Perovskite manganites hosting versatile multiferroic phases with symmetric and antisymmetric exchange strictions

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Complete magnetoelectric (ME) phase diagrams of orthorhombic $RMnO_3$ with and without magnetic moments on the R ions have been established. Three kinds of multiferroic ground states, the *ab*-cycloidal, the *bc*-cycloidal, and the collinear *E*-type phases, have been identified by the distinct ME responses. The electric polarization of the *E*-type phase dominated by the symmetric spin exchange $(S_i \cdot S_j)$ is more than ten times as large as that of the *bc*-cycloidal phase dominated by the antisymmetric one $(S_i \times S_j)$ and the ME response is enhanced near the bicritical phase boundary between these multiferroic phases of different origins. These findings may provide generic features of the magnetically induced multiferroics.

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It has been a long standing problem how to enhance the correlation between magnetism and ferroelectricity in a solid. Possible solutions can be found in recent studies on the magnetically induced multiferroics,^{1,2} which can be classified into two types; one is driven by antisymmetric exchange striction in the cycloidal spin structure, typified by TbMnO₃,^{3–5} and the other is driven by symmetric exchange striction in the commensurate collinear spin structure. Since orthorhombic (o-) perovskite RMnO₃ (R=rare earth and Y) contains the both types, it provides an ideal laboratory to compare the respective magnetoelectric (ME) properties and extract the essential ingredients of them.

The ferroelectricity with a cycloidal spin order in o-RMnO₃ has been found for R=Gd, Tb, Dy, and $Eu_{1-x}Y_x$ ^{3,6-8} Upon the application of magnetic field B along the *a* axis in the *ab*-cycloidal phase (in *Pbnm* notation) of $Eu_{1-x}Y_{x}MnO_{3}$, the spin-cycloidal plane rotates from *ab* to bc, accompanied by the polarization P rotation from a to $c.^{9,10}$ The origin of the ferroelectricity has been discussed in terms of the spin-current model or the inverse Dzyaloshinskii-Moriya (DM) interaction represented by the relation, $P \sim \sum A e_{ii} \times (S_i \times S_i)$,¹¹⁻¹³ in which e_{ii} is the unit vector connecting the neighboring spins $(S_i \text{ and } S_j)$ and both the spin-orbit and superexchange interactions are relevant to the coefficient A. On the other hand, o-RMnO₃ with R=Ho, Tm, Yb, Lu shows a commensurate collinear spin order with a propagation vector q = (0, 1/2, 1), which is so-called *E*-type antiferromagnetic order.^{14–17} This phase is allowed to possess P along a due to symmetric exchange striction, which is independent of the spin-orbit interaction. Sergienko et al. proposed the emergence of substantially large P up to 0.12 C/m^2 in the *E*-type phase with considering the ferromagnetic nearest-neighbor interaction J_1 mediated by e_g electrons as a major source of the exchange striction.^{18,19}

The rich variety of magnetic phases of o-RMnO₃ reflects the J_1 - J_2 competition that varies as a function of R-ion radius under the staggered orbital ordering of $3x^2 - r^2$ and $3y^2 - r^2$ type orbitals; as GdFeO₃-type lattice distortion becomes large, the antiferromagnetic next-nearest-neighbor interaction J_2 in the *ab* plane becomes competitive with the ferromagnetic interaction J_1 . So far, ME phase diagrams of o-RMnO₃ have been made for R from La to Dy or Eu_{0.5}Y_{0.5} [see Figs. 1(a) and 1(b)].^{7,8,20–22} However, since $o-RMnO_3$ with a smaller R ion than Dy needs a high-pressure (HP) technique to synthesize, a complete ME phase diagram of o-RMnO₃ including the neighboring area between the cycloidal and the *E*-type phases is absent. In fact, although the ferroelectricity in the E-type phase has been confirmed for o-RMnO₃ with R=Y, Ho,²³ and Tm,¹⁷ intrinsic ME properties inherent to the Mn-spin arrangement alone remain unclear because of the intervention by the magnetic R ions. Besides, the reported P values vary widely depending on the materials. As a result, quantitative estimations of P of the *E*-type phase and its microscopic origin are under intensive debate.

This Rapid Communication reports complete ME phase diagrams of a series of o-*R*MnO₃ with nonmagnetic *R* ions (R=Eu_{1-x}Y_x and Y_{1-y}Lu_y) together with the system containing the magnetic *R* ion (R=Dy, Ho, Er, Tm, and Yb). By using high-quality polycrystalline samples of o-*R*MnO₃, we have confirmed substantially large *P* of nearly 5000 μ C/m² for the *E*-type phases, and clearly demonstrated the generic transition of the multiferroic ground state from the *ab* cycloidal, to *bc* cycloidal, and eventually to the *E*-type phase upon decreasing the size of the *R* ion. Furthermore, we found an enhanced ME response characteristic of a bicritical phase boundary formed by the different multiferroic phases.

Polycrystalline samples of o-RMnO₃ with R=Dy and R not larger than Y were synthesized under HP by using hexagonal (h-) RMnO₃ as precursors. First, single crystals of h-RMnO₃ were prepared by a floating-zone method and then heat treated for 1 h in the range of 1323–1373 K under a HP of 5.5 GPa. Just for comparison between the single crystalline and polycrystalline specimens, polycrystalline samples of o-Eu_{1-x}Y_xMnO₃ with x=0.2, 0.4, 0.6 were synthesized by heat treatment under HP using polycrystalline precursors prepared by grinding the single crystals of the o phase. The grain-boundary effects on ME measurements are expected to



FIG. 1. (Color online) Phase diagrams of o-RMnO₃ with (a) magnetic R=Gd, Tb, Dy, Ho, Er, Tm, and Yb, and (b) nonmagnetic $R = Eu_{1-x}Y_x$ and $Y_{1-y}Lu_y$ (PM, IC, and AF denote paramagnetic, incommensurate, and antiferromagnetic, respectively), (c) polarization P value at 2 K for the compounds with nonmagnetic R ions $(P^{\rm HP},$ which is multiplied by a calibration factor (=6) (Ref. 26), is for P of the polycrystals, and P_a is for P along a of the single crystals taken from Ref. 9, (d) k in the magnetic propagation vector q = (0, k, 1) (taken from Refs. 5, 9, 14–17, 24, and 25), and (e) unit-cell volume as a function of the R-ion radius, of which coordination number is assumed to be 8 for an orthorhombically distorted perovskite lattice. T_{N1} was determined by magnetic and dielectric measurements. $T_{\rm N2}$ and $T_{\rm N3}$ were determined from the measurements of P. T_{N1} and T_{N2} for RMnO₃ with R=Eu, Gd, and Tb were taken from Ref. 7. The shaded area in (b) represents a possible phase-coexisting region.

be minimized owing to the high-quality precursors. However, as for $o-\text{Eu}_{1-x}Y_x\text{MnO}_3$ with x=0.75, 0.9, the attempts to grow crystals gave the mixtures of the *h* and the *o* phases, which were adopted as the precursors. The linear change in the unit-cell volume as a function of the *R*-ion radius ensures the successful syntheses of a series of $o-R\text{MnO}_3$ [Fig. 1(e)].



FIG. 2. (Color) Temperature dependence of polarization *P* of o-*R*MnO₃ with (a) nonmagnetic *R* and (b) magnetic *R* in the absence of external magnetic field *B*, and (c) *P* of o-Eu_{0.1}Y_{0.9}MnO₃ (multiplied by 2) and (d) *P* of o-LuMnO₃ measured in external *B* perpendicular to the applied electric field, and χ measured in external *B* of 0.1 T on increasing temperature. *B* was applied during the poling procedure as well as the measuring process.

For measurements of P, gold electrodes were deposited on the polished faces (4 mm²) of the platelet samples with typically 0.2 mm in thickness. As a poling procedure, an electric field of 800 V/mm was applied at 40 K, followed by cooling to 2 K. The displacement current was measured with increasing temperature at a rate of 5 K/min or sweeping *B* at a rate of 100 Oe/sec and was integrated as a function of time to obtain *P*. Magnetic susceptibility was measured by a superconducting quantum interference device magnetometer.

Figure 2 shows temperature dependence of *P* o-RMnO3 prepared by the HP technique with highlighting different responses of P to external B for o-Eu_{0.1}Y_{0.9}MnO₃ and o-LuMnO₃. As exemplified in Figs. 2(c) and 2(d), the compounds undergo a transition to an incommensurate (IC) sinusoidal phase at $T_{\rm N1}$ where χ takes a maximum. Then, they show a second transition to the *E*-type phase at T_{N2} where P sets in. In Figs. 2(a) and 2(b), P in zero magnetic field is plotted against temperature for nonmagnetic and magnetic R ions, respectively. The compounds with $R = Eu_{0,1}Y_{0,9}$, $Y_{1-v}Lu_v$, Ho, Er, Tm, and Yb are supposed to possess the E-type phase, and all the compounds but for R=Er and $Eu_{0.1}Y_{0.9}$ exhibit fairly large P of about 800 μ C/m² at the lowest temperature. By contrast, $R = Eu_{1-x}Y_x$ (x=0.4, 0.6, 0.75) and Dy samples possess relatively small P values, implying the different mechanism of P generation. We should note here that o-YMnO₃ and o-ErMnO₃ show the stepwise temperature dependence of P. The neutron diffraction studies have suggested the IC q vectors of (0, 0.435, 1) and (0, 0.433, 1) for the magnetic ground states in o-YMnO₃ and o-ErMnO₃, respectively.^{24,25} Thus, these stepwise changes may indicate the presence of a small amount of the cycloidal or the sinusoidal phase which competes and thus coexists with the E-type phase. The sign for such a phase coexistence was found for o-Eu_{0.1}Y_{0.9}MnO₃ showing smaller P and the larger effect of applied B on P in comparison with LuMnO₃ [see Figs. 2(c) and 2(d)]. The possible phase coexistence in o-YMnO₃ and o-Eu_{0.1}Y_{0.9}MnO₃ can be attributed to proximity to the firstorder phase boundary, as discussed later. As for o-ErMnO₃, a long-range order of the *E*-type phase is likely to be amenable to the large magnetic moments of Er ions with planar anisotropy, leading to the stepwise change in *P*.

Based on the polarization and magnetization measurements, we established the ME phase diagrams for a series of o-RMnO₃ with magnetic and nonmagnetic R ions, respectively, [Figs. 1(a) and 1(b)]. Here, one may find that the variation in T_{N1} as a function of the ionic radius of R is well systematic when the R ions are nonmagnetic whereas it is rather irregular when the R ions are magnetic. This difference suggests that the presence of the magnetic moment of the R ions gives substantial influence on the transition temperatures as well as the relative stability of the two kinds of cycloidal phases and the *E*-type phase. Figure 1(c) shows a plot of *P* at 2 K multiplied by a correction factor (=6) for the compounds with nonmagnetic R ions,²⁶ together with P_a in $Eu_{1-x}Y_{x}MnO_{3}$, as a function of the *R*-ion radius. Contour plot of the corrected value of P is shown in Fig. 3(a). In the *E*-type phase, *P* reaches nearly 5000 μ C/m², which is more than ten times as large as P (=384 μ C/m²) in $Eu_{0.25}Y_{0.75}MnO_3$, and almost independent of the *R*-ion radius except for Eu_{0.1}Y_{0.9}MnO₃. In the cycloidal phases, on the other hand, P is critically dependent on the R-ion radius. According to the inverse DM model, P is proportional to $\sin \theta$, where θ denotes the angle between the nearestneighbor Mn spins. When the ab-cycloidal spins are of ideal cycloid and k is less than 0.5 ($\theta < \pi/2$), P ($\propto \sin \theta$) is expected to increase monotonically, as indicated by the broken line in Fig. 1(c). However, in the range of 0.4 < x < 0.75, P decreases as increasing x. This is probably because the ellipticity of the cycloidal spins becomes larger²² or a fraction of the bc-cycloidal phase with smaller P becomes larger. By further increasing x, P shows a local minimum at x=0.75, where the *bc*-cycloidal state is the magnetic ground state, followed by an abrupt increase that indicates the emergence of the *E*-type phase.

In order to reveal the difference in ME response of each phase of o-RMnO₃ with nonmagnetic R, we have investigated B dependence of P at representative points (T=2 K)marked with open circles in Fig. 3(a). In $Eu_{0,1}Y_{0,9}MnO_3$ and LuMnO₃ [Figs. 3(d) and 3(e)], the normalized P shows nearly quadratic behavior as a function of B and little dependence on the *B* direction with respect to the *P* direction. The results are in accord with the *E*-type phase as the magnetic ground state for both compounds because P of this phase should be proportional to $S_i \cdot S_i$ and thus to $1 - aB^2$ when the magnetization is proportional to applied B. As clearly seen in Fig. 3(a), the *R*-ion dependence of the transition temperature $T_{\rm N2}$, below which the system becomes ferroelectric, makes a V-shaped feature near the boundary between the *bc*-cycloidal and the *E*-type phases. This feature reminds us of the bicriticality of two competing phases separated by the first-order phase boundary in the presence of weak randomness.²⁷ The *bc*-cycloidal and the *E*-type phases are likely separated by the first-order phase boundary,^{28,29} and $Eu_{1-x}Y_xMnO_3$ has also weak randomness introduced via alloying the A site. Since $Eu_{0,1}Y_{0,9}MnO_3$ is located on the verge of the transition from the *E*-type phase to the *bc*-cycloidal phase, the system should be subject to the phase coexistence, leading to the



FIG. 3. (Color) (a) Corrected magnitude of polarization *P* displayed as a contour plot in the phase diagram of *o*-*R*MnO₃ with nonmagnetic *R* (Eu_{1-x}Y_x, Y_{1-y}Lu_y). Magnetic-field *B* dependence of normalized *P* at 2 K for (b) Eu_{0.6}Y_{0.4}MnO₃, (c) Eu_{0.25}Y_{0.75}MnO₃, (d) Eu_{0.1}Y_{0.9}MnO₃, and (e) LuMnO₃ (*P* was normalized by the value at 2 K). Schematic of the spin rotation from the *bc* cycloidal to the *ab* cycloidal (f) under $B \parallel c$, (g) under $B \parallel P_m$ deviating from *c* to *a* by ϕ , and (h) under $B \perp P_m$ deviating from *c* to *a* by ϕ ($\phi < \pi/4$). These illustrations depict the spin rotation in a single domain both for lattice and spin. The relational expressions between P_m or P'_m and ϕ are shown.

enhancement of the effect of *B* on *P* [Fig. 3(d)] and the suppression of *P* [Fig. 2(c)]. As a hallmark of the phase coexistence, the large hysteresis was observed in the *P*-*B* curve of Eu_{0.1}Y_{0.9}MnO₃, suggesting that the decrease in *P* by the application of *B* is not only due to the cant of spins but partly due to the increase in the volume fraction of the cycloidal phase, which is dominated by the Zeeman energy gain $(\propto \chi B^2)$. In accord with this expectation, χ is larger for Eu_{0.1}Y_{0.9}MnO₃ containing a small amount of the large- χ cycloidal phase than for LuMnO₃ with the purely *E*-type order [see Figs. 2(c) and 2(d)].

On the other hand, in the *ab*-cycloidal phase of $Eu_{0.6}Y_{0.4}MnO_3$ [Fig. 3(b)] and the *bc*-cycloidal phase of $Eu_{0.25}Y_{0.75}MnO_3$ [Fig. 3(c)], *P* changes drastically at critical *B* of about 5 and 3 T, respectively, which signifies the occurrence of the *P* rotation. In fact, the study of a single crystal of $Eu_{0.6}Y_{0.4}MnO_3$ showed the *B*-induced *P* rotation from P_a

to P_c at 4.5 T ($B \parallel a$).⁹ However, because our samples are polycrystals, we have to consider the relative orientations between B, P, and the crystallographic axes of a single domain. To explain the drastic change in P under B, we considered specific situations which satisfy the conditions for the occurrence of the B-induced P rotation (hereafter we formally use P_m and P'_m for measured P before and after the P rotation, respectively). The panels in Fig. 3(f) describe the P rotation from P_c to P_a upon the application of B along c. After the *P* rotation, two kinds of domains with positive and negative P_a coexist in an equal amount [only for the P rotation to positive P_a is shown in Fig. 3(f)] so that P'_m should be zero. When B is slightly deviated from c to a by ϕ , finite P'_m remains after the P rotation as shown in Figs. 3(g) and 3(h)because the plane of spin spiral inclines to be perpendicular to B to gain the Zeeman energy during the P rotation (see the middle panels) and thereby one of the rotation directions is selected as reported in Eu_{0.55}Y_{0.45}MnO₃.¹⁰ If B is slightly deviated from c to b, P'_m should be zero similarly to the case in Fig. 3(f). Here, the normalized P after the P rotation (P'_m/P_m) for $B \parallel P_m$ and $B \perp P_m$ can be expressed as $P'_m/P_m = -\tan \phi P_a/P_c$ and $P'_m/P_m = \tan(\pi/2 - \phi)P_a/P_c$, re-

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spectively. Taking the inequations $P_a > P_c$ and $\phi < \pi/4$ into account, the drastic decrease (and even sign reversal) of P

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for $B \parallel P_m$ and the increase in *P* for $B \perp P_m$ shown in Fig. 3(c) can be well explained. The same explanation is applicable for the *P* rotation from P_a to P_c [Fig. 3(b)].

In summary, we have revealed the magnetoelectric properties of a whole series of multiferroic o-RMnO₃ with both nonmagnetic and magnetic R and estimated the genuine values of P (~5000 μ C/m²) in the E-type phase, which is more than ten times as large as that of the bc-cycloidal phase, yet one order of magnitude smaller than the predicted values.^{18,19} Furthermore, we found the bicritical feature near the phase boundary between the competing bc-cycloidal and E-type phases.

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