Magnetism in $Nb_{1-\nu}Fe_{2+\nu}$: Composition and magnetic field dependence

D. Moroni-Klementowicz, M. Brando,* C. Albrecht, W. J. Duncan, and F. M. Grosche[†] Department of Physics, Royal Holloway, University of London, Egham TW20 0EX, United Kingdom

D. Grüner and G. Kreiner

Max-Plank-Institut für Chemische Physik fester Stoffe, Nöthnitzer Strasse 40, D-01187 Dresden, Germany (Received 11 March 2009; revised manuscript received 15 May 2009; published 9 June 2009)

We present a systematic study of transport and thermodynamic properties of the Laves phase system $Nb_{1-y}Fe_{2+y}$. Our measurements confirm that Fe-rich samples, as well as those rich in Nb (for $|y| \ge 0.02$), show bulk ferromagnetism at low temperature. For stoichiometric NbFe₂, on the other hand, magnetization, magnetic susceptibility, and magnetoresistance results point toward spin-density wave (SDW) order, possibly helical, with a small ordering wave vector $Q \sim 0.05$ Å⁻¹. Our results suggest that on approaching the stoichiometric composition from the iron-rich side, ferromagnetism changes into long-wavelength SDW order. In this scenario, Q changes continuously from 0 to small, finite values at a Lifshitz point in the phase diagram, which is located near y=+0.02. Further reducing the Fe content suppresses the SDW transition temperature, which extrapolates to zero at $y \approx -0.015$. Around this Fe content magnetic fluctuations dominate the temperature dependence of the resistivity and of the heat capacity which deviate from their conventional Fermi-liquid forms, inferring the presence of a quantum critical point. Because the critical point is located between the SDW phase associated with stoichiometric NbFe₂ and the ferromagnetic order which reemerges for very Nb-rich NbFe₂, the observed temperature dependences could be attributed both to proximity to SDW order or to ferromagnetism.

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I. INTRODUCTION

Magnetic quantum phase transitions, where the threshold of magnetism can be reached at low temperature, guide toward new ordering phenomena in metals. While examples of quantum phase transitions and of the associated quantum critical behavior abound in 4f-electron metals, such as the heavy fermion Ce and Yb compounds,¹ comparatively few transition metal compounds have been studied in detail. Among these are the nearly or weakly ferromagnetic (FM) materials MnSi, Ni₃Al/Ni₃Ga, *e*-Fe, FeGe, and ZrZn₂, as well as layered oxides such as the high- T_c cuprates and the ruthenates.²⁻⁸ Outside the oxide family, itinerant antiferromagnetism, or spin-density-wave (SDW) order, is rare in transition metal compounds. The most thoroughly studied example is chromium and its alloy series with vanadium, in which signatures of Fermi liquid (FL) (Ref. 9) breakdown have been observed at the quantum critical composition $Cr_{1-x}V_{x}$ ¹⁰ Recently, we have reported anomalous temperature dependences of the heat capacity and of the electrical resistivity in slightly off-stoichiometric NbFe2.11 Positioned very close to the threshold of magnetism at ambient pressure and stoichiometric composition, NbFe2 may provide fresh opportunities in this long-standing problem.

The C14 Laves phase system Nb_{1-y}Fe_{2+y} exhibits three magnetically ordered low-temperature states within a narrow composition range at ambient pressure. At slightly off-stoichiometric compositions, both toward the iron-rich and the niobium-rich side, it has been reported to be ferromagnetic at low temperature, with composition-dependent, low transition temperatures T_c of the order of tens of Kelvin. Moreover, at and very close to the stoichiometric composition, NbFe₂ has been reported to assume antiferromagnetic or spin-density-wave order below a second transition tem-

perature $T_m \simeq 20$ K.^{12–16} The easy accessibility of these ordered states in a nearly stoichiometric compound presents a number of interesting opportunities for studies of quantum phase transitions and quantum criticality in a transition metal compound.

First, it may be possible to examine ferromagnetic quantum criticality by tuning T_c toward absolute zero. This requires starting with a niobium-rich sample and either changing the composition or applying hydrostatic pressure. In iron-rich samples, ferromagnetism transforms into the presumed spin-density-wave state on approaching stoichiometry. The precise nature of this nonferromagnetic ordered phase in nearly stoichiometric NbFe₂, and the associated quantum critical phenomena, represent a second area of interest. Finally, the intermediate composition ranges, at which ferromagnetism is replaced by the nonferromagnetic order characteristic of stoichiometric NbFe₂ invite detailed investigation.

Past studies of the $Nb_{1-v}Fe_{2+v}$ system have explored the composition-temperature phase diagram in polycrystals by magnetic measurements and by nuclear magnetic resonance.^{13,14} While in broad agreement, these studies differ in their classification of the slightly niobium-rich region of the phase diagram. The former (Ref. 13) describes the region around $y \approx -0.01$ as paramagnetic down to 2 K, whereas the latter (Ref. 14) reports a mixed (FM and SDW) phase. Further microscopic probes have been Mößbauer spectroscopy, muon spin relaxation (μ sr) and neutron scattering studies. While μ sr has shown evidence of static moments in stoichiometric NbFe₂, neutron scattering has so far not revealed any information about the nature of this magnetic order in NbFe₂. Here, we re-examine the magnetic phase diagram of $Nb_{1-\nu}Fe_{2+\nu}$ in well-characterized high-quality samples. We aim to address key questions thrown up by earlier studies: (i)

TABLE I. Results of the phase analysis: the deviation from stoichiometry given by y has been derived from Eq. (1).

Nb _{1-y} Fe _{2+y}	у	Meth.	Phase	Nom. (at. %)	WDXS (at. %)	a (Å)	с (Å)	RRR	<i>Т_N</i> (К)	<i>Т_{с,s}</i> (К) ^b	$\mu_0 H_c$ (T) ^c
Nb _{0.8} Fe _{2.2}	0.2	(A)	C14+Fe(Nb)	26.0	27.9(2)	4.8136(4)	7.8509(5)				
Nb _{0.825} Fe _{2.175}	0.175	(A)	C14+Ti ₂ Ni	28.0	28.4(3)	4.8159(7)	7.857(1)				
Nb _{0.865} Fe _{2.135}	0.135	(A)	C14 ^a	29.0	29.2(2)	4.8211(5)	7.8642(7)				
Nb _{0.897} Fe _{2.103}	0.103	(A)	C14 ^a	30.0	30.3(3)	4.8249(6)	7.872(1)				
Nb _{0.96} Fe _{2.04}	0.04	(A)	C14	32.0	32.5(2)	4.8346(5)	7.8880(8)			72	
Nb _{0.992} Fe _{2.008}	0.008	(A)	C14	33.3	33.2(2)	4.8419(4)	7.8978(6)		14		
Nb _{1.035} Fe _{1.965}	-0.035	(A)	C14	34.0	34.5(3)	4.8461(6)	7.9048(9)			14.5	
Nb _{1.06} Fe _{1.94}	-0.06	(A)	C14+Ti ₂ Ni	35.0	34.6(1)	4.8503(5)	7.9106(7)			30	
Nb _{1.09} Fe _{1.91}	-0.09	(A)	C14+ <i>µ</i>	36.0	34.8(2)	4.8556(4)	7.9175(7)				
Nb _{1.093} Fe _{1.907}	-0.093	(A)	C14+ <i>µ</i>	38.0		4.8562(6)	7.918(1)				
Nb _{1.097} Fe _{1.903}	-0.097	(A)	C14+ <i>µ</i>	40.0	35.2(2)	4.8569(7)	7.919(2)				
Nb _{1.006} Fe _{1.994}	-0.006	(A)	C14		33.3	4.8414(6)	7.8989(1)	13	5		
Nb _{1.012} Fe _{1.988}	-0.012	(A)	C14		33.53	4.842(1)	7.900(2)		2		
NbFe ₂	0	(B)	C14	33.3	33.3(4)	4.8401(2)	7.8963(6)	9.1	10		0.6
Nb _{0.997} Fe _{2.003}	0.003	(B)	C14+Ti ₂ Ni			4.8397(6)	7.8959(9)		10	80	
Nb _{0.993} Fe _{2.007}	0.007	(B)	C14 ^a			4.8414(6)	7.8981(9)	5.3	18	6	0.15
Nb _{0.985} Fe _{2.015}	0.015	(B)	C14					6	32	25	
Nb _{1.01} Fe _{1.99} ^d	-0.01	(C)	C14					18	2.8		0.2
Nb _{1.022} Fe _{1.978} d	-0.022	(C)	C14			4.8440(4)	7.9015(5)			4	

^aSome traces of Ti₂Ni-structure type phase.

 ${}^{b}T_{s}$ indicates the FM transition temperature of the second phase with Ti₂Ni-structure type, while T_{c} refers to the FM state of the C14 Laves phase.

^cValues estimated at 100 m K

^dSingle crystal obtained using Czochralski method (Refs. 11 and 20).

are both FM phases previously observed truly belonging to the C14 phase of Nb_{1-y}Fe_{2+y} or is the ferromagnetism due to the presence of a FM second phase;^{17,18} (ii) which is then the real origin of the SDW order; (iii) how does the SDW state grow out of the FM state; (iv) does the antiferromagnetism disappear in slightly Nb-rich samples, and, if so, (v) is there a quantum critical point (QCP)?

II. EXPERIMENTAL DETAILS

A. Sample preparation

Most measurements in previous studies^{12–14} were carried out on polycrystalline samples of Nb_{1-y}Fe_{2+y} prepared by arc melting, followed by annealing for one week at 1000 °C. X-ray powder diffraction was used to check for phase purity and to determine the lattice parameters, from which the composition was estimated. Since C14-Nb_{1-y}Fe_{2+y} crystallizes in a broad homogeneity range from 27.4 to 36.3 at. % Nb at 1100 °C, and because the magnetic properties are very sensitive to *y*, it is important to verify not only the structure but also the final composition. We decided, therefore, to prepare samples in the range from 26 to 40 at. % Nb, covering the whole homogeneity range of the C14 phase, with two similar methods (indicated as method A and B in Table I) accompanied by characterization with x-ray powder diffraction, metallographic analysis, and wavelength and energy dispersive x-ray spectroscopy (WDXS and EDXS). The results are displayed in Table I. Method A: pellets with a mass of ≈ 2 g were prepared by arc melting from pure niobium (H. C. Starck, granules, 99.9%) and iron (Chempur, foil, 99.997%) on a water-cooled copper hearth in an argon atmosphere. In order to assure homogeneity, the pellets were turned over and remelted several times. The pellets were subsequently enclosed in weld-sealed niobium ampoules which were in turn jacketed in quartz tubes, and annealed at 1100 °C for three weeks. Method B: polycrystals were prepared by melting together the elements in a radio-frequency induction heating system (in argon atmosphere) on a water-cooled copper boat, immediately followed by a short anneal above 1000 °C.

X-ray powder diffraction data (Fig. 1) have been collected on a Huber G670 diffractometer equipped with a Ge monochromator. Co $K\alpha_1$ radiation was used instead of standard Cu $K\alpha_1$ because of the strong absorption of iron. Silicon [National Institute of Standards & Technology SRM 640 c, a=5.43119(1)Å] was used as the internal standard. The lattice parameters were refined by least-squares fits of the diffraction angles in the range $20^\circ \le 2\theta \le 100^\circ$, where 2θ is the diffraction angle, using the program PPLP.¹⁹ Together with the expected hexagonal C14 Laves phase (space group $P6_3/mmc$), X-ray powder diffractograms show evidence of a second phase with Ti₂Ni structure type (space group $Fd\bar{3}m$, $a \approx 11.3$ Å) in almost all investigated samples. The structure type of the second phase was confirmed for the sample at the



FIG. 1. (Color online) X-ray powder diffractogram of a sample extracted from the upper slice of a pellet prepared with method B. Besides reflections of the C14 phase a second phase with Ti_2Ni structure type is clearly observable (possible reflection positions are indicated). Reflections of the internal silicon standard are marked with asterisks.

composition Nb_{1.06}Fe_{1.94} (see Table I) by electron-backscatter diffraction. According to EDXS analysis this phase is richer in Nb compared to the Laves phase. The actual composition of this phase could not be accurately determined due to its small grain size. It is plausible that in a sample with a large amount of this Nb-rich second phase the C14 main phase will be Fe rich. This argument may explain the difference in lattice parameters between the samples with y=0 and y=0.003, which were both extracted from the same ingot.

The main difference between the two methods emerged in the metallographic analysis: while the first method resulted in high homogeneity in all regions of the final pellet with very small isolated filaments of the Ti_2Ni -like phase, the second method produced drastic differences in homogeneity between the upper and the lower zone inside the pellet. The zone with almost no extra phases appears to be the middle of the pellet.

After having cut the samples into horizontal slices, we ground and polished them carefully (in two steps) and observed the surface with a high resolution optical microscope. The inset in Fig. 2 shows the optical micrograph taken from one of the upper slices of the NbFe₂ polycrystal prepared with method B. It is the same sample as that measured in Fig. 1, where we observed the richest amount of second phase. The phase can be seen clearly near the C14-NbFe₂ grain boundaries, which have dimensions of about 50 μ m.

Afterward, we measured the ac susceptibility at low magnetic fields ($H \le 250$ Oe) of samples extracted from the upper and the middle slice (Fig. 2): in the upper slice, we observed a FM signal with a transition temperature $T_s \approx 80$ K, in addition to the expected SDW peak at $T_N \approx 10$ K. In the middle slice, only the SDW signature is present. Comparing these measurements with the metallographic analysis, we conclude that the second phase is responsible for the FM part of the signal. Repeating the same procedure with different samples, we observed that T_s is different in every sample, varying from 250 to 15 K. It probably depends on the real



FIG. 2. (Color online) ac susceptibility of samples cut from the upper and middle slice of a pellet prepared with method B (samples with y=0 and y=0.003 in Table I). The measurements performed at low magnetic field (100 Oe) allow to detect the presence of FM phases ($T_s \approx 80$ K) in the upper slice sample in addition to the expected SDW peak ($T_N \approx 10$ K). Comparing this measurement with the micrograph shown in the inset (the scale length is 100 μ m), we deduce that the FM signal is caused by the second phase present around the NbFe₂ grain boundaries.

Nb concentration at the grain boundaries. Looking at the $Nb_{1-y}Fe_{2+y}$ phase diagram (Fig. 19), this hypothesis makes sense because the Nb-rich region is FM and the transition temperature varies strongly with the quantity of Nb. In magnetic fields higher than 250 Oe these FM features disappear. It is therefore necessary to evaluate the quality of any sample by not only measuring the residual resistance ratio (RRR), but also the ac susceptibility in zero magnetic field.

The lattice parameters of all polycrystals are listed in Table I, together with the initial nominal compositions, WDXS results and transition temperatures. The discrepancy between the values of our lattice parameters for NbFe₂ and those reported in literature is less than 0.001 Å.^{12,17,18} The measurements that we are going to present have been performed on a selected number of polycrystals, which show none or at most very low presence of the second phase.

Figure 3 shows the average volume per atom of the C14 structure versus the nominal (solid line) and WDXS (dashed line) Nb content. The final composition of most samples, as determined by WDXS, is slightly richer in Nb than the nominal composition. In the range from 28 to 34 at. % Nb, a linear dependence of the atomic volume on composition determined by WDXS according to Vegard's volume is observed. Around the upper border of the homogeneity range (\approx 35 at. % Nb), the sample composition could not be determined accurately by WDXS, probably because of the large amount of the second phase. Using the relation

$$V_{\text{atom}}(\text{\AA}^3) = 12.0103 + 0.04024x(\text{at. \%Nb}_{\text{WDXS}})$$
 (1)

obtained by a least-squares fit of the data, we can determine the composition of a sample by measuring the lattice parameters. This is especially useful for detecting small differences in composition that are difficult to observe using WDXS.



FIG. 3. Unit-cell volume of the hexagonal C14 Laves phase $Nb_{1-y}Fe_{2+y}$ calculated from the unit cell parameters, displayed in Table I, versus the nominal and WDXS composition. The limits of the homogeneity region are 27.4 and 36.3 at. % Nb. The dependence of the average atomic volume on the composition determined by WDXS (dashed line) is used to extract the compositions of samples with small values of *y*.

The accuracy in the WDXS experiments is of the order of 0.1% limiting the accuracy on y to \sim 0.003. We note that a small systematic error of about 0.2 at. % cannot be ruled out. However, this error would only shift the absolute value without changing the shape of the phase diagram.

B. Experimental setup

Measurements of the resistivity, heat capacity, magnetization, and magnetic susceptibility down to 1.8 K have been carried out in a 9 T Physical Properties Measurements System and in a 7 T Magnetic Properties Measurements System (Quantum Design). High-resolution measurements of the resistivity at temperatures down to 50 mK were obtained in an adiabatic demagnetization refrigerator (Cambridge Magnetic Refrigeration) by a standard four-terminal lock-in technique. The low-temperature heat capacity was determined in the same refrigerator by a relaxation-time technique. Due to the low molar specific heat of NbFe2 compared to that of heavy fermion compounds ($C_p \sim 48 \text{ mJ/Kmol}$ at 1 K), and due to the low sample mass of $\sim 5\,$ mg, we had to assemble a particular platform for measuring the heat capacity with high resolution: We used sapphire single crystals of 6 mm diameter and 100 μ m thickness held mechanically by four Pt_{0.9}Ir_{0.1} wires, which provide a well defined thermal link with a thermal conductance $K \sim 0.2 \ \mu W/K$ at 1 K; this configuration provides relaxation-time constants of about au $=C_{p}/K \sim 6$ s. Details of the heat capacity setup will be described in a forthcoming article.

III. RESULTS AND DISCUSSION

A. FM phases: $y \ge 0.015$ and $y \le -0.02$

We focus our attention first on a very Nb-rich sample (y = -0.035), as well as on one very rich in Fe (y=+0.04). For



FIG. 4. Temperature dependence of the dc magnetic susceptibility of samples with y=+0.04 and y=-0.035. Inset: hysteretic magnetization curves for the same samples. The Nb-rich sample shows a larger coercive field and smaller remanent magnetization.

both compositions, literature reports FM order.^{13,14} They are still in the C14 homogeneity range and the amount of the second phase appears to be negligible. Our results confirm that both Nb- and Fe-rich $Nb_{1-y}Fe_{2+y}$ are FM, and indicate that the two ferromagnetically ordered states are different in character.

Figure 4 shows the dc magnetic susceptibility of the two samples measured at 50 Oe. Clear jumps in magnetization are observed at $T \approx 72$ K and $T \approx 14.5$ K for the Fe-rich and the Nb-rich samples, respectively. When increasing the external field, the transition temperature remains practically constant, suggesting that the ferromagnetism is not a consequence of the second phase. In addition, hysteresis signatures are evident in both samples, as shown in the inset of the same figure. Although in both samples the remanent magnetizations are very small, of the order of $10^{-2}\mu_B$ per atom, the coercive fields are quite large: 200 and 700 Oe. With increasing Nb excess the coercive field of Nb-rich samples decreases to a value that is not detectable anymore by the measurement technique, as in the case of soft magnets: the Arrott plot for sample y=-0.06 ($T_c=30$ K) in Fig. 5 at 2 K does not show any hysteresis, although the finite M^2 intercept for zero H/M is a clear indication of a remanent magnetization and hence of ferromagnetism. For very slight Nb doping, v =-0.012, no FM signal has been observed down to 2 K and also the Arrott plot suggests a PM ground state. On the Ferich side, all samples exhibited large hysteresis (see inset of Fig. 5). The FM transition in samples y=+0.04 and y=-0.035 is also inferred by the behavior of χ^{-1} versus T shown in Fig. 6. All samples in the homogeneity range follow a Curie-Weiss law below 100 K with a slope C =1/($T\chi$) corresponding to a large fluctuating moment $\mu_{\rm eff}$ compared to the ordered moment (~0.02 μ_B), where C^{-1} $=\frac{1}{3}\frac{N}{V}\mu_0\mu_{\rm eff}^2/k_B$ and $\frac{N}{V}$ is the atomic number density. As the composition is varied from slightly Fe rich (y > 0) to slightly Nb rich (y<0), μ_{eff} remains constant at $\approx 0.9 \mu_B$ per atom (inset of Fig. 6), whereas the Curie-Weiss temperature θ_{CW} changes sign, from $\theta_{CW} \approx 3$ K for y=0 to $\theta_{CW} \approx -3$ K for y = -0.012.



FIG. 5. Arrott plots for Nb-rich samples at 2 K. Inset: the same is for Fe-rich samples.

To study the ground-state properties of the FM phase, we measured the electrical resistivity and the heat capacity of a FM sample with y=+0.015 down to 100 mK. Although the sample shows a small SDW signal (peak in the magnetic susceptibility) at $T_N=32$ K, its ground-state properties are dominated by the FM transition at $T_c=25$ K: the resistivity $\rho(T)$, shown in Fig. 7, suddenly decreases only at the FM transition temperature $T_c=25$ K, following a $T^{5/3}$ law above and below T_c (Ref. 11), which indicates a strong influence of FM fluctuations on the electron scattering rate. Below $T_{FL} \approx 2$ K, $\rho(T)$ recovers a Fermi-liquid T^2 dependence (lower inset). The magnetoresistance is almost flat at 100 mK up to 1 T (upper inset).



FIG. 6. Inverse of the magnetic susceptibility at three selected compositions versus temperature. All samples in the homogeneity range follow a Curie-Weiss law below 100 K. Inset: dependence of the fluctuating moment μ_{eff} on y.



FIG. 7. (Color online) Resistivity ρ of sample with y=+0.015: the FM transition temperature T_c is visible at 25 K where ρ suddenly decreases. Interestingly, ρ follows a $T^{5/3}$ law above and below T_c and recovers a Fermi-liquid T^2 dependence below 2 K (lower inset) (Ref. 11). The magnetoresistance is almost flat at 100 mK (upper inset).

The temperature dependence of the heat capacity divided by temperature, C_p/T , of the sample with y=+0.015 is shown in Fig. 8. No clear heat capacity anomalies are visible at the SDW and FM transition temperatures (arrows), which suggests small ordered moments. Below 2 K, C_p/T levels off at a value of 42 mJ/K² mol, consistent with a Fermi-liquid ground state.

Below 300 mK, however, C_p/T increases dramatically, probably because of the nuclear Schottky contribution of the Nb atoms subject to the internal field of a ferromagnet (inset). By fitting the points below 2 K with a Schottky contribution, we obtain $C_p/T = (41.6 + 0.02(T/K)^{-3}) \text{mJ/K}^2 \text{ mol.}$ From this equation we can estimate the internal magnetic field B_{eff} experienced by the Nb atoms,



FIG. 8. (Color online) Heat capacity divided by temperature Cp/T vs T of the sample with y=+0.015: the SDW transition is slightly visible as a tiny hump at 32 K. Below 2 K Cp/T is constant and increases dramatically below 300 mK, probably because of the nuclear Schottky contribution of the Nb atom surrounded by a strong FM field (inset). The line in the inset is a fit with a Schottky function $Cp/T \sim T^{-3}$.



FIG. 9. (Color online) Real and imaginary part of the ac susceptibility in different magnetic fields for NbFe₂.

0.02 KmJ/mol =
$$\frac{R}{3}I(I+1)\left(\frac{h\gamma_{\rm NMR}}{k_B}\right)^2 B_{\rm eff}^2$$

where γ_{NMR} is given by NMR experiments and *I* is the nuclear spin of the Nb atoms. For ⁹³Nb atoms with *I*=9/2 and γ_{NMR} =10.4 MHz/T, the resulting field is B_{eff} =1.08 T. Only in this FM sample (the only FM sample measured down to 50 mK), has the Schottky contribution to the heat capacity been clearly observed.

We conclude that the ground state of the Fe-rich compounds is ferromagnetic and that the low-temperature transport and thermodynamic properties are well explained by the Fermi-liquid theory, as in many other FM transition metal compounds.³

B. SDW phase: y=0

We focus now on stoichiometric NbFe₂. Based on the broadening of the NMR line width and on the "S"-like feature in the magnetization curves below $T_N=10$ K, NbFe₂ had been reported to exhibit spin-density-wave order at low temperature.^{12–14} This interpretation is consistent with our measurements of the magnetic susceptibility, specific heat capacity, and magnetization.^{11,21,22} Based on these earlier results, we suggested that the low-temperature magnetic order in stoichiometric NbFe₂ may take the form of a SDW with an helical arrangement of the Fe spins. Direct evidence for SDW order in NbFe₂ from neutron scattering is still outstanding,²³ but a renewed attempt, using large single crystals grown in an infrared mirror furnace, is scheduled.

Figure 9 shows the real and imaginary components of the ac susceptibility. The distinct SDW peak is visible at 10 K, cutting off the Curie-Weiss-like behavior with decreasing temperature (see Fig. 6). The low transition temperature, for



FIG. 10. (Color online) Magnetization (upper panel) and (lower panel) ac susceptibility vs field *H*.

a *d*-metal compound, as well as the geometrically frustrated Kagomé structure of the Fe atoms in the C14 phase, might suggest a spin-glass transition. However, the measured Curie-Weiss temperature θ_{CW} is close to 1 K giving a frustration factor $f = \theta_{CW}/T_N$ of only 0.1. By contrast, the hallmark of frustration in local moment systems is a frustration factor f > 1. Moreover, the lack of a loss peak in χ'' at T_N , as well as the absence of a distinct frequency dependence of χ' argue against a spin-glass transition.

The analysis of the irreversible field-cooled (FC) and zero-field-cooled (ZFC) measurements (not shown) indicates a slight splitting of the magnetization curves at T_N , but the effect is so small that it would be difficult to attribute it to the main phase. Only at much lower temperatures ($T \le 3$ K) and at very low fields does this splitting become more pronounced, suggesting the formation of magnetic domains.²⁴ This behavior will be analyzed in the next section, where it resurfaces in a slightly Fe-doped sample.

An applied magnetic field *H* shifts the χ' peak to lower *T* and suppresses the magnetic order at a critical field $\mu_0 H_c \approx 0.6$ T at 100 mK, switching the system to a paramagnetic state. This can be seen more clearly in the magnetization and susceptibility plots in Fig. 10. Such a behavior could be interpreted as a metamagnetic transition from a paramagnetic (PM) state to another PM state with a higher magnetization in field. Such transitions are not uncommon in nearly FM metals, such as YCo₂,²⁵ but here, in contrast to the metamagnetic PM-to-PM case, H_c is shifted to lower values with increasing temperature (see color plot of Fig. 16).¹¹

A further indication of a bulk SDW transition at T_N is given by the Arrott plot in Fig. 11, in which the high field, high-temperature linear dependence of M^2 on H/M changes over to an arc with negative slope below T_N and H_c .¹³ Moreover, in the specific heat capacity (Fig. 12), the bulk SDW transition is visible as a small hump at T_N , when the data are



FIG. 11. (Color online) Arrott plots for four samples with y varying from small positive to negative values. The M^2 vs H/M lines are changing slope exactly at T_N where the extrapolation of the linear dependency crosses the y axis at negative value of M^2 .

compared with measurements in magnetic fields $H \ge H_c$.²² In Fig. 12, this hump is shown not only for y=0 but also for the slightly Fe-rich sample with y=0.007, which shows signatures of an SDW transition as well, at $T_N \simeq 18$ K. The entropy below the transition is very low, of the order of 4 mJ/Kmol. Figure 12 also shows the heat capacity of a slightly Nb-rich sample (y=-0.012). At this composition, no indication of a transition has been observed in the heat capacity down to 2 K. The inset of the Fig. 12 shows how the high-temperature data can be fitted with the electronic and phononic contribution,



FIG. 12. (Color online) Cp/T vs T^2 of two SDW samples with y=0.007 and y=0 and a paramagnetic one with y=-0.012. The inset shows the high-temperature data fitted with a linear function [Eq. (2)].



FIG. 13. Magnetoresistance of the stoichiometric sample (upper panel) and the slightly Fe-rich sample (lower panel) at different temperatures.

$$Cp/T = \gamma_{\rm el} + 5832 \, \mathrm{J/Kmol}\left(\frac{T^2}{\theta_D^3}\right),$$
 (2)

where the Sommerfeld coefficient is $\gamma_{el} = 14.2 \text{ mJ/K}^2 \text{ mol}$ and the Debye temperature is $\theta_D = 348$ K. This Sommerfeld coefficient agrees very well with that predicted by bandstructure calculations.^{26,27} However, Cp/T increases toward low temperatures T < 20 K and saturates below 1 K near a value of $\approx 48 \text{ mJ/K}^2 \text{ mol}$ (cf. Fig. 17),²⁸ three times larger than the band-structure value. The most dramatic signature of the SDW state is the behavior of the magnetoresistance ρ_M : although the SDW transition cannot be detected in the temperature dependence of the resistivity, the lowtemperature magnetoresistance ρ_M shows a jump of about 10% at the critical field in NbFe₂, and even more (\approx 17%) in the Fe-rich sample with y=0.007, as shown in Fig. 13. The magnitude of the jump thus appears to be related to T_N . At T_N this feature disappears and ρ_M follows a linear field dependence at high field, as in a paramagnetic metal.

In NbFe₂, all magnetic signatures below T_N are practically identical to those observed in the helically ordered compound MnAu₂, although the latter has a much higher transition temperature $T_N \approx 380$ K.²⁹ Furthermore, the large magnetic anisotropy observed in single crystals,¹⁶ and the critical field $\mu_0 H_c = 0.6$ T required to destroy the SDW state, which is a surprisingly low value when compared to T_N , suggest a long-wavelength incommensurate magnetic ground state. Moreover, the hexagonal C14 structure may favor helical order with propagation wave vector along the *c* axis, as in rare earth metals.³⁰ Finally, the large Stoner factor $S \approx 180$ $(I \cdot N(\epsilon_F) = 0.99 \approx 1)$, computed by comparing the value of χ at T_N to the one expected from band-structure DOS,^{26,27} and the high Wilson ratio of about 60,¹¹ indicates that this com-



FIG. 14. ac susceptibility for y=0.007 in a field of 0.025 T. The inset shows the ZFC-FC measurement performed with 50 Oe.

pound is on the border of a ferromagnetic instability. All these observations led us to propose that the SDW state grows out of the Q=0 ferromagnetism existing in Fe-rich NbFe₂, and develops into a $Q \neq 0$ helical state (as illustrated in Fig. 20.

C. SDW and FM mixed phase: $0 \le y \le 0.02$

To examine our hypothesis that the SDW state grows continuously out of the FM phase in Fe-rich samples, we measured a slightly Fe-doped sample with y=0.007, which is located in the phase diagram around the point of crossover between the pure FM phase and the SDW one. The ac susceptibility, plotted in Fig. 14, shows the same sharp peak as in NbFe₂, but at a higher temperature $T_N=18$ K. The bulk transition is also visible in the specific heat measurement shown in Fig. 12. The ZFC-FC curves separate exactly at T_N , whereas in the stoichiometric (y=0) sample a clear separation only occurred well below T_N .³¹ This might indicate spinglass behavior. However, as in the stoichiometric sample, we did not observe any shift of the χ' maximum at T_N with measurement frequency.

Hysteresis phenomena appear only at temperatures below 10 K, as shown, for example, in the Arrott plot of Fig. 5 at 2 K: in this plot the sample seems to be SDW between 10 and 18 K, but develops a remanent magnetization suggesting FM at lower temperatures. These features are illustrated more drastically in the ρ_M isotherms of Fig. 13: below T_N , the sample develops a clear negative magnetoresistance, and ρ_M jumps toward lower values as the critical field H_c is exceeded. The magnitude of the jump increases with decreasing temperature. Around 10 K, ρ_M begins to show hysteresis until at 2 K the "up" and "down" lines are well separated. The resulting curves suggest that the hysteresis phenomena detected in the stoichiometric samples are not a consequence of secondary FM phases, but are an intrinsic property of the SDW state.

Supporting evidence for this interpretation is provided in Fig. 15, which shows a sequence of magnetization and acsusceptibility isotherms for the same sample. As already



FIG. 15. Sequence of isothermals curves (magnetization and ac susceptibility) taken at temperatures above $T_N=18$ K and far below. This sequence shows how with decreasing temperature the hysteresis opens first at the critical field $\mu_0 H_c=0.15$ T and then becomes broad also at H=0.

pointed out by Crook and Cywinski,¹⁴ the hysteresis lines separate first not at H=0 but at finite fields. Following the measurements from the top to the bottom frame of the figure, we recognize the double peak at 15 K and $\mu_0 H_c = 0.15$ T as a signature of the SDW phase. As the temperature is lowered further, between 15 and 4 K hysteresis develops, which straddles H_c but does not yet reach H=0. Finally, close to 2 K, the size of the hysteretic field region includes zero field, so that the two hysteresis regions associated with $+H_c$ and $-H_c$ merge, giving rise to a FM remanent magnetization. All investigated samples with Fe concentrations $0 \le y \le 0.02$ display qualitatively the same phenomena, only the transition temperatures and the magnitudes of the effects change with increasing y. For $y \ge 0.02$ the samples seem to show only the FM character. The fact that in these very iron rich, purely FM samples the magnetoresistance ρ_M is much weaker and does not show any hysteresis suggests that the hysteretic signatures in M(H), $\rho_M(H)$, and $\chi'(H)$ are a characteristic property of this specific SDW state and indicates the presence of SDW domains, as recently discussed by Jaramillo et al.32

D. PM phase: $y \approx -0.015$

While Fe excess increases the SDW ordering temperature T_N , and eventually induces low-ordered-moment ferromagnetism, slight Nb excess weakens the SDW state and tunes the system toward paramagnetism and therefore to a QCP. This has been observed in the behavior of the magnetic susceptibility χ in two samples with y=-0.006 and y=-0.012. Figure 16 shows the contour plots of $\chi(H,T)$ for the two

MAGNETISM IN Nb_{1-v}Fe_{2+v}: COMPOSITION...



FIG. 16. (Color online) Contour plots of the magnetic susceptibility for Nb-rich samples at 2 K. The SDW phase in y=-0.006, defined by the bright zone, shrinks to a point for y=-0.012.

samples at 2 K: The ordered SDW phase for the y=-0.006 sample, defined by the bright zone in the *H*-*T* diagram, shrinks to a point for y=-0.012. A similar plot for a stoichiometric sample (y=0) is shown in Ref. 11.

Additional evidence for this evolution toward a paramagnetic state close to y=-0.015 is given by the sequence of Arrott plots shown in Fig. 11. For y=0 and y=-0.006, the M^2 vs H/M curves change appearance at T_N : for $T>T_N$, the curves bend toward the left, whereas for $T < T_N$ they bend toward the right. By contrast, for y=-0.012, no such change in behavior occurs down to 2 K.

The heat capacity is less sensitive to the magnetic transition and so the weak anomaly that was observed in the stoichiometric sample at 10 K, has not yet been observed in either Nb-rich sample. The surprisingly high value of the Sommerfeld coefficient γ at low temperature for all the samples close to the stoichiometric composition has been considered as indication of the presence of strong spin fluctuations and of the vicinity of the system to a QCP.^{20,22} Approaching the QCP by increasing the Nb content, γ gradually increases and, for y=-0.012, it follows a logarithmic temperature dependence down to about 0.6 K (Fig. 17).

Non-Fermi-liquid behavior has also been observed in the temperature dependence of the electrical resistivity. The inset of Fig. 18 shows the temperature dependence of the resistiv-



FIG. 17. (Color online) Specific heat of samples with y close to 0. The low-temperature Schottky contribution for the sample with y=0.015 has been subtracted.



FIG. 18. (Color online) Magnetoresistance of the sample with y=-0.012 at three temperatures. Inset: $T^{3/2}$ dependence of the resistivity; below 0.6 K the measured curve deviates strongly from the linear fit.

ity $\rho(T)$: its power-law exponent $n=d(\log \Delta \rho)/d(\log T)$ varies from 3/2 to 5/3 between 3 and 0.6 K and crosses over to the FL exponent 2 below 0.6 K.²⁰ Combining both observations, the *T* dependence of the heat capacity and of the resistivity below 3 K, the sample Nb_{1.012}Fe_{1.988} seems to be close to a ferromagnetic QCP, where $C/T \propto \log(T)$ and $\rho \propto T^{5/3}$ have been predicted for a three-dimensional (3D) FM metal.^{33,34} Below 0.6 K, Nb_{1.012}Fe_{1.988} experiences a crossover to the conventional FL state.

Magnetoresistance measurements below 2 K (Fig. 18) suggest that Nb_{1-y}Fe_{2+y} does not have a paramagnetic ground state at y=-0.012. A little drop (less than 1%) in ρ_M is observed at 2 K and clear hysteresis phenomena develop below 1 K. This indicates the presence of very weak SDW order below $T_N \approx 2$ K, which develops domains at $T \leq 1$ K.

These findings contrast with our observations on a highquality Nb-rich ($y \approx -0.01$) single crystal, in which non-Fermi-liquid phenomena have been observed down to the lowest temperature.¹¹ Despite residual low-temperature order below $T_N \approx 2.8$ K, presumably with a very small ordered moment, the resistance of the single crystal follows a $T^{3/2}$ power law down to 50 mK, and its heat capacity coefficient C_p/T increases logarithmically with decreasing temperature, without any indication of a crossover to a FL state. The return of our Nb-rich polycrystal with y=-0.012 to FL behavior at low temperature, when such a return to FL behavior is not observed in a single crystal with a similar composition, may be attributed to the lower quality of the polycrystal either in terms of its purity or its homogeneity.

We interpret these findings as the first clear indication of a logarithmic breakdown of the Fermi-liquid state in a transition metal antiferromagnet: the fact that the precise power-law exponent varies between 3/2 and 5/3 and the discrepancies between the phenomena observed in single and polycrystals suggest that the precise form of $\rho(T)$ may depend on stoichiometry and on the sample orientation and that present data cannot decide whether FM or SDW spin fluctuations determine the low-temperature behavior.



FIG. 19. (Color online) Proposed new phase diagram for $Nb_{1-\nu}Fe_{2+\nu}.$

IV. PHASE DIAGRAM AND CONCLUSION

We propose a refined version of the magnetic phase diagram for $Nb_{1-y}Fe_{2+y}$ (Fig. 19), obtained by combining all the thermodynamic and transport measurements discussed in this paper. Dashed lines indicate the likely limits of the FM and SDW state. We can address the five questions outlined in the introduction with the help of this phase diagram.

For $|y| \ge 0.02$, distinct FM phases have been observed at low temperature. Our data confirm that the FM features do not originate from an impurity phase, but are intrinsic to the C14 Laves phase Nb_{1-y}Fe_{2+y}. The *M*-*H* isotherms are qualitatively different in the Fe-rich and Nb-rich FM phases, suggesting that the two states are of a different nature.

Between the FM phases, a distinct non-FM phase has been observed: susceptibility, magnetization, resistivity, and heat capacity results, many of which are similar to observations in well-known spin helical compounds such as MnAu₂,²⁹ Y₂Fe₁₇,³⁵ and MnSi,^{36,37} suggest that the magnetic order in stoichiometric NbFe2 assumes the form of a SDW, possibly a long-wavelength helical or spiral state. On the Fe-rich side, we believe that the SDW state with a small wave vector $Q \neq 0$ connects continuously to the FM state at the Lifshitz point, which is located around y=+0.02.³⁸ The schematic Fig. 20 shows the possible evolution of the wavevector-dependent generalized susceptibility $\chi(q)$, as $Nb_{1-\nu}Fe_{2+\nu}$ is tuned away from the Lifshitz point: with decreasing iron content y the minimum shifts from q=0 to low $q \rightarrow Q_1 \rightarrow Q_2$, and Nb_{1-v}Fe_{2+v} changes from FM to longwavelength SDW order. The small value of the $\chi(q)^{-1}$ intercept at q=0 would explain the high value of the measured ac susceptibility at low temperature. Assuming that the dispersion stiffness value of NbFe₂ is similar to the one measured in ZrZn₂, we could estimate an ordering wave vector of about 0.05 Å⁻¹ for NbFe₂.¹¹ Until the precise nature of the SDW state can be observed using a microscopic probe, ideally neutron scattering, this scenario remains strongly hypothetical. Alternative origins of the dual nature of NbFe2-nearly ferromagnetic and yet antiferromagneticinclude (i) coincidental peaks of the wave-vector-dependent susceptibility at large wave vector (SDW) and at q=0 (FM)



FIG. 20. (Color online) Postulated evolution of the inverse of the wave vector q dependent susceptibility χ_q with doping. With decreasing iron content y the minimum of the dispersion χ_q^{-1} may shift from q=0 to finite q. This shift is accompanied by a change from FM to SDW, possibly helical, magnetic order. The point in the phase diagram, where this crossover takes place, is called Lifshitz point (Fig. 19).

and (ii) a very weakly dispersive wave-vector-dependent susceptibility $\chi(q)$, peaking at a large, antiferromagnetic wave vector but remaining unusually large even at q=0. Neutron-scattering experiments on recently grown, large single crystals of NbFe₂ are planned, and will hopefully help to resolve this controversy.

Once the SDW state is entered, SDW domains form with different order parameter polarizations or ordering wavevector directions, creating a set of domain walls. A weak magnetic field H is sufficient to destroy the SDW and induce a weakly spin-aligned paramagnetic state. The transition in H from helical to the paramagnetic state probably occurs through a number of successive steps, which are linked with domain wall jumps and define the size of the hysteresis. We are planning to investigate this interpretation by high precision measurements of the magnetic field dependence of the magnetization and of the electrical resistivity at very low temperatures. By reducing the Fe content (for small negative y), the SDW state can be suppressed completely, leading to a magnetic quantum critical point and leaving a small paramagnetic region around $y \approx -0.015$. Here, magnetic fluctuations dominate the temperature dependence of resistivity and specific heat. Since the critical exponents fall in between those expected for FM and SDW spin fluctuations and the QCP is located between the two different phases, the nature of the fluctuations requires clarification.

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- *Present address: Max-Plank-Institut für Chemische Physik fester Stoffe, Nöthnitzer Strasse 40, D-01187 Dresden, Germany; manuel.brando@cpfs.mpg.de
- [†]Present address: Cavendish Laboratory, University of Cambridge, J J Thomson Avenue, Cambridge CB3 0HE, UK.
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