

Crystal and magnetic structure of $\text{La}_{1-x}\text{Sr}_{1+x}\text{MnO}_4$: Role of the orbital degree of freedom

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The crystal and magnetic structure of $\text{La}_{1-x}\text{Sr}_{1+x}\text{MnO}_4$ ($0 \leq x \leq 0.6$) has been studied by diffraction techniques and high resolution capacitance dilatometry. There is no evidence for a structural phase transition related to octahedron tilting like those found in isostructural cuprates or nickelates, but there are significant structural changes induced by the variation of temperature and doping which we attribute to a rearrangement of the orbital occupation.

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I. INTRODUCTION

Manganates have drawn a large interest due to the observation of a colossal magneto-resistivity in $\text{RE}_{1-x}(\text{Sr}/\text{Ca})_x\text{MnO}_3$ and in $\text{La}_{2-2x}\text{Sr}_{1+2x}\text{Mn}_2\text{O}_7$.^{1,2} The magnetoresistivity appears to result from the competition between the ferromagnetic metallic state and the ordering of charge, orbital, and magnetic degrees of freedom. The basic phase diagrams of the single layer materials $\text{La}_{1-x}\text{Sr}_{1+x}\text{MnO}_4$, of the K_2NiF_4 -structure type, have been elaborated in the work by Bao *et al.*³ and Moritomo *et al.*⁴ However, these materials have been much less studied, because in these compounds high magnetic fields of the order of 30 T are needed to induce a large magnetoresistivity.⁵ The high value of the transition field, however, indicates that the complex ordered state is particularly stable in the single layer manganates. Many techniques have been applied to characterize the order in the half-doped layered compound $\text{La}_{0.5}\text{Sr}_{1.5}\text{MnO}_4$.⁶⁻¹⁰

The composition $\text{La}_{1.0}\text{Sr}_{1.0}\text{MnO}_4$ corresponds to LaMnO_3 in the perovskite series, since all Mn-ions are three-valent. Like the perovskite LaMnO_3 , $\text{La}_{1.0}\text{Sr}_{1.0}\text{MnO}_4$ exhibits an antiferromagnetic order. Kawano *et al.* report a small ordered moment of only $0.8\mu_B$ ¹¹ compared to the value of $3.87(3)$ observed in LaMnO_3 .¹² A more recent study on $\text{La}_{1.0}\text{Sr}_{1.0}\text{MnO}_4$ finds a larger moment of $3.3\mu_B$.¹³ Through the substitution of La by divalent Sr, part of the Mn is oxidized, which may suggest physics similar to that of the perovskite materials based on the Zener double exchange mechanism. However, in the $\text{La}_{1-x}\text{Sr}_{1+x}\text{MnO}_4$ series there is no metallic ferromagnetic phase at ambient conditions. Instead, the charges, the orbital degrees of freedom and the spins order in closely coupled patterns. In this sense the $\text{La}_{1-x}\text{Sr}_{1+x}\text{MnO}_4$ phase diagram resembles that of $\text{Pr}_{1-x}\text{Ca}_x\text{MnO}_3$ where there is no metallic phase at zero magnetic field neither. Upon increasing the Sr content in $\text{La}_{1-x}\text{Sr}_{1+x}\text{MnO}_4$ the commensurate antiferromagnetic order disappears near $x=0.15$ ^{3,4} and charge order appears for x

larger than 0.4.^{4,10} For the intermediate concentrations spin-glass behavior is observed.⁴

Concerning the crystal structure, so far only the lattice parameters were determined as a function of doping.³ Upon introduction of Sr, the c parameter decreases, whereas the in-plane parameter a increases. We have performed a systematic study of the crystal and magnetic structure of $\text{La}_{1-x}\text{Sr}_{1+x}\text{MnO}_4$ by neutron and by x-ray diffraction techniques combined with measurements of the thermal expansion. We find strong evidence of an orbital rearrangement, both as a function of temperature and as a function of doping.

II. EXPERIMENT

Single crystals of $\text{La}_{1-x}\text{Sr}_{1+x}\text{MnO}_4$ were grown as described elsewhere.¹⁴ Powder samples were obtained by crushing single crystalline samples with a poor mosaic spread. Magnetic properties of the samples were reported by Baumann *et al.*¹⁵ The crystal and magnetic structure of $\text{La}_{1-x}\text{Sr}_{1+x}\text{MnO}_4$ was studied by powder and by single crystal diffraction using neutrons and x-rays. All neutron experiments were performed on instruments of the Laboratoire Léon Brillouin installed at the Orphée reactor in Saclay. The crystal and magnetic structure of single crystals of composition $\text{La}_{1.0}\text{Sr}_{1.0}\text{MnO}_4$ and $\text{La}_{0.5}\text{Sr}_{1.5}\text{MnO}_4$ were studied on the neutron four-circle diffractometer 5C.2 at room temperature and at ~ 20 K. Powder neutron diffraction experiments on samples of $\text{La}_{1-x}\text{Sr}_{1+x}\text{MnO}_4$ with $x=0.125, 0.25, 0.4,$ and 0.6 were performed on the high resolution neutron diffractometer 3T.2 using a wavelength of 1.23 \AA . The temperature dependence of the lattice parameters was studied by powder x-ray diffraction using $\text{Cu } K_\alpha$ radiation and by measurements of the thermal expansion coefficients with a high resolution capacitance dilatometer on single crystalline samples. All diffraction patterns were analyzed by the Rietveld method using the FULLPROF program.¹⁶ Typical diffraction patterns, together with the profile-fitted description, are

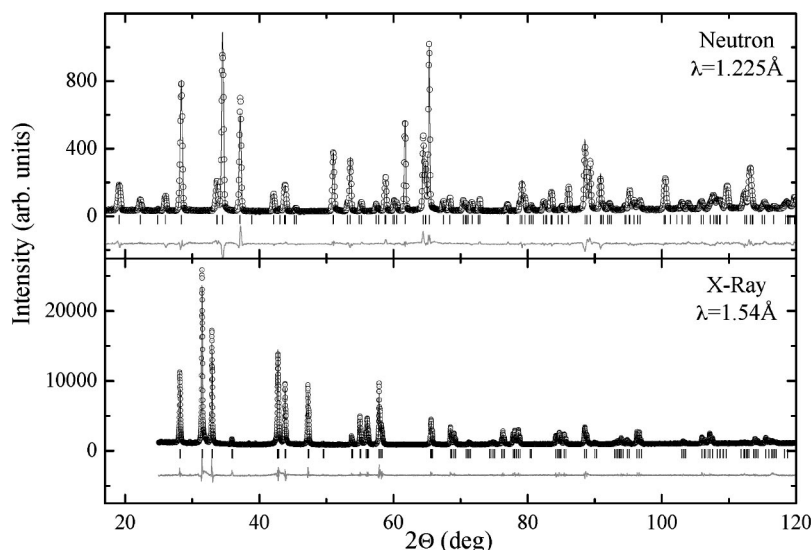


FIG. 1. Typical diffraction patterns obtained for the layered manganates: (top) a neutron powder diffraction pattern measured on the 3T.2 diffractometer for $\text{La}_{0.75}\text{Sr}_{1.25}\text{MnO}_4$; (bottom) an x-ray diffraction pattern determined on a D5000-diffractometer. Points designate the intensity data, dark lines the profile fits, and fine lines the intensity differences. Vertical bars indicate the Bragg positions.

shown in Fig. 1. The temperature dependence of magnetic superstructure peaks was determined using the 4F triple-axis spectrometers.

III. MAGNETIC AND CRYSTAL STRUCTURE OF LOW-DOPED SAMPLES WITH COMMENSURATE ANTIFERROMAGNETIC ORDER

$\text{La}_{1.0}\text{Sr}_{1.0}\text{MnO}_4$ exhibits antiferromagnetic order with the magnetic moments aligned along the c axis.¹¹ Using a small single crystal of $\sim 10 \text{ mm}^3$ volume a set of 1100 Bragg reflection intensities was collected on the four-circle diffractometer 5C.2 in respect to the $I4/mmm$ lattice (dimension $3.8 \times 3.8 \times 12 \text{ \AA}^3$). At room temperature, no evidence for superstructure reflections was found indicating that the K_2NiF_4 -structure remains undistorted in this material. At low temperature, superstructure intensities were found at positions $(h/2k/2l)$ in respect with the $I4/mmm$ lattice which agree with the antiferromagnetic order reported previously.^{4,11,13} A set of 152 magnetic Bragg reflection intensities and 965 fundamental intensities was recorded at 20 K. Due to the high crystal quality, extinction effects turned out to be important in this experiment and that on the composition $\text{La}_{0.5}\text{Sr}_{1.5}\text{MnO}_4$; see below. Therefore, we have corrected the data for extinction using the Becker-Coppens formalism for an anisotropic mosaic spread. It is important to take into account the facts that the reliability of the determination of the strong Bragg intensities is hampered by the extinction and that the multiple diffraction may add intensity to weak reflections. Therefore, we have enhanced the statistical errors σ_{stat} by

$$\sigma^2 = \sigma_{stat}^2 + \text{Int} \cdot * c_{mult} + c_{const}$$

using the observed intensity Int and two additional constants c_{mult} and c_{const} , following the procedure described in detail for a similar problem.¹⁷ In addition, reflections with an ex-

inction correction larger than 0.75 were excluded from the refinements. Using this procedure we obtain a satisfying description of the Bragg reflections. The sets of 1037(871) reflections taken at room temperature ($T=20 \text{ K}$) are described with reliability factors of $R_w(F^2)=4.71\%$ (3.16%) and $R_{unw}(F^2)=3.99\%$ (3.56%). The structural results are given in Table I.

In the K_2NiF_4 -structure of space group $I4/mmm$, there is an intrinsic difference between the antiferromagnetic ordering of spins oriented along the c direction or oriented along an in-plane direction. If the spins are oriented within the plane, the phase of the order in the neighboring plane leads to two symmetrically different phases usually called La_2CuO_4 and La_2NiO_4 -type. In contrast the different phases of ordering along the c direction yield two domains of the same symmetry if the ordered moment points along the c direction. The three-dimensional coupling in $\text{La}_{1.0}\text{Sr}_{1.0}\text{MnO}_4$ arises from the coupling with the next-nearest layer, which is ferromagnetic and rather weak; the two magnetic domains are drawn in Fig. 2.

The collected magnetic Bragg intensities correspond to a superposition of contributions of both domains. The refinement of the magnetic structure was performed with the FULLPROF program using the magnetic form factor of Mn^{3+} and the crystal structure determined above. The two-domain structure was taken into account. We obtain an ordered magnetic moment of $3.21(3)\mu_B$ in strong disagreement with a first report,¹¹ but in good agreement with more recent work.¹³ The difference is most likely due to a poor sample quality of the crystal studied first. Our sample exhibits the antiferromagnetic ordering at $T_N=127 \text{ K}$, in good agreement with the study by Moritomo *et al.*⁴ Attempts to refine an additional ordered moment component, aligned perpendicular to the c axis, did not yield a significant value. The ordered moment is still lower than what is expected for a Mn^{3+} -ion without an orbital contribution, i.e., $4\mu_B$, and it is still lower than the ordered moment reported for LaMnO_3 .¹² A part of

TABLE I. Results of the single crystal and powder neutron diffraction experiments on $\text{La}_{1-x}\text{Sr}_{1+x}\text{MnO}_4$ for room (RT) and low temperature (LT); the O1 position is at (0,0.5,0).

			$x=0^a$	$x=0.125^b$	$x=0.25^b$	$x=0.4^b$	$x=0.5^a$	$x=0.6^b$
a (Å)		RT	3.786	3.814	3.846	3.857	3.863	3.857
		LT	3.768	3.794	3.840	3.852	3.855	3.857
c (Å)		RT	13.163	12.938	12.676	12.548	12.421	12.402
		LT	13.195	12.985	12.651	12.524	12.397	12.405
V (Å) ³		RT	188.676	188.204	187.500	186.670	185.356	184.498
		LT	187.348	186.879	186.588	185.855	184.257	184.555
Mn	U_{iso} (10^{-4} Å ²)	RT	13(3)/68(3) ^c	15(6)	18(6)	5(4)	19(2)/30(3) ^c	28(5)
		LT	9(3)/30(5) ^c	5(4)	29(7)	10 ^d	5(3)/5(3) ^c	6(4)
La/Sr	z	RT	0.35598(2)	0.35688(8)	0.35815(9)	0.35917(11)	0.35816(3)	0.35799(8)
		LT	0.35564(2)	0.35699(8)	0.35779(9)	0.35864(15)	0.35814(4)	0.35786(7)
	U_{iso} (10^{-4} Å ²)	RT	46(8)/38(8) ^c	34(3)	57(3)	63(3)	40(1)/30(1) ^c	7(4)
		LT	19(1)/17(2) ^c	18(3)	41(3)	10(4)	5(3)/5(3) ^c	6(2)
O(1)	U_{11} (10^{-4} Å ²)	RT	31(1)	46(8)	63(9)	122(9)	62(2)	67(6)
		LT	27(2)	63(8)	83(9)	43(8)	36(3)	54(6)
	U_{22} (10^{-4} Å ²)	RT	71(1)	46(7)	75(8)	65(8)	69(2)	47(6)
		LT	45(2)	24(8)	87(8)	97(8)	68(3)	66(6)
	U_{33} (10^{-4} Å ²)	RT	100(2)	74(8)	55(11)	94(11)	67(2)	87(7)
		LT	58(3)	62(8)	55(10)	6(10)	27(4)	42(6)
O(2)	z	RT	0.17221(3)	0.16900(11)	0.16494(17)	0.16386(15)	0.16106(8)	0.15966(10)
		LT	0.17270(4)	0.17023(11)	0.16449(15)	0.16260(19)	0.16138(10)	0.15968(9)
	U_{11} (10^{-4} Å ²)	RT	179(2)	191(7)	167(15)	153(7)	117(2)	93(4)
		LT	131(2)	86(6)	131(7)	134(9)	70(2)	63(4)
	U_{33} (10^{-4} Å ²)	RT	74(2)	163(9)	191(11)	137(10)	68(2)	61(7)
		LT	51(3)	180(9)	185(10)	56(8)	52(3)	33(6)
Bond length	$d_{\text{Mn-O}(1)}$ (Å)	RT	1.89335(1)	1.90714(3)	1.92329(4)	1.92903(4)	1.93164(3)	1.92853(3)
		LT	1.88403(8)	1.89684(4)	1.92018(4)	1.92615(5)	1.92763(6)	1.92859(3)
	$d_{\text{Mn-O}(2)}$ (Å)	RT	2.2668(4)	2.1873(14)	2.0915(21)	2.0574(19)	2.0005(9)	1.9805(10)
		LT	2.2788(5)	2.2104(14)	2.0810(19)	2.0360(24)	2.0006(13)	1.9808(5)

^aRefinement of single-crystal data.^bRefinement of powder-crystal data.^cThe single-crystal data refinement allows one to determine anisotropic thermal parameters on the Mn and La/Sr sites, $U_{11}=U_{22}$ and U_{33} .^dThis value was fixed during the refinement since it tended to negative values.

the moment reduction may be attributed to the two-dimensional character of the antiferromagnetic coupling in $\text{La}_{1.0}\text{Sr}_{1.0}\text{MnO}_4$, but this reduction gives only $\Delta m=0.22\mu_B$. However, there is evidence that part of the magnetic order in $\text{La}_{1.0}\text{Sr}_{1.0}\text{MnO}_4$ remains two-dimensional in character and does not transform into the 3-dimensional Bragg intensities.¹⁸

The crystal and magnetic structure of the $\text{La}_{0.875}\text{Sr}_{1.125}\text{MnO}_4$ sample was studied by powder neutron diffraction. We find the same magnetic superstructure peaks and obtain an ordered magnetic moment of $2.4(1)\mu_B$. The further reduction of the moment compared to pure $\text{La}_{1.0}\text{Sr}_{1.0}\text{MnO}_4$ agrees with the reduced Néel temperature, $T_N \sim 62$ K, and the finding that, for this composition, the amount of two-dimensional diffuse magnetic scattering is further enhanced.¹⁸

The related structural phase diagrams of $\text{La}_{2-x}\text{Sr}_x\text{CuO}_4$, $\text{La}_{2-x}\text{Sr}_x\text{NiO}_4$, and $\text{La}_{2-x}\text{Sr}_x\text{CoO}_4$ exhibit all the same type of

structural phase transition related to the tilting of the octahedra around an in-plane axis. We do not find such a long range tilt distortion in the $\text{La}_{1-x}\text{Sr}_{1+x}\text{MnO}_4$ samples studied here. However, the tilt transitions vanish for relatively small Sr concentrations in all systems. The critical Sr concentration amounts to $x \sim 0.2$ in the cuprates,¹⁹ nickelates,²⁰ and cobaltates,^{21,22} i.e., much below the concentration range studied here (note that the common notation in the manganates is different). Therefore, it is not astonishing that the $\text{La}_{1-x}\text{Sr}_{1+x}\text{MnO}_4$ samples studied here do not exhibit the tilt distortion.

The most interesting structural aspect concerns the elongation of the MnO_6 octahedron. Three-valent Mn possesses four electrons in the 3d shell with one occupying the e_g orbitals; therefore Mn^{3+} is strongly Jahn-Teller active, and one expects a strong octahedron elongation. In the layered material there is a competition between the crystal field of the Mn planar structure favoring an elongation along the c

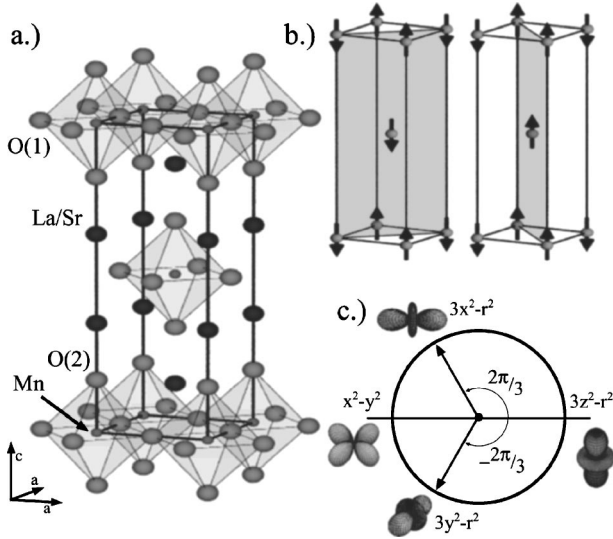


FIG. 2. (a) Crystal structure of $\text{La}_{1.0}\text{Sr}_{1.0}\text{MnO}_4$ at room temperature; (b) antiferromagnetic structure of $\text{La}_{1.0}\text{Sr}_{1.0}\text{MnO}_4$; the two domains are shown (the ferromagnetic planes are shaded for the two domains); (c) schematic drawing of the e_g orbital state in the pseudospin space. The angles $\theta = -2\pi/3, \pi, 2\pi/3,$ and 0 correspond to the $3y^2-r^2, x^2-y^2, 3x^2-r^2,$ and $3z^2-r^2$ orbitals, respectively.

axis, like in La_2CuO_4 or in La_2NiO_4 (note that the octahedron in the nickelates is elongated though Ni^{2+} is not a Jahn-Teller ion), and the exchange energy.²³ In order to allow for the virtual hopping of e_g electrons to their nearest neighbors in the a, b plane, it is preferable to have e_g orbitals occupied with the lobes within this plane. Furthermore, the elastic interactions will favor an arrangement such that the long axis of one octahedron is pointing towards the short axis of a neighboring site.²⁴ A similar arrangement is realized in the perovskite LaMnO_3 in the a, b planes of the $Pbnm$ crystal structure,²⁵ where it causes a ferromagnetic coupling.¹² In the layered manganate $\text{La}_{1.0}\text{Sr}_{1.0}\text{MnO}_4$, the crystal field appears to overrule these mechanisms, since the elongation axes point along the c direction. The Mn-O bond distances at room temperature amount to $2 \times 2.267 \text{ \AA} + 4 \times 1.893 \text{ \AA}$ compared to $2 \times 1.968 \text{ \AA} + 2 \times 1.907 \text{ \AA} + 2 \times 2.178 \text{ \AA}$ for the perovskite LaMnO_3 .²⁵ The difference of the Mn-O distances is even stronger in the layered material than in LaMnO_3 due to the combined effect of the Jahn-Teller effect and the K_2NiF_4 -structure crystal field. The ferro-orbital ordering of the e_g orbital along the c direction agrees with the antiferromagnetic ordering of the spins within the planes.

In view of the large splitting in the bond distances, one might expect a complete orbital ordering and a stable crystal structure. However, we find that the bond distances vary significantly with temperature (see Table I). The Mn-O2 to Mn-O1 distance ratio increases from 1.197(2) to 1.209(3), upon cooling from room temperature to 20 K. Further insight is obtained from the analysis of the lattice constants as a function of temperature; see Fig. 3. Both the determination of the lattice parameters by x-ray powder diffraction and the measurement of the thermal expansion coefficient with a high resolution capacitance dilatometer, yield large and anisotropic anomalies around the Néel ordering. Upon cooling,

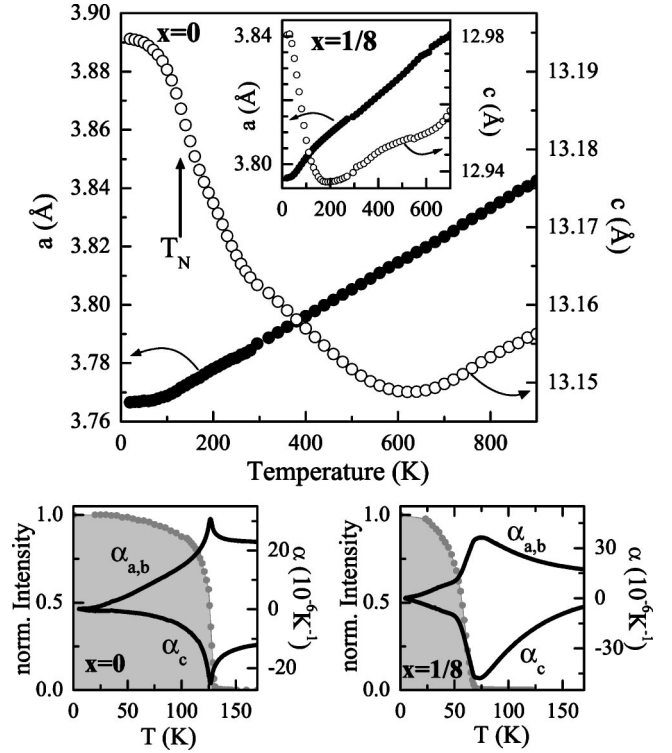


FIG. 3. (Upper panel) Temperature dependence of the lattice constants in $\text{La}_{1.0}\text{Sr}_{1.0}\text{MnO}_4$ as a function of temperature (the arrow indicates the Néel temperature). The inset shows the temperature dependence of the lattice constants in $\text{La}_{0.875}\text{Sr}_{1.125}\text{MnO}_4$ where Néel order occurs at ~ 62 K. (Lower panel) Thermal expansion coefficient (parallel and perpendicular to the MnO_2 layers) determined by capacitance dilatometry and the normalized intensity of the antiferromagnetic superstructure reflection (0.5 0.5 0) measured by neutron diffraction for $\text{La}_{1.0}\text{Sr}_{1.0}\text{MnO}_4$ (left) and for $\text{La}_{0.875}\text{Sr}_{1.125}\text{MnO}_4$ (right).

the c axis expands while the in-plane parameters shrink, in agreement with the observation of the more elongated MnO_6 -octahedron at low temperature. The thermal expansion coefficients, corresponding to the temperature derivatives of the lattice constants, exhibit a λ -like anomaly at the Néel temperature. In the antiferromagnetically ordered phase, the c axis expansion and the in-plane shrinking are less pronounced. The clear coupling between the structural effect and the magnetic ordering indicates an orbital origin. Most likely the orbital order is not complete in $\text{La}_{1.0}\text{Sr}_{1.0}\text{MnO}_4$, but some of the in-plane e_g orbitals are still occupied. The anisotropic thermal expansion indicates that there is a change in the orbital occupation. Upon cooling, less in-plane orbitals are occupied. The temperature dependence of the lattice parameters suggests that the orbital rearrangement is spread over a wide temperature interval, 10–600 K. Only at very high temperatures does the c axis exhibit a normal, positive thermal expansion. Since we do not find a change of the crystal symmetry, this structural variation should not be associated with a phase transition.

In Fig. 2 the e_g orbitals relevant in $\text{La}_{1.0}\text{Sr}_{1.0}\text{MnO}_4$ are depicted. The mixing of in-plane and out-of-plane e_g orbitals has been modeled by a combination of the $3z^2-r^2$ and x^2-y^2 orbitals,^{23,26}

$$|\theta\rangle = \cos(\theta/2)|3z^2 - r^2\rangle + \sin(\theta/2)|x^2 - y^2\rangle.$$

The $x^2 - y^2$ and $3z^2 - r^2$ orbitals correspond to the coefficients $\theta = \pi$ and $\theta = 0$, respectively. In-plane elongated e_g -orbitals correspond to values of $\theta = \pm 2\pi/3$. With such a linear combination one assumes at least a local distortion of the tetragonal symmetry. The value of θ may fluctuate around the average value of $\theta = 0$ corresponding to the $3z^2 - r^2$ -orbital. However, at the moment there is no evidence for such symmetry reduction. The orbital occupation in $\text{La}_{1.0}\text{Sr}_{1.0}\text{MnO}_4$ may eventually correspond to a linear combination with imaginary coefficients corresponding to an admixture of orbitals which would not break the local symmetry.²⁷

The second sample presenting an antiferromagnetic ordering, $\text{La}_{0.875}\text{Sr}_{1.125}\text{MnO}_4$, exhibits qualitatively similar anomalies in the lattice constants. However, the anomalous c axis expansion and the in-plane shrinking occur over a smaller temperature interval upon cooling. The c parameter of $\text{La}_{0.875}\text{Sr}_{1.125}\text{MnO}_4$ increases by the same amount between 200 and 10 K, as does the c axis in $\text{La}_{1.0}\text{Sr}_{1.0}\text{MnO}_4$ between 600 and 10 K. Concerning the effects in the in-plane parameters, it is less obvious that one may separate the electronically-induced effects from the normal thermal expansion, as they have the same sign. The thermal expansion temperature dependence, see lower part of Fig. 3, again shows the maxima just at the Néel temperature and a reduced effect in the ordered phase. So, both compositions exhibit a continuous elongation of the MnO_6 octahedron when approaching the Néel order from high temperature and this tendency gets stopped in the ordered state.

Doping $\text{La}_{1.0}\text{Sr}_{1.0}\text{MnO}_4$ with additional Sr induces Mn^{4+} sites where no e_g -orbitals are occupied. Such Mn^{4+} sites may attract the e_g orbitals of neighboring Mn^{3+} sites, and the concept of an orbital polaron has been proposed.²⁸ In the layered material the doping-induced Mn^{4+} sites will destabilize the ordering of the e_g orbitals elongated along the c direction and—simultaneously—the antiferromagnetic order. The balance between the crystal field preferring a c elongation of the e_g orbitals and the kinetic energy favoring an orientation of the orbitals along the planes, is shifted towards the in-plane orientation through the doping, since, only in this case, the e_g electron may easily hop. Therefore, compared to $\text{La}_{1.0}\text{Sr}_{1.0}\text{MnO}_4$, there seems to be more in-plane elongated e_g states occupied in $\text{La}_{0.875}\text{Sr}_{1.125}\text{MnO}_4$ at high temperature. The increase of the magnetic correlations induced upon cooling leads then to a redistribution of electrons which is stronger than in $\text{La}_{1.0}\text{Sr}_{1.0}\text{MnO}_4$.

The interpretation of the structural anomalies by an orbital redistribution is supported by recent x-ray absorption measurements which find a finite and temperature-dependent occupation of the e_g -orbital oriented in the planes.²⁹

The anomalies in the thermal expansion suggesting a rearrangement of the orbital occupation induced by the magnetic interaction resemble recent observations on LaTiO_3 ^{30,31} and on Ca_2RuO_4 .³² In these compounds, one finds an enhancement of an octahedron distortion upon approach of the Néel order. In the case of the ruthenate the accompanying change in the orbital occupation has been directly observed by x-ray absorption spectroscopy.³³ As in $\text{La}_{1.0}\text{Sr}_{1.0}\text{MnO}_4$,

the octahedron distortion implies a stronger crystal field splitting of the orbital energies in these t_{2g} systems. Observing a similar effect in the strongly Jahn-Teller active e_g -system $\text{La}_{1.0}\text{Sr}_{1.0}\text{MnO}_4$ appears more astonishing.

IV. AVERAGE CRYSTAL STRUCTURE FOR INTERMEDIATE DOPING

Crystal structure. Upon further increase of the Sr content the commensurate antiferromagnetic order, as illustrated in Fig. 2, disappears, and rather complex schemes of charge, orbital and magnetic order appear, among which that of the half-doped composition is best studied.⁶⁻¹⁰ The single crystal of composition $\text{La}_{0.75}\text{Sr}_{1.25}\text{MnO}_4$ studied here does not only show the superstructure peaks indexed by $(h/2k/2l)$ but magnetic scattering at an incommensurate \mathbf{q} -position $(0.5, 0.16, 0)$, which one may not relate with the simple commensurate magnetic ordering.¹⁸ With the diffraction techniques used here, we are not able to study the structural distortion arising from the charge and orbital order, we may only discuss the average crystal structure. A single crystal of $\text{La}_{0.5}\text{Sr}_{1.5}\text{MnO}_4$ was studied on the four-circle diffractometer. As extinction effects were again severe, we used the same methods as for the $\text{La}_{1.0}\text{Sr}_{1.0}\text{MnO}_4$ sample. The data sets of 1030(554) reflections taken at room temperature ($T=25$ K) are described with reliability factors of $R_w(F^2) = 5.65\%$ (6.24%) and $R_{int}(F^2) = 5.65\%$ (6.07%). The structural results are given in Table I.

The lattice constants, see Fig. 4(a), show that the trend already discussed continues with a further increase of the Sr content up to $x=0.6$. The c parameter shrinks and the in-plane parameter slightly expands. Simultaneously, the Mn-O2 bond shrinks and the Mn-O1 bond expands [see Figs. 4(b) and 4(c)]. The increase of the Sr content corresponds to an oxidation of the Mn sites or to an increase in the number of Mn^{4+} sites, which, in first view, should lead to shorter average Mn-O bond distances. The Sr dependence of the lattice constants and bond lengths indicates that this effect superposes with the one discussed above: the continuing change in the orbital occupation at the Mn^{3+} sites. For $x=0$ mainly the c -elongated orbitals are occupied, whereas, for $x=0.5$, there is a strong occupation of the in-plane components. The reminiscent elongation of the MnO_6 octahedra along the c -direction at high Sr-concentrations can be understood due to the remaining influence of the planar crystal structure. Note that the non-Jahn-Teller configuration $3d^8$ in La_2NiO_4 still exhibits an octahedron elongation due to the purely structural effects.³⁵ The temperature dependence of the bond distance ratio [see Fig. 4(c)] shows that the enhancement of the octahedron elongation at low temperature occurs only in the low-doped compounds which exhibit the commensurate antiferromagnetic order.

The change of the Mn-O bond distances may be analyzed quantitatively using the bond valence sum formalism:

$$V_{Mn} = \sum_i \exp\left(\frac{d_0 - d_i(\text{Mn}-\text{O})}{B}\right),$$

where $d_0 = (1-x) \times 1.76 \text{ \AA} + x \times 1.753 \text{ \AA}$ and $B = 0.37 \text{ \AA}$ are empirical parameters³⁶ and $d_i(\text{Mn}-\text{O})$ denotes the six Mn-O

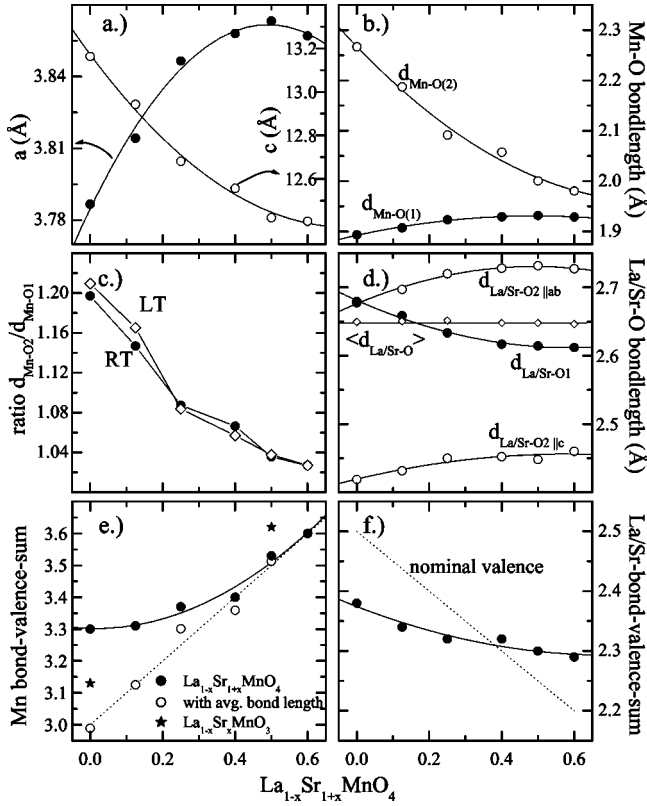


FIG. 4. Doping dependence of the lattice parameters (a), the Mn-O bond lengths (b), the ratio of the two Mn-O bond lengths (c), and the three distinct bond lengths in the (La,Sr)O₉ polyhedron (d) for the series La_{1-x}Sr_{1+x}MnO₄. Calculated bond-valence sum for the central Mn-ion (e) [closed circles denote the values obtained with the split bond lengths, open circles those calculated with the average Mn-O bond length and stars values calculated for LaMnO₃ (Ref. 25) and La_{0.5}Sr_{0.5}MnO₃ (Ref. 34)], and bond valence sums for the La/Sr place (f). Drawn lines are guides to the eye. All data are taken at room temperature if not specified.

bond distances. The Mn bond valence sums are shown in Fig. 4(e); only for Sr content higher than $x=0.4$ one does find the nominal valence, suggesting that for low doping the crystal structure exhibits an internal stress similar to the isostructural cuprates.³⁷ The internal stress seems to be released with further doping as the calculated bond valence sum approaches the nominal value. In Fig. 4(e) we also show the

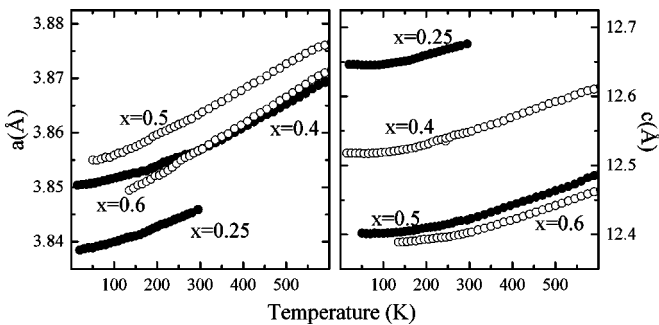


FIG. 5. Temperature dependence of the lattice parameters in La_{1-x}Sr_{1+x}MnO₄ for concentrations $x \geq 0.25$.

bond-valence sums calculated with the averaged Mn-O distances, which follow the nominal values. It seems that part of the internal stress is related to the strong splitting of the Mn-O bond distances, which is probably insufficiently described with the bond valence scenario. For comparison we also show the bond valence sums obtained for the perovskites La_{1-x}Sr_xMnO₃ which are quite close to the nominal values for $x=0$ and 0.5 .^{25,34}

The La-O coordination polyhedron, too, changes significantly with Sr content, in particular the shortest La-O2 bond, parallel to the c axis, which increases rapidly with doping in the range $0 \leq x < 0.4$; see Fig. 4(d). The analysis of the bond valence sums calculated via

$$V_{La/Sr} = (0.5 - 0.5x) \times \sum_i \exp\left(\frac{d_{0-La} - d_i(La/Sr - O)}{B}\right) + (0.5 + 0.5x) \times \sum_i \exp\left(\frac{d_{0-Sr} - d_i(La/Sr - O)}{B}\right),$$

is shown in Fig. 4(f). At low doping, the obtained values are significantly lower than the expected values, confirming the interpretation that there is an internal strain in this material. This strain is reduced with doping and even turns into the opposite as the bond valence sums are larger than the expected values in the samples with Sr concentration of 0.4 or higher.

Temperature dependence of the lattice parameters. The temperature dependencies of the lattice constants for large doping are shown in Fig. 5. In contrast to the anomalous behavior found for the low-doped samples, the thermal expansion is quite normal in this Sr-concentration range. The complex charge, orbital and magnetic ordering schemes occurring in these samples does not imply structural changes similar to those induced through the commensurate antiferromagnetic order.

Debye Waller factors. Our interpretation of the doping-induced reorientation of the e_g orbitals is well supported by the anisotropic atomic displacement parameters (ADP). Due to the intrinsic disorder induced by the occupation of the same site by La and Sr, some of the displacement parameters are significantly higher than one would expect from the phonon contributions. The disorder effect has been studied in detail for La_{2-x}Sr_xCuO₄, where, however, only 8% of the La are replaced by Sr.³⁸ In La_{1-x}Sr_{1+x}MnO₄ the doping disorder is maximal for La_{1.0}Sr_{1.0}MnO₄ and decreases with further doping. Since the (La/Sr)-O bonds are perpendicular to the Mn-O bonds, the doping disorder will displace the O-atoms mainly perpendicular to the Mn-O bonds. The U_{11} parameter of O2 and the U_{33} parameter of O1 are most sensitive. These displacement parameters are indeed maximal for $x=0$ and decrease with increasing Sr content. The interatomic interaction potentials should be similar to those in La_{2-x}Sr_xCuO₄; therefore one may compare to the additional ADP obtained for the cuprate system.³⁸ Extrapolating the disorder induced ADP contributions calculated for La_{2-x}Sr_xCuO₄ to $x=1.0$ one obtains $\Delta U_{11}(O2)=0.023 \text{ \AA}^2$ and $\Delta U_{33}(O1)=0.0047 \text{ \AA}^2$ and these values are of the order of the low temperature ADP's measured in La_{1.0}Sr_{1.0}MnO₄. The manganese seems to be less

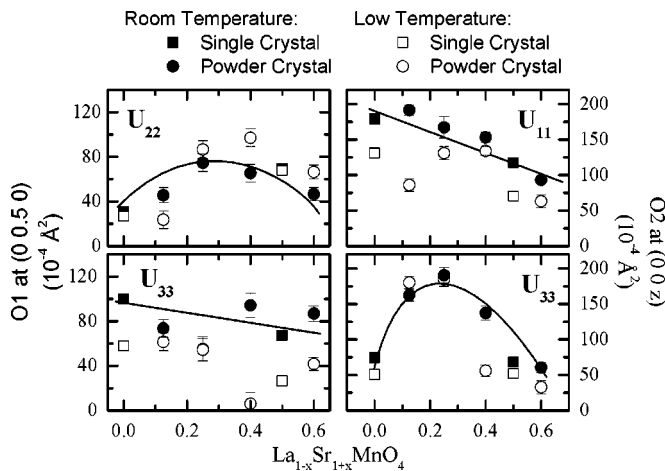


FIG. 6. Anisotropic displacement parameters of the two oxygen sites as function of temperature and doping. Note that $U_{22}(O1)$ and $U_{33}(O2)$ correspond to the displacement of the oxygen parallel to the bonds.

sensitive to the disorder most likely due to the more stable character of its crystal structure. In the $La_{1-x}Sr_xMnO_4$ phase diagram there are no structural phase transitions, whereas $La_{2-x}Sr_xCuO_4$ exhibits the octahedron tilt instabilities.

The orbital occupation is related to the atomic displacement parameters parallel to the bonds, U_{33} of O2 and the U_{22} parameter of O1. These parameters are smaller in pure $La_{1.0}Sr_{1.0}MnO_4$ than in intermediate Sr concentrations, although the doping disorder is maximal for that composition. U_{33} of O2 exhibits a maximum around $x=0.25$ where the orbital orientation is less defined. This concentration roughly represents the border between the majority occupation of c -axis elongated orbitals around $La_{1.0}Sr_{1.0}MnO_4$ and the ma-

jority occupation of in-plane elongated orbitals at higher doping. (See Fig. 6.)

The longitudinal displacement parameter of the in-plane oxygen is maximal near half-doping.³⁹ This finding agrees well with the model of the CE-type ordering of both the Mn^{3+}/Mn^{4+} sites and orientation of the e_g orbitals.^{6,7,10} Both displacements have not been taken into account in our refinement of the average structure. The doping dependence of the U_{22} -O1 parameter suggests local displacements of the order of 0.1 Å near half doping.

V. CONCLUSIONS

The crystal and magnetic structure of $La_{1-x}Sr_xMnO_4$ was studied by several diffraction techniques and high resolution capacitance dilatometry. At low doping $0.0 \leq x \leq 0.25$, the MnO_6 octahedra are elongated along the c direction in accordance with a majority occupation of the $3z^2-r^2$ orbital. However, pronounced and anisotropic anomalies in the thermal expansion yield evidence that the orbital occupation varies with temperature. Upon cooling there seems to occur a shift of electrons towards the $3z^2-r^2$ orbitals. Upon increase of the Sr concentration, the amount of Mn^{3+} ions is reduced and the octahedron elongation along the c direction diminishes. This effect has to be attributed to a majority occupation of the in-plane elongated e_g orbitals, as it was predicted to arise from the magnetic coupling between Mn^{3+} and Mn^{4+} sites.²⁸

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