

Incommensurate interactions and nonconventional spin-Peierls transition in TiOBr

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Temperature-dependent x-ray diffraction of the low-dimensional spin-1/2 quantum magnet TiOBr shows that the phase transition at $T_{c2}=47.1(4)$ K corresponds to the development of an incommensurate superstructure. Below $T_{c1}=26.8\pm 0.3$ K the incommensurate modulation locks in into a twofold superstructure similar to the low-temperature spin-Peierls state of TiOCl. Frustration between intra- and interchain interactions within the spin-Peierls scenario, and competition between the two-dimensional magnetic order and one-dimensional spin-Peierls order are discussed as possible sources of the incommensurability.

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I. INTRODUCTION

Recently, TiOCl was proposed to be a quasi-one-dimensional $S=1/2$ quantum spin system that develops a spin-Peierls state at low temperatures.¹ The spin-Peierls state is now well established by the temperature dependence of the magnetic susceptibility (χ_m), which is zero below the phase transition at $T_{c1}=67$ K, the observation by NMR of two independent Ti atoms below T_{c1} , the twofold crystallographic superstructure below T_{c1} , and electronic band-structure calculations.¹⁻⁴ The atomic displacements in the superstructure, as well as the calculated band structure, with the single valence electron of Ti^{3+} occupying the d_{xy} orbital, indicate that the spin-Peierls state is formed on the chains of Ti atoms parallel to \mathbf{b} via direct exchange interactions.^{1,3,4}

Although the properties of the low-temperature phase of TiOCl are those of a true spin-Peierls system, TiOCl is not a conventional spin-Peierls compound, because the phase transition at T_{c1} is first order. The temperature dependencies of χ_m , electron spin resonance (ESR), nuclear magnetic resonance (NMR), specific heat (C_p), and x-ray diffraction have shown that a second-order phase transition occurs at $T_{c2}=91$ K.^{1-3,5,6} The one-dimensional (1D) character of the magnetic interactions was also supported by the temperature dependencies of optical reflectivity and angle-resolved photoelectron spectroscopy (ARPES),⁷⁻⁹ although it was suggested that on cooling from room temperature, a crossover from two-dimensional (2D) towards 1D interactions occurs.^{7,10} The nature of the state above T_{c1} is not understood yet. Orbital, spin, and structural fluctuations have been proposed to be responsible for the properties of TiOCl.^{2,6,11,12} However, Rückamp *et al.*⁹ suggested that orbital fluctuations can be ruled out.

TiOBr and TiOCl crystallize in the FeOCl structure type.^{13,14} Physical properties of both compounds are similar, with the two transition temperatures scaled down towards $T_{c1}=27$ K and $T_{c2}=47$ K in TiOBr.^{9,15-18} The twofold superstructures below T_{c1} are similar in TiOCl and TiOBr, suggesting a spin-Peierls state for TiOBr too.¹⁹ In the present paper we report on incommensurate satellite reflections in x-ray diffraction of TiOBr at temperatures T with $T_{c1} < T < T_{c2}$. Complete crystal structures are presented, but

the data do not allow us to distinguish between a one-dimensional and a two-dimensional incommensurate modulation wave. These two models provide two possible interpretations for the understanding of the interatomic interactions in TiOBr and TiOCl.

II. EXPERIMENTAL

Single crystals of TiOBr were prepared by gas transport reaction.^{13,19} A single crystal of dimensions $0.27 \times 0.13 \times 0.002$ mm³ was glued on a carbon fiber that was attached to a closed-cycle helium cryostat mounted on a four-circle Huber diffractometer. Single-crystal x-ray diffraction with synchrotron radiation was measured at beamline D3 of HASYLAB (DESY, Hamburg), employing monochromatized radiation of wavelength $0.5000(1)$ Å and a point detector.

Diffraction at room temperature confirmed the FeOCl structure type.¹⁹ The temperature dependence of the component q_1 of the modulation wave vector $\mathbf{q}=(q_1, 0.5, 0)$ was determined from q scans along \mathbf{a}^* centered on the positions $(-2, -3.5, -1)$ and $(1, -3.5, -2)$. Up to $T_{c1}=27$ K a single peak was found at $q_1=0$, while for $T_{c1} < T < T_{c2}=47$ K two peaks appeared in each scan, at positions $\pm q_1$ [Fig. 1(a)]. Above T_{c2} any diffraction at these positions had disappeared. These results show that below T_{c1} TiOBr has a twofold superstructure, while in the intermediate phase TiOBr is incommensurately modulated. The component q_1 of the modulation wave vector was found to continuously decrease on decreasing temperature, and it jumps to zero at T_{c1} [Fig. 2(a)]. This

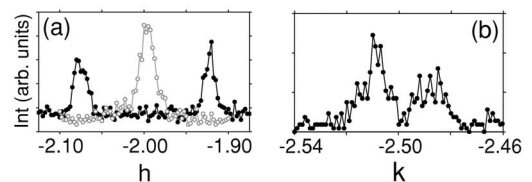


FIG. 1. (a) Q scans along \mathbf{a}^* centered on $(-2, -3.5, -1)$ at $T=17.5$ K (open circles) and $T=37$ K (full circles). (b) Q scans along \mathbf{b}^* centered on $(0.075, -2.5, -1)$ at $T=35$ K. The lines are a guide for the eye. The stepwidths of the scans were 0.0025 . The intensities in (a) are higher than those in (b).

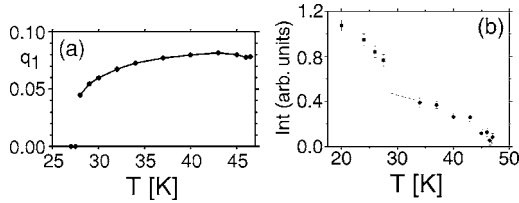


FIG. 2. (a) Temperature dependence of q_1 in $\mathbf{q}=(q_1, 0.5, 0)$. q_1 was obtained as half the distance between the two maxima in q scans, like those given in Fig. 1(a). (b) Temperature dependencies of the intensities of the satellite reflections $(-2, -3.5, -1)$ (squares) and $(2+q_1, -3.5, -1)$ (circles). The line represents a fit with the function $I(T)=I_0\sqrt{1-T/T_{c2}}$ with $T_{c2}=47.0(6)$ K. Similar plots lead to $T_{c2}=47.1(6)$ K for $(-2-q_1, -3.5, -1)$, $T_{c2}=46.8(8)$ K for $(1+q_1, -3.5, -2)$, and $T_{c2}=48(2)$ K for $(1-q_1, -3.5, -2)$.

result corroborates the incommensurate character of the modulation in the intermediate phase, and it shows the first-order character of the transition at T_{c1} . The transition temperature was determined from Fig. 2(a) as $T_{c1}=26.8\pm 0.3$ K.

The integrated intensities of reflections were measured by ω scans centered at the expected reflection positions. In this way, the temperature dependencies were measured of the intensities of two commensurate satellites (below 27 K) and of four incommensurate satellites [Fig. 2(b)]. Due to an experimental error, reliable intensities were not obtained for temperatures $27.5\text{ K}\leq T\leq 32\text{ K}$. The intensities $I(T)$ above T_{c1} could be fitted by a function $I(T)=I_0\sqrt{1-T/T_{c2}}$, which provided consistent estimates for the transition temperature from all four reflections, with an average value of $T_{c2}=47.1(4)$ K. The gradual loss of intensity of the incommensurate satellites on approaching T_{c2} is an indication for the second-order character of the phase transition at this temperature.

Incited by the observation that some reflections were broader than others in the ω scans of the data collection (see below), the possibility of a second incommensurate component of the modulation was investigated, which resulted in a modulation wave vector $\mathbf{q}=(q_1, 0.5+\delta, 0)$. Evidence for a splitting along \mathbf{b}^* was obtained from q scans along \mathbf{b}^* with narrow slits [Fig. 1(b)]. These results show that each incommensurate satellite $(q_1, 0.5, 0)$ actually is the superposition of two satellites $(q_1, 0.5\pm\delta, 0)$ with $\delta=0.011$ at $T=35$ K. In ω scans they are not resolved, because of the relatively large width of the reflections due to the less than optimal crystal quality.

The twofold superstructure below T_{c1} of TiOBr was determined from the integrated intensities of the Bragg reflections measured at $T=17.5$ K.¹⁹ The result was in complete accordance with the twofold superstructure of TiOCl,³ and it thus shows that the low-temperature phase of TiOBr is a spin-Peierls state like that in TiOCl.¹⁹

At $T=35$ K the integrated intensities of the Bragg reflections up to $[\sin(\theta)/\lambda]_{\max}=0.62\text{ \AA}^{-1}$ were measured by ω scans. As noticed above, each measured satellite intensity corresponds to the superposition of two reflections at $(h+q_1, k+0.5\pm\delta, l)$, or at $(h-q_1, k+0.5\pm\delta, l)$, while the two groups are separated from each other. Structure refinements were performed within the superspace approach for incom-

mensurate structures,^{20,21} employing the computer program JANA2000.²² The diffraction data have orthorhombic mmm symmetry. Refinements of the average structure against the intensities of the main reflections showed that the room-temperature structure in $Pm\bar{m}n$ is also valid as the average structure at 35 K. Cooling had negatively affected crystal quality, resulting in full widths at half maximum (FWHMs) of the reflections up to 0.06° at 50 K. A systematic variation of reflection widths could not be observed between 50 and 35 K. However, the splitting of reflections due to a possible lowering of symmetry is expected to be small, probably less than the observed FWHM above the transition. As a consequence, information on a possible lowering of the symmetry towards monoclinic could not be derived from the reflection profiles.

Without direct information on the symmetry, two possibilities exist for the superspace group. If the orthorhombic symmetry is preserved, the modulation is two dimensional with modulation wave vectors

$$\mathbf{q}^1 = (0.075, 0.511, 0),$$

$$\mathbf{q}^2 = (-0.075, 0.511, 0), \quad (1)$$

at $T=35$ K. The $(3+2)$ -dimensional superspace group is $Pm\bar{m}n(\alpha, \beta, 0)(-\alpha, \beta, 0)000$. The incommensurate modulation is described by one harmonic for each of the two waves [Eq. (1)]. Because all atoms are in the mirror plane m_a , the two harmonics are equivalent by symmetry and three of the six independent components are zero. This leads to modulation functions for each of the three atoms Ti, O, and Br given by three parameters only,

$$u_x(\bar{x}_{s4}, \bar{x}_{s5}) = u_x^1[\sin(2\pi\bar{x}_{s4}) - \sin(2\pi\bar{x}_{s5})],$$

$$u_y(\bar{x}_{s4}, \bar{x}_{s5}) = u_y^1[\sin(2\pi\bar{x}_{s4}) + \sin(2\pi\bar{x}_{s5})],$$

$$u_z(\bar{x}_{s4}, \bar{x}_{s5}) = u_z^1[\cos(2\pi\bar{x}_{s4}) + \cos(2\pi\bar{x}_{s5})], \quad (2)$$

with $\bar{x}_{s4}=\mathbf{q}^1\cdot(\mathbf{x}+\mathbf{L})$ and $\bar{x}_{s5}=\mathbf{q}^2\cdot(\mathbf{x}+\mathbf{L})$. \mathbf{x} is the average position of the atom in the unit cell, and \mathbf{L} enumerates the unit cells.

A reduction of the symmetry to monoclinic with a unique axis \mathbf{a} gives the same model for the modulation functions as was obtained in orthorhombic symmetry. Therefore, this possibility was not considered any further. Alternatively, monoclinic symmetry with a unique axis \mathbf{c} corresponds to a one-dimensional modulation with wave vector $\mathbf{q}=\mathbf{q}^1$, and with the $(3+1)$ -dimensional superspace group $P2/n(\alpha, \beta, 0)00$. The modulation now is a single wave, which can be obtained from Eq. (2) by removing all terms containing \bar{x}_{s5} . The diffraction symmetry mmm implies that the crystal is twinned, if the symmetry is monoclinic, with the first domain modulated by \mathbf{q}^1 and the second domain modulated by \mathbf{q}^2 . Structure refinements, employing the nine independent modulation parameters, gave excellent fits to the diffraction data with equal R factors for both monoclinic symmetry (assuming twinning) and orthorhombic symmetry.²⁴ Furthermore,

TABLE I. Structural parameters for the three independent atoms in the incommensurate structure of TiOBr at $T=35$ K. Basic structure coordinates $\mathbf{x}=(x^0, y^0, z^0)$ are relative to the lattice parameters $a=3.7849(9)$, $b=3.4685(7)$, and $c=8.500(2)$ Å. Modulation parameters are given in angstroms [Eq. (2)]. Ti is at $(0, 0.5, z^0)$; O and Br are at $(0, 0, z^0)$.

| Atom | z^0 | u_x^1 (Å) | u_y^1 (Å) | u_z^1 (Å) |
|--|---------------|----------------|----------------|----------------|
| <i>Pmnm</i> ($\alpha, \beta, 0$)($-\alpha, \beta, 0$)000 000 | | | | |
| Ti | 0.11096 (6) | -0.0043 (7) | 0.0262 (6) | -0.0116 (6) |
| O | -0.05127 (23) | 0.0031 (33) | 0.0150 (20) | 0.0160 (18) |
| Br | 0.32947 (3) | -0.0033 (6) | -0.0062 (4) | -0.0148 (3) |
| <i>P2/n</i> ($\alpha, \beta, 0$)00 (<i>c</i> unique) | | | | |
| Ti | 0.11096 (6) | -0.0057 (11) | 0.0369 (9) | -0.0165 (8) |
| O | -0.05126 (23) | 0.0050 (46) | 0.0213 (29) | 0.0225 (26) |
| Br | 0.32947 (3) | -0.0049 (8) | -0.0088 (5) | -0.0210 (4) |

both models give rise to equal values for the structure factors, so that they cannot be distinguished on the basis of the diffraction data.

III. DISCUSSION

Both structure models have equal values for the parameters, except for a scale factor of $\sqrt{2}$ between the modulation amplitudes, that is explained by the different assumptions on twinning (Table I). The two models share several features that allow for conclusions on the state of the intermediate phase. The major amplitude is the displacement of Ti along **b**. This is similar to the displacements in the low-temperature phase, and it suggests that antiferromagnetic interactions between Ti atoms along **b** are important in the incommensurate phase. Displacements along **a** are small, in accordance with the low-temperature phase, where symmetry requires them to be zero. Thus, the intermediate phase appears to be an incommensurate version of the low-temperature twofold superstructure. The incommensurability determines that the modulation amplitudes, as well as Ti—Ti distances, assume all values between a maximum and minimum. Alternatively, it cannot be excluded that the modulation is a block wave, which would correspond to domains with a twofold superstructure separated by domain walls where the spin pairs would be broken. Both the continuous modulation as well as the domain model are in accordance with incomplete spin pairing and with the observed finite value of χ_m in the intermediate phase.

Rückamp *et al.*⁹ have proposed that the incommensurate phase is the result of frustration between intra- and interchain interactions, for which a finite amplitude of the modulation along **a** is required. The monoclinic model with a 1D modulation is in accordance with this interpretation. This model features displacements along **a** that are small but vary in phase with the major displacement along **b** [Eq. (2) and Fig. 3(a)].

Interference of the two waves in the orthorhombic model implies that the displacements along **a** and **b** are out of phase [Eq. (2) and Fig. 3(b)], i.e., a large displacement along **b**

corresponds to zero displacement along **a**. This is exactly as can be expected for a competition between intra- and interchain interactions, the former dominating in regions with large u_y displacements and the latter being important in regions with small u_y . The model of frustrated spin-Peierls interactions could also be supported by this interpretation of the crystal structure.⁹ Alternatively, the 2D nature of the modulation wave in orthorhombic symmetry suggests that the incommensurability might be the result of a competition between spin-Peierls interactions on the chains and 2D magnetic interactions between neighboring Ti atoms along $[0.5, 0.5, \sim 0.25]$, resulting in a helical magnetic structure coupled

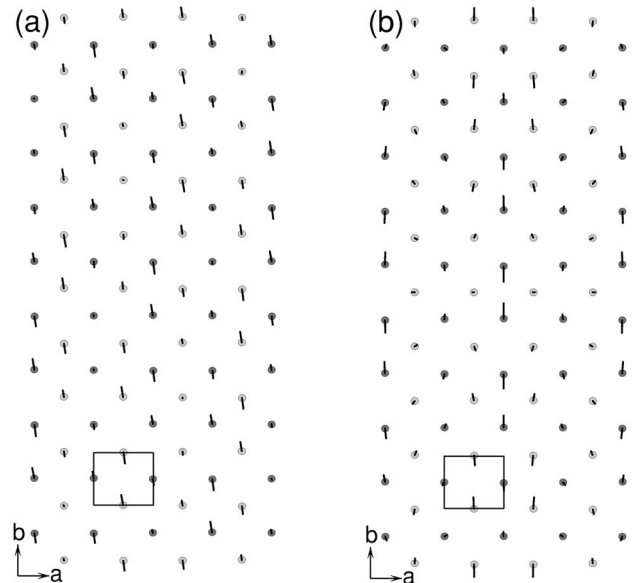


FIG. 3. One double layer of Ti atoms parallel to the **a, b** plane (a) for the 1D modulation in monoclinic symmetry; (b) for the 2D modulation in orthorhombic symmetry. Atoms are depicted at their basic positions, with shifts towards their true position indicated by arrows with a length of 20 times the true displacements. Dark and light circles represent Ti atoms at $-z_0$ and z_0 , respectively. The modulation was computed in a 5×11 supercell approximation.

to the lattice distortion.²³ The latter interpretation is in accordance with the observed increase of the two-dimensional character on increasing temperature,^{7,10} and with the proposed contributions of d_{xz} and d_{yz} symmetries to the orbital of lowest energy and the increased admixture of these orbitals at higher temperatures due to phonons.^{7,10–12,17}

The two symmetries lead to completely different properties of the phase transition at T_{c1} . Structurally, the orthorhombic model can be considered a precursor to the spin-Peierls state, and the phase transition is a lock-in transition. On the other hand, the monoclinic model requires that the unique axis switches from **c** to **a** at T_{c1} . Accordingly, the domain structure of the material must change, and a much more sluggish behavior of the transition, e.g., hysteresis, would be expected.

Previously, we have failed to observe incommensurate satellites in the intermediate-temperature phase of TiOCl. This is easily explained by the experimental method, which involved scans along reciprocal lattice directions, and thus might have missed possible satellites, especially if the incommensurate components in TiOCl will be larger than in TiOBr.

IV. CONCLUSIONS

We have found that TiOBr exhibits a second-order phase transition at $T_{c2}=47.1(4)$ K towards an incommensurately modulated structure. Available experimental data indicate that this structure is either orthorhombic with a two-dimensional modulation or monoclinic with a one-dimensional modulation. Both structure models give rise to similar local structures. Arguments have been given that favor or disfavor mechanisms for the existence of the incommensurate phase, that involve either frustrations between dimerized chains⁹ or two-dimensional magnetic interactions²³ in competition with one-dimensional spin-Peierls interactions. The equivalence of the low-temperature structures of TiOBr and TiOCl shows that TiOBr is in a spin-Peierls state below $T_{c1}=26.8\pm 0.3$ K.^{3,19}

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¹A. Seidel, C. A. Marianetti, F. C. Chou, G. Ceder, and P. A. Lee, Phys. Rev. B **67**, 020405(R) (2003).

²T. Imai and F. C. Chou, cond-mat/0301425 (unpublished).

³M. Shaz, S. van Smaalen, L. Palatinus, M. Hoinkis, M. Klemm, S. Horn, and R. Claessen, Phys. Rev. B **71**, 100405(R) (2005).

⁴T. Saha-Dasgupta, R. Valenti, H. Rosner, and C. Gros, Europhys. Lett. **67**, 63 (2004).

⁵V. Kataev, J. Baier, A. Moller, L. Jongen, G. Meyer, and A. Freimuth, Phys. Rev. B **68**, 140405(R) (2003).

⁶J. Hemberger, M. Hoinkis, M. Klemm, M. Sing, R. Claessen, S. Horn, and A. Loidl, cond-mat/0501517 (unpublished).

⁷G. Caimi, L. Degiorgi, N. N. Kovaleva, P. Lemmens, and F. C. Chou, Phys. Rev. B **69**, 125108 (2004).

⁸M. Hoinkis, M. Sing, J. Schafer, M. Klemm, S. Horn, H. Benthien, E. Jeckelmann, T. Saha-Dasgupta, L. Pisani, R. Valenti, and R. Claessen, cond-mat/0506203 (unpublished).

⁹R. Rückamp, J. Baier, M. Kriener, M. Haverkort, T. Lorenz, G. Uhrig, L. Jongen, A. Möller, G. Meyer, and M. Grüninger, cond-mat/0503409 (unpublished).

¹⁰P. Lemmens, K. Y. Choi, G. Caimi, L. Degiorgi, N. N. Kovaleva, A. Seidel, and F. C. Chou, Phys. Rev. B **70**, 134429 (2004).

¹¹L. Craco, M. Laad, and E. Müller-Hartmann, cond-mat/0410472 (unpublished).

¹²L. Pisani and R. Valenti, Phys. Rev. B **71**, 180409(R) (2005).

¹³H. Schäfer, F. Wartenpfuhl, and E. Weise, Z. Anorg. Allg. Chem.

295, 268 (1958).

¹⁴H. G. von Schnering, M. Collin, and M. Hassheider, Z. Anorg. Allg. Chem. **387**, 137 (1972).

¹⁵G. Caimi, L. Degiorgi, P. Lemmens, and F. C. Chou, J. Phys.: Condens. Matter **16**, 5583 (2004).

¹⁶C. Kato, Y. Kobayashi, and M. Sato, J. Phys. Soc. Jpn. **74**, 473 (2005).

¹⁷P. Lemmens, K. Choi, R. Valenti, T. Saha-Dasgupta, E. Abel, Y. Lee, and F. Chou, New J. Phys. **7**, 074 (2005).

¹⁸T. Sasaki, M. Mizumaki, K. Kato, Y. Watabe, Y. Nishihata, M. Takata, and J. Akimitsu, cond-mat/0501691 (unpublished).

¹⁹L. Palatinus, A. Schönleber, and S. van Smaalen, Acta Crystallogr., Sect. C: Cryst. Struct. Commun. **61**, i48–i49 (2005).

²⁰P. M. de Wolff, T. Janssen, and A. Janner, Acta Crystallogr., Sect. A: Cryst. Phys., Diffr., Theor. Gen. Crystallogr. **37**, 625 (1981).

²¹S. van Smaalen, Crystallogr. Rev. **4**, 79 (1995).

²²V. Petricek, M. Dusek, and L. Palatinus, *The Crystallographic Computing System JANA2000* (Institute of Physics, Praha, Czech Republic, 2000).

²³R. J. Beynon and J. A. Wilson, J. Phys.: Condens. Matter **5**, 1983 (1993).

²⁴502 measured reflections contain 159 observed main reflections and 149 observed superlattice reflections. Structure refinements in monoclinic and orthorhombic symmetries converged to models with the same reliability factors $R_{all}=0.022$ with $R_{main}=0.018$ and $R_{sat}=0.080$.