Determination of the infrared complex magnetoconductivity tensor in itinerant ferromagnets from Faraday and Kerr measurements

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We present measurement and analysis techniques that allow the complete complex magnetoconductivity tensor to be determined from midinfrared (11–1.6 μ m; 100–800 meV) measurements of the complex Faraday (θ_F) and Kerr (θ_K) angles. Since this approach involves measurement of the geometry (orientation axis and ellipticity of the polarization) of transmitted and reflected light, no absolute transmittance or reflectance measurements are required. Thick-film transmission and reflection equations are used to convert the complex θ_F and θ_K into the complex longitudinal conductivity σ_{xx} and the complex transverse (Hall) conductivity σ_{xy} . θ_F and θ_K are measured in a Ga_{1-x}Mn_xAs and SrRuO₃ films. The resulting σ_{xx} is compared to the values obtained from conventional transmittance and reflectance measurements, as well as the results from Kramers-Kronig analysis of reflectance measurements on similar films.

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I. INTRODUCTION

Conventional dc Hall effect measurements have been essential in revealing the unusual character of unconventional electronic materials including high-temperature superconducting cuprates (HTSCs),¹ diluted magnetic semiconductors $(DMSs)^2$ and ruthenate perovskite³⁻⁵ (RP) materials. In many of these materials, the Hall angle θ_H and transverse (Hall) conductivity σ_{xy} provide information critical to understanding their electronic properties. The frequency dependence of θ_H and σ_{xy} is very sensitive to the electronic structure, and, in many cases, exposes insights that are hidden from the longitudinal conductivity σ_{xx} that is measured by conventional spectroscopy The longitudinal and Hall resistivities, ρ_{xx} and ρ_{xy} , respectively, are related to the conductivities as follows: $\sigma_{xx} = 1/\rho_{xx}$ and $\sigma_{xy} = \rho_{xy}/\rho_{xx}^2$. The midinfrared (MIR: $11-1.6 \mu m$; 100-800 meV) energy range is particularly interesting in many of these materials. For example, the band structure of RP (Ref. 6) and III-V(Mn) DMS (Refs. 7 and 8) leads to predictions of strong spectral features in σ_{xy} in the MIR. In the electron-doped cuprate, $Pr_{2-x}Ce_xCuO_4$, evidence of a spin-density wave gap has been observed in the MIR behavior of σ_{xy} .

Reflection and transmission magnetopolarimetry measurements allow one to determine the complex Faraday θ_F and Kerr θ_K angles, respectively. $\theta_F(\theta_K)$ describes the change in polarization of transmitted (reflected) light produced by a sample in a magnetic field. Although θ_F and θ_K are useful, they depend on the optical geometry of the sample, such as the thickness of the film and index of refraction of the substrate. Furthermore, calculations of θ_F and θ_K depend on which assumptions are made for the optical formulas, e.g., thin film, thick film, and bulk. Theoretical models typically

calculate response functions such as σ_{xy} and σ_{xx} . Since these conductivities are related to the electronic behavior in a more fundamental way, it is useful to convert θ_F and θ_K into these more elementary quantities.

In principle, one can determine σ_{xx} using conventional, polarization-insensitive spectroscopy techniques, such as Kramers-Kronig analysis of reflectance measurements or analysis of transmittance and reflectance measurements. These measurements, however, do not access σ_{rv} , which is critical to understanding many unconventional materials. Although conventional spectroscopic techniques may be experimentally simpler than the Faraday/Kerr measurements presented in this paper, polarization-insensitive approaches to measuring σ_{xx} rely on absolute transmittance and reflectance measurements, which can limit the accuracy of these techniques. On the other hand, since θ_F and θ_K measurements are absolute measurements that do not require normalization to a reference sample, the accuracy of this technique can be very high. In this paper, we obtain the entire complex magnetoconductivity tensor by measuring the geometry (orientation axis and ellipticity of the polarization) of reflected and transmitted light; no absolute intensity measurements are required. A key advantage of determining both σ_{xx} and σ_{xy} from the same set of θ_F and θ_K measurements is that the behavior of σ_{xx} , which may already be fairly well known, can provide a consistency test for σ_{xy} , which typically is not well known in the MIR. For experimental systems such as ours that are designed for magnetopolarimetry measurements, obtaining σ_{xx} through θ_F and θ_K measurements is experimentally more straightforward and more accurate than using conventional polarization-insensitive spectroscopic techniques.

Of course, this approach has limitations. First, one must sensitively measure polarization changes produced in trans-

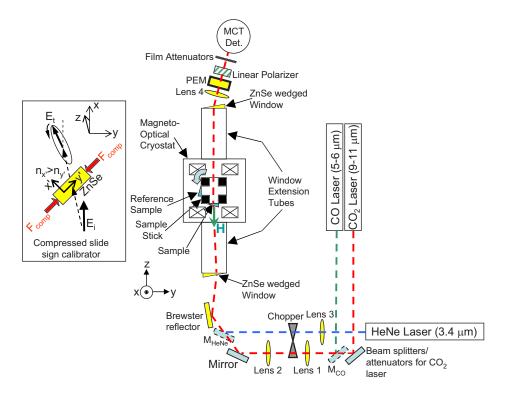


FIG. 1. (Color online) Overall schematic of the optical path. The sample stick is rotated to select either the sample or reference. The polarization change induced by compressively straining a ZnSe slide is shown in the boxed inset on the left.

mitted and reflected light by a magnetic field. Second, the samples must produce measurable θ_F and θ_K signals, which is not always the case. For example, in ordinary metals, such as Au, as well as unconventional metals, such as high-temperature superconductors, θ_K is very small and would be difficult to measure accurately. Fortunately, magnetic metals, such as SrRuO₃ and Ga_{1-x}Mn_xAs, produce θ_K signals that can be measured readily.

In this paper, we present a MIR polarimetry technique that determines the entire complex conductivity tensor using θ_F and θ_K measurements. We shall first introduce a sensitive magneto-optical measurement system, next we develop formulas for θ_F and θ_K in terms of σ_{xx} and σ_{xy} , and finally we shall test these techniques on SrRuO₃ and Ga_{1-x}Mn_xAs films.

II. EXPERIMENTAL SYSTEM

When illuminating a material in a magnetic field H using linearly polarized light, the transmitted and reflected beams can be modified in two ways: the plane of polarization can be rotated with respect to the incident linear polarization and the beam may acquire ellipticity. The polarization change of the transmitted light is characterized by the complex Faraday angle θ_F , the optical analog of the Hall angle $\theta_H = \sigma_{xy} / \sigma_{xx}$, where σ_{xx} is the longitudinal conductivity and σ_{xy} is the transverse (Hall) conductivity. θ_F relates the magnitudes and phases of the transmitted electric fields that are perpendicular (t_{xy}) and parallel (t_{xx}) to the incident linear electric polarization, which is along the x direction in this case: $\tan \theta_F$ $\equiv t_{xy}/t_{xx}$, where t_{xx} and t_{xy} are the diagonal and off-diagonal components of the complex magnetotransmission tensor. The complex polar Kerr angle θ_K describes the change in the polarization of reflected light for near-normal incidence, with tan $\theta_K \equiv r_{xy}/r_{xx}$, where r_{xx} and r_{xy} are the diagonal and offdiagonal components of the complex magnetoreflection tensor. For small changes in the incident polarization, $\text{Re}(\theta_F)$ [$\text{Re}(\theta_K)$] is related to the rotation and $\text{Im}(\theta_F)$ [$\text{Im}(\theta_K)$] is related to the ellipticity of the transmitted (reflected) beam's polarization.

The experimental technique used in this paper is based on Ref. 10. There are four experimental aspects that are added here: (1) precision translational mount for a magneto-optical cryostat; (2) extended wavelength range using additional lasers; (3) reflection measurements to determine θ_K ; and (4) a more powerful calibration technique. After reviewing the basic experimental technique, this paper will focus on these four areas. The experimental setup is shown in Fig. 1. The Faraday and Kerr angles are measured using discrete lines from CO₂ (115–133 meV), CO (215–232 meV), and HeNe (366 meV) lasers. Measurements have also been made using laser diodes operating at 500 and 775 meV located near lens 3 in Fig. 1. One can probe the samples down to 6 K and magnetic fields up to 7 T in the magneto-optical cryostat. The different laser beams are aligned in one optical path before entering the cryostat (Fig. 1). The original optical path is designed for the CO_2 laser. The CO laser is sent to probe the sample by inserting mirror $M_{\rm CO}$, which is mounted on a kinematic base before lens 1. The HeNe laser is sent to probe the sample by inserting mirror M_{HeNe} , which is mounted on a kinematic base near the Brewster reflector. An optical chopper modulates the laser intensity with a frequency ω_0 . To prevent étalon artifacts from multiple reflections within optical components, the cryostat windows and the sample substrate are wedged 1°-2°, and the photoelastic modulator (PEM), which is used to analyze the polarization of the transmitted/reflected radiation, is tilted forward 25°. As is discussed in more detail in Sec. IV B, a compressively strained ZnSe slide produces a well-characterized rotation and ellipticity (see boxed inset in Fig. 1) in the polarization,

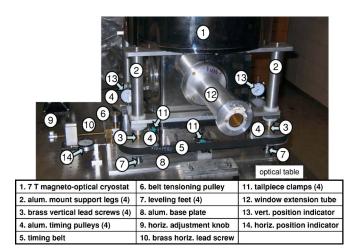


FIG. 2. (Color online) The mount that allows the 700 lb magneto-optical cryostat (labeled 1 in figure) to be translated vertically and horizontally with $\sim 25 \ \mu m$ precision. The knob (9) attached to the lead screw (10) for horizontal translation is indicated in the left part of the photograph. Vertical adjustment is made by pulling on the timing belt (5), which rotates the four timing pulleys (4) simultaneously, causing the large brass lead screws (3) to move the support legs (2) up or down.

which can be used to determine the absolute signs of θ_F and θ_K .

The 7 T magneto-optical cryostat has only two windows, with the sample in vacuum. These room temperature ZnSe windows are placed on 30 cm extension tubes (as can be seen in Fig. 2) to minimize their contribution to the Faraday rotation due to the stray magnetic field. The absence of cold windows is important in several respects. First, other magneto-optical cryostat can have up to four cold windows (a pair of liquid helium temperature and a pair of liquid nitrogen temperature windows) that are located close to the sample and, therefore, experience similar magnetic fields as the sample. Since the cold windows typically are several millimeters thick, they can produce a large magnetopolarization signal that can readily overwhelm the signal produced by the sample, typically several hundred nanometer-thick films. Furthermore, the absence of cold windows increases transmission and reduces artifacts due to multiple reflections.

A PEM makes it possible to measure sensitively both real and imaginary parts of θ_F and θ_K .¹⁰ The optical axis of the PEM is oriented vertically along the *x* axis and modulates the phase of the transmitted light that is polarized along the y direction at a frequency $\omega_{\text{PEM}}/2\pi \approx 50$ kHz. A linear polarizer P_2 is oriented 45° from vertical and mixes the x and y polarization components of the light that has passed through the PEM. A liquid-nitrogen-cooled mercury-cadmiumtelluride (MCT) detector measures the intensity of the modulated beam. If the sample does not cause any changes in the incident polarization, the light entering the PEM is linearly polarized along the x direction and there will be no signals at the detector related to the PEM. Three lock-in amplifiers demodulate the detector signal. One lock-in amplifier is referenced to chopper frequency ω_0 to provide a measurement of the average laser intensity I_0 at the detector. The other two lock-in amplifiers are referenced to harmonics of ω_{PEM} to detect the polarization of the beam. The even harmonics of ω_{PEM} are related to a rotation of the polarization vector $[\text{Re}(\theta_F) \text{ or } \text{Re}(\theta_K)]$ and the odd harmonics are related to the ellipticity $[\text{Im}(\theta_F) \text{ or } \text{Im}(\theta_K)]$.¹⁰ Typically, one measures the rotation using the second-harmonic signal $I_{2\omega_{\text{PEM}}}$ and the ellipticity using the third-harmonic signal $I_{3\omega_{\text{PEM}}}$.

The mount holding the 7 T magneto-optical cryostat can be translated with high accuracy (~25 μ m) both horizontally and vertically. As can be seen in Fig. 2, the cryostat rests on a large 5/16 in. thick aluminum base plate (17 \times 24 in.²) that has two long turcite (grade A, blue) strips attached to the bottom of the base plate directly below the vertical supports. The strips are 0.030 in. thick, 23 in. long, and 2 in. wide, and are epoxied into shallow pockets in the bottom of the plate. Turcite is optimized for high load, low friction applications and does not cold flow. The base plate and cryostat are moved horizontally with respect to the optical table by turning a 3/8 in. $\times 24$ brass lead screw that is attached to the base plate and the optical table. The height of the cryostat is adjusted by simultaneously turning four 3/4 in. $\times 16$ brass lead screws that are threaded into turcite bushings in each of the four cryostat mount support legs. To ensure that the cryostat is raised and lowered uniformly, without tipping, the brass lead screws are coupled to each other by a 1 in. wide, 80 in. long timing belt (T10, 10 mm pitch) that connects 5 in. diameter timing pulleys that are attached to each lead screw. The design and construction of this mount are challenging, since it must be made of nonmagnetic materials and since the cryostat weighs approximately 700 lb. Separate leveling feet at the bottom of each brass lead screw allow the cryostat to be leveled. The magneto-optical signals are critically affected when the probe laser beam is close to the edge of the aperture in the copper plate on which the sample is held, so centering the sample, which can be as small as 3 mm, on the laser beam is very important. Since the lasers are invisible, this can be difficult. The procedure for centering the sample on the laser beam is greatly improved and simplified using the translating cryostat mount. The transmitted or reflected signal at the detector is simply maximized by translating the sample (along with the cryostat) vertically and horizontally. Precision indicators (shown in Fig. 2) are placed to measure the absolute position of the magneto-optical cryostat. By monitoring the detector signal as one translates the cryostat and sample aperture across the beam, one can determine the laser beam profile and accurately center the sample on the beam. Since they contain magnetic components, the indicators are removed before the running measurements with the superconducting magnet energized. As with the magneto-optical cryostat described in Ref. 10, unwanted motion of the sample is minimized by clamping the lower end of the sample tube to a horizontal Delrin rod, which is attached to an aluminum plate in the side window opening of the tailpiece. Furthermore, the tailpiece of the cryostat itself is clamped to the base plate by four bolts that are capped with turcite pads, as shown in Fig. 2. This greatly increases the rigidity of the cryostat mount and minimizes sample movement when the magnetic field is energized.

Unlike Ref. 10, which describes measurements in the 112–136 meV (1100–900 cm⁻¹; 11–9 μ m) range, in this

paper the measurements have been extended up to 775 meV (6250 cm⁻¹, 1.6 μ m). Since the PEM, the lenses, and the cryostat windows are ZnSe, the optical system is compatible with sources over the 0.5–20 μ m range. The main challenge in using sources with shorter wavelength λ is that the background Faraday rotation due to stray magnetic fields at the windows and sample substrate increases as λ^{-2} .¹¹ This makes careful measurement and subtraction of the background at shorter wavelengths especially critical. Since this background is reduced when the energy gap of the optical material is increased relative to the photon energy, higher bandgap optical materials, such as BaF₂ and CaF₂, will be used for the cryostat windows in future measurements. A further advantage of BaF_2 and CaF_2 is their smaller index (1.5 compared to 2.4 for ZnSe) that reduces Fresnel losses in transmitting light through them. Initially, the 3.4 μ m HeNe laser was placed approximately 1.5 m away from the magnet. However, the stray magnetic field at that distance induces both rotation and ellipticity in the output of the HeNe laser that is comparable to the signals produced by many of our samples (10⁻³ rad at 1 T).¹² This background is dramatically reduced by moving the HeNe laser farther away (2.5 m) from the magneto-optical cryostat.

In this paper, both transmission and reflection measurements are made to determine both θ_F and θ_K , whereas only transmission measurements (θ_F) were reported in Ref. 10. Although θ_F and θ_K provide similar information, there are several advantages to measuring both. First, in the metallic films that are reported here, the transmittance can be very low, as is discussed in the Sec. III of this paper. The reflectance amplitudes, on the other hand, are typically on the order of 50% or higher. Although the substrates used in our measurements are relatively transparent in the MIR, θ_K measurements are not limited to transparent substrates and can even be applied to metallic bulk materials where transmission measurements (θ_F) would be impossible. The fact that θ_F and θ_K are related is also an advantage in that the self-consistency of the results may be confirmed.

In Ref. 10, we describe several techniques to calibrate the polarimetry system. We have developed a more powerful technique which allows the simultaneous calibration of the PEM retardance R_{PEM} and the angle α_2 of the final linear polarizer P_2 . One of the calibration techniques in Ref. 10 is to rotate the PEM by a known small angle ϕ and use the change in the normalized detector signal S_2 at frequency $2\omega_{\text{PEM}}$ to calibrate θ_F . The PEM and polarizer P_2 are rotated as a single unit, so that the angle α_2 between the PEM and P_2 is kept constant. In this case, we measure the normalized signals S_2 and S_4 at $2\omega_{\text{PEM}}$ and $4\omega_{\text{PEM}}$, respectively. For $\phi \ll 1$, these signals depend on ϕ as follows:

$$S_2 = \frac{I_{2\omega_{\text{PEM}}}}{I_0} = \frac{4J_2(R_{\text{PEM}})\phi\tan(\alpha_2)}{1 + J_0(R_{\text{PEM}})\phi\tan(\alpha_2)},$$
(1)

$$S_4 = \frac{I_{4\omega_{\rm PEM}}}{I_0} = \frac{4J_4(R_{\rm PEM})\phi\tan(\alpha_2)}{1+J_0(R_{\rm PEM})\phi\tan(\alpha_2)},$$
 (2)

$$\frac{S_2}{S_4} = \frac{J_2(R_{\rm PEM})}{J_4(R_{\rm PEM})},$$
(3)

where J_n are the *n*th order Bessel functions, I_0 is the signal at the chopper frequency, $I_{2\omega_{\text{PEM}}}$ is the signal at $2\omega_{\text{PEM}}$, and $I_{4\omega_{\text{PEM}}}$ is the signal at $4\omega_{\text{PEM}}$. Note that the ratio of S_2 and S_4 in Eq. (3) only depends on R_{PEM} , as all the other factors cancel. One can use the measurement of S_2/S_4 to determine R_{PEM} . Since neither ϕ nor α_2 enter into Eq. (3), this calibration is not affected by the amount the PEM is rotated, as long $\phi \ll 1$, nor by the precise orientation of polarizer P_2 . Once R_{PEM} is determined, it can be entered into either Eq. (2) or (3) to determine α_2 , which is nominally 45°. Although α_2 can be calibrated, when doing measurements at a various wavelengths, errors can be reduced by keeping α_2 constant. The errors in adjusting P_2 to make $\alpha_2 = 45^\circ$ at each wavelength can be significant. If α_2 is kept constant, there will be no variation from wavelength to wavelength due to different settings of P_2 . A further check when performing this calibration for different wavelengths and different lasers is that α_2 from the fits should be the same, since P_2 was not moved with respect to the PEM. In our calibration measurements, α_2 determined from this calibration typically remains constant to within 0.2° over the entire measurement range. As in Ref. 10, the roll-off attenuation of the detector and its associated electronics are included in the final calibration. We have found that the roll-off also depends on the gain setting of the detector preamplifier.

III. SAMPLES

The SrRuO₃ sample consists of a 282 nm thick SrRuO₃ film on a LaSrGaO₄ substrate and was grown by pulsed-laser deposition at Oak Ridge National Laboratory, as described in Ref. 13. The LaSrGaO₄ substrate is transparent in the MIR, which allowed both transmission and reflection measurements to be made. The thickness of the SrRuO₃ film and the fact that the substrate begins to absorb strongly below 117 meV resulted in transmittances that could be below 0.01%. Fortunately, high power coherent sources such as the CO₂ and CO lasers coupled with a liquid-nitrogen-cooled detector provide the measurement system with the necessary dynamic range to measure small changes in the polarization even for such small transmittance values. The back of the LaSrGaO₄ substrate was polished to a 2° wedge after the film was grown.

The Ga_{1-x}Mn_xAs sample having a Curie temperature of 95 K was synthesized using Mn⁺ ion implantation followed by pulsed-laser melting.^{14,15} A semi-insulating GaAs (001) wafer was implanted with 80 keV Mn⁺ to a dose of 1.8 $\times 10^{16}$ cm⁻² and irradiated in air with a single 0.4 J/cm² pulse from a KrF excimer laser. The total Mn concentration depth profile measured by secondary-ion-mass spectrometry was nearly Gaussian with a peak value of approximately 8% and a width of 50 nm. The back of the GaAs substrate was polished to a 1° wedge after the film was grown.

IV. ANALYSIS

Although many experimental improvements have been made since our report in Ref. 10, the primary contribution of

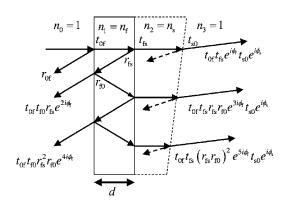


FIG. 3. Optical path of light passing through a thick film (left) with thickness d on a wedged substrate. The multiply reflected beams in the film are drawn at non-normal angles for clarity.

this paper involves an analysis of θ_F and θ_K measurements. In this section, we develop thick-film formulas for θ_F and θ_K in terms of σ_{xx} and σ_{xy} . We also discuss the sign conventions that are used for θ_F and θ_K .

A. Thick-film equations for θ_F and θ_K

Since θ_F and θ_K are defined in terms of the complex transmission and reflection amplitudes t_{xy} , t_{xx} , r_{xy} , and r_{xx} , we begin by determining these amplitudes. As light passes through a thick-film sample on a wedged substrate, part of the beam is reflected and part is transmitted at each interface, as shown in Fig. 3. The beam that reflects off the first air-film interface combines with beams that have been multiply reflected within the film to produce the reflected light. Since the back of the substrate is wedged, beams that reflect from it do not combine with the beams reflecting off the film. The first-pass beam combines with beams that are multiply reflected within the film to produce the transmitted light entering the substrate. This transmitted light is also multiply reflected within the substrate, but since the substrate is wedged, each order of reflection exits the substrate at a different angle. Therefore, the main advantage of using a wedged substrate is that one can spatially separate the firstpass beam from beams that are multiply reflected within the substrate.

Consider a thick film with complex index of refraction n_f on a wedged substrate with complex index of refraction n_s . The Fresnel coefficients t_{ij} and r_{ij} describe the transmission and reflection amplitudes, respectively, at each interface separating a material with index n_i from a material with index n_j . Summing over all transmitted beams including only multiple reflections within the film, the complex transmission coefficient for the sample is given by

$$t(n_{f}) = \frac{t_{0f}t_{fs}e^{i\phi_{f}}}{1 - r_{fs}r_{f0}e^{2i\phi_{f}}t_{s0}e^{i\phi_{s}}}$$
$$= \frac{2}{(1 + n_{s})\cos(kd) - i\left(n_{f} + \frac{n_{s}}{n_{f}}\right)\sin(kd)}t_{s0}e^{i\phi_{s}}, \quad (4)$$

where $k = \omega n_f / c$ is the wave number of the light within the

film and *d* is the thickness of the film. The index 0 represents air. The phase shift of the beam that passes through the film is given by $\phi_f = kd$. For the wedged substrate, we are only interested in the first-pass beam, which experiences a phase shift of $\phi_s = (\omega n_s d_{sub})/c$, where d_{sub} is the thickness of wedged substrate.

Similarly adding all contributions to the reflected light, the total reflection coefficient becomes

$$r(n_f) = \frac{r_{0f} + r_{fs}e^{2i\phi_f}}{1 - r_{fs}r_{f0}e^{2i\phi_f}} = \frac{(n_s - 1)\cos(kd) - i\left(n_f - \frac{n_s}{n_f}\right)\sin(kd)}{(1 + n_s)\cos(kd) - i\left(n_f + \frac{n_s}{n_f}\right)\sin(kd)}.$$
(5)

Note that the $r(n_f)$ on a wedged substrate is the same as for a film on an infinitely thick substrate, since the reflection from the back of the substrate never reaches the detector.

The following calculation connects θ_F to the optical conductivities σ_{xx} and σ_{xy} . We shall assign the *z* axis as the direction of light propagation and also the direction of the magnetic field *H*. Assuming that the sample is cylindrically symmetric along the *z* axis, the conductivity tensor would be indistinguishable between *x* and *y* axes, $\sigma_{xx} = \sigma_{yy}$ and σ_{xy} $= -\sigma_{yx}$. Therefore, the optical conductivity tensor is purely diagonal in the circular polarization basis. One can write the complex dielectric function ε_{\pm} for a circularly polarized basis with either positive or negative helicity. ε_{\pm} is related to the conductivity σ_{\pm} , represented in the circular basis, by

$$\varepsilon_{\pm} = \varepsilon_b - \frac{4\pi}{i\omega}\sigma_{\pm} = \varepsilon_b - \frac{4\pi}{i\omega}(\sigma_{xx} \pm i\sigma_{xy}) = \varepsilon_0 \mp \frac{4\pi}{\omega}\sigma_{xy},$$
(6)

where ε_b is the response of bound charges and $\varepsilon_0 \equiv \varepsilon_b - 4\pi\sigma_{xx}/i\omega$ is the longitudinal component of dielectric function. The complex index of refraction $n_{f,\pm}$ for the film using a circular polarization basis can be expressed in terms of ε_{\pm} and σ_{xy} .

$$n_{f,\pm} = \sqrt{\varepsilon_{\pm}} = \sqrt{\varepsilon_0} \left(1 \mp \frac{\sigma_{xy}}{\sigma_b + i\sigma_{xx}} \right)^{1/2}, \tag{7}$$

where σ_b is defined as $\frac{\omega}{4\pi}\varepsilon_b$. Since $\sigma_{xx} \gg \sigma_{xy}$ in most cases in the MIR, $n_{f,\pm}$ can be simplified to

$$n_{f,\pm} \simeq \sqrt{\varepsilon_0} \mp \frac{1}{2} \sqrt{\varepsilon_0} \frac{\sigma_{xy}}{\sigma_b + i\sigma_{xx}} = n_{f,0} \mp \delta n_f, \qquad (8)$$

where $n_{f,0} = \sqrt{\varepsilon_0}$ is longitudinal component of complex index of refraction and $\delta n_f = 2\pi\sigma_{xy}/\omega n_{f,0}$ is the transverse component. The diagonal (longitudinal) transmission coefficient t_{xx} and the off-diagonal (transverse) transmission coefficient t_{xy} in a linear polarization basis are related to the diagonal transmission coefficients t_+ and t_- in the circular polarization basis as follows. Since $t_{\pm}(n_{f,\pm}) \approx t_0(n_{f,0}) \pm (\partial t_0/\partial n_{f,0}) \delta n_f$,

$$t_{xx} = \frac{t_{+} + t_{-}}{2} \approx t_0(n_{f,0}), \qquad (9)$$

$$t_{xy} = \frac{t_+ - t_-}{2i} \approx \frac{1}{i} \left(\frac{\partial t_0}{\partial n_{f,0}} \right) \delta n_f.$$
(10)

Equations (9) and (10) show that diagonal transmission measurements probe the sum of t_+ and t_- , whereas offdiagonal transmission measurements, e.g., Faraday measurements, probe the difference, making them more sensitive to small changes in t_+ and t_- induced by magnet fields or other symmetry-breaking mechanisms. Putting Eqs. (9) and (10) together, the Faraday angle θ_F can be represented in terms of σ_{xy} and σ_{xx} (found in $n_{f,0} = \sqrt{\varepsilon_b - 4\pi\sigma_{xx}/i\omega}$) and σ_{xy} as

$$\tan \theta_F = \frac{t_{xy}}{t_{xx}} = -i\delta n_f \frac{1}{t_0(n_{f,0})} \frac{\partial t_0}{\partial n_{f,0}} = \frac{-2\pi i}{\omega n_{f,0}} \left(\frac{1}{t_0(n_{f,0})} \frac{\partial t_0}{\partial n_{f,0}}\right) \sigma_{xy}.$$
(11)

One can use the complex transmission coefficient $t_0(n_{f,0})$ [Eq. (4)] to calculate the Faraday angle θ_F .

$$\tan \theta_F = \left(\frac{-2\pi i\sigma_{xy}}{\omega n_{f,0}}\right) \frac{\left[(1+n_s)\left(\frac{\omega d}{c}\right) + i\left(1-\frac{n_s}{n_{f,0}^2}\right)\right]\sin(kd) + i\left(n_{f,0} + \frac{n_s}{n_{f,0}}\right)\left(\frac{\omega d}{c}\right)\cos(kd)}{(1+n_s)\cos(kd) - i\left(n_{f,0} + \frac{n_s}{n_{f,0}}\right)\sin(kd)}.$$
(12)

Taking the approximation $kd = (2\pi d)/\lambda \ll 1$ $(d \rightarrow 0, \omega \rightarrow 0)$, Eq. (12) yields the simple thin-film formula

$$\tan \theta_F \approx \left(\frac{\sigma_{xy}}{\sigma_{xx}}\right) \left[1 + \frac{1}{Z_+ \sigma_{xx}}\right]^{-1}, \tag{13}$$

where $Z_{\pm} \equiv (Z_0 d) / (n_s \pm 1)$, Z_0 is the impedance of free space, and the unit of conductivity is Ω^{-1} cm⁻¹. Note that Eq. (13) is slightly different from Eq. (2.2) in Ref. 16 due to a typographical error on the right side of Eq. (2.2).

One can use the same approach to calculate θ_K in terms of σ_{xx} and σ_{xy} , using the reflection coefficients. The diagonal and the off-diagonal reflection amplitudes, r_{xx} and r_{xy} , can be expressed in a linear polarization basis in terms of the diagonal reflection coefficients r_+ and r_- in the circular polarization basis.

$$r_{xx} = \frac{r_+ + r_-}{2} \approx r_0(n_{f,0}), \qquad (14)$$

$$r_{xy} = \frac{r_+ - r_-}{2i} = \frac{1}{i} \left(\frac{\partial r_0}{\partial n_{f,0}}\right) \delta n_f.$$
(15)

Again, we have assumed that $r_{\pm}(n_{f,\pm}) \approx r_0(n_{f,0})$ $\pm (\partial r_0 / \partial n_{f,0}) \delta n_f$. Combining Eqs. (14) and (15) produces an expression for θ_K in terms of σ_{xx} and σ_{xy} ,

$$\tan \theta_K = \frac{r_{xy}}{r_{xx}} = -i \delta n_f \frac{1}{r_0(n_{f,0})} \frac{\partial r_0}{\partial n_{f,0}} = \frac{-2\pi i}{\omega n_{f,0}} \left(\frac{1}{r_0(n_{f,0})} \frac{\partial r_0}{\partial n_{f,0}} \right) \sigma_{xy}.$$
(16)

To calculate the Kerr angle θ_K , we can use the complex reflection coefficient r_0 [Eq. (5)].

$$\tan \theta_{K} = \left(\frac{-2\pi i\sigma_{xy}}{\omega n_{f,0}}\right) \frac{\left[-\left(n_{s}-1\right)\left(\frac{\omega d}{c}\right)-i\left(1+\frac{n_{s}}{n_{f,0}^{2}}\right)\right]\sin(kd)-i\left(n_{f,0}-\frac{n_{s}}{n_{f,0}}\right)\left(\frac{\omega d}{c}\right)\cos(kd)}{\left(n_{s}-1\right)\cos(kd)-i\left(n_{f,0}-\frac{n_{s}}{n_{f,0}}\right)\sin(kd)} + \left(\frac{-2\pi i\sigma_{xy}}{\omega n_{f,0}}\right)\frac{\left[\left(n_{s}+1\right)\left(\frac{\omega d}{c}\right)+i\left(1-\frac{n_{s}}{n_{f,0}^{2}}\right)\right]\sin(kd)+i\left(n_{f,0}+\frac{n_{s}}{n_{f,0}}\right)\left(\frac{\omega d}{c}\right)\cos(kd)}{\left(n_{s}+1\right)\cos(kd)-i\left(n_{f,0}+\frac{n_{s}}{n_{f,0}}\right)\sin(kd)}.$$

$$(17)$$

Applying the same approximation $(kd \leq 1)$ used for θ_F to Eq. (17) results in the thin-film formula for θ_K ,

$$\tan \theta_K \approx \left(\frac{\sigma_{xy}}{\sigma_{xx}^2}\right) \left(-\frac{2}{Z_0 d}\right) \left[\left(1 + \frac{1}{Z_+ \sigma_{xx}}\right) \left(1 + \frac{1}{Z_- \sigma_{xx}}\right)\right]^{-1},$$
(18)

where the unit of conductivity is Ω^{-1} cm⁻¹.

Equations (11) and (16) can be simplified by using the relation between σ_{xx} and $n_{f,0}$ obtained by the Eqs. (6) and (8).

$$\tan \theta_F = -\frac{1}{t_0} \left(\frac{\partial t_0}{\partial \sigma_{xx}} \right) \sigma_{xy},$$
$$\tan \theta_K = -\frac{1}{r_0} \left(\frac{\partial r_0}{\partial \sigma_{xx}} \right) \sigma_{xy}.$$
(19)

Note that since both $\tan \theta_F$ and $\tan \theta_K$ are proportional to σ_{xy} , the magneto-optical signals vanish when $\sigma_{xy}=0$, as expected. Dividing $\tan \theta_F$ in Eq. (11) by $\tan \theta_K$ in Eq. (16) allows σ_{xy} to divide out and produces

$$\frac{\tan \theta_F}{\tan \theta_K} = \frac{\frac{1}{t_0} \left(\frac{\partial t_0}{\partial n_{f,0}}\right)}{\frac{1}{r_0} \left(\frac{\partial r_0}{\partial n_{f,0}}\right)} = F(n_{f,0}).$$
(20)

Also, it can be expressed as

$$\frac{\tan \theta_F}{\tan \theta_K} = \frac{\frac{1}{t_0} \left(\frac{\partial t_0}{\partial \sigma_{xx}}\right)}{\frac{1}{r_0} \left(\frac{\partial r_0}{\partial \sigma_{xx}}\right)} = G(\sigma_{xx}).$$
(21)

Since $\delta n_f \ll n_{f,0}$, $t_0 \approx t_{xx}$, and $r_0 \approx r_{xx}$ in the linear polarization basis, the complex function $F(n_{f,0})$ only depends on the longitudinal index of refraction of the film. If the complex θ_F and θ_K are measured experimentally, one can solve Eq. (20) numerically to obtain $n_{f,0}$. Once $n_{f,0}$ is determined, we can use Eqs. (6) and (8) to calculate the complex longitudinal conductivity σ_{xx} of the film:

$$\sigma_{xx} = \frac{i\omega}{4\pi} (\varepsilon_b - n_{f,0}^2), \qquad (22)$$

where ε_b is the contribution to the dielectric function from bound carriers, which allows the conductivity to be determined for free carriers. In these measurements, we typically are interested in the response of all the carriers, bound and free, so ε_b is set to 0. Using the measured values for θ_F or θ_K , and plugging $n_{f,0}$ back into Eqs. (11) or (16), one can now determine the complex σ_{xy} .

B. Sign calibration

The sign calibration is important for our Faraday and Kerr measurements because the solutions for σ_{xx} and σ_{xy} depend on which signs are assigned to θ_F and θ_K . There are two sign conventions in describing the time evolution of the electric

field in magneto-optical measurements: $\exp(-i\omega t)$ or $\exp(+i\omega t)$. Determining the correct signs for θ_F and θ_K can be challenging, both experimentally and theoretically.

Experimentally, one must make sure that the direction of the magnetic field is known and that the direction of the changes in orientation and ellipticity in the polarization of the transmitted and reflected light are properly determined. Since the signals are demodulated using lock-in amplifiers, one must keep track of the phase of each lock-in in order to avoid sign errors.

In our measurements, the signs of the polarization signals are determined in three independent, yet overlapping, ways. The sign of the polarization rotation $[\operatorname{Re}(\theta_F) \text{ and } \operatorname{Re}(\theta_K)]$ is determined by rotating the PEM and linear polarizer P_2 together in a counterclockwise direction, as viewed along the beam's propagation direction toward the detector. This is equivalent to the sample rotating the transmitted/reflected polarization in the clockwise direction. The change in signal is compared with that produced by the sample in a magnetic field. Second, the sign of both the rotation and ellipticity can be verified by placing a compressively strained ZnSe slide, oriented at approximately 45° from vertical, in the beam, as described in Refs. 10 and 17. The index of refraction in ZnSe is decreased for the linear polarization along the compressive strain direction,¹⁸ and, hence, this polarization exits the ZnSe slide before the polarization that is perpendicular to the compressively strained axis. For the geometry shown in the boxed inset of Fig. 1, this phase shift between perpendicular linear polarizations produces a counterclockwise rotation of the linear polarization as well as a counterclockwise ellipticity, as viewed toward the detector. The strain on the ZnSe slide is applied in the y' direction by hanging a weight on a compression lever, as can be seen in greater detail in Ref. 17. Since the orientation of the slide critically affects the transmitted polarization, the slide holder is clamped to the optical table to prevent small movements of the slide when the weight is added or removed. The same calibration can be made using a wave plate, as described in Ref. 10, but a ZnSe slide is much less expensive than a zeroth-order infrared wave plate, and since the strain is applied externally, there is no ambiguity for the direction of the fast axis, which is not always clear on a wave plate. The compressed slide sign calibration technique can be used over a large wavelength range (500 nm-20 μ m) without any sign changes for two reasons: (1) the strain, and therefore the retardance, of the slide is small, so even at the shortest wavelengths in this range the retardance never approaches π , where fast and slow axes would reverse. (2) The piezobirefringence coefficient for ZnSe does not change sign in this wavelength range.¹⁸ These sign calibrations are performed with the sample in place and under exactly the same conditions that are used for measuring the sample. Once the calibrations are completed, the PEM is aligned with the laser polarization and the ZnSe slide is removed before the magnetic field is energized. Once the directions of the rotation and ellipticity signals are determined with respect to the magnetic-field direction and the direction of light propagation, one can use Ref. 19, which is a critically important paper for this work, to determine the proper signs of θ_F and θ_K .

Finally, the signs of polarization signals are verified by measuring $\text{Re}(\theta_F)$ and $\text{Im}(\theta_F)$ produced by a gold film at

120 meV. Since θ_F in gold is produced by free electrons, one can characterize the response of a sample in $Re(\theta_F)$ and $Im(\theta_F)$ as "electronlike" or "holelike." One can represent the complex σ_{xx} and σ_{xy} of gold using a simple Drude model, which in turn can be used to calculate θ_F using the thick-film transmission formula in Eq. (11). This same formula is then used to calculate "backward" from the measured θ_F and θ_K produced by other samples to obtain σ_{xx} and σ_{xy} . The signs of $\operatorname{Re}(\theta_F)$ and $\operatorname{Im}(\theta_F)$ determined by this calculation can be used to determine the signs produced by other samples. For example, applying Eq. (11) to a Drude model for a gold film produces $\operatorname{Re}(\theta_F) < 0$ and $\operatorname{Im}(\theta_F) < 0$ below 200 meV. Therefore, if the $\operatorname{Re}(\theta_F)$ signal for a sample has the same polarity with applied magnetic field as the signal from a gold film (electronlike), a negative value of $\operatorname{Re}(\theta_F)$ is used in the thickfilm equation to determine σ_{xx} and σ_{xy} . The sign of Im(θ_F) is determined the same way. This convention ensures that a film with an electronlike response in both $\operatorname{Re}(\theta_F)$ and $\operatorname{Im}(\theta_F)$ will produce the correct signs and magnitudes of σ_{xx} and σ_{xy} . Since θ_K from a gold film is too small to measure, we cannot use the gold film to calibrate the sign of θ_K .

The signs obtained using these three techniques are all consistent with each other. For example, rotating the PEM counterclockwise produced a signal with the opposite polarity as applying strain to the ZnSe slide. Furthermore, the signs (determined by rotating the PEM or straining a ZnSe slide) of the transmitted polarization signals produced by a gold film were consistent with the lower frequency (<200 meV) behavior of electrons in a magnetic field.

When the calibrated signs for the measured θ_F and θ_K are used in our thick-film equations, we obtain reasonable optical properties. First, the real (*n*) and imaginary (*k*) parts of the index of refraction are positive. Second, for probe frequencies below the plasma frequency of metallic samples such as Au and SrRuO₃, the real part of the dielectric function $\epsilon_1 = n^2 - k^2$ is negative, implying that n < k. The final signs for σ_{xx} and σ_{xy} are not arbitrarily assigned, but are determined from the measurements and the calibration procedure described here.

V. RESULTS

The measurements on the SrRuO₃ and Ga_{1-x}Mn_xAs films reported here probe the anomalous Hall effect (AHE), which is the Hall effect that arises from the sample magnetization. Therefore, to eliminate contributions from the ordinary Hall effect (OHE), which depends on the applied magnetic field *H*, these measurements were performed on films that are fully magnetized out of plane with zero applied magnetic field. In the case of Ga_{1-x}Mn_xAs where the remanent magnetization was very small, the finite-field linear behavior is extrapolated back to $H=0.^{20}$ The measurement and analysis techniques described in this paper can be applied to finite magnetic fields equally well.

Figure 4 shows θ_F and θ_K at 10 K and 0 T from the SrRuO₃ film with the sample fully magnetized out of plane as a function of probe energy *E*. θ_F and θ_K exhibit strong energy dependence, with Re(θ_F) and Im(θ_K) changing sign at

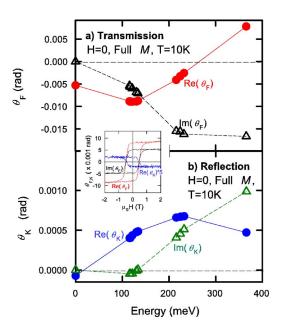


FIG. 4. (Color online) Energy dependence of (a) θ_F and (b) θ_K from a SrRuO₃ film with the sample fully magnetized perpendicular to the plane at 0 T and 10 K. Since the measurement is at H=0 T, the OHE as well as background signals from the substrate and windows, which are linear in H, do not contribute to the signal. Note the strong energy dependence and the sign changes in Re(θ_F) and Im(θ_K) at 250 K and 130 meV, respectively. The $\theta_F(E \rightarrow 0)$ plotted in (a) is determined using Eq. (13) with the four-probe dc measurements for the dc θ_H and σ_{xx} . Inset shows the relationship between Re(θ_F), Im(θ_F), and Re(θ_K) (which is multiplied by a factor of 5) at 117 meV and 10 K.

250 and 130 meV, respectively. Below 300 meV, both $\operatorname{Re}(\theta_F)$ and $\operatorname{Im}(\theta_F)$ are negative, indicating that the change in polarization is in the same sense as the Faraday signals from free electrons in a gold reference film. Strictly speaking, θ_F and θ_K are not defined at zero energy, but they approach well-defined values as $E \rightarrow 0$. The dc values of θ_F and θ_K in Fig. 4 are determined using the thick-film equations [Eqs. (11) and (16)], as $E \rightarrow 0$ with the four-probe dc measurements for σ_{xx} and σ_{xy} . The dc θ_H at 10 K and 0 T with the sample fully magnetized is -0.0052 rad, which is within 2% of $\theta_F(E \rightarrow 0)$ and confirms the expected relationship for metallic materials, where $\theta_F \approx \theta_H$ as frequency goes to zero.¹⁶ The θ_F and θ_K signals from SrRuO₃ are significantly larger than what has been measured in nonmagnetic metals. For example, in gold, copper, and HTSC films, where the Faraday signals are produced by free carriers in a magnetic field, $\operatorname{Re}(\theta_F) \approx \operatorname{Im}(\theta_F) \approx 0.001 \text{ rad at } 8 \text{ T.}^{16,21}$

The inset of Fig. 4 shows θ_F and θ_K from the SrRuO₃ film as a function of applied magnetic field at a probe energy of 117 meV at 10 K. The response of θ_F to the applied magnetic field is electronlike, as determined by comparing the signals to a gold film reference sample. Although the intensity of transmitted light can be as small as 0.01%, the magneto-optical signals in transmission (θ_F) are approximately an order of magnitude larger than those obtained in reflection (θ_K). This is due to the fact that in metallic films $\theta_F \propto \sigma_{xy}/\sigma_{xx} = \theta_H$ while $\theta_K \propto \sigma_{xy}/(\sigma_{xx})^2 = \rho_{xy}$, as is suggested

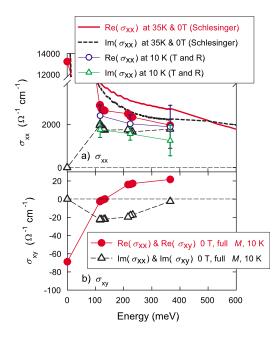


FIG. 5. (Color online) The longitudinal conductivity σ_{xx} (a) and transverse (AHE) conductivity σ_{xy} (b) for a SrRuO₃ film as a function of probe energy. The solid symbols are obtained from θ_F and θ_K measurements at 10 K and 0 T with the sample fully magnetized out of plane. The smaller symbols with error bars in (a) are determined from conventional transmittance and reflectance measurements of the same SrRuO₃ film. The heavy solid [Re(σ_{xx})] and dashed [Im(σ_{xx})] lines in (a) are from Kramers-Kronig analysis of reflectance measurements at 35 K and 0 T on a different SrRuO₃ sample by Schlesinger and co-workers.

by Eqs. (13) and (18). Therefore, for highly metallic films in the MIR, where $\sigma_{xx} \gg \sigma_{xy}$, θ_F is typically larger than θ_K for the same σ_{xy} .

Figure 5 shows the measured complex (a) longitudinal conductivity σ_{xx} and (b) transverse conductivity σ_{xy} for the SrRuO₃ film. The large symbols are obtained from θ_F and θ_K measurements at 10 K and 0 T with the sample fully magnetized out of plane. The signs of σ_{xx} and σ_{xy} represented by the symbols are not assigned arbitrarily, but are determined experimentally using the techniques described in Sec. IV B. The conductivity was defined to include contributions from both bound and free charges, although the bound charge contribution in the MIR was found to be small and had almost no affect on the results. The heavy solid $[\operatorname{Re}(\sigma_{xx})]$ and dashed $[Im(\sigma_{xx})]$ lines in (a) are from H=0 Kramers-Kronig analysis of reflectance measurements at 35 K and 0 T on a different SrRuO₃ film by Schlesinger and co-workers.²² The qualitative agreement between the σ_{xx} obtained by θ_F and θ_K measurements and that obtained by reflectance measurements is excellent over the entire energy range. The quantitative differences could be readily accounted for by the differences in the two samples, especially since the sample measured by Schlesinger and co-workers had a dc resistivity at 10 K of approximately 20 $\mu\Omega$ cm,^{3,22} a factor of 3 smaller than the resistivity of the sample used in the θ_F and θ_K measurements. The complex σ_{xx} obtained using θ_F and θ_K measurements can be compared with values determined by transmittance and reflectance measurements on the same sample

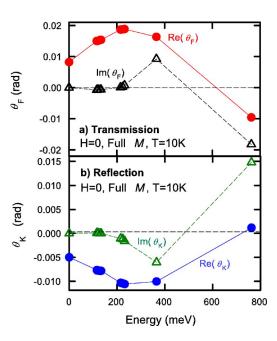


FIG. 6. (Color online) Energy dependence of the AHE (a) θ_F and (b) θ_K from a Ga_{1-x}Mn_xAs film with the sample fully magnetized perpendicular to the plane at 0 T and 10 K. Note the strong energy dependence and the sign changes in both the real and imaginary parts of θ_F and θ_K . θ_F and θ_K exhibit qualitatively similar behavior in this case.

[small symbols with error bars in Fig. 5(a)]. In the 117–224 meV range, σ_{xx} obtained using θ_F and θ_K measurements is within 20% of the average values obtained using transmittance and reflectance measurements. At 366 meV, the average difference is closer to 30% due to the difficulty in aligning the weak HeNe laser, which has an output of 2 mW that is 2 orders of magnitude lower than the CO_2 and CO lasers. Furthermore, the differences in the quality of the polish of the wedged sample and reference substrates are much more critical at this shorter wavelength. At all wavelengths, σ_{xx} obtained by Faraday and Kerr measurements is within the error bars of σ_{xx} obtained using transmittance and reflectance measurements. Challenges in absolute transmission and reflection measurements using discrete laser lines on a wedged sample could easily account for the differences and suggest that in this case, θ_F and θ_K measurements, which are self-normalizing, may allow a more accurate determination of σ_{xx} .

Figure 6 shows θ_F and θ_K at 10 K and 0 T from the $Ga_{1-x}Mn_xAs$ film with the sample fully magnetized out of plane as a function of probe energy. θ_F and θ_K show strong energy dependence, including sign changes. θ_F and θ_K exhibit qualitatively similar behavior in this case.

Figure 7 shows the measured complex (a) longitudinal conductivity σ_{xx} and (b) transverse conductivity σ_{xy} for the Ga_{1-x}Mn_xAs film. These were obtained from θ_F and θ_K measurements at 10 K and 0 T with the sample fully magnetized out of plane. The *E*=0 results are from dc magnetotransport measurements by Dubon and co-workers on similar samples also grown by them. The symbols with error bars in Fig. 7(a) show σ_{xx} obtained from transmittance and reflectance measurements on the same sample. Since it was more difficult to

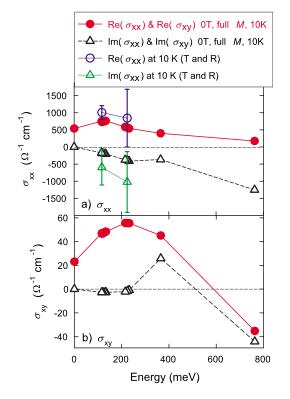


FIG. 7. (Color online) The longitudinal conductivity σ_{xx} (a) and transverse (AHE) conductivity σ_{xy} (b) from a Ga_{1-x}Mn_xAs film as a function of probe energy. The measurements are at 10 K and 0 T with the sample fully magnetized out of plane. The smaller symbols with error bars in (a) are determined from conventional transmittance and reflectance measurements of the same Ga_{1-x}Mn_xAs film. The E=0 results are from dc measurements by Dubon and coworkers on similar samples, which were also grown by them.

obtain an optically smooth polish on GaAs (compared to LaSrGaO₄), and perhaps because it has a larger index of refraction (compared to LaSrGaO₄), differences in the quality of the GaAs substrate polish led to more than a factor of 2 uncertainty in the transmittance at 366 meV due to scattering. Therefore, transmittance and reflectance measurements at 366 meV and shorter wavelengths are not included. The quality of the polish made no measurable impact at the longer wavelengths. Since the Faraday and Kerr measurements involve changes in polarization with applied magnetic field, substrate roughness did not measurably affect the accuracy of the θ_F and θ_K measurements. The transmittance and reflectance measurements led to a σ_{xx} that qualitatively agrees with the σ_{xx} obtained by Faraday and Kerr measurements, which was within the error bars of the transmittance and reflectance measurements.

The magnitude and frequency dependence of σ_{xx} compare reasonably well with results from other experiments and from theoretical models. The MIR $\text{Re}(\sigma_{xx})$ extrapolates smoothly to the dc σ_{xx} . $\text{Re}(\sigma_{xx})$ is similar to that obtained in Refs. 23 and 24 for $\text{Ga}_{1-x}\text{Mn}_x\text{As}$ samples. The frequency dependence of $\text{Re}(\sigma_{xx})$ is also similar to that predicted by theoretical models.^{7,8} One can also represent the complex conductivity in terms of the complex dielectric constant ϵ . In this case, the $\text{Re}(\epsilon)$ remains at a fairly constant value of 9 over the measured frequency range, whereas $Im(\epsilon)$ strongly decreases with increasing energy as $E^{-1.3}$, reaching a value of approximately 3 at 760 meV. This compares well with ellipsometry measurements on $Ga_{1-x}Mn_xAs$ films where $Re(\epsilon)$ levels off at a value between 10 and 12 as $E \rightarrow 0$, while Im(ϵ) is close to 2 at 750 meV and begins to rise as $E \rightarrow 0.25$ Note that the negative sign in $Im(\sigma_{xx})$ in Fig. 7(a) is consistent with the positive $\operatorname{Re}(\epsilon)$ found in ellipsometry measurements in the MIR. The reasonable behavior of σ_{xx} in Fig. 7(a) provides added confidence that the σ_{xy} found in Fig. 7(b) accurately represents the response of the $Ga_{1-x}Mn_xAs$ film. The MIR $\operatorname{Re}(\sigma_{xy})$ extrapolates smoothly to the dc σ_{xy} . Note that the low-energy behavior of the $\operatorname{Re}(\sigma_{xy})$ is holelike, suggesting that there is no sign reversal in the AHE between 0 and 100 meV.

VI. CONCLUSION

We have demonstrated experimental and analytical techniques that can be used to determine the complete complex magnetoconductivity tensor from Faraday and Kerr measurements in ferromagnetic metals and semiconductors. It is interesting to note that no absolute intensity measurements are required; the polarization of the reflected and transmitted light is sufficient to completely determine the complex magnetoconductivity tensor. In both SrRuO₃ and Ga_{1-x}Mn_xAs films, σ_{xx} obtained using Faraday and Kerr measurements was quantitatively consistent with the values for σ_{xx} that were obtained using conventional transmittance and reflectance measurements. Furthermore, in both materials, σ_{rv} showed strong spectral features, including peaks and sign changes, which will be discussed in future papers. With the increasing application of magneto-optical measurements to study magnetic and nonmagnetic materials, and with the specific interest in the infrared longitudinal and transverse conductivities of magnetic oxides and semiconductors, these techniques may have a significant impact in a number of fields.

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