} \equiv (1 - q^2) \frac{d\eta}{dq} - 2q\eta(q) \quad (q \equiv \cos \theta), \quad (3b)$$

$$\Rightarrow I_e^{\text{crit}} = \frac{\Delta m}{(\hbar/2e) P_{\text{eff}}} \frac{\alpha(H'_{y'y'} + H'_{z'z'})}{(1 - q^2) d\eta/dq - 2q\eta(q)}, \quad (3c)$$

where α is the Gilbert damping. The I_e -scaling of the terms in Eq. (3a) follows just from the form of Eq. (2). The result in Eq. (3b) was derived earlier⁴ in the present approximation of rigid $\hat{\mathbf{m}}_{\text{RL}}$.

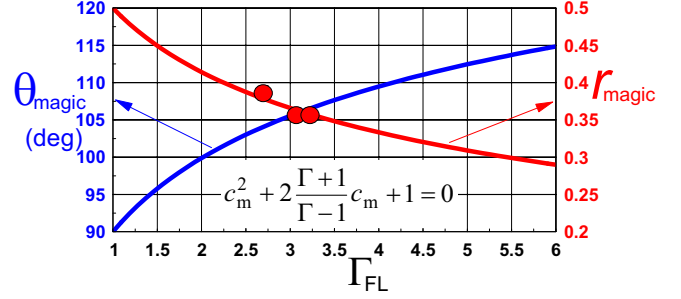


FIG. 2. (Color online) Graph of θ_{magic} and $r_{\text{magic}} = r_{\text{bias}}(\theta_{\text{magic}})$ vs Γ_{FL} as described by Eq. (4). The equation for $c_m = \cos(\theta_{\text{magic}})$ follows from Eqs. (3) and (4). The solid circles are measured $(\Gamma_{\text{FL}}, r_{\text{magic}})$ from Figs. 3, 4, and 6.

With θ the angle between $\hat{\mathbf{m}}_{\text{RL}}$ and $\hat{\mathbf{m}}_{\text{FL}}$ (at equilibrium), it follows from Eq. (3c) that at a “magic angle” θ_{magic} where the right side of Eq. (3b) vanishes, $I_e^{\text{crit}} \rightarrow \infty$ and *spin-torque effects are effectively eliminated from the system at finite I_e* . To pursue this point further, explicit results for $\eta(\cos \theta)$ in Eq. (2) will be used from the prototypical case where the CPP-GMR stack [Fig. 1(b)] is approximately symmetric about the Cu spacer, which is roughly equivalent to the less restrictive situation where the RL and FL are similar materials with thicknesses that are *not* small compared the spin-diffusion length. For this quasisymmetric case, both quasiballistic⁶ and fully diffusive⁷ transport models yield the following simple functional form:

$$\eta(\cos \theta) = \Gamma / [\Gamma + 1 + (\Gamma - 1) \cos \theta],$$

$$r(\cos \theta) \equiv \frac{\delta R \equiv R(\cos \theta) - R_{\text{min}}}{\Delta R \equiv R_{\text{max}} - R_{\text{min}}} = \frac{\eta(\cos \theta)}{\Gamma} (1 - \cos \theta) \quad (4)$$

which also relates η to the normalized magnetoresistance r ($0 \leq r \leq 1$) which is directly measurable experimentally. The transport parameter Γ is theoretically related to the Sharvin resistance^{6,8} or mixing conductance⁸ at the Cu/FL interface but will here be estimated via I_e^{crit} measurement in Sec III. Using $\eta(q)$ from Eq. (4) in Eq. (3), θ_{magic} and $r(\theta_{\text{magic}})$ vs Γ curves are shown in Fig. 2.

The magic-angle concept also applies to the power spectral density (PSD) $S_V = [I_{\text{bias}} \Delta R (dr/d\theta)_{\text{bias}}]^2 S_\theta$ at bias current I_{bias} , arising from thermal fluctuations in θ about equilibrium bias angle θ_{bias} . Assuming $\hat{\mathbf{m}}_{0,\text{RL,FL}}$ in/near the film plane ($\hat{\mathbf{z}}' \equiv \hat{\mathbf{z}} = \text{plane-normal}$), and *requiring* $|I_{\text{bias}}| < |I_e^{\text{crit}}|$, it can be shown⁵ from fluctuation-dissipation arguments that

$$S_\theta(f) \equiv \frac{4k_B T}{\gamma \Delta m} (\vec{\chi} \cdot \vec{D} \cdot \vec{\chi}^\dagger)_{y'y'}, \quad \vec{\chi}(\omega) = [\vec{H}' - i\omega(\vec{D} + \vec{G})]^{-1}$$

$$\Rightarrow S_\theta(f) \equiv \frac{4\gamma k_B T \alpha \gamma^2 (H'_{z'z'} + H'_{y'y'}) + \omega^2}{\Delta m (\omega^2 - \omega_0^2)^2 + (\omega \Delta \omega)^2}, \quad (5)$$

where

$$\omega_0 = \gamma \sqrt{H'_{y'y'} H'_{z'z'} - H'_{y'z'} H'_{z'y'}}$$

and

$$\Delta\omega = \gamma[\alpha(H'_{z'z'} + H'_{y'y'}) + H'_{z'y'} - H'_{y'z'}].$$

Comparing Eq. (5) with Eq. (3), it is seen that the spectral linewidth $\Delta\omega$ is predicted to be a linear function of I_e but with $d\Delta\omega/dI_e \rightarrow 0$ when $\theta_{\text{bias}} \rightarrow \theta_{\text{magic}}$. Since $H'_{z'z'} \gg H'_{y'y'}$ (due to ~ 10 kOe out-of-plane demag fields) and $H'_{y'y'} \gg H'_{y'z'}$ (e.g., for the $I_e < I_e^{\text{crit}}$ measurements in Sec. III), it is only in the linewidth $\Delta\omega$ that the off-diagonal terms $H'_{y'z'}$, $H'_{z'y'}$, can be expected to influence $S_\theta(f)$. Therefore, measurement of $S_V(f)$ with $\theta_{\text{bias}} \equiv \theta_{\text{magic}}$ ideally allows direct measurement of the natural thermal-equilibrium mag-noise spectrum $S_\theta(f)$, from which can be extracted the intrinsic (i.e., I_e -independent) Gilbert damping constant α . This is the subject of Sec. III.

III. EXPERIMENTAL RESULTS

The results to be shown below were measured on CPP-GMR spin valves of stack structure: seed layers/IrMn (6nm)/RL/Cu (3nm)/FL/cap layers. The films were fabricated by magnetron sputtering onto AlTiC substrates at room temperature, with 2 mTorr of Ar sputter gas. The bottom contact was a ~ 1 - μm -thick NiFe layer, planarized using chemical-mechanical polishing. To increase $\Delta R/R$, both the RL and FL were made from $(\text{CoFe})_{70}\text{Ge}_{30}$ magnetic alloys.⁹ The RL includes a thin CoFe layer between IrMn and CoFeGe to help maximize the exchange-coupling strength, and both RL and FL include very thin CoFe at the Cu interface. The resultant $M_s t$ product for the RL and FL were about 0.64 emu/cm^2 . After deposition, SV films were annealed for 5 h at 245°C in 13 kOe applied field to set the exchange pinning direction. The IrMn/RL exchange pinning strength of $J_{\text{pin}} \approx 0.75 \text{ erg/cm}^2$ was measured by vibrating sample magnetometry. After annealing, patterned devices with ≈ 60 nm diameter (measured at the FL) were fabricated using e-beam lithography and Ar-ion milling. A 0.2 - μm -thick Au layer was used as the top contact to devices.

Figure 3 illustrates a full measurement sequence. Devices are first prescreened to find samples with approximate ideal in-plane δR - H loops [Fig. 3(a)] for circular pillars: nonhysteretic, unidirectionally square loops when $H=H_{\parallel}$ is collinear with the exchange pinning direction ($+\hat{x}$), and symmetric loops about $H=0$ when $H=H_{\perp}$ is transverse (\hat{y} axis). The right shift in the δR - H_{\parallel} loop indicates a large demagnetizing field of ~ 500 Oe from the RL on the FL.

As shown previously,⁴ narrow-band “low”-frequency N - I_e measurements [$N \equiv \sqrt{\text{PSD}}(f=100 \text{ MHz})$, 1 MHz bandwidth] can reveal spin-torque criticality as the very rapid onset of excess ($1/f$ -like) noise when $|I_e|$ exceeds $|I_e^{\text{crit}}|$. N - I_e loops are measured with I_e sourced from a continuous sawtooth generator (2 Hz) which also triggers 1/2 s sweeps of an Agilent-E4440 spectrum analyzer (in zero-span averaging mode) for ≈ 50 cycles. With high sweep repeatability and virtually no I_e hysteresis, this averaging is sufficient so that after (quadratically) subtracting the mean $N(I_e \approx 0) \approx 1 \text{ nV}/\sqrt{\text{Hz}}$ electronics noise, the resultant N - I_e loops [Fig. 3(b)] indicate stochastic uncertainty $\ll 0.1 \text{ nV}/\sqrt{\text{Hz}}$.

With $\cos \theta = \pm 1$, it readily follows from Eqs. (3c) and (4) that

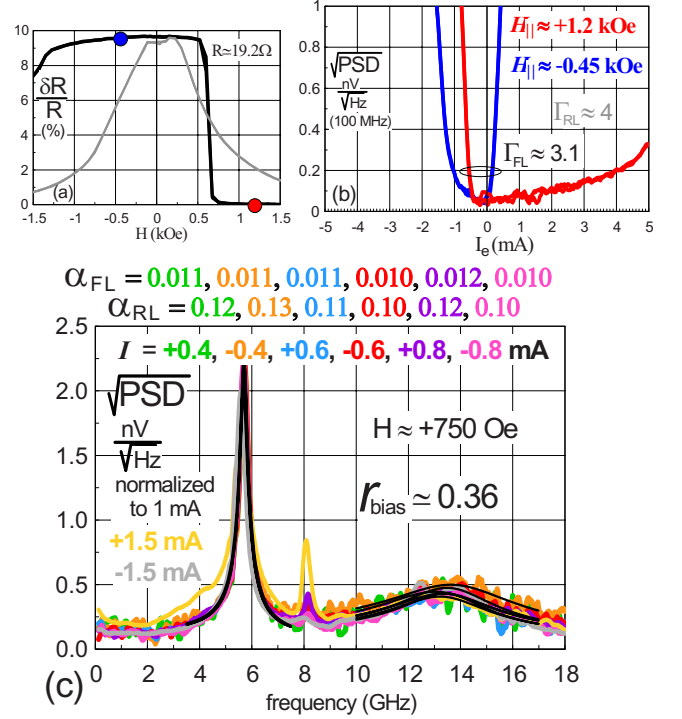


FIG. 3. (Color online) Measurement set for 60 nm device. (a) δR - H_{\parallel} (black) and δR - H_{\perp} (gray) loops at -5 mV bias; solid circles denote discrete H fields used for data in Fig. 3(b). (b) P-state N - I_e loops at $H_{\parallel} \approx +1.2$ kOe, and AP-state N - I_e loops at $H_{\parallel} \approx -0.45$ kOe; FL critical currents to determine Γ_{FL} [via Eq. (6)] enclosed by oval. (c) rms PSD(f, I_e) (normalized to 1 mA) with I_e as indicated by color. Thin black curves are least-squares fits via Eq. (7), fitted values for α_{FL} , α_{RL} listed on top of graph. Measured r_{bias} and applied field H listed inside graph. Field strength and direction (see Fig. 9) adjusted to achieve magic angle. ± 1.5 mA spectra shown but not fit.

$$\Gamma = -\frac{I_e^{\text{crit}}(\theta = \pi) \equiv I_{\text{AP}}^{\text{crit}}}{I_e^{\text{crit}}(\theta = 0) \equiv I_{\text{P}}^{\text{crit}}}. \quad (6)$$

Hence, to estimate Γ , N - I_e are measured with applied fields $H_{\parallel} \approx -0.45, +1.2$ kOe [Fig. 3(b)], which are more than sufficient to align \hat{m}_{FL} antiparallel (AP) or parallel (P) to \hat{m}_{RL} , respectively [see Fig. 3(a)], thereby reducing possible sensitivity to Oersted-field and/or thermal effects. (Reducing $|H_{\parallel}|$ by ~ 200 – 300 Oe did not significantly change either N - I_e curve.) With $I_e > 0$ denoting *electron* flow from RL to FL, it is readily found from Eq. (3) that $I_{\text{AP-FL}}^{\text{crit}} > 0$ and $I_{\text{P-FL}}^{\text{crit}} < 0$ for the FL. By symmetry, it must follow that $I_{\text{AP-RL}}^{\text{crit}} < 0$ and $I_{\text{P-RL}}^{\text{crit}} > 0$ for spin-torque-induced instability of the RL. This sign convention readily identifies these four critical points by inspection of the N - I_e data. To account for possible small (thermal) spread in apparent critical onset, specific values for I_e^{crit} (excluding $I_{\text{P-RL}}^{\text{crit}}$) are defined by where the N - I_e curves cross the $0.2 \text{ nV}/\sqrt{\text{Hz}}$ line, which is easily distinguished from the $\sim 0.05 \text{ nV}/\sqrt{\text{Hz}}$ /mA residual magnetic/thermal background. $I_{\text{P-RL}}^{\text{crit}}$ is estimated in Fig. 3(b) (and repeatedly in Figs. 4–7) to be $\approx +4.5$ mA. Arbitrariness in the value of I_e^{crit} from using the $0.2 \text{ nV}/\sqrt{\text{Hz}}$ criterion is thought to only be of minor significance for $I_{\text{AP-RL}}^{\text{crit}}$, due to the rounded shape of the

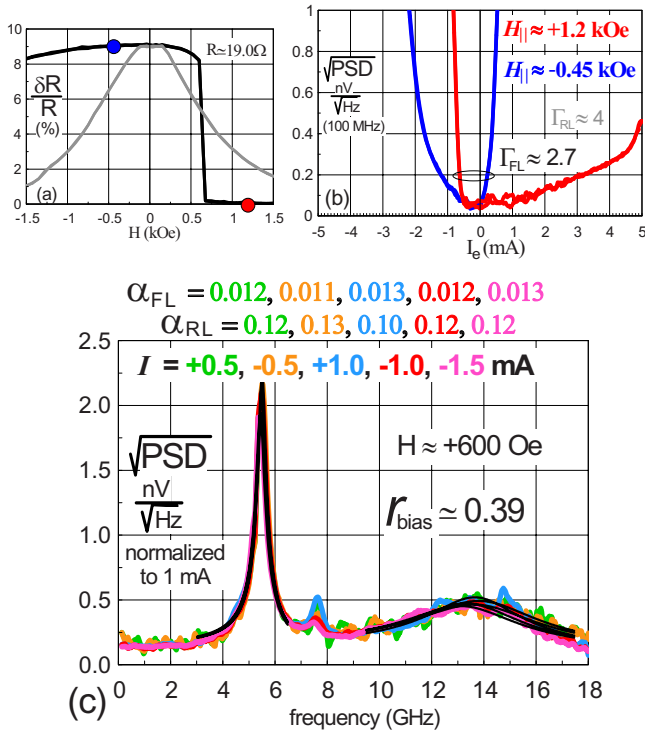


FIG. 4. (Color online) Analogous measurement set for a different (but nominally identical) 60 nm device. as that shown in Fig. 3.

AP N - I_e curves near this particular critical point, which may in part explain why $\Gamma_{\text{RL/Cu}}$ estimated from N - I_e is found to be systematically somewhat larger than $\Gamma_{\text{Cu/FL}}$.

However, the key results here are the 0.1–18 GHz broadband (rms) PSD($f; I_e$) spectra [Fig. 3(c)]. They are measured at discrete dc bias currents with the same Miteq preamp (and in-series bias-T) used for the N - I_e data, the latter being *in situ* gain calibrated vs frequency (with $\approx 50 \Omega$ preamp input impedance and additionally compensating the present ≈ 0.7 pF device capacitance) to yield quantitatively absolute values for these PSD($f; I_e$) [each averaged over ~ 100 sweeps, with PSD($f; I_e=0$) subtracted postprocess]. To confirm the real existence of an effective magic angle, the applied field H was carefully adjusted (by repeated trial and error) in both amplitude and direction to eliminate as much as possible any real-time observed dependence of the raw PSD($f; I_e$) near the FL FMR peak (~ 6 GHz) on the *polarity* as well as amplitude of I_e over a sufficient range. This procedure was somewhat tedious and delicate, and initial attempts using a nominally transverse field H_{\perp} were empirically found inferior to additionally adjusting the direction of the field, here rotated somewhat toward the pinning direction for the RL. Using a mechanically positioned permanent magnet as a field source, this field rotation was only crudely estimated at the time to be $\sim 20^\circ - 30^\circ$ (see also Sec. IV). With both H and bias-point θ_{bias} “optimized” as such, an I_e series of PSD($f; I_e$) were measured, after which the bias resistance R_{bias} , and finally R_{min} and R_{max} were measured at a common (low) bias of -10 mV to determine r_{bias} [as in Eq. (4)].

The key feature of the rms PSD($f; I_e$) in Fig. 3(c) is that these measured spectra (excluding $I_e = +1.5$ mA) appear essentially independent of both the polarity and magnitude of

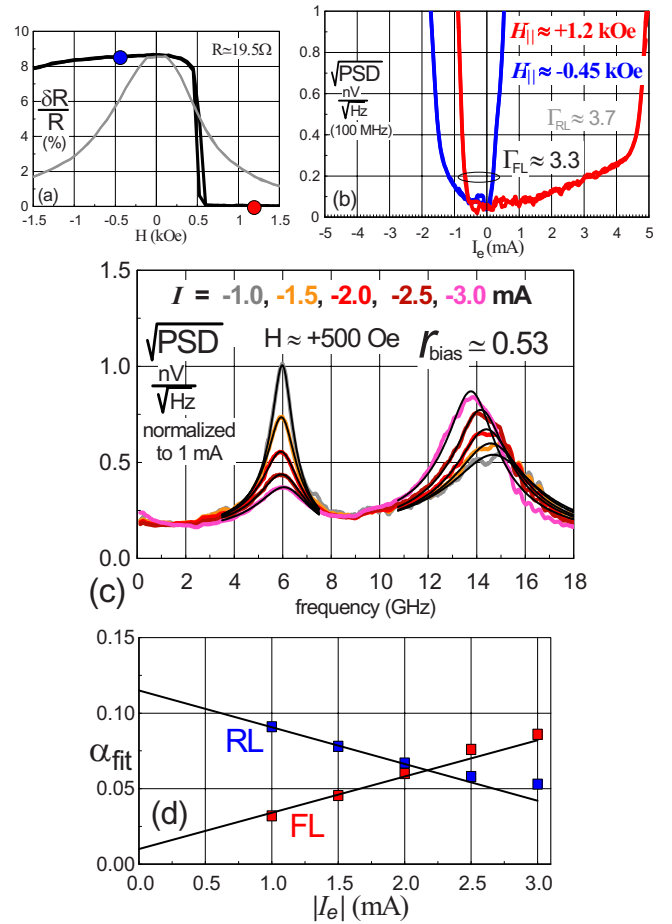


FIG. 5. (Color online) Measurement set for a different (but nominally identical) 60 nm device as that shown in Figs. 3 and 4. (c) rms spectra (with least-squares fits) measured at *larger* r_{bias} and $\theta_{\text{bias}} > \theta_{\text{magic}}$. (d) I_e -dependent values of $\alpha_{\text{fit}}(I_e)$ for FL and RL, with suggested $I_e \rightarrow 0$ extrapolation lines.

I_e (after 1 mA normalization), defining a “universal” spectrum curve over the entire 18 GHz bandwidth, including the *unexpectedly wide, low-amplitude* RL-FMR peak near 14 GHz (more on this below). Because of the relatively large $\sqrt{\text{PSD}(f; I_e=0)} \sim 1$ nV/ $\sqrt{\text{Hz}}$ background, these RL peaks were not well discernible during raw spectrum measurements, and were practically revealed only after electronics background noise subtraction. As suggested in Fig. 3(c), eventual breakdown of the magic-angle condition was generally found to first occur from spin-torque instability of the FL at larger *positive* I_e .

The spectra in Fig. 4 shows the equivalent set of measurements on a physically different (though nominally identical) 60 nm device. They are found to be remarkably alike in all properties to those of Fig. 3, providing additional confirmation that the magic-angle method can work on real nanoscale structures *to directly obtain the intrinsic* $S_{\theta}(f; I_e=0)$ *in the absence of spin-torque effects*. This appears further confirmed by the close agreement of measured ($r_b, \Gamma_{\text{Cu/FL}}$) pairs (from data of Figs. 3, 4, and 6) and the macrospin model predictions described in Fig. 2.

To obtain values for linewidth $\Delta\omega$ and then damping α from the measured PSD($f; I_e$), regions of spectra several gi-

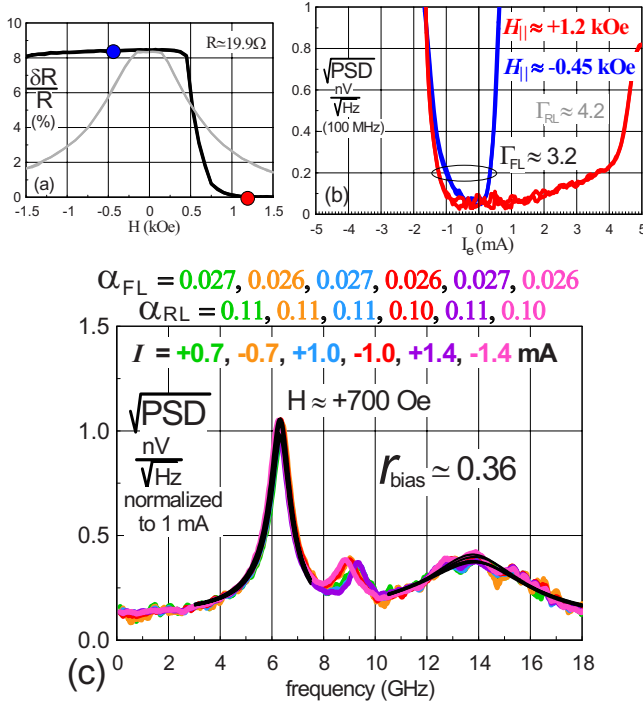


FIG. 6. (Color online) Analogous measurement set as in Figs. 3 and 4, for (an otherwise identical) device with a 10 A Dy-cap layer in direct contact with the FL.

gahertz wide, surrounding the FL and RL-FMR peaks are each nonlinear least-squares fitted to the functional form for $S_\theta(f; I_e=0)$ in Eq. (5). In particular, the fitting function is taken to be

$$S_V\left(f = \frac{\omega}{2\pi}\right) = S_0 \frac{\omega_0^2[\omega_0^2 + (H'_{y'y'} / H'_{z'z'})\omega^2]}{(\omega^2 - \omega_0^2)^2 + (\omega\Delta\omega)^2}$$

with

$$\omega_0 = \gamma\sqrt{H'_{y'y'}H'_{z'z'}}, \quad \Delta\omega = \gamma\alpha_{\text{fit}}(H'_{z'z'} + H'_{y'y'})$$

and

$$H'_{y'y'} \rightarrow [(\omega_{\text{peak}}/\gamma)^2 + (\alpha_{\text{fit}}H'_{z'z'})^2/2]/H'_{z'z'}. \quad (7)$$

Three fitting parameters are used: $S_0 = S_V(f=0)$, α_{fit} , and ω_{peak} , the latter being already well defined by the data itself. The substitution for $H'_{y'y'}$ is accurate to order α^2 , leaving $H'_{z'z'}$, as yet unknown. With $H'_{z'z'} \gg H'_{y'y'}$, dominated by out-of-plane demagnetizing fields, $S_V(f)$ depends mostly on the product $\alpha_{\text{fit}}H'_{z'z'}$. For simplicity, fixed values $H'_{z'z'}^{\text{FL}} = 8$ kOe and $H'_{z'z'}^{\text{RL}} = 10$ kOe, were used here, based on macrospin calculations that approximately account for device geometry and net $M_s t$ product for FL and RL films. The fitted PSD($f; I_e$) curves, and the values obtained for $\alpha_{\text{fit}}^{\text{FL}}$ and $\alpha_{\text{fit}}^{\text{RL}}$ are also included in Figs. 3(c) and 4(c). These values are notably *independent* of (or show no significant trend with) I_e .

Although the $\alpha_{\text{fit}}^{\text{FL}} \approx 0.01$ repeatedly found from these data is a quite typical magnitude for Gilbert damping in CoFe alloys, the *extremely large*, $\sim 10\times$ greater value of $\alpha_{\text{fit}}^{\text{RL}} \approx 0.1$ is quite noteworthy since the RL and FL are not too

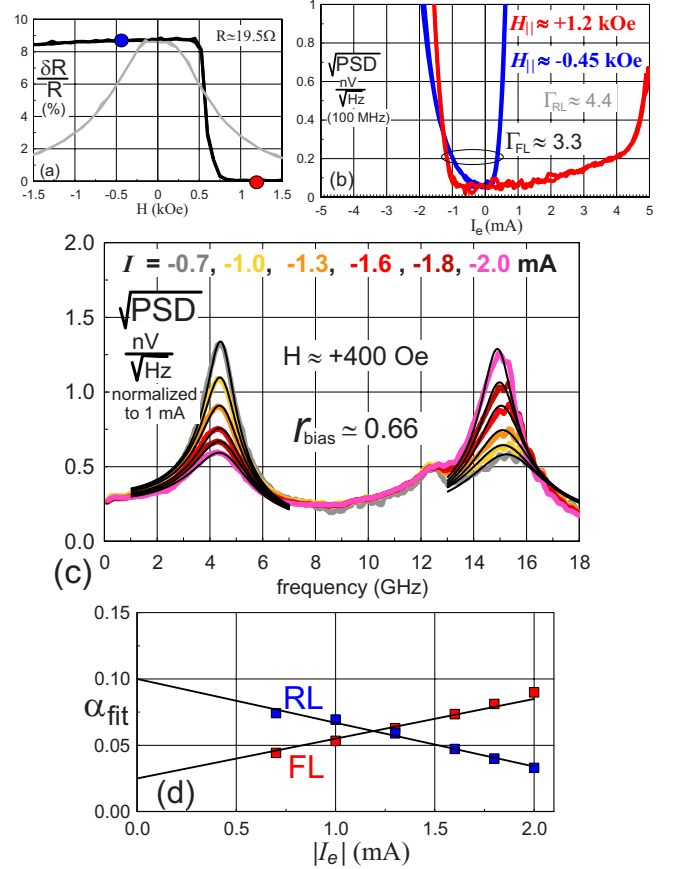


FIG. 7. (Color online) Analogous measurement set as in Fig. 5 for a different (but nominally identical) device as that in Fig. 6 with a 1nm Dy cap layer.

dissimilar in thickness and composition. Although the small amplitude of the RL-FMR peaks in Figs. 3 and 4 (everywhere below the raw $1 \text{ nV}/\sqrt{\text{Hz}}$ electronics noise), may suggest a basic unreliability in this fitted value for $\alpha_{\text{fit}}^{\text{RL}}$, this concern is seemingly dismissed by the data of Fig. 5. Measured on a third (nominally identical) device, an alternative “extrapolation method” was used, in which the applied field was purposefully reduced in magnitude (and more transversely aligned than for magic-angle measurements) to increase r_{bias} and thus align \hat{m}_{FL} to be more antiparallel to \hat{m}_{RL} . As a result, spin-torque effects at larger negative I_e will decrease $\Delta\omega$ and concomitantly *enhance* RL-FMR peak amplitude (and vice versa for the FL), bringing this part of the measured spectrum above the raw electronics noise background.

Using the same fitting function from Eq. (7), it is now necessary to extrapolate the $\alpha_{\text{fit}}^{\text{RL}}(I_e)$ to $I_e \rightarrow 0$ [Fig. 5(d)] in order to obtain the intrinsic damping. This method works well in the case of the RL since $d\alpha_{\text{fit}}^{\text{RL}}/d|I_e| < 0$ and the extrapolated $I_e=0$ intercept value of $\alpha_{\text{fit}}^{\text{RL}}$ is necessarily *larger* than the measured $\alpha_{\text{fit}}^{\text{RL}}(I_e)$, and hence will be (proportionately) less sensitive than uncertainty in the estimated extrapolation slope. As can be seen from Fig. 5(d), the *extrapolated values for intrinsic* α_{RL} are virtually identical to those obtained from the data of Figs. 3 and 4. The extrapolated α_{FL} is also quite consistent as well. The extrapolation data also

TABLE I. Summary of critical voltages (measured over ≈ 8 devices each) and damping parameter values α for the present experiment. Estimated statistical uncertainty in the α values is $\sim 10\%$.

Stack	$-RI_{\text{FL}}^{\text{crit}}$ (mV)	α_{FL}	α_{RL}
Control	10.4 ± 0.1	0.011	0.11
Dy cap	24.5 ± 0.5	0.026	0.11
SF-FL	44.9 ± 2.0	0.011	0.12

confirm the expectation [noted earlier following Eq. (5)] that linewidth $\Delta\omega$ will vary *linearly* with I_e .

Comparing with Figs. 3(c) and 4(c), the spectra in Fig. 5(c) illustrate the profound effect of spin torque on altering the linewidth and peak height of both FL and RL-FMR peaks even if the system is only moderately misaligned from the magic-angle condition. By contrast, for other frequencies [where the $\Delta\omega$ term in the denominator of Eq. (5) is unimportant], the 1 mA normalized spectra are *independent* of I_e . Being consistent with Eq. (5), this appears to verify that this second form of fluctuation-dissipation theorem remains valid despite that the system of Eq. (1) is not in thermal equilibrium¹⁰ at nonzero I_e . [Alternatively stated, spin torques lead to an *asymmetric* \vec{H} but do not alter the damping tensor \vec{D} in Eq. (1).] The α proportionality in the prefactor of $S_\theta(f)$ in Eq. (5) relatedly shows that the effect of spin torque on $\Delta\omega$ is *not* equivalent to additional damping (positive or negative) as may be commonly misconstrued. The data further indicate that Oersted-field effects, or other I_e -dependent terms in \vec{H} not contributing to $\Delta\omega$, are insignificant in this experiment.

Analogous to Figs. 4 and 5, the data of Figs. 6 and 7 are measured on CPP-GMR SV stacks differing only by an additional 1-nm-thick Dy-cap layer deposited directly on top of the FL. The use of Dy in this context (presumed spin pumping from FL to Dy but possibly including Dy intermixing near the FL/Dy interface¹¹) was found in previous work¹² to result in an $\sim 3\times$ increase in FL damping, then inferred from the $\sim 3\times$ increase in measured $|I_{\text{FL}}^{\text{crit}}|$. Here, a more direct measure from the FL FMR linewidth indicates a roughly similar, $\approx 2.3\times$ increase in α_{FL} (now using somewhat thicker FL films). This ratio is closely consistent with that inferred from $|I_{\text{FL}}^{\text{crit}}|$ data measured in this experiment over a population of devices (see Table I). Notably, the values found for α_{RL} remain virtually the same as before.

Finally, Fig. 8 shows results for a synthetic-ferrimagnet (SF) free layer of the form FL1/Ru(8Å)/FL2. The Ru spacer provides an interfacial *antiferromagnetic* coupling of ≈ 1.0 erg/cm². Here, FL1 has a thicker CoFeGe layer than used for prior FL films, and FL2 is a relatively thin CoFe layer chosen so that $(M_s t)_{\text{FL1}} - (M_s t)_{\text{FL2}} \approx (M_s t)_{\text{FL}} \approx 0.64$ erg/cm². Although having similar static M - H or R - H characteristics to that of the simple FL (of similar *net* $M_s t$ product) used in earlier measurements, the transport of the SF-FL in regard to spin-torque effects in particular is fundamentally distinct. The basic physics of this phenomenon was described in detail previously.¹³ In summary, a spin-torque-

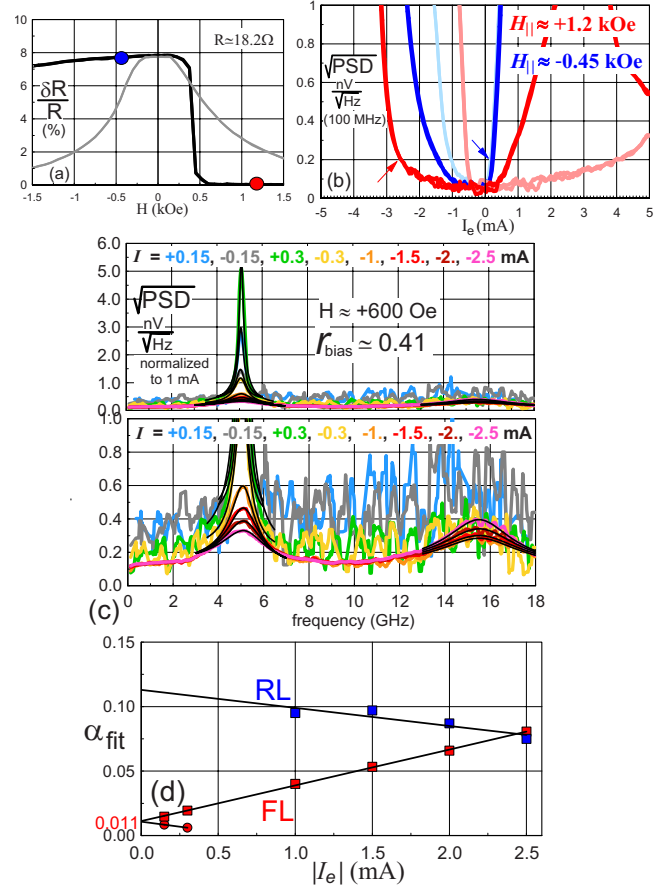


FIG. 8. (Color online) Analogous measurement set as in Fig. 5, for (an otherwise identical) device with a synthetic-ferrimagnet FL (SF-FL) as described in text. (b) includes for comparison N - I_e loops (in lighter color) from Fig. 3(b); arrows show SF-FL I_{crit} for P state (red) and AP state (blue). (c) spectral data and fits are repeatedly shown (for clarity) using two different ordinate scales. (d) Square or circular symbols denote measurements with negative or positive I_e , respectively.

induced quasicore resonance between the two natural oscillation modes of the FL1/FL2 couple in the case of *negative* I_e and $\hat{m}_{\text{FL1}} \cdot \hat{m}_{\text{FL2}} > 0$, can act to transfer energy out of the mode that is destabilized by spin torque, thereby delaying the onset of criticality and substantially increasing $|I_{\text{P-FL}}^{\text{crit}}|$. Indeed, the side-by-side comparison of N - I_e loops provided in Fig. 8(b) indicate a *nearly* $5\times$ increase in $|I_{\text{P-FL}}^{\text{crit}}|$, despite that $|I_{\text{AP-FL}}^{\text{crit}}|$ remains virtually unchanged.

For the SF-FL devices, attempts at finding the magic angle under similar measurement conditions as used for Figs. 3(c), 4(c), and 6(c) were not successful, and so the extrapolation method at similar $r_{\text{bias}} \approx 0.4$ was used instead. To improve accuracy for extrapolated α_{FL} , the data of Fig. 8(c) include measurements for $|I_e| \leq 0.3$ mA (so that $I_e < I_{\text{FL}}^{\text{crit}}$) for which electronics noise overwhelms the signal from the RL FMR peaks. Showing excellent linearity of $\alpha_{\text{fit}}^{\text{FL}}$ vs. I_e over a wide I_e range, the extrapolated intrinsic $\alpha_{\text{FL}} \approx 0.01$ is, as expected, unchanged from before. The same is true for the extrapolated α_{RL} as well.

Table I summarizes the mean critical voltages $-RI_{\text{P-FL}}^{\text{crit}}$ (less sensitive than $I_{\text{P-FL}}^{\text{crit}}$ to lithographic variations in actual

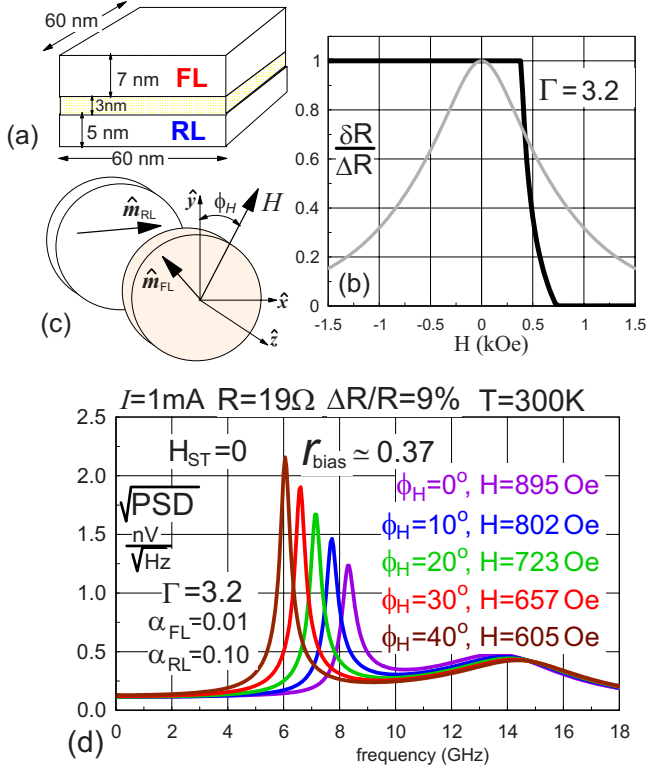


FIG. 9. (Color online) Two-macrospin model results. (a) Cartoon of model geometry. (b) Simulated δR - H loops analogous to data of Figs. 3–8. (c) Cartoon defining vector orientations (RL exchange pinned along $+x$ direction). (d) Simulated rms PSD assuming parameter values indicated, with variable $|H|$ to maintain a fixed r_{bias} at each ϕ_H (as indicated by color).

device area) from a larger set of PSD- J_e measurements. The $\approx 2.3 \times$ increase in $|RI_{\text{P-FL}}^{\text{crit}}|$ with the use of the Dy cap is in good agreement with that of the ratio of measured α_{FL} .

IV. MICROMAGNETIC MODELING

For more quantitative comparison with experiment than afforded by the one-macrospin model of Sec. II, a two-macrospin model equally treating both \hat{m}_{RL} and \hat{m}_{FL} is now considered here as a simpler, special case of a more general micromagnetic model to be discussed below. The values $M_s^{\text{FL}} = 950$ emu/cc, $t_{\text{FL}} = 7$ nm, $M_s^{\text{RL}} = 1250$ emu/cc, and $t_{\text{RL}} = 5$ nm will be used as simplified, combined representations (of similar thickness and $M_s t$) to the actual CoFe/CoFeGe multilayer films used for the RL and FL. The magnetic films are geometrically modeled as 60 nm squares which (in the macrospin approximation) have zero shape anisotropy (as do circles) but allow analytical calculation of all magnetostatic interactions. The effect of IrMn exchange pinning on the RL is simply included as a uniform field $\mathbf{H}_{\text{pin}} = [J_{\text{pin}} / (M_s t)_{\text{RL}}] \hat{x}$ with measured $J_{\text{pin}} \approx 0.75$ erg/cm². First, Fig. 9(b) shows simulated $r_{\text{bias}}-H_{\parallel}$ and $r_{\text{bias}}-H_{\perp}$ curves computed assuming $\Gamma = 3.2$, roughly the mean value found from the $J_{\text{FL}}^{\text{crit}}$ data of Sec. III. The agreement with the shape of the measured R - H is very good (e.g., Figs. 6 and 7, in particular), which reflects how remarkably closely these actual de-

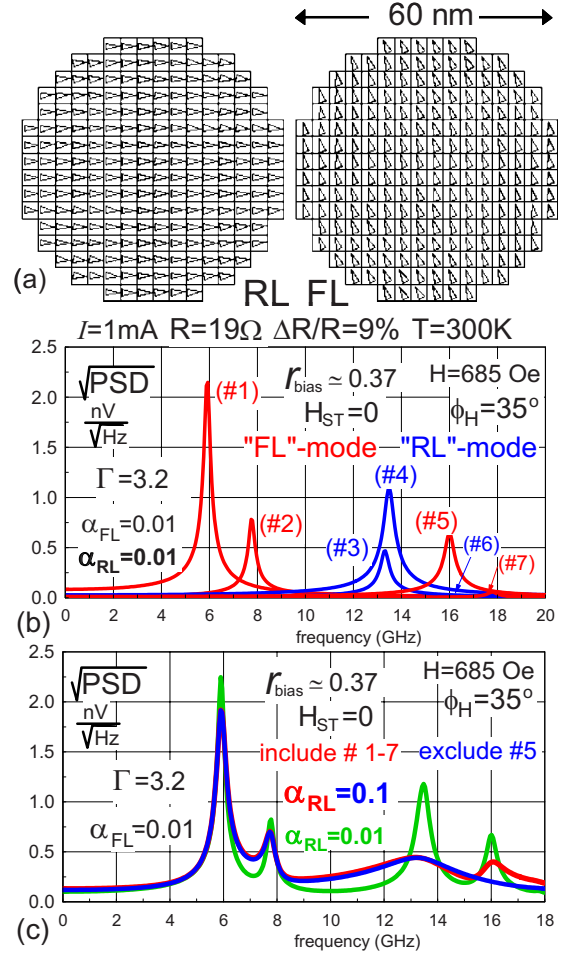


FIG. 10. (Color online) Micromagnetic model results. (a) Cell discretizations with arrow heads showing magnetization orientation when $|H| = 685$ Oe and $\phi_H = 35^\circ$ [see Fig. 9(c)]. (b) simulated partial rms PSD for first seven eigenmodes (as labeled) computed individually with $\alpha_{\text{FL}} = 0.01$ and $\alpha_{\text{RL}} = 0.01$; other parameter values indicated. (c) Simulated total rms PSD with $\alpha_{\text{FL}} = 0.01$ and $\alpha_{\text{RL}} = 0.01$ (green) or $\alpha_{\text{RL}} = 0.1$ (red or blue); blue curve excludes contribution from fifth (FL) eigenmode at 16 GHz.

vices resemble idealized (macrospin) behavior.

Next, Fig. 9(d) shows simulated PSD curves $S_V(f)$ computed (see Appendix) in the absence of spin torque (i.e., $H_{\text{ST}} = 0$) but otherwise assuming typical experimental values $R = 19 \Omega$, $\Delta R / R = 9\%$, and $T = 300$ K, as well as $\alpha_{\text{FL}} = 0.01$ and $\alpha_{\text{RL}} = 0.1$, so to be compared with the magic-angle spectra of Figs. 3 and 4. Since (as stated in Sec. III) the experimental field angle was not accurately known, the field angle ϕ_H was varied systematically for the simulations, and in each case the field magnitude H was iterated until $r_{\text{bias}} \approx 0.37$, approximately matching the mean measured value. In terms of both absolute values and the ratio of FL to RL-FMR peak amplitudes, the location of f_{peak} (particularly for the FL), and the magnitude of H (on average 650–700 Oe from the three magic-angle data in Sec. III), the best match with experiment clearly occurs with $30^\circ \leq \phi_H \leq 40^\circ$. The agreement, both qualitatively and quantitatively, is again remarkable given the simplicity of the two-macrospin model.

Finally, results from a discretized micromagnetic model are shown in Fig. 10. Based on Fig. 9, the value $\theta_H=35^\circ$ was fixed, and $H=685$ Oe was determined by iteration until $r_{\text{bias}} \approx 0.37$. The equilibrium bias-point magnetization distribution is shown in Fig. 10(a) for this 416-cell model. Estimated values for exchange stiffness, $A_{\text{FL}}=1.4$ $\mu\text{erg}/\text{cm}$ and $A_{\text{RL}}=2$ $\mu\text{erg}/\text{cm}$ were assumed. The simulated spectra in Fig. 10(b) are shown *one eigenmode at a time* (see Appendix), for the seven eigenmodes with predicted FMR frequencies below 20 GHz (the eighth mode is at 22.9 GHz). Although all modes involve to some degree coupled motion of both FL and RL, the first, second, fifth, and seventh modes involve mostly FL motion, and the nearly degenerate third and fourth modes (and the sixth) mostly that of the RL. [The amplitudes of $S_V(f)$ from the sixth or seventh mode are negligible.] For illustration purposes only, Fig. 10(b) assumed identical damping $\alpha_{\text{FL}}=\alpha_{\text{RL}}=0.01$ in each film.

For Fig. 10(c), the computation of $S_V(f)$ is more properly computed using either six or all seven eigenmodes simultaneously, which includes damping-induced coupling between the modes. Including higher-order modes makes negligible change to $S_V(f < 20$ GHz) (but rapidly increases computation time). As was observed earlier, the agreement between simulated and measured spectra in Figs. 3(c) and 4(c) is quite good (with $\alpha_{\text{RL}}=0.1$), and the simulations now include the small, secondary FL peak near 8 GHz clearly seen in the measured data [including that of Fig. 6(c)], though it is somewhat more pronounced in the model results. Notably, the computed spectrum near the RL-FMR peak more resembles the measurements after removing the 16 GHz fifth mode from the calculation, as this (FL) mode does not appear to be physically present in the Figs. 3(c) and 4(c) spectra.

While it is perhaps expected that higher-order modes in a micromagnetic simulation assuming perfectly homogeneous magnetic films would show deviations from real devices with finite grain size, edge roughness/damage, etc., the situation is actually more interesting. Figure 11 shows measured spectra on yet another device (again, nominally identical to that of Figs. 3–5) in which the experiment was perhaps slightly off from the optimum magic-angle condition, as evidenced by the very small shift in the 6 GHz FL FMR peak position with polarity of I_e . More noteworthy, however, is the clear polarity asymmetry and nonlinear in I_e peak amplitude (for $I_e > 0$, in particular) of *both* the 8 GHz secondary FL mode *and* a higher-order mode close to 15 GHz. Both resemble typical spin-torque effect at angles more antiparallel than the magic-angle condition. This similarity in behavior indicates with near certainty that this 15 GHz mode is also FL-like in origin, and is thus a demonstration of the “missing” fifth mode predicted in Fig. 10. [In hindsight, there is discernible a small but similar 15 GHz peak in the spectra of Fig. 4(c).] It is worth remembering that the magic-angle argument was based on a simple one-macrospin model, and so remarkably there appears to be circumstances (e.g., Figs. 3(c)–7(c) where this “spin-torque null” actually does apply simultaneously to both the FL and RL, as well as to higher-order modes.

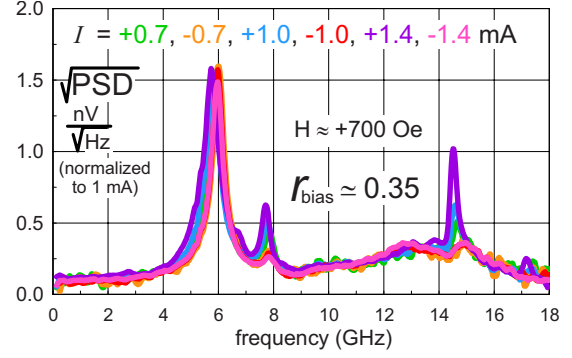


FIG. 11. (Color online) The rms PSD measured on a physically different (but nominally identical) device as that generating the analogous magic-angle spectra shown in Figs. 3(c) and 4(c).

V. DISCUSSION

In addition to direct observation from the measured spectral linewidth in Figs. 3–8, evidence for large Gilbert damping $\alpha_{\text{RL}} \gg \alpha_{\text{FL}}$ for the RL is also seen in the I_e^{crit} data. As ratios $I_{\text{P-RL}}^{\text{crit}}/I_{\text{P-FL}}^{\text{crit}}$ and $I_{\text{AP-RL}}^{\text{crit}}/I_{\text{AP-FL}}^{\text{crit}}$ are (from Figs. 3–5 data) both roughly ~ 7 , this conclusion is consistent with the basic scaling [from Eq. (3c)] that $I_e^{\text{crit}} \propto \alpha$. This, as well as the substantial, $2\text{--}3\times$ variation in $\alpha_{\text{fit}}^{\text{RL}}$ with I_e in Figs. 5(d) and 7(d), appears to rather conclusively (and expectedly) confirm that inhomogeneous broadening is not a factor in the large linewidth-inferred values of α_{RL} found in these *nanoscale* spin valves.

Large increases in effective damping of “bulk” samples of ferromagnetic (FM) films in contact with antiferromagnet (AF) exchange pinning layers has been reported previously.^{14–16} The excess damping was generally attributed to two-magnon scattering processes¹⁷ arising from an inhomogeneous AF/FM interface. However, the two-magnon description applies to the case where the uniform, ($\mathbf{k}=0$, $\omega \equiv \omega_0$) mode is pumped by a external rf source to a high excitation (magnon) level, which then transfers energy via two-magnon scattering into a large (quasicontinuum) number of *degenerate* ($\mathbf{k} \neq 0$, $\omega_{\mathbf{k}} = \omega_0$) spin-wave modes, all with low (thermal) excitation levels and mutually coupled by the same two-magnon process. In this circumstance, the probability of energy transfer back to the uniform mode (just one among the degenerate continuum) is negligible, and the resultant one-way flow of energy out of the uniform mode resembles that of intrinsic damping to the lattice. By contrast, for the nanoscale spin-valve device, the relevant eigenmodes (Fig. 10) are discrete and generally *nondegenerate* in frequency. Even for a coincidental case of a quasidegenerate pair of modes (e.g., RL modes #3 and #4 in Fig. 10), both modes are equally excited to thermal-equilibrium levels (as are all modes), and have similar intrinsic damping rates to the lattice. Any additional energy transfer via a two-magnon process should flow both ways, making impossible a large (e.g., $\sim 10\times$) increase in the effective net damping of either mode.

Two alternative hypotheses for large α_{RL} which are essentially independent of device size are 1) large spin-pumping effect at the IrMn/RL interface, or 2) strong interfacial exchange coupling at the IrMn/RL resulting in nonresonant

TABLE II. Summary of bulk film FMR measurements by Mewes (Ref. 18) for reduced film stack structure: seed/IrMn(t_{AF})/Cu(t_{Cu})/RL/Cu(3nm)/cap. Removal of IrMn, or alternatively a lack of proper seed layer and/or use of a sufficiently thick $t_{Cu} \approx 3$ nm can each effectively eliminate exchange pinning strength to RL.

Sample type	$\alpha_{RL} = \frac{\sqrt{3}}{2} \gamma (d\Delta H/d\omega)$
$t_{AF}=t_{Cu}=0$	0.011
$t_{AF}=6$ nm, $t_{Cu}=0$ (no seed layer for IrMn)	0.010
$t_{AF}=6$ nm, $t_{Cu}=3$ nm	0.013
$t_{AF}=t_{Cu}=0$ (out-of-plane FMR)	0.013

coupling to high-frequency modes in either the RL *and/or* the IrMn film. However, these two alternatives can be distinguished since the exchange-coupling strength can be greatly altered without necessarily changing the spin-pumping effect. In particular, α_{RL} was very recently measured by conventional FMR methods by Mewes¹⁸ on bulk film samples (grown by us with the same RL films and IrMn annealing procedure as that of the CPP-GMR SV devices reported herein) of the reduced stack structure: seed/IrMn(t_{AF})/Cu(t_{Cu})/RL/Cu(30 A)/cap. For all four cases described in Table II, the exchange coupling was deliberately reduced to zero, and the measured $\alpha_{RL} \approx 0.012$ was found to be nearly identical to that found here for the FL of similar CoFeGe composition. However, for the two cases with $t_{AF}=60$ A, excess damping due to spin pumping of electrons from RL into IrMn should not have been diminished (e.g., the spin-diffusion length in Cu is $\sim 100\times$ greater than $t_{Cu} \approx 3$ nm). This would appear to rule out the spin-pumping hypothesis.

The second hypothesis emphasizes the possibility that the energy loss takes place inside the IrMn, from oscillations excited far off resonance by locally strong interfacial exchange coupling to a fluctuating \hat{m}_{RL} . This local interfacial exchange coupling J_{ex} can be much greater than J_{pin} since the latter reflects a surface average over inhomogeneous spin alignment (grain-to-grain and/or from atomic roughness) within the IrMn sublattice that couples to the RL. Further, though such strong but inhomogeneous coupling cannot truly be represented by a uniform H_{pin} acting on the RL, the similarity between measured and modeled values of ~ 14 GHz for the “uniform” RL eigenmode has clearly been demonstrated here. Whatever are the natural eigenmodes of the real device, the magic-angle spectrum measurements of Sec. III reflect the thermal excitation of all eigenmodes for which “one-way” intermodal energy transfer should be precluded by the condition of thermal equilibrium and the orthogonality¹⁹ of the modes themselves. Hence, without an additional energy sink *exclusive* of the RL/FL spin-lattice system, the linewidth of all modes should arguably reflect the intrinsic Gilbert damping of the FL or RL films, which the data of Sec. III and Table II indicate are roughly equal with $\alpha \sim 0.01$. Inclusion of IrMn as a combined AF/RL system, would potentially provide that extra energy loss channel for the RL modes.

A rough plausibility argument for the latter hypothesis may be made with the following crude AF/FM model. For

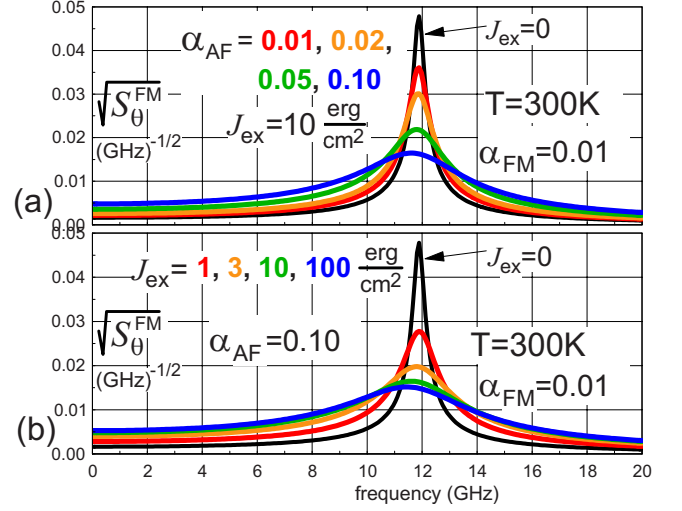


FIG. 12. (Color online) Simulated rms PSD $S_{\theta}^{FM}(f)$ for a three-macrospin model of an AF/FM couple as described via Eq. (8) and in the text. The FM film parametrics are the same as used for macrospin RL model in Fig. 9, with $\alpha_{FM}=0.01$ and $J_{pin}=0.75$ erg/cm². (a) Varied α_{AF} (denoted by color) with $J_{ex}=10$ erg/cm². (b) Varied J_{ex} (denoted by color) with $\alpha_{AF}=0.1$. The black curve in (a) or in (b) corresponds to $J_{ex}=0$. For AF, M_s is taken to be 500 emu/cc.

this purpose, the (two-sublattice) AF film is modeled as two ferromagnetic layers (#1 and #2) occupying the *same* physical location.²⁰ Excluding magnetostatic contributions, the free energy/area for this three-macrospin system is taken to be

$$(M_s t)_{AF} H_{AF} \hat{m}_1 \cdot \hat{m}_2 - \frac{1}{2} (K t)_{AF} [(\hat{m}_1 \cdot \hat{x})^2 + (\hat{m}_2 \cdot \hat{x})^2] - J_{ex} \hat{m}_2 \cdot \hat{m}_{FM} + [J_0 - J_{pin}] \hat{m}_{FM} \cdot \hat{x}. \quad (8)$$

For IrMn with Neel temperature of $T_N \approx 700$ K, the internal AF exchange field $H_{AF} \sim k_B T_N / \mu_B \sim 10^7$ Oe.²¹ With $t_{AF} = 60$ A, AF uniaxial anisotropy is estimated to be $K_{AF} \sim 10^6$ erg/cc.²² A rough estimate $J_{ex}^{max} \sim 8(A/t)_{FM}$ defining a limiting value of *strong* interfacial exchange is obtained by equating interface energy $J_{ex} \bar{\phi}^2/2$ to the bulk exchange energy $4A \bar{\phi}^2/t$ of a hypothetical, small-angle Bloch wall ($0 \leq \phi \leq 2\bar{\phi}$) twisting through the FM film thickness. Taking $t \approx 5$ nm and $A \sim 10^{-6}$ erg/cm yields $J_{ex}^{max} \sim 20$ erg/cm². The value of $J_0 \sim [1/J_{ex} + 1/(K t)_{AF}]^{-1}$ in the last field-like term in Eq. (8) is more precisely determined (iteratively) to maintain a constant eigenfrequency for the FM layer *independent* of J_{ex} or K_{AF} , thus accounting for the weaker inhomogeneous coupling averaged over an actual AF/FM interface.

As shown in Fig. 12, this crude model can explain a $\sim 10\times$ increase in the FM linewidth provided $J_{ex} \geq 5-10$ erg/cm² and $\alpha_{AF} \geq 0.05-0.1$. It is worthwhile to note²¹ that for the two-sublattice AF, the linewidth $\Delta\omega/\omega_0 \approx \alpha_{AF} \sqrt{2H_{AF}} / [H_K \equiv (K/M_s)_{AF}]$ is larger by a factor of $\sqrt{2H_{AF}/H_K} \sim 100$ compared to high order exchange-

dominated FM spin-wave modes in cases of comparable α and ω_0 (with $\omega_0 \approx \gamma\sqrt{2H_{\text{AF}}H_K} \sim 10^{12}$ Hz for the AF). Since the *lossy* part of the low-frequency susceptibility for FM or AF modes scales with $\Delta\omega$, it is suggested that the IrMn layer can effectively sink energy from the ~ 14 GHz RL mode despite the $\sim 100\times$ disparity in their respective resonant frequencies. *Size-independent* damping mechanisms for FM films exchange coupled to AF layers such as IrMn are worthy of further, detailed study.

ACKNOWLEDGMENTS

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APPENDIX

As was described in detail elsewhere,²³ the generalization of Eq. (1) and (5) from a single macrospin to that for an N -cell micromagnetic model takes the form

$$\begin{aligned} (\vec{D} + \vec{G}) \cdot \frac{d\vec{m}'}{dt} + \vec{H}' \cdot \vec{m}' &= \vec{h}'(t), \\ \vec{S}(\omega) &\equiv \frac{2k_B T}{\gamma\Delta m} \vec{\chi} \cdot \vec{D} \cdot \vec{\chi}^\dagger, \quad \vec{\chi}(\omega) = [\vec{H}' - i\omega(\vec{D} + \vec{G})]^{-1}, \end{aligned} \quad (\text{A1})$$

where \vec{m}' (or \vec{h}') is an $2N \times 1$ column vector built from the N 2D vectors $\vec{m}'_{j=1\dots N}$, and \vec{D} , \vec{G} , and \vec{H}' are $2N \times 2N$ matrices formed from the $N \times N$ array of 2D tensors \vec{D}_{jk} , \vec{G}_{jk} , and \vec{H}_{jk} . Here, $\vec{D}_{jk} = \vec{D}\delta_{jk}$ and $\vec{G}_{jk} = \vec{G}\delta_{jk}$, though \vec{H}_{jk} is *nonlocal* in cell indices j, k due to exchange and (particularly) the magnetostatic interaction.

The PSD $S_Q(f)$ for any scalar quantity $Q(\{\hat{m}_j\})$ is²³

$$S_Q(f) = 2 \sum_{j,k=1}^N \vec{d}'_j \cdot \vec{S}_{jk}(\omega) \cdot \vec{d}'_k, \quad \vec{d}'_j \equiv \frac{\partial m'_j}{\partial \hat{m}_j} \cdot \frac{\partial Q}{\partial \hat{m}_j}. \quad (\text{A2})$$

The computations for the PSD of Figs. 9 and 10 took $Q(\vec{m})$ to be

$$Q = \frac{1}{N_i} \sum_{i=1}^{N_i} \frac{I_{\text{bias}} \Delta R (1 - \hat{m}_i^{\text{RL}} \cdot \hat{m}_i^{\text{FL}})}{\Gamma + 1 + (\Gamma - 1) \hat{m}_i^{\text{RL}} \cdot \hat{m}_i^{\text{FL}}} \quad (\text{A3})$$

averaged over the $N_i = N/2$ cell pairs at the RL-FL interface.

For a *symmetric* \vec{H} (e.g., $H_{\text{ST}}=0$), the set of eigenvectors $\vec{e} \leftarrow \vec{m}$ of the system [Eq. (A1)] can be defined from the following eigenvalue matrix equation:

$$(\vec{G}^{-1} \cdot \vec{H}') \cdot \vec{e}_{n=1\dots 2N} = i\omega_n \vec{e}_n. \quad (\text{A4})$$

The eigenvectors come in N complex-conjugate pairs \vec{e}^+, \vec{e}^- with real eigenfrequencies $\pm\omega$. With suitably normalized \vec{e}_n , matrices $H'_{mn} = \delta_{mn}$ and $G_{mn} = \delta_{mn}/i\omega_n$ are diagonal in the eigenmode basis.²³ The analog to Eq. (A1) becomes

$$[\chi^{-1}(\omega)]_{mn} = (1 - \omega/\omega_n) \delta_{mn} - i\omega(D_{mn} \equiv \vec{e}_m^* \cdot \vec{D} \cdot \vec{e}_n),$$

$$S_{mn}(\omega) \equiv \frac{2k_B T}{\gamma\Delta m} \sum_{m',n'} \chi_{mm'}(\omega) D_{m'n'} \chi_{nn'}^*(\omega),$$

$$S_Q(f) = 2 \sum_{m,n} d_m'^* S_{mn}(\omega) d_n', \quad d_n' \equiv \vec{e}_n^* \cdot \vec{d}'. \quad (\text{A5})$$

The utility of eigenmodes for computing PSD, e.g. in the computations of Fig. 10, is that only a *small* fraction (e.g., 7 rather than 416 eigenvector pairs) need be kept in Eq. (A5) (with all the rest simply ignored) in order to obtain accurate results in practical frequency ranges (e.g., <20 GHz). Despite that D_{mn} is (in principle) a full matrix, the reduction in matrix size for the matrix inversion to obtain $\chi(\omega)$ at *each frequency* more than makes up for the cost of computing the (ω_n, \vec{e}_n) which need be done only once independent of frequency or α values.

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RL and FL. For simplicity, these are here simply lumped into the Gilbert damping parameter α for either RL or FL. In this context, “intrinsic” damping refers approximately to that in the limit $I_e \rightarrow 0$, but also in the presence of the complete stack structure of the spin-valve device of interest.

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- ¹⁹A nonzero damping matrix $\vec{\vec{D}}$ can technically cause a coupling of nondegenerate eigenmodes if $\vec{e}_{m \neq n}^* \cdot \vec{\vec{D}} \cdot \vec{e}_n \neq 0$, though this does not change the argument citing this footnote. This small effect can be seen in Fig. 10(c) as the slight change in the 6 GHz FL FMR peak height and linewidth when α_{RL} is grossly varied from 0.01 to 0.1.
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