Crystal field analysis and emission cross sections of Ho3+ in the locally disordered single-crystal laser hosts $M^+Bi(XO_4)_2$ $(M^+ = Li, Na; X = W, Mo)$

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The spectroscopic properties of Ho³⁺ laser channels in locally disordered tetragonal NaBi(WO₄)₂ (NaBiW), $Nabi(MoO₄)₂$ (NaBiMo), and LiBi(MoO₄)₂ (LiBiMo) single crystals grown by the Czochralski method have been studied in the 5–300-K temperature range using several holmium concentrations $[H_0] \approx 0.05-0.6$ 10^{20} cm⁻³. Here 5-K polarized optical absorption and photoluminescence measurements have been used to determine the energy position of 85, 56, and 39 $Ho³⁺$ Stark levels in NaBiW, NaBiMo, and LiBiMo crystals, respectively. These energy levels were labeled with irreducible representations corresponding to the *S*⁴ local symmetry of an average optical center. Single-electron Hamiltonians combining together free-ion and crystalfield interactions have been used in the fit of experimental energy levels and in the simulation of the corresponding $4f^{10}$ Ho³⁺ configuration for NaBiW and NaBiMo crystals. Very satisfactory correlations were obtained between experimental and calculated crystal-field levels, with rms deviations σ =8.8 and 7.3 cm⁻¹ for NaBiW and NaBiMo, respectively. The radiative properties and emission cross sections of Ho^{3+} laser channels in these hosts were calculated by the Judd-Ofelt theory and compared with experimental results. The emission cross sections of Ho3+ in NaBiW are similar to those observed in other crystal laser hosts, and positive gain cross sections can be achieved in extended spectral ranges. These properties make the Ho^{3+} -doped double tungstates and double molybdates feasible materials for tunable and short-pulse laser operation.

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I. INTRODUCTION

Disordered crystals with inhomogeously broadened optical bands are presently being searched for in the context of ultrafast (femtosecond) laser systems.¹ Tetragonal double tungstate (DT) and double molybdate (DMo) single crystals with general formula $MT(XO_4)_2$ (*M*=monovalent, *T* =trivalent, and $X = W^{6+}$ or Mo⁶⁺ ions) are a class of locally disordered compounds related to the CaWO₄ scheelite type. For brevity we shall refer to each particular DT or DMo compound as *MTX*. The substitution of two Ca^{2+} ions by an M^+T^{3+} pair gives rise to a large number of compounds, some of them exhibiting polymorphism; therefore, different crystallographic structures (triclinic, monoclinic, orthorhombic, etc.) can be selected.² LiLaW,³ LiTMo $(T=Bi,4 Y,5 La,5)$ $(T=Bi,4 Y,5 La,5)$ $(T=Bi,4 Y,5 La,5)$ $(T=Bi,4 Y,5 La,5)$ $(T=Bi,4 Y,5 La,5)$ Ce-Lu⁵), NaTW (*T*=Bi,⁴ Y,⁶ La,⁶ Nd-Er⁶), NaTMo (*T*=Bi,⁴ Y,⁷ La,⁸ Gd⁸), KLaMo,⁹ and AgLaW,¹⁰ compounds have congruent melt, and the tetragonal crystallographic phase is stable upon cooling from their congruent melting points to room temperature. Single crystals of these compounds can be grown in air by the Czochralski method, with high growth rates (typically 1 mm/h) and a relatively low crystal cost, which ease their applications.

The most common trivalent lanthanide $Ln³⁺$ ions of interest for laser operation are $Ln³⁺ = Pr$, Nd, Ho, Er, Tm, and Yb. The *f* electrons of Yb^{3+} and Tm^{3+} experience a stronger interaction with the vibrational lattice environment, and therefore these ions are well suited for tuning and mode-locked laser operation even in ordered crystals. The broad bandwidths are also useful for pumping with commercial (nonwavelength-matched) semiconductor diodes without thermal stabilization, leading to low-cost systems. Another outstanding application already demonstrated in some DT and DMo crystals is solid-state stimulated Raman laser shifting

(SRS).^{[11](#page-12-12)} Tetragonal DT and DMo crystals have larger Raman bandwidths [full width at half maximum (FWHM) ≈ 15 cm⁻¹] than the ordered phases of DT and DMo compounds, like monoclinic \angle KY(WO₄)₂ (FWHM ≈ 5 cm⁻¹), leading thus to operation in shorter SRS picosecond regimes.¹² When the other Ln^{3+} (Pr-Er) are considered, the above applications require spectral inhomogeneous broadening. In tetragonal DT and DMo the spectral broadening derives from two features of the crystal host: the overlapping Ln^{3+} contributions arising from two shared crystal sites 2*b* and 2*d* and the broadening associated to structural disorder that is, the optical centers resulting from different shortrange *M* and *T* distributions around each one of the two above Ln^{3+} lattice sites.

Laser tunability and mode-locked operation have been recently achieved by using Yb^{3+} - and Tm^{3+} -doped tetragonal DT and DMo hosts: for Yb^{3+} a continuous tuning range up to 65 nm has been reached around 1.05 μ m,^{[13](#page-12-14)} and 212 nm around 1.93 μ m has been also realized with Tm³⁺.^{[14](#page-12-15)} The tuning ranges so far demonstrated were limited by the spectral responses of the optical cavity elements (mirrors and birrefringent filter), and then even broader tuning ranges are expected. The estimated full tuning range can be used in mode-locked operation to produce \leq 50-fs laser pulses. First steps in this direction have already shown pulses of 120 fs in Yb-doped NaGdW¹³ and 90 fs in Yb-doped NaLuW.¹⁵ To achieve similar results for other laser-active Ln^{3+} (Pr-Er) in DT and DMo still remains a challenging task. Knowledge of the spectroscopic ion properties is required to evaluate this possibility.

 $NaBi(WO_4)_2$ (NaBiW), $NaBi(MoO_4)_2$ (NaBiMo), and $LiBi(MoO₄)₂$ (LiBiMo) single crystals have a locally disordered tetragonal crystallographic structure with space group $(SG) I⁴$.^{[16](#page-12-17)} Laser operation of $Ln³⁺$ in NaBiW, and NaBiMo

was first shown using Nd^{3+} .^{[17](#page-12-18)} In the present work we study Ho^{3+} in NaBiW, NaBiMo, and LiBiMo as a potential laser ion for the visible and midinfrared regions.¹⁸ Although Ho^{3+} lacks a strong absorption for diode pumping at λ $=800-1000$ nm, it can be sensitized in this region by Tm^{3+} and Yb^{3+} codoping¹⁹ or pumped directly with the λ \approx 1.9 μ m emissions of GaInAsSb/AlGaAsSb quantum-well diodes and Tm lasers using a pumping cascade scheme. Therefore, Ho^{3+} remains of wide interest in present laser technology. 20

Using low-temperature (5 K) polarized spectroscopic techniques and crystal field modeling we have determined the Ho^{3+} Stark energy levels from ${}^{5}I_8$ up to ${}^{3}H_6$ (for NaBiW) or 5G_5 (for NaBiMo and LiBiMo) manifolds, which have been labeled with the adequate irreducible representation (IR) for the current S_4 local symmetry, and the large bandwidth of the spectral bands has been confirmed. Radiative and nonradiative processes of these Ho^{3+} materials have been studied in detail to evaluate the application as solid-state laser systems.

II. GROWTH PROCEDURES AND CRYSTAL STRUCTURE

The Czochralski crystal growth procedures of NaBiW, NaBiMo, and LiBiMo have been described previously.^{4,[16](#page-12-17)} Ho-doped single crystals with optical quality were obtained reacting the host compounds with 99.9% Ho₂O₃ from CERAC Inc. during \sim 100 h at 1133 K for tungstates and 1058 K for molybdates. The products were melted in Pt crucibles, and crystals were pulled at the following rates: 0.8–4 mm/h for NaBiW, 1.3–2.3 mm/h for NaBiMo, and 1.7–1.9 mm/h for LiBiMo.

The Ho concentration in the crystal $[H_0]_{\text{crys}}$ was determined by proton-induced x-ray emission (PIXE) spectroscopy using for each host samples of the highest Ho concentration available. For this purpose we used $L_{\alpha1,\alpha2}$ holmium x-ray emission at about 6.719 keV, which was free of interference with x-ray emissions from other crystal constituents. Samples were irradiated in vacuum with 2.0-MeV protons supplied by a 2.5-MV Van de Graaff accelerator. The proton beam was collimated down to 1.5 mm in diameter, and beam currents close to 3 nA were used. A 165-eV-resolution Si(Li) detector was placed at 110° relative to the proton beam direction, and the face of the sample was tilted to an angle of 15° with the incident beam. Spectra analyses were performed by the AXIL-PC V3.1 computer code. Quantitative analysis was carried out through the DATTPIXE program, which accounts for proton energy $loss.²¹$

The Ho content for lower concentrations was determined by comparison of the optical absorption (OA) intensities. Table [I](#page-1-0) summarizes the results obtained. It is worth noting that the segregation coefficient $S = [Ho]_{crys}/[Ho]_{melt}$ of holmium in Bi-based DT and DMo is larger than 1, in agreement with previous results of other lanthanides.⁴

In the noncentrosymmetric tetragonal SG $I\overline{4}$ (No. 82), $Z=2$, Li⁺/Na⁺ and Bi³⁺ cations share two nonequivalent lattice sites 2*d* and 2*b*, both with *S*⁴ local symmetry and the same high-symmetry axis parallel to the crystal *c* axis, but with different occupancy factors: namely, $63(60)\%$ Bi

TABLE I. Holmium concentration in NaBiW, NaBiMo, and LiBiMo crystals used in this work. The Ho concentration in lowdoped samples was deduced from a comparison of the $Ho³⁺$ optical absorption (OA) taken as reference the PIXE results.

	$[\mathrm{Ho}]_{\mathrm{melt}}$ [mol $%$]	$[Ho]_{\text{crys}}$ $\lceil \text{mol} \; \% \rceil$	$[Ho]_{\text{cyrs}}$ $[10^{20} \text{ cm}^{-3}]$
NaBiW	0.02	0.11	0.07 OA
	0.31	0.56	0.35 PIXE
NaBiMo	0.02	0.10	0.06 OA
	0.10	0.48	0.30 OA
	0.33	1.09	0.68 PIXE
LiBiMo	0.02	0.08	0.05 OA
	0.10	0.31	$0.20 \Omega A$
	0.32	0.64	0.41 PIXE

+37(40)%Na and 46(40)%Bi+54(60)%Na for 2*d* and 2*b* sites, respectively, in NaBiW,^{16[,22](#page-12-23)} 45%Bi+55%Na and 53%Bi+47%Na in NaBiMo[,16](#page-12-17) or 0.52%Bi+0.48%Li and 0.51% Bi+0.49% Na in LiBiMo, for 2d and 2b sites,¹⁶ respectively. In these crystals as well as in other isostructural DT and DMo, detailed crystallographic analyses have shown that Ln^{3+} active ions replace $\text{Bi}^{3+}(T^{3+})$ in both sites: Er^{3+} in NaBiW, NaBiMo, and LiBiMo,¹⁶ Yb³⁺ in NaGdW,¹³ and Yb^{3+} in LiGdMo.²³

III. EXPERIMENTAL TECHNIQUES

Ground-state optical absorption (GSA) was recorded in a Varian spectrophotometer model CARY-5E. Photoluminescence (PL) was excited with a cw Ar⁺ laser or with Rhodamine R-6G ($\lambda \approx 591$ nm) and Coumarin-480 (λ \approx 485 nm) emissions of a dye laser (LSI, model DUO-220). The holmium emission was dispersed with a SPEX 340E spectrometer using 1200- or 600-l/mm gratings. The light was detected with a water-cooled R2658 Hamamatsu InGaAs photomultiplier $(\lambda = 0.0185 - 0.1030 \mu m)$ and a 77-K-cooled Ge photodiode $(\lambda = 0.8 - 1.7 \mu m)$. PL emissions were corrected by the spectral response of the equipment, which was previously determined using a calibrated W lamp standard. The intensity of the ${}^5F_5 \rightarrow {}^5I_7$ and ${}^5S_2 \rightarrow {}^5I_6$ transitions has been used to relate the spectral responses of the photomultiplier and Ge photodiode. To record the PL spectral distributions we used a lock-in amplifier. Lifetime measurements were excited with the dye laser described above. Coumarine 440 (or 540A) and Rhodamine 6G dyes were used for excitation of the 4S_2 and 5F_5 holmium levels, respectively. The laser pulse width has $FWHM \approx 3$ ns. The nominal spectral bandwidth is ≈ 0.3 nm, but in practice the wavelength accuracy is limited to ± 1 nm by the mechanics of the tuning system. The pulsed detector signal outputs were monitored with a 500-MHz Tektronix TDS520 digital oscilloscope. In the spectroscopic experiments the sample temperature was selected in the 5–300-K temperature range by using a He closed-cycle cryostat connected to a suitable temperature controller.

FIG. 1. Comparison of the 5-K (solid line) and 75-K (dashed line) selected optical absorption of Ho^{3+} in NaBiW. [Ho]=0.35 \times 10²⁰ cm⁻³. π spectra are arbitrarily displaced in the *y* axis for clarity.

IV. Ho3+ ENERGY LEVELS

A. Low-temperature ground-state absorption and photoluminescence

 Ho^{3+} energy-level positions were investigated by lowtemperature optical absorption and photoluminescence measurements. The large degeneracy of the ground $^{5}I_{8}$ multiplet of the $4f^{10}$ configuration of Ho^{3+} , 13 Stark energy levels in the current S_4 symmetry, hampers this task. Usually ${}^5I_8(1)$ and ${}^{5}I_{8}(2)$, and sometimes even ${}^{5}I_{8}(3)$, excited Stark levels are only few cm⁻¹ above the ground ${}^{5}I_8(0)$ (Ref. [18](#page-12-19)) level, and therefore they can be electronically populated even at 5 K, giving rise to *hot* bands, ${}^5I_8(n \ge 1) \rightarrow {}^{2S+1}L_J$, merged together with those corresponding to true ground-state transitions, ${}^{5}I_8(0) \rightarrow {}^{2S+1}L_J$. Measurements at very low temperatures \approx 1.4 K can be thus required to fully depopulate the excited ${}^{5}I_8(n \ge 1)$ Stark levels.

To overcome this uncertainty, we have measured the optical absorption of several excited multiplets at different temperatures in the 5–75-K range. Figure [1](#page-2-0) shows a representative example of the OA thermal evolution. It is worth noting that while the intensity of some bands in a given polarization decreases with increasing temperature, some new bands may appear in the alternative polarization but shifted to low energy. This behavior will be explained later in the framework of the transition polarization rules expected for the *S*⁴ point symmetry. At this point we just want to note that from the systematic analysis of the energy position of the bands whose observation has been induced by raising the sample temperature we determine ${}^{5}I_8(n \ge 1)$ levels at 11, 29, and 51 cm−1 for Ho3+ in NaBiW. Similar results were achieved for Ho^{3+} in NaBiMo and LiBiMo; see Table [II.](#page-3-0)

The energy positions of the ${}^{5}I_8(n)$ Stark levels have been further assessed by low-temperature PL to the ground ${}^{5}I_8$ multiplet. Figure [2](#page-5-0) shows a comparison of the ${}^5S_2 \rightarrow {}^5F_5 \rightarrow$, and ${}^{5}I_6 \rightarrow {}^{5}I_8(n)$ emissions of Ho-doped NaBiMo taken as an example. Results for Ho-doped NaBiW were very similar, and they are not shown for the sake of brevity, while the bands observed in Ho^{3+} -doped LiBiMo have lower resolution due to the broader bands observed and therefore led to a lower number of experimentally determined ${}^{5}I_8$ Stark levels [see Fig. $2(c)$ $2(c)$ and Table [II](#page-3-0)].

For two $({}^5F_5$ and 5I_6) of the Ho³⁺-emitting multiplets in NaBiMo the PL shows a first set of four overlapped bands $(0, 0)$ 22, 49, and 72 cm^{-1}) followed by an energy gap, an isolated band (179 cm^{-1}) , and finally two overlapped bands $(233, 12)$ 260 cm^{-1}). The photoluminescence from the ${}^{5}S_2$ multiplet is slightly different in terms of the relative intensity of the last three bands. This suggests that the ground ${}^5S_2(0)$ Stark level belongs to a different IR than those of ${}^5F_5(0)$ and ${}^5I_6(0)$.

Taking into account the possible *hot* OA bands, the relative Ho^{3+} energy-level positions in the considered Bi-based DT and DMo have been investigated by 5-K-polarized OA. The wider range of ultraviolet transparency of NaBiW allows identifying electronic transitions to multiplets up to 28000 cm^{-1} above the ground $5I_8$ multiplet—i.e., up to ${}^{3}H_{4}$ —with only the exception of ${}^{5}I_{4}$. In NaBiMo and LiBiMo hosts the Ho^{3+} spectra are limited by the UV band edge at \sim 24 000 cm⁻¹—i.e., up to ⁵ G_5 or a lower-energy multiplet. Moreover, the Ho^{3+} spectra in LiBiMo are less resolved as a consequence of the presence of broader bands, hampering the correct identification of hot bands, and therefore a lower number of Stark levels was determined and the polarization assignment is more uncertain. Apart from these facts the results are basically similar in the three matrices. For the sake of brevity we present in Fig. [3](#page-5-1) only the results concerning to Ho^{3+} -doped NaBiW crystal and Table [II](#page-3-0) includes the energy levels obtained for each of the three crystals. It is worth remarking that the π (E||c and $B \perp c$) spectra are clearly different from the σ (**E** \perp **c** and **B**||**c**) and α (**E** \perp **c** and $\mathbf{B} \perp c$) spectra; **E** and **B** are the electromagnetic fields and *c* the crystal axis. The two latter spectra are very similar between them, as expected from induced electric-dipole (ED) transitions for the *S*⁴ point symmetry discussed later.

The OA results achieved at 5 K and displayed in Figs. [1](#page-2-0)[–3](#page-5-1) do not show any resolved contribution from the two possible lattice sites or multiple host environments for Ho. Further measurements made at 1.2 K on selected multiplets showed bandwidths similar to those obtained at 5 K, and no new band structure was found. Therefore, the present energylevel sets must be understood, in each case, as corresponding to an average Ho center including the nonresolved contributions of different Ho sites and environments.

B. Crystal-field analysis and simulation of 4*f* **¹⁰ energy levels**

Although for Ln^{3+} ions with large ionic radii, like Nd and Pr, the C_2 local symmetry was required to adequately describe the 4f^N energy sequences and polarization behavior of transitions observed in NaBiW, NaBiMo, and LiBiMo crystals, 24.25 the first option for a similar analysis of Ho³⁺ spectral properties must be the S_4 symmetry of T sites in the host. In fact the latter symmetry described successfully the spectroscopic properties of Er^{3+} (with ionic radii close to Ho^{3+}) in the same hosts.²⁶ In the current analysis the free-ion (FI) and crystal-field (CF) interactions have been simulta-

			NaBiW		NaBiMo	LiBiMo				NaBiW		NaBiMo	LiBiMo				NaBiW		NaBiMo	LiBiMo
${}^{2S+1}L_I$	IR	E_{o}	$\ensuremath{{E_c}}$	E_{o}	$\ensuremath{E_{c}}$	E_{o}	$^{2S+1}L_J$ – IR		E_{o}	E_c	E_{o}	$E_{\rm c}$	E_{o}	$^{2S+1}L_J$ – IR		E_{o}	E_c	E_o E_c		E_{o}
$^{5}I_{\rm g}$	$\Gamma_{3,4}$	$\overline{0}$	3	$\boldsymbol{0}$	\boldsymbol{I}	$\overline{0}$	5F_5	Γ_2	15416	15411	15404	15397	15398		$\Gamma_{3,4}$	22247	22230	22215	22205	
	Γ_2	11	10	9	9			$\Gamma_{3,4}$		15416		15402			Γ_1		22238	22228	22214	
	Γ_2	29	28	$22\,$	$18\,$			Γ_1	15441	15432	15426	15418	15508		Γ_2	22252	22246		22215	22221
	Γ_1	51	52	45	43	55		Γ_1	15480	15475	15469	15462	15463							
	Γ_1		61		52			$\Gamma_{3,4}$	15530	15542	15514	15523		5F_1	Γ_1		22281	22261	22253	22277
	$\Gamma_{3,4}$	79	76	72	64			Γ_{2}	15548	15552		15531	15528		$\Gamma_{3,4}$	22323	22326	22308	22307	
	Γ_1		<i>201</i>	179	178	178		Γ_1		15561	15529	15542								
	$\Gamma_{3,4}$		254	233	234			$\Gamma_{3,4}$	15575	15580	15559	15563	15551	5G_5	Γ_1	23874	23876	23860	23848	23819
	Γ_1		260		243										$\Gamma_{3,4}$	23879	23883	23838	23855	23842
	Γ_2	266	262	260	262	252	$5S_2$	Γ_2		18410		18395			Γ_2		23905		23870	
	Γ_1		283		259			Γ_1	18414	18413	18404	18397	18400		Γ_2		23941		23907	
	$\Gamma_{3,4}$		289		272			$\Gamma_{3,4}$	18438	18440	18422	18422	18424		$\Gamma_{3,4}$		23944		23912	
	Γ_2		304		272			Γ_{2}	18449	18448		18430			Γ_1	23963	23960		23928	
															$\Gamma_{3,4}$	23967	23975	23946	23946	23927
$^{5}I_{7}$	Γ_2		5130		5125		5F_4	Γ_1		18525	18508	18512	18502		Γ_1	23998	23992		23957	23968
	$\Gamma_{3,4}$	5135	5135	5130	5130	5135		$\Gamma_{3,4}$	18519	18534		18513								
	Γ_2	5135	5138	5130	5131			Γ_2	18529	18526	18513	18519		5G_4	Γ_2		25725			
	Γ_1		5146 5144		5141			$\Gamma_{3,4}$	18600	18597	18579	18582	18571		Γ_1		25776 25770			
	$\Gamma_{3,4}$		5160		5150			Γ_1		18599		18589	18581		$\Gamma_{3,4}$		25779 25781			
	Γ_1		5182	5165	5174			Γ_2	18604	18614	18585	18594			Γ_1		25829			
	$\Gamma_{3,4}$	5205	5202	5193	5192	5188		Γ_1	18625	18625	18605	18608			$\Gamma_{3,4}$	25825	25836			
	Γ_2		5207	5201	5198										Γ_2	25850	25863			
	$\Gamma_{3,4}$	5264	5267	5255	5258		5F_3	$\Gamma_{3,4}$	20562	20550	20547	20538	20531		Γ_1	25914	25891			
	Γ_2	5265	5265	5255	5260	5258		Γ_1	20577	20571	20564	20556	20545							
	Γ_1		5268		5261			Γ_2	20621	20615	20606	20598	20605	3K_7	$\Gamma_{3,4}$		26125			
								$\Gamma_{3,4}$	20672	20669	20655	20653	20637		Γ_2		26126			
$^{5}I_{6}$	Γ_1		8651		8651			Γ_2	20666	20668	20645	20653			Γ_1		26127			
	Γ_2		8656 8651		8651										Γ_2	26145	26143			
	$\Gamma_{3,4}$		8657		8654		5F_2	Γ_2		21043		21034			$\Gamma_{3,4}$	26151	26137			
	Γ_2	8665	8663		8657			Γ_1	21046	21046	21035	21035	21032		Γ_1		26166			
	$\Gamma_{3,4}$	8667	8664	8662	8664	8659		$\Gamma_{3.4}$	21085	21085	21073	21073	21058		$\Gamma_{3,4}$		26172			
	Γ_1		8673		8670			Γ_2	21137	21138	21120	21126	21114		Γ_1		26176			
	Γ_2	8677	8677		8673	8676									$\Gamma_{3,4}$		26179 26185			
	Γ_1		8752		8751		3K_8	Γ_1		21327 21321 21321 21316					Γ_2		26180 26182			

TABLE II. Energy levels (in cm⁻¹) of Ho³⁺, observed at 5 K (E₀) and calculated (E_c) in S₄ symmetry for NiBiX, X=Mo or W, and LiBiMo crystals. IR indicates the corresponding irreducible representation, Γ_1 , Γ_2 (σ spectra) and $\Gamma_{3,4}$ (π spectra).

174208-4

PHYSICAL REVIEW B **75**, 174208 (2007)

CRYSTAL FIELD ANALYSIS AND EMISSION CROSS CRYSTAL FIELD ANALYSIS AND EMISSION CROSS ...

TABLE II. *(Continued.)*

${}^{2S+1}L_I$	IR	E_{o}	E_{c}	E_{o}	E_{c}	E_{o}	${}^{2S+1}L_J$ IR		E_{o}	E_{c}	E_{o}	$E_{\rm c}$	E_{o}	$^{2S+1}L_{J}$	IR	$E_{\rm o}$	E_c	E_{o}	E_c	E_{c}
	$\Gamma_{3,4}$	8761	8763	8762	8759	8744		Γ_2	21327	21324	21321	21320	21318		Γ_2		26186			
	Γ_2		8773		8768			$\Gamma_{3,4}$	21331	21329	21323	21324								
								Γ_1		21338		21329		5G_5	Γ_{2}	27484	27495			
$^{5}I_5$	$\Gamma_{3,4}$		11215		11208			$\Gamma_{3,4}$	21341	21340		21333			$\Gamma_{3,4}$	27507	27502			
	Γ_1		11217		11212			Γ_2		21340	21329	21330			Γ_1	27507	27516			
		$\Gamma_{3,4}$ 11224 11220			11220			Γ_1		21362		21352			Γ_1	27583	27590			
	Γ_1		11224		11219			$\Gamma_{3,4}$	21400	21411	21390	21396			$\Gamma_{3,4}$	27585	27574			
	Γ_2		11226		11226			Γ_1		21412		21397			Γ_2	27635	27629			
	Γ_1		11284		11285			Γ_1		21414		21399	21400			$\Gamma_{3,4}$ 27653	27629			
	$\Gamma_{3,4}$		11286 11306		11301			Γ_2	21423	21431	21410	21415	21412		Γ_1		27632			
	Γ_2		11310		11305			$\Gamma_{3,4}$	21441	21437	21428	21419								
								Γ_2		21437 21438		21419		$^{5}H(4)_{6}$	Γ_1	27728	27733			
$^{5}I_4$	Γ_1		13170		13180										Γ_2		27738			
	$\Gamma_{3,4}$		13240		13243		5G_6	Γ_2	21996	22007	21980	21988	21963		$\Gamma_{3,4}$	27756	27750			
	Γ_2		13292		13296			$\Gamma_{3,4}$	22006	22015	21982	21994			Γ_2	27771	27780			
	Γ_1		13307		13297			Γ_1		22026		22005			$\Gamma_{3,4}$	27865	27846			
	Γ_2		13333		13329			Γ_2	22077	22082	22056	22057	22047		Γ_2		27846			
	$\Gamma_{3,4}$		13384		13382			Γ_2		22083		22058			Γ_1		27866			
	Γ_1		13494		13477			Γ_1	22155	22156		22132			Γ_1		27874			
								$\Gamma_{3,4}$	22158	22155	22133	22130	22118		$\Gamma_{3,4}$	27897	27910			
															Γ_2	27918	27927			

FIG. 2. 5-K ${}^{2S+1}L_J \rightarrow {}^{5}I_8$ unpolarized photoluminescence of Ho^{3+} in NaBiMo (solid line) and in LiBiMo (dashed line).

neously considered in the procedure of fitting the observed energy levels. In this manner all intermediate coupling and crystal-field *J*-mixing effects are included in the calculations and arbitrary adjustments of the barycenter of the multiplets have been avoided. The theoretical background, including the expressions of the corresponding FI and CF Hamiltonians, and a short description of the IMAGE routine used to perform the current CF analysis, can be found in Ref. [27.](#page-13-3)

For the S_4 symmetry of Ho³⁺ in *MBiX* hosts the CF potential involves five real B_q^k and one complex S_q^k parameters (after setting S_4^4 to zero by choosing adequate reference axes for the system), which must be adjusted to give the best agreement between the calculated and observed Stark levels. These levels are characterized by the IR of the S_4 point group, monodimensional, Γ_1 , Γ_2 , or bidimensional, $\Gamma_{3,4}$. The selections rules for induced ED or magnetic dipolar (MD) transitions for S_4 symmetry and an even number of electrons are collected in Table [III.](#page-5-2) Since the $4f¹⁰$ configuration does not includes any nondegenerate excited state, the attribution of the IR for the ${}^{5}I_8(0)$ ground level is not straightforward. To proceed with this analysis we have first simulated the sequence of the Ho^{3+} energy levels using the previously reported CF parameters of Er^{3+} in NaBiW²⁶ and FI parameters calculated for Ho^{3+} in monoclinic $KGd(WO_4)_2$ α -KGW.²⁸ In the calculation process, the S_4 $4f^{10}$ states—i.e., the observed energy levels—are distributed in three submatrices, each being a submatrix associated with only one IR. These matrices contain 257, 254, and 245 energy levels, and consequently the last one will correspond to the IR $\Gamma_{3,4}$. This initial simulation showed that ${}^{5}I_8(0)$ belongs to IR $\Gamma_{3,4}$, in agreement with the results found for Ho^{3+} in the closely related $CaWO_4$

FIG. 3. Polarized 5-K optical absorption of Ho³⁺ in NaBiW single crystal. $[Ho]=0.35\times10^{20}$ cm⁻³. The arrows indicate transitions considered as starting in ${}^{5}I_{8}(0)$.

crystal,²⁹ while ${}^{5}I_8(1)$ and ${}^{5}I_8(2)$ correspond to the same IR, either Γ_1 or Γ_2 , and ${}^5I_8(3)$, which appears in the remaining submatrix, to the other IR—that is, Γ_2 or Γ_1 .

Once the IR of the ground level was assessed, we assigned the excited ones as Γ_1 or Γ_2 if the corresponding band appears in the σ (or α) spectrum or as $\Gamma_{3,4}$ when the band appears in the π spectrum. Finally, the Γ_1 and Γ_2 assignment was made on the basis of the best adjustment to the calculated energy levels in the corresponding submatrices, which sometimes only differs in a few cm⁻¹—that is, inside the experimental error of the Ho^{3+} bands. Current fits indicate that for the NaBiW host the observed ${}^{5}I_8(1)$ and ${}^{5}I_8(2)$ levels, at 11 and 29 cm⁻¹, are Γ_2 singlets, which once again agrees with the energy and IR found for Ho^{3+} levels in CaWO₄,^{[30](#page-13-6)} and in the also scheelite-type LiYF₄ crystals,³¹ while ${}^{5}I_{8}(3)$, at 51 cm⁻¹, corresponds to Γ_{1} . The thermal OA

TABLE III. Selection rules for ED and MD transitions for *S*⁴ symmetry and an even number of *f* electrons.

		ED			MD	
	Γ_1	Γ_2	$\Gamma_{3,4}$	Γ_1	Γ_2	$\Gamma_{3,4}$
Γ_1		π	α, σ	σ		σ, π
Γ_2	π		α, σ		σ	σ, π
$\Gamma_{3,4}$	α, σ	α, σ	π	σ, π	σ, π	σ

behavior presented in Fig. [1](#page-2-0) can now be rationalized: Transitions to a given excited Stark level and starting from ${}^{5}I_{8}(1,2,3)$ levels must appear either in a different polarization than that observed when starting from the ${}^{5}I_8(0)$ or become forbidden.

The optical absorption of Ho^{3+} in NaBiW, NaBiMo, and LiBiMo was basically identical for α and σ spectra; therefore, the intensity of the observed transitions is dominated by the induced ED contribution, even for those with $\Delta J=1$, which have non-negligible MD contributions.³² For the present analysis we have assigned each band to the polarization where it is observed with higher intensity, although often a residual intensity is also observed in the other polarization configuration. It must be noted that several centers associated with the two independent 2*b* and 2*d* lattice sites for *T* $=$ Ho³⁺, as well as with different Na (or Li) and Bi environments around both sites, are simultaneously contributing to the experimental spectra. This is the likely reason for the residual contributions described above, and the present energy levels must be understood as corresponding to an average Ho^{3+} center.

The CF calculation was done first for Ho^{3+} in NaBiW because of the larger number of observed energy levels. The total Hamiltonian includes 26 parameters, and among them some FI parameters have been held constant through the adjustment, while others were constrained to vary within determined ratios. The results achieved in NaBiW were taken as the model for the IR and submatrix assignments of $Ho³⁺$ in NaBiMo. The values of the Judd parameters, initially those obtained for NaBiW, after a few cycles of refinement were kept constant along the adjustment process.

In both hosts and despite the modest number of observed Stark levels, the simulation reproduces very adequately the experimental Ho^{3+} -NaBiW and NaBiMo level sequences, with overall agreements of $\sigma = 8.8$ and 7.3 cm⁻¹, respectively, and in no case have large individual discrepancies between experimental and calculated energy levels been found; see Table [II.](#page-3-0) These uncertainties are small considering the precision of the experimental data inherent to the large linewidth. The final results of the refinements are summarized in Table II , for the energy levels, and in Table [IV,](#page-6-0) for adjusted FI and CF parameters. EPAPS MS Word Supplementary Documents with calculated energy levels of the Ho^{3+} configuration in NaBi(WO₄)₂ and in NaBi(MoO₄)₂, indicating the three main components of their corresponding associated wave functions, up to ${}^{3}H(4)_{6}$, ~28 000 cm⁻¹ and 5G_5 , ~24 000 cm⁻¹, respectively, are available.²⁷ Lists of energy levels in these configurations up to 60 000 cm−1 can be also obtained from the authors. The confidence in the obtained phenomenological parameters and the physical meaning of the fits are supported not only by the low σ values obtained, but also by the very similar results of previous calculations performed either independently for the same $4f¹⁰$ Ho³⁺ configuration in the isostructural scheelite CaWO₄ crystal³⁰ or those parallel for the closest $4f¹¹$ Er³⁺ configuration in the same crystalline matrices,²⁶ which provide S_4 CF parameter sets with only smooth variations.

V. Ho3+ RADIATIVE PROCESSES

 $Ho³⁺$ has a large number of electronic transitions which can lead to laser action.¹⁸ The most important ones are those

TABLE IV. Free-ion and CF parameters $\text{ (cm}^{-1}\text{)}$ in S_4 symmetry for Ho^{3+} in $Nabi(XO_4)_2$, $X = W$, Mo, single crystals. Values in parentheses refer to estimate standard deviations in the indicated parameter. Values in square brackets were not allowed to vary in the parameter fitting.

	NaBiW	NaBiMo
E^0	47647.9(9)	47624.2(9)
E^2	6381.6(3)	6390.0(3)
\mathbb{E}^4	31.78(2)	31.92(3)
E^6	637.33(5)	633.26(6)
α	17.37(3)	16.60)
β	$-670(2)$	$-729(1)$
γ	$[1750]$	$[1750]$
ζ	2140.5(5)	2148.9(5)
M^{0} a	3.5(6)	$[3.5]$
P^2 b	750(20)	$[825]$
T^2	$[330]$	$[330]$
T^3	37(2)	$[38]$
T^4	112(2)	$[172]$
T^6	$-197(10)$	$[-127]$
T^7	237(22)	$[336]$
T^8	$[336]$	$[336]$
B_0^2	411(31)	440(30)
B_0^4 B_4^4 B_0^6	$-601(30)$	$-565(34)$
	$\pm 832(18)$	$\pm 800(21)$
	$-67(34)$	$-54(32)$
B_4^6	±576(20)	±509(19)
S_4^6	±195(54)	$\pm 85(42)$
S_2^c	184	197
S_4	440	422
S_6	239	203
S_T	308	293
L	85	56
σ ^d	8.8	7.3
Residue	5289.2	2296.5

 $^{a}M^{0}$ $M^{2}=0.56M^{0}$, $M^{4}=0.32M^{0}$.

 $^{b}P^{4}=0.75P^{2}$, $P^{6}=0.50P^{2}$.

^cThe crystal field strength parameters S_T and S_K are defined (Ref. [33](#page-13-9)) as $S_T = \left[\frac{1}{3} \sum_k S_k^2\right]^{1/2}$, $S_k = \left\{\frac{1}{2k+1} \left[(B_0^k)^2 + 2 \sum_q \left[(B_q^k)^2 + (S_q^k)^2\right]\right]\right\}^{1/2}$.
 ${}^d\sigma = \left[\sum(\Delta_i)^2 / (L-P)\right]^{1/2}$, $\Delta_i = E_o - E_c$, *L* number of levels, *P* number of parameters.

operating at room temperature and related to deexcitations from three different multiplets: ${}^5I_6 \rightarrow {}^5I_7$ ($\lambda = 3 \mu m$), 5I_7 \rightarrow ⁵ I_8 (λ =2 μ m), and ⁵S₂ \rightarrow ⁵ I_7 (λ =760 nm), \rightarrow ⁵ I_6 (λ $= 1.03 \mu m$, $\rightarrow {}^{5}I_{5}$ ($\lambda = 1.4 \mu m$), and $\rightarrow {}^{5}F_{5}$ ($\lambda = 3.37 \mu m$). Moreover, the ${}^5F_5 \rightarrow {}^5I_{5,6,7}$ deexcitations produce laser operation at cryogenic temperatures and the ${}^{5}I_{5} \rightarrow {}^{5}I_{6}$ (λ $=$ 3.9 μ m) has been also reported in selected systems. In this section we study first the expected radiative properties of Ho^{3+} in the considered Bi-based DT and DMo, and later we show the actual properties as a function of the temperature and concentration, with particular emphasis on the abovementioned laser-related multiplets.

FIG. 4. 300-K ground state $\binom{5}{8}$ \rightarrow ^{2S+1}L_J) optical absorption of Ho³⁺ in NaBiW single crystal. [Ho]_{crys}=0.35 \times 10²⁰ cm⁻³.

A. Judd-Ofelt calculations

The Judd-Ofelt (JO) theory^{34,[35](#page-13-11)} is often used to obtain the radiative properties of $4f$ transitions of Ln^{3+} in solid and liquid hosts. The theory uses 300-K integrated absorption cross sections $[Ho]^{-1} \int \alpha \partial \lambda$ to calculate the JO parameters Ω_k $(k=2,4,6)$, from which radiative transition rates $A_{JJ'}$, luminescence branching ratios $\beta_{JJ'}=A_{JJ'}/\Sigma_{J'}A_{JJ'}$, and radiative lifetimes $\tau_{r,J} = 1/\sum_{J'} A_{JJ'}$ for each *J* manifold can be achieved. Details of the JO treatment and definitions can be found in previous works.^{28[,34,](#page-13-10)[35](#page-13-11)}

To perform these calculations we measured the roomtemperature polarized OA of Ho^{3+} in the three considered Bi-based DT and DMo crystal hosts. Figure [4](#page-7-0) shows the results obtained in Ho-doped NaBiW. For Ho-doped NaBiMo and LiBiMo crystals the spectra (not shown for the sake of brevity) were similar to those shown in Fig. [4](#page-7-0) but a lower number of transitions were determined due to the smaller energy gap of the molybdate hosts. Table [V](#page-7-1) summarizes the experimental integrated cross sections, which were weighted as $(2\sigma + \pi)/3$ to obtain the experimental oscillator strengths \overline{f}_o . In this process the refractive index values of each matrix at the corresponding average wavelength were considered[.4](#page-12-5)[,16](#page-12-17)

The JO Ω_k parameter sets obtained for the two molybdate hosts are very similar but slightly different from that obtained for the tungstate host. In particular, the Ω_4 parameter in NaBiW is significantly larger. Ω_2 is mainly determined by the ${}^{5}I_{8} \rightarrow {}^{5}G_{6}$ transition probability which is rather similar in the three hosts. Ω_4 is determined by the ${}^5I_8 \rightarrow {}^5F_5$, 5F_4 , and 5G_6 transition probability; therefore, the Ω_4 difference between NaBiW and Na(Li)BiMo is related to the larger absorption cross section of the 5F_5 and 5F_4 multiplets in the latter host, particularly for the π contribution. All the Ω_k sets obtained are within the usual values found for Ho^{3+} in solids. 32 In particular they agree rather well with those reported for Ho^{3+} in the NaYMo,³⁶ but are clearly different from those reported for Ho:NaYW.³⁷ Our analysis is later validated by a comparison of the calculated spectroscopic magnitudes with the experimental results.

The radiative properties obtained by using the Ω_k set for each host are summarized in Table [VI](#page-8-0) for transitions with fluorescence branching ratios larger than $\approx 20\%$ and, when possible, compared with experimental results. Magnetic-

TABLE V. Room-temperature spectroscopic absorbance properties of Ho³⁺ in NaBiW, NaBiMo, and LiBiMo crystals. σ - and π -polarized integrated ground state $({}^5I_8 \rightarrow {}^{25+1}L_J)$ absorption cross section, $\int \sigma = [H\sigma]^{-1} \int \alpha \partial \lambda$. $\overline{\lambda}$ is the corresponding average wavelength of each ${}^{25+1}L_J$ multiplet. Observed oscillator strength (\bar{f}_o) , calculated electric-dipole oscillator strength (f_{ED}) , $|\Delta f| = |\bar{f}_o - f_{ED}|$, JO parameters Ω_k $(\times 10^{-20}$ cm²).

			NaBiW						NaBiMo						LiBiMo			
$^{5}I_8 \rightarrow$	$\int_0^{\sigma} \sigma$	$\int_0^{\pi} \sigma$	$\bar{\lambda}$	\bar{f}_o	$f_{\rm ED}$	$ \Delta f $	$\int^\sigma \!\!\! \sigma$	$\int_0^{\pi} \sigma$	$\bar{\lambda}$	\bar{f}_o	$f_{\rm ED}$	$ \Delta f $	$\int^{\sigma} \sigma$	$\int_0^{\pi} \sigma$	$\bar{\lambda}$	\overline{f}_o	$f_{\rm ED}$	$ \Delta f $
$^{3}H(4)_{6}$	100	170	361	1066	1494	428												
${}^3K_7+{}^5G_4$	17	18	388	132	135	3												
5G_5	145	66	419	764	209	555												
${}^5G_6 + {}^5F_1$	1279	1390	453	7247	7203	44	1062	1465	453	6589	6589	$\overline{0}$	1286	1674	453	7976	7976	$\overline{0}$
${}^5F_3 + {}^5F_2 + {}^3K_8$	64	114	478	402	259	143	56	73	478	306	210	96	68	89	477	371	271	100
${}^5F_4 + {}^5S_2$	115	261	541	632	497	135	86	130	541	387	307	80	104	158	542	469	383	86
5F_5	111	209	648	478	575	97	90	161	648	304	326	22	109	195	649	368	391	23
$^{5}I_{5}$	16	16	899	22	17	5	5	8	899	13	19	6						
$^{5}I_{6}$	96	142	1170	94	82	12	65	106	1165	66	60	6	105	113	1165	90	77	13
$^{5}I_{7}$	514	568	1960	91	129	38	413	585	1960	71	91	20	539	605	1960	93	116	23
f_{MD}^{a}				$+65.2$						$+67.2$						$+71.2$		
Ω_2	10.5								9.5						10.1			
Ω_4	5.2							2.6						2.7				
Ω_6			0.6					0.4						0.5				
rms			2×10^{-6}					0.7×10^{-6}						0.8×10^{-6}				

^aThe magnetic-dipole f_{MD} contribution has been discounted from the ${}^{5}I_{7}$ experimental oscillator strength to obtain the reported observed \bar{f}_{o} value.

TABLE VI. Radiative properties of Ho³⁺ in NaBiW, NaBiMo, and LiBiMo crystals. Total electric-dipole, ED, plus magnetic-dipole, MD (in parentheses) radiative transitions rates $A_{JJ'}$. Calculated and 300 K observed branching ratios $\beta_{JJ'}$, β_0 (in parentheses). Radiative, τ_r , and 5 K observed τ_o , lifetimes. The observed branching ratios and lifetimes were measured in samples with low (1×10^{19} cm⁻³) Ho concentration.

			NaBiW			NaBiMo			LiBiMo	
	λ $\lceil nm \rceil$	$A_{JJ'}$ $[s^{-1}]$	$\beta_{JJ'}$ (β_0) $[\%]$	$\tau_{\rm r}$ ($\tau_{\rm o}$) $[\mu s]$	$A_{JJ'}$ $[s^{-1}]$	$\beta_{JJ'}$ (β_o) $[\%] % \begin{center} \includegraphics[width=\textwidth]{figures/cdf} \end{center} % \vspace*{-1em} \caption{The figure of the \textit{NCT} model. The \textit{NCT} model is a function of the \textit{NCT} model. The \textit{NCT} model is a function of the \textit{NCT} model.} \label{fig:cdf}$	$\tau_{\rm r}$ $(\tau_{\rm o})$ $[\mu s]$	$A_{JJ'}$ $[s^{-1}]$	$\beta_{JJ'}$ (β_o) $[\%]$	$\tau_{\rm r}$ ($\tau_{\rm o}$) $[\mu s]$
${}^3K_7 \rightarrow {}^5I_7$	477	526	21	403						
$^{5}I_{8}$	385	1379	56							
${}^5G_4 \rightarrow {}^5I_5$	583	35321	52	15						
$5I_7$ $5G_5 \rightarrow 5I_7$	484	14049	21							
	534	34253	69	20	36586	75	20	44212	74	17
${}^5F_1 \rightarrow {}^5I_4$	1115	1310	24	181	734	20	277	852	19	223
$^{5}I_5$	903	2792	50		1656	46		1972	44	
					760	21		1010	23	
$^5G_6 \rightarrow {^5I_8}$	455	156444	94	6	161136	95	6	197868	95	5
	472	2475	91	369	2253	92	408	2806	92	328
	807	2178	39	180	1309	33	250	1562	31	196
${}^{3}K_{8} \rightarrow {}^{5}I_{8}$ ${}^{5}F_{2} \rightarrow {}^{5}I_{6}$ ${}^{5}I_{8}$	478	1818	33		1657	41		2258	44	
$\begin{gathered} ^{5}I_{8}\\ ^{5}F_{3} \rightarrow \begin{matrix} ^{5}I_{7}\\ ^{5}I_{8}\\ ^{5}F_{4} \rightarrow \begin{matrix} ^{5}I_{8}\\ ^{5}S_{8}\\ ^{5}S_{8}\\ \end{matrix} \end{gathered}$	650	5872	56	96	3554	51	142	4244	49	115
	490	2148	21		1960	28		2664	31	
	548	9430	73	77	6448	75	116	7974	76	95
$^{5}S_{2} \rightarrow ^{5}I_{6}$	1010	864	11(14)	381 (16)	183	9(12)	469 (12)	228	20(8)	353 (11)
	746	278	33 (34)		701	33(27)		931	33 (33)	
$5I_7$ $5I_8$ $5F_5 \rightarrow 5I_6$ $5I_7$ $5I_8$	543	1372	52(51)		1173	55 (61)		1578	56 (46)	
	1456	214	3(12)	117(1.9)	148	3	189(2.0)	180	\mathfrak{Z}	157(1.7)
	965	1605	19(28)		1017	19(38)		1210	19(39)	
	650	6699	78 (60)		4120	78 (62)		4952	78 (60)	
$^{5}I_{4} \rightarrow ^{5}I_{6}$ $^{5}I_{7}$	2151	42	39	9084	33	38	11524	42	38	9084
	1227	46	42		36	42		47	43	
$5I_5 \rightarrow 5I_7$ $5I_8$	1657	92	36	3857	73	31	4280	93	32	3465
	905	144	55		140	60		172	60	
$\begin{array}{c} \n \stackrel{5}{\rightarrow} I_6 \rightarrow \stackrel{5}{I_7} \\ \n \stackrel{5}{\rightarrow} I_8 \n \end{array}$	2860	(30) 65	22	3345	(32) 56	24	4225	(34) 63	22	3437
	1174	234	78		180	76		227	78	
$^{5}I_{7} \rightarrow ^{5}I_{8}$	1993	(57) 165	100	6047	(60) 141	100	7089	(77) 178	100	5613

dipole contributions have been explicitly considered when needed. These contributions are particularly relevant for the ${}^{5}I_{6} \rightarrow {}^{5}I_{7}$ ($\lambda = 3$ μ m) and ${}^{5}I_{7} \rightarrow {}^{5}I_{8}$ ($\lambda = 2$ μ m) laser transitions with $\Delta J = 1$. The uncertainty of the radiative JO magnitudes obtained is estimated about $\pm 15\%$ due mainly to the experimental uncertainty in the measurement of the Ho concentration. Within this uncertainty, the Ho^{3+} radiative results obtained for the three hosts must be considered as similar.

B. Ho3+ fluorescence

The Ho^{3+} fluorescence has been studied in several spectral ranges to evaluate the observed branching ratios β _o of corresponding multiplets as a test for JO radiative results. Figure [5](#page-9-0) shows the fluorescence obtained for Ho-doped NaBiW. Again, the results obtained for Ho-doped NaBiMo and LiBiMo were very similar, and therefore they are not shown for brevity. The β results are included in Table [VI.](#page-8-0)

Due to the considerable number of possible electronic transitions in Ho^{3+} and the large energy of the excited multiplets, an overlap of emissions from several multiplets is possible. For instance, in the 1000–1030-nm emission range contributions from ${}^5F_2 \rightarrow {}^5I_5$, ${}^5F_4 \rightarrow {}^5I_6$, and ${}^5S_2 \rightarrow {}^5I_6$ can be found. However, for the ${}^{5}F_{5}$ and ${}^{5}S_{2}$ fluorescence spectra shown in Fig. [5](#page-9-0) these possible contributions are weak and correspond to transitions with small branching ratios; therefore, the observed branching ratios summarized in Table [VI](#page-8-0) are a good approximation. The general agreement between the observed and calculated branching ratios of the 5S_2 and ${}^{5}F_{5}$ multiplets shows the correctness of the radiative calcu-lations in Table [VI.](#page-8-0) In particular, the ${}^5S_2 \rightarrow {}^5I_5$, 5F_5 emissions were not observed, what agrees with the very low branching ratios obtained for these transitions.

FIG. 5. 300-K fluorescence of Ho³⁺ in NaBiW. (a) Unpolarized fluorescence, $\lambda_{\text{exc}} = 454 \text{ nm}$ (${}^5G_6 + {}^5F_1$). (b) Unpolarized fluorescence, $\lambda_{\text{exc}} = 488 \text{ nm}$ (⁵ F_3). (c) π -polarized (dashed line) and σ -polarized (solid line) fluorescences, $\lambda_{\rm exc}$ =488 nm (⁵ F_3).

C. Ho3+ lifetimes

Lifetime measurements of Ln^{3+} in DT and DMo crystals have been most commonly provided in samples prepared for laser demonstration and, therefore, with a large concentration of active ions, typically above 1 at. % in the crystal. These measurements are difficult to compare with the radiative lifetime calculated by the JO analysis because several nonradiative deexcitation mechanisms coexist. In this work, as well as in our previous ones related to Nd, Pr, and Er in the same DT and DMo, we considered first samples with very low Ln^{3+} concentration for which Ln-Ln interactions can be ignored, and later on we studied these Ln-Ln interactions using samples with increasing Ln^{3+} concentration.

The Ho^{3+} lifetime results obtained in the three DT and DMo hosts here considered were similar; therefore, we shall describe the methodology and results for NaBiW and refer to Table [VI](#page-8-0) for the minor differences found for Ho-doped NaBiMo and LiBiMo.

For each host we used first the sample with lowest Ho concentration; see Table [I.](#page-1-0) All fluorescence intensity decays observed at this doping level $(\approx 5-7 \times 10^{18} \text{ cm}^{-3})$ were single exponential. This shows that the influence of the crystalline disorder on the energy levels is not very strong and the average centre assumption used in Sec. IV A is also valid to describe the emission properties. Figure $6(a)$ $6(a)$ shows examples for the Ho-doped NaBiW sample. The multiplet 5G_6 is convenient for excitation of the ${}^{5}S_2$ photoluminescence because of its strong absorption; however, the large radiative lifetime of the intermediate 3K_8 multiplet can *a priori* introduce uncertainty in the measurements. To be sure that our measurements were free of this uncertainty, we compared the fluorescence intensity decays exciting at 5G_6 and 5F_4 multiplets at 300 K and 5 K. The 5S_2 lifetime obtained was independent of the excitation; therefore, we excited at the 5G_6 multiplet, which provided very intense 5S_2 fluorescence signals. This result is also consistent with the low-energy gap

FIG. 6. Lifetimes of 5S_2 and 5F_5 levels of Ho³⁺ in NaBiW. [Ho]=0.07 × 10²⁰ cm^{−3}. (a) Light intensity ${}^5S_2 \rightarrow {}^5I_8$ decays (λ_{exc} $=455$ nm, $\lambda_{\text{emi}} = 550$ nm) at several temperatures (points) and fits to a single-exponential law (lines). (b) Temperature dependence of the ${}^{5}S_{2}$ experimental lifetime (points), fit to a multiphonon model [Eq. (3) (3) (3)] (dashed line) and fit taking into account Eq. (4) (4) (4) (solid line). (c) Temperature dependence of the ${}^{5}F_{5}$ (λ_{exc} =650 nm, λ_{emi} =660 nm) experimental lifetime (points) and fit to the multiphonon model [Eq. (3) (3) (3)] (line).

between ${}^{3}K_{8}$ and ${}^{5}F_{2}$ multiplets (<200 cm⁻¹); therefore, the actual 3K_8 lifetime must be must shorter than the calculated radiative lifetime.

At 5 K the experimental ${}^5S_2 \rightarrow {}^5I_8$ lifetime found in Hodoped NaBiW was $\tau_0=16 \mu s$ (see Table [VI](#page-8-0) for other hosts) and this decreased up to τ_0 =4.[6](#page-9-1) μ s at 300 K. Figure 6(b) shows the thermal dependence. It must be first noted that $\tau_0(10 \text{ K})$ of 5S_2 is much lower than the calculated radiative lifetime τ_r =381 μ s. Ho-Ho energy transfer can be ignored at such low doping concentration because of the large average Ho-Ho distance $\bar{r} = (4\pi[\text{Ho}]/3)^{-1/3} \ge 320$ nm, and since phonon emission is also minimized at 5 K, the difference between the experimental and radiative lifetimes must be ascribed to a Ho-host nonradiative interaction.

The ${}^{5}F_{5}$ lifetime in the different hosts was excited resonantly $({}^5I_8 \leftrightarrow {}^5F_5)$ at $\lambda_{\text{exc}} = 650 \text{ nm}$ and monitored at λ_{emi} =660 nm. Procedures similar to the case above and the same low-concentrated samples were used. The $\text{Ho}^{3+5}F_5$ lifetime at 10 K was $\tau_0 \approx 1.9 \mu s$, also much smaller than the calculated radiative lifetime τ =117 μ s, showing again the strong Ho-host nonradiative interaction.

The decrease of the experimental lifetime with increasing temperature is usually ascribed to nonradiative release of the energy by phonon emission. These nonradiative processes are described by the characteristic nonradiative multiphonon relaxation probability $W_{\text{nr}}^{\text{ph}} = \tau_{\text{ph}}^{-1}$ and energy transfer probability $W_{\text{nr},c} = \tau_{C_1}^{-1}$, which are related to the observed, τ_0^{-1} , and radiative, τ_{r}^{-1} , lifetimes as $\tau_{0}^{-1} = \tau_{r}^{-1} + \tau_{ph}^{-1} + \tau_{C}^{-1}$. The temperature dependence of the multiphonon relaxation rate is given $by³⁸$ $by³⁸$ $by³⁸$

$$
W_{\rm nr}^{\rm ph}(T) = W_{\rm nr}^{\rm ph}(0) \prod_i (1 + n_{\rm eff}^{(i)})^{\rm ph}(i), \tag{1}
$$

where $n_{\text{eff}}^{(i)} = [\exp(\hbar \omega_{\text{eff}}^{(i)}/kT) - 1]^{-1}$, $\bar{h} \omega^{(i)}$ being the energy of the *i* phonon emitted and *k* the Boltzmann constant. To maintain the energy conservation in a nonradiative transition between levels separated by ΔE , the gap energy must be equal to the total thermal energy released—i.e., $\Delta E = \sum_i ph^{(i)}$ $\times \hbar \omega^{(i)}$, where ph^(*i*) is the number of required *i* phonons. In the simplest case phonons of a single energy, usually the largest phonon observed in Raman or infrared absorption spectra, are emitted.

The analysis of the 5S_2 and 5F_5 lifetime thermal dependences with this nonradiative model produced different results. For 5S_2 the multiphonon emission model, given by Eq. ([1](#page-9-2)), does not account for the behavior observed. As example, Fig. [6](#page-9-1)(b) shows (dashed line) the behavior expected for the ⁵ S_2 → ⁵ F_5 ($\Delta E \approx 2830$ cm⁻¹) nonradiative deexcitation by emission of three phonons with $\bar{h}\omega$ =912 cm⁻¹. The assumption of a single phonon type could not reproduce neither the thermal behavior of the ${}^5\overline{F}_5$ lifetime [see dashed line of Fig. $6(c)$ $6(c)$, but in this case the assumption of the emission of two phonons with $\bar{h}\omega$ =912 cm⁻¹ and one of $\bar{h}\omega$ =390 cm⁻¹ (ΔE \approx 2000 cm⁻¹ for the ⁵ $F_5 \rightarrow {}^5I_4$ gap) provides an acceptable fit of the experimental results.

For ${}^{5}S_{2}$ it must be realized that the upper ${}^{5}F_{4}$ multiplet with lower radiative lifetime $(\tau_r=77 \mu s)$ is very close, 100 cm^{-1} (see Table [II](#page-3-0)). The increase of temperature induces a redistribution of the electronic population between 5S_2 and 5F_4 multiplets. In this case the effective radiative probability becomes temperature dependent and it can be conveniently described as³⁹

$$
W_{\rm rad}(T) = \frac{g_1 W_{\rm rad1} \exp(-\Delta E_{1-2}/kT) + g_2 W_{\rm rad2}}{g_1 \exp(-\Delta E_{1-2}/kT) + g_2},\qquad(2)
$$

where $g_1 = 9$ and $g_2 = 5$ are the 5F_4 and 5S_2 degeneracies and $\Delta E_{1-2} \approx 100 \text{ cm}^{-1}$ is the energy gap between these two multiplets. Under this assumption the experimental 5S_2 lifetime temperature dependence can be properly described; see the solid line of Fig. $6(b)$ $6(b)$.

We further studied the concentration dependence of the 5S_2 and 5F_5 room-temperature lifetimes. For the Ho concentration range available, [Ho] $< 0.6 \times 10^{20}$ cm⁻³, all the light intensity decays observed are single exponential and the lifetime of each host remains constant. This indicates that for the Ho concentrations used nonefficient Ho-Ho energy transfer took place.

VI. DISCUSSION

First, the large linewidth of Ho^{3+} transitions in NaBiW, NaBiMo, and LiBiMo must be noted. Figure [7](#page-10-1) shows a comparison of one of the Stark levels of the ${}^{3}I_{8} \rightarrow {}^{5}F_{5}$ multiplet with an equivalent one corresponding to α -KGdW, an ordered double tungstate. The FWHM of the ${}^{3}I_{8} \rightarrow {}^{5}F_{5}$ band in NaBiW, NaBiMo, and LiBiMo is 34.5 cm⁻¹, 28.7 cm⁻¹, and 35.7 cm^{-1} , respectively. These values are about 4 times larger than the 7.7 cm⁻¹ observed in α -KGdW.

The broadening in *M*+Bi*X* crystals must be understood as induced by the coexistence of several Ho environments around each of the two possible lattice sites 2*b* and 2*d*. The detailed structures of these environments or the importance of their contributions to the band shape are not yet well

FIG. 7. Optical absorption line shape of ${}^{5}I_{8} \rightarrow {}^{5}F_{5}$ Ho³⁺ Stark transition at 5 K in ordered monoclinic α -KGdW and in disordered tetragonal $MBi(XO_4)_2$ crystals.

known but the present data show that differences between them cannot be large since, although broadened, a single band is observed at 10 K for most of the transition between Stark levels. The spectroscopic contributions of these sites have been resolved for the ${}^4F_{3/2} \leftrightarrow {}^4I_{11/2}Nd^{3+}$ transitions in NaBiW⁴⁰ and for the ${}^2F_{7/2}(0) \leftrightarrow {}^2F_{5/2}(0')$ transition of Yb³⁺ in NaGdW,¹³ NaLuW,¹⁵ and LiGdMo.²³ Similar site-resolved contributions have not been specifically searched in this work, but some of the band structures observed in Fig. [3—](#page-5-1)for instance, at 25 824 cm⁻¹ $(5G_4)$ —could be related to this fact. The polarization features shown in Figs. [1](#page-2-0) and [3](#page-5-1) indicate that the optical axes of all Ho centers must be close to the lattice c axis. Therefore, within the spectral resolution of our measurements the assumption of an average Ho^{3+} center to discuss the observed spectroscopic characteristics seems a good first approximation although clearly the actual situation is more complex and should include centers with symmetry lower than *S*4.

The nonradiative probability W_{nr} related to the nonradiative interaction with the host is given by the *energy gap law*

$$
W_{\rm nr} = \tau^{-1}(0 \text{ K}) - \tau_{\rm rad}^{-1} = \beta \exp(-\alpha \Delta E_g),\tag{3}
$$

where for a given lanthanide multiplet ΔE_g is the energy difference to the low-lying energy level, and α and β characterize the host. Figure 8 shows a representation of Eq. (3) (3) (3)

FIG. 8. Gap law representation of the nonradiative probability W_{nr} for tetragonal NaBiW (solid symbols), NaBiMo (open symbols), and LiBiMo (crossed symbols). $Ho^{3+} : {}^{5}F_5$, \square ; ${}^{5}S_2$, \square . $\text{Er}^{3+}: {}^{4}S_{3/2}, \triangle; {}^{4}I_{13/2}, \star.$

FIG. 9. Emission cross sections at 300 K of Ho³⁺ in NaBiW. The ground-state absorption cross section σ_{GSA} is indicated with the dashed line. The experimental photoluminescence by points and the calculated emission cross section σ_{emi} by the solid lines. Unpolarized ${}^5S_2 \rightarrow {}^5I_J$ (a)–(c). Unpolarized ${}^5F_5 \rightarrow {}^5I_J$ (d)–(f). Polarized 5I_6 \rightarrow ⁵*I*₈ (g). Polarized ⁵*I*₇ \rightarrow ⁵*I*₈ (f)–(i).

for the Bi-based DT and DMo. For this purpose we have used the Ho results presented in this work as well as previous results achieved for Er.¹⁶ This fit provides $\beta = 3.4 \times 10^7$ s⁻¹ and α =2.2×10⁻³ cm as average values for the three Bibased DT and DMo hosts.

We finally discuss the emission cross sections σ_{emi} of several Ho³⁺ transitions of interest for laser application taking NaBiW for reference. σ_{emi} represents the laser gain per unit of population inversion; therefore, high values are desirable. $\sigma_{\rm emi}$ of transitions to the ground $^{5}I_8$ multiplet can be calculated from the ground-state absorption cross section σ_{GSA} $=\alpha_{\text{GSA}}/[H\text{o}]$ using the the reciprocity principle⁴¹

$$
\sigma_{\text{emi}} = \sigma_{\text{GSA}} \frac{Z_l}{Z_u} e^{(E_{zl} - h\nu)/k_B T},\tag{4}
$$

where Z_u and Z_l are the partition functions of the upper and lower multiplets, respectively, and E_{z} is the energy difference between the lowest Stark levels of both multiplets. For a given transition these latter parameters can be calculated from the energy-level results of Table [II.](#page-3-0) For the transition to

FIG. 10. Gain cross section for the ${}^{5}I_{7} \rightarrow {}^{5}I_{8}$ transition of Ho³⁺ in NaBiW. (a) σ and π configurations. (b) Gain cross section in π configuration for increasing population inversion ratios β .

excited multiplets σ_{emi} can be obtained by the Füchtbauer-Ladenburg method:⁴²

$$
\sigma_{\rm emi} = \sigma_{\rm emi}^{\rm ref} \frac{I\lambda^5}{I_{\rm ref}\lambda_{\rm ref}^5},\tag{5}
$$

where $\sigma_{\text{emi}}^{\text{ref}}$, I_{ref} , and λ_{ref} are taken from the corresponding ${}^{2S+1}L_J \rightarrow {}^{5T}I_8$ transition.

Figure [9](#page-11-1) summarizes the emission cross sections of the laser channels of Ho³⁺ with the exception of ${}^5I_6 \rightarrow {}^5I_7$ (λ \approx 3 μ m) and $^{5}I_5 \rightarrow ^{5}I_6$ ($\lambda \approx$ 3.9 μ m), both expected beyond the spectral response of our equipment. Table [VII](#page-11-2) shows a comparison with other laser crystal hosts taken as reference transitions to the fundamental $\frac{5I_8}{4}$ multiplet. The emission cross sections obtained for Ho^{3+} in NaBiW are similar to those obtained in common oxide and fluoride laser hosts.

For transitions with the ground ${}^{5}I_8$ multiplet as terminal level the laser characteristics are determined by the balance between the radiative and absorption properties. This is ex-

TABLE VII. Room-temperature Ho³⁺ peak emission cross sections $(\times 10^{-20} \text{ cm}^2)$ in several crystal laser hosts. In parentheses the corresponding wavelength (nm).

	$NaBi(WO4)_{2}$	$Y_3Al_5O_{12}^a$	YAlO ₃ ^a	$\text{YVO}_4^{\ b}$	$LiYF_4^c$
${}^5S_2 \rightarrow {}^5I_8$	2.35(545)				π , 2.4 (536)
${}^5F_5 \rightarrow {}^5I_8$	2.23(660)			σ , 8.4 (660)	π , 3.3 (638)
$^{5}I_{6} \rightarrow ^{5}I_{8}$	π , 2.24 (1189)				π , 1.5 (1190)
$5I_7 \rightarrow 5I_8$	π , 1.45 (2047)	1(2090)	1.1(1975)	π , 2.5 (2030)	π , 2.24 (1950)

a Reference [43.](#page-13-19)

b Reference [44.](#page-13-9)

c Reference [45.](#page-13-20)

pressed by the gain cross section σ_{gain} , defined as

$$
\sigma_{\text{gain}} = \beta \sigma_{\text{emi}} - (1 - \beta) \sigma_{\text{GSA}},\tag{6}
$$

where β is the inversion ratio between fundamental and excited populations.

Figure [10](#page-11-3) shows the ${}^5I_7 \rightarrow {}^5I_8$ gain cross sections under different polarization and inversion ratio conditions. The wavelength corresponding to maximum σ_{gain} determines the free running laser wavelength. This occurs at 2053 nm with little difference for both polarizations. Moreover, it shifts to slightly shorter wavelength with increasing inversion ratios.

VII. CONCLUSIONS

Despite differences on the lattice cell parameters between NaBi*X* ($V \approx 320 \text{ Å}^3$) and LiBiMo ($V \approx 312 \text{ Å}^3$), the observed energy levels and radiative properties of Ho^{3+} in the three crystals remain similar each other and also close to those observed for Ho^{3+} in NaYMo ($V \approx 306$ Å³). While broadened, the bands preserve well-defined $S₄$ polarization characteristics according to the uniaxial symmetry of the above DT and DMo crystals. These Ho^{3+} properties are representative of an average center with local *S*⁴ symmetry for the smaller Ln³⁺-doped structurally disordered tetragonal DT and DMo hosts with general formula $MT(XO₄)₂$. The average center

includes the effects induced by the two crystallographic sites for Ho^{3+} and the random Na^{+} and Bi^{+} distributions in the first cationic environment. Energy-level sequences for both NaBiW and NaBiMo have been fit very satisfactorily with rather similar sets of FI and *S*₄ CF parameters. The fits confirm the IR assignments derived from experiments as well as from the S_4 Er³⁺-based initial CF simulation.

Although large nonradiative losses have been found in these DT and DMo due to large phonon energies $(\approx 980 \text{ cm}^{-1})$ associated with WO₄ or MoO₄ vibrations, the emission cross sections of Ho^{3+} in NaBiW are similar to those observed in other crystal laser hosts, and positive gain cross sections can be achieved in extended spectral ranges. These properties make the Ho^{3+} -doped double tungstates and double molybdates to be promising materials for tunable and short-pulse laser operation.

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