

Soft x-ray absorption spectroscopy and magnetic circular dichroism study of the valence and spin states in spinel MnFe_2O_4

J.-S. Kang,^{1,*} G. Kim,¹ H. J. Lee,¹ D. H. Kim,¹ H. S. Kim,¹ J. H. Shim,² S. Lee,² Hangil Lee,³ J.-Y. Kim,³ B. H. Kim,⁴ and B. I. Min⁴

¹*Department of Physics, The Catholic University of Korea, Bucheon 420-743, Korea*

²*Department of Physics, Korea Advanced Institute of Science and Technology, Daejeon 305-701, Korea*

³*Pohang Accelerator Laboratory (PAL), POSTECH, Pohang 790-784, Korea*

⁴*Department of Physics, POSTECH, Pohang 790-784, Korea*

(Received 23 July 2007; published 17 January 2008)

The valence and spin states of MnFe_2O_4 spinel oxide have been investigated by employing soft x-ray absorption spectroscopy (XAS) and x-ray magnetic circular dichroism (XMCD). The measured Mn $2p$ and Fe $2p$ XAS spectra indicate that Mn and Fe ions are nearly divalent (Mn^{2+}) and trivalent (Fe^{3+}), respectively. Our XAS and XMCD spectra for MnFe_2O_4 do not show clear evidence of mixed-valent states but provide evidence for the inversion of both Mn and Fe ions, with $T_d/O_h \sim 0.8/0.2$ for Mn ions and $T_d/O_h \sim 0.1/0.9$ for Fe ions, respectively. Based on our data, MnFe_2O_4 can be described either by the single-valence states of $(\text{Mn}_{0.8}^{2+}\text{Fe}_{0.2}^{3+})_A[\text{Fe}_{1.8}^{3+}\text{Mn}_{0.2}^{2+}]_B\text{O}_4$ or by the mixed-valence states of $(\text{Mn}_{0.8}^{2+}\text{Fe}_{0.2}^{3+})_A[\text{Fe}_{1.6}^{3+}\text{Fe}_{0.2}^{2+}\text{Mn}_{0.2}^{3+}]_B\text{O}_4$.

DOI: [10.1103/PhysRevB.77.035121](https://doi.org/10.1103/PhysRevB.77.035121)

PACS number(s): 78.70.Dm, 87.64.K-, 75.50.Gg, 74.25.Jb

The $AB_2\text{O}_4$ -type spinel oxides, which are the most abundant structures on earth, have attracted renewed attention because of their possible application as new functional materials as well as many interesting phenomena observed, such as Jahn-Teller (JT) effects and phase separations.¹ In $AB_2\text{O}_4$ spinels, a cation occupies either the tetrahedral (T_d) A site or the octahedral (O_h) B site. When both A and B ions are magnetic ions, such as in Fe_3O_4 , a ferrimagnetic ordering is often observed since the main A - B interaction is antiferromagnetic (AFM).² MnFe_2O_4 is one of the well-known ferrimagnetic spinel oxides with a partially inverted spinel structure described by the formula $(\text{Mn}_{1-y}\text{Fe}_y)_A[\text{Fe}_{2-y}\text{Mn}_y]_B\text{O}_4$, where y is called the inversion parameter. The inversion parameter is known to vary from 0 to 0.2 depending on the ways of thermal treatment, among which 0.2 is the most usual.³ If all the Mn ions are divalent (Mn^{2+}) and all the Fe ions are trivalent (Fe^{3+}), then the magnetic moment is expected to be $5\mu_B/\text{f.u.}$ in the ionic model. From magnetic scattering measurements,³ the magnetic moment of MnFe_2O_4 was found to be $\sim 4.6\mu_B/\text{f.u.}$ This reduced magnetic moment of MnFe_2O_4 has drawn much attention.⁴

There have been several models that explain the reduced magnetic moment of MnFe_2O_4 . Harrison *et al.*⁵ proposed the mixed valency for Mn and Fe ions such that some of Mn ions are trivalent (Mn^{3+}) and that some of Fe ions are divalent (Fe^{2+}), assuming the collinear ferrimagnetic spin order between Mn^{3+} and Fe^{2+} pairs in B sites. The structural formula in this case was assumed to be $(\text{Mn}_{1-y}^{2+}\text{Fe}_y^{3+})_A[\text{Fe}_{2-2y}^{3+}\text{Fe}_y^{2+}\text{Mn}_y^{3+}]_B\text{O}_4$. On the other hand, Mössbauer measurements of MnFe_2O_4 revealed little Fe^{2+} ions in B sites, and so the canted ferrimagnetic spin order was proposed.⁶ The canted spin structure was supported by the $M(T)$ curve.⁷ However, nuclear magnetic resonance (NMR) experiments showed no evidence of spin canting.^{8,9} Based on recent NMR measurements, some of the authors of this work¹⁰ reported the existence of Fe^{3+} ions in A sites and Mn^{3+} and Fe^{2+} ions in B sites. Then, in order to explain the reduced magnetic moment, an AFM order among Fe spins in

B sites was proposed in addition to the AFM order between Fe spins in A and B sites. Hence, the above models are controversial to one another, and the valence states and the spin ordering in MnFe_2O_4 are still open issues.

Therefore, it is important to determine the valence states of Mn and Fe ions in MnFe_2O_4 and their distribution between $T_d(A)$ and $O_h(B)$ sites more precisely. Soft x-ray absorption spectroscopy¹¹⁻¹³ (XAS) and soft x-ray magnetic circular dichroism^{14,15} (XMCD) are powerful experimental tools for studying the valence and spin states of transition-metal (T) ions in solids and the element-specific local magnetic moments of spin (m_s) and orbital (m_l) components, respectively. In both XAS and XMCD, photons at specific characteristic energies are absorbed to produce the transition of a core electron to an empty state above the Fermi level, which is governed by the dipole selection rules. $T\ 2p$ XAS and XMCD spectra of transition-metal oxides often show the multiplet structures, and their line shapes are strongly dependent on the occupied $3d$ electron configurations, the crystal field, the spin-orbit and electron-electron interactions within the T atom, and the hybridization of $3d$ electrons to other valence electrons.

We have investigated the electronic structures of MnFe_2O_4 by employing XAS and XMCD. We have determined the valence and spin states of Mn and Fe ions and the amount of the inverted structure. This is the first XMCD study on *bulk* stoichiometric MnFe_2O_4 . As to the related spinel oxides, the XMCD study has been reported for $\text{Zn}_{1/3}\text{Mn}_{2/3}\text{Fe}_2\text{O}_4$ (Ref. 16) and nanocrystalline thin films of $\text{Mn}_x\text{Fe}_{3-x}\text{O}_4$.¹⁷

The MnFe_2O_4 sample used in this study is a commercially available polycrystalline powder of 99.9% purity synthesized by KOJUNDO Chemical. Scanning electron microscope measurements showed that the sizes of samples are larger than a few micrometers. X-ray powder diffraction measurements also showed the sharp peaks that are characteristic of the single-phase bulk spinel structure.¹⁸ XAS and XMCD experiments were performed at the 2A EPU beamline of the

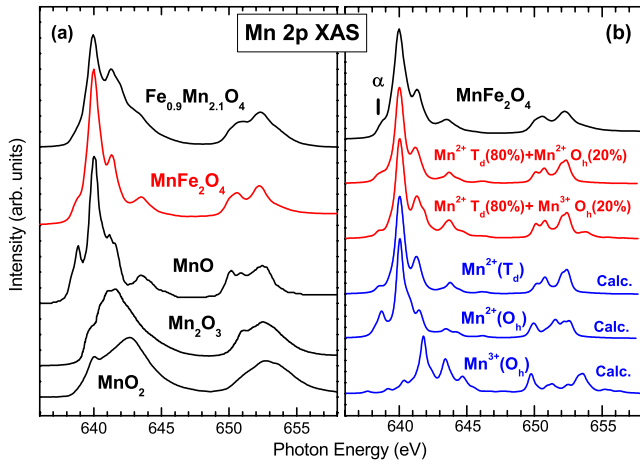


FIG. 1. (Color online) (a) Comparison of the Mn 2*p* XAS of MnFe₂O₄ to those of Fe_{0.9}Mn_{2.1}O₄ and the reference Mn oxides, such as MnO (Mn²⁺) (Ref. 21), Mn₂O₃ (Mn³⁺) (Ref. 22), and MnO₂ (Mn⁴⁺) (Ref. 21). (b) Comparison of the Mn 2*p* XAS of MnFe₂O₄ to the calculated Mn 2*p* XAS for *T_d* Mn²⁺, *O_h* Mn²⁺, and *O_h* Mn³⁺ ions and their weighted sums (red lines).

PAL by using the circularly polarized light with the degree of circular polarization >90%. The XAS and XMCD data were collected in the total electron yield (TEY) mode.¹⁹ The XMCD spectra were taken for a fixed helicity of light by reversing the applied magnetic field (~0.7 T) at each *hν*. The base pressure of the system was ~5 × 10⁻¹⁰ Torr. All the data were obtained at *T* ~ 80 K. The total resolution for XAS was less than 100 meV, while that for XMCD was ~120 meV at the Mn 2*p* and Fe 2*p* edges.

Figure 1(a) compares the measured Mn 2*p* XAS spectrum of MnFe₂O₄ to those of Fe_{0.9}Mn_{2.1}O₄ single crystal²⁰ and the reference Mn oxides of MnO (Mn²⁺),²¹ Mn₂O₃,²² and MnO₂ (Mn³⁺).²¹ The Mn 2*p* XAS spectrum of MnFe₂O₄ is qualitatively similar to that of MnO but quite different from those of Mn₂O₃ and MnO₂, suggesting that Mn ions in MnFe₂O₄ are nearly divalent. In contrast, the line shape of Fe_{0.9}Mn_{2.1}O₄ is rather different from that of MnFe₂O₄, which appears to be a mixture of MnO and Mn₂O₃, indicating the Mn²⁺-Mn³⁺ mixed-valent states in Fe_{0.9}Mn_{2.1}O₄. This comparison reveals clearly that Mn ions in MnFe₂O₄ are nearly divalent, while Mn ions in Fe_{0.9}Mn_{2.1}O₄ are mixed valent.

To confirm this argument, we have compared the Mn 2*p* XAS spectrum of MnFe₂O₄ to the calculated Mn 2*p* XAS spectra in Fig. 1(b). The calculated spectra were obtained by employing the ligand-field multiplet (LFM) model^{11,12} by including the spin-orbit interaction between 3*d* electrons. The three calculated XAS spectra (blue lines) represent those for Mn²⁺(3*d*⁵) under the *T_d* symmetry, Mn²⁺(3*d*⁵) under the *O_h* symmetry, and Mn³⁺(3*d*⁴) under the *O_h* symmetry, with the crystal field energy 10*Dq*=0.6 eV, 10*Dq*=1.2 eV, and 10*Dq*=1.2 eV, respectively. The two red curves represent the weighted sums of [80% Mn_A²⁺(*T_d*) and 20% Mn_B²⁺(*O_h*)] to describe the single-valence states of Mn ions and [80% Mn_A²⁺(*T_d*) and 20% Mn_B³⁺(*O_h*)] to describe the mixed-valence states of Mn ions. In Fig. 1(b), it is seen that the fitting with the single-valent Mn ions seems to be slightly

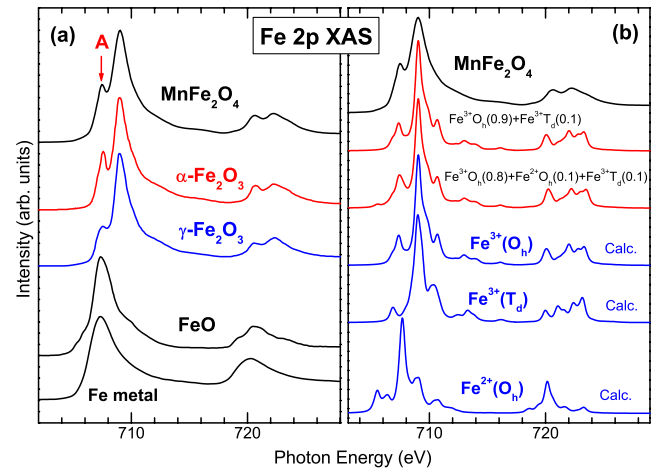


FIG. 2. (Color online) (a) Comparison of the Fe 2*p* XAS of MnFe₂O₄ to those of α-Fe₂O₃ (Ref. 24), γ-Fe₂O₃ (Ref. 23), FeO (Ref. 24), and Fe metal (Ref. 25). (b) Comparison of the Fe 2*p* XAS of MnFe₂O₄ to the calculated Fe 2*p* XAS for *O_h* Fe³⁺, *T_d* Fe³⁺, and *O_h* Fe²⁺ ions and their weighted sums (red lines).

better than that with the mixed-valent Mn ions, in particular, for the feature α around *hν* ~ 639 eV. However, considering the uncertainties in fitting, both fits should be considered to be good. Hence, at the moment, we cannot finalize whether MnFe₂O₄ is (Mn_{0.8}Fe_{0.2}³⁺)_A[Fe_{1.8}Mn_{0.2}²⁺]_BO₄ or (Mn_{0.8}Fe_{0.2}³⁺)_A[Fe_{1.6}Fe_{0.2}²⁺Mn_{0.2}³⁺]_BO₄. In both cases, the inversion parameter *y* ≈ 0.2 gives the best fitting for Mn 2*p* XAS of MnFe₂O₄.

Figure 2(a) shows the measured Fe 2*p* XAS spectrum of MnFe₂O₄. As a guide of the valence states of Fe ions, it is compared to those of α-Fe₂O₃ (Refs. 23,24) and γ-Fe₂O₃ (Refs. 23) both as formally trivalent Fe³⁺ oxides (3*d*⁵) but having different local symmetries, FeO (Ref. 24) as a formally divalent Fe²⁺ oxide (3*d*⁶), and Fe metal.²⁵ The line shape of the Fe 2*p* XAS spectrum of MnFe₂O₄ is very similar to those of both α-Fe₂O₃ and γ-Fe₂O₃ but quite different from those of FeO and Fe metal, indicating that the valence states of Fe ions in MnFe₂O₄ are mainly trivalent (3+). The minor difference in the feature “A” for MnFe₂O₄ from those for α-Fe₂O₃ and γ-Fe₂O₃ seems to indicate that Fe ions in MnFe₂O₄ occupy not only *B*(*O_h*) sites but also *A*(*T_d*) sites. This is because Fe³⁺ ions in α-Fe₂O₃ are located only at *O_h* sites, whereas Fe³⁺ ions in γ-Fe₂O₃ are at both *T_d* and *O_h* sites.²⁸ A similar trend has been observed in the calculated XAS for Fe_B³⁺(*O_h*) and Fe_A³⁺(*T_d*) ions.^{26–28} This argument is also supported in Fig. 4.

In Fig. 2(b), we have compared the measured Fe 2*p* XAS spectrum of MnFe₂O₄ to the calculated Fe 2*p* XAS spectra, obtained from the LFM calculations. The three calculated XAS spectra (blue lines) represent those for Fe³⁺(3*d*⁵) under *O_h* with 10*Dq*=1.5 eV, Fe³⁺(3*d*⁵) under *T_d* with 10*Dq*=1.2 eV, and Fe²⁺(3*d*⁶) under *O_h* with 10*Dq*=1.5 eV. The two red curves represent the weighted sums of [90% Fe_B³⁺(*O_h*) and 10% Fe_A³⁺(*T_d*)] to describe the single-valence states of Fe ions and [80% Fe_B³⁺(*O_h*), 10% Fe_B²⁺(*O_h*), and 10% Fe_A³⁺(*T_d*)] to describe the mixed-valence states of Fe

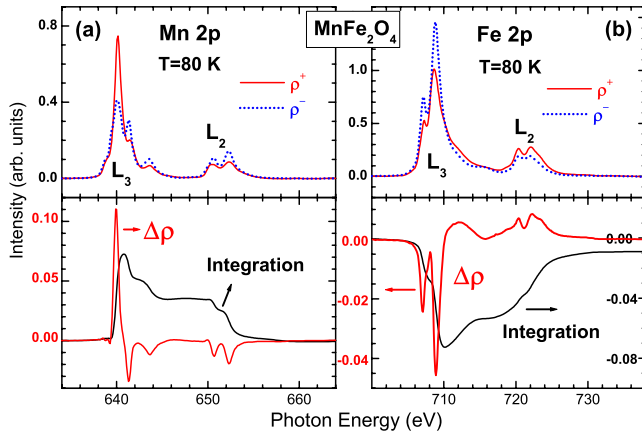


FIG. 3. (Color online) (a) The Mn 2p XAS spectra of MnFe_2O_4 , obtained with different photon helicities (ρ_+ and ρ_-), the Mn 2p XMCD spectrum (red) ($\Delta\rho = \rho_+ - \rho_-$), and its integrated value (black). (b) Similarly for the Fe 2p states.

ions. The agreement between the LFM calculations and the measured Fe 2p XAS spectrum is not as good as that for Mn 2p XAS [Fig. 1(b)], and the quality of fitting is comparable to each other. Hence, MnFe_2O_4 can be described either by $(\text{Mn}_{0.8}^{2+}\text{Fe}_{0.2}^{3+})_A[\text{Fe}_{1.8}^{3+}\text{Mn}_{0.2}^{2+}]_B\text{O}_4$ or by $(\text{Mn}_{0.8}^{2+}\text{Fe}_{0.2}^{3+})_A[\text{Fe}_{1.6}^{3+}\text{Fe}_{0.2}^{2+}\text{Mn}_{0.2}^{3+}]_B\text{O}_4$.

Figure 3 shows the two Mn 2p and Fe 2p absorption spectra of MnFe_2O_4 for different magnetization directions, which are obtained with the photon helicity parallel to (ρ_+) and antiparallel to (ρ_-) the applied magnetic field. The bottom panels show the XMCD spectra obtained from the difference ($\Delta\rho$) between ρ_+ and ρ_- ($\Delta\rho = \rho_+ - \rho_-$) and their integrated values ($\int\Delta\rho$). The spectra of ρ_+ and ρ_- are roughly divided into the L_3 ($2p_{3/2}$) and L_2 ($2p_{1/2}$) regions. According to the sum rule,¹⁵ $\int\Delta\rho$ can be used to estimate the element-specific m_l value.

The Mn 2p XMCD is similar to that in the literature.^{17,29} This figure reveals the following features. First, the polarity of the Fe 2p XMCD is opposite to that of the Mn 2p XMCD, which indicates the antiparallel alignment of the spin moments between Fe and Mn ions. This finding is consistent with the ferrimagnetic behavior of MnFe_2O_4 . Note, however, that this opposite polarity between Mn 2p and Fe 2p XMCD does not tell whether the ferrimagnetic spin order is collinear or noncollinear. Secondly, the integrated value of the Fe 2p XMCD over the whole (L_3+L_2) range, $\int_{L_3+L_2}\Delta\rho$, is finite, while that for the Mn 2p XMCD is zero. Since $\int_{L_3+L_2}\Delta\rho$ is proportional m_l ,¹⁵ this difference indicates that the orbital moments of Fe ions are not completely quenched in contrast to the quenched orbital moments of Mn ions. The orbital moment should be quenched for Fe^{3+} ions under both O_h and T_d symmetries. Thus, the nonzero m_l of Fe ions might reflect the existence of Fe^{2+} ions at $B(O_h)$ sites, which is favored by the mixed-valent Fe states with a structural formula of $(\text{Mn}_{0.8}^{2+}\text{Fe}_{0.2}^{3+})_A[\text{Fe}_{1.6}^{3+}\text{Fe}_{0.2}^{2+}\text{Mn}_{0.2}^{3+}]_B\text{O}_4$.

Figure 4 compares the Fe 2p XMCD spectrum of MnFe_2O_4 to those of GaFeO_3 (Ref. 23) and $\gamma\text{-Fe}_2\text{O}_3$.^{23,28} The weighted sum (black line) of [GaFeO_3 (70%) and $\gamma\text{-Fe}_2\text{O}_3$ (30%)] is superposed upon MnFe_2O_4 . Note that the

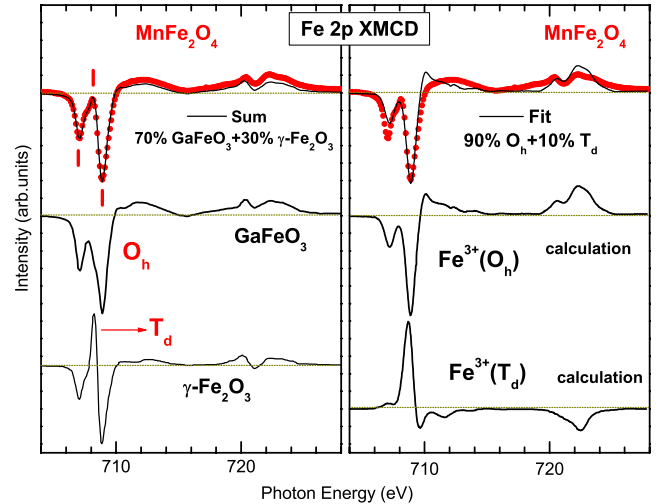


FIG. 4. (Color online) Comparison of the Fe 2p XMCD spectrum of MnFe_2O_4 to those of $\gamma\text{-Fe}_2\text{O}_3$ (Ref. 28) and GaFeO_3 (Ref. 23) (left) and their weighted sum (left) and to those of the calculated Fe 2p XMCD spectra for $\text{Fe}_B^{3+}(O_h)$ (Ref. 28) and $\text{Fe}_A^{3+}(T_d)$ (Ref. 28) and their weighted sum (right).

XMCD spectrum of GaFeO_3 has the contribution only from $O_h \text{Fe}^{3+}$ ions,²³ whereas that of $\gamma\text{-Fe}_2\text{O}_3$ has a mixture of $\text{Fe}_B^{3+}(T_d)$ and $\text{Fe}_A^{3+}(O_h)$ ions²⁸ (see the labels in Fig. 4). In $\gamma\text{-Fe}_2\text{O}_3$, the sign of the dichroic contribution of $\text{Fe}_A^{3+}(T_d)$ ions to XMCD is opposite to that of $\text{Fe}_B^{3+}(O_h)$ ions because the main magnetic coupling between A and B sites is antiferromagnetic. The good agreement between MnFe_2O_4 and the weighted sum of GaFeO_3 and $\gamma\text{-Fe}_2\text{O}_3$ indicates the existence of $\text{Fe}_A^{3+}(T_d)$ ions in MnFe_2O_4 (see the feature around $h\nu \sim 708$ eV). Considering 100% $O_h \text{Fe}^{3+}$ ions for GaFeO_3 and $\text{Fe}_A^{3+}/\text{Fe}_B^{3+} = T_d/O_h = 3/5$ for $\gamma\text{-Fe}_2\text{O}_3$, the fitting yields the estimated T_d/O_h ratio in MnFe_2O_4 to be $T_d/O_h = 0.11/0.89 \approx 0.1/0.9$. This value agrees with the estimated inversion parameter of $y \sim 0.2$ in Fig. 1 within the experimental error.

The right panel of Fig. 4 compares the Fe 2p XMCD spectrum of MnFe_2O_4 to the calculated XMCD spectra for $\text{Fe}_B^{3+}(O_h)$ (Ref. 28) and $\text{Fe}_A^{3+}(T_d)$ (Ref. 28) and their weighted sum (black line) of [$\text{Fe}_B^{3+}(O_h)$ (90%) and $\text{Fe}_A^{3+}(T_d)$ (10%)]. Here, we have assumed the single-valence state of Fe^{3+} at both $B(O_h)$ and $A(T_d)$ sites. A reasonably good agreement is found between experiment and calculations. Therefore, this comparison supports the inversion of $\sim 10\%$ of Fe ions from $B(O_h)$ sites to $A(T_d)$ sites.

Figure 5 compares the Mn 2p XMCD spectrum of MnFe_2O_4 to the calculated Mn 2p XMCD spectra for different valence states of Mn ions,³⁰ including $\text{Mn}^{2+}(3d^5)$ and $\text{Mn}^{3+}(3d^4)$ and their weighted sums of [50% Mn^{2+} and 50% Mn^{3+} and 80% Mn^{2+} and 20% Mn^{3+}]. These calculations describe the multiplet features of XMCD of $\text{Mn}^{2+}(3d^5)$ and $\text{Mn}^{3+}(3d^4)$ qualitatively. In particular, the negative feature (marked with an arrow) around $h\nu \sim 641$ eV increases with the increasing $\text{Mn}^{3+}(3d^4)$ component.¹⁷ On the other hand, both 100% Mn^{2+} calculation and the sum of 80% Mn^{2+} and 20% Mn^{3+} calculations seem to agree to the measured Mn 2p

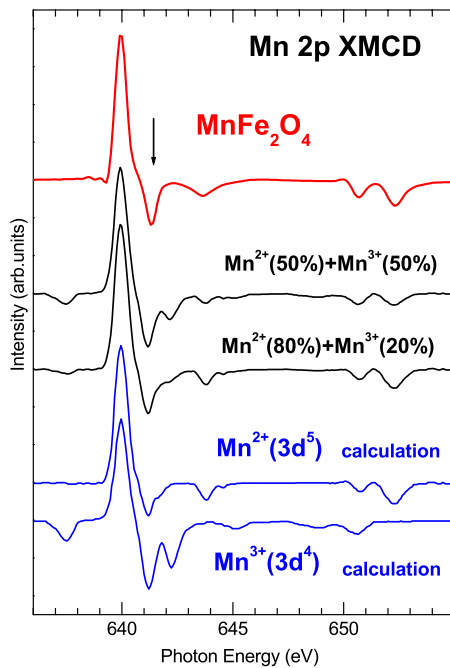


FIG. 5. (Color online) Comparison of the Mn 2p XMCD of MnFe_2O_4 to the calculated Mn 2p XMCD for $\text{Mn}^{2+}(3d^6)$, $\text{Mn}^{2+}(3d^5)$, and $\text{Mn}^{3+}(3d^4)$ ions (Ref. 30) and their weighted sums.

XMCD of MnFe_2O_4 within the fitting uncertainty. This finding is again in agreement with that of Mn 2p XAS (see Fig. 1).

Our XAS/XMCD data indicate that Mn ions are nearly divalent (Mn^{2+}) and that Fe ions are nearly trivalent (Fe^{3+}), with the inversion of $y \sim 0.2$. On the other hand, we cannot determine exactly whether MnFe_2O_4 is $(\text{Mn}_{0.8}^{2+}\text{Fe}_{0.2}^{3+})_A[\text{Fe}_{1.8}^{3+}\text{Mn}_{0.2}^{2+}]_B\text{O}_4$ or $(\text{Mn}_{0.8}^{2+}\text{Fe}_{0.2}^{3+})_A[\text{Fe}_{1.6}^{3+}\text{Fe}_{0.2}^{2+}\text{Mn}_{0.2}^{3+}]_B\text{O}_4$, as explained above. Note that the dominating Fe^{3+} and Mn^{2+} ions in MnFe_2O_4 obtained from the present analysis are compatible with the spinel cubic structure of MnFe_2O_4 with the negligible JT distortion because both $\text{Fe}^{3+}(3d^5)$ and $\text{Mn}^{2+}(3d^5)$ ions are not JT active.

Now, we discuss on the reduced magnetic moment in MnFe_2O_4 . In $AB_2\text{O}_4$ -type spinel oxides, the antiferromag-

netic coupling between magnetic moments of A and B sites is dominating. Then, in the ionic model, the configuration of $(\text{Mn}_{0.8}^{2+}\text{Fe}_{0.2}^{3+})_A[\text{Fe}_{1.8}^{3+}\text{Mn}_{0.2}^{2+}]_B\text{O}_4$ for MnFe_2O_4 has the magnetic moment of $5\mu_B/\text{f.u.}$, while $(\text{Mn}_{0.8}^{2+}\text{Fe}_{0.2}^{3+})_A[\text{Fe}_{1.6}^{3+}\text{Fe}_{0.2}^{2+}\text{Mn}_{0.2}^{3+}]_B\text{O}_4$ for MnFe_2O_4 has the magnetic moment of $4.6\mu_B/\text{f.u.}$ The magnetic moment of the latter configuration is consistent with the observed magnetic moment. Nevertheless, one should be cautious because, in solids, the magnetic moment would not be explained simply in terms of the ionic model due to the rather complicated solid-state effects, such as the hybridization effect between T and ligand oxygen ions. Indeed, the existing band-structure calculations^{31–34} indicate that the magnetic moments of Mn and Fe ions are reduced a lot from the values predicted by the ionic model. For example, Penicaud *et al.*³¹ obtained the magnetic moments of $(2.96–3.96)\mu_B$ for Mn and $(3.29–3.75)\mu_B$ for Fe, which are much smaller than $5.0\mu_B$ of both Mn^{2+} and Fe^{3+} and $4.0\mu_B$ of both Mn^{3+} and Fe^{2+} in the ionic model. The reference oxides employed in Figs. 2 and 4, which are assumed to have the formal integer valences, might also have some mixed configurations of valence states through the hybridization between T and ligand oxygen ions.

In conclusion, the valence and spin states of MnFe_2O_4 have been investigated by performing XAS and XMCD measurements near the Mn 2p and Fe 2p absorption edges. The valence states of Mn and Fe ions are found to be nearly divalent (Mn^{2+}) and trivalent (Fe^{3+}), respectively. The analysis of the XAS and XMCD line shapes provides evidence for the inversion with $y \sim 0.2$. Based on our data, MnFe_2O_4 can be described either by $(\text{Mn}_{0.8}^{2+}\text{Fe}_{0.2}^{3+})_A[\text{Fe}_{1.8}^{3+}\text{Mn}_{0.2}^{2+}]_B\text{O}_4$ or by $(\text{Mn}_{0.8}^{2+}\text{Fe}_{0.2}^{3+})_A[\text{Fe}_{1.6}^{3+}\text{Fe}_{0.2}^{2+}\text{Mn}_{0.2}^{3+}]_B\text{O}_4$. The polarity of the Fe 2p XMCD is opposite to that of the Mn 2p XMCD, which reveals the antiparallel alignment of the spin moments between Fe and Mn ions. Further, the orbital moments of Fe ions are not completely quenched, while the orbital moments of Mn ions are nearly quenched in MnFe_2O_4 .

This work was supported by the KRF (KRF-2006-311-C00277), by the KOSEF (R01-2006-000-10369-0), and by the KOSEF through the CSCMR at SNU and the eSSC at POSTECH. B.I.M. also acknowledges the support from POSTECH research fund. The PAL is supported by the MOST and POSCO in Korea.

*kangjs@catholic.ac.kr

¹C. L. Zhang, S. Yeo, Y. Horibe, Y. J. Choi, S. Guha, M. Croft, and S.-W. Cheong, *Appl. Phys. Lett.* **90**, 133123 (2007).

²J. B. Goodenough, *Magnetism and the Chemical Bond* (Krieger, New York, 1976).

³J. M. Hastings and L. M. Corliss, *Phys. Rev.* **104**, 328 (1956).

⁴Z. Šimša and V. A. M. Brabers, *IEEE Trans. Magn.* **11**, 1303 (1975).

⁵F. W. Harrison, W. P. Osmond, and R. W. Teale, *Phys. Rev.* **106**, 865 (1957).

⁶G. A. Sawatzky, F. van der Woode, and A. H. Morrish, *Phys. Rev.*

187, 747 (1969).

⁷G. F. Dionne, *J. Appl. Phys.* **63**, 3777 (1988).

⁸H. Yasuoka, *J. Phys. Soc. Jpn.* **19**, 1182 (1964).

⁹T. Shirakashi and T. Kubo, *Am. Mineral.* **64**, 599 (1979).

¹⁰J. H. Shim, S. Lee, and B. I. Min, *Phys. Rev. B* **75**, 134406 (2007).

¹¹F. M. F. de Groot, J. C. Fuggle, B. T. Thole, and G. A. Sawatzky, *Phys. Rev. B* **42**, 5459 (1990).

¹²G. van der Laan and I. W. Kirkman, *J. Phys.: Condens. Matter* **4**, 4189 (1992).

¹³S. C. Wi, J.-S. Kang, J. H. Kim, S.-B. Cho, B. J. Kim, S. Yoon, B.

- J. Suh, S. W. Han, K. H. Kim, K. J. Kim, B. S. Kim, H. J. Song, H. J. Shin, J. H. Shim, and B. I. Min, *Appl. Phys. Lett.* **84**, 4233 (2004).
- ¹⁴B. T. Thole, P. Carra, F. Sette, and G. van der Laan, *Phys. Rev. Lett.* **68**, 1943 (1992).
- ¹⁵C. T. Chen, Y. U. Idzerda, H.-J. Lin, N. V. Smith, G. Meigs, E. Chaban, G. H. Ho, E. Pellegrin, and F. Sette, *Phys. Rev. Lett.* **75**, 152 (1995).
- ¹⁶S. Suga and S. Imada, *J. Electron Spectrosc. Relat. Phenom.* **92**, 1 (1992).
- ¹⁷L. Stichauer, A. Mirone, S. Turchini, T. Prospero, S. Zennaro, N. Zema, F. Lama, R. Pontin, Z. Šimša, Ph. Tailhades, and C. Bonnaingue, *J. Appl. Phys.* **90**, 2511 (2001).
- ¹⁸We performed x-ray diffraction (XRD) measurements several times over a long time, in which the same sharp XRD patterns were observed. This finding reveals that MnFe_2O_4 samples are stable at room temperature.
- ¹⁹The TEY mode has proving depth of 50–100 Å
- ²⁰Single crystal of $\text{Fe}_{0.9}\text{Mn}_{2.1}\text{O}_4$ was cleaved *in situ* and measured under the same conditions as for MnFe_2O_4 .
- ²¹C. Mitra, Z. Hu, P. Raychaudhuri, S. Wirth, S. I. Csiszar, H. H. Hsieh, H.-J. Lin, C. T. Chen, and L. H. Tjeng, *Phys. Rev. B* **67**, 092404 (2003).
- ²²P. Ghigna, A. Campana, A. Lascialfari, A. Caneschi, D. Gatteschi, A. Tagliaferri, and F. Borgatti, *Phys. Rev. B* **64**, 132413 (2001).
- ²³J.-Y. Kim, T. Y. Koo, and J.-H. Park, *Phys. Rev. Lett.* **96**, 047205 (2006).
- ²⁴T. J. Regan, H. Ohldag, C. Stamm, F. Nolting, J. Lüning, J. Stöhr, and R. L. White, *Phys. Rev. B* **64**, 214422 (2001).
- ²⁵Hangil Lee *et al.* (unpublished).
- ²⁶J. P. Crocombette, M. Pollak, F. Jollet, N. Thromat, and M. Gautier-Soyer, *Phys. Rev. B* **52**, 3143 (1995).
- ²⁷A. Mirone, M. Sacchi, and S. Gota, *Phys. Rev. B* **61**, 13540 (2000).
- ²⁸ $\gamma\text{-Fe}_2\text{O}_3$ is often represented by $(\text{Fe}^{3+})_A[\text{Fe}_{5/3}^{3+}]_B\text{O}_4$ spinel with vacancies at *B* sites. S. Brice-Profeta, M.-A. Arrio, E. Tronc, N. Menguy, I. Letard, C. Cartier dit Moulin, M. Nogués, C. Chaneac, J.-P. Jolivet, and Ph. Sainctavit, *J. Magn. Magn. Mater.* **288**, 354 (2005).
- ²⁹P. Ferriani, G. Ghiringhelli, G. Ferrari, C. M. Bertoni, A. Tagliaferri, L. Braicovich, and N. B. Brookes, *Nucl. Instrum. Methods Phys. Res. B* **200**, 220 (2003).
- ³⁰A. P. Holm, S. M. Kauzlarich, S. A. Morton, G. dan Waddil, W. E. Pickett, and J. G. Tobin, *J. Am. Chem. Soc.* **124**, 9894 (2002).
- ³¹M. Penicaud, B. Siberchicot, C. B. Sommers, and J. Kübler, *J. Magn. Magn. Mater.* **103**, 212 (1992).
- ³²D. J. Singh, M. Gupta, and R. Gupta, *Phys. Rev. B* **65**, 064432 (2002).
- ³³V. N. Antonov, B. N. Harmon, and A. N. Yaresko, *Phys. Rev. B* **67**, 024417 (2003).
- ³⁴Z. Szotek, W. M. Temmerman, D. Kodderitzsch, A. Svane, L. Petit, and H. Winter, *Phys. Rev. B* **74**, 174431 (2006).