Doping phase diagram of $Y_{1-x}Ca_xBa_2(Cu_{1-y}Zn_y)_3O_{7-\delta}$ from transport measurements: Tracking the pseudogap below T_c

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The effects of planar hole concentration, p, and magnetic field on the resistivity, $\rho(T)$, of high-quality c-axis oriented crystalline thin films and sintered $Y_{1-x}Ca_xBa_2(Cu_{1-y}Zn_y)_3O_{7-\delta}$ samples were investigated over a wide range of Ca, Zn, and oxygen contents. Zn was used to suppress superconductivity and this enabled us to extract the characteristic pseudogap temperature, $T^*(p)$ below $T_{c0}(p)[\equiv T_c(x=0, y=0)]$. We have also located the characteristic temperature, T_{scf} , marking the onset of significant superconducting fluctuations above T_c , from the analysis of $\rho(T, H, p)$ and $\rho(T, p)$ data. From this we are able to identify $T^*(p)$ near the optimum doping level where the values of $T^*(p)$ and $T_{scf}(p)$ are very close and hard to distinguish. We again found that $T^*(p)$ depends only on the hole concentration p, and not on the level of disorder associated with Zn or Ca substitutions. We conclude that (i) $T^*(p)$ (and therefore, the pseudogap) persists below $T_{c0}(p)$ on the overdoped side and does not merge with the $T_{c0}(p)$ line and (ii) $T^*(p)$ and the pseudogap energy extrapolate to zero at the doping $p=0.19\pm0.01$.

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I. INTRODUCTION

One of the most remarkable phenomena in high- T_c cuprates is the observation of the pseudogap in the spectra of charge and spin excitations.^{1–3} The pseudogap (PG) is detected by various experimental techniques¹⁻³ over a certain range of planar hole concentrations, p (the number of added holes per CuO_2 plane), extending from the underdoped (UD) to the slightly overdoped (OD) regions. In the pseudogap state various anomalies are observed both in the normal and superconducting (SC) states, which can be interpreted in terms of a depletion of the single particle density of states.³ A number of theoretical explanations have been proposed as to the origin for the PG, which is believed to be an essential feature of the physics of the normal state (NS) and possibly of the SC state. Existing theories of the PG can be classified broadly into two categories. The first is based upon incoherent Cooper pair formation for $T < T^*$ well above the SC transition temperature, T_c , ⁴⁻⁶ with long-range phase coherence appearing only at $T \leq T_c$. In this scenario T^* may be the mean-field T_c . $T^*(p)$ merges with the $T_c(p)$ phase curve in the OD region ($p \ge 0.20$) where the pair formation temperature is essentially the same as the phase coherence temperature. In the second scenario, the PG is ascribed to fluctuations of some other type that coexist and generally compete with superconductivity. The most popular picture here is that of short-range antiferromagnetic (AFM) fluctuations, but similar effects have been attributed to charge density waves, a structural phase transition or electronic phase separation on a microscopic scale (e.g., the stripe scenario).⁷⁻¹⁰ One key constraint, which may rule out many of the above models, is that the single-particle density of states has a mysterious states nonconserving property in the PG state.¹¹ The experimental situation is often thought to be rather inconclusive regarding the origin of the $PG.^{1,2}$

The evolution of $\rho(T)$ with p provides a way of establishing the T-p phase diagram of high- T_c superconductors (HTS) and can give estimates of $T^*(p)$, which is identified from a characteristic downturn in $\rho(T)$. One particular difficulty associated with this method is that when $p \approx p_{opt}$ [where $T_c(p)$ is maximum], T^* is close to the temperature, T_{scf} , at which the effect of SC fluctuation is clearly seen in $\rho(T)$. This is not a problem for theories belonging to the first group where $T^*(p)$ itself is essentially derived from strong SC fluctuations. For the second scenario, this poses a problem as SC fluctuations (and superconductivity itself) mask the signatures of the predicted PG in the vicinity of (and below) T_c . With the notable exception of specific heat measurements, ^{3,11} most experimental techniques are unable to track the predicted $T^*(p)$ below $T_c(p)$. One way of avoiding this is to suppress superconductivity with a magnetic field and reveal the normal state below T_c , because the PG is comparatively insensitive to magnetic field.¹² In practice this is very hard to accomplish because of the large upper critical field, H_{c2} , of the hole-doped HTS. The other way is to destroy superconductivity by introducing disorder, e.g., by alloying with Zn. Because of the *d*-wave SC order parameter, Zn suppresses superconductivity most effectively and, like a magnetic field, has little effect on T^* irrespective of hole concentration.^{13–15} We took this second route to look for $T^*(p)$ below $T_{c0}(p)$, though we also employ magnetic fields to help distinguish between SC fluctuations and PG effects.

In this paper we report a systematic study of the transport properties of the superconducting compound $Y_{1-x}Ca_xBa_2(Cu_{1-y}Zn_y)_3O_{7-\delta}$. We have measured resistivity, room-temperature thermopower, *S*[290 K], and the AC sus-

ceptibility (ACS) of a series of *c*-axis oriented thin films and sintered $Y_{1-x}Ca_xBa_2(Cu_{1-y}Zn_y)_3O_{7-\delta}$ samples with different levels of Zn, Ca, and oxygen contents. The motivation for the present study is to locate $T^*(p)$ from the evolution of $\rho(T)$ of Y123 with different amounts of Zn and Ca extending from UD to OD states. While pure Y123 with full oxygenation (δ =0), is slightly overdoped, further overdoping can be achieved by substituting Y^{3+} by Ca^{2+} . The advantages of using Zn are (i) it mainly substitutes the in-plane Cu(2) sites, thus the effects of planar impurity can be studied and (ii) the doping level remains nearly the same when Cu(2) is substituted by Zn, enabling one to look at the effects of disorder on various normal and SC state properties at almost the same hole concentration.^{13–15} From the analysis of $\rho(T,p)$ and $\rho(T, p, H)$ data we extracted the p dependence of the characteristic temperatures T^* and T_{scf} , and find that indeed T^* falls below T_{c0} , providing strong evidence for the second scenario, namely that the PG arises from a correlation independent of, and coexisting with, superconductivity.

II. EXPERIMENTAL DETAILS AND RESULTS

Single-phase polycrystalline sintered samples of $Y_{1-x}Ca_xBa_2(Cu_{1-y}Zn_y)_3O_{7-\delta}$ were synthesized by solid-state reaction methods using high-purity (>99.99%) powders. The details of sample preparation and characterization can be found in Refs. 16 and 17. High-quality *c*-axis oriented thin films were grown on (100) SrTiO₃ substrates using the pulsed laser deposition (PLD) method. Details of the PLD parameters and characterization of the films can be found in Ref. 17.

The NS and SC properties, including T^* , of HTS are highly sensitive to p and, therefore, it is important to determine p as accurately as possible. The room temperature thermopower, S[290 K], has a systematic variation with p for various HTS over the entire doping range extending from very UD to heavily OD regimes,¹⁸ also S[290 K] is insensitive to in-plane disorder like Zn and the crystalline state of the sample.¹⁹ For these reasons S[290 K] is a good measure of p even in the presence of strong in-plane electronic scattering by Zn²⁺ ions. For all our samples we have used $S[290 \text{ K}]^{20}$ to determine p. Using these values of p, we find that the parabolic $T_c(p)$ formula,²¹ given by

$$\frac{T_c(p)}{T_c(p_{\rm opt})} = 1 - Z(p - p_{\rm opt})^2$$
(1)

is obeyed for all samples. Usually, for Zn-free samples, Z and p_{opt} take universal values of 82.6 and 0.16, respectively,²¹ but these parameters increase systematically with increasing Zn content.²² In the present case, Z increases from the usual 82.6 for the Zn-free sample to 410 for the 6%-Zn sample²² and p_{opt} rises from 0.16 for the Zn-free compound to 0.174 for the 6%Zn–20%Ca sample.²² The physical meaning of these changes in the fitting parameters is that as the Zn content is increased, the region of superconductivity shrinks and becomes centered on higher values of p, before finally forming a small "bubble" around $p \sim 0.19$ and disappearing completely for $y \sim 0.1$.^{17,22,23} From the observed evolution of $p_{opt}(y)$ we previously inferred that superconductivity for this system is at its strongest at $p \sim 0.185$,^{17,22} as this remains the last point of superconductivity at a critical Zn concentration (defined as the highest possible Zn concentration for which superconductivity just survives, considering all *p* values). This point has been made earlier in other studies²³ and the value $p \sim 0.19$ is indeed a special value at which the PG vanishes quite abruptly, as seen from the analysis of specific heat data of a number of different HTS cuprates.^{11,23}

The hole concentration was varied by changing both the oxygen deficiency and the Ca content. We have obtained T_c from both resistivity and low field ACS measurements $(H_{\rm rms}=0.1 \text{ Oe}; f=333.3 \text{ Hz})$. T_c was taken at zero resistance (within the noise level of $\pm 10^{-6} \Omega$) and at the point where the line drawn on the steepest part of the diamagnetic ACS curve meets the T-independent base line associated with the negligibly small NS signal. T_c values obtained from these two methods agree within 1 K for most of the samples.²² We placed particular emphasis on the determination of p and T_c as accurately as possible because of the extreme sensitivity of various ground-state SC and NS properties to p. This is especially true near the optimum doping level, where, although $T_c(p)$ is nearly flat, the SC condensation energy, superfluid density, PG energy scale and other quantities change quite abruptly and substantially for a small change in p.11,23-25

Patterned thin films with evaporated gold contact pads and high density (89% to 93% of the theoretical density) sintered bars were used for resistivity measurements. Resistivity was measured using the four-terminal method with an ac current of 1 mA (77 Hz), using 40 μ m diam. copper wire and silver paint to make the low resistance contacts. We have tried to locate the PG temperature, T^* , with a high degree of accuracy. At this point we would like to emphasize that $T^*(p)$ may not represent a phase transition temperature but instead $k_B T^*(p)$ is some kind of a characteristic energy scale of a lightly doped Mott insulator, reflecting probably the energy of correlated holes and spins.¹¹ As discussed previously²² plots of $d\rho(T)/dT$ vs T and $[\rho(T)-\rho_{\rm LF}]$ vs T yield very similar T* values (within ± 5 K) (ρ_{LF} is a linear fit $\rho_{\text{LF}} = b + cT$ in the high-T linear region of $\rho(T)$, above T^*). Actually $d\rho(T)/dT$ vs T is more useful in the sense that it gives a more general measure for the $T^*(p)$ [characterized by an increase in the slope of $\rho(T)$ in the vicinity of T^* , than the deviation from the high-T linear $\rho(T)$. Using T^*/T as a scaling parameter, it is also possible to normalize $\rho(T)$. The result of the scaling was shown in Ref. 22, where, leaving aside the SC transitions, all resistivity curves collapsed reasonably well on to one *p*-independent universal curve over a wide temperature range.

It is important to note that Zn does not change the $T^*(p)$ values but it suppresses $T_c(p)$ very effectively. For example, $T^*(p \sim 0.115) \sim 250 \pm 5$ K for both Zn-free and the 3%-Zn samples but T_c itself is suppressed from 70 K (Zn-free) to 29 K (3%-Zn).²² Similar results have been obtained for Zn-substituted Y123 by other studies.¹³ This very different Zn-dependence for T_c and T^* has often been stated as an indication for the non-SC origin of the PG.²⁶

Magnetotransport measurements were made using an Oxford Instruments superconducting magnet system. The sample temperature was measured using a Cernox thermometer with an absolute accuracy of ~ 50 mK. For thin film samples magnetic field was always applied along the c axis, perpendicular to the plane of the film. We have located $T_{\rm scf}$, from the analysis of $\rho(T,H)$ and $d^2\rho(T)/dT^2$ vs T data, in the T range from T_c to $\sim T_c + 30$ K. Here we have used the facts that (i) $\rho(T,H)$ becomes strongly field sensitive only below $T_{\rm scf}$, where experimentally we find that conventional strong SC fluctuations set in quite abruptly, but T^* is unaffected by magnetic field, at least for H up to 12 Tesla and (ii) $\rho(T)$ shows a much stronger, and progressively accelerating, downturn at T_{scf} than that present at T^* . As a consequence, plots of $d^2\rho(T)/dT^2$ vs T mask the PG feature and visually enhance the effects of SC fluctuations near T_{scf} . This is clearly seen in Fig. 1. Notice that for nearly identical values of $p (=0.134 \pm 0.004)$, both the sintered and the thin film samples with different Ca contents have almost the same values for T^* and ΔT_{scf} , where $\Delta T_{scf} = T_{scf} - T_c$. The sintered sample has a much larger resistivity due to the percolative effect and larger contributions from the grain boundaries.²⁷ We have defined $T_{\rm scf}$ from $d^2\rho(T)/dT^2$ as the temperature at which strong and accelerating downturn in $\rho(T)$ becomes evident near T_c . $d\rho(T)/dT$ and $d^2\rho(T)/dT^2$ were obtained numerically from the $\rho(T)$ data. $d^2\rho(T)/dT^2$ is particularly sensitive to any noise in the data, in some cases smoothing over a temperature window of 1 K was used. This does not affect the location of $T_{\rm scf}$ in any noticeable way because the final divergence of $d^2\rho(T)/dT^2$ is so abrupt. Linear fits of $\rho(T)$ in the temperature range from $\sim T_c + 35$ K to $T_c + 25$ K also locate T_{scf} within ±2.5 K (see the insets of Fig. 2). Extraction of T_{scf} from the $\rho(T, H)$ data is shown next in Fig. 2 for both sintered and thin film samples. T_{scf} from $\rho(T,H)$ is characterized by the onset of strong magnetic field dependence in the $\rho(T,H)$ data. In summary, three different methods [including the linear fit of $\rho(T)$ to locate the onset of the strong deviation in $\rho(T)$ from linearity near the superconducting transition] are applied and all of them give the same value of $T_{\rm scf}$ to within ±2.5 K.

Figure 3 shows $T_c(p)$ and $T_{scf}(p)$ of two representative sintered 20%-Ca samples with 0%-Zn and 4%-Zn. $\Delta T_{scf}(p)$ is insensitive to Zn content (y) (see the inset of Fig. 3), and is also found to be independent of Ca content (x) and crystalline state of the samples (see the inset of Fig. 3).

We have modeled the evolution of $\Delta T_{\rm scf}(p)$ from the temperature dependence of ρ at high temperatures and from the Aslamazov-Larkin (AL) contribution to the fluctuation conductivity.²⁸ It is widely accepted now that the Maki-Thomson²⁹ contribution to the fluctuation conductivity is not important for cuprate superconductors due to the presence of strong pair breaking effects for the *d*-wave SC order parameter.^{30–32} As far as the second temperature derivative of the resistivity is concerned, a simple AL type analysis yields the same result as a more rigorous theoretical analysis of the fluctuation conductivity. We show this in Fig. 4(a). The excess conductivity due to superconducting fluctuation, $\Delta \sigma(T)$, can be defined as $\Delta \sigma(T)=1/\rho(T)-1/\rho_{\rm BG}(T)$ where $\rho_{\rm BG}(T)$ is the background resistivity taken as the high-*T* fit ($T>T_c$

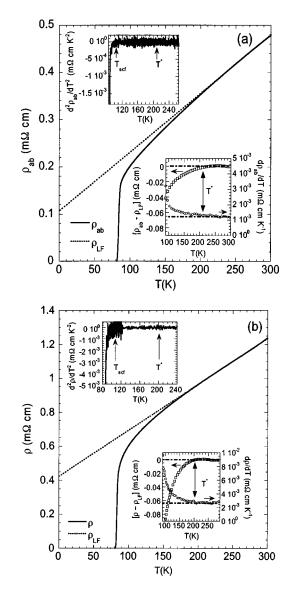


FIG. 1. Resistivity data for (a) $Y_{0.95}Ca_{0.05}Ba_2Cu_3O_{7-\delta}$ (T_c = 81.8 K, p=0.133, T_{scf} =105 K, and T^* =210 K) thin film and (b) $Y_{0.80}Ca_{0.20}Ba_2Cu_3O_{7-\delta}$ (T_c =81 K, p=0.136, T_{scf} =107 K, and T^* =202 K) sintered sample. Insets show the identification of T^* and T_{scf} . Notice that $d^2\rho(T)/dT^2$ is completely featureless at T^* . \Box denotes $[\rho(T)-\rho_{LF}]$ data and \bigcirc denotes $d\rho(T)/dT$ data. The dashed-dotted straight lines in the insets are drawn as guides to the eye.

+50 K) of the $\rho(T)$ data. The AL contribution to the fluctuation conductivity for a two-dimensional (2D) superconductor, $\Delta \sigma^{AL}$, is expressed as,²⁸ $\Delta \sigma^{AL} = [e^2/(16\hbar d)]\epsilon^{-1}$, where $\epsilon = \ln(T/T_c)$ and *d* is the periodicity of the superconducting layers ($d \sim 11.7$ Å for Y123, assuming that the CuO₂ bilayers fluctuate as one unit). In the AL analysis we have taken T_c at zero resistivity (within the noise level). Figures 4(a) and 4(b) show a comparison of the measured *T* dependence of $d^2\rho(T)/dT^2$ and that of $d^2\rho_{tot}(T)/dT^2$, where $1/\rho_{tot}(T)=1/\rho_{BG}(T)+\Delta\sigma^{AL}(T)$. We see that $d^2\rho(T)/dT^2$ and $d^2\rho_{tot}(T)/dT^2$ show remarkable agreement. In Fig. 4(a) we have also shown the theoretical $d^2\rho_{tot-Ec}(T)/dT^2$ following a rigorous analysis of the fluctuation conductivity incorporating the Lawrence-Doniach parameter and an appropriate en-

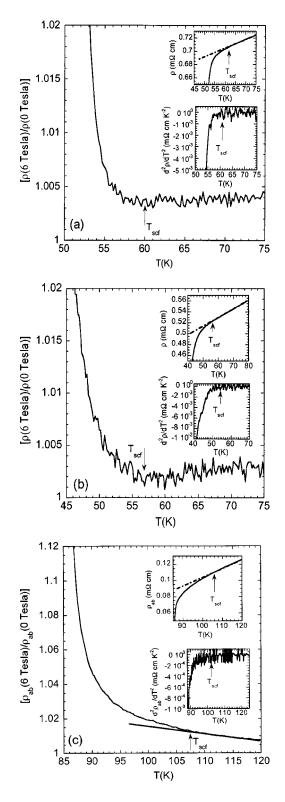


FIG. 2. Determination of $T_{\rm scf}$ from $\rho(T,H)$ data. Main panel, [$\rho(H=6 \text{ Tesla})/\rho(H=0 \text{ Tesla})$] vs *T* data. Insets, $\rho(T)$ and its linear fit (the dashed-dotted thick straight line) near T_c and $d^2\rho(T)/dT^2$ (a) $Y_{0.80}\text{Ca}_{0.20}\text{Ba}_2(\text{Cu}_{0.97}\text{Zn}_{0.03})_3\text{O}_{7-\delta}$ (sintered, p=0.185, and $T_c=45.8 \text{ K}$), (b) $Y_{0.80}\text{Ca}_{0.20}\text{Ba}_2(\text{Cu}_{0.96}\text{Zn}_{0.04})_3\text{O}_{7-\delta}$ (sintered, p=0.198, and $T_c=36.7 \text{ K}$), and (c) $Y_{0.95}\text{Ca}_{0.05}\text{Ba}_2\text{Cu}_3\text{O}_{7-\delta}$ (thin film, p=0.148 and $T_c=85.2 \text{ K}$). The straight line in the main panel of (c) is drawn as a guide to the eye.

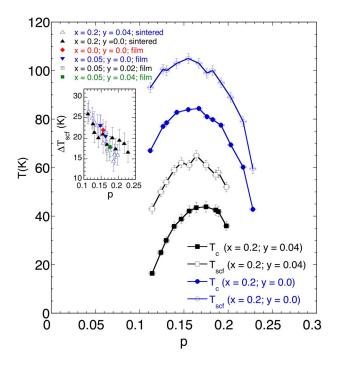


FIG. 3. (Color online) Main panel, $T_c(p)$ and $T_{scf}(p)$ for sintered $Y_{0.80}Ca_{0.20}Ba_2(Cu_{1-y}Zn_y)_3O_{7-\delta}$. Inset, $\Delta T_{scf}(p)$ for sintered and thin films of $Y_{1-x}Ca_xBa_2(Cu_{1-y}Zn_y)_3O_{7-\delta}$. Zn and Ca contents (x and y) are shown in the figure.

ergy cutoff, using the procedures adopted in Refs. 32 and 33. Here, $1/\rho_{\text{tot-Ec}}(T) = 1/\rho_{\text{BG}}(T) + \Delta\sigma^{\text{Ec}}(T)$ where, $\Delta\sigma^{\text{Ec}}(T)$ takes account of short-wavelength fluctuation effects present at high reduced-temperature regions.^{32,33} Figure 4(a) clearly illustrates that locating $T_{\rm scf}$ from $d^2\rho(T)/dT^2$ is not affected in any way by the presence of finite fluctuation conductivity at high reduced temperatures. In fact $d^2\rho(T)/dT^2$ near T_c and at higher temperatures is insensitive to different theoretical scenarios^{28,32,33} for SC fluctuations which may be discernible in $\rho(T)$ up to $3T_c$. Our aim was only to study the strong SC fluctuation near the transition temperature so that T^* can be distinguished from T_{scf} . A rigorous recent analysis of thin film data has also found the paraconductivity to be independent of the PG.³² We have also compared the amplitudes of the fluctuation conductivity for sintered and thin film samples with similar T_c (both Zn substituted and Zn-free) and found them to be almost identical. Similar results were obtained by earlier studies.34

In an effort to understand the systematic trends in ΔT_{scf} shown in the inset to Fig. 3 we have examined various terms in the second temperature derivative of $\rho_{tot}(T)$ as follows. Writing

$$\frac{1}{\rho_{\rm tot}(T)} = \frac{1}{\rho_{\rm BG}(T)} + C\varepsilon^{-1},\tag{2}$$

where $C = e^2/(16\hbar d)$, we obtain (ignoring the insignificant terms)

$$\frac{d^2 \rho_{\text{tot}}}{dT^2} \approx C \left(\frac{\rho_{\text{tot}}}{\varepsilon}\right)^2 \left(\frac{1}{T^2}\right) \left\{-\frac{2}{\varepsilon} - 1\right\}$$

Now, assuming $-\eta$ to be the threshold value of the second derivative for which we can identify a characteristic tempera-

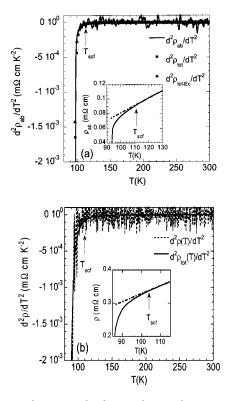


FIG. 4. (a) $d^2 \rho_{ab}(T)/dT^2$, $d^2 \rho_{tot}/dT^2$, and $d^2 \rho_{tot-Ec}/dT^2$ vs *T* for optimally doped YBa₂Cu₃O_{7- δ} thin film (T_c =92.4 K, p=0.163) and (b) $d^2 \rho(T)/dT^2$ and $d^2 \rho_{tot}/dT^2$ vs *T* of an underdoped sintered Y_{0.90}Ca_{0.10}Ba₂Cu₃O_{7- δ} sample (T_c =83 K, p=0.149). Arrows indicate T_{scf} . Insets show $\rho(T)$ and its linear fit (dashed-dotted straight line) near the superconducting transition.

ture $T=T_{scf}$ then the above equation becomes, after some rearrangements

$$\frac{\varepsilon_{\rm scf}^3}{2 + \varepsilon_{\rm scf}} T_{\rm scf}^2 \approx \frac{C\rho_{\rm scf}^2}{\eta},\tag{3}$$

where, $\varepsilon_{\rm scf} = \ln(T_{\rm scf}/T_c)$ and $\rho_{\rm scf} = \rho_{\rm tot}(T_{\rm scf})$. We have plotted the left-hand side of Eq. (3) versus experimental values of $\rho^2(T_{\rm scf})$ in Fig. 5 using the approximation $\rho_{\rm tot}(T_{\rm scf}) \sim \rho(T_{\rm scf})$. A remarkably linear trend is found and this gives further credence to our fluctuation analysis. The decreasing trend in $\Delta T_{\rm scf}$ with increasing doping shown in Fig. 3 is therefore primarily associated with the decreasing absolute resistivity as summarized by Eq. (3).

Once we have located $T_{\rm scf}$, we are in a position to look at $T^*(p)$ [below $T_{c0}(p)$] for Zn substituted samples. There is one disadvantage of Zn substitution that can hamper the identification of T^* from $\rho(T)$ measurements, namely, Zn tends to localize low-energy quasiparticles (QP) and induces an upturn in $\rho(T)$. This upturn starts at increasingly higher temperatures as p decreases and, to a lesser extent, as the Zn content increases, and thus can mask the downturn due to the PG at T^* .^{13,22} Indeed, the upturn temperature, $T_{\rm min}$, has been found to scale with T^* and is evidently also associated with the pseudogap, trending to zero as $p \rightarrow 0.19$.³⁵ In this study we have taken care of this fact by confining our attention to the $\rho(T, p)$ data for lower Zn contents in the UD region (p)

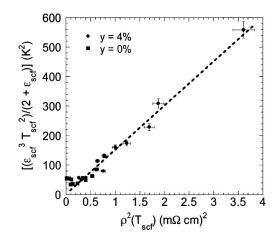
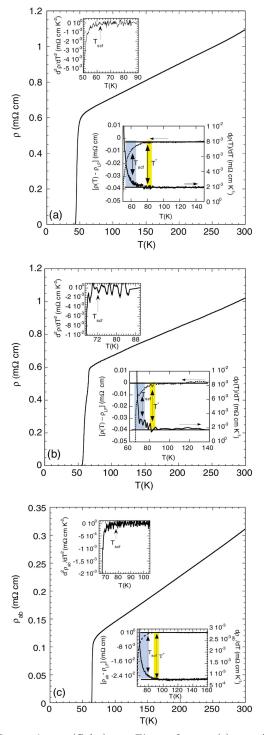


FIG. 5. $[(\varepsilon_{scf}^{3}T_{scf}^{2})/(2+\varepsilon_{scf})]$ vs $\rho^{2}(T_{scf})$ [see Eq. (3)] for the sintered $Y_{0.80}Ca_{0.20}Ba_{2}(Cu_{1-y}Zn_{y})_{3}O_{7-\delta}$ samples. Zn contents (y in %) are shown in the figure. The dashed straight line is drawn as a guide to the eye.

<0.14). Figure 6(a) shows the $\rho(T)$ data of sintered $Y_{0.80}Ca_{0.20}Ba_2(Cu_{0.96}Zn_{0.04})_3O_{7-\delta}$ for $p=0.174\pm0.003$. The insets of Fig. 6 clearly show the downturn associated with the PG at around 80 K. It should be noted that T_c of this compound is 43 K and T_{scf} =62.5 K (see Fig. 3). Figure 6(b) shows a similar analysis for sintered $Y_{0.80}Ca_{0.20}Ba_2(Cu_{0.985}Zn_{0.015})_3O_{7-\delta}$ with $p=0.177\pm0.003$. The $\rho(T)$ features of this compound once again show a T^* ~80 K with now a T_c of 54 K and T_{scf} =73 K. Considering the facts that $T^*(p)$ values are the same irrespective of Ca and Zn contents (at least for the level of substitutions used in the present study,^{17,22} and T_{c0-max} for pure Y123 is 93 K, these $T^*(p)$ are substantially below the respective $T_{co}(p)$ values (\sim 90.5 K) (see Fig. 8). A similar result is shown for $Y_{0.95}Ca_{0.05}Ba_2(Cu_{0.98}Zn_{0.02})_3O_{7-\delta}$ thin film with р =0.167±0.003, T_c =63.6 K, and T_{scf} =80 K in Fig. 6(c). Notice that $T^* = 92.5$ K of this film almost coincides with its $T_{c0}(p) \sim 92$ K. In the insets, the blue shaded parts show the range of regions of extremely strong SC fluctuations extending down from T_{scf} to T_c . For comparison the error bars on T^* are shown by the yellow shaded regions. This illustrates that these two regions can be clearly distinguished, and that they both fall significantly below T_{c0} . This study, to our knowledge, is the first instance where $T^*(p)$ has been tracked down below $T_{c0}(p)$ from any transport measurement, although of course this has effectively been done earlier by analysis of specific heat data by Loram et al.^{3,11} and NMR data by Tallon *et al.*³⁶

The field dependence of T^* and $T_{\rm scf}$ for two thin film samples with different values of p (one is UD and the other is slightly OD) are shown in Fig. 7. It is clearly seen that $T^*(p)$ is completely field independent up to 12 Tesla while $T_{\rm scf}(p)$ is strongly field dependent and shifts to lower values with increasing magnetic field just as T_c itself does. A magnetic field of 6 Tesla decreases $T_{\rm scf}$ by (5 ± 1) K for the two compounds shown in Fig. 7. This clearly relates $T_{\rm scf}$ to superconductivity and T^* to a different type of correlation. In Fig. 8 we construct a doping "phase diagram" for $Y_{1-x}Ca_xBa_2(Cu_{1-y}Zn_y)_3O_{7-\delta}$, including the PG energy scale,



 T^* FIG. (Color) 6. for (a) sintered $Y_{0.80}Ca_{0.20}Ba_2(Cu_{0.96}Zn_{0.04})_3O_{7-\delta}((p=0.174\pm0.003), (b)$ sintered $Y_{0.80}Ca_{0.20}Ba_2(Cu_{0.985}Zn_{0.015})_3O_{7-\delta}$ (p=0.177±0.003), and (c) thin film of $Y_{0.95}Ca_{0.05}Ba_2(Cu_{0.98}Zn_{0.02})_3O_{7-\delta}$ (p=0.167±0.003). Top inset shows $d^2\rho(T)/dT^2$ data. Notice that $d^2\rho(T)/dT^2$ is featureless at T^* (yellow region in the bottom inset). T_{scf} is also marked in the bottom inset. The straight lines in the bottom inset are drawn as a guide to the eye. Strong superconducting fluctuations persist in the blue region.

 $E_g(p)/k_B$, from specific heat measurements.²³ E_g is the energy scale for the PG and $E_g(p)/k_B \sim \theta T^*(p)$, where (θ

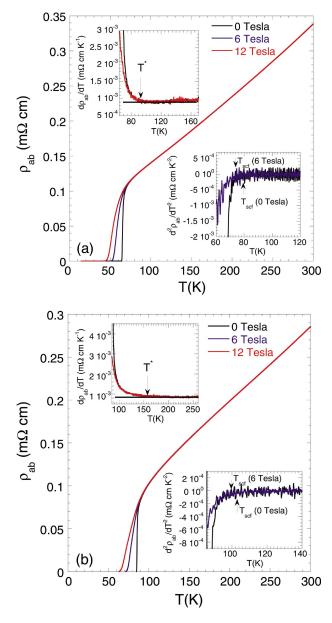


FIG. 7. (Color) Magnetic field dependence of T^* and T_{scf} . (a) A second slightly overdoped $Y_{0.95}\text{Ca}_{0.05}\text{Ba}_2(\text{Cu}_{0.98}\text{Zn}_{0.02})_3\text{O}_{7-\delta}$ thin film $(p=0.168\pm0.003 \text{ and } T_c=64 \text{ K})$ and (b) an underdoped $Y_{0.95}\text{Ca}_{0.05}\text{Ba}_2\text{Cu}_3\text{O}_{7-\delta}$ thin film $(p=0.148\pm0.003 \text{ and } T_c=85.2 \text{ K})$.

=1.3±0.1) is a certain proportionality constant.²⁶ This figure clearly shows that T^* and T_{c0} have very different doping dependence, most importantly so in the OD regions. In particular $T^*(p)$ does not merge with T_c in the OD region but falls almost linearly to zero at $p=0.19\pm0.01$, following the same trend as found for $E_g(p)/k_B$.

III. DISCUSSION

From a careful analysis of the resistivity data we have been able to track $T^*(p)$ below $T_{c0}(p)$. At this point we would like to stress once again that $T^*(p)$ values obtained from $\rho(T)$ are independent of the crystalline state and are the same for polycrystalline and single crystal samples.²² Con-

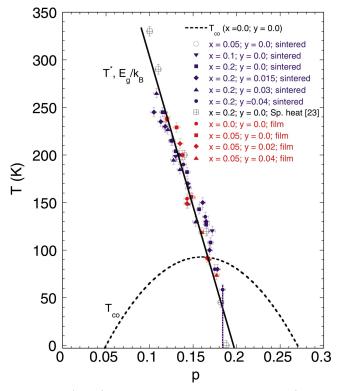


FIG. 8. (Color) Characteristic pseudogap energies (T^* and E_g/k_B) for $Y_{1-x}Ca_xBa_2(Cu_{1-y}Zn_y)_3O_{7-\delta}$: The thick black dashed line shows $T_{c0}(p)$ for pure Y123, drawn using Eq. (1) with $T_{c0-max}=93$ K. The thick black straight line is a guide to the eye. The dashed blue line represents the large asymmetric error bar associated with the data point for $Y_{0.80}Ca_{0.20}Ba_2(Cu_{0.96}Zn_{0.04})_3O_{7-\delta}$ ($p=0.184\pm0.003$) compound (see text for details).

sidering the disorder (Zn and Ca) independence of $T^*(p)$, our findings confirm that the PG exists in the SC region below T_{c0} and $T^*(p)$ does not merge with $T_c(p)$ in the OD region as proposed by various precursor pairing scenarios for PG. As explained in the present work, if one is unable to distinguish between $T^*(p)$ and $T_{scf}(p)$, one can easily be led to the wrong conclusion that $T^*(p)$ exists for p > 0.19 and merges with $T_c(p)$ on further overdoping. As T_{scf} is $\sim T_c$ +22 K near the optimum doping level, the close proximity between T_{scf} and T^* makes an accurate identification of T^* very difficult unless both T_c and thus T_{scf} are suppressed by some means that does not affect T^* . Zn serves this purpose very well. Unfortunately, we have been unable to track $T^*(p)$ for samples with p > 0.180 with enough accuracy. This is because $T^*(p)$ decreases very sharply with increasing p [compared with the decreases in $T_c(p)$ and $T_{scf}(p)$, see Figs. 3 and 8] and becomes very close to or even goes below T_{scf} and eventually $T_c(p)$.^{3,11} We discuss further the following examples (see Fig. 9): (i) $Y_{0.90}Ca_{0.10}Ba_2(Cu_{0.985}Zn_{0.015})_3O_{7-\delta}$ with $p=0.180\pm0.003$ has a $T_c = 64$ K and $\rho(T)$ is linear over 310 K to 80 K, then a marked and accelerating downturn in $\rho(T)$ starts at ~79 K. Considering the Zn and Ca independence of $\Delta T_{scf}(p)$ and $T^*(p)$, we expect T_{scf} to be $T_c + \Delta T_{\text{scf}} \approx (64+17)$ K=81 K, where $\Delta T_{\rm scf} \approx 17$ K is read off the inset of Fig. 3. So this downturn at \sim 79 K must be the onset of strong SC fluctua-

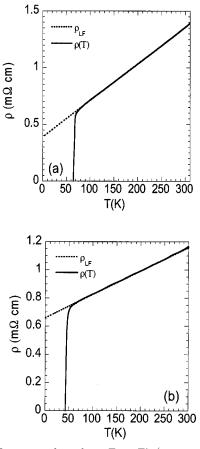


FIG. 9. Two examples where $T_{scf} \ge T^*$ (see text for details): (a) $Y_{0.90}Ca_{0.10}Ba_2(Cu_{0.985}Zn_{0.015})_3O_{7-\delta}$ ($p=0.180\pm0.003$ and T_c = 64 K) and (b) $Y_{0.80}Ca_{0.20}Ba_2(Cu_{0.96}Zn_{0.04})_3O_{7-\delta}$ ($p=0.184\pm0.003$ and $T_c=42.25$ K).

tions. Consequently T^* must lie below 79 K in this sample. (ii) Another example is $Y_{0.80}Ca_{0.20}Ba_2(Cu_{0.96}Zn_{0.04})_3O_{7-\delta}$ with $p=0.184\pm0.003$, which has $T_c=42.25$ K and $\rho(T)$ is linear over 310 K to 60 K. A marked and accelerating down-turn in $\rho(T)$ starts at ~58.5 K. We expect T_{scf} to be T_c + $\Delta T_{scf} \approx (42.25+16.5)$ K=58.75 K for this sample. Thus for this compound $T^* < 58.5$ K, much less than $T_{c0} \sim 89$ K.

 $T_{\rm scf}(p)$ has been taken as the onset temperature of strong SC fluctuations in the present study. The disorder independence of $\Delta T_{\rm scf}(p)$ (inset of Fig. 3), a simple AL-type analysis of $\rho(T)$ data (see Figs. 4 and 5), and the magnetic field dependence of $T_{\rm scf}$ strongly support this assumption. Disorder and magnetic field suppress both $T_c(p)$ and $T_{\rm scf}(p)$ in the same qualitative way, unlike $T^*(p)$ which remains unaffected.

IV. CONCLUSIONS

In summary, we have analyzed resistivity data to determine the $T^*(p)$ of $Y_{1-x}Ca_xBa_2(Cu_{1-y}Zn_y)_3O_{7-\delta}$ over a wide doping range and compositions. We have shown that $T^*(p)$ falls below $T_{c0}(p)$ in the OD side, does not merge with $T_c(p)$, and the extrapolated $T^*(p)$ becomes zero at $p=0.19\pm0.01$. We have also extracted a characteristic temperature, T_{scf} , at which strong SC fluctuations become detectable. It is perhaps surprising that T_{scf} is so well-defined experimentally in all our samples, but the very different p, H, and Zn dependence of T^* and T_{scf} points towards their different origins, e.g., T_{scf} is related to precursor SC, whereas T^* has a non-SC origin. This also agrees with a recent fluctuation conductivity study over a wide range of hole content by Curras *et al.*,³² where paraconductivities were found to be independent of the presence of the PG. Our findings support the picture proposed by Loram *et al.* based on their specific heat study that the PG vanishes at a critical doping, $p_{crit} \sim 0.19$, and coexists with SC for p < 0.19.^{11,23} Recently Krasnov *et al.* have reached similar conclusions based on their intrinsic tunneling spectroscopy studies of Bi-2212 single crystals.^{12,37}

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- ¹B. Batlogg and C. Varma, Phys. World **13**, 33 (2000); M.V. Sadovskii, Usp. Fiz. Nauk **171**, 539 (2001).
- ²T. Timusk and B. Statt, Rep. Prog. Phys. **62**, 61 (1999).
- ³J.W. Loram, K.A. Mirza, J.R. Cooper, W.Y. Liang, and J.M. Wade, J. Supercond. **7**, 243 (1994).
- ⁴V.B. Geshkenbein, L.B. Ioffe, and A.I. Larkin, Phys. Rev. B 55, 3173 (1997).
- ⁵V. Emery, S.A. Kivelson, and O. Zachar, Phys. Rev. B 56, 6120 (1997).
- ⁶J. Maly, B. Janko, and K. Levin, Phys. Rev. B 59, 1354 (1999).
- ⁷A.P. Kampf and J.R. Schrieffer, Phys. Rev. B 42, 7967 (1990).
- ⁸P. Monthoux, Phys. Rev. B **68**, 064408 (2003).
- ⁹J.J. Deisz, D.W. Hess, and J.W. Serene, Phys. Rev. Lett. **76**, 1312 (1996).
- ¹⁰ M. Langer, J. Schmalian, S. Grabowski, and K.H. Bennemann, Phys. Rev. Lett. **75**, 4508 (1995); J. Schmalian, D. Pines, and B. Stojkovic, *ibid.* **80**, 3839 (1998); A.V. Cubukov, D. Pines, and J. Schmalian, Report No. cond-mat/0201140 (unpublished).
- ¹¹J.W. Loram, K.A. Mirza, J.R. Cooper, and J.L. Tallon, J. Phys. Chem. Solids **59**, 2091 (1998).
- ¹²V.M. Krasnov, A.E. Kovalev, A. Yurgens, and D. Winkler, Phys. Rev. Lett. **86**, 2657 (2001).
- ¹³D.J.C. Walker, A.P. Mackenzie, and J.R. Cooper, Phys. Rev. B 51, 15653 (1995).
- ¹⁴K. Mizuhashi, K. Takenaka, Y. Fukuzumi, and S. Uchida, Phys. Rev. B 52, R3884 (1995).
- ¹⁵H. Alloul, P. Mendels, H. Casalta, J.F. Marucco, and J. Arabski, Phys. Rev. Lett. **67**, 3140 (1991).
- ¹⁶G.V.M. Willams and J.L. Tallon, Phys. Rev. B 57, 8696 (1998).
- ¹⁷S.H. Naqib, Ph.D. thesis, University of Cambridge, UK, 2003; S.H. Naqib, R.A. Chakalov, and J.R. Cooper, Physica C **407**, 73 (2004).
- ¹⁸S.D. Obertelli, J.R. Cooper, and J.L. Tallon, Phys. Rev. B 46, 14928 (1992).
- ¹⁹J.L. Tallon, J.R. Cooper, P.S.I.P.N. DeSilva, G.V.M. Williams, and J.L. Loram, Phys. Rev. Lett. **75**, 4114 (1995).

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- ²⁰J.L. Tallon, C. Bernhard, H. Shaked, R.L. Hitterman, and J.D. Jorgensen, Phys. Rev. B **51**, 12911 (1995).
- ²¹M.R. Presland, J.L. Tallon, R.G. Buckley, R.S. Liu, and N.E. Flower, Physica C **176**, 95 (1991).
- ²²S.H. Naqib, J.R. Cooper, J.L. Tallon, and C. Panagopoulos, Physica C **387**, 365 (2003); Report No. cond-mat/0209457 (unpublished).
- ²³J.L. Tallon, J.W. Loram, G.V.M. Williams, J.R. Cooper, I.R. Fisher, J.D. Johnson, M.P. Staines, and C. Bernhard, Phys. Status Solidi B **215**, 531 (1999).
- ²⁴C. Panagopoulos, B.D. Rainford, J.R. Cooper, W. Lo, J.L. Tallon, J.W. Loram, J. Betouras, Y.S. Wang, and C.W. Chu, Phys. Rev. B **60**, 14617 (1999).
- ²⁵C. Bernhard, J.L. Tallon, Th. Blasius, A. Golnik, and Ch. Niedermayer, Phys. Rev. Lett. 86, 1614 (2001).
- ²⁶J.L. Tallon and J.W. Loram, Physica C 349, 53 (2001).
- ²⁷I.R. Fisher and J.R. Cooper, Physica C **272**, 125 (1996).
- ²⁸L.G. Aslamazov and A.I. Larkin, Phys. Lett. **26A**, 238 (1968).
- ²⁹K. Maki and R.S. Thomson, Phys. Rev. B **39**, 2767 (1989).
- ³⁰S.-K. Yip, Phys. Rev. B **41**, 2612 (1990).
- ³¹M.V. Ramallo, A. Pomar, and F. Vidal, Phys. Rev. B **54**, 4341 (1996).
- ³²S.R. Curras, G. Forro, M.T. Gonzalez, M.V. Ramallo, M. Ruibal, J.A. Veira, P. Wagner, and F. Vidal, Phys. Rev. B 68, 094501 (2003).
- ³³C. Carballeira, S.R. Curras, J. Vina, and J.A. Viera, Phys. Rev. B 63, 144515 (2001).
- ³⁴J. Maza and F. Vidal, Phys. Rev. B **43**, 10560 (1991); J.A. Viera and F. Vidal, *ibid.* **42**, 8748 (1990).
- ³⁵J.L. Tallon, *Advances in Superconductivity*, edited by T. Yamashita and K. Tanabe (Springer, Tokyo, 2000), p. 185.
- ³⁶J.L. Tallon, G.M.V. Williams, M.P. Staines, and C. Bernhard, Physica C 235, 1821 (1994).
- ³⁷ V.M. Krasnov, Phys. Rev. B 65, 140504 (2002); V.M. Krasnov,
 A. Yugens, D. Winkler, P. Delsing, and T. Claeson, Phys. Rev. Lett. 84, 5860 (2000).