Soft-Phonon Response Function: Inelastic Neutron Scattering from LaAlO[†]

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High-resolution inelastic-neutron-scattering measurements have been performed at the cubic-rhombohedral (Pm3m \rightarrow R3c) phase transitions in a low-mosaic flux-grown single crystal of LaAlO₃. The transition occurred at (489 ± 2) °C, somewhat lower than reported for Verneuil-grown samples. The scattering from the soft R_{25} mode at the corner of the Brillouin zone in the cubic phase exhibits the central component up to 610 °C. In addition, phonon sidebands centered around $\pm \omega_{\infty}$ are observed which are overdamped below 550 °C. The observed intensities are well described by a model which relates the central component to the phonon sidebands through an anharmonic coupling parameter $\delta(T)$, as recently reported by Shapiro *et al.* for SrTiO₃. $\omega_{\omega}^2(T)$ is found to be nearly constant. A comparison is made between LaAlO₃, SrTiO₃, and KMnF₃ with respect to these parameters as well as the anisotropy of the dispersion surfaces near the *R* point.

I. INTRODUCTION

The structural phase changes in the materials that crystallize in the perovskite structure are often accompanied by soft-phonon modes whose symmetries correspond to the distortion that takes place at the transition.¹ The phase transition in LaAlO₃ around 500 °C belongs to a class where the condensing mode is found at the R corner $(\frac{1}{2}\frac{1}{2}\frac{1}{2})$ of the Brillouin zone in the cubic phase.² The triply degenerate phonon mode has the R_{25} irreducible representation³ and its components can be thought of as alternate librations of the AlO_8 octahedra around the cubic axis. The low-temperature phase is rhombohedral⁴ $(D_{3d}^6 \equiv R\overline{3}c)$ and the distortion from cubic symmetry corresponds to a condensation of a linear combination of all three cubic components of the R_{25} mode.⁵ Other materials of the same class such as $SrTiO_3$ ($T_c \sim 100 \text{ K}$)⁶ and $KMnF_3$ $(T_c \sim 180 \text{ K})^7$ have a tetragonal distortion in the lowtemperature phase resulting from the condensation of only one of the cubic components of the R_{25} mode. The transitions in SrTiO₃ and LaAlO₃ are continuous. Close to the transition the rotation angle of the octahedra, which is the order parameter,⁸ shows a deviation from mean-field behavior as measured by EPR.⁹

A neutron-scattering study of SrTiO₃ by Riste *et al.*¹⁰ has revealed, unexpectedly, a narrow central component in the scattering, in addition to the phonon sidebands. A recent extension of this study including both SrTiO₃ and KMnF₃ by Shapiro *et al.*¹¹ has yielded detailed information on the full response function. It was shown¹² that the soft-phonon frequency ω_{∞} remains finite at the transition point and that instead another frequency ω_0 , which characterizes the total scattering, phonons plus central peak, becomes infinitely small. This behavior is explained in terms of a phenomenological theory which introduces a low-frequency resonance with amplitude δ^2 in the phonon self-energy and which subsequently relates the characteristic frequencies by $\omega_0^2 = \omega_\infty^2 - \delta^2$. The temperature dependence of these parameters could be determined quite close to T_c in SrTiO₈ because the phonons remained underdamped, in contrast to KMnF₃ where the overdamped nature of the phonons prevented detailed quantitative studies. In this respect LaAlO₃ represents an intermediate case and, since the phase change takes place at a much higher temperature and results in a different low-temperature structure, we expect a quantitative study to yield new information on the central-mode features.

Additional motivation was given by a recent theoretical paper by Enz^{13} in which the appearance of the central peak is related to the symmetry of the lowest Γ -R phonon dispersion curve by a touching condition. This curve is nearly symmetric in $SrTiO_3^6$ and in $KMnF_3^7$ but it is asymmetric in LaAlO₃,² and, according to the model, a somewhat weaker central peak is expected for the latter case. Our findings of a strong central component very similar to the findings for $SrTiO_3$ and for $KMnF_3$ most likely do not support this model.

II. EXPERIMENTAL DETAILS

The measurements were carried out at the Brookhaven high-flux-beam reactor on a triple-axis spectrometer in the constant-Q mode of operation. The spectrometer was equipped with pyro-

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lytic graphite monochromator (bent) and analyzer, and filters were used to eliminate the higher-order contamination of the monochromated beam. At 5.1-meV incident energy, the filter was an 8-in. block of polycrystalline beryllium cooled to liquidnitrogen temperature; at 13.7 meV, pyrolytic graphite single crystals with a total thickness of 4 in. were used; and at 41 meV, 2 in. of pyrolytic graphite served as a filter. The bulk of the measurements were carried out with 20-sec horizontal collimation throughout the spectrometer and the effective vertical resolution was 1.8° [full width athalf-maximum (FWHM)]. The resulting high resolution is exemplified by the projection of the resolution ellipsoid on the energy axis (FWHM): 0.08 meV at 5.1-meV incident energy, 0.4 meV at 13.7 meV, and 2.2 meV at 41 meV. The resolution features of a triple-axis spectrometer are well known and they may be reliably simulated on a computer. In the reduction of the data we have used computer programs based on the treatment by Cooper and Nathans¹⁴ and the calculated resolution profiles were checked against measurements at Bragg reflections in the sample.

The LaAlO₃ single crystal was grown from fluxed melts using lead-based solvents. Several mixtures of PbO, PbF_2 , and B_2O_3 were tried with various ratios of La_2O_2 : Al_2O_3 . The best results were obtained with a $PbO-PbF_2$ solvent containing a small amount of V_2O_5 . In a typical growth experiment 4.8-g La₂O₃ (>99.9%, Fluka, Buchs/Switzerland), 1.6-g Al₂O₃ (high purity, Merck, Darmstadt), 26-g PbO and 24-g PbF₂ (both > 99.9% from Associated Lead Manufacturers, Greenford, Great Britain), and 1-g V_2O_5 (>99%, Fluka) were mixed and premelted at about 900 °C in a 30-cm³ platinum crucible. The crucible was sealed with a lid by argon arc welding leaving a hole of less than 0.1 mm for pressure release. Thus evaporation of solvent was minimized. The crucible was placed into a Superkanthal chamber furnace in such a way that a temperature difference of two opposite crucible walls facilitated natural convection, with the coolest spot at the bottom of the crucible. The furnace chamber was then heated to 1300 °C for 15 h, slowly cooled to 1050 °C, heated again to 1180 °C and then cooled at a rate of $0.5^{\circ}/h$ by means of a Eurotherm electronic programmer (ramp generator). The temperature of the heating elements was regulated to ± 0.5 °C, and temperature fluctuations in the furnace chamber filled with crucibles and with ceramics were less than 0.2 °C. When the temperature of 980 °C was reached, holes were punched into the lid and excess solution was poured out. The rest of the solidified flux was dissolved in dilute nitric acid.

Three crystals weighing a total of 5.4 g were obtained. The larger two were intergrown at a cor-

ner. The largest crystal of dimensions 11×9 $\times 6$ mm contained few inclusions and cracks and was by more than 95% optically clear. After the corner with the second crystal had been cut off, most of the included flux could be dissolved. This crystal was used in the measurements. The mosaic spread measured in the course of the experiments was less than 6 sec (FWHM). The exceptionally large size of this inclusion-free crystal and the prevention of multinucleation are attributed to the addition of V_2O_5 to the solution and could be explained by the formation of LaVO₄ complexes in the solution and especially in front of the growing crystals where vanadate is rejected. Formation of such complexes extends the width of the metastable Ostwald-Miers region and favors stable growth, and further examples are discussed by Elwell and Scheel.¹⁵ An electron-microprobe analysis of the crystal showed a locally varying lead concentration between 0.7 and 1.1 wt% and a vanadium content of 0.01-0.02 wt%.

The crystal was mounted with an $[03\overline{1}]$ axis vertical so that the Γ -M branch was accessible. The transition temperature T_c was determined from measurements of the scattering at the R point $\frac{1}{2}(1, 1, 3)$ of the cubic Brillouin zone which becomes a reciprocal-lattice point in the distorted phase. The intensity variation on very slow cooling is shown in Fig. 1. The peaking feature at small energy transfers is due to the soft-phonon mode which exists on both sides of T_c giving rise to a nearly



FIG. 1. Determination of the transition temperature at the $\frac{1}{2}(1,1,3)$ R point. The soft-phonon mode gives rise to a peak in the inelastic scattering at the transition temperature, and the elastic scattering is due to Bragg scattering which only occurs in the distorted phase and to the critical scattering at the phase transition.

symmetric peak as long as elastic Bragg scattering is not picked up by the finite resolution. From these traces a transition temperature of $(489 \pm 2)^{\circ}$ C was deduced. In earlier experiments on Verneuilgrown crystals transition temperatures near 525°C were reported, ^{2,4,16} and the difference is probably caused by the lead impurities.

III. RESPONSE FUNCTION

Microscopic phonon theory with self-consistent treatment of the anharmonic effect is adequate in the description of soft-phonon modes in the temperature ranges where the mean-field approximation is valid. Such a treatment yields the correct temperature dependence of both the order parameter and of the soft-phonon frequency. 13, 17, 18 However, near the transition temperature where fluctuations are large mean-field theory breaks down and so far only few attempts have been made to calculate the behavior in this regime. Feder¹⁹ has presented thermodynamic arguments for the appearance of a central diffusive mode in the response function and both Silberglitt²⁰ and Schwabl²¹ have shown that microscopic theory assuming a low-frequency resonance in the phonon self-energy also produces a triple-peaked response, ²⁰ but neither approach can actually predict the temperature dependence. This has been done by Schneider, but close to T_c only.²² Nevertheless, such theories are useful to the experimentalists in that the derived functional form of the response function can be tried out in fits to experimental observations, and values for the model parameters may be found. In this work we have applied the following form for the response function

$$S(\mathbf{\bar{q}}, \omega) = \frac{k_B T}{\pi} \frac{\delta^2(T)}{\omega_0^2(\mathbf{\bar{q}}, T) \omega_\infty^2(\mathbf{\bar{q}}, T)} \frac{\gamma'}{\omega_0^2 + \gamma'^2} + \frac{k_B T}{\pi} \frac{\Gamma_0}{[\omega_\infty^2(\mathbf{\bar{q}}, T) - \omega^2]^2 + \omega^2 \Gamma_0^2} , \quad (1)$$

$$\omega_0^2(\mathbf{q}, T) \equiv \omega_\infty^2(\mathbf{q}, T) - \delta^2(T), \ \gamma' \equiv \gamma(\omega_0/\omega_\infty)^2$$

which is derived under the conditions that $\Gamma_0 \ll \delta^2 / \gamma$ and $\omega_{\infty}^2 \gg \gamma^2$ and which obeys the following sum rule:

$$\int_{-\infty}^{\infty} S(\vec{\mathbf{q}},\omega) \, d\omega = \frac{1}{\omega_0^2(\vec{\mathbf{q}},T)} = \frac{\delta^2(T)}{\omega_0^2(\vec{\mathbf{q}},T)\omega_\infty^2(\vec{\mathbf{q}},T)} + \frac{1}{\omega_\infty^2(\vec{\mathbf{q}},T)} \, . \tag{2}$$

The first term in Eq. (1) represents the central peak and the second term represents the phonon bands at $\pm \omega_{\infty}(\mathbf{q}, T)$. This form results from the incorporation of the simplest possible low-frequency resonance at frequency γ in the self-energy with the amplitude $\delta^2(T)$. The arithmetic derivation of Eq. (1) is given by Shapiro et al.¹¹ It relates with a single parameter $\delta^2(T)$ both the amplitude and the d dependence of the central peak to the phonon bands, and it was shown to give an adequate description of the observed intensities near the phase transition in the cubic phase of SrTiO₂. Shapiro et al. pointed out the paramount importance of the resolution effects when dealing with very singular cross sections such as Eq. (1) and in the following we wish to extend their discussion.

Given a functional form of the cross section the expected intensities may be calculated by a folding with four-dimensional resolution function $R(\bar{q}, \omega)$:

$$I_{j}(\vec{q}_{0},\omega) = A(k_{i}/k_{j}) |F_{j}(\vec{k})|^{2} \int d\vec{q} \, d\omega \, S(\vec{q},\omega) \\ \times R(\vec{q} - \vec{q}_{0},\omega - \omega_{0}) , \quad (3)$$

where \vec{k}_i and \vec{k}_f are the initial and final wave vectors of the scattered neutrons $\vec{k} = \vec{k}_i - \vec{k}_f$, and $F_j(\vec{k})$ is the inelastic structure factor for the *j*th mode at $\vec{q} = \vec{k} - \vec{\tau}$, $\vec{\tau}$ being a reciprocal-lattice vector. Equation (3) shows that the observed intensities even in constant-*q*-scans with $\vec{q}_0 = 0$ will depend on the behavior of $S(\vec{q}, \omega)$ at finite \vec{q} which means that we have to insert some information about the dispersion around the *R* point in our fitting attempts to the soft-mode response function. Gesi *et al.*⁷ have shown that the dispersion of the triply degenerate R_{25} mode to a good approximation is given by the following truncated dynamical matrix which neglects all other degrees of freedom than the librations of the AlO₆ octahedra:

$$\overrightarrow{D}(\overrightarrow{q}_{R}+q) = \begin{cases}
\omega_{\infty}^{2} + \lambda_{1}q_{x}^{2} + \lambda_{2}(q_{y}^{2}+q_{z}^{2}) & \lambda_{3}q_{x}q_{y} & \lambda_{3}q_{x}q_{x} \\
\lambda_{3}q_{x}q_{y} & \omega_{\infty}^{2} + \lambda_{1}q_{y}^{2} + \lambda_{2}(q_{z}^{2}+q_{x}^{2}) & \lambda_{3}q_{y}q_{z} \\
\lambda_{3}q_{x}q_{x} & \lambda_{3}q_{y}q_{z} & \omega_{\infty}^{2} + \lambda_{1}q_{z}^{2} + \lambda_{2}(q_{x}^{2}+q_{y}^{2})
\end{cases} .$$
(4)

A comparison of the above form of the dynamical matrix to a small-q expansion of the form derived by Pytte and Feder¹⁷ by self-consistent anharmonic theory shows that the correlation functions which determine the temperature dependence of ω_{∞}^2 also enter in the expressions for λ_i . However, meaningful estimates of the temperature variation of λ_1

cannot be obtained from this theory because the results depend crucially on the assumed force constants. Detailed knowledge of the dispersion near the R point at temperatures close to T_c in LaAlO₃ is not experimentally accessible owing to the overdamped nature of the phonons, and so the information we are able to extract about the soft-mode re-



FIG. 2. Observed intensities at 13.7-meV incident neutron energy. The broken and full lines represent the best fit to the data using Eq. (1) of the text with the appropriate resolution corrections.

sponse function will be limited by this deficiency. To illustrate how the uncertainty in the values of λ_i introduces uncertainties in the derived model parameters, we may perform an integration of the central-peak cross section [Eq. (2)] along the vertical component of \mathbf{q} ; the integration is inherent to the experiment due to the relaxed vertical collimation. We also integrate over ω since the width of the central peak is always found much smaller than the instrumental energy resolution. The result is



FIG. 3. Observed intensities at 5.1-meV incident neutron energy. The full line is calculated on the basis of the parameters derived from fits to the 13.7-meV data.

$$S_{\text{centr.}}(\vec{\mathbf{q}}_{s}) = \int d\omega \, dq_{z} S_{\text{centr.}}(\vec{\mathbf{q}}, \omega) = \frac{\partial^{2}(T)}{[\lambda_{2}(T)]^{1/2}} \\ \times \frac{k_{B}T}{\omega_{\infty}^{2}(\vec{\mathbf{q}}_{s}, T)\omega_{0}(\vec{\mathbf{q}}_{s}, T) + \omega_{\infty}(\vec{\mathbf{q}}_{s}, T)\omega_{0}^{2}(\vec{\mathbf{q}}_{s}, T)},$$
(5)

where \bar{q}_s lies in the scattering plane. It shows that $\delta^2(T)$ can only be determined to the accuracy of our knowledge of $[\lambda_2(T)]^{1/2}$.

IV. RESULTS AND DISCUSSION

The existence of a central component in the response function of the soft R_{25} mode in cubic LaAlO₃ is convincingly seen in Figs. 2 and 3, which also illustrates the influence of the finite resolution on the observed line shapes. At 13.7-meV incident energy, only in the high-temperature scan in Fig. 2 are the phonon bands and the central peak distinct; however, at 5.1-meV incident energy, we still can distinguish the narrow central component from the phonons at the lower temperature as seen in Fig. 3. It is noted that a small amount of incoherent scattering has been subtracted from the data so that the central feature is entirely due to the response function. The rapid temperature variation of the central peak is illustrated in Fig. 4 where the inverse peak intensity (extrapolated



FIG. 4. Temperature over the observed peak intensity of the central component plotted on double logarithmic scale vs the reduced temperature. The solid line is based on the lines drawn through the data points in Fig. 6.



FIG. 5. Dispersion of the lowest phonon branch along R to M on the edge of the Brillouin zone. The dispersion of the equivalent branches in KMnF₃ and SrTiO₃ is indicated for comparison.

phonon background subtracted) is plotted against the reduced temperature. For $(T - T_c)/T_c > 0.02$ the q_s width of the central peak is larger than the q_s width at zero-energy transfer of the resolution function, whereas the energy width is smaller than the resolution width so as an approximation we may use Eq. (5) to describe the peak intensity variation. This is done for the solid line using the parameters derived below.

In order to obtain the dispersion parameters for the fitting procedure we measured the dispersion of the lowest R-M branch in the cubic phase 100 °C above the transition and the result is shown in Fig. 5. Together with previous measurements by Axe et al.² along the Γ -R direction our data yielded the following parameters $\lambda_1 = \lambda_2 = 2000 \text{ meV}^2 \text{ Å}^2$. These isotropic values were used at all temperatures together with the assumption $\lambda_3 = 0$. Also, in Fig. 5 we have as a comparison indicated the dispersion of the equivalent branch in KMnF₃ and SrTiO₃. Our finding of a nearly isotropic dispersion in LaAlO₃ as opposed to the marked anisotropy found most pronounced in KMnF₃ but also in SrTiO₃ may intuitively be related to the fact that the latter two undergo a tetragonal distortion at the phase change whereas LaAlO, distort rhombohedrally. However, mean-field theory does not show such a connection.

The soft phonon was followed at $\frac{1}{2}(1, 1, 3)$ in the temperature range 500-600 °C and, independent of the central peak, the damped harmonic oscillator response was fitted to the observed phonon intensities using Γ_0 and ω_{∞}^2 as variables. Below 550 °C,

the response is overdamped and no shoulders were explicit in the data. The fits were still reliable down to 530 °C since the scaling factor was already determined at high temperatures and the separation of the central peak was readily made. The broken line in Fig. 2 gives an example of the fitted curves. After the phonon parameters were determined only $\delta^2(T)$ was allowed to vary in order to fit the central peak. Below 530 °C the full cross section was used and the full line in Fig. 2 represent the best fit to the 504 °C data. The same parameters except for a scaling parameter were used to draw the full line through the higher resolution data on Fig. 3 and the consistency is satisfactory but not overwhelming, the main reason being the approximate way in which the dispersion is taken into account in the resolution correction. It is seen that the central component is calculated too broad which indicates that the value used for λ_i is too small but the data do not contain enough information to include λ_i as parameters to be fitted. At 569 °C where the q width of the central peak is much larger than the instrumental resolution, we measured the q profile and the observed full width of 0.068 ± 0.008 Å⁻¹ compared well with the calculated value of 0.072 Å⁻¹ based on the derived crosssection parameters. The results of the fits for ω_{∞}^2 and δ^2 are shown in Fig. 6. The error bars only refer to the statistical uncertainty in the determination. The phonon width was found to be essentially constant through the temperature range Γ_0 $= 2.8 \pm 0.2$ meV and the fits were quite insensitive to the value of γ' as long as it was smaller than half the resolution width so values of 0.02 and 0.03



FIG. 6. Temperature dependence of the parameters ω_{∞}^2 and δ^2 of the response function as determined by the least-squares-fitting procedure. The error bars do not include systematic error.

meV were used but these values can only be taken as crude upper limits for the actual γ' . Recent analysis of the EPR line width of the Fe³⁺- V_0 center in SrTiO₃ has shown that γ' (ρ in Ref. 23). 0.8 °C above T_c is above 70 MHz which is more than one order of magnitude below the resolution of the present neutron spectrometers.

The general picture that emerges of the response function associated with the soft mode in LaAlO₃ is similar to the findings for SrTiO₃ and KMnF₃ in spite of the fact that the transition temperature is much higher. Away from T_c , δ^2 is constant and ω_{∞}^2 varies linearly with temperature but extrapolates to zero well above the observed transition temperature. Near T_c a relatively rapid temperature variation of δ^2 (from 0.9 to 0.4 meV²) is found in SrTiO₃ of which we find no indication in LaAlO₃;

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however, we could not derive meaningful parameters in the immediate vicinity of T_c where the phonon is extremely overdamped. Within the limitations set by the uncertainty in the dispersion parameters we find that the phenomenological form of the response function as given by Eq. (1) is an adequate representation of the observed scattering. Further theoretical work is needed to elucidate the temperature dependence of the involved parameters.

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