

Eu³⁺ luminescence in Eu-doped KMgF₃ crystals investigated by site-selective laser-excitation spectroscopy

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Luminescence properties of Eu³⁺ ions in KMgF₃ are investigated by site-selective laser-excitation spectroscopy. Two different types of Eu³⁺ centers are identified in KMgF₃:Eu³⁺ (0.01 mol %). Two lines at 570.16 and 573.87 nm corresponding to two Eu³⁺ centers are observed in the excitation spectrum of the ⁷F₀→⁵D₀ transition. The former line is much stronger than the latter line. The dominant Eu³⁺ center is due to the isolated Eu³⁺-V_c (charge-compensating vacancies) pair and the other Eu³⁺ centers are attributable to a clustering of Eu³⁺-V_c pairs. At high concentration of Eu ions (above 0.5 mol %), new excitation lines due to clustering of Eu³⁺-V_c pairs appear in the excitation spectra of the ⁷F₀→⁵D₀ transition. Luminescence from most of the Eu³⁺ centers is quenched as the temperature is raised to 305 K. Such a luminescence quenching is explained by an energy transfer from Eu³⁺ to an H center formed in the vicinity of Eu³⁺-V_c pair.

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I. INTRODUCTION

When rare-earth (RE) ions are introduced into host materials, they may substitute for cation sites. If the valence state of the rare-earth ions is different from that of the cation ions, anion or cation vacancies are formed for charge compensation. The charge-compensating vacancies are expected to locate in the vicinity of the rare-earth ions. The different positions of the vacancies relative to the rare-earth ions give rise to different types of rare-earth centers. In addition to the isolated rare-earth ions with the vacancies, clustering of such rare-earth ions occurs, resulting in the presence of various cluster configurations in the host materials such as alkali-metal halides and alkaline-earth halides.¹⁻⁶

Optical properties of rare-earth-doped KMgF₃ have been extensively investigated not only for the purpose of scientific interest, but also from the practical viewpoint of UV scintillators⁷ and radiation dosimetry.^{8,9} Divalent europium and samarium ions substitute for K⁺ ions at the sites with cubic (*O_h*) and three different noncubic (*C_{4v}*, *C_{2v}*, and *C_{3v}*) symmetries depending on the location of charge-compensating K⁺-ion vacancies in KMgF₃.¹⁰⁻¹⁴ Doping of KMgF₃ by trivalent rare-earth ions is expected to cause quite a different situation from the case of doping by divalent rare-earth ions because RE³⁺ has different charge-compensating vacancies and ionic radius from the case of RE²⁺. No detailed study has been done on the defect structure of trivalent rare-earth-doped KMgF₃, although various investigations on the trivalent rare-earth-doped KMgF₃ by ESR (Refs. 15-17) and optical spectroscopy^{9,18,19} have been reported. Luminescence properties of Ce³⁺ in KMgF₃ were investigated by Francini *et al.*¹⁸ and Martini *et al.*⁹ They suggested that a Ce³⁺ ion substitutes for a K⁺ site and there exist two different Ce³⁺ centers causing the two luminescence bands. Francini *et al.* attributed the two centers to the unperturbed (cubic) site and a site perturbed (noncubic) by two K⁺-ion vacancies, while Martini *et al.* suggested that they are ascribed to the

isolated Ce³⁺ ions and complex centers formed by aggregation of Ce³⁺ ions.

Among rare-earth ions, europium can be stabilized in a divalent or trivalent state depending on the chemical composition of the host. In the Eu-doped KMgF₃, divalent europium is certainly observed by absorption and luminescence spectroscopy and there are many detailed reports on the optical properties of Eu²⁺ in KMgF₃.¹⁰⁻¹⁴ Only a few works have been published on Eu³⁺ luminescence in KMgF₃:Eu. Shiran *et al.*²⁰ reported narrow Eu³⁺ emission bands in the region of 520-630 nm, and Merkle *et al.*²¹ observed weak luminescence lines which seem to be due to Eu³⁺ in KMgF₃. Sommerdijk and Bril²² reported the Eu³⁺ emission line due to the ⁵D₀→⁷F₀ transition at 578.0 nm in KMgF₃. However, no figure was presented to show the experimental results in their paper. To our knowledge, no report has been given in detail so far regarding the Eu³⁺ luminescence spectra and the characteristics of the Eu³⁺ ion in KMgF₃ crystal.

We used site-selective laser-excitation spectroscopy to investigate substitution sites for Eu³⁺ and the defect structure of the trivalent europium-ion-doped KMgF₃ crystals. The narrow *4f*→*4f* transitions characteristic of trivalent europium in KMgF₃ offer a convenient method for directly monitoring the different sites. Eu³⁺ has a simple multiplet pattern of its energy level, which makes this ion particularly suited as a probe of local structure.^{23,24} Since each Eu³⁺ center with different crystal-field symmetry possesses a unique optical transition in the ⁷F₀→⁵D₀ excitation spectrum, the evolution of the cluster sites can be reliably followed as a function of doping concentration of Eu³⁺ ions. In this paper, we show that there exist two different types of Eu³⁺ centers, isolated Eu³⁺-V_c pairs, and clusters of Eu³⁺-V_c pairs.

II. EXPERIMENTAL PROCEDURE

Single crystals of KMgF₃ doped with europium ions were grown from the melt in Ar-gas atmosphere by the Czochral-

ski method at Pukyong National University. The concentrations of europium ions were 0.01, 0.5, and 2.5 mol % in the melt. For comparison with the experimental results on our crystals, we took the same measurements on a $\text{KMgF}_3:\text{Eu}$ (0.25 mol %) crystal grown by the Stockbarger method at Changwon National University and obtained nearly the same results as the results of our sample (0.5 mol % Eu-doped KMgF_3). This indicates that the luminescence properties of Eu^{3+} investigated in our KMgF_3 are not dependent on the individual crystal samples.

Annealing and quenching experiments were taken on the samples at argon gas atmosphere. The samples were heated up to 700 °C, kept at this temperature for 5 h, and then quenched down to room temperature by letting it cool down in the platinum boat in which it was heated. The excitation source was a dye laser (Spectron Laser Sys. SL4000) pumped by the second harmonic (532 nm) of a pulsed Nd:YAG (yttrium aluminum garnet) laser (Spectron Laser Sys. SL802G). The laser beam was focused inside the sample with a cross-sectional area of about 3 mm². The pulse energy was about 5 mJ with a 10 Hz repetition rate and 5 ns duration. A 355-nm pulsed laser with about 7-mJ pulse energy was used for UV irradiation on the samples. The beam path of the 355-nm laser beam coincided with that of the excitation laser beam on the samples. The samples were placed in a liquid-helium flow cryostat for measurements in the variable-temperature region (10–305 K). The luminescence was dispersed by a 75-cm monochromator (Acton Research Corp. Pro-750) and observed with a photomultiplier tube (PMT) (Hamamatsu R928) with which the signal is detectable up to about 850 nm. The excitation spectra for the ${}^7F_0 \rightarrow {}^5D_0$ transition were obtained by monitoring total luminescence detecting with the monochromator in zero order of diffraction. Suitable filters were used to eliminate the intense scattering peaks in the spectrum due to the scattered laser irradiation. The slit widths of the monochromator were normally set to give a resolution of 0.025 nm for emission spectra.

III. RESULTS

The presence of Eu^{2+} ions in our Eu-doped KMgF_3 crystals was confirmed by means of absorption and two-photon excitation spectroscopy. Eu^{2+} absorption bands were the same as those observed by Tsuboi and Scacco.¹² The parity forbidden $4f^7 \rightarrow 4f^7$ transition of Eu^{2+} was investigated by two-photon excitation spectroscopy and the results were published in Ref. 10.

Figure 1 shows the excitation spectrum corresponding to the ${}^7F_0 \rightarrow {}^5D_0$ transition at 10 K for the sample with Eu concentration of 0.01 mol %. Two excitation lines (labeled A and B) are observed in the spectrum. No other line was detected in the temperature range 10–305 K. The dominant excitation line is at 570.16 nm (A) and the weak line at 573.87 nm (B). Each line appearing in the spectrum corresponds to a different Eu^{3+} center since the transition between the nondegenerate ${}^7F_0 \rightarrow {}^5D_0$ levels can have only one line per site. We will call the Eu^{3+} ion that gives rise to the A excitation line site A hereafter. The position of the dominant

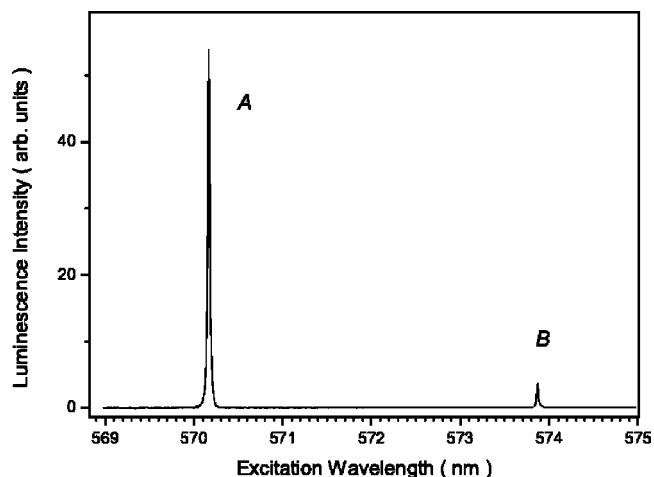


FIG. 1. Excitation spectrum for the ${}^7F_0 \rightarrow {}^5D_0$ transition of Eu^{3+} in the KMgF_3 crystal (sample I: Eu concentration of 0.01 mol %). The spectrum was obtained by monitoring the luminescence due to the ${}^5D_0 \rightarrow {}^7F_J$ ($J=0,1,2,\dots,6$) transitions in the 560–800 nm region in KMgF_3 at 10 K. The peaks are labeled by A and B.

line (570.16 nm) in $\text{KMgF}_3:\text{Eu}^{3+}$ is similar to that of the $\text{CsMgF}_3:\text{Eu}^{3+}$ (568.1 nm) and $\text{RbMgF}_3:\text{Eu}^{3+}$ (569.1 nm),²² in which monovalent ions (K^+ , Cs^+ , and Rb^+) are surrounded by 12 F^- ions in these crystals.

Figures 2 and 3 show the emission spectra of the ${}^5D_0 \rightarrow {}^7F_J$ ($J=1,2,\dots,5$) transitions from the individual Eu^{3+} center obtained by tuning the laser to resonance with each excitation line in Fig. 1. Emission lines for the ${}^5D_0 \rightarrow {}^7F_6$ transition are expected to appear at longer wavelengths than 800 nm, but they were not observed, presumably because they were too weak. As observed in usual crystals containing Eu^{3+} ions, the ${}^5D_0 \rightarrow {}^7F_2$ luminescence intensity is stronger than the other ${}^5D_0 \rightarrow {}^7F_J$ luminescence. Each Eu^{3+} center exhibits numerous emission lines between 12 500 and 17 500 cm^{-1} originating from the 5D_0 level. The energy levels of 5D_0 and 5F_J ($J=1,2,\dots,5$) multiplets of the two Eu^{3+} centers (sites A and B) were identified from the emission and excitation spectra. The transitions between 5D_0 and 7F_J and energy levels are given in Tables I and II, respectively.

Figure 4 shows the excitation spectra corresponding to the ${}^7F_0 \rightarrow {}^5D_0$ transition for three KMgF_3 crystals with different Eu concentrations at 10 K. We refer to samples doped with Eu concentrations of 0.01, 0.5, and 2.5 mol % as sample I, sample II, and sample III, respectively. With increasing Eu concentration, the excitation lines (A and B) are broadened and the peak height of line B relative to line A is increased. The ratio of the peak height of line B to that of line A was estimated to be 1.6 times larger for sample II than for sample I, and the ratio is much larger (6.4 times) for sample III. The bandwidth of line A of sample I is estimated to be 0.92 cm^{-1} and the line is broadened by 2.2 cm^{-1} (sample II) and 5.1 cm^{-1} (sample III) [see Fig. 4(b)]. New excitation lines, which are not observed for sample I, appear in the enlargement of the excitation spectrum for sample II, although their peaks are very weak, as shown in Fig. 5. The relative intensities of these lines are increased for sample III, but the lines are overlapped by broad and strong lines A and B. The peaks

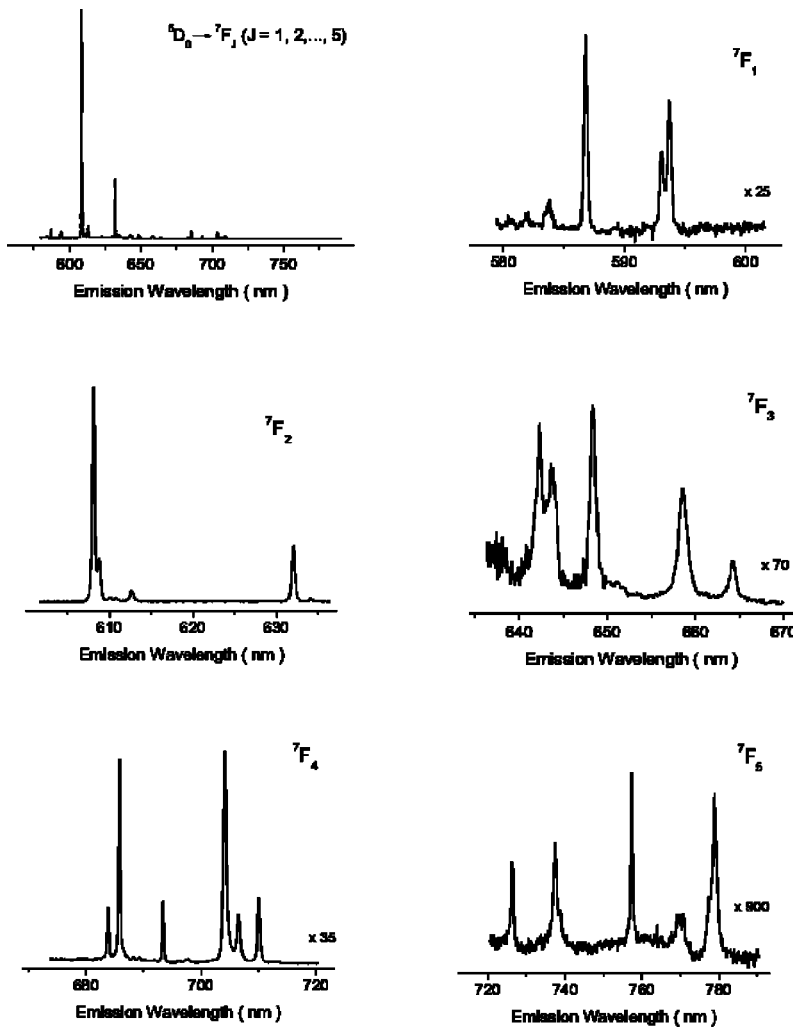


FIG. 2. ${}^5D_0 \rightarrow {}^7F_J$ ($J=0, 1, 2, \dots, 5$) emission spectrum for site A exciting at 570.16 nm in 5D_0 at 10 K.

are labeled by *C*, *D*, *E*, *F*, *G*, *H*, *I*, and *J*, as indicated in Fig. 5. Relatively intense but broad lines are observed at 566.39 (*D*) and 567.74 nm (*E*). Additionally, several weak lines are at 564.36 nm (*C*), 568.45 nm (*F*), 569.43 nm (*G*), 571.40 nm (*H*), 573.30 nm (*I*), and 575.51 nm (*J*). Each line (except for the phonon side band) appearing in the spectrum corresponds to a different Eu³⁺ center.

Figure 6 shows the 10 K emission spectra of the ${}^5D_0 \rightarrow {}^7F_J$ ($J=0, 1, 2$) transitions from individual sites obtained by tuning the laser to resonance with each excitation line in Fig. 5. The emission spectra should be different from each other if all the excitation lines are responsible for the different Eu³⁺ centers. However, the emissions obtained by exciting at some excitation lines show the same spectral and temporal behaviors. The same emission profiles are observed for the spectra obtained by irradiation with the *E*, *F*, and *A* excitation lines. The emission spectrum obtained by irradiation with the *C* excitation line consists of the spectrum obtained by irradiation with the *D* excitation line and the spectrum obtained by irradiation with the *A* excitation line. Two excitation lines (called *C_A* and *C_D* lines) are superimposed at the broad *C* excitation lines. From these results, we identify that the *E*, *F*, and *C_A* lines are phonon side bands associated with the main sharp *A* line [zero phonon line (ZPL) *A*], while the *C_D* phonon side bands are associated with the zero phonon

line *C*. These are also confirmed by the appearance of the zero phonon line at 570.16 nm for the site *A* in each emission spectrum (see the left side of Fig. 6). The observed phonon side bands are at 53 cm⁻¹ (*E*), 75 cm⁻¹ (*F*), and 172 cm⁻¹ (*C_A*) from the zero phonon line *A* and at 55 cm⁻¹ (*C_D*) from the zero phonon line *D*. As a result, there are seven measurable Eu³⁺ centers (i.e., the sites *A*, *B*, *D*, *G*, *H*, *I*, and *J*) in 0.5 mol % Eu-doped KMgF₃ at 10 K.

Figure 7(a) shows the excitation spectrum of the ${}^7F_0 \rightarrow {}^5D_0$ transition for the sample irradiated by a 355 nm pulsed laser at 10 K. In addition to the lines *A* and *B*, new broadband peaking at 571.7 nm appears with a weak band at 573.3 nm. The intensity of these new bands increases by increasing 355 nm pulsed laser output and irradiation time and is weaker for the sample with low Eu concentration. When the 355-nm-irradiated sample was heated to room temperature, the new bands completely disappeared in the excitation spectrum measured at 10 K. We attribute the new bands to the Eu³⁺ centers due to the formation of [*F₂⁻*] centers near the Eu³⁺-*V_c* which is discussed in Sec. IV. The emission spectrum of the ${}^5D_0 \rightarrow {}^7F_J$ transitions obtained by exciting at 571.7 nm is shown in Fig. 7(b). The most intense luminescence line of the ${}^5D_0 \rightarrow {}^7F_2$ transition is observed at 633.8 nm, which is quite different from those for the sites *A* and *B*, in which the strongest lines of the ${}^5D_0 \rightarrow {}^7F_2$ transi-

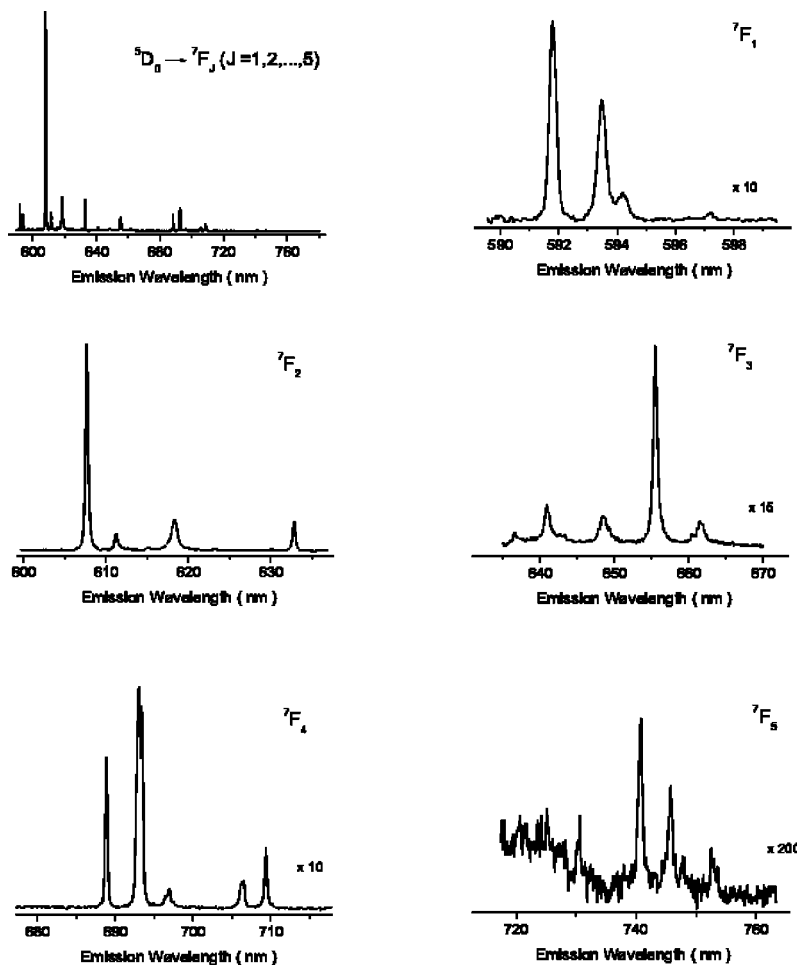


FIG. 3. ${}^5D_0 \rightarrow {}^7F_J$ ($J=0,1,2,\dots,5$) emission spectrum for site *B* exciting at 573.87 nm in 5D_0 at 10 K.

tions are around 610 nm. The decay of the 5D_0 level was single exponential with a lifetime of 790 μs .

The excitation spectra of the ${}^7F_0 \rightarrow {}^5D_0$ transition for different crystal temperatures are displayed in Fig. 8. The luminescence intensities of all the lines decrease rapidly above a certain temperature with increasing temperatures. The lines *B* and *D* disappear in the spectra obtained at 150 and 220 K, respectively. The intensity of the luminescence from the site *A* drastically decreases with increasing temperature, but it is still detectable even at 305 K. The *A* excitation line shifts about 4 cm^{-1} to lower energy with increasing temperature from 10 K to 300 K, as shown in the inset of Fig. 8. A new line (called *K*) appears at 569.82 nm above 260 K and its peak becomes higher than the peak of line *A* at room temperature. The *K* line may be hidden under the intense *A* line at low temperature. The *K* line was also observed in the excitation spectrum for sample III and the intensity ratio of the excitation lines between sites *K* and *A* is the same as that of sample II. This indicates that site *K* is not due to the cubic-site Eu^{3+} as discussed in Sec. IV. We note that only three excitation lines (*K*, *A*, and *I*) are observed at room temperature.

The decay times of the 5D_0 luminescence of all the sites were measured in the temperature range from 10 to 305 K. The decay times for the two major sites *A* and *B* are estimated to be 616 and 470 μs , and those for the minority sites *D*, *G*, *H*, *I*, and *J* are 394, 636, 644, 810, and 787 μs , re-

spectively, at 10 K and 1.5 ms for site *K* at room temperature. Figure 9 plots lifetimes of the 5D_0 luminescence for sites *A*, *B*, *D*, and *I* as a function of temperature. We observe a quenching of decay times. The temperatures giving rise to the lifetime quenching are the same as the temperatures giving rise to the intensity quenching of the 5D_0 luminescence. The critical quenching temperatures are 60, 150, 210, 230, and 300 K for sites *D*, *A*, *C*, and *I*, respectively. The lifetimes for sites *G* and *H* are nearly constant up to about 120 K, but those were not measurable due to overlapping of those excitation lines with the broadened *A* excitation line above this temperature. It is noted that the site with shorter decay time tends to have lower quenching temperature, and site *K* with a much longer decay time of 1.5 ms exhibits no quenching up to 305 K.

IV. DISCUSSION

A Mg^{2+} ion forms an octahedron composed of six F^- ligands in KMgF_3 . The K^+ ion has 12 F^- nearest neighbors with cuboctahedral O_h symmetry and is surrounded by eight $[\text{MgF}_6]^{-4}$ octahedrons. The small Mg^{2+} ion (ionic radius is 0.86 Å) bonds strongly with the F^- ligands in the $[\text{MgF}_6]^{-4}$ octahedron, while the large Eu -F distance in the $[\text{KF}_{12}]^{-11}$ cuboctahedron and the presence of the surrounding $[\text{MgF}_6]^{-4}$ octahedrons leads to low covalency between the K^+ ion (1.78 Å) F^- .¹⁴ Consequently, the site environment of Mg^{2+} is

TABLE I. Observed transitions of Eu³⁺ for the sites *A* and *B* in KMgF₃ obtained from excitation and emission spectra at 10 K.

	Site <i>A</i>		Site <i>B</i>	
	nm	cm ⁻¹	nm	cm ⁻¹
⁷ F ₀ → ⁵ D ₀	570.16	17539	573.87	17426
⁵ D ₀ → ⁷ F ₁	586.81	17041	591.80	16898
	593.07	16861	593.49	16849
⁵ D ₀ → ⁷ F ₂	593.72	16843	594.21	16829
	608.12	16444	607.71	16455
⁵ D ₀ → ⁷ F ₃	608.82	16425	611.27	16359
	612.70	16321	618.35	16172
	632.06	15821	623.85	16029
	642.34	15568	636.59	15709
	643.74	15534	640.96	15602
⁵ D ₀ → ⁷ F ₄	648.35	15424	648.54	15419
	658.55	15185	655.50	15256
	664.21	15055	661.64	15114
	683.88	14622	688.86	14517
	685.89	14580	693.01	14430
	693.48	14420	693.44	14421
	704.17	14201	696.97	14348
⁵ D ₀ → ⁷ F ₅	706.57	14153	706.40	14156
	710.10	14083	709.36	14097
	726.32	13768	730.54	13689
	737.45	13560	740.72	13500
	757.40	13203	745.72	13410
	770.27	12982	747.86	13371
	778.93	12838	752.72	13285

significantly different from that of K⁺. Eu³⁺ has an ionic radius of 1.21 Å, which is much larger than Mg²⁺. Therefore, it is reasonable to suggest that an Eu³⁺ ion preferentially substitutes for a larger K⁺ ion (1.78 Å) as the case of Ce³⁺ ions (1.28 Å) in KMgF₃ suggested by Francini *et al.*¹⁸ and Martini *et al.*⁹

The site symmetries of Eu³⁺ can be determined from the number of emission lines of the ⁵D₀→⁷F_{*J*} (*J*=1, 2, ..., 6) transitions.²³ Theoretically, for example, the ⁷F₁ and ⁷F₂ multiplets split by crystal-field perturbation into two and four sublevels for the tetragonal site and three and five sublevels for the orthorhombic site. Induced electric dipole (ED) and magnetic dipole (MD) selection rules predict the permitted transitions between excited ⁵D₀ and ground ⁷F_{*J*} levels of Eu³⁺. The transitions between *J* sublevels are further restricted by symmetry selection rules.²³ In the case of ⁵D₀→⁷F₁ and ⁵D₀→⁷F₂ transitions of Eu³⁺ with tetragonal crystal-field symmetry, only two and two transition lines are allowed, respectively. It is known that the transitions between ⁵D₀ and ⁷F_{*J*} (*J*=0, 1, and 2) states of Eu³⁺ or Sm²⁺ doped in fluoride crystals such as LiYF₄:Eu³⁺ (Ref. 24) and KMgF₃:Sm²⁺ (Ref. 11) strictly obey the induced ED and MD selection rules and the symmetry selection rules. By applying the selection rules to the transitions from ⁵D₀ to ⁷F_{*J*}

TABLE II. Energies (in cm⁻¹) of the ⁵D₀ and ⁷F_{*J*} levels of Eu³⁺ for the sites *A* and *B* in KMgF₃.

	Site <i>A</i>	Site <i>B</i>
⁷ F ₀	0	0
⁷ F ₁	498	528
	678	576
⁷ F ₂	696	596
	1095	970
⁷ F ₃	1114	1066
	1218	1254
	1718	1624
⁷ F ₄	1971	1717
	2005	1824
	2115	2014
	2354	2170
⁷ F ₅	2484	2312
	2917	2909
	2959	2996
	3119	3005
	3338	3078
⁵ D ₀	3386	3269
	3456	3328
	3771	3737
	3979	3925
	4336	4016
	4557	4052
	4701	4140
	17539	17426

levels of Eu³⁺ in KMgF₃, we have found the sites *A* and *B* have lower symmetries belonging neither in tetragonal (*C*_{4*v*}) nor orthorhombic (*C*_{2*v*}) symmetries. We note that the tetragonal and orthorhombic sites for Eu²⁺ and Sm²⁺ are dominantly identified with cubic symmetry (*O*_{*h*}) in KMgF₃:Eu²⁺ and KMgF₃:Sm²⁺.^{10,11,13}

The induced ED transition cannot occur when the point group of the site contains a center of symmetry (e.g., cubic symmetry) and the ⁷F₀→⁵D₀ transition for the cubic site is further forbidden by the selection rules. The MD transition is allowed only from the ⁷F₁ level to the ⁵D₀ level in cubic symmetry. The ⁷F₁ levels of Eu³⁺ ions can be populated thermally by the ⁷F₀ levels at high temperature due to the small energy difference of about 550 cm⁻¹. To ascertain the presence of cubic-site Eu³⁺ in KMgF₃, we measured the excitation spectrum of the ⁷F₁→⁵D₀ transition at temperatures higher than 150 K. However, no transition line from cubic-site Eu³⁺ was observed, but only the lines at 586.8 and 593.7 nm due to the site *A* were observed.

The substitution of Eu³⁺ for K⁺ gives rise to two K⁺-ion vacancies (*V*_{*c*}) close to Eu³⁺ forming an Eu³⁺-*V*_{*c*} pair for charge compensation. The trivalent europium ion is more tightly bound to the vacancies than a divalent europium ion in KMgF₃ because two extra charges on the Eu³⁺ ion increase effectively the Coulomb attraction. The cubic-site

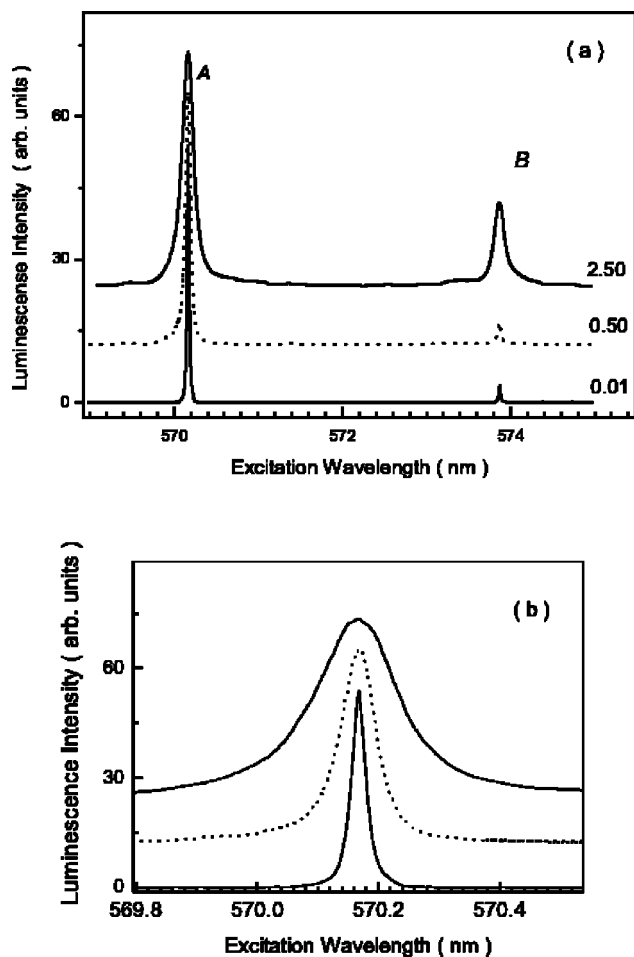


FIG. 4. (a) Excitation spectra for three different $\text{KMgF}_3:\text{Eu}$ crystals with an Eu concentration of 0.01, 0.5, and 2.5 mol % at 10 K. (b) Extended spectra near site A in the spectral range of 569.8–570.5 nm.

Eu^{3+} can be formed by an Eu^{3+} ion with two vacancies located far from the Eu^{3+} ion. However, no cubic-site Eu^{3+} is produced in $\text{KMgF}_3:\text{Eu}^{3+}$, which seems to be due to the strong electrostatic attraction between Eu^{3+} and the vacancies unlikely to the cubic sites Eu^{2+} and Sm^{2+} produced in KMgF_3 .^{10,11,13} Eu^{2+} and Sm^{2+} in KCl, similarly to Eu^{3+} in KMgF_3 , do not occupy a site with cubic symmetry but a site with C_{2V} or C_{4V} symmetry.^{1,6}

In case of Ce^{3+} -doped KMgF_3 (Ref. 9) and CaF_2 ,³ an increase of Ce^{3+} concentration causes significant growth of clustered sites. In Eu^{3+} -doped KMgF_3 , the growth of the excitation line B relative to the line A on doping concentration (Fig. 4) is behavior typical of clustering. We thus attribute the site A to the isolated $\text{Eu}^{3+}-V_c$ pairs, while the site B and other minority sites can be attributed to the sites due to the clustering of $\text{Eu}^{3+}-V_c$ pairs. The clustering behavior of Eu^{3+} is similar to that of Ce^{3+} in KMgF_3 .

The two charge-compensating vacancies have the same charge and so they repel each other. Therefore, the positions where the vacancies are favored lie on the opposite sites in terms of an Eu^{3+} ion giving a C_{4V} site symmetry of Eu^{3+} . However, the crystal-field symmetries of all the sites appearing in the excitation spectra are assumed to be lower than

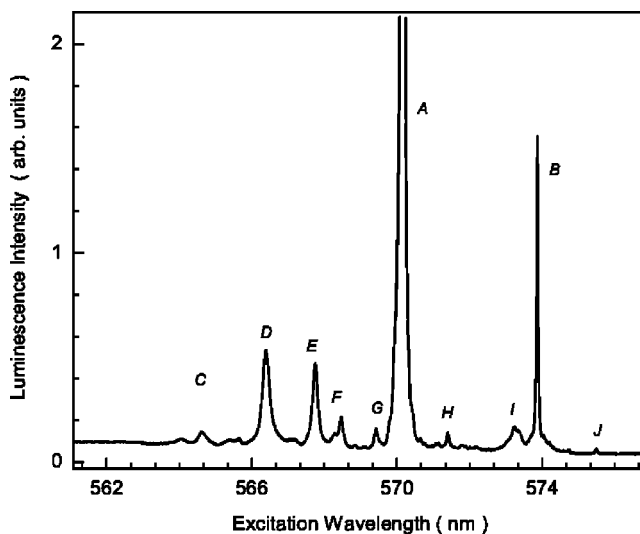


FIG. 5. Enlarged excitation spectrum for the ${}^7F_0 \rightarrow {}^5D_0$ transition of Eu^{3+} in the KMgF_3 crystal (sample II: Eu concentration of 0.05 mol %).

C_{4V} , as discussed above. The C_{4V} symmetry seems to be distorted by a defect center formed together with the $\text{Eu}^{3+}-V_c$ pair. One possible defect center is an interstitial fluorine atom (H center) which enters the KMgF_3 lattice adjacent to the $\text{Eu}^{3+}-V_c$ pair due to enough space created by two charge-compensating vacancies. It is known that the intrinsic interstitial fluorine atom (H center) exists adjacent to

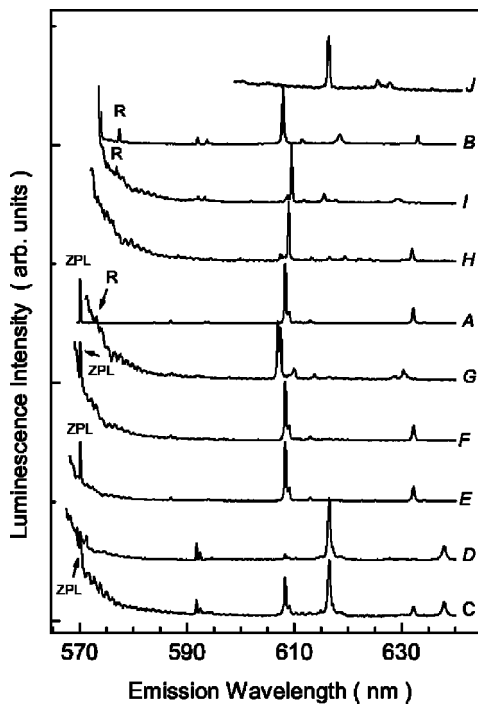


FIG. 6. Emission spectra for the ${}^5D_0 \rightarrow {}^7F_J$ ($J=0,1,2$) transitions of Eu^{3+} under the resonant site-selective excitations at 10 K. The excitation lines are indicated at the right side of the emission spectra. Intensities are normalized to the most intense ${}^5D_0 \rightarrow {}^7F_2$ peak. Zero phonon lines (ZPL) are designated in the spectra A, F, E, D and C at 570.16 nm. R denotes Raman line at 102 cm^{-1} .

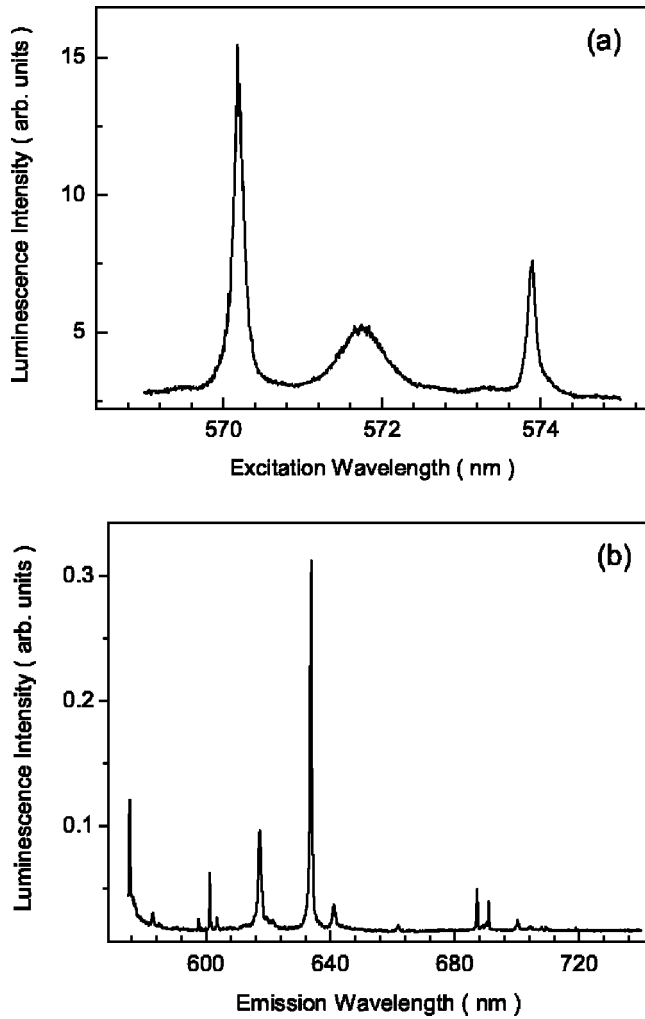


FIG. 7. (a) Excitation spectrum of the ${}^7F_0 \rightarrow {}^5D_0$ transition for the 355-nm-irradiated $\text{KMgF}_3:\text{Eu}$ crystal at 10 K. (b) Emission spectrum of the ${}^5D_0 \rightarrow {}^7F_J$ transitions obtained by exciting at 571.7 nm appearing in (a).

impurity cation ions (Li^+, Na^+) whose ionic radii are smaller than the K^+ ions in KMgF_3 .²⁵ The new site in the excitation spectrum (Fig. 9) obtained after 355-nm irradiation on the sample is evidence of the presence of interstitial fluorine atoms. The interstitial fluorine atom combines with a neighboring fluorine ion to form a molecular ion (F_2^-) by UV irradiation.^{26,27} The F_2^- molecular ion perturbs the crystal field at the Eu^{3+} ion, resulting in the production of a new Eu^{3+} center.

Clustering occurs in various rare-earth-doped alkali-metal halides^{1,2,4-6} and alkaline-earth halides.³ The clustering state of rare-earth ions in host materials can be changed by thermal treatment, and its behavior depends on the host materials. Santiuste *et al.*⁴ observed different types of EuX_2 ($X = \text{Cl}, \text{Br}, \text{or I}$) precipitates in Eu^{2+} -doped KCl in which the Eu^{2+} luminescence exhibits no temperature-dependent quenching in the range of 10–300 K, unlike Eu^{3+} luminescence in KMgF_3 . The temperature-dependent quenching was reported for Sm^{2+} luminescence in KCl crystals by Guzzi *et al.*²⁸ and Ramponi *et al.*¹ All the Sm^{2+} centers exhibit quenching behavior at low temperatures in $\text{KCl}:\text{Sm}^{2+}$, and

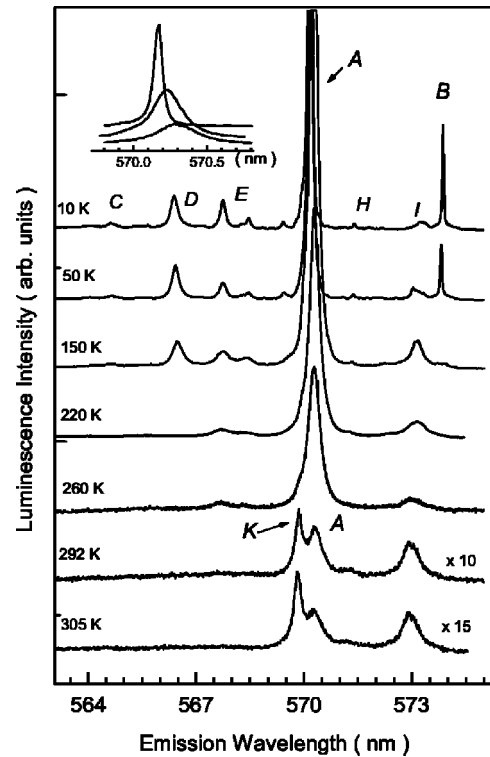


FIG. 8. Excitation spectra for the luminescence due to the ${}^5D_0 \rightarrow {}^7F_J$ ($J=1,2,\dots,6$) transitions measured at different temperatures. The temperatures are indicated at the left side of the spectra. The inset is the extended spectra near the site C in the spectral range 569.8–570.8 nm for 10, 150, and 220 K.

the luminescence even from the isolated $\text{Sm}^{2+}-V_c$ pairs is quenched at 12 K.¹ Guzzi *et al.* explained the phenomena in terms of a nearby $4f^5 5d$ configuration which thermally depopulates the emitting 5D_0 level. In the case of Eu^{3+} , however, the nonradiative relaxation processes do not take place via the thermal population of the $4f^5 5d$ configuration even at room temperature because the $4f^5 d$ configuration of Eu^{3+} lies at much higher energy with respect to the 5D_0 level.

The thermal quenching of the Eu^{3+} luminescence is ascribed to an energy transfer from Eu^{3+} to a defect center such as the H center in KMgF_3 .²⁵⁻²⁷ The clustering of $\text{Eu}^{3+}-V_c$ pairs further enhances the energy transfer rate between Eu^{3+}

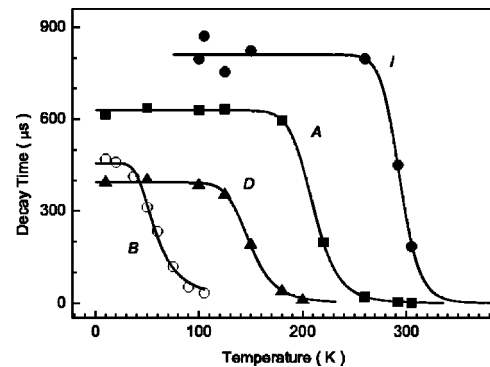


FIG. 9. Decay times for the sites A, B, D, and I as a function of temperature. Solid curves are fits to a three-level decay system.

ions to quench the luminescence at lower temperature in $\text{KMgF}_3:\text{Eu}^{3+}$. The excited Eu^{3+} ions due to the ${}^7F_0 \rightarrow {}^5D_0$ transitions may relax to the ground state by very rapid energy transfer among Eu^{3+} ions and finally return to the ground state by transferring their energy to the sink. As shown in Figs. 8 and 9, the thermal quenching tends to occur at lower temperatures for the sites with shorter lifetimes, except for the site K , which exhibits no quenching behavior up to 305 K.

The aggregation of rare-earth ions can be controlled by appropriate annealing and subsequent quenching procedures. The Sm^{2+} clusters in alkali-metal halides are dissolved in part by annealing and quenching processes (e.g., see Ref. 1). We took the annealing and quenching experiments on the samples under Ar-gas atmosphere to observe the change in clusters. However, no change in spectral features of all the sites was observed in the temperature range 10–305 K. The same behavior was reported in $\text{KMgF}_3:\text{Ce}^{3+}$ by Francini *et al.*, from which they ruled out the possibility of clustering of

Ce^{3+} ions and argued that cubic-site Ce^{3+} ions are present. We suggest that the clustering of Eu^{3+} occurs in KMgF_3 , although the clusters are not destroyed by annealing and quenching treatment, unlike those in $\text{KCl}:\text{Sm}^{2+}$. The $\text{Eu}^{3+}-V_c$ pairs dissolved by annealing treatment may recombine with each other very rapidly or an association energy of the clusters is too large to be dissolved by a given thermal energy in KMgF_3 . Ramponi *et al.*¹ reported that the annealing and quenching treatment does not completely dissolve the clusters even by very fast quenching for the annealed sample.

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