

Determination of the micromagnetic parameters in (Ga,Mn)As using domain theory

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The magnetic domain structure and magnetic properties of a ferromagnetic (Ga,Mn)As epilayer with perpendicular magnetic easy axis are investigated. We show that despite strong hysteresis domain theory at thermodynamical equilibrium can be used to determine the micromagnetic parameters. Combining magneto-optical Kerr microscopy, magnetometry, and ferromagnetic resonance measurements, we obtain the characteristic parameter for magnetic domains λ_c , the domain wall width and specific energy, and the spin stiffness constant as a function of temperature. The nucleation barrier for magnetization reversal and the Walker breakdown velocity for field-driven domain wall propagation are also estimated.

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In thin layers based on ferromagnetic (FM) metallic or semiconducting materials domain wall (DW) propagation is the subject of extensive research with the ultimate aim of information storage and/or transport by domain walls.¹⁻⁸ In particular, the FM semiconductor (Ga,Mn)As has received a lot of attention recently since the critical current for DW propagation was shown to be two orders of magnitude smaller than in metallic systems.⁵ Both current-driven and field-driven DW propagation have been recently investigated in (Ga,Mn)As layers and stripes.⁶⁻⁸ In this context, it is important to determine experimentally the micromagnetic parameters such as the DW width and spin stiffness constant since they play a crucial role in DW dynamics. In the hydrodynamic regime for field-driven DW propagation, the DW mobility and the Walker breakdown velocity are proportional to the DW width,⁹ which depends on the spin stiffness constant and the magnetic anisotropy constants.¹⁰ Spin stiffness also governs the exchange length, which characterizes the width of Bloch lines and sets an upper limit for the size of superparamagnetic particles. Moreover, the determination of the spin stiffness constant also gives access to the J_{pd} exchange parameter characterizing the p - d exchange interaction between localized Mn spins and itinerant carriers (holes).¹¹⁻¹³ The experimental determination of the spin stiffness is not straightforward. It was extracted from spin wave resonances despite the discrepancy between theoretical and experimental dispersion curves.^{13,14} The obtained value was larger than the predicted ones. In this paper, we determine the spin stiffness constant and the other micromagnetic parameters using combined experimental techniques and analyzing the magnetic domain structure and the hysteresis cycles in the framework of domain theory.

Owing to the complex structure of the valence band in zinc-blende (Ga,Mn)As, the direction of the magnetic easy axis can be set parallel or perpendicular to the plane by tuning the layer strain and carrier concentration.^{11,15-18} In the case of strong perpendicular anisotropy, the ground state of the FM layer at thermodynamical equilibrium corresponds to the self-organized periodic stripe or bubble pattern of the up- and down-magnetized domains.¹⁰ Domain theory derived

from the micromagnetic equations for FM films provides a quantitative description of the equilibrium domain patterns. Long-range self-organization results from the competition between the DW energy and the magnetostatic energy. The domain pattern is controlled by a single parameter $\lambda_c = \ell_c / d = \sigma / 2K_d d$, where ℓ_c is a characteristic length equal to the ratio of the specific wall energy σ and the magnetostatic energy with $\sigma = 4\sqrt{AK_u}$ and $K_d = \mu_0 M_s^2 / 2$. A is the spin stiffness constant, K_u the uniaxial anisotropy constant, M_s the saturation magnetization, and d the sample thickness. Experimentally, the parameter λ_c can be extracted from the zero-field period of domain array. This method was used by Dietl *et al.* to determine λ_c for perpendicular-axis (Ga,Mn)As layers showing magnetic domains.¹⁹ However, the question arises whether the observed domain structures correspond to the equilibrium state. Actually, for perpendicular-axis (Ga,Mn)As the hysteresis cycle is square,^{17,20} thereby indicating that metastable states and/or DW pinning play an important role in the field dependence of the magnetization. Defect-assisted nucleation processes and DW pinning were recently investigated using magneto-optical imaging.^{17,20,21} In particular it was reported that the equilibrium demagnetized stripe or bubble state at zero applied field cannot be reached using ac demagnetization.

In this paper we show that domain theory can be applied to determine upper and lower boundaries for the parameter λ_c even in the presence of metastability and DW pinning. The upper boundary is obtained by comparing the domain width close to magnetization saturation with the predicted equilibrium width. The lower boundary is obtained from the comparison of major and minor hysteresis loops with the equilibrium magnetization curve. Using the saturation magnetization and the anisotropy constants obtained by superconducting quantum interference device (SQUID) magnetometry and ferromagnetic resonance (FMR) experiments, respectively, we obtain the temperature dependence of the lower and upper boundaries for the micromagnetic parameters: the specific DW energy, the DW width, and the spin stiffness constant, which are determined within a narrow range.

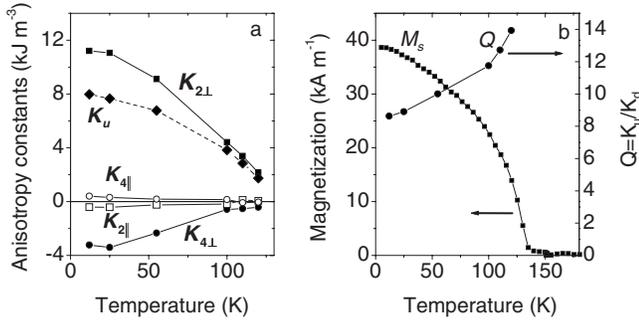


FIG. 1. (a) Temperature dependence of the anisotropy constants. (b) Temperature dependence of the saturation magnetization and of the Q parameter.

The sample was prepared by molecular beam epitaxy. It consists of a 50 nm thick $\text{Ga}_{0.93}\text{Mn}_{0.07}\text{As}$ layer grown on a $\text{Ga}_{0.902}\text{In}_{0.098}\text{As}$ relaxed buffer layer deposited on a semi-insulating (001) GaAs substrate. After postgrowth annealing the Curie temperature T_C was 130 K. The magnetic domain structure was investigated using magneto-optical Kerr (MOKE) microscopy. The hysteresis cycle was obtained from the average intensity of MOKE images as a function of the applied field. Experimental details can be found elsewhere.¹⁷

The anisotropy constants were obtained from FMR measurements. The FMR spectra were measured with X-band and Q-band spectrometers in the first derivative absorption mode with a modulation frequency of 100 kHz. To determine the anisotropy constants the spectra were measured for both the in-plane and out-of-plane orientation of the static magnetic field. From the resonance field positions we can deduce that the easy axis of the sample is along the [001] direction for all temperatures $T < T_C$. For the in-plane angular variation of the magnetic field an almost isotropic behavior is observed. Using the Smit-Beljers equation and the minimization of the free energy for different alignments of the magnetization,^{22,23} the magnetic anisotropy constants $K_{2\perp}$, $K_{4\perp}$, $K_{2\parallel}$, and $K_{4\parallel}$ were obtained. They are plotted in Fig. 1(a) as a function of temperature. The small values of the in-plane anisotropy constants $K_{2\parallel}$ and $K_{4\parallel}$ explain the almost isotropic in-plane behavior. In the following we will neglect them and consider the first term of the development of the anisotropy energy $K_u \sin^2 \theta$ with θ the angle between the [001] direction and the magnetization. We obtain K_u from the FMR results as $K_u = K_{2\perp} + K_{4\perp}$. The temperature dependence of K_u is plotted in Fig. 1(a). The uniaxial anisotropy is characterized by the parameter $Q = K_u/K_d$. From FMR and SQUID results we obtain a Q parameter ranging from 8.6 to 14 [Fig. 1(b)].

Since $Q \gg 1$ domain theory for FM films with strong uniaxial anisotropy can be applied.¹⁰ In this framework the free energy of a stripe array F_{str} is the sum of the DW energy, the Zeeman energy, and intra- and interdomain magnetic interaction terms. F_{str} is a function of two variables: the reduced average magnetization along the direction of the applied field $m = \langle M \rangle / M_s$ and the stripe period P . Minimization of F_{str} provides the dependence of m and P on λ_c and on the reduced applied field $h = H/M_s$. $m(h)$ curves are shown in Fig. 2(a) for several values of λ_c . The stripe width of the

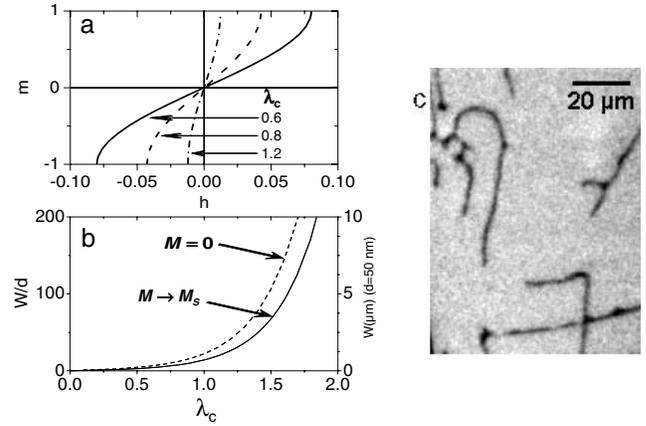


FIG. 2. (a) Dependence of the reduced average magnetization $m = \langle M \rangle / M_s$ on $h = H/M_s$ for a stripe array with $\lambda_c = 0.6, 0.8$, and 1.2 . (b) Stripe width as a function of the parameter λ_c for stripes of the minority phase near saturation (solid curve) and stripes at $m=0, H=0$ (dashed curve). (c) Lamellar domain structure in (Ga, Mn)As near the saturation field at $T=80$ K.

minority phase (M opposite to H) decreases as h increases from zero up to the saturation field $h_s = h_{(m \rightarrow 1)}$ [Fig. 2(b)]. It is worth noting that this width keeps a finite value W_s when $m \rightarrow 1$ due to intradomain magnetic interaction. The dependence of W_s on λ_c is obtained by solving the equation $2\pi\lambda_c = w_s^2 \ln(1 + w_s^{-2}) + \ln(1 + w_s^2)$, where $w_s = W_s/d$. It is shown in Fig. 2(b). Let us note the quasieponential increase of the stripe width with λ_c , which means that domains cannot exist for $\lambda_c \gg 1$.

In (Ga,Mn)As with perpendicular easy axis, DW pinning is strong enough with respect to interdomain magnetic interaction to hinder the long-range ordering of magnetic domains.¹⁷ However, lamellar domains with a narrow distribution of their width are observed close to magnetization saturation, as shown in Fig. 2(c). This indicates that DW pinning is not sufficiently strong to prevent the formation of lamellar domains resulting from intradomain magnetic interaction. An upper boundary for the parameter λ_c is obtained by comparing the prediction for the equilibrium domain width W_s with the average width W of these lamellar domains. At $T=80$ K, W is found equal to $1.7 \pm 0.3 \mu\text{m}$, which corresponds to $\lambda_c = 1.3$. Since DW pinning impedes the reversible evolution of the domain width with the field, the width of lamellar domains should be larger than the predicted equilibrium stripe width. Moreover, owing to the limited spatial resolution ($\approx 1 \mu\text{m}$), the measured width is probably larger than the actual one. Finally, since W_s is an increasing function of λ_c , the value of λ_c deduced experimentally corresponds to an upper boundary for λ_c . The same analysis was repeated at different temperatures. The results are reported in Fig. 3(a). A small variation of λ_c^{max} (1.24 to 1.7) is found as the temperature T varies from 12 to 120 K.

A lower boundary for λ_c is determined by comparing the theoretical predictions for the magnetization curve $m(h)$ with the hysteresis cycles obtained experimentally. Because of hysteresis, the $m(H)$ curve predicted for thermodynamical equilibrium and without pinning should lie *inside* the experimental hysteresis cycle. A major hysteresis cycle at

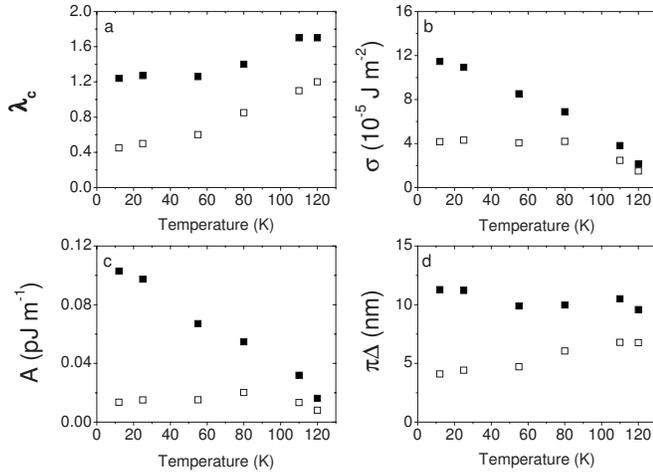


FIG. 3. Temperature dependence of the upper (black squares) and lower (empty squares) boundaries of (a) the parameter λ_c , (b) the specific wall energy σ , (c) the exchange spin stiffness A , and (d) the DW width $\pi\Delta$. They are obtained from the lamellar domain width (upper values) and hysteresis cycle (lower values).

$T=80$ K is shown in Fig. 4(b) (black squares). The applied field is swept up to 60 mT, i.e., beyond the saturation field (≈ 12 mT), which is much larger than the coercive field $\mu_0 H_c = 1.6$ mT. When the field is swept down, the onset of magnetization reversal and therefore the width of the cycle is determined by defect-assisted nucleation (see below). In order to obtain a narrower cycle, we recorded a minor hysteresis loop (empty squares). It is obtained by sweeping the field up to $\mu_0 H_m = 6$ mT, i.e., below the saturation field, in order to leave a few reversed domains in the sample. Hence, upon sweeping the field down, the nucleation stage is avoided. The shape of the hysteresis loop is determined by the dependence of the DW velocity on the applied field. Care is taken to use a low field ramp rate in order to allow for DW propagation at low velocity in the vicinity of the coercive field. The shape and width of the minor hysteresis loop does not depend on H_m provided it is larger than the coercive field. The theoretical $m(h)$ curves for stripes were calculated for a set of λ_c values [Fig. 2(a)]. The saturation field increases for decreasing λ_c . Therefore, by selecting the $m(h)$ curve which is tangent to the hysteresis cycle close to saturation, one can obtain a lower boundary for λ_c . This analysis can be further refined by considering that the saturation field calculated for stripes is lower than the theoretical collapse field, i.e., the field at which domains of the minority phase disappear. As $m \rightarrow 1$, lamellar domains keep a finite width but their length decreases until they reach the bubble circular shape and finally collapse.²⁴ The bubble collapse field h_{coll} can be calculated from the free energy of an isolated bubble with reverse magnetization^{10,24}

$$F_{sb} = \pi K_d d^3 [2\lambda_c p + (h-1)p^2 + p^2 f(p)], \quad (1)$$

where $p=2R/d$ is the reduced diameter of the bubble and $f(p) = 1 + 4p \{1 - [(2k^2 - 1)E(k) + (1 - k^2)K(k)]k^{-3}\} / 3\pi$ with $k^2 = p^2 / (1 + p^2)$. E and K are complete elliptic integrals of the first and the second kind, respectively. F_{sb} varies non-

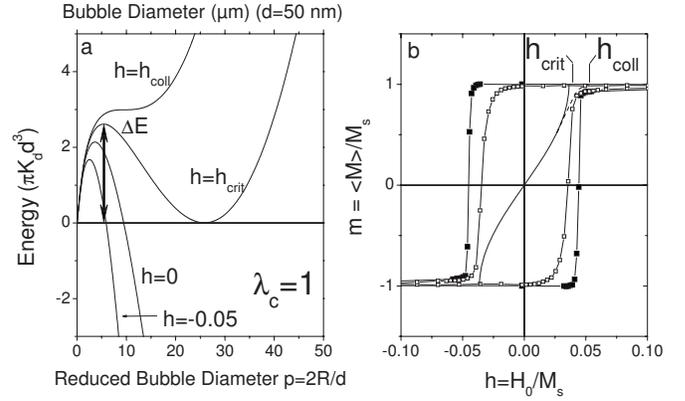


FIG. 4. (a) Energy of an isolated bubble with reverse magnetization for several magnetic fields ($\lambda_c = 1$). (b) Major (black squares) and minor (empty squares) hysteresis cycles obtained from Kerr microscopy images at $T=80$ K. The full curve is the $m(h)$ curve for a stripe array with $\lambda_c = 0.85$. The bubble critical and collapse fields for $\lambda_c = 0.85$ are indicated as h_{crit} and h_{coll} , respectively. The dashed curve links the $m(h)$ curve for stripes to the bubble collapse point (h_{coll} , $m=1$). It shows the expected magnetization path resulting from the existence of the collapse barrier.

monotonously with p as shown in Fig. 4(a). Above the critical field h_{crit} the system remains in a metastable state until the collapse field is reached. h_{coll} is the field at which $\partial F_{sb} / \partial p = \partial^2 F_{sb} / \partial p^2 = 0$. It increases with decreasing λ_c .

Figure 4(b) shows the critical and collapse fields. The dashed curve shows the expected magnetization path from the stripe array curve to the bubble collapse field for the metastable domain pattern. The lower boundary for λ_c is taken as the value that makes this curve tangent to the minor hysteresis cycle close to saturation. At $T=80$ K, $\lambda_c^{min} = 0.85$ is obtained. This procedure was repeated at different temperatures.

Finally we obtain the micromagnetic parameters. They are displayed in Fig. 3. The specific wall energy σ , the DW width $\pi\Delta = \pi\sqrt{A/K_u}$, and the exchange spin stiffness constant A are derived from the λ_c values using the temperature dependence of M_s and K_u . σ is found in the range $1.5 - 11.5 \times 10^{-5} \text{ J m}^{-2}$, which is about two orders of magnitude smaller than for iron or cobalt films.^{25,26} This explains that propagating DWs easily skirt around pinning defects.^{17,20} The DW width is found in the range 4–11 nm. This is larger than the mean distance between Mn ions (0.9 nm for 7% Mn) and smaller than the domain width, which validates the use of domain theory. The ratio of the DW parameter to the sample thickness Δ/d is larger than 0.025. Together with the large Q value, this ensures that the specific DW energy is very close to $4\sqrt{AK_u}$, as assumed in our analysis.²⁷

From the micromagnetic parameters one can estimate the Walker breakdown velocity V_W for field-driven DW propagation in the flow regime.¹⁰ Using $V_W = \gamma\sqrt{2\mu_0 A Q}(\sqrt{1+1/Q} - 1) \approx \gamma\mu_0 M_s \Delta / 2$ with γ the gyromagnetic ratio, one finds $V_W \approx 5.5 \text{ m s}^{-1}$ for the lower Δ values and 15 to 9.6 m s^{-1} for the largest Δ values for T between 12 and 80 K. These last V_W values are consistent with the largest velocities measured in the depinning regime for the same sample.⁸ This

indicates that the values of λ_c , σ , A , and Δ are very likely the largest ones, i.e., those determined from the lamellar domain width.

From the determination of λ_c , one also gains some insight into the magnetization reversal process. We obtain the height of the intrinsic nucleation barrier. As seen in Fig. 4(a), below the critical field the energy of the system with a reverse bubble domain is lower than the energy of the saturated state. However the system can stay in the saturated state since there is an energy barrier ΔE for nucleation. Taking $0.85 < \lambda_c < 1.3$ at $T=80$ K yields $280 < \Delta E/KT < 1310$ at $h=h_{crit}$. The barrier height only weakly decreases with decreasing field. At the onset of magnetization reversal ($h \approx -0.05$) one still finds large values for ΔE , namely $194 < \Delta E/KT < 660$. This means that the system is highly metastable and that nucleation proceeds through local defects that strongly decrease the barrier height. This conclusion also holds at $T=120$ K, i.e., close to T_C .

The spin stiffness constant is smaller by one order of magnitude than the value determined by Goennenwein *et al.* from spin wave resonances in FMR spectra for (Ga,Mn)As with in-plane magnetization.¹⁴ Let us note that a linear gradient of the magnetic properties had to be assumed by Goen-

nenwein *et al.*¹⁴ in order to fit the results with theoretical spin wave dispersion curves. Theoretical predictions for A were made by König *et al.*¹³ for various carrier densities and for two values of the J_{pd} exchange constant, taking a Mn concentration of 4.5%. For our sample we obtain an effective Mn concentration $x_{eff}=3.8\%$ [we neglect the hole magnetization and assume $M_s(T \rightarrow 0) = x_{eff} N_0 5 \mu_B$, with N_0 the density of cation sites]. With this Mn concentration, our results for the spin stiffness are consistent with the extrapolated theoretical estimations of König *et al.*¹³ provided that a low value of the J_{pd} exchange constant is used ($J_{pd} \leq 50$ meV nm³).

To summarize, we have applied domain theory to the case of a metastable system in order to obtain the ratio of DW specific energy and magnetic energy. Combining the study of domain structure, magnetization, and anisotropy constants in a (Ga,Mn)As thin film with perpendicular magnetization, we have determined the micromagnetic parameters: the specific DW energy ($1.5-11.5 \times 10^{-5}$ J m⁻²), the DW width (4–11.3 nm), and the spin stiffness constant (0.01–0.1 pJ m⁻¹). From these results we estimate the nucleation barrier and the Walker velocity for DW propagation.

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