Nonvolatile bistability effect based on the electrically controlled phase transition in scaled magnetic semiconductor nanostructures

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We explore the bistability effect in a dimensionally scaled semiconductor nanostructure consisting of a dilute magnetic semiconductor quantum dot (QD) and a reservoir of itinerant holes separated by a barrier. The bistability stems from the magnetic phase transition in the QD mediated by the changes in the hole population. Our calculation shows that when properly designed, thermodynamic equilibrium of the scaled structure can be achieved at two different configurations; i.e., the one with the QD in a ferromagnetic state with a sufficient number of holes and the other with the depopulated QD in a paramagnetic state. The parameter window suitable for this bistability formation is discussed along with the conditions for maximum robustness/ nonvolatility. To examine the issue of scaling, an estimation of the bistability lifetime is made by considering the thermal fluctuation in the QD hole population via the spontaneous transitions. A numerical evaluation is carried out for a typical carrier-mediated magnetic semiconductor [e.g., (Ga,Mn)As] as well as for a hypothetical case of high Curie temperature for potential room-temperature operation.

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The magnetism in semiconductors is the basis for the emerging field of spin-polarized electronics, or spintronics.¹ Substantial progress has been made during the past few years, particularly in materials development. The advantages of the semiconductor-based systems over the metallic counterparts include the controllability of the ferromagnetism (via the bias² and/or doping) and the potential compatibility with modern Si-based processing technology.

Recently, a theoretical study further explored opportunities offered by electrically controlled magnetism. It was found that a properly designed structure consisting of magnetic and nonmagnetic semiconductor quantum wells (QWs) can exhibit bistability with respect to the paramagneticferromagnetic (PM-FM) phase transition when the process is controlled by the itinerant carriers (holes).³ This bistability effect in the *two-dimensional* system was predicted to persist even at temperatures nearly as high as the critical temperature T_c of the PM-FM phase transition. Subsequently, a nonvolatile memory application was suggested, citing the successful growth of transition metal doped semiconductors that are FM at or above room temperature [e.g., the nitrides,⁴ Ge,⁵ as well as some II-VI's (see Refs. 6 and 7)].

For practical realization of the proposed device application, it is highly desirable to reduce the size of the magnetic layer (i.e., the active part of the memory) without the loss of high-temperature operability and nonvolatility. Consequently, a magnetic semiconductor quantum dot (QD) that can exchange itinerant holes with a reservoir provides an interesting opportunity. However, unlike the QW case examined earlier,³ the thermal fluctuation resulting from spontaneous hopping between the two possible stable states cannot be neglected in the scaled structure due to the limited number of carriers populating the QD. Hence, it is the purpose of this paper to theoretically investigate the effect of the reduced dimensionality and explore the potential bistability conditions based on the electrically controlled magnetic phase transition in the magnetic semiconductor nanostructures. The specific structure under consideration consists of a single dilute magnetic semiconductor (DMS) quantum dot (QD) separated from a large reservoir of itinerant holes, which controls the chemical potential μ_0 of the system. For simplicity, we assume $\mu_0 \gg k_B T$ and ignore the possible temperature dependence of μ_0 . A nonmagnetic QW filled with itinerant holes (for example, through modulation doping, etc.) can be used as the desired reservoir (see Fig. 1).

To accurately describe this system, one needs to know the energy structure of the QD, which is a function of multiple parameters. In particular, the magnetic interactions that lead to the PM-FM phase transition in the DMS QD must be taken into account. Note that the analysis of different magnetic phase transition mechanisms is beyond the scope of the present study (see, for details, Ref. 8). Instead, we assume that the main magnetic properties [such as the critical temperature T_c and its dependence on the hole concentration, and the magnetization dependence on temperature M = M(T)] can be obtained from measurements of the relevant DMS. This allows the use of a semiphenomenological approach in calculating the free energy of the system.

In the present model, we approximate the free energy of the QD as the sum of two terms: the magnetic (F_M) and nonmagnetic (F_N) contributions. If the DMS QD is not far from the PM-FM transition, the Landau expansion in the magnetization M can be applied for F_M :

$$F_M = -a(T_c - T)M^2 + bM^4.$$
 (1)

The parameters *a*, *b*, and T_c are functions of the number of holes *j* in the QD, and in particular, the dependence $T_c = T_c(j)$ plays a crucial role in the magnetic instability.⁹ In addition, *a* and *b* can be expressed in terms of the fundamental properties of the magnet: The Curie-Weiss law for magnetic susceptibility $\chi = C_0/(T-T_c)$ at $T > T_c$ defines $a=1/4C_0$, while spontaneous magnetization $M_s = M_0\sqrt{1-T/T_c}$ at $T < T_c$ implies $b=aT_c/2M_0^2$. Since Eq. (1) is



FIG. 1. Schematic diagram of the structure containing a DMS QD (in two different phases) and a nonmagnetic quantum well (NQW) reservoir separated by a barrier. The energy orientation is provided from the hole point of view in the valence band. Left: The PM phase corresponds to disordered magnetic ion spins (small arrows) and the lack of a magnetic contribution to the hole energy. This is achieved when the holes (small circles) only weakly populate the QD with a discrete energy spectrum and, thus, cannot change its magnetic state. The ground state U of the QD is high relative to the chemical potential μ_0 of the hole reservoir. Right: Another thermodynamically stable state (at the same external conditions) is possible when the magnetic ions are ferromagnetically ordered. Magnetic interactions can decrease the hole potential so that the ground state of the DMS QD is now substantially below μ_0 ; i.e., the equilibrium hole population is high enough to stabilize the FM phase. Switching between the PM and FM states can be achieved by applying a gate bias V_g that populates or depopulates the DMS QD.

assumed to fully describe the magnetization of the DMS QD, the thermodynamically stable state corresponds to the specific magnitude of M that gives rise to the free energy minimum

$$F_{M} = -T_{c} \frac{M_{0}^{2}}{C_{0}} \left(1 - \frac{T}{T_{c}}\right)^{2} \theta \left(1 - \frac{T}{T_{c}}\right), \qquad (2)$$

where $\theta(x)$ is the Heaviside step function. It is important to note that the localized spin *S* of the magnetic ions provides the main contribution to the DMS magnetization, whereas the itinerant carriers play a minor role. The parameters M_0 and C_0 can be easily estimated for N_m localized spin moments leading to the estimation $M_0^2/C_0=3SN_m/8(S+1)$, which is independent of carriers. Thus, the only dependence of F_M on the hole population *j* comes from the term T_c = $T_c(j)$ in our approximation.

To obtain a numerical value for j, one must also incorporate the nonmagnetic part F_N of the free energy for j holes located in the QD. Unfortunately, the calculation of F_N requires approaches very specific to each individual case as F_N depends on the details such as the material composition, the size and shape of the QD, the presence of dopants and external fields, etc. Consequently, this problem cannot be solved in a general manner. To proceed further, we treat the QD as a scaled QW with a finite lateral size (and thickness); this is analogous to a nanodot embedded in a barrier. The potential profile of the sample structure along the growth (z) direction is schematically illustrated in Fig. 1 from the *hole* representation, as is the case throughout the paper. It is convenient to split F_N into two parts: $F_N = E_i + F_1(T, j)$. The first term,

$$E_{j} = jU + \frac{1}{2}j(j-1)C,$$
(3)

accounts for the energy acquired by *j* holes due to their localization in the QD with the ground state energy *U* as well as their Coulomb repulsion energy $(C=e^2/\epsilon\sqrt{A_0})$, where *e* is the electron charge, ϵ the dielectric constant, and A_0 the lateral cross section of the QD).¹⁰ The remaining part,

$$F_1(T,j) = \Omega(T,\mu_1) + j\mu_1(j), \tag{4}$$

is similar to the free electron gas contribution with the thermodynamic potential:¹¹

$$\Omega(T,\mu_1) = -k_B T \sum_n \ln[1 + e^{(\mu_1 - \varepsilon_n)/k_B T}].$$
 (5)

 ε_n and μ_1 represent the energy spectrum (with the quantum number *n*) and the chemical potential of the QD when the influence of the magnetic interaction is excluded (i.e., the nonmagnetic version of the QD). In what follows, we consider the lateral dimension A_0 of the QD to be relatively sizable so that the energy gaps in the discrete energy spectra are smaller than k_BT . The sum in Eq. (5) can then be approximated by an integral with the density of states $mA_0/\pi\hbar^2$. From the relation $j=-\partial\Omega(T,\mu_1)/\partial\mu_1,\mu_1(j)$ is found to be

$$\mu_1(j) = k_B T \ln(e^{\xi j} - 1), \tag{6}$$

where $\xi = \pi \hbar^2 / m A_0 k_B T$ and *m* is the in-plane hole effective mass. Consequently, the thermodynamic potential [Eq. (5)] can be expressed in the form

$$\Omega[T,\mu_1(j)] = -\frac{k_B T}{\xi} \int_0^\infty \ln[1+(e^{\xi j}-1)e^{-x}]dx.$$
(7)

Equations (3) and (4) along with $\mu_1(j)$ and $\Omega(T, \mu_1)$ from Eqs. (6) and (7) determine the nonmagnetic part of the free energy, while the total free energy of the DMS QD is the sum $F = F_M + F_N$.

Now we take into account the fact that the QD is in contact with a large reservoir providing two-way exchange of carriers through the potential barrier (see Fig. 1). This leads to the establishment of a unified chemical potential that coincides with μ_0 in the reservoir. Thus, the equation that determines the population of the QD takes the form

$$\mu_{\rm QD}(j) = \mu_0. \tag{8}$$

Note that $\mu_{\text{QD}}(j) \neq \mu_1(j)$ since both the nonmagnetic and magnetic interactions are considered for $\mu_{\text{QD}}(j)$. Since the chemical potential $\mu_{\text{QD}}(j)$ of the QD can be expressed as $\mu_{\text{QD}}(j)=dF/dj$ in general, the stable solutions of Eq. (8) must correspond to the local minima of F=F(j) or equivalently $d\mu_{\text{QD}}(j)/dj > 0$.

Finally, the desired solutions require a specific expression for $T_c = T_c(j)$ in Eq. (2). Following the experimental data of



FIG. 2. Chemical potential of the DMS QD with a thickness of 5 nm and cross section of 25×25 nm² as a function of the QD hole population [Eq. (10)]. The parameters of Ga_{0.95}Mn_{0.05}As are assumed with T=77 K as discussed in the text. The solutions of Eq. (10) can be found as intersections of the solid curve with the horizontal line corresponding to a certain value of $\Delta \mu (=\mu_0 - U)$. Two cases $\Delta \mu / T_c^0 = 6$ and 0.5 (dashed lines 1 and 3) provide monostable FM and PM states, while dashed line 2 ($\Delta \mu / T_c^0 = 2.7$) depicts the bistable state. Stable solutions are indicated by single-head arrows and the unstable one by the horizontal double-head arrow.

Ref. 12, we adopt a semiphenomenological description

$$T_c = T_c^0 (1 - e^{-\alpha j \xi t}),$$
(9)

where T_c^0 is the asymptotic value of the critical temperature (at a sufficiently high hole concentration), $t=T/T_c^0$, and $\alpha(=1)$ is the fitting parameter that adjusts Eq. (9) to the experiments. Subsequently, the QD hole population *j* can be obtained from Eqs. (2)–(4) and (8) as

$$\mu_{0} - U = (j - 1/2)C + k_{B}T \ln(e^{\xi j} - 1) - k_{B}T\xi \frac{3SN_{m}}{8(S+1)}$$
$$\times e^{-\alpha\xi t j} \left[1 - \left(\frac{T}{T_{c}}\right)^{2}\right] \theta \left(1 - \frac{T}{T_{c}}\right).$$
(10)

For a numerical evaluation, let us assume the following set of parameters "typical" for a carrier-mediated DMS (e.g., Ga_{0.95}Mn_{0.05}As): $m = 0.13m_0$ (m_0 is the free electron mass), ϵ =12.9, S=5/2, N_m=1.3×10²¹ cm⁻³×(QD volume), and T_c^0 =110 K. Figure 2 depicts the dependence of $[\mu_{QD}(j)]$ $-U]/k_BT_c^0$ as a function of j. Clearly, the results indicate that only one solution for *j* exists at sufficiently low or high energies U in reference to μ_0 (e.g., dashed line 1 or 3) corresponding to the only stable QD population at a given U. However, the moderate values of U can support multiple solutions. In the case of dashed line 2, two of them (with the smallest and largest i) are stable considering the positive derivative $(d\mu_{\rm OD}/dj > 0)$, while the intermediate solution is not $(d\mu_{\rm OD}/dj < 0)$. Note also that the stable solutions with the larger (smaller) *j* are realized in the FM (PM) phase of the DMS QD. Hence, this demonstrates a bistable state for a properly designed QD in terms of the hole population or the magnetic phase. Figure 3 provides the ranges of U (assuming



FIG. 3. Phase diagram of the parameter space indicating the potential bistability region (the shaded area). The PM and FM labels denote the monostable areas corresponding to the PM and FM QD states, respectively. The same parameters as in Fig. 2 are assumed $(T_c^0=110 \text{ K})$.

a fixed μ_0) and *T* where the bistability can be expected in the system under consideration. Clearly, the calculated results indicate a large window in the system parameter space where the bistability is possible. The maximum operating temperature may be nearly as high as T_c^0 .

To achieve a bistable state robust against thermal fluctuations (i.e., nonvolatile) for possible memory applications, it is necessary to select a condition that provides the maximal separation $\Delta F = \min\{F_{\max} - F_{P_{\min}}, F_{\max} - F_{F_{\min}}\}\)$ between the local maximum F_{\max} and each of the local minima $F_{P_{\min}}$ and $F_{F_{\min}}$ (for the PM and FM phases of the DMS QD, respectively) of the free energy F(j). At a given temperature, one can find the QD potential alignment that results in the maximal ΔF . Figure 4 illustrates the behavior of F(j) at three different shifts $\Delta \mu = \mu_0 - U$: curve 1 (3) represents the case of the monostable state in the FM (PM) state, whereas curve 2



FIG. 4. Free energy of the QD calculated as a function of hole population for three different values of $\Delta \mu (=\mu_0 - U)$: $\Delta \mu / T_c^0 = 6$ (curve 1, dotted line); 2.7 (curve 2, solid line); 0.5 (curve 3, dashed line). The single minima of curves 1 (FM phase) and 3 (PM phase) correspond to the vicinities of the right and left boundaries of the bistable area in Fig. 3; curve 2 represent the bistable case with the optimal free energy barrier height separating two local minima. The same parameters as in Fig. 2 are assumed (T=77 K, $T_c^0=110$ K).



FIG. 5. Maximal free energy barrier height (in units of k_BT) between the local maximum and the closest local minimum in the bistable case (curve 2) and the corresponding optimal chemical potential (curve 1) as a function of temperature. Curves 3–5 in the bottom pane shows the hole population at the FM minimum (curve 3), the PM minimum (curve 5), and the local maximum (curve 4) of F(j) for the bistable case shown in the top pane at the corresponding temperature. The same parameters as in Fig. 2 are assumed $(T_c^0=110 \text{ K})$.

exhibits two local minima in the FM and PM states, respectively, separated by the maximal ΔF . The temperature dependence of the maximal ΔF and the associated optimal potential shift $\Delta \mu$ is plotted in Fig. 5 along with the mean values of the particle numbers j_P , j_m , and j_F corresponding to $F_{P_{\min}}$, F_{\max} , and $F_{F_{\min}}$. Our analysis shows that the "thermodynamic barrier" ΔF decreases drastically as *T* approaches T_c^0 [curve 2 in Fig. 5(a)]. Obviously, the system based on the PM-FM transition becomes much less stable near the critical temperature.

Note that the mean value *j* is reached through the balance of the particle flux to and from the QD. Each of these events transfers one particle via thermal activation with a characteristic time τ_0 , which depends on the temperature, height, and width of the energy barrier separating the QD and reservoir, etc. The lifetime T_{lt} is defined as the time it takes to develop a sufficiently large fluctuation to induce a transition from the state initially at $F_{F_{\min}} = F(j_F)$ to that at $F_{P_{\min}} = F(j_P)$. Apparently, if the system reaches the state at the local maximum $F_{\max} = F(j_m)$, further evolution can result in either the PM (with $j=j_P$) or the FM (with $j=j_F$) QD state with an approximately equal probability; hence, T_{lt} can be estimated as the reciprocal probability for the process $F_{F_{\min}} \rightarrow F_{\max}$ or $j_F \rightarrow j_m$.

The fluctuation $\Delta j = j_F - j_m$ can be found by considering the sequential process of hole withdrawal from the QD. For the first hole transfer out of the QD, the characteristic time of this process is τ_0 as defined above. Due to the finiteness of the QD hole population, this reduces the chemical potential



FIG. 6. Bistability lifetime vs temperature at T_c^0 =110 K. Two different QD dimensions are considered: (1) $25 \times 25 \times 5$ nm³ and (2) $15 \times 15 \times 5$ nm³. The mean time τ_0 of particle exchange between the QD and reservoir via thermal processes is assumed to be 1 ns. Other parameters are the same as in Fig. 2.

by $\Delta\mu(1) = \mu_{\text{QD}}(j_F) - \mu_{\text{QD}}(j_F-1)$. Consequently, the new time constant becomes $\tau_0 \exp[\Delta\mu(1)/k_BT]$ when the next hole escapes from the QD, provided no particles are injected into the QD from the reservoir. Hence, the mean time necessary for the Δj fluctuation through the sequential withdrawal of holes can be estimated to be

$$T_{w}(\Delta j) = \tau_{0} \sum_{j=0}^{j_{m}-1} \exp[\Delta \mu(j)/k_{B}T].$$
 (11)

On the other hand, the probability of no hole injection from the reservoir during this time span T_w is P_w =exp $[-T_w(\Delta j)/\tau_0]$. The frequency of appearance for such a rare occasion is P_w/τ_0 and the desired lifetime is

$$T_{lt} = \tau_0 \exp[T_w(\Delta j)/\tau_0]. \tag{12}$$

A similar expression is shown to apply to the fluctuations $F_{P_{\min}} \rightarrow F_{\max}$ or $j_P \rightarrow j_m$.

One can see that the estimated lifetime (or the bit retention time) depends crucially on the operating temperature



FIG. 7. Bistability lifetime vs temperature for a hypothetical DMS material with characteristics similar to (Ga,Mn)As, in which T_c^0 in Eq. (9) is treated as a variable over a wide temperature range. *T* is fixed at 300 K and three different QD sizes are considered: (1) $25 \times 25 \times 5$ nm³, (2) $15 \times 15 \times 5$ nm³, and (3) $25 \times 25 \times 3$ nm³. The mean time τ_0 of particle exchange between the QD and reservoir via thermal processes is assumed to be 1 ns as in Fig. 6.

and the QD sizes, which determine the number of terms j_m in the sum of Eq. (11). Figure 6 analyzes the results for the (Ga,Mn)As QD with dimensions of $25 \times 25 \times 5$ nm³ and $15 \times 15 \times 5$ nm³ assuming $T_c^0 = 110$ K. As expected, the bistable state becomes less stable (i.e., shorter T_{lt}) as the QD size shrinks. The thermal fluctuation clearly has a bigger impact in this case due to the finite number of holes in the QD. For the two structures considered (or those of similar sizes), a practically nonvolatile condition (i.e., sufficiently long bit retention) may be achieved when operating below ~75 K. $\tau_0 = 1$ ns is used for the calculation.

To examine the feasibility of room-temperature operation, it is desirable to extend our consideration of magnetic semiconductors to those with potentially much higher critical temperatures. Note, for instance, two recent reports of DMS with $T_c \ge 300$ K.^{5,7} Since the search for an ideal DMS is only at the beginning stage, we assume a hypothetical material with characteristics similar to (Ga,Mn)As except for T_c^0 in Eq. (9), which is treated as a variable over a wide temperature range. Figure 7 depicts the estimated lifetime as a function of T_c^0 , while T is fixed at 300 K. For a sufficiently long

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 T_{lt} in this case, the desired material needs a T_c^0 of approx. 550 K or higher.

In summary, we investigate theoretically the effect of reduced dimensionality in magnetic semiconductor nanostructures and explore the potential bistability conditions based on the electrically controlled magnetic phase transition. The analysis is based on a semiphenomenological model that assumes common magnetic behavior and a simple hole energy spectrum in a DMS QD. When properly designed, the calculation predicts the possibility of controlled switching between the stable PM and FM states in the QD. The parameter window suitable for bistability is discussed along with the conditions for maximum robustness/nonvolatility. An estimation of the bistability lifetime as limited by thermal fluctuations provides a guideline for its potential application as a room-temperature low-power, high-density memory element.

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