

Probing exciton localization in nonpolar GaN/AlN quantum dots by single-dot optical spectroscopy

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We present an optical spectroscopy study of nonpolar GaN/AlN quantum dots by time-resolved photoluminescence and by microphotoluminescence. Isolated quantum dots exhibit sharp emission lines, with linewidths in the 0.5–2 meV range due to spectral diffusion. Such linewidths are narrow enough to probe the inelastic coupling of acoustic phonons to confined carriers as a function of temperature. This study indicates that the carriers are laterally localized on a scale that is much smaller than the quantum dot size. This conclusion is further confirmed by the analysis of the decay time of the luminescence.

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I. INTRODUCTION

III-N heterostructures are usually grown along the polar [0001] axis (c axis), giving rise to a built-in internal field in quantum wells (QWs) or quantum dots¹ (QDs) due to the polarization discontinuity along the c axis at the well-barrier interface. For quantum wells grown along a nonpolar axis, the c axis, being in the plane of the well, does not cross any polarization discontinuity so that such a quantum well is free from the internal electric field.² For III-N quantum dots grown along a nonpolar axis, the situation is not as straightforward as the c axis still crosses a polarization discontinuity at the quantum dot lateral facets. The buildup of an internal field is then very sensitive to the exact dot geometry (shape and strain)³ as well as to possible screening due to residual doping. In this paper, we study GaN in AlN quantum dots grown along the [11–20] axis (a axis). While a strongly reduced quantum confined Stark effect (QCSE) compared to polar QDs of similar sizes was already observed,^{4,5} the optical study presented here aims at probing the lateral extent of the confined wave functions using time-resolved photoluminescence and single QD photoluminescence spectroscopy.

II. SAMPLES AND EXPERIMENTS

A. Samples and optical properties of QD ensembles

The sample used for the present study was grown by plasma assisted molecular-beam epitaxy (MBE) on a (11–20) 6H-SiC substrate provided by INTRINSIC Semiconductor and polished by NOVASiC. It consists of a 30 nm AlN buffer layer followed by a first array of self-organized GaN quantum dots, then a 20 nm AlN cap layer is grown followed by a surface quantum dot array similar to the first one. It was checked on a control sample containing only surface quantum dots that these do not contribute to the photoluminescence (PL) signal. The two-dimensional (2D)–three-dimensional (3D) transition was controlled during the growth by the appearance of 3D reflection high-energy electron diffraction patterns, which happens after the deposition

of 3.7 monolayers of GaN. Details of the growth procedure can be found elsewhere.⁴ The uncapped quantum dots, studied by atomic force microscope (AFM), show a rather high areal density of $1.2 \times 10^{11} \text{ cm}^{-2}$, a typical height of 2 nm, and a diameter of 20 nm.⁴

The samples were mounted in a helium-flow cryostat for time-resolved and time-integrated PL measurements and the temperature could be controlled from 5 K to more than 300 K. We present in Fig. 1 time-resolved and time-integrated photoluminescence measurements of the buried quantum dot plane at 5 K. Experiments were performed using a tripled Ti:sapphire laser with an average power of 3 mW at the wavelength of 250 nm and pulsed at the frequency of 76 MHz. The quantum dot luminescence was analyzed using a 320 mm monochromator and a streak camera with a time resolution of 5 ps.

Compared to the optical properties of the polar GaN quantum dots (0001), the nonpolar ones (11–20) present three main features evidencing a much weaker QCSE, as

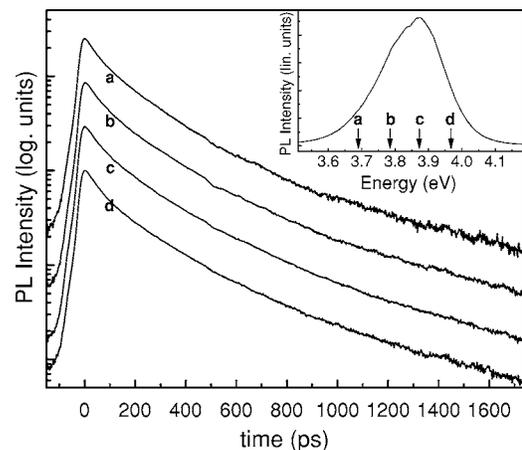


FIG. 1. Photoluminescence decay curves on ensembles of QDs at 5 K for various detection wavelengths (a–d). The decay curves are not monoexponential and do not vary as a function of wavelength.

already observed in previous works.^{4,5} First, the PL intensity presents a maximum around 3.8 eV, and even for large dots, the emission is at a higher energy than the GaN strained band gap. This clearly shows that quantum confinement effects are dominant over any quantum confined Stark effect, in contrast to (0001) QDs of similar dimensions.^{6,7} Second, no blueshift has been reported under high excitation power,⁵ contrary to the polar dots for which the internal field can be screened by the photocreated carriers.^{7,8} Third, previous studies⁴ show that nonpolar GaN quantum dots present PL decay times at low temperature in the 200–500 ps range instead of a few nanoseconds and more for the (0001) dots.^{6,9} Indeed, the presence of an internal electric field for *c*-plane QDs results in a spatial separation of the electron and hole wave functions, thus decreasing the oscillator strength of the transitions.

For the samples studied in this paper, the decay curves do not depend on the emission wavelength (Fig. 1). This is again in strong contrast with polar QDs, for which the quantum confined Stark effect depends on the QD size: there is thus a strong dependence of the decay rate as a function of the emission wavelength.⁹ The decay curves we observe are not monoexponential: the decay on the short time scale is about 70 ps, while the decay on the longer time scale is about 600 ps, the mean decay time being 280 ps. We shall discuss these decay values later in this paper. The decay curves remain stable up to 100 K, so that we can exclude the role of nonradiative processes in the low-temperature recombination dynamics. As a first conclusion, the luminescence decay in the studied *a*-plane samples at low temperature is radiative, with much shorter decay values than those for *c*-plane QDs, the radiative lifetimes being independent of the QD size. The optical characterization of ensembles of *a*-plane QDs and especially the time-resolved experiments do not show any evidence of QCSE.

B. Microphotoluminescence properties

In order to access the optical properties of a single GaN quantum dot, we aim at isolating the luminescence of the smallest possible number of dots by etching mesas¹⁰ or by opening small apertures in an aluminum mask.¹¹ For the mesas, square openings in a resist layer with sizes ranging from 5 μm down to 200 nm were designed by e-beam lithography. After deposition of a 50-nm-thick nickel film and a lift-off, the mesas were formed by SiCl_4 inductively coupled plasma etching. The nickel mask was then removed in a HNO_3 solution. For the aluminum mask, we put a drop of polystyrene nanobeads (diameter=300 nm) dispersed in water on the sample and let it dry. Then, we deposited 100 nm aluminum and subsequently removed the beads to leave apertures in the mask.¹⁰ Both techniques (mesas and aluminum mask) give the same optical spectroscopy results.

As the average density of dots is rather high (5×10^{10} – 1.2×10^{11} cm^{-2}), the smallest holes or mesas contain on average 30–80 dots. However, it is possible to find mesas and apertures containing only a few dots since the areal density of the dots is quite inhomogeneous on an ~ 1 μm length scale for this sample. It is thus possible, by

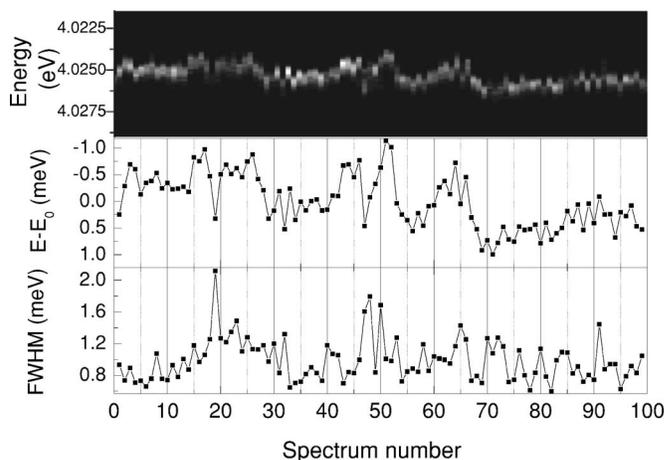


FIG. 2. Time-dependent spectral diffusion. The upper graph displays a single QD photoluminescence spectrum recorded as a function of time (100 consecutive spectra of 1 s integration time each). The lower graphs show the variation of the emission energy compared to its time-averaged value E_0 and the evolution of the full width at half maximum of the line.

selecting the correct apertures or mesas, to study spectrally isolated QDs.

Single QDs are studied by low-temperature microphotoluminescence.¹² The excitation is provided by a cw doubled argon laser line at 244 nm creating electron-hole pairs in the wetting layer. The beam is focused on the sample by an UV-optimized microscope objective (numerical aperture of 0.4), leading to a 1 μm spot. The photoluminescence is collected through the same microscope objective, analyzed by a single grating monochromator (with a resolution of 0.3 meV around 4 eV), and detected by a liquid-nitrogen-cooled charge coupled device camera.

The photoluminescence of the smallest mesas presents isolated sharp lines in the high-energy tail of the quantum dot distribution (>3.8 eV), but for reasons that are still not fully understood, we could not find such sharp structures on the low-energy side of the spectrum. We should mention that these sharp lines display a linear dependence with the excitation laser at low excitation power, so that they can be attributed to fundamental single electron-hole pair transitions of the QDs. In general, the linewidths of the studied isolated peaks are between 0.5 and 2 meV, which is much larger than the lifetime limited linewidth (5 μeV). We attribute this broadening to a Coulomb interaction between the confined exciton and loosely trapped charges moving in the vicinity of the dot, leading to a time-dependent spectral diffusion.^{13–15} All phenomena that occur on a shorter time scale than our accumulation time (typically 1 s) are averaged on the spectra. Even when accumulating several spectra one after the other with an integration time of 1 s, a spectral diffusion of the lines of the order of 1 meV can be observed (Fig. 2). Compared with the study of single polar GaN QDs, such linewidths are 1 order of magnitude smaller than the values reported in the first study by Kako *et al.*¹⁶ and comparable with the values reported recently by Bardoux *et al.*¹⁵ In this later publication, the authors analyzed the spectral diffusion dynamics as a function of excitation power and proposed that

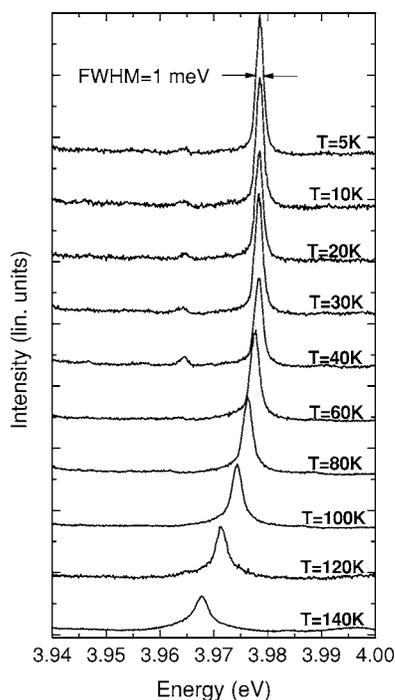


FIG. 3. Temperature-dependent microphotoluminescence on a single QD line.

the spectral diffusion is due to the interaction with loosely trapped charges in the vicinity of the QDs. In this framework, the spectral diffusion depends essentially on the density of such traps, and thus on the sample quality. While one would expect that the internal field enhances the spectral diffusion and thus that the spectral diffusion should be smaller in nonpolar QDs, it is difficult to compare polar and nonpolar QDs as it would require a knowledge of the trap densities and properties in the various samples.

The lines obtained in the nonpolar GaN single quantum dots studied here are narrow enough to be sensitive to temperature-induced broadening mechanisms. In Fig. 3, we present a single-dot spectrum taken at different temperatures from 5 up to 140 K and at low excitation power. The spectra are normalized to the spectrally integrated intensity. Below 60 K, the line shape does not evolve: spectral diffusion is the dominant broadening mechanism. Above 60 K, temperature-induced broadening takes over as the dominant broadening mechanism. This includes the appearance of asymmetric phonon wings (Fig. 4) due to absorption and emission of acoustic phonons in the luminescence process (inelastic phonon scattering). As detailed in Sec. III, the analysis and modeling of these phonon wings allow us to finely probe the confined electronic wave functions.

III. MODELING AND ANALYSIS: LOCALIZATION OF THE EXCITON

A. Exciton-phonon interaction

The observation of phonon wings on single QD spectra when raising the temperature has already been reported for CdTe (Ref. 17), InAs (Refs. 18 and 19), and GaAs (Ref. 20) QDs.

This mechanism was first described and modeled by Besombes *et al.*¹⁷ by extending the Huang-Rhys theory of localized electron-phonon interaction to the case of an exciton localized in a quantum dot. Let us recall the framework of this model that we shall apply to our particular experimental situation. In this model, the dispersion of the acoustic phonons is discretized into N modes, and for each of them the Huang-Rhys theory defines new eigenstates where the exciton and each monochromatic acoustic-phonon mode are in strong coupling: the acoustic polaron states. This nonperturbative coupling creates a discrete set of polaron states which can recombine radiatively but with different probabilities depending on the phonon part of each exciton-acoustic-phonon state. The exciton-phonon interaction Hamiltonian is given by $H_{X-LA-ph} = c^+ c \sum_{\vec{q}} M_{\vec{q}} (b_{\vec{q}}^+ + b_{\vec{q}})$, where c^+ and $b_{\vec{q}}^+$ (c and $b_{\vec{q}}$) are, respectively, the creation (annihilation) operators of the ground-state exciton $|X\rangle$ and of the phonon (of wave vector q and energy $\hbar\omega_q$). To calculate the matrix element $M_{\vec{q}}$, only the deformation potential induced by the longitudinal-acoustic phonon is considered. This approximation is, in general, valid for zinc-blende compounds for which the piezoelectric terms are much smaller than the deformation-potential terms.^{21,22} For bulk III-N compounds, the piezoelectric coupling parameters are much larger so that neglecting these terms is not obvious. While the deformation-potential coupling is mainly sensitive to the overall spatial extent of the confined electron and hole wave functions, the piezoelectric coupling is sensitive to the wave-function difference between the hole and the electron (in the case of perfect local neutrality, the piezoelectric coupling vanishes).^{21,22} In particular, in the case of an electron-hole spatial separation due to a static electric field, the piezoelectric coupling is much enhanced.^{22,23} In the case of the nonpolar QDs discussed in this paper, as there is no effect of the internal electric field in the cw and time-resolved characterization on ensembles of dots, it can be assumed that the permanent dipole of the confined electron-hole pair is small enough so that the piezoelectric coupling to the acoustic phonons can be neglected compared to the deformation-potential coupling. This assumption is justified *a posteriori* by the good fit obtained for the temperature-dependent phonon wings and by the consistency of our proposed global picture of an electron-hole pair exhibiting a strong lateral confinement and a negligible internal electric field effect.

It is also important to remember that the following results are valid as long as the separation between the ground and the excited states is large compared to the acoustic-phonon energies, as no mixing between electronic states is taken into account. $M_{\vec{q}}$ is thus given by $M_{\vec{q}} = \sqrt{\hbar|\vec{q}|/2\rho u_s v} (D_c \langle X | e^{i\vec{q}\cdot\vec{r}_e} | X \rangle - D_v \langle X | e^{i\vec{q}\cdot\vec{r}_h} | X \rangle)$, where ρ is the mass density, u_s the isotropic averaged sound velocity stemming from a Debye approximation for the LA-phonon relation dispersion ($\omega_q = u_s q$), v the quantization volume, and D_c (D_v) the deformation potential of the conduction (valence) band.

As the Bohr radius of the GaN bulk exciton is $a_{\text{GaN}} = 2.8$ nm,²⁴ the exciton located in our quantum dots is still strongly confined along the height (~ 2 nm), but weakly confined in the a -plane (QD diameter ~ 20 nm). As a result,

the exciton state $|X\rangle$ should *a priori* be well described by a quasi-two-dimensional wave function, laterally limited by a Gaussian distribution of the center of mass.

$$\psi_X(r_e, r_h) = \left(\frac{1}{\xi\sqrt{\pi}} e^{-R_p^2/2\xi^2} \right) \left(\sqrt{\frac{2}{\pi a_{2D}^2}} e^{-r/a_{2D}} \right) \times \left[\frac{2}{L} \cos(\pi z_e/L_Z) \cos(\pi z_h/L_Z) \right]. \quad (1)$$

The third part of this wave function is the solution of an infinite barrier quantum well with a height of L_z , which is a good approximation because band offsets between GaN and AlN are large enough for AlN to behave as an infinite barrier. The in-plane electron-hole correlation is described by a 2D hydrogenoid wave function, r being the electron/hole relative position and a_{2D} the 2D Bohr radius of the infinite barrier quantum well. The lateral extension of the center-of-mass position R_p is described by the localization parameter ξ .

With these assumptions, M_q can be calculated, ξ being so far the only unknown parameter of the model. The most relevant parameter describing the exciton-phonon coupling is the Huang-Rhys factor $g(q)$ obtained by integrating over all directions of \mathbf{q} (the coupling constant) defined by $g_{\vec{q}} = |M_{\vec{q}}|^2 / (\hbar\omega_{\vec{q}})^2$. The study of $g(q)$ shows that the coupling is maximum for phonons with $q \sim 1/\xi$ and becomes negligible for $q > 2/\xi$. The maximum value of $g(q)$ increases for more localized wave functions (smaller ξ).

As shown first by Huang and Rhys,²⁵ the probability W_p^q that the optical transition involves p phonons with a wave-vector modulus q can be calculated from $g(q)$. The expression for W_p^q can be found in Ref. 17. p can be positive or negative, corresponding respectively to the emission or the absorption of p phonons. W_0^q is the probability to recombine without phonon emission or absorption (zero-phonon line). To apply the model to LA phonons, we discretize the dispersion relation into $N=50$ modes in the wave-vector regions where the phonons couple efficiently with the confined exciton. It is then possible to calculate the spectral shape function by the combination of every simultaneous emission probability $\prod_{i<j} W_{p_j}^{q_i}$ for every set of p_j . We, however, checked that in the temperature range that we consider here (<200 K), the probability of a transition involving more than three phonons is negligible so that such events need not be taken into account in the calculation. The calculated spectral shape function is a set of δ -like functions, each representing the probability to recombine with a combination of different LA phonons. To finally get an emission spectrum simulation, we convolute the spectral shape function by the zero-phonon line.

As the spectral diffusion is important in our GaN QDs, the zero-phonon line has no reason to be well described by a simple Lorentzian. Indeed, at low temperature for which the spectral diffusion is the dominant broadening process, the line shape is mainly Gaussian with a width $w_g = 1.1$ meV, while at high temperature (>120 K) the line shape is mainly Lorentzian. To fit the zero-phonon line, we thus choose to use a Voigt function with full width at half maximum w_g and w_l for the Gaussian and the Lorentzian part, respectively. w_g

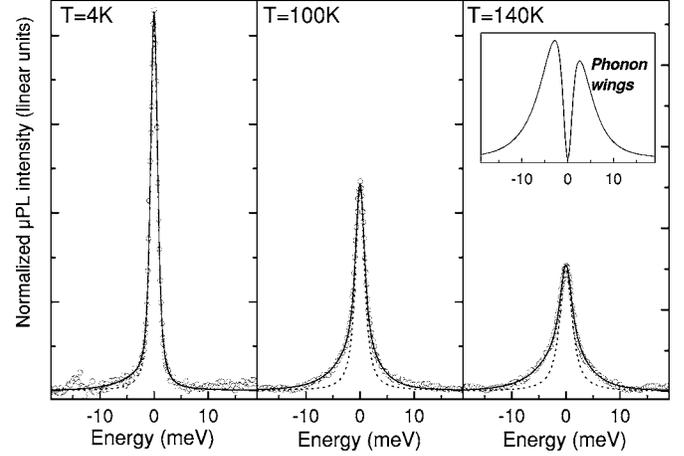


FIG. 4. QD line-shape analysis. The open dots display the same experimental data as Fig. 3. The dotted line is the Voigt function fit of the zero-phonon line, while the solid line is the result of the model described in the text that accounts for acoustic-phonon emission and absorption. The inset in the upper right corner represents the acoustic-phonon emission and absorption to the emission line shape at $T=140$ K.

corresponds to the spectral diffusion, and w_l is related to the additional temperature-dependent broadening mechanisms of the zero-phonon line.

For the calculations, we used the parameter values $D_c = -9$ eV, $D_v = 1$ eV,²⁶ $\rho = 6150$ g cm⁻³, $u_s = 8000$ m s⁻¹, $m_e = 0.2m_0$, and $m_h = m_0$. For a luminescence energy of $E = 3.98$ eV, a simple infinite barrier quantum well model gives an estimated height $L_z = 1.6$ nm, and a value of the two-dimensional Bohr radius $a_{2D} = 2.1$ nm. We set w_g to the low-temperature experimental value of 1.1 meV for every temperature, assuming that the spectral diffusion is constant with temperature. Now, the way to describe the temperature-dependent behavior of the single exciton line with this model consists in finding the correct localization parameter ξ and changing only w_l at each temperature to take the zero-phonon line broadening into account.

By following this procedure, we find a localization parameter $\xi = 2.1$ nm and an increase of w_l from 0.7 meV at 5 K to 2.4 meV at 140 K. The increase of the Voigt function linewidth ($\sqrt{w_l^2 + w_g^2}$) with temperature is larger than expected in a quantum dot, but comparable to already reported results in Ref. 27. Figure 4 presents the experimental points at three different temperatures compared with the calculated spectra. In order to visualize the amplitude and the asymmetry of the phonon sidebands, Fig. 4 also represents the zero-phonon line in a dotted line. An important point is that at high temperature, the phonon wings have a much larger extension than the width of the zero-phonon line so that ξ is the most relevant parameter to fit the phonon wings part of the spectra— w_l and w_g playing here a minor role—while w_l and w_g are mostly important to fit the zero-phonon line. The theoretical curves fit very well the experimental data over the experimentally accessible temperature range. The fits are clearly important for all high-temperature curves (>80 K) for values of ξ that differ from the optimum value of 2.1 nm by more than 0.2 nm.

The probing of the exciton wave function in a -plane QDs by the analysis of the coupling to acoustic phonons thus leads to the determination of the lateral localization parameter $\xi=2.1$ nm, which corresponds to a full width at half maximum of the center-of-mass wave function of 4.9 nm. The exciton is thus localized on a lateral scale which is much smaller than the average QD diameter of 20 nm. Further evidence of this localization and discussion of possible localization mechanisms are presented below.

B. Decay time analysis

Another way of probing the exciton confinement in a “flat” QD is to analyze its radiative decay time. Two extreme confinement regimes are of particular interest. In the “strong confinement regime,” the electron and the hole are localized on a scale that is smaller than the Bohr radius of the QW exciton. In that case, excitonic effects become negligible. The exciton oscillator strength (or equivalently its radiative lifetime) does not depend explicitly on the QD size, but only on the electron-hole wave-function overlap. In the opposite case, for which the exciton is loosely confined laterally (with respect to the scale of the Bohr radius of the QW exciton), the oscillator strength increases proportionally to the coherence volume of the exciton. This regime, which has been studied for flat QDs by Kavokin²⁸ and Andreani *et al.*,²⁹ is known as the “giant oscillator strength regime.” Following the approach developed in Refs. 28 and 29, we find that for the quasi-two-dimensional wave function [Eq. (1)] and for $2\xi > a_{2D}$, the oscillator strength is given by

$$f = \frac{8E_p}{E} \left(\frac{\xi}{a_{2D}} \right)^2, \quad (2)$$

where E is the exciton energy and E_p the interband matrix element (Kane energy). The Kane energy for the “heavy-hole” valence band in wurtzite GaN was calculated by Chuang and Chang to be 15.7 eV.³⁰

In order to discuss the nature of the exciton confinement in our GaN QDs, let us now calculate the oscillator strength and radiative exciton lifetime for three different cases: (i) an exciton wave function that extends laterally over the entire quantum dots, its coherence volume being only limited by the lateral confining potential of the AlN barrier, (ii) an exciton wave function with an additional lateral localization mechanism as deduced from the single-dot phonon coupling experiments, and (iii) a laterally strongly confined exciton for which the confinement effects dominate over the Coulomb effects. Let us first recall at this point that the radiative lifetime τ_r is related to the oscillator strength f by

$$\tau_r = \frac{3m_e c^3 4\pi\epsilon_0}{2ne^2 \omega^2 f}, \quad (3)$$

where m_e is the electron mass and n the refractive index of the medium.

In the first case (i), we expect the QD to be in the giant oscillator strength regime since its lateral size is much larger

than the Bohr radius of the QW exciton. More precisely, the transmission electron microscopy and high-resolution AFM images show QDs with truncated pyramid shapes (rectangular base), with facets orientated at about 30°. Given the strong band-gap difference between AlN and GaN, the center of mass of the exciton can be considered to be laterally limited by an infinite barrier in a cylinder of diameter corresponding to the top dimension of the truncated pyramid. The top diameter D_t of the QD is given by the geometric relation $D_t = D - 2\sqrt{3}h$, where D and h are, respectively, the base diameter and the height of the QD. In this simple model, the localization parameter ξ is given by the Gaussian closest to the sinusoidal shape of the center-of-mass exciton wave function in the infinite barrier potential. This approximation underestimates ξ but leads approximately to $\xi \approx D_t/4$. By applying this formula to the center of the QD distribution, $h=1.6$ nm and $D=19$ nm, we obtain $\xi \approx 3.4$ nm. From Eqs. (2) and (3), we deduce $f \approx 85$ and thus $\tau_r \approx 24$ ps. This estimate of τ_r is well below experimentally measured values although our wave function calculation underestimates ξ and thus overestimates τ_r . This is consistent with the fact that an additional exciton localization mechanism occurs in these QDs.

In case (iii), the oscillator strength tends toward the value $f = E_p/E$ (Ref. 29) for the ideal case of a perfect electron-hole overlap. This gives in our case an oscillator strength of 4 or conversely a decay time value of $\tau_r \approx 520$ ps, which is in the upper decay time values that we experimentally measure for an ensemble of QDs. The actual situation is thus intermediate between the strong confinement regime and the quasi-2D case.

In case (ii), with the localization parameter found in the μ -PL studies ($\xi=2.1$ nm), we deduce from Eqs. (2) and (3) an oscillator strength $f \approx 33$ and a decay time $\tau_r \approx 62$ ps. This value is still a factor of 4 smaller than the measured average decay time, which calls for a comment. Firstly, the Kane energy might be smaller than the value we used, while the authors of Ref. 30 calculate a value of 15.7 eV and the authors of Ref. 26 recommend a similar value of 14 eV, and another study deduced a value of 7.7 eV.³¹ For this lower value of E_p , the measured oscillator strength would still be between the strong confinement limit and the quasi-2D framework with $\xi=2.1$ nm, but less toward the strong confinement limit. Secondly, the strong enhancement factor that appears in formula (2) is due to the excitonic nature of the quasi-2D wave function. It is clear that when ξ is comparable to the 2D Bohr radius, the wave function cannot anymore be factorized into an excitonic part and a center-of-mass motion, so that the excitonic enhancement of the oscillator strength is reduced.

At this point, we recall that in the deformation-potential coupling of acoustic phonons to the localized carriers, the excitonic nature of the electron-hole pair does not play an important role. Through the deformation potential, the acoustic phonons are sensitive mainly to the overall extent of the localized electron and hole wave functions so that the subtleties of the interplay between Coulomb and confinement energies do not affect the model much.

Finally, let us recall that the measured decay curve for an ensemble of QDs at a given emission wavelength is not mo-

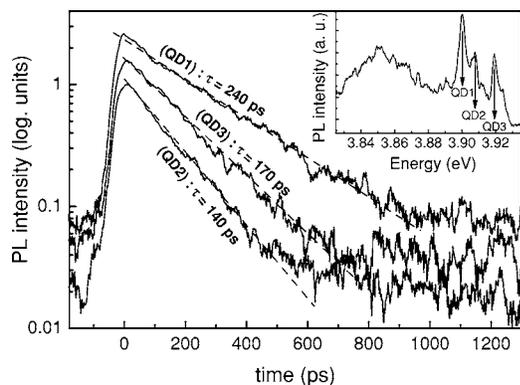


FIG. 5. Photoluminescence decay curves for three single QDs. The dashed lines are single exponential fits. The decays are clearly monoexponential and the decay times are not correlated to the emission wavelengths. The inset displays the time-integrated spectrum as obtained on the streak camera.

noexponential and is better described by a distribution of decay times ranging from 70 to 600 ps. We then probed the photoluminescence decay on single QDs (Fig. 5) by adapting the microphotoluminescence setup on the time-resolved experiment. These experiments are difficult to perform due to the low detection sensitivity of the streak camera. Nonetheless, sharp peaks could be isolated and studied as a function of time. Each single peak decays monoexponentially, but the decay times vary from peak to peak, which confirms that each peak can be attributed to the emission of a single QD. The decay time for each QD is not correlated to its emission wavelength. These observations suggest that excitons are further randomly localized by an additional potential in each individual QD. Thus, the QD exciton lifetime is not governed by the lateral extension of the QD, but by the lateral extension of its wave function resulting from this additional localization mechanism.

Let us sum up the key results and conclusion so far exposed in this paper. (i) In microphotoluminescence, the study of the coupling of a confined electron-hole pair to acoustic phonons is consistent with a quasi-two-dimensional wave function with a localization parameter $\xi=2.1$ nm. This could, however, only be shown for the smallest QDs emitting in the high-energy part of the QD emission spectrum. (ii) The analysis of the oscillator strength confirms that for all dot sizes, the wave function is more laterally localized than expected from the sole AlN barrier lateral confinement.

We shall now compare these conclusions with the results obtained on other III-N heterostructures. For instance, *a*-plane GaN/AlGaIn quantum wells were studied in Ref. 32, in particular, by using time-resolved spectroscopy. The authors of Ref. 32 discuss the variation of decay time as a function of QW width but do not discuss the absolute value of the decay time. Indeed, for the thinnest QW in Ref. 32 that has a height comparable to our QD samples, the decay time at low temperature is similar (within the experimental accuracy) to the one we find for QDs. This corresponds to a long decay time for a QW emission, which would be consistent

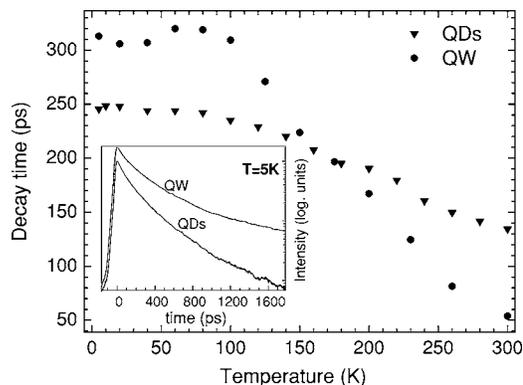


FIG. 6. Evolution of the decay time for *a*-plane GaN/AlN QDs and an *a*-plane GaN/AlN QW (1.5 nm thick) as a function of temperature. The inset displays the decay curves at 5 K.

with a lateral localization on dimensions comparable to the Bohr radius. We also note that the lifetime measured in cubic phase GaN/AlN QDs, for which there is no internal field, is about 200 ps (Ref. 6) and is thus similar to the decay times measured here. Again, the giant oscillator strength regime is not reached for these cubic QDs although they are quite large (3 nm high, 30 nm wide) so that an additional localization should occur. We also studied a 1.5-nm-thick GaN/AlN QW grown along the *a* axis. The decay curve for the photoluminescence at 5 K is presented in the inset of Fig. 6 and compared to the decay curve for *a*-plane QDs. The decay curve for the QW is not monoexponential. Still, an average decay time can be extracted and is reported as a function of temperature in Fig. 6. The first striking feature is that the photoluminescence decay curves do not change between 5 and 100 K, which indicates that the excitons are laterally localized. The low-temperature decay time (300 ps) corresponds to a small oscillator strength for a QW emission, again leading to the conclusion that the QW exciton has a small coherence surface and that a strong lateral localization occurs. We can deduce that in our *a*-plane samples, at low temperature the excitons are laterally localized in QWs as in QDs. This is further confirmed by the fact that the decay curves remain the same between 5 and 100 K for both the QW and QD emissions. Above 100 K, delocalization probably occurs, leading to a stronger non radiative decay probability for the QW than for the QDs.

IV. CONCLUSION

All these results are consistent with the hypothesis that a strong lateral localization comparable with the two-dimensional Bohr radius occurs in GaN/AlN heterostructures for state of the art MBE growth. Such localization could, for instance, be induced by interface fluctuations. Localization effects were already pointed out, for instance, in shallow polar GaN/AlGaIn quantum wells;³³ however, in that case alloy fluctuations in the ternary barrier alloy are likely to be responsible for localizing potentials. If this

localizing effect also occurs in polar GaN/AlN heterostructures, then for the larger QDs the electrons and holes localized at the top and bottom of the QDs by the internal electric field might be laterally separated as well. This effect might have an important role on the very low oscillator strengths reported for large polar GaN/AlN QDs.⁹

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