Magneto-optics of Gd and Tb in the soft x-ray resonance regions

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We present x-ray absorption spectra around the $3d \rightarrow 4f$ and $4d \rightarrow 4f$ excitation thresholds of in-plane magnetized Gd and Tb films measured by total electron yield using circularly polarized synchrotron radiation. By matching the experimental spectra to tabulated absorption data far below and above the thresholds, the imaginary parts of the complex refractive index are determined quantitatively. The associated real parts for circularly polarized light propagating nearly parallel or antiparallel to the magnetization direction are obtained through the Kramers-Kronig relations. The derived magnetooptical parameters are used to calculate soft x-ray reflectivity spectra of a magnetized Gd film at the $3d \rightarrow 4f$ threshold, which are found to compare well with our experimental spectra.

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I. INTRODUCTION

Magneto-optical (MO) effects in the visible-light region are widely used for analyzing magnetic materials¹ and have important technological applications, e.g., in the reading process of MO disks.² They are based on the fact that left- and right-hand circularly polarized (CP) light is reflected with different intensities from a magnetic material depending on the local magnetization direction. Although MO effects are small in the visible-light region, sensitive detection methods yield enough contrast to distinguish bits of opposite magnetization and allow the observation of domain structures in optical microscopy.3 One powerful feature of MO techniques is their capability to monitor magnetization reversal processes in applied magnetic fields, which is not readily achievable with electron detection techniques due to Lorentz forces. MO techniques in the visible-light region involve optical transitions between *delocalized* valence states, which renders it extremely difficult to spectrally separate the magnetic contributions of different elements in compounds and in advanced layered or nanostructured materials. This can be a severe limitation in analyzing magnetic nanostructures 4.5 or heteromagnetic systems for information storage. $2,6$

Element sensitivity is naturally achieved by employing optical transitions that involve core electrons. Large magneto-optical signals in the x-ray region were theoretically predicted for resonant scattering (XRS) , i.e., when the x-ray photon energy is tuned to the transition energy from a core level into a partially filled shell that contributes to an ordered magnetic moment.7 In fact, XRS has been used extensively to study the magnetic structure of lanthanide⁸ and actinide 9 materials in the hard x-ray regime.

In the soft x-ray region, large changes in the specularly reflected x-ray intensity at the $L_{2,3}$ edges of transition metals upon magnetization reversal have been detected $10-12$ and used in element-specific studies of heteromagnetic systems.13–20

In present-day thin films and multilayers, the wavelength of soft x rays is comparable to the system dimensions (typical thicknesses range from 1 to 10 nm). It has been shown by several studies $13,15,19$ that in order to extract layer-resolved magnetization profiles from measured soft x-ray MO signals

of layered systems, a comparison with model calculations²¹ of reflected intensities is usually required. These are based on the Fresnel equations and need accurate values of the MO constants as input.

Several experimental determinations of soft x-ray MO constants^{11,22,23} and reflection coefficients²⁴ have been reported for the $L_{2,3}$ thresholds of ferromagnetic transition metals, but results on rare-earth elements have been scarce, in spite of the fact that these are often employed to achieve high coercive fields in magnetic layers (e.g., in spring magnets²⁵) or large perpendicular magnetic anisotropies.²⁶ Only recently has it been demonstrated that sizeable MO signals are obtained from lanthanide elements in the soft x-ray region at the $N_{4.5}$ thresholds.²⁷ In a further study, a huge Faraday rotation has been predicted for Gd films at the photon energy corresponding to this transition, making use of the experimentally determined difference in the refractive index for oppositely magnetized material.²⁸

It is the aim of this paper to perform a quantitative determination of the MO constants for the $4d \rightarrow 4f$ and $3d \rightarrow 4f$ absorption thresholds of ferromagnetic Gd and Tb metals. Absorption spectra, calibrated by matching to tabulated data far from the resonances, provide the data basis for a determination of the imaginary parts of the refractive index. The associated real parts are obtained through a Hilbert transformation using the Kramers-Kronig (KK) relations. In this sense, the magneto-optical parameters derived in this work are quantitative; they are consistent with the tabulated values far from the resonances, but they have not been determined in an absolute way. Finally, in order to illustrate the applicability of the MO constants obtained in this way, we calculate soft x-ray reflectivity spectra of a magnetized Gd film at the $M_{4.5}$ thresholds and compare them with our experimental reflectivity spectra.

II. EXPERIMENTAL

Experiments at the $4d \rightarrow 4f$ absorption thresholds were performed at the UE56 undulator beamline²⁹ of the Berliner Elektronenspeicherring für Synchrotronstrahlung (BESSY II), while those at the $3d \rightarrow 4f$ thresholds were performed at the beamline ID12-B/HELIOS-I of the European Synchro-

tron Radiation Facility (ESRF).^{30,31} In the UE56 experiments, the photon energy resolution was set to about 100 meV (full width at half maximum), which is well below the intrinsic width of the narrow $N_{4,5}$ pre-edge absorption lines of Gd and Tb^{32} By scanning the photon energy at slow speed through a synchronized movement of monochromator and undulator, an easy normalization of the spectra was made possible; this also allows one to exploit the high flux of the undulator beamline of about 10^{14} photons/(s×100 mA $\times 0.1\%$ bandwidth) over a wide energy range. The degree of circular polarization at this Sasaki-type undulator beamline is $98\pm2\%$ ²⁹ The reflectivity measurements at the Gd $M_{4.5}$ lines were also performed at BESSY II. For the absorption measurements at the $M_{4,5}$ thresholds at ID12, the energy resolution was set to about 0.4 eV and the degree of circular polarization was $94\pm3\%$.³³

Absorption spectra were recorded in total-electron yield (TEY) mode using a high-current channeltron. To suppress the background of secondary electrons from the chamber walls, both the sample and a retarding grid placed in front of the channeltron were biased using a low-voltage battery. For signal stability, high voltage was supplied by a 3.2-kV battery box. The electron-yield current was amplified by an electrometer. TEY spectra of the $4d \rightarrow 4f$ thresholds were normalized to the mirror current measured in the last refocusing element of the beamline located in a separate vacuum system in front of the experimental chamber. Because it is essentially the same physical effect (secondary electron generation) that is used for the measurement of both absorption and normalization signals, possible distortions due to a lack of proportionality between TEY and absorption coefficient³⁴ in the wide photon energy range measured for the $N_{4,5}$ transitions are expected to be minimal. The film thicknesses were always sufficiently larger than the electron's inelastic mean free path (IMFP) relevant for TEY measurements so that contributions of the W substrate to the TEY signal are negligible. In addition, such a contribution would be structureless in the measured photon-energy ranges and would be removed by the procedure of matching the ends of the spectra to tabulated values (see below). Reflectivity spectra were recorded in specular geometry with a photodiode mounted on a rotatable feedthrough; the diode current was measured by means of a low-noise electrometer. The experimental arrangement is shown schematically in Fig. 1.

Epitaxial Gd and Tb metal films with thicknesses ranging from 10 to 50 nm were prepared *in situ* by vapor deposition in ultrahigh vacuum on a $W(110)$ single-crystal substrate. The base pressure in the chamber was in the 10^{-11} mbar range, rising to about 4×10^{-10} mbar during film deposition. Compared to transmission methods, which give the absorption coefficient in a more direct way, TEY detection has the advantage of allowing the use of metallic single crystals as substrates. The growth of epitaxial films on them is well characterized. In particular, controlled annealing of the deposited lanthanide films can be performed at the optimum temperatures for achieving smooth films with homogeneous thicknesses. (for details of film preparation, see Ref. 35). For remanent *in-plane* sample magnetization, an external field was applied along the $\lceil 1\overline{1}0 \rceil$ direction of the substrate using

FIG. 1. (a) Schematic experimental arrangement used for measuring absorption of CP light by TEY, using a channeltron, and specular reflectivity by means of a photodiode. In-plane magnetized Gd and Tb epitaxial films were prepared on $W(110)$ and measured *in situ*. (b) XMOKE hysteresis loop of an 8-nm-thick Gd film measured at a photon energy of 147.5 eV.

a rotatable electromagnet.³⁶ This corresponds to the easy axis of magnetization of the Gd and Tb films. In the case of the magnetically harder Tb films, the sample was cooled from room temperature down to the measurement temperature $(30 K)$ in an external magnetic field on the order of 0.1 T to achieve a single-domain magnetic structure with a high remanent magnetization. This has been veryfied by means of the magneto-optical Kerr effect (MOKE) in the visible-light range in the laboratory and in the soft x-ray range $(XMOKE)$ at the beamline. As an example, Fig. $1(b)$ shows an elementspecific hysteresis loop of a thin Gd film measured by XMOKE.

III. RESULTS AND DISCUSSION

A. Absorption

Figure 2 displays experimental absorption spectra in the region of the Gd and Tb $N_{4,5}$ (4 $d \rightarrow 4f$) thresholds for nearly parallel and antiparallel orientation of the magnetization with respect to the spin of the incoming CP photons. Figure 3 shows the $M_{4.5}$ (3*d*→4*f*) absorption thresholds of both elements. The spectra have been corrected for saturation and scaled to fit the tabulated values of Henke *et al.*³⁷ at the ends of the measured photon energy ranges following a procedure that is described in detail below.

The measured TEY spectra are affected by intrinsic saturation, which becomes significant when the attenuation length of the electromagnetic radiation μ^{-1} gets comparable to d_e , the inelastic mean free path (IMFP) of electrons in the solid.^{38,39} In that case, the detected signal at the channeltron is no longer proportional to the absorption coefficient μ . Due to the large absorption cross sections at the lanthanide 4*d* \rightarrow 4*f* and 3*d* \rightarrow 4*f* thresholds, the intrinsic saturation effect is significant and the TEY spectra must be corrected for it in order to achieve a reliable quantitative estimation of the absorption coefficient. The saturation correction was made by inverting the relation 39

$$
Y = C \frac{\mu d_e}{\mu d_e + \sin \theta} \tag{1}
$$

for the measured yield *Y*. Since the constant *C* depends on geometrical parameters of the detection system and is not

FIG. 2. Absorption coefficients of (a) Gd and (b) Tb at the $N_{4,5}$ (4*d* \rightarrow 4*f*) thresholds of remanently magnetized films at *T*=30 K. CP light was incident at θ =30° with respect to the film plane, i.e., mainly parallel (filled symbols) and antiparallel (open symbols) to the in-plane sample magnetization. The inserts show the regions of the prepeaks measured with higher point densities.

precisely known, we need to determine the value of μd_e at some fixed photon energy from independent measurements. In the case of Gd $M_{4,5}$, the saturation correction is based on previous results obtained from the (6×6) Eu/Gd (0001) surface.⁴⁰ It consists of a quasi-close-packed Eu monolayer on top of the $Gd(0001)$ surface. Since divalent Eu has the same electronic 4f shell configuration $(4f^7)$ as trivalent Gd, both elements show the same multiplet structure in absorption at the $M_{4,5}$ thresholds. The transitions are well separated in energy. A value of (0.30 ± 0.05) for the product $(\mu d_e)_{peak}$ of Gd at the M_5 threshold has been determined by comparison of the experimental M_5 peaks of Gd and Eu; for the latter, negligible saturation can be safely assumed due to the small thickness of the Eu overlayer (1 atomic layer) .⁴¹ Furthermore, by comparing the total (energy-integrated) M_5 absorption signals of Gd and Tb and taking into account the different number of $4f$ holes of both elements (seven in Gd, six in Tb), we estimate a value of (0.20 ± 0.05) for $(\mu d_e)_{\text{peak}}$ at the Tb M_5 threshold. A consistency argument can be applied to justify the use of the same values for the saturation correction at the $4d \rightarrow 4f$ thresholds: the relevant electron

FIG. 3. Absorption coefficients of (a) Gd and (b) Tb at the $M_{4.5}$ (3*d* \rightarrow 4*f*) thresholds of remanently magnetized films at $T=30$ K. CP light was incident at $\theta=30^{\circ}$ with respect to the film plane, i.e., mainly parallel (filled symbols) and antiparallel (open symbols) to the in-plane sample magnetization. The raw data were taken from Refs. 30 and 31.

IMFP at the $4d \rightarrow 4f$ thresholds corresponds to about 150 eV kinetic energy of the Auger electrons which trigger the secondaries detected at the channeltron; it is typically 3 times smaller than at \sim 1200 eV, according to the "universal" curve, 42 while the values obtained for the absorption length μ^{-1} change roughly by the same factor (see below).

The saturation-corrected absorption spectra were scaled to fit tabulated values. The photon energy ranges of the present spectra are significantly wider than those of previous studies; $32,43$ they include the wide asymmetric flanks in the case of the $4d \rightarrow 4f$ resonances. This allows us to calibrate the absorption spectra by matching both ends to the tabulated absorption coefficients³⁷ at photon energies where the influence of the resonances is expected to be negligible. To this end we fixed the absorption coefficients μ_{\pm} at the low- and high-energy sides of the measured spectra to the values given by the tables of Henke *et al.*³⁷ This procedure defines the ordinate scales in Figs. 2 and 3. The values for the absorption lengths μ^{-1} determined in this way are given in Tables I and II. The error bars for the quoted values have

TABLE I. Absorption lengths μ^{-1} (in nm) for Gd and Tb at the $4d \rightarrow 4f$ thresholds for nearly parallel and antiparallel orientations of (in plane) magnetization and photon spin. Values are given at the antiresonance (AR) , the largest prepeak, and the maximum of the giant resonance of each element.

Element	Magnet./spin	AR	Prepeak	Peak
Gd	par.	250 ± 130	17 ± 3	3.0 ± 0.6
	antipar.	360 ± 270	16 ± 3	2.8 ± 0.6
Tb	par.	105 ± 20	58 ± 12	5.4 ± 1.6
	antipar.	106 ± 20	31 ± 8	8.1 ± 2.4

been carefully estimated. They result from an experimental precision of $\pm 1\%$ at the considered energy and at both ends of the photon energy range, where the spectra were matched to the tabulated data. This has been determined from the scatter of the data points about their mean values. An additional contribution results from the uncertainty in the parameter $(\mu d_e)_{\text{peak}}$ used in the saturation correction (see above). A full error propagation calculation was made considering the explicit functional relation of these quantities.

The absorption spectra at the $4d \rightarrow 4f$ thresholds display overall shapes of Fano resonances. $44,45$ In addition, up to 10 eV below the *giant resonance*, small sharp *prepeaks* can be seen, which correspond to absorption final states reached from the respective ground states (Gd ${}^{8}S_{7/2}$, Tb ${}^{7}F_{6}$) through violation of the $\Delta L=0,\pm 1$ selection rule.^{32,35}

Note that the x-ray absorption lengths of a few nanometers at the Gd and Tb $N_{4,5}$ maxima are the shortest in the periodic table. 37 They are accompanied by a huge magnetic contrast: the dichroic (difference) signals for Gd and Tb amount respectively to 33 and 60 % of the nonmagnetic signals, approximated by the average of the spectral pairs in Figs. 2 and 3.

The absorption cross section determined here for the Gd $N_{4,5}$ threshold is more than a factor of 2 higher than the calculated one for atomic Eu in Ref. 46. These two elements have the same $4d^{10}4f^7$ ground state configuration and hence the same $4d^9 4f^8$ multiplet lines, but owing to the lower (screened) nuclear charge of Eu one clearly expects differences in all parameters of the giant resonance spectral profile

TABLE II. Absorption lengths μ^{-1} (in nm) for Gd and Tb at the $3d \rightarrow 4f$ thresholds for nearly parallel and antiparallel orientation of the (in plane) magnetization and the photon spin. Values are given at the M_5 and M_4 maxima of each element as well as at the energy equidistant from the two maxima: 1200.9 eV for Gd and 1255.6 eV for Tb.

Element	Magnet./spin	$M_{\rm S}$	Betw. maxima	M_A
Gd	par.	8.4 ± 1.0	280 ± 50	36 ± 4
	antipar.	14.7 ± 1.7	300 ± 50	27 ± 3
Tb	par.	6.7 ± 1.3	330 ± 100	33 ± 7
	antipar.	15 ± 3	390 ± 130	29 ± 6

(height, width and asymmetry). In particular the Fano q parameter depends very sensitively on the radial matrix elements. Therefore a relative increase of the $4d \rightarrow 4f$ absorption maximum by a factor of 2 in going from Eu to Gd is not unreasonable. In fact a factor of 2 is found in other calculations.47

In the case of the lanthanide $M_{4,5}$ thresholds $(3d \rightarrow 4f)$ transitions), the matrix elements for Auger decay are reduced compared to the $4d \rightarrow 4f$ resonances. This causes the Fano *q* parameter to be of the order of 100.³⁵ In the limit of large \hat{q} , the Fano shape approaches a Lorentzian and, in fact, the $M_{4.5}$ absorption line shape is Lorentzian, containing hundreds of multiplet components that cluster into two main groups, the M_5 and M_4 thresholds⁴⁸ (depending on whether the spin of the 3*d* hole state is oriented parallel or antiparallel to the *l* $=$ 2 orbital angular momentum). For several characteristic photon energies, the values of the absorption length, determined after the matching to tabulated data, are presented in Table II. A quantitative determination of these values seems particularly pertinent because standard tables³⁷ do not include the $3d \rightarrow 4f$ transitions, but only the steplike $3d \rightarrow 6p$ threshold. Our results for the absorption lengths of Gd at the M_5 peak are comparable, though somewhat larger than the magnetization-averaged value measured in transmission in Ref. 49. However, as stated by the authors, their absolute

FIG. 4. Optical constants (β and δ) for (a) Gd and (b) Tb corresponding to the $N_{4,5}$ (4*d*→4*f*) thresholds of remanently inplane magnetized films with the magnetization vector nearly parallel (filled symbols) and antiparallel (open symbols) to the spin of CP photons.

FIG. 5. Optical constants (β and δ) for Gd and Tb corresponding to the M_{45} ($3d\rightarrow4f$) thresholds of remanently in-plane magnetized films with the magnetization vector nearly parallel (filled symbols) and antiparallel (open symbols) to the spin of the CP photons.

values might be affected by systematic errors. Note also the huge experimental magnetic contrast at M_5 , which leads to asymmetries of 40 and 57 % for Gd and Tb, respectively. Some time ago, Goedkoop *et al.* proposed to apply the large magnetization dependence of lanthanide $M_{4,5}$ absorption for constructing line filters to produce CP x rays.⁵⁰ Furthermore, the $M_{4,5}$ lines are clearly separated in energy even for neighboring elements in the periodic table like Gd and Tb; this represents an advantage for element-specific studies.

B. Magneto-optical constants

From the values obtained for the absorption coefficient we are able to calculate the magneto-optical constants, i.e., the real and imaginary parts of the complex index of refraction, defined as

$$
n_{\pm}(E) = 1 - \delta_{\pm}(E) - i\beta_{\pm}(E), \tag{2}
$$

where the $+$ and $-$ signs refer to the magnetization pointing either parallel or antiparallel to the CP photon spin vector, respectively. The imaginary part is directly related to the absorption coefficient through

$$
\beta_{\pm}(E) = \frac{1}{4\pi} \frac{hc}{E} \mu_{\pm}(E). \tag{3}
$$

The real parts are calculated by means of a Hilbert transformation using the KK relations. Due to the broken time reversal symmetry inside a magnetized medium, the complex refractive index satisfies the symmetry relation $n_{\pm}(-E^*)$ $=n^*$ (*E*), and one has to apply the modified KK equations⁵¹

$$
\delta_{+}(E) + \delta_{-}(E) = -\frac{2}{\pi} \int_{0}^{\infty} E' \frac{\beta_{+}(E') + \beta_{-}(E')}{(E')^{2} - E^{2}} dE',
$$

$$
\delta_{+}(E) - \delta_{-}(E) = -\frac{2E}{\pi} \int_{0}^{\infty} \frac{\beta_{+}(E') - \beta_{-}(E')}{(E')^{2} - E^{2}} dE'. \quad (4)
$$

In order to perform these integrations over the largest possible energy range, we resumed to the tabulated values for the optical constants outside the measured regions in the entire range from 0 to 30 keV taken from the compilation of Henke *et al.*³⁷ We have included the relativistic correction affecting the asymptotic behavior of the real part of the atomic scattering factor at large photon energies

FIG. 6. Calculated (a) and measured (b) x-ray reflectivity spectra in the $M_{4,5}$ region of a remanently magnetized Gd film for nearly parallel (filled symbols) and antiparallel (open symbols) orientation of the magnetization with respect to the light propagation direction. Nominal film thickness is 11 ± 1 nm and the light incidence angle is $10\pm1^\circ$. The values used for the calculated spectra are 12 nm and 11° . The inset in (a) shows the two light paths considered in the calculation.

$$
f'(E) \to Z^* = Z - (Z/82.5)^{2.37},\tag{5}
$$

where *Z* is the atomic number, as given by the fit of Henke et al.³⁷ to the tabulated values of Kissel and Pratt.⁵² The atomic scattering factor and the optical constants are related through

$$
\delta_{\pm}(E) = \frac{r_0 h^2 c^2 n_{at}}{2 \pi E^2} f'_{\pm}(E),
$$

$$
\beta_{\pm}(E) = \frac{r_0 h^2 c^2 n_{at}}{2 \pi E^2} f''_{\pm}(E),
$$
 (6)

where r_0 is the classical electron radius and n_{at} is the atomic density of the element.

The calculated real parts of the refractive index of Gd and Tb at the $N_{4,5}$ and $M_{4,5}$ absorption thresholds are shown in Figs. 4 and 5 for opposite orientations of the magnetization and the photon spin vector. They exhibit the well-known dispersive behavior, with tails ranging far beyond the associated imaginary parts. While the latter give the highest magnetic contrast in the absorption maxima, the real parts also provide magnetic contrast in regions where the absorption is small. This allows the performance of magnetizationdependent measurements in reflectivity in regions of different light-penetration depths, as the reflected signal is determined by both the real and the imaginary parts of the refractive index.

The exact values for n_+ can only be obtained in an ideal experiment where the magnetization direction and the photon spin are strictly colinear. Since this is not feasible for inplane magnetized films, there is a contribution of transitions with $\Delta m_J = 0$ with a weight $1/2 \sin^2 \theta$ (Ref. 53) to both the experimentally obtained n_+ and n_- . This amounts to only a few percent for incidence angles up to 30°. Contributions from the opposite magnetization, scaling as $1/2(1-\cos\theta)^2$, can be safely neglected for those angles. The contribution of Δm_J =0 transitions cancels out if only the difference in the MO constants $n_{+}-n_{-}$ is required, as in, for example, the calculation of the Faraday effect.²⁸

C. Reflectivity

As an application of the magneto-optical constants determined in this way, we present in Fig. $6(a)$ calculated reflectivity spectra of a Gd film in the region of the $M_{4.5}$ threshold. The reflectivity was calculated using the Jones matrix formalism and the Fresnel equations for transmission and reflectivity at interfaces of magnetized media.^{54,55} We considered interference between two channels [see inset in Fig. $6(a)$: (1) represents the reflection at the vacuum/Gd interface and (2) comprises transmission through this interface, propagation in the Gd film (which includes absorption and Faraday effect), reflection at the $Gd/W(110)$ surface, propagation back through the film, and transmission through the Gd/vacuum interface. Higher-order paths, including multiple reflections, are found to contribute negligibly to the reflected intensity. In the calculation, we employed the MO constants for Gd at the $M_{4,5}$ thresholds shown in Fig. 5, together with the values for the W substrate taken from Ref. 37.

Figure $6(b)$ shows the corresponding experimental reflectivity spectra of a Gd film on $W(110)$. The nominal film thickness is $d=11\pm1$ nm and the light incidence angle is θ =10^{\pm}1[°]. The experimental spectra are well reproduced by the calculation shown in Fig. $6(a)$, including their fine structure. The best agreement between calculated and experimental spectra was achieved by setting $d=12$ nm and $\theta=11^{\circ}$ in the calculation, in good agreement with the experimental values. Quantification of the experimental reflectivity *R* was done by normalizing using the diode signal in the direct beam. Considering the uncertainties in detector position and the simplicity of the calculation (for example, no roughness in the film was considered) the agreement of the calculated absolute intensities with the measured ones can be considered satisfactory.

In conclusion, by quantifying magnetization-dependent absorption spectra of Gd and Tb, we were able to obtain the values of the MO constants of both elements at the $4d \rightarrow 4f$ and $3d \rightarrow 4f$ excitation thresholds in the soft x-ray region. For the example of Gd $3d \rightarrow 4f$, we have shown that the x-ray MO constants obtained here can serve as input for reflectivity model calculations of thin lanthanide films. In this way, our results open the possibility to apply the power of soft x-ray reflectivity to element-specific studies of nano-

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scaled layered magnetic systems containing lanthanide elements.

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