Selective optical bleaching of Cr³⁺ luminescence at low temperature in MgO codoped near stoichiometric LiNbO₃:Cr crystals

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Selective *R*-line emission of the Cr^{3+} ions in near stoichiometric LiNbO₃ crystals codoped with MgO has been studied at low temperature under excitation into the ${}^{4}A_{2} \rightarrow {}^{4}T_{1}$ transition. The emission intensity of *R* lines associated with Cr^{3+} ions located in Nb⁵⁺ sites, $[Cr]_{Nb}$ centers, was observed to decrease with the illumination time, whereas those ascribed to other Cr^{3+} ions, such as those located in the Li site, $[Cr]_{Li}$ centers, are optically stable. The selective bleaching is explained by considering the photoinization of Cr^{3+} ions in Nb sites and the formation of Nb⁴⁺ complexes (small polaron).

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LiNbO₃ doped with rare earth and transition metals is an interesting material for applications in linear and nonlinear optically active devices. In pure LiNbO3 crystals of congruent composition (where the Li/Nb ratio is 0.946) an optically active dopant such as rare-earth or transition-metal ions are incorporated into the host matrix in the octahedral Li⁺ lattice site.¹ In the case of Cr^{3+} ions, the as-grown crystal is uniformly green in color because of the broad absorption band in the visible region ascribed to transitions to the ${}^{4}T_{1}$ and ⁴ T_2 energy levels.^{2,3} Several different Cr³⁺ defect centers have been identified.^{2,4,5} the center denoted γ -[Cr]_{Li} corresponds to undisturbed Cr³⁺ in the Li⁺ lattice site; the centers α -[Cr]_{Li} and β -[Cr]_{Li} which are perturbed by [Nb]_{Li} and [V]_{Li} (V: vacancy), respectively. In order to substitute the Nb^{5+} sites with Cr^{3+} ions additional codoping with other divalent cations, such as Mg²⁺ or Zn²⁺, above a certain threshold corresponding with the composition of the LiNbO₃, is required. In the case of congruent composition this is \sim 4.5 mol %. At cation concentrations slightly higher than this concentration, there is an abrupt change in the crystal's color, from green to pink.³ At these high concentrations of cations both optical and paramagnetic resonance techniques detected new Cr³⁺ centers, in coexistence with the $[Cr]_{Li}$ centers, which have been ascribed to Cr^{3+} ions located in the Nb⁵⁺ sites ($[Cr]_{Nb}$ centers).³ The isotropic electron paramagnetic resonance (EPR) signal associated with a new center is explained considering that a Mg²⁺ ion close to the Cr^{3+} ion perturbs the center, producing a more cubic symmetry.^{6,7} As the threshold value of \sim 4.5 mol % coincides with the concentration of Li⁺ defect sites-Li⁺ vacancy, $[V]_{Li}$, and antisites ($[Nb]_{Li}$, Nb^{5+} ion located in Li^+ vacancy) in congruent LiNbO3 samples-it has been suggested that the appearance of the $[Cr]_{Nb}$ center starts when all Li⁺ defect sites are filled by dopant divalent cations. In such circumstances, Cr³⁺ ions can be incorporated in both Li⁺ and Nb⁵⁺ sites.³

In near stoichiometric LiNbO₃ samples, where the ratio of Li/Nb is \sim 1, the total number of defects (Li⁺ vacancy and antisites) is expected to be much less and hence the cation concentration threshold is lower than in congruent samples. In fact, it has recently been reported that pink-colored near stoichiometric crystals can be grown by incorporating only 1 mol % of MgO in the melt.⁸

Electron traps, in LiNbO₃ crystals, have been related to cation impurities as well as Nb⁵⁺ antisites and intrinsic Nb⁵⁺ ions. In the case associated with the Nb⁵⁺ ions the electron capture is expected to be delocalized onto the surrounding ions, forming a Nb⁴⁺ complex, which can be considered as a small polaron⁹ involving either a regular Nb_{Nb} or a Nb_{Li} antisite. In an early work, it has also been suggested that Mg²⁺ ions are also an effective electron trap, forming Mg⁺ complexes similar to the Nb⁴⁺ complex when the Mg²⁺ ions concentration is above the 4.5 mol %.¹⁰ However, more recently it has been reported that in MgO and ZnO doped LiNbO₃, the electron trap is associated with the Nb⁵⁺ ions (antisites and intrinsic sites) and not with divalent cation impurities.¹¹

The system of LiNbO₃:Cr:MgO doped with a cation concentration high enough to induce the coexistence of the two Cr^{3+} centers would therefore be a particularly interesting case to study the electron trapping property of this system. This paper reports preliminary results of the observed optical bleaching of the spectral lines of Cr^{3+} centers in near stoichiometric LiNbO₃:Cr (0.2 mol %) codoped with 2 mol % of MgO.

Near stoichiometric LiNbO₃ crystal doped with 0.2 mol % of Cr_2O_3 and with 2% of MgO in the melt was grown in air by the top-seeded solution pulling method at a pull rate of 0.3 mm/h. The as-grown boule shows a uniform pink color along the full length of the boule. Absorption spectra were measured with a Varian Cary-SE spectrophotometer. Emission spectra were excited using an Ar-ion laser at 514.5 nm.



FIG. 1. 10-K photoluminescence spectra for different illumination time periods.

The emitted light was dispersed in a SPEX 500M monochromator and detected with a Hamamatsu R-2949 photomultiplier or a charge-coupled device (CCD) array detector when a fast wavelength scan was required.

Figure 1 shows the emission spectra under continuouswave (cw) 514.5-nm argon laser excitation (50 mW) at 20 K for different illumination time periods. The 514.5-nm wavelength excites the ${}^{4}A_{2} \rightarrow {}^{4}T_{1}$ transition of the [Cr]_{Nb} centers and also the [Cr]_{Li} centers.¹² The *R* lines ascribed to the α -and β -[Cr]_{Li} centers^{2,4,5} are observed together with those associated with the $[Cr]_{Nb}$ center (ε in Macfarlane's notation⁴) centered at 740.3 and 738.0 nm. The emission of these two R lines has been previously reported in congruent MgO codoped LiNbO₃ and MgO codoped near stoichio-metric LiNbO₃ crystals.^{2,4} The emission of the R lines associated with γ centers is only observed at higher temperatures (>50 K).^{5,12} It is important to note that, at difference with the spectra previously published, the intensity of the R lines ascribed to the $[Cr]_{Nb}$ center are initially more intense than that of the α -[Cr]_{Li} center and decreases with increasing illumination time, whereas the emission intensity of the Rlines ascribed to the α - and β -[Cr]_{Li} centers remains constant. The emission intensity ratio of the two R lines ascribed to the [Cr]_{Nb} center remains relatively constant as they decrease with increasing illumination time. This result clearly illustrates that a selective optical bleaching mechanism is taking place in this crystal system and is only related to this $[Cr]_{Nb}$ center. The previously published spectra are similar to our spectra after long-time excitation.

Figure 2 shows the dependence of the 740.3-nm *R*-line emission time decays as a function of excitation power density. These intensity decay curves show a bleaching of the emission intensity of 33% and 86% for \sim 3.3 and \sim 6 W/mm² excitation power density, respectively, and reaching a similar "plateau" level for long illumination time, suggesting some sort of equilibrium condition was reached. After a dark period of several minutes at 10 K, a recovery of only a few percent of the luminescence intensity is observed, whereas a full recovery is obtained only after heating the sample up to room temperature.



FIG. 2. Variation of the 740.3-nm *R*-line emission time decays as a function of excitation power density at 10 K.

Figure 3 shows the absorption spectra before and after illumination (\sim 1 h) with the 514.5-nm argon laser line at 10 K. In this experiment the front and back surfaces of the crystal are masked showing only an aperture of $\approx 2 \text{ mm}$ in diameter. The laser beam was expanded in order to obtain uniform illumination over the aperture, which was subsequently used for the absorption measurements. An illumination power density of $\sim 1 \text{ W/mm}^2$ was estimated. The absorption spectrum before bleaching is essentially due to the broad absorption bands ascribed to [Cr]_{Li} and [Cr]_{Nb} centers.¹² The presence of [Cr]_{Li} centers in the overall spectrum only becomes more noticeable after subtracting the contribution of the absorption due to this $[Cr]_{Nb}$ center. Figure 3 shows that after illumination at 514.5 nm the optical absorption due to this [Cr]_{Nb} center decreases, which is in agreement with the bleaching behavior observed of the R-line emission associated with this center. However, the bleaching magnitude is less than that reported in Fig. 1 for the R lines, due to the different illumination light power density used in the two experimental setups. It is important to note that a broadband



FIG. 3. Absorption spectra before and after illumination with the 514.5-nm argon laser line at 10 K (power density \sim 1 W/mm²).

centered at 1200 nm (\sim 1 eV) emerged after 514.5 nm illumination. A similar absorption band has been reported in MgO and ZnO doped LiNbO₃ crystals and has been ascribed to a small polaron (Nb⁴⁺) formed by a trapped electron.¹¹ Although Nb⁴⁺ polarons have also been reported located in antisites, however in the present case this possibility is discarded because in near stoichiometric LiNbO₃:Cr:MgO crystals the expectation of much reduced intrinsic defects suggest a very low concentration of antisites. Furthermore, the absorption bands ascribed to antisite polaron is at 1.6 eV.¹³

In previous works it has been established that the highenergy side of the broad ${}^{4}A_{2} \rightarrow {}^{4}T_{1}$ transition overlapped with the charge transfer region in MgO and ZnO codoped LiNbO₃:Cr crystals.³ Taking this into account, the optical bleaching of Cr³⁺ ions could be achieved if a stable electron trap is involved in the process. As commented above, electron traps together with the Nb⁴⁺ complexes can be formed in MgO doped congruent LiNbO₃ crystals when the cation concentrations exceed the 4.5% threshold. It is therefore not unreasonable that a similar situation is found in near stoichiometric crystals, but at a lower MgO concentration.

The structure suggested by EPR and electron-nuclear double-resonance (ENDOR) measurements for this $[Cr]_{Nb}$ center is a Cr^{3+} ion substituting into a Nb^{5+} site with a Mg^{2+} ion in close proximity forming a $Cr^{3+}-Mg^{2+}$ complex.⁷ The precise position of the Mg^{2+} ion is undetermined: however, a theoretical calculation by Yang *et al.*¹⁴ suggested the Mg^{2+} ions may enter the structural vacancy sites on the [111] axis above the Cr^{3+} ions. If the high-energy band of the $[Cr]_{Nb}$ centers is photoexcited, the ionized electron could be trapped by the Nb^{5+} ions, resulting in the formation of small polarons (Nb^{4+} complex) and $Cr^{4+}-Mg^{2+}$ complexes. This is supported by the presence of the absorption band centered at 1200 nm ascribed to small polarons and the decrease of the concentration of the initial $Cr^{3+}-Mg^{2+}$ complexes.

In a first approximation calculation, taking N(t) to be the ε -[Cr]_{Nb} center concentration and the intrinsic Nb⁵⁺ ions the unique electron traps, the time dependence (dN/dt) under illumination can be written as

$$dN(t)/dt = -\sigma(\lambda)I(\lambda)DN(t) + kN'(t), \qquad (1)$$

where $\sigma(\lambda)$ the absorption cross section of the $[Cr]_{Nb}$, $I(\lambda)$ the light intensity, D a factor proportional to the decay time of the ${}^{4}T_{1} \rightarrow {}^{4}A_{2}$ radiative transition, and k the Nb⁴⁺ thermal ionization coefficient. $N_{0} = N(t) + N'(t)$ is the initial concentration of $[Cr]_{Nb}$ centers where N(t) is the $[Cr]_{Nb}$ center concentration and N'(t) is the concentration of the Nb⁴⁺ complex. The analytical solution of Eq. (1) is

$$N = A + B \exp[-(t/\tau_0)], \qquad (2)$$

$$A = kN_0 / (\sigma DI + k),$$
$$B = I\sigma DN_0 / (\sigma DI + k),$$



FIG. 4. (a) Variation of the luminescence decay curves of the *R*-line emission with the illumination wavelengths. (b) Dependence of the B_0 and $1/\tau_0$ parameters. The line corresponds to the optical density of the sample before bleaching.

$$1/\tau_0 = I\sigma D + k. \tag{3}$$

The luminescence intensity L(t) is given by

$$L(t) = I\sigma(\lambda)N(t) = A_0 + B_0 \exp[-(t/\tau_0)], \quad (4)$$

where the expression for the A_0 and B_0 parameters is

$$A_0 = \sigma I k N_0 / (\sigma D I + k),$$

$$B_0 = I^2 \sigma^2 D N_0 / (\sigma D I + k),$$
 (5)

and the expression for $1/\tau_0$ is unchanged.

Figure 4(a) shows the luminescence decay curves of the R-line emission (740.3 nm) of the $[Cr]_{Nb}$ center at 10 K for three different excitation wavelengths with the same excitation power (~3.3 W mm²). An acceptable fit to the experimental decay curves is achieved using expression (4). According to expression (5) and considering that $\sigma DI > k$ (for a weak thermal recovery), both parameters B_0 and $1/\tau_0$ are proportional to σ and therefore they have to follow the form of the broadband associated with the ${}^{4}A_{2} \rightarrow {}^{4}T_{1}$ transition of the $[Cr]_{Nb}$ centers. This is presented in Fig. 4(b) where the variation of B_0 and $1/\tau_0$ as a function of the excitation wavelengths is plotted along with the high-energy absorption broadband of the $[Cr]_{Nb}$ centers.

In summary, this preliminary study revealed the following important points: (i) the photoionization of Cr^{3+} ions is

only effective when the active ion is located in the Nb^{5+} site forming a [Cr]_{Nb} center; (ii) the role played by cations such as Zn^{2+} and Mg^{2+} in the optical stability is twofold: forcing the substitution of Cr^{3+} ions in the Nb^{5+} sites and destroying the Nb^{5+} antisites; (iii) the presence of the absorption band centered at 1200 nm and the bleaching of the [Cr]_{Nb} center seems to give more evidence for the polaron electron trap model; (iv) a single exponential decay function described in the simple model used in this work can explain the general characteristic of the observed bleaching behavior.

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Finally, this work reinforced the role of the Cr^{3+} ions as a successful optical probe for the study of defect structure especially in LiNbO₃ crystal.

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