

Angle-dependent magnetotransport in cubic and tetragonal ferromagnets: Application to (001)- and (113)A-oriented (Ga,Mn)As

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General expressions for the longitudinal and transverse resistivities of single-crystalline cubic and tetragonal ferromagnets are derived from a series expansion of the resistivity tensor with respect to the magnetization orientation. They are applied to strained (Ga,Mn)As films, grown on (001)- and (113)A-oriented GaAs substrates, where the resistivities are theoretically and experimentally studied for magnetic fields rotated within various planes parallel and perpendicular to the sample surface. We are able to model the measured angular dependences of the resistivities within the framework of a single ferromagnetic domain, calculating the field-dependent orientation of the magnetization by numerically minimizing the free-enthalpy density. Angle-dependent magnetotransport measurements are shown to be a powerful tool for probing magnetic anisotropy. The anisotropy parameters of the (Ga,Mn)As films inferred from the magnetotransport measurements agree with those obtained by ferromagnetic resonance measurements within a factor of 2.

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I. INTRODUCTION

Realization of ferromagnetism in III-V semiconductors by introducing high concentrations of magnetic elements has motivated intense research on the dilute magnetic semiconductor (Ga,Mn)As. This system is considered a potential candidate for spintronic applications due to its compatibility with conventional semiconductor technology.^{1,2} In (Ga,Mn)As, magnetic Mn acceptors are predominantly incorporated on cation sites as Mn^{2+} ions having a total spin of $S=5/2$. (Ga,Mn)As is paramagnetic at room temperature and undergoes a transition to the ferromagnetic phase at the Curie temperature T_C , where maximum values of up to ~ 170 K have been reported so far.² The ferromagnetism has been successfully explained within the Zener mean-field model by an indirect Mn-Mn exchange interaction mediated by delocalized holes.³ (Ga,Mn)As is grown by low-temperature molecular-beam epitaxy and, if necessary, subjected to postgrowth annealing to reduce the density of compensating defects. Considerable progress has been made in understanding its structural, electronic, and magnetic properties. In particular, anisotropic magnetoresistance (AMR),⁴⁻⁹ planar Hall effect (PHE),¹⁰ and magnetic anisotropy (MA),¹¹⁻¹⁷ have been identified as characteristic features, making (Ga,Mn)As potentially suitable for field-sensitive devices and nonvolatile memories. These properties have been shown to be governed by several parameters, such as Mn concentration, hole density, strain, or temperature. Most of the work carried out on AMR in (Ga,Mn)As, however, has been restricted to special cases where the magnetic field was applied parallel or perpendicular to the layer and equations for the angular dependence of the longitudinal and transverse resistivities, describing the AMR and the PHE,

respectively, have been given only for in-plane configuration and polycrystalline films. A comprehensive theoretical model describing the resistivities as a function of arbitrary field orientation is still missing. Moreover, (Ga,Mn)As layers are usually grown on GaAs(001) substrates and only little is known about the magnetic properties of films grown on high-index substrates.¹⁶⁻²¹

In this work, the longitudinal and transverse resistivities of (Ga,Mn)As layers, grown on (001)- and (113)A-oriented GaAs substrates, are studied for arbitrarily orientated magnetic fields. The anisotropy of the resistivities and the MA are experimentally probed by rotating the magnetic field \mathbf{H} at fixed field strengths within different planes parallel and perpendicular to the sample surface. General expressions for the resistivities, holding for single-crystalline cubic and tetragonal ferromagnets, are derived from a series expansion of the resistivity tensor with respect to the direction cosines of the magnetization \mathbf{M} . The measured data are well modeled by applying the expressions to the given experimental configurations, assuming the (Ga,Mn)As films to consist of a single ferromagnetic domain. Analytical expressions, widely used in the literature to describe the angular dependence of AMR and PHE, are shown to be inappropriate to single-crystalline materials. Finally, anisotropy parameters are estimated from the low-field magnetotransport data and compared with those obtained from ferromagnetic resonance (FMR) spectroscopy.

II. EXPERIMENTAL DETAILS

40-nm-thick (001) and (113)A (Ga,Mn)As films with Mn concentrations of $\sim 5\%$ were simultaneously grown by low-temperature molecular-beam epitaxy (MBE) in a RIBER

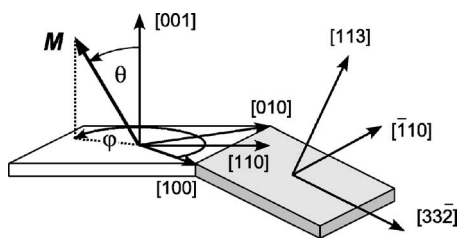


FIG. 1. Orientation of the (001) and (113)A samples with respect to the crystallographic axes.

32 MBE machine on semi-insulating GaAs(001) and GaAs(113)A substrates mounted together on the same Mo holder. A conventional Knudsen cell and a hot-lip effusion cell were used to provide the Ga and Mn fluxes, respectively. A valved arsenic cracker cell was operated in the noncracking mode to supply As_4 with a maximum V to III flux ratio of about 5. First, a 30-nm-thick GaAs buffer layer was grown at a temperature of $T_s \approx 580^\circ\text{C}$ (conventional growth temperature for GaAs), then the growth was interrupted and T_s was lowered to $\sim 250^\circ\text{C}$. The Mn concentrations in the (Ga,Mn)As films were determined by flux measurements.

For the magnetotransport and FMR studies, the (001) and (113)A samples were cleaved into small rectangular pieces with edges along $[110]$ and $[\bar{1}10]$, and along $[33\bar{2}]$ and $[\bar{1}10]$, respectively. The $[110]$ and $[\bar{1}10]$ directions of the (001) sample were identified by selective wet chemical etching and the orientation of the $[001]$ crystal axis in the (113)A sample was determined by x-ray diffraction. The crystallographic orientations of the samples are shown in Fig. 1. Hall bars with Ti-AuPt-Au contacts were prepared on several pieces of the cleaved (001) and (113)A samples with the current directions along $[110]$ and $[33\bar{2}]$, respectively. The width of the Hall bars was 0.3 mm and the longitudinal voltage probes were separated by 1 mm. Hole densities of $3 \times 10^{20} \text{ cm}^{-3}$ for the (001) sample and $1.2 \times 10^{20} \text{ cm}^{-3}$ for the (113)A sample were determined by means of high-field magnetotransport measurements (up to 14.5 T) at 4.2 K using an Oxford SMD 10/15/9 VS liquid helium cryostat with superconducting coils. Least-squares fits were performed to separate the contributions of the normal and anomalous Hall effects. Curie temperatures of $T_C \approx 65$ and 54 K, respectively, were estimated from the peak positions of the temperature-dependent sheet resistivities at 10 mT. For the angle-dependent magnetotransport measurements carried out at 4.2 K, the Hall bars were mounted on the sample holder of a liquid-He-bath cryostat, which was positioned between the poles of a LakeShore electromagnet system providing a maximum field strength of 0.7 T. The sample holder possesses two perpendicular axes of rotation, allowing for an arbitrary alignment of the Hall bars with respect to the applied magnetic field \mathbf{H} . Using a dc current density of $8 \times 10^2 \text{ A cm}^{-2}$, the longitudinal and transverse resistivities ρ_{long} and ρ_{trans} were measured at fixed magnitudes $\mu_0 H = 0.1, 0.25, \text{ and } 0.7 \text{ T}$ of \mathbf{H} while rotating its orientation. Prior to each angular scan, the magnetization \mathbf{M} was put into a clearly defined initial state by raising the field to 0.7 T where \mathbf{M} is nearly saturated and aligned with the external

field. The field was then lowered to one of the above mentioned magnitudes and the scan was started.

The FMR measurements were carried out at 5 K in a commercial Bruker ESP 300 electron paramagnetic resonance spectrometer operated at a fixed frequency of $\omega_{\text{HF}}/2\pi \approx 9.3 \text{ GHz}$ (X band). The spectrometer consists of a microwave bridge for the high-frequency radiation and an electromagnet providing a variable dc magnetic induction up to 1 T. To increase the sensitivity, lock-in techniques were used in which the dc induction is superimposed by a 100 kHz modulation field of 3.2 mT.

III. THEORETICAL OVERVIEW

In our theoretical considerations the total magnetic moment arising from the Mn ion-hole spin complex is treated within the framework of the Stoner-Wohlfarth model,²² i.e., for temperatures below T_C the whole (Ga,Mn)As layer is assumed to consist of a single homogeneous ferromagnetic domain. This simple model has been astoundingly successful in describing a large variety of magnetization-related phenomena in (Ga,Mn)As. Under the given experimental conditions described above, domain nucleation and expansion, which have been shown to accompany in-plane and perpendicular magnetization-reversal processes,^{11,13} are expected to play only a minor role. Accordingly, we may write the magnetization as a vector $\mathbf{M} = M\mathbf{m}$ where M denotes its magnitude and the unit vector \mathbf{m} its direction. In terms of the polar and azimuth angles θ and φ , respectively, which are defined in Fig. 1, the components of \mathbf{m} read as $m_x = \sin \theta \cos \varphi$, $m_y = \sin \theta \sin \varphi$, and $m_z = \cos \theta$. The equations used in the discussion of the angle-dependent magnetotransport data can be written in a concise way by introducing the unit vectors \mathbf{j} , \mathbf{n} , and \mathbf{t} , which specify the current direction, the surface normal, and an in-plane vector defined by $\mathbf{t} = \mathbf{n} \times \mathbf{j}$, respectively. Throughout this work, all vector components refer to the cubic coordinate system with the $[100]$, $[010]$, and $[001]$ directions of the crystal denoted by x , y , and z , respectively.

A. Longitudinal and transverse resistivities

In standard magnetotransport measurements the longitudinal and transverse voltages, measured along and across the current direction, arise from the components $E_{\text{long}} = \mathbf{j} \cdot \mathbf{E}$ and $E_{\text{trans}} = \mathbf{t} \cdot \mathbf{E}$ of the electric field \mathbf{E} , respectively. Starting from Ohm's law $\mathbf{E} = \bar{\rho} \cdot \mathbf{J}$, where $\bar{\rho}$ represents the resistivity tensor and $\mathbf{J} = J\mathbf{j}$ the current density, the corresponding longitudinal resistivity ρ_{long} (sheet resistivity) and transverse resistivity ρ_{trans} (Hall resistivity) can be written as

$$\rho_{\text{long}} = \frac{E_{\text{long}}}{J} = \mathbf{j} \cdot \bar{\rho} \cdot \mathbf{j},$$

$$\rho_{\text{trans}} = \frac{E_{\text{trans}}}{J} = \mathbf{t} \cdot \bar{\rho} \cdot \mathbf{j}. \quad (1)$$

In (Ga,Mn)As, as in many other ferromagnets, the resistivity tensor sensitively depends on the orientation of \mathbf{M} with respect to the crystallographic axes.²³ Thus, in order to quan-

tatively model the measured resistivities in the general case of an arbitrarily oriented magnetization, a universal mathematical relationship between ρ_{long} and ρ_{trans} and the direction cosines m_i of \mathbf{M} has to be derived. For these purposes, we follow the ansatz of Birss²³ and Muduli *et al.*²⁴ and write the resistivity tensor $\bar{\rho}$ as a series expansion in powers of m_i using the Einstein summation convention:

$$\rho_{ij} = a_{ij} + a_{kij}m_k + a_{klj}m_k m_l + \dots \quad (2)$$

For cubic symmetry T_d , most of the components a_{ij}, a_{kij}, \dots of the galvanomagnetic tensors vanish and, considering terms up to the second order, we obtain

$$\begin{aligned} \bar{\rho}^{\text{cubic}} = & A \begin{pmatrix} 1 & 0 & 0 \\ 0 & 1 & 0 \\ 0 & 0 & 1 \end{pmatrix} + B \begin{pmatrix} m_x^2 & 0 & 0 \\ 0 & m_y^2 & 0 \\ 0 & 0 & m_z^2 \end{pmatrix} \\ & + C \begin{pmatrix} 0 & m_x m_y & m_x m_z \\ m_x m_y & 0 & m_y m_z \\ m_x m_z & m_y m_z & 0 \end{pmatrix} \\ & + D \begin{pmatrix} 0 & m_z & -m_y \\ -m_z & 0 & m_x \\ m_y & -m_x & 0 \end{pmatrix}, \quad (3) \end{aligned}$$

with the resistivity parameters

$$\begin{aligned} A &= a_{11} + a_{1122}, & B &= a_{1111} - a_{1122}, \\ C &= a_{2323}, & D &= a_{123}. \end{aligned} \quad (4)$$

Insertion of Eq. (3) into Eqs. (1) and elementary vector algebra yields the general expressions

$$\begin{aligned} \rho_{\text{long}}^{\text{cubic}} &= A + C(\mathbf{j} \cdot \mathbf{m})^2 + (B - C) \sum_i j_i^2 m_i^2, \\ \rho_{\text{trans}}^{\text{cubic}} &= C(\mathbf{j} \cdot \mathbf{m})(\mathbf{t} \cdot \mathbf{m}) + (B - C) \sum_i t_i j_i m_i^2 - D(\mathbf{n} \cdot \mathbf{m}), \end{aligned} \quad (5)$$

which apply to single-crystalline ferromagnetic materials of cubic symmetry. The transverse resistivity in Eqs. (5) includes the contribution of the anomalous Hall effect, which correlates with the perpendicular component of \mathbf{M} , but it does not account for the ordinary Hall effect. For magnetic field strengths $\mu_0 H < 1$ T and hole concentrations $p > 10^{20} \text{ cm}^{-3}$ as in our experiments, however, the maximum contribution of the ordinary Hall effect is $\mu_0 H / e p \approx 6 \times 10^{-6} \Omega \text{ cm}$ (e denotes the elementary charge), and thus about two orders of magnitude smaller than the measured peak values of $\rho_{\text{trans}}^{\text{cubic}}$ (see Sec. IV).

In the following, Eqs. (5) are applied to three different experimental configurations using the relation

$$m_i = j_i(\mathbf{j} \cdot \mathbf{m}) + t_i(\mathbf{t} \cdot \mathbf{m}) + n_i(\mathbf{n} \cdot \mathbf{m}). \quad (6)$$

In the simplest case of a sample with (001) surface, where the current flows along the [100] or [010] direction, Eqs. (5) reduce to

$$\rho_{\text{long}}^{\text{cubic}} = A + B(\mathbf{j} \cdot \mathbf{m})^2,$$

$$\rho_{\text{trans}}^{\text{cubic}} = C(\mathbf{j} \cdot \mathbf{m})(\mathbf{t} \cdot \mathbf{m}) - D(\mathbf{n} \cdot \mathbf{m}). \quad (7)$$

The magnetotransport studies presented in this work were performed on (001)- and (113)A-oriented samples with the current directions \mathbf{j} along [110] and $[33\bar{2}]$, respectively. The corresponding resistivities are as follows.

(001) surface and $\mathbf{j} \parallel [110]$

$$\rho_{\text{long}}^{\text{cubic}} = A + \frac{1}{2}(B - C) + C(\mathbf{j} \cdot \mathbf{m})^2 + \frac{1}{2}(C - B)(\mathbf{n} \cdot \mathbf{m})^2,$$

$$\rho_{\text{trans}}^{\text{cubic}} = B(\mathbf{j} \cdot \mathbf{m})(\mathbf{t} \cdot \mathbf{m}) - D(\mathbf{n} \cdot \mathbf{m}). \quad (8)$$

(113)A surface and $\mathbf{j} \parallel [33\bar{2}]$

$$\begin{aligned} \rho_{\text{long}}^{\text{cubic}} &= A + \frac{9}{22}(B - C) + \frac{1}{121}(126C - 5B)(\mathbf{j} \cdot \mathbf{m})^2 \\ &+ \frac{45}{242}(C - B)(\mathbf{n} \cdot \mathbf{m})^2 + \frac{15\sqrt{2}}{121}(B - C)(\mathbf{j} \cdot \mathbf{m})(\mathbf{n} \cdot \mathbf{m}), \end{aligned}$$

$$\begin{aligned} \rho_{\text{trans}}^{\text{cubic}} &= \frac{1}{11}(9B + 2C)(\mathbf{j} \cdot \mathbf{m})(\mathbf{t} \cdot \mathbf{m}) + \frac{3\sqrt{2}}{11}(B - C)(\mathbf{t} \cdot \mathbf{m})(\mathbf{n} \cdot \mathbf{m}) \\ &- D(\mathbf{n} \cdot \mathbf{m}). \end{aligned} \quad (9)$$

The $(\mathbf{j} \cdot \mathbf{m})^2$ terms of $\rho_{\text{long}}^{\text{cubic}}$ give rise to a dependence of the sheet resistivity on the relative orientation between magnetization \mathbf{M} and current density \mathbf{J} , commonly referred to as AMR. Microscopically, it is explained by a strong spin-orbit coupling in the semiconductor valence band. Experimentally observed differences in the in-plane and out-of-plane AMR (\mathcal{A}), often defined as

$$\begin{aligned} \mathcal{A}_{\text{ip}} &= \frac{\rho_{\text{long}}(\mathbf{m} \parallel \mathbf{j}) - \rho_{\text{long}}(\mathbf{m} \parallel \mathbf{t})}{\rho_{\text{long}}(\mathbf{m} \parallel \mathbf{j})}, \\ \mathcal{A}_{\text{op}} &= \frac{\rho_{\text{long}}(\mathbf{m} \parallel \mathbf{j}) - \rho_{\text{long}}(\mathbf{m} \parallel \mathbf{n})}{\rho_{\text{long}}(\mathbf{m} \parallel \mathbf{j})}, \end{aligned} \quad (10)$$

have been ascribed to biaxial strain in the layer.⁵⁻⁸ According to Eqs. (8) and (9), however, such differences may be expected even in the case of perfect cubic symmetry due to the $(\mathbf{n} \cdot \mathbf{m})^2$ terms of $\rho_{\text{long}}^{\text{cubic}}$. As will be shown below, a strain-induced tetragonal distortion leads to further $(\mathbf{n} \cdot \mathbf{m})^2$ terms, additionally affecting the difference between \mathcal{A}_{ip} and \mathcal{A}_{op} . The PHE, represented by the $(\mathbf{j} \cdot \mathbf{m})(\mathbf{t} \cdot \mathbf{m})$ terms of ρ_{trans} , is closely related to the AMR and describes the appearance of a transverse voltage in the presence of an in-plane magnetic field, or, more precisely, of an in-plane magnetization. From the summation terms in Eqs. (5) it becomes clear that ρ_{long} and ρ_{trans} , and thus the AMR and PHE, depend not only on the relative orientation between \mathbf{m} and \mathbf{j} , but also on the orientations of \mathbf{m} , \mathbf{j} , and \mathbf{t} with respect to the crystal axes.

So far, quantitative studies on the angular dependences of the AMR and the PHE in (Ga,Mn)As were restricted to in-plane configurations with $\mathbf{n} \cdot \mathbf{m} = 0$ and the discussions were based on the well-known expressions^{26,27}

$$\begin{aligned}\rho_{\text{long}} &= \rho_{\perp} + (\rho_{\parallel} - \rho_{\perp}) \cos^2 \phi_j, \\ \rho_{\text{trans}} &= (\rho_{\parallel} - \rho_{\perp}) \sin \phi_j \cos \phi_j,\end{aligned}\quad (11)$$

with ϕ_j denoting the angle between \mathbf{j} and \mathbf{M} . These expressions, however, hold only for polycrystalline films, whereas (Ga,Mn)As layers are normally of high crystalline quality with a uniform crystallographic orientation of the layer. Accordingly, the expressions given in Eqs. (7)–(9) are incompatible with Eqs. (11) and cannot be brought into agreement by simply setting $\mathbf{n} \cdot \mathbf{m} = 0$. In fact, Eqs. (11) result from Eqs. (5) by averaging the summation terms over all possible crystal orientations in space²⁵ with \mathbf{M} lying in the plane spanned by \mathbf{j} and \mathbf{t} :

$$\begin{aligned}\sum_i \overline{j_i^2 m_i^2} &= \frac{1}{5} [2(\mathbf{j} \cdot \mathbf{m})^2 + 1], \\ \sum_i \overline{t_i j_i m_i^2} &= \frac{2}{5} (\mathbf{j} \cdot \mathbf{m})(\mathbf{t} \cdot \mathbf{m}).\end{aligned}\quad (12)$$

Inserting the averaged terms into Eqs. (5) and using the relations $\mathbf{j} \cdot \mathbf{m} = \cos \phi_j$ and $\mathbf{t} \cdot \mathbf{m} = \sin \phi_j$ (the latter holds only for $\mathbf{n} \cdot \mathbf{m} = 0$), we obtain

$$\begin{aligned}\rho_{\text{long}}^{\text{poly}} &= A + \frac{1}{5}(B - C) + \frac{1}{5}(2B + 3C)\cos^2 \phi_j, \\ \rho_{\text{trans}}^{\text{poly}} &= \frac{1}{5}(2B + 3C)\sin \phi_j \cos \phi_j.\end{aligned}\quad (13)$$

These equations are formally identical to Eqs. (11) and apply to polycrystalline materials. A comparison between Eqs. (11) and Eqs. (13) allows us to connect the quantities ρ_{\parallel} and ρ_{\perp} to the components of the galvanomagnetic tensors for cubic symmetry up to second order:

$$\begin{aligned}\rho_{\parallel} &= A + \frac{1}{5}(3B + 2C) = a_{11} + \frac{1}{5}(3a_{1111} + 2a_{1122} + 2a_{2323}), \\ \rho_{\perp} &= A + \frac{1}{5}(B - C) = a_{11} + \frac{1}{5}(a_{1111} + 4a_{1122} - a_{2323}).\end{aligned}\quad (14)$$

Thus, in general Eqs. (11) are not appropriate to describe the in-plane AMR and the PHE in single-crystalline (Ga,Mn)As layers. Only in the limiting case where $B = C$ do Eqs. (7)–(9) simplify to Eqs. (11).

Analyzing the angle-dependent magnetotransport data presented in Sec. IV, it turns out that additional terms proportional to $(\mathbf{n} \cdot \mathbf{m})^2$ have to be introduced in the expressions of $\rho_{\text{long}}^{\text{cubic}}$ to achieve a satisfactory description of the experimental results.¹⁷ They are supposed to originate from a distortion of the crystal lattice due to compressive strain in the (Ga,Mn)As layer. To account for such strain-induced effects in a correct way, we extend our model to a tetragonal distortion of the cubic lattice along the [001] direction. As will be

shown below, our FMR and magnetotransport data suggest that this applies not only for the (001)- but also for the (113)A-oriented sample.

In the case of a tetragonal lattice distortion along [001], the symmetry reduces to D_{2d} and the series expansion of $\bar{\rho}$ in Eq. (2) yields further contributions which can be subsumed into an additional term $\Delta\bar{\rho}$. The resistivity tensor then reads as

$$\bar{\rho}^{\text{tetra}} = \bar{\rho}^{\text{cubic}} + \Delta\bar{\rho},\quad (15)$$

with

$$\begin{aligned}\Delta\bar{\rho} &= \begin{pmatrix} 0 & 0 & 0 \\ 0 & 0 & 0 \\ 0 & 0 & a \end{pmatrix} + \begin{pmatrix} 0 & dm_z & 0 \\ -dm_z & 0 & 0 \\ 0 & 0 & 0 \end{pmatrix} \\ &+ \begin{pmatrix} b_1 m_z^2 & cm_x m_y & 0 \\ cm_x m_y & b_1 m_z^2 & 0 \\ 0 & 0 & b_2 m_z^2 \end{pmatrix}.\end{aligned}\quad (16)$$

The additional resistivity parameters are given by

$$\begin{aligned}a &= a_{33} - a_{11} + a_{1133} - a_{1122}, \\ b_1 &= a_{3311} - a_{1122}, \\ b_2 &= a_{3333} - a_{1111} - a_{1133} + a_{1122}, \\ c &= a_{1212} - a_{2323}, \\ d &= a_{312} - a_{123}.\end{aligned}\quad (17)$$

Accordingly, Eqs. (7)–(9) have to be rewritten as follows.

(001) surface and $\mathbf{j} \parallel [100]$

$$\begin{aligned}\rho_{\text{long}}^{\text{tetra}} &= A + B(\mathbf{j} \cdot \mathbf{m})^2 + b_1(\mathbf{n} \cdot \mathbf{m})^2 \\ \rho_{\text{trans}}^{\text{tetra}} &= (C + c)(\mathbf{j} \cdot \mathbf{m})(\mathbf{t} \cdot \mathbf{m}) - (D + d)(\mathbf{n} \cdot \mathbf{m}).\end{aligned}\quad (18)$$

(001) surface and $\mathbf{j} \parallel [110]$

$$\begin{aligned}\rho_{\text{long}}^{\text{tetra}} &= A + \frac{1}{2}(B - C - c) + (C + c)(\mathbf{j} \cdot \mathbf{m})^2 \\ &+ \left(\frac{1}{2}(C - B + c) + b_1 \right) (\mathbf{n} \cdot \mathbf{m})^2 \\ \rho_{\text{trans}}^{\text{tetra}} &= B(\mathbf{j} \cdot \mathbf{m})(\mathbf{t} \cdot \mathbf{m}) - (D + d)(\mathbf{n} \cdot \mathbf{m}).\end{aligned}\quad (19)$$

(113)A surface and $\mathbf{j} \parallel [3\bar{3}\bar{2}]$

$$\begin{aligned}\rho_{\text{long}}^{\text{tetra}} &= A + \frac{2}{11}a + \frac{9}{22}(B - C - c) + \frac{1}{121}(126C - 5B + b + 90c) \\ &\times (\mathbf{j} \cdot \mathbf{m})^2 + \frac{9}{242}(5C - 5B + b + 13c)(\mathbf{n} \cdot \mathbf{m})^2 \\ &+ \frac{3\sqrt{2}}{121}(5B - 5C - b + 9c)(\mathbf{j} \cdot \mathbf{m})(\mathbf{n} \cdot \mathbf{m}),\end{aligned}$$

$$\rho_{\text{trans}}^{\text{tetra}} = \frac{1}{11}(9B + 2C)(\mathbf{j} \cdot \mathbf{m})(\mathbf{t} \cdot \mathbf{m}) + \frac{3\sqrt{2}}{11}(B - C)(\mathbf{t} \cdot \mathbf{m})(\mathbf{n} \cdot \mathbf{m}) - \left(D + \frac{9}{11}d\right)(\mathbf{n} \cdot \mathbf{m}) + \frac{3\sqrt{2}}{11}d(\mathbf{j} \cdot \mathbf{m}), \quad (20)$$

where $b = 18b_1 + 4b_2$. For perfect cubic symmetry the parameters a , b_1 , b_2 , c , and d vanish and Eqs. (18)–(20) reduce to Eqs. (7)–(9). It should be emphasized that the expressions for ρ_{long} and ρ_{trans} derived above generally apply to ferromagnets of cubic or tetragonal symmetry, provided that the angular dependence of the resistivity tensor is exclusively determined by the direction cosines of the magnetization. Effects correlated with the magnitude B of the magnetic induction \mathbf{B} , such as the negative magnetoresistance, can be easily taken into account by considering B -dependent resistivity parameters.

B. Magnetic anisotropy

The pronounced MA in (Ga,Mn)As is associated with a density of the free enthalpy²⁸ G being highly anisotropic with respect to the orientation of \mathbf{M} . The direction of \mathbf{M} , i.e., the vector \mathbf{m} which enters the equations for ρ_{long} and ρ_{trans} given above, aligns in such a way that G takes its minimum value. In addition to the single-domain model, we assume that the magnitude M of the magnetization is nearly constant under the given experimental conditions while its orientation \mathbf{m} is strongly affected by the applied magnetic field \mathbf{H} . Instead of G we therefore consider the normalized quantity $G_M = G/M$, allowing for a more concise description of the MA. For a biaxially strained (Ga,Mn)As film grown on GaAs(001) substrate, it can be written as¹³

$$G_M^{001} = -\mu_0 \mathbf{H} \cdot \mathbf{m} + B_{\text{cl}}(m_x^4 + m_y^4) + B_{c\perp}m_z^4 + B_{001}(\mathbf{n} \cdot \mathbf{m})^2 + B_{\bar{1}10}(\mathbf{t} \cdot \mathbf{m})^2, \quad (21)$$

with $\mathbf{n} \parallel [001]$ and $\mathbf{t} \parallel [\bar{1}10]$. The terms refer, respectively, to the Zeeman energy, the cubic anisotropy under tetragonal distortion, an effective uniaxial anisotropy perpendicular to the film including demagnetization and magnetoelastic effects, and a uniaxial in-plane contribution whose origin is still under discussion.^{14,29,30} The anisotropy parameters B_i introduced in Eq. (21) are in SI units. Expressed by the anisotropy fields H_i and $4\pi M_{\text{eff}}$ used in Refs. 13 and 31, they read as $B_{\text{cl}} = -\mu_0 H_{4\parallel}/4$, $B_{c\perp} = -\mu_0 H_{4\perp}/4$, $B_{\bar{1}10} = -\mu_0 H_{2\parallel}/2$, and $B_{001} = \mu_0 4\pi M_{\text{eff}}/2$. Note that, by using the trivial identity

$$|\mathbf{m}|^2 = (\mathbf{j} \cdot \mathbf{m})^2 + (\mathbf{t} \cdot \mathbf{m})^2 + (\mathbf{n} \cdot \mathbf{m})^2 = 1, \quad (22)$$

Eq. (21) can be easily converted to a completely equivalent expression where the in-plane contribution along $[\bar{1}10]$ is formally replaced by a contribution along $[110]$. The only consequence is a redefinition of the anisotropy parameters and the addition of a constant term which does not alter the physical information provided by G_M . In Fig. 2, the free enthalpy is visualized by a three-dimensional (3D) plot, calculated for a weak magnetic field $\mu_0 H = 0.15$ T and a set of anisotropy parameters with arbitrarily chosen values $B_{\text{cl}} = B_{c\perp} = -0.1$ T, $B_{001} = 0.15$ T, and $B_{\bar{1}10} = -0.05$ T. The direc-

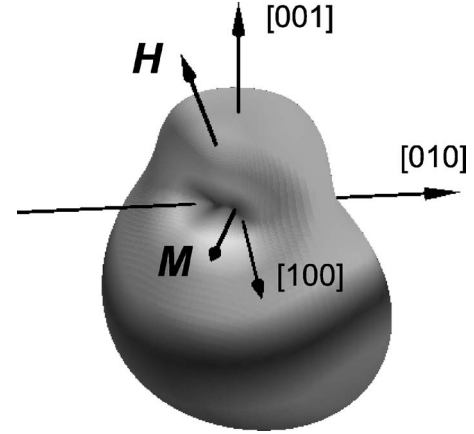


FIG. 2. G_M as a function of \mathbf{m} , calculated for a given magnetic field $\mu_0 H = 0.15$ T and a set of anisotropy parameters with arbitrarily chosen values $B_{\text{cl}} = B_{c\perp} = -0.1$ T, $B_{001} = 0.15$ T, and $B_{\bar{1}10} = -0.05$ T. The equilibrium position of \mathbf{M} is determined by the minimum of G_M .

tion of \mathbf{M} was calculated, as throughout the present work, by numerically minimizing G_M with respect to θ and φ (see Fig. 1). In doing so, we are able to trace the motion of \mathbf{M} , starting from a given orientation, while sweeping or rotating \mathbf{H} . Figure 3 shows as an example the simulated polar (θ_H , θ) and azimuth (φ_H , φ) angles of \mathbf{H} (dashed line) and \mathbf{M} (solid line), respectively, while \mathbf{H} is rotated within the (111) plane. For the simulation the same field strength and the same anisotropy parameters have been chosen as for the 3D plot in Fig. 2. While \mathbf{H} smoothly rotates within the (111) plane, \mathbf{M} remains very close to the (001) plane ($\theta \approx 90^\circ$) and undergoes sudden jumps in φ whenever the minimum of G_M discontinuously changes its position.

In the case of the (Ga,Mn)As films grown on GaAs(113)A substrates, the best fits to the experimental data (see Sec. IV B) are achieved for a normalized free-enthalpy density of the form¹⁷

$$G_M^{113} = -\mu_0 \mathbf{H} \cdot \mathbf{m} + B_{\text{cl}}(m_x^4 + m_y^4) + B_{c\perp}m_z^4 + B_{113}(\mathbf{n} \cdot \mathbf{m})^2 + B_{\bar{1}10}(\mathbf{t} \cdot \mathbf{m})^2 + B_{001}m_z^2. \quad (23)$$

The first five terms correspond to those already presented in

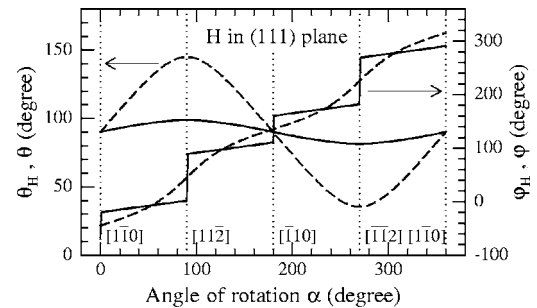


FIG. 3. Simulated polar (left axis) and azimuth (right axis) angles of the magnetic field \mathbf{H} (dashed lines) and the magnetization \mathbf{M} (solid lines) for \mathbf{H} rotated in the (111) plane. The same field strength and anisotropy parameters have been used as in Fig. 2.

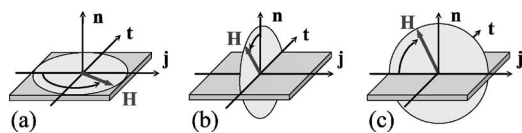


FIG. 4. The angle-dependent magnetotransport measurements were carried out for \mathbf{H} rotated within (a) the layer plane, (b) a plane perpendicular to the current direction \mathbf{j} , and (c) a plane spanned by \mathbf{j} and the normal vector \mathbf{n} .

Eq. (21). The sixth term, which has to be additionally introduced to obtain an optimal agreement between experiment and theory, is an inclined uniaxial contribution along [001], i.e., neither parallel nor perpendicular to the film.¹⁶ We attribute it to a lattice distortion along the [001] direction.

C. Ferromagnetic resonance

A highly efficient and widely used tool to study MA is ferromagnetic resonance spectroscopy.³² Most recently, a detailed review on FMR in (Ga,Mn)As has been given by Liu and Furdyna.³¹ In the FMR experiments, the total magnetic moment of the Mn ion–hole spin complex and thus the magnetization \mathbf{M} precesses around its equilibrium position (which in general is not identical to the orientation of \mathbf{H}) at the Larmor frequency ω_L . Sweeping the magnitude of the magnetic field \mathbf{H} at a fixed microwave frequency ω_{HF} , the resonance condition $\omega_L = \omega_{\text{HF}}$ is satisfied at the resonance field H_{res} which strongly depends on the orientation of \mathbf{H} due to MA. The resonance condition is given by³³

$$\left(\frac{\omega_{\text{HF}}}{\gamma}\right)^2 = \frac{1}{\sin^2 \theta} \left[\frac{\partial^2 G_M}{\partial \theta^2} \frac{\partial^2 G_M}{\partial \varphi^2} - \left(\frac{\partial^2 G_M}{\partial \theta \partial \varphi} \right)^2 \right], \quad (24)$$

where $\gamma = g\mu_B\hbar^{-1}$ denotes the gyromagnetic ratio, g the g factor, μ_B the Bohr magneton, and \hbar the Planck constant. At any given direction of \mathbf{H} , the resonance field is obtained by solving Eq. (24) at the equilibrium position of \mathbf{M} , i.e., for $\partial G_M / \partial \varphi = 0$ and $\partial G_M / \partial \theta = 0$. The anisotropy parameters can then be derived from a fit to the measured H_{res} recorded as a function of field orientation. In the present work, most of the calculations were carried out numerically.

IV. RESULTS AND DISCUSSION

The longitudinal and transverse resistivities of the (001) and (113)A (Ga,Mn)As layers were measured as a function of the magnetic field orientation at fixed field strengths $\mu_0 H = 0.1, 0.25, \text{ and } 0.7$ T. In order to probe the anisotropy in all three directions in space, the applied magnetic field \mathbf{H} was rotated within three different crystallographic planes perpendicular to \mathbf{n} , \mathbf{j} , and \mathbf{t} , respectively, as shown in Fig. 4. In the following, the measured angular dependences of ρ_{long} and ρ_{trans} are discussed and values for the resistivity and anisotropy parameters are derived by fits to the experimental data using the theoretical formalism presented in Sec. III. The results of FMR measurements, carried out on the same samples, are presented for reference. Note, however, that it is not the aim of the present study to yield a detailed or complete set of anisotropy parameters. In fact, the work is meant

to provide a comprehensive theoretical tool for the description of the resistivities in arbitrarily oriented (Ga,Mn)As layers and to demonstrate the potential of angle-dependent magnetotransport studies for the investigation of magnetic anisotropy.

A. (001) orientation

The anisotropy of the (001)-oriented sample was probed by rotating \mathbf{H} within the (001), (110), and $(\bar{1}10)$ planes. The corresponding angular dependences of ρ_{long} and ρ_{trans} , measured with the current direction along [110], are shown in Fig. 5. At 0.7 T the Zeeman energy dominates the free enthalpy and MA only plays a minor role. As a consequence, \mathbf{M} is expected to nearly align with the applied magnetic field and to continuously follow the motion of \mathbf{H} . In fact, the curves of ρ_{long} and ρ_{trans} at 0.7 T are smooth and largely reflect the anisotropy of the resistivity tensor. With decreasing magnetic field the influence of the MA increases and the orientation of \mathbf{M} deviates more and more from the field direction. Accordingly, jumps and kinks occur in the curves at 0.25 and 0.1 T, arising from sudden movements of \mathbf{M} caused by discontinuous displacements of the minimum of the free enthalpy. The observed angular dependences of ρ_{long} and ρ_{trans} can be understood in great detail by modeling the measured curves within the theoretical framework presented in Sec. III. For this purpose, the resistivity and anisotropy parameters from Eqs. (4), (17), and (21) were determined by an iterative fit procedure. Starting with an initial guess for the anisotropy parameters, the resistivity parameters were obtained by fitting Eqs. (19) to the experimental data recorded at 0.7 T. Then the anisotropy parameters were modified for an optimal agreement at 0.25 and 0.1 T, and the whole procedure was repeated until no further improvement of the fit could be achieved. The unit vector \mathbf{m} at any given magnetic field \mathbf{H} was calculated by numerically minimizing G_M^{001} in Eq. (21) with respect to the polar and azimuth angles of \mathbf{M} . With the exception of A , the resistivity parameters turned out to be field independent within the accuracy of the fit and are given by $B = -2.3 \times 10^{-4}$ Ω cm, $C + c = -1.7 \times 10^{-4}$ Ω cm, $b_1 = 0.9 \times 10^{-4}$ Ω cm, and $D + d = -4.4 \times 10^{-4}$ Ω cm. The resistivity parameter A was found to decrease from 7.24×10^{-3} Ω cm at 0.1 T to 7.11×10^{-3} Ω cm at 0.7 T, reflecting the negative-magnetoresistance behavior of ρ_{long} .¹⁷ For the anisotropy parameters we obtained the values $B_{\text{cl}} = -0.015$ T, $B_{c\perp} = 0$ T, $B_{001} = 0.17$ T, and $B_{\bar{1}10} = 0.002$ T. The theoretical curves calculated with these parameters are in excellent agreement with the experiment and are drawn as solid lines in Fig. 5.

Once the anisotropy parameters are known, the orientations of the easy axes can be determined by minimizing G_M^{001} with respect to \mathbf{m} at zero magnetic field. The easy axes are found to lie within the (001) layer plane ($\theta = 90^\circ$) at the azimuth angles $\varphi_1 = 1.9^\circ$ and $\varphi_2 = 88.1^\circ$ (see Fig. 1). The slight deviation from the cubic [100] and [010] axes toward the [110] direction arises from the positive value of $B_{\bar{1}10}$.

Using the expression for $\rho_{\text{long}}^{\text{tetra}}$ in Eqs. (19), the in-plane and out-of-plane AMR coefficients defined in Eqs. (10) read for a (001) layer with $\mathbf{j} \parallel [110]$ as

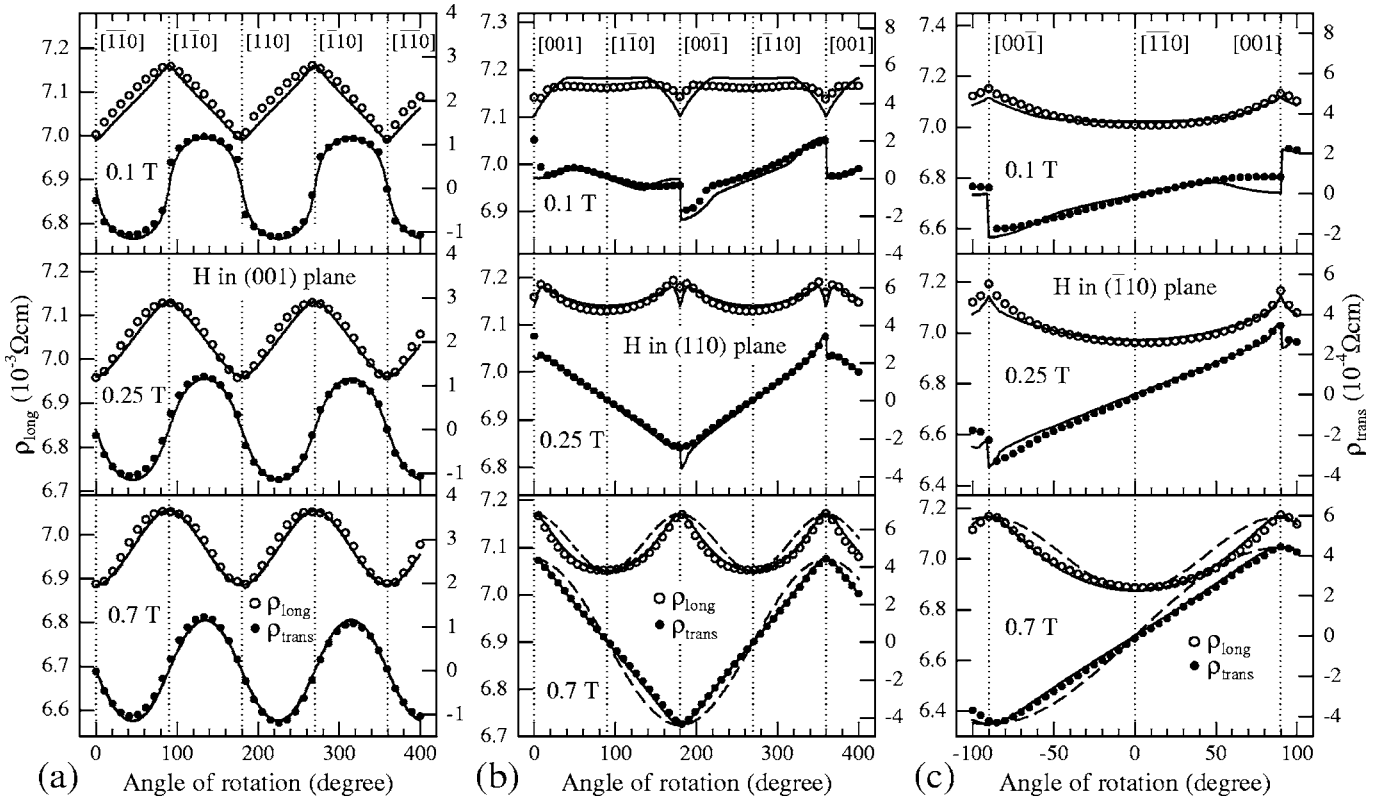


FIG. 5. Angle-dependent resistivities ρ_{long} (circles) and ρ_{trans} (dots) of the (001) (Ga,Mn)As sample at 4.2 K. The measurements were carried out at fixed field strengths of $\mu_0 H = 0.1, 0.25,$ and 0.7 T with \mathbf{H} rotated in (a) the (001), (b) the (110), and (c) the $(\bar{1}10)$ plane. The solid lines represent fits to the experimental data using Eqs. (19) and one set of resistivity and anisotropy parameters. The dashed lines at 0.7 T simulate the limiting case where \mathbf{M} perfectly aligns with \mathbf{H} . In (a) the dashed lines completely coincide with the solid lines.

$$A_{\text{ip}} = \frac{C + c}{A + (B + C + c)/2},$$

$$A_{\text{op}} = \frac{(B + C + c)/2 - b_1}{A + (B + C + c)/2}. \quad (25)$$

Inserting the resistivity parameters derived above, we find $A_{\text{ip}} = -2.5\%$ and $A_{\text{op}} = -4.2\%$. Similar values, namely, -3.4% and -4.0% , respectively, have been reported by Wang *et al.*⁸ for a 50-nm-thick (Ga,Mn)As layer with 5% Mn.

Based on the resistivity and anisotropy parameters given above, plots similar to Fig. 3 can be drawn for each configuration and field strength, revealing in great detail the motion of \mathbf{M} . Figure 6 shows as an example the polar (θ_H, θ) and azimuth (φ_H, φ) angles of \mathbf{H} and \mathbf{M} , respectively, plotted as functions of the angle of rotation α for \mathbf{H} rotated in the (110) plane at 0.1 T [see Fig. 5(b)]. While θ_H passes through all values between 0° and 180° (dashed line, left axis), \mathbf{M} remains very close to the (001) plane with $75^\circ < \theta < 105^\circ$ (solid line, left axis). For $40^\circ < \alpha < 140^\circ$ and $220^\circ < \alpha < 320^\circ$, where \mathbf{H} is closer to the (001) plane than to the $[00\bar{1}]$ axis, the azimuth angles φ_H and φ of \mathbf{H} and \mathbf{M} , respectively, almost perfectly coincide. When \mathbf{H} approaches the hard $[001]$ axis, however, \mathbf{M} tends toward the easy $[100]$ axis (azimuth angles 0° and 180°) or to the easy $[010]$ axis (azimuth angles -90° and 90°). At $\alpha = 180^\circ$ and $\alpha = 360^\circ$, where

\mathbf{H} exactly aligns with the $[00\bar{1}]$ and $[001]$ directions, respectively, the azimuth angle of \mathbf{M} undergoes a sudden jump by 90° .

Considerable information can also be obtained by comparing the measured angular dependences with those expected for the limiting case where \mathbf{M} perfectly aligns with \mathbf{H} . To this end, ρ_{long} and ρ_{trans} were calculated with \mathbf{m} replaced by the vector $\mathbf{h} = \mathbf{H}/H$ in Eqs. (19). The resulting curves are depicted by the dashed lines in Fig. 5. For \mathbf{H} rotated within

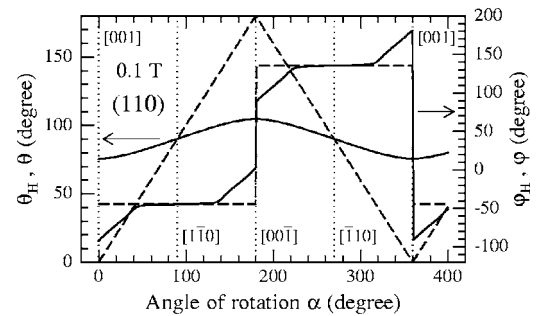


FIG. 6. Calculated polar (left axis) and azimuth (right axis) angles of the magnetic field \mathbf{H} (dashed lines) and the magnetization \mathbf{M} (solid lines) for \mathbf{H} rotated in the (110) plane of the (001) (Ga,Mn)As sample at 0.1 T. The resistivity and anisotropy parameters used in the calculation were derived from a fit to the angle-dependent resistivities shown in Fig. 5.

the (001) plane [Fig. 5(a)], the linear and quadratic ($\mathbf{n} \cdot \mathbf{m}$) terms in Eqs. (19) vanish and we obtain the well-known $\cos^2 \phi_j$ and $\cos \phi_j \sin \phi_j$ dependences of ρ_{long} and ρ_{trans} , respectively. At 0.7 T the dashed curves coincide with the solid curves, meaning that for \mathbf{H} rotated within the layer plane the magnetization almost perfectly follows the motion of the magnetic field. For lower fields, \mathbf{M} remains in the layer plane since [001] is a hard axis, but it increasingly deviates from \mathbf{H} toward the easy [100] and [010] axes. At 0.1 T it abruptly switches whenever \mathbf{H} approaches the somewhat harder [110] and $[\bar{1}\bar{1}0]$ axes, leading to the kinks observed for ρ_{long} .

The rotation of \mathbf{H} within a plane perpendicular to the layer is accompanied by significant differences in the orientations of \mathbf{H} and \mathbf{M} , even for 0.7 T. This is clearly demonstrated in Figs. 5(b) and 5(c), where the dashed curves represent the $(\mathbf{n} \cdot \mathbf{m})^2 = \cos^2 \phi_n$ and $(\mathbf{n} \cdot \mathbf{m}) = \cos \phi_n$ dependences of ρ_{long} and ρ_{trans} in Eqs. (19), respectively, with ϕ_n denoting the angle between \mathbf{m} and \mathbf{n} . At 0.7 T, \mathbf{H} and \mathbf{M} coincide whenever \mathbf{H} is orientated parallel or perpendicular to the layer plane. At lower fields this is no longer true and \mathbf{M} avoids the perpendicular direction by tending toward the easy [100] and [010] axes (see Fig. 6). Accordingly, the differences between the minimum and maximum values of ρ_{long} and ρ_{trans} are drastically reduced at 0.25 and 0.1 T.

It should be emphasized that the magnetotransport measurements give clear evidence for the tetragonal distortion of the (Ga,Mn)As layer. First, the parameters B_{cl} and $B_{c\perp}$, representing the in-plane and out-of-plane contributions to the cubic-anisotropy term in G_M^{001} , significantly differ. Second, Eqs. (8), which have been derived for the case of perfect cubic symmetry, correctly reproduce the measured amplitudes of ρ_{long} for the in-plane configuration in Fig. 5(a), but do not so for the two out-of-plane configurations in Figs. 5(b) and 5(c). Moreover, it is worth noting that the fits presented in Fig. 5 could be improved even further by taking into account higher-order terms in the series expansion of the resistivity tensor in Eq. (2). However, since the agreement achieved in second order is more than satisfactory and since the mathematical expressions for the resistivities would become much more complicated, higher-order terms have not been considered in the present study.

The results of FMR measurements are presented in Fig. 7. It shows the measured and simulated angular dependences of the resonance field H_{res} for \mathbf{H} rotated within the (001), $(\bar{1}\bar{1}0)$, and (110) planes. The dashed lines, reproducing only roughly the experimental curves, were numerically calculated using Eq. (21), Eq. (24), $g=2.0$, and the anisotropy parameters derived from the magnetotransport measurements. The agreement between experiment and theory is significantly improved using $g=1.9$ and the slightly higher values $B_{\text{cl}} = -0.02$ T, $B_{c\perp} = 0$ T, $B_{001} = 0.24$ T, and $B_{\bar{1}\bar{1}0} = 0.002$ T, which were obtained by a least-squares fit based on Eqs. (21) and (24). The values agree within 30% with the anisotropy parameters determined from magnetotransport. The reason for the remaining difference between the two sets of parameters is not yet clear. Inevitable sample heating up to 150 °C for less than 30 min during the Hall-bar preparation as well as the different lateral sizes of our samples (shape anisotropy) are not expected to account for the observed variation.

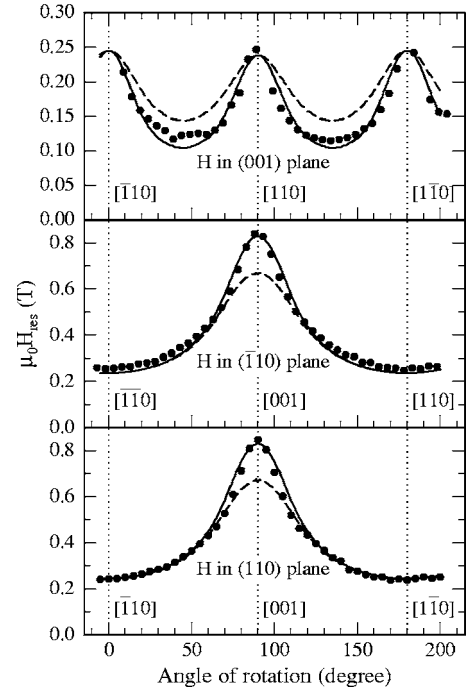


FIG. 7. Angle-dependent FMR fields of the (001) (Ga,Mn)As sample at 5 K for \mathbf{H} rotated in the (001), $(\bar{1}\bar{1}0)$, and (110) planes. The solid lines represent the result of a least-squares fit; the dashed lines were calculated using $g=2.0$ and the anisotropy parameters estimated from the magnetotransport data.

Angular dependences of H_{res} almost identical to those shown in Fig. 7 have been reported by Liu *et al.* in Refs. 12 and 31 for a 300-nm-thick (Ga,Mn)As layer with 3% Mn. They obtained an optimal fit to the experimental data by using values for the anisotropy fields H_i which are, written in terms of the anisotropy parameters B_i , very close to the values found in the present study, namely, $B_{\text{cl}} = -0.025$ T, $B_{c\perp} = 0$ T, $B_{001} = 0.17$ T, and $B_{\bar{1}\bar{1}0} = 0.02$ T.

B. (113)A orientation

As already mentioned in Sec. III B, our experimental data suggest the existence of a lattice distortion along [001] even in the (Ga,Mn)As film grown on GaAs(113)A substrate. This is demonstrated in Fig. 8, which shows the measured and simulated angular dependences of the FMR field H_{res} for \mathbf{H} rotated within the $(3\bar{3}\bar{2})$, $(1\bar{1}0)$, and (113) planes. The solid and dashed lines depict the results of least-squares fits using Eqs. (23) and (24) with and without considering the uniaxial term $B_{001}m_z^2$, respectively. Even though no perfect simulation of the measured curves could be achieved, the solid curve is much closer to the experimental data than the dashed one. Optimal agreement is obtained for $g=2.0$ and the values $B_{\text{cl}} = B_{c\perp} = -0.046$ T, $B_{113} = 0.032$ T, and $B_{001} = 0.053$ T of the anisotropy parameters (solid line in Fig. 8). Similar to the case of the (001) layer, the uniaxial in-plane contribution is almost negligible with $B_{\bar{1}\bar{1}0} = -0.005$ T.

The magnetotransport data, measured with the current direction along $[3\bar{3}\bar{2}]$, are depicted in Fig. 9. The figure shows

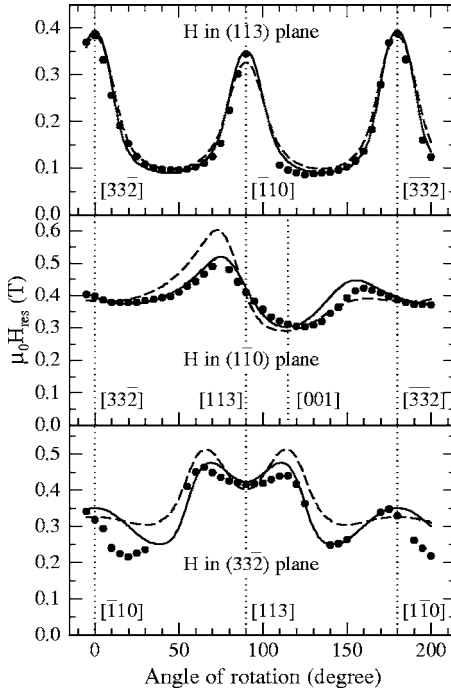


FIG. 8. Angle-dependent FMR fields of the (113)A (Ga,Mn)As sample at 5 K for \mathbf{H} rotated in the $(33\bar{2})$, $(\bar{1}\bar{1}0)$, and $(113)A$ planes. The solid and dashed lines represent the results of least-squares fits with and without considering a uniaxial term along $[001]$ in G_M^{113} , respectively.

the experimental and simulated angular dependences of ρ_{long} and ρ_{trans} for \mathbf{H} rotated within the (113) , $(33\bar{2})$, and $(\bar{1}\bar{1}0)$ planes. Using the fit procedure described in Sec. IV A, the values of the resistivity parameters in Eqs. (20) are obtained as $B = -8.1 \times 10^{-4} \Omega \text{ cm}$, $C = -8.7 \times 10^{-4} \Omega \text{ cm}$, $b = 7.2 \times 10^{-3} \Omega \text{ cm}$, $c = 5.1 \times 10^{-4} \Omega \text{ cm}$, $D = -2.1 \times 10^{-3} \Omega \text{ cm}$, and $d = 2.1 \times 10^{-4} \Omega \text{ cm}$. The sum $A + 2a/11$ decreases from $21.61 \times 10^{-3} \Omega \text{ cm}$ at 0.1 T to $20.79 \times 10^{-3} \Omega \text{ cm}$ at 0.7 T. For the anisotropy parameters we obtained the values $B_{c\perp} = -0.046 \text{ T}$, $B_{c\perp} = -0.03 \text{ T}$, $B_{113} = 0.018 \text{ T}$, $B_{001} = 0.02 \text{ T}$, and $B_{\bar{1}\bar{1}0} = -0.008 \text{ T}$. As for the (001) sample, they are smaller than the values determined by the FMR study; however, the discrepancy is less than a factor of about 2. The calculated curves, represented by the solid lines in Fig. 9, are in excellent agreement with the measured data. Again, the dashed curves simulate the case of a magnetization that perfectly aligns with \mathbf{H} . A comparison between the dashed and solid curves reveals that the magnetization at 0.7 T almost perfectly follows the motion of the magnetic field in the two out-of-plane configurations [Figs. 9(b) and 9(c)], whereas for \mathbf{H} rotated within the layer plane [Fig. 9(a)] \mathbf{M} significantly deviates from \mathbf{H} . According to the model calculations, the latter behavior arises from the cubic terms and the uniaxial $[001]$ contribution in G_M^{113} , resulting in a deflection of \mathbf{M} toward the (001) plane. The asymmetry of ρ_{trans} in Fig. 9(a) partly results from this deflection and partly from the last term in Eqs. (20) which originates from the tetragonal distortion. A similar asymmetry has been observed by Muduli *et*

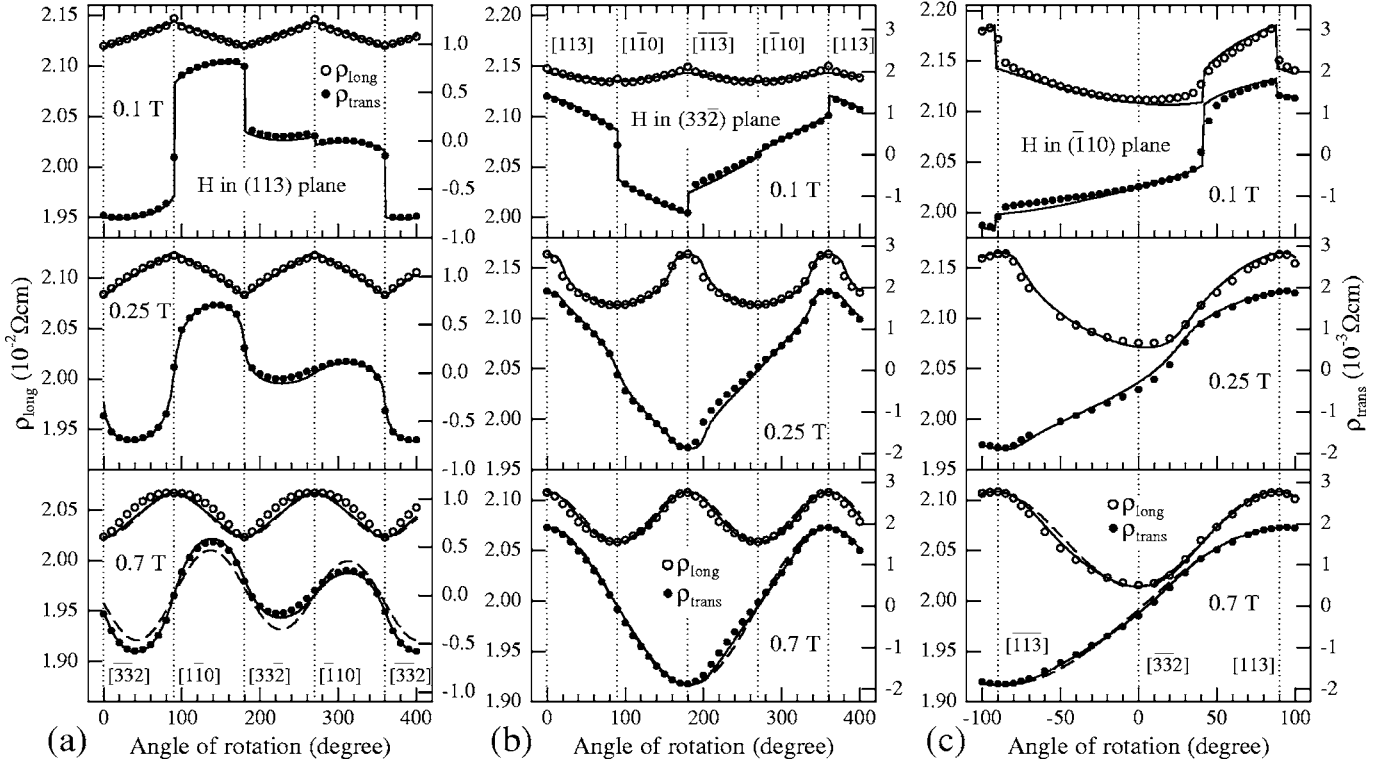


FIG. 9. Angle-dependent resistivities ρ_{long} (circles) and ρ_{trans} (dots) of the (113)A (Ga,Mn)As sample at 4.2 K. The measurements were carried out at fixed field strengths of $\mu_0 H = 0.1, 0.25$, and 0.7 T with \mathbf{H} rotated in (a) the (113) , (b) the $(33\bar{2})$, and (c) the $(\bar{1}\bar{1}0)$ plane. The solid lines represent fits to the experimental data using Eqs. (20) and one set of resistivity and anisotropy parameters. The dashed lines at 0.7 T simulate the limiting case where \mathbf{M} perfectly aligns along \mathbf{H} .

*al.*²⁴ in Fe₃Si films grown on GaAs(113)A substrates. There, the asymmetry has been explained by third-order terms in ρ_{trans} . We cannot rule out that in the (113)A (Ga,Mn)As sample under study higher-order terms contribute to the asymmetry, too. However, since the present model, including terms up to second order, fully accounts for the observed angular dependence of ρ_{long} and ρ_{trans} , it has not been considered as mandatory to include them.

Using the anisotropy parameters obtained from the curve fits, the orientations of the easy axes were determined by minimizing G_M^{113} with respect to \mathbf{m} at zero magnetic field. We find the easy axes in the (113)A layer at the angles (see Fig. 1) $\theta_1=92^\circ$, $\varphi_1=-3.4^\circ$, and $\theta_2=92^\circ$, $\varphi_2=93.4^\circ$, i.e., very close to the [100] and [010] axes, in qualitative agreement with the results presented in Refs. 16 and 21.

V. SUMMARY

A series expansion of the resistivity tensor with respect to the magnetization components yields general expressions for

the longitudinal and transverse resistivities in single-crystalline ferromagnets with cubic and tetragonal symmetry. The expressions, applicable to (Ga,Mn)As layers with arbitrary surface index, were used to quantitatively model the angular dependences of the resistivities, measured in (001) and (113)A (Ga,Mn)As films as a function of magnetic field orientation. Whereas the curves at 0.7 T largely reflect the anisotropy of the resistivity tensor, the curves at 0.25 and 0.1 T are strongly affected by magnetic anisotropy, allowing access to anisotropy parameters. The magnetotransport data and comparative ferromagnetic resonance studies reveal an inclined uniaxial anisotropy along [001] in the (113)A-oriented (Ga,Mn)As layers in addition to the usual in- and out-of-plane contributions known from (001) layers.

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