

Vortex matter in layered superconductors without Josephson coupling: Numerical simulations within a mean-field approach

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We study vortex matter in layered superconductors in the limit of zero Josephson coupling. The long range of the interaction between pancake vortices in the c direction allows us to employ a mean-field method: all attractive interlayer interactions are reduced to an effective substrate potential, which pancakes experience in addition to the same-layer pancake repulsion. We perform numerical simulations of this mean-field model using two independent numerical implementations with different simulation methods (Monte Carlo sampling and Langevin molecular dynamics). The substrate potential is updated self-consistently from the averaged pancake density. Depending on temperature, this potential converges to a periodic profile (crystal) or vanishes (liquid). We compute thermodynamic properties of the system, such as the melting line, the instability line of the crystal, and the entropy jump across the melting transition. The simulation results are in good agreement with approximate analytical calculations.

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I. INTRODUCTION

The vortex state in type-II superconductors is a complex physical system. Within the layered high-temperature materials, such as $\text{Bi}_2\text{Sr}_2\text{CaCu}_2\text{O}_8$ (BSCCO) or $\text{YBa}_2\text{Cu}_3\text{O}_7$ (YBCO), vortex lines can be understood as wiggling stacks of pancake vortices.^{1–5} The thermodynamic properties of the vortex state are determined by the interaction between pancake vortices. There are two mechanisms of pancake interaction: (i) electromagnetic interaction and (ii) Josephson coupling. The electromagnetic interaction is mediated by supercurrents circulating around each pancake, whereas the Josephson coupling results from the energy cost due to a phase shift between the superconducting order parameters in the neighboring layers.

To understand the phase diagram of high-temperature superconductors and in particular the melting line of the vortex lattice,^{6,7} we need to gain insight into the behavior of vortex matter under a variety of experimental conditions. In moderately anisotropic materials, such as YBCO, the short-range Josephson coupling is the dominant interlayer interaction, and the vortices are well described as elastic strings.^{8–13} In very anisotropic materials, on the other hand, such as BSCCO, the Josephson coupling is weak, and the long-range electromagnetic interaction between the pancakes should be taken into account. In this paper we consider very anisotropic materials in the absence of Josephson coupling and neglect pinning. Even after keeping only the electromagnetic coupling, the problem remains challenging, due to the long range of the interactions: the energy of electromagnetic in-

teraction between two pancakes depends logarithmically on the separation along the layers and decays exponentially with the number of layers between the pancakes. More specifically, the interaction is repulsive between pancakes in the same layer and attractive between pancakes in different layers, and the decay length of the exponential dependence is the London penetration depth λ , which is typically 100 times larger than the layer spacing s . Approximately, this system has been investigated within the density-functional theory.^{14–16}

For a numerical investigation of the system, one can in principle simulate directly a stack of two-dimensional (2D) pancake systems taking into account all of the interlayer interactions. However, the computational challenge is that the interlayer attraction between pancakes extends over a range of $2\lambda/s \sim 100$ –150 layers. In addition, realistic simulation of the melting transition requires at least several hundred point vortices per layer. So far, direct numerical investigations have been performed only on small systems using about 10 layers and of the order of 100 vortices.^{17–22} This is not sufficient to describe realistically the vortex state in BSCCO. With today's computational resources, it is not feasible to perform realistic direct three-dimensional (3D) simulations of this system because the necessary computational effort grows quadratically with the number of layers.

Fortunately, one can benefit from the long range of the interlayer coupling. As the interlayer force on a pancake is the result of a sum of a large number ($\sim 2\lambda/s$) of small contributions, it can be calculated by a mean-field approach.

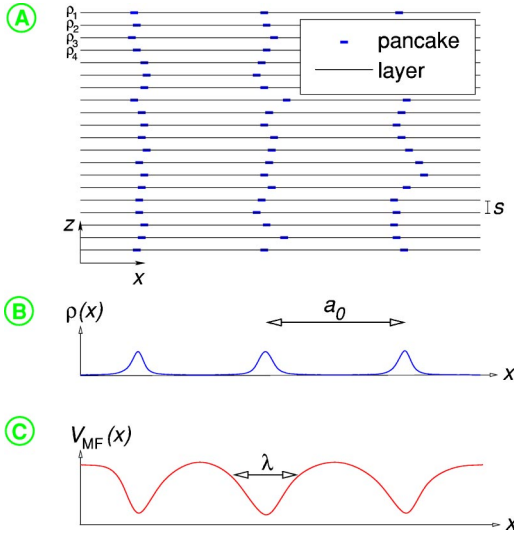


FIG. 1. Schematic representation of the substrate model. (A) The pancake positions $\rho_n(x)$ in each layer n are (B) averaged over the layers in order to obtain the averaged pancake density $\rho(x)$. From the average pancake density we compute (C) the substrate potential $V_{MF}(x)$, which is smeared over a length of the order of λ . The vortex lattice spacing is a_0 and s is the layer spacing.

The exact value of this force is determined by the instantaneous pancake densities in the large number of layers. In the crystal state the instantaneous density can be decomposed into the average density, which is a periodic function of the in-plane coordinates, and a fluctuating contribution. In the mean-field approach to the interlayer interactions, one replaces the instantaneous densities in the other layers by the average density. This approach gives a quantitatively correct description of the system, because due to the law of large numbers, the neglected force from the fluctuating densities is typically smaller than the average interlayer force by the factor $\sim \sqrt{s/\lambda} \ll 1$. The calculation then takes the form of independent layers, with the pancakes in each layer subject to an effective ‘‘substrate potential.’’²³ This substrate potential is the cumulative affect of the attraction of pancakes in all other layers as illustrated in Fig. 1. Pancakes within one layer interact directly with each other, whereas the interaction with pancakes in other layers is mediated via the substrate potential. Thus, each layer is treated individually, until a new substrate potential can be computed. This process is iterated, until the substrate has converged to a steady solution. In this paper, we present numerical implementations of this substrate model and show results that we compare with the semianalytic approximations given in Ref. 23.

We summarize this work in Fig. 2. On the top, the central idea is visualized: pancakes experience attractive interlayer interactions through the substrate potential that stabilizes the pancake crystal. On the bottom, we show the computed melting line separating a 3D pancake vortex lattice from decoupled 2D liquids. We express magnetic induction in units of $B_\lambda \equiv \Phi_0/\lambda^2$, where Φ_0 is the magnetic flux quantum, such that the pancake spacing in a triangular lattice is $a_0 = (2/\sqrt{3})^{1/2}\lambda \approx 1.07\lambda$ at $B = B_\lambda$. We use a dimensionless

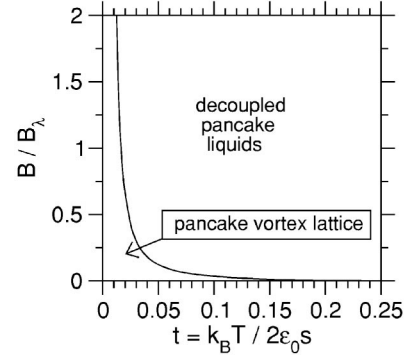
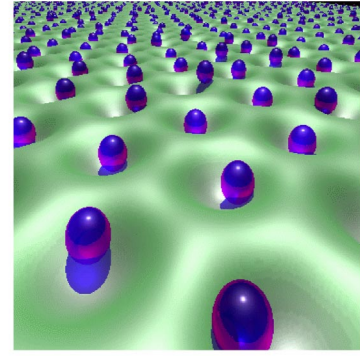


FIG. 2. Top: Snapshot of the pancakes (visualized by spheres) placed onto the substrate potential (visualized as a surface) at $B = B_\lambda$ and $t = 1/59 \approx 0.017$ just below the melting transition to demonstrate how the substrate potential constrains the pancake motion. Bottom: The phase diagram we have computed using the substrate method.

temperature, t , which is the ratio of the thermal energy $k_B T$ to the prefactor $2s\epsilon_0$ of the logarithmic pancake-pancake interaction,

$$t \equiv \frac{1}{\Gamma} \equiv \frac{k_B T}{2\epsilon_0 s}, \quad (1)$$

where $\epsilon_0 = \Phi_0^2/(4\pi\mu_0\lambda^2)$, μ_0 is the vacuum permeability, and s is the layer spacing. This allows us to compare our results with outcomes from 2D one-component Coulomb plasma simulations,^{24–26} where frequently $\Gamma = 1/t$ is used to express temperatures. At low fields, the electromagnetic attraction of range $\lambda \gg s$ between pancakes in different layers stabilizes the 3D pancake-vortex lattice. Increasing the magnetic field decreases the relative strength of the interlayer coupling. At high fields, $B \gg B_\lambda$, the long-range repulsive interaction within the layer dominates, and the 3D pancake lattice melts at a temperature close to the 2D melting temperature.

In Sec. II we describe the substrate model in detail, including three different methods for the efficient computation of the substrate potential (Sec. II C). The results, including the equilibrium phase diagram, are shown in Sec. III before we conclude in Sec. IV. The Appendix gives a derivation of the correlation correction to the free energy, and shows that our mean-field approach should be accurate to order s/λ .

II. MEAN-FIELD APPROACH (SUBSTRATE MODEL)

A. The mean-field interlayer coupling

The in-layer energy \mathcal{E}^{in} and the interlayer energy $\mathcal{E}^{\text{inter}}$ of a system of electromagnetically interacting pancakes in a layered superconductor are, respectively,

$$\mathcal{E}^{\text{in}} = \sum_n E_n^{\text{in}} = \sum_n \frac{1}{2} \sum_{j' \neq j} U(\mathbf{R}_j^n - \mathbf{R}_{j'}^n, 0) \quad (2)$$

and

$$\mathcal{E}^{\text{inter}} = \frac{1}{2} \sum_{n' \neq n} \sum_{j, j'} U(\mathbf{R}_j^n - \mathbf{R}_{j'}^{n'}, n - n'). \quad (3)$$

Indices n and n' count over layers and j and j' over pancakes in the layers, \mathbf{R}_j^n is the (2D) position of pancake j in layer n , and $U(\mathbf{R}, n)$ is the coupling energy for two pancakes separated by a vector (\mathbf{R}, z) , where $z = ns$, with s being the layer spacing. The z axis is chosen perpendicular to the layers.

The in-layer pancake interaction⁴ is

$$U(\mathbf{r}, 0) = 2\epsilon_0 s \left[\left(1 - \frac{s}{2\lambda} \right) \ln\left(\frac{L}{r}\right) + \frac{s}{2\lambda} \int_r^\infty dr' \frac{\exp(-r'/\lambda)}{r'} \right] \quad (4)$$

and the interlayer interaction ($n \neq 0$) is

$$U(\mathbf{r}, n) = -\frac{\epsilon_0 s^2}{\lambda} \left[\exp\left(-\frac{ns}{\lambda}\right) \ln\left(\frac{L}{r}\right) - \int_r^\infty dr' \frac{\exp[-\sqrt{r'^2 + (ns)^2}/\lambda]}{r'} \right]. \quad (5)$$

Using

$$\rho_n(\mathbf{r}) = \sum_j \delta(\mathbf{r} - \mathbf{R}_j^n), \quad (6)$$

we rewrite

$$\mathcal{E}^{\text{inter}} = \frac{1}{2} \sum_{n \neq n'} \int d^2 r d^2 r' \rho_n(\mathbf{r}) \rho_{n'}(\mathbf{r}') U(\mathbf{r} - \mathbf{r}', n - n'). \quad (7)$$

We separate pancake density fluctuations from the layer average density

$$\rho(\mathbf{r}) \equiv \langle \rho_n(\mathbf{r}) \rangle, \quad (8)$$

$$\rho_n(\mathbf{r}) = \rho(\mathbf{r}) + \delta\rho_n(\mathbf{r}), \quad (9)$$

and obtain from Eq. (7)

$$\begin{aligned} \mathcal{E}^{\text{inter}} = & \frac{1}{2} \sum_{n \neq n'} \int d^2 r d^2 r' U(\mathbf{r} - \mathbf{r}', n - n') \\ & \times [\rho(\mathbf{r})\rho(\mathbf{r}') + 2\rho(\mathbf{r}')\delta\rho_n(\mathbf{r}) + \delta\rho_n(\mathbf{r})\delta\rho_{n'}(\mathbf{r}')]. \end{aligned} \quad (10)$$

Because the difference $n - n'$ in the last sum extends over a very large number of layers ($\sim \lambda/s$), a typical value of the sum $\sum_{n'} U(\mathbf{r} - \mathbf{r}', n - n') 2\rho(\mathbf{r}')$ is larger than a typical value of the sum $\sum_{n'} U(\mathbf{r} - \mathbf{r}', n - n') \delta\rho_{n'}$ by the factor $\sim \sqrt{\lambda/s}$. Again, the law of large numbers allows us to neglect the last term in Eq. (10), leading to the mean-field description of the interlayer interactions. A more precise justification is given in the Appendix, where the free-energy correction due to the correlation term is shown to be smaller than the mean-field free energy by the factor s/λ .

Separating the pancake density into the average value $\rho = \langle \rho_n(\mathbf{r}) \rangle$ and a modulating part, we can split the total magnetic coupling energy into two parts, each with a quite different meaning. The part containing the average density does not depend on temperature and formally diverges due to the logarithmic term in Eq. (5). This divergence exactly compensates a similar divergence in the in-plane energy. Within the mean-field approach the part of the coupling energy sensitive to density variations is finite only in the crystal state. In the liquid state it vanishes.

For the mean-field interlayer energy \mathcal{E}^{MF} we obtain from Eq. (10)

$$\begin{aligned} \mathcal{E}^{\text{MF}} = & \sum_n E_n^{\text{MF}} \\ = & \frac{1}{2} \sum_{n \neq n'} \int d^2 r d^2 r' U(\mathbf{r} - \mathbf{r}', n - n') \\ & \times [\rho(\mathbf{r})\rho(\mathbf{r}') + 2\rho(\mathbf{r}')\delta\rho_n(\mathbf{r})] \end{aligned} \quad (11)$$

$$= \frac{1}{2} \sum_n \int d^2 r V_{\text{MF}}(\mathbf{r}) \rho(\mathbf{r}) + \sum_n \int d^2 r V_{\text{MF}}(\mathbf{r}) \delta\rho_n(\mathbf{r}) \quad (12)$$

$$\begin{aligned} = & -\frac{1}{2} \sum_n \int d^2 r V_{\text{MF}}(\mathbf{r}) \rho(\mathbf{r}) \\ & + \sum_n \int d^2 r V_{\text{MF}}(\mathbf{r}) \rho_n(\mathbf{r}) \quad \text{via Eq. (9)}. \end{aligned} \quad (13)$$

The last term describes fluctuations in the fixed substrate potential V_{MF} ,

$$V_{\text{MF}}(\mathbf{r}) = \int d^2 r' \left[\sum_{n \neq 0} U(\mathbf{r} - \mathbf{r}', n) \right] \rho(\mathbf{r}') \quad (14)$$

$$= \int d^2 r' \mathcal{U}(\mathbf{r} - \mathbf{r}') \rho(\mathbf{r}') \quad (15)$$

$$\equiv (\mathcal{U} * \rho)(\mathbf{r}), \quad (16)$$

where $*$ is the convolution operator and

$$\mathcal{U}(\mathbf{r}) \equiv \sum_{n \neq 0} U(\mathbf{r}, n). \quad (17)$$

$\mathcal{U}(\mathbf{r})$ is the interaction potential of a pancake separated by \mathbf{r} from a stack of pancakes minus the interaction of the (missing) pancake in the same layer and is given by⁴

$$\mathcal{U}(\mathbf{r}) = 2\epsilon_0 s K_0\left(\frac{r}{\lambda}\right) - U(\mathbf{r}, 0), \quad (18)$$

with $K_0(x)$ being a modified Bessel function of the second kind. Ignoring terms of the order of $s/2\lambda$, the pancake-pancake repulsion (4) simplifies to

$$U(\mathbf{r}, 0) = 2\epsilon_0 s \ln\left(\frac{L}{r}\right). \quad (19)$$

In our calculations we find it useful to use the form in Fourier space,²⁷

$$\mathcal{U}(\mathbf{q}) = 4\pi\epsilon_0 s \left(\frac{1}{\lambda^{-2} + q^2} - \frac{1}{q^2} \right) \quad (20)$$

$$= -4\pi\epsilon_0 s \frac{\lambda^{-2}}{q^2(\lambda^{-2} + q^2)}. \quad (21)$$

B. Algorithm

In principle, the substrate model can be implemented as follows:

(1) Assume initial pancake densities $\rho_n(\mathbf{r})$, for example, a hexagonal lattice in each layer n .

(2) Average the pancake density $\rho_n(\mathbf{r})$ over all layers to obtain $\rho(\mathbf{r})$, Eq. (8).

(3) Compute the substrate potential $V_{\text{MF}}(\mathbf{r})$, Eq. (16), by convoluting the substrate interaction kernel $\mathcal{U}(\mathbf{r})$, Eq. (18), with the average pancake density $\rho(\mathbf{r})$:

$$V_{\text{MF}}(\mathbf{r}) = (\mathcal{U}^* \rho)(\mathbf{r}). \quad (22)$$

(4) For each layer n compute the pancake distribution $\rho_n(\mathbf{r})$ using Monte Carlo or Langevin dynamics simulations. The total energy for layer n contains the direct pancake-pancake interaction within the layer [Eq. (2)]

$$E_n^{\text{in}} = \frac{1}{2} \sum_{j' \neq j} U(\mathbf{R}_j^n - \mathbf{R}_{j'}^n, 0), \quad (23)$$

and the relevant interaction with pancakes in other layers via the substrate potential (13)

$$\begin{aligned} E_n^{\text{MF}} &= -\frac{1}{2} \int d^2r V_{\text{MF}}(\mathbf{r}) \rho(\mathbf{r}) + \int d^2r V_{\text{MF}}(\mathbf{r}) \rho_n(\mathbf{r}) \\ &= -E^{\text{coup}} + \sum_j V_{\text{MF}}(\mathbf{R}_j^n) \quad \text{via Eq. (6)}. \end{aligned} \quad (24)$$

$E_n^{\text{coup}} = \frac{1}{2} \int d^2r V_{\text{MF}}(\mathbf{r}) \rho(\mathbf{r})$ is constant for a given $\rho(\mathbf{r})$ and can therefore be ignored within the Monte Carlo and Langevin simulation as it only shifts the energy scale.

(5) Go to (2), until V_{MF} (or ρ) has converged.

Since the substrate potential V_{MF} in step (4) is the same for all layers, we can compute $\rho_n(\mathbf{r})$ for many Langevin-dynamics time steps (or Monte Carlo sweeps) rather than many layers. Therefore, in order to obtain the averaged pancake density $\rho(\mathbf{r})$ in step (2), we average over time steps (or sweeps) computed in one layer rather than averaging over

layers. Using the substrate potential, we reduce the solution of the 3D problem to performing one 2D simulation in the presence of the iteratively refined substrate potential.

C. Numerical implementation

We exploit the convolution theorem and compute the substrate potential in Fourier space²⁷

$$V_{\text{MF}}(\mathbf{r}) = (\mathcal{U}^* \rho)(\mathbf{r}) \quad \text{via Eq. (16)} \quad (25)$$

$$= \int \frac{d^2q}{(2\pi)^2} \mathcal{U}(\mathbf{q}) \rho(\mathbf{q}) \exp(i\mathbf{q} \cdot \mathbf{r}) \quad (26)$$

using the analytical Fourier transform $\mathcal{U}(\mathbf{q})$ as given in Eq. (21), and the numerically computed

$$\rho(\mathbf{q}) = \int d^2r \rho(\mathbf{r}) \exp(-i\mathbf{q} \cdot \mathbf{r}). \quad (27)$$

This has two advantages: First, we do not cut off the interaction kernel \mathcal{U} within the simulation cell as would be the case in the real-space convolution. Second, this is numerically more efficient than performing the convolution (15) directly. We have used three different methods for computing $V_{\text{MF}}(\mathbf{r})$ numerically.

1. The full method

The ‘‘full method’’ computes the substrate potential V_{MF} using the full spectrum $\rho(\mathbf{q})$ of Fourier components of the average pancake density $\rho(\mathbf{r})$ as shown in Eq. (26). In our simulations we use a resolution of $\approx 100^2$ grid cells per pancake in order to compute $\rho(\mathbf{r})$ as an average over time steps or sweeps. This results in reciprocal lattice vectors up to magnitudes of $\approx 100Q_0$, where $Q_0 = 4\pi/(\sqrt{3}a_0)$, because $|\mathbf{Q}_{\text{max}}|/Q_0 \approx 2\pi/(\Delta x Q_0) \approx a_0/\Delta x \approx 100$. The necessary discrete Fourier transform of $\rho(\mathbf{r})$ and the inverse transform of $V_{\text{MF}}(\mathbf{q}) = \mathcal{U}(\mathbf{q})\rho(\mathbf{q})$ can be done efficiently using an implementation of the fast Fourier transform.²⁸

We precompute the substrate potential $V_{\text{MF}}(\mathbf{r})$ on a mesh and interpolate subsequently for intermediate pancake positions while performing Langevin dynamics in the fixed substrate. We compute a new substrate every 200 000 time steps. It is important to average over so many time steps to reduce density fluctuations (due to poor statistics) in the pancake histogram, which would result in a deformed substrate potential. Note that $\rho(\mathbf{r})$ and $\rho(\mathbf{q})$ are discretized out of numerical necessity to compute a histogram but not for conceptual reasons.

2. The Fourier-filtered method

The average density $\rho(\mathbf{r})$ should be a periodic function, which can be represented by a discrete set of Fourier components. Therefore, the second method uses only a subset \mathbf{Q}_μ of the Fourier components \mathbf{q} to represent $\rho(\mathbf{q})$,

$$\rho^{\text{FF}}(\mathbf{q}) = (2\pi)^2 \sum_\mu \rho_{\mathbf{Q}_\mu} \delta^2(\mathbf{q} - \mathbf{Q}_\mu), \quad (28)$$

which we determine from the maxima of the structure factor and

$$\rho_{\mathbf{Q}} = \left\langle \frac{1}{L_x L_y} \sum_j \exp(-i\mathbf{R}_j^c \cdot \mathbf{Q}) \right\rangle_c, \quad (29)$$

with $L_x L_y$ being the area of the simulation cell. We average over a set of configurations c of pancake positions \mathbf{R}_j^c (either sweeps or time steps) to compute $\rho_{\mathbf{Q}}$.

Using $\rho^{\text{FF}}(\mathbf{r}) = (2\pi)^{-2} \int d^2q \rho^{\text{FF}}(\mathbf{q}) \exp(i\mathbf{q} \cdot \mathbf{r})$ to present $\rho(\mathbf{r})$, we Fourier-filter (FF) $\rho(\mathbf{r})$, and keep only the relevant components for the computation of the periodic substrate. We can write

$$V_{\text{MF}}^{\text{FF}}(\mathbf{r}) = \int \frac{d^2q}{(2\pi)^2} \mathcal{U}(\mathbf{q}) \rho^{\text{FF}}(\mathbf{q}) \exp(i\mathbf{q} \cdot \mathbf{r}) \quad \text{via Eq. (26)} \quad (30)$$

$$= \sum_{\mu} \mathcal{U}(\mathbf{Q}_{\mu}) \rho_{\mathbf{Q}_{\mu}} \exp(i\mathbf{Q}_{\mu} \cdot \mathbf{r}) \quad \text{via Eq. (28)} \quad (31)$$

$$= -4\pi\epsilon_0 s \sum_{\mu} \frac{\rho_{\mathbf{Q}_{\mu}} \exp(i\mathbf{Q}_{\mu} \cdot \mathbf{r})}{Q_{\mu}^2 (1 + \lambda^2 Q_{\mu}^2)} \quad \text{via Eq. (21)}. \quad (32)$$

This is equivalent to using the full method, but setting $\rho(\mathbf{Q}) = 0$ if $\mathbf{Q} \notin \{\mathbf{Q}_{\mu}\}$.

The advantage of the Fourier-filtered method is that we need to average over less iterations before we can compute a new pancake density and subsequently a new substrate, because the substrate is periodic per construction. Using the Fourier-filtered method we use 500 time steps or sweeps for each substrate iteration.

It turns out that it is not necessary to take the average (29) over different configurations, but it is sufficient to use just one configuration (i.e., one time step or sweep):

$$\rho_{\mathbf{Q}} = \frac{1}{L_x L_y} \sum_j \exp(-i\mathbf{R}_j \cdot \mathbf{Q}). \quad (33)$$

Nevertheless, we run a simulation for 500 time steps or sweeps with the same fixed substrate potential to reduce re-computation of $\rho_{\mathbf{Q}}$, and to give the pancakes some time to experience the system with a new substrate potential.

3. The small-harmonics (Fourier-filtered) approximation

In addition to Fourier-filtering $\rho(\mathbf{r})$ we can speed up the computation further because close to the melting temperature, $\rho_{\mathbf{Q}_{\mu}}$ decays quickly for higher-order \mathbf{Q}_{μ} due to the Debye-Waller factor. We can estimate the reduction of $\rho_{\mathbf{Q}}$ due to the Debye-Waller factor

$$\exp\left(-\frac{\langle u^2 \rangle Q^2}{4}\right) = \exp\left(-\frac{1}{4} \frac{\langle u^2 \rangle}{a_0^2} \frac{16\pi^2 Q^2}{Q_0^2}\right), \quad (34)$$

where $Q_0 = 4\pi/(\sqrt{3}a_0)$ and $\langle u^2 \rangle$ is the mean-squared fluctuation displacement. Depending on $\langle u^2 \rangle$ we can ignore all

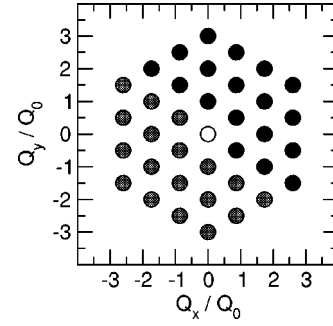


FIG. 3. Set of \mathbf{Q}_{μ} vectors up to third order (i.e., three “shells” around the origin) in reciprocal space used in the small-harmonics Fourier-filtered method to compute $\rho_{\mathbf{Q}_{\mu}}$. Due to the reality of $\rho(\mathbf{r})$ we have $\rho(\mathbf{q}) = \bar{\rho}(-\mathbf{q})$, and it is therefore sufficient to compute only half of the 36 coefficients $\rho_{\mathbf{Q}_{\mu}}$.

$\rho(\mathbf{Q}_{\mu})$ with $|\mathbf{Q}_{\mu}| > Q'$. For all but the smallest fields, we find close to the melting transition $\langle u^2 \rangle / a_0^2 \approx 0.02 - 0.03$ (see Sec. III E), and it is sufficient to include vectors up to third order \mathbf{Q}_{μ} in the summation in Eq. (32) as shown in Fig. 3.

For the small-harmonics Fourier-filtered method it is more efficient to evaluate Eq. (32) for each pancake position occurring in the Langevin–Monte Carlo simulation rather than pre-computing V_{MF} on a mesh. We demonstrate the equivalence of the full and the Fourier-filtered method for the determination of the instability line in Sec. III A, and we compare with the small-harmonics Fourier-filtered method in Sec. III C.

D. Monte Carlo and Langevin dynamics simulations

We have two independent implementations of the small-harmonics Fourier-filtered method: One of us (A.E.K.) has written a Monte Carlo simulation that is based on energy evaluations, and another (H.F.) has implemented a Langevin dynamics simulation based on force calculations. The results of both implementations agree perfectly.

We follow standard vortex-state simulation techniques,^{29,30} including periodic boundary conditions for the in-plane interactions. We use a smooth cutoff for the vortex in-plane interactions.^{29,31} For the Langevin dynamics simulations, we compute the substrate forces numerically from the precomputed mesh (Sec. II C 1) for the full method and the Fourier-filtered method. For the small-harmonics Fourier-filtered method we use the analytical derivative of Eq. (32). The Monte Carlo simulations were only implemented with the small-harmonics method. If not stated otherwise we use a system with 1020 pancakes. The Monte Carlo method is more efficient in converging to a steady solution, whereas the Langevin method allows a generalization of the model to study dynamic properties.

III. RESULTS

A. Time convergence of the substrate potential

As described in Sec. II B, we start each run with a hexagonal pancake distribution corresponding to zero temperature. Figure 4 shows results for the Fourier-filtered method at

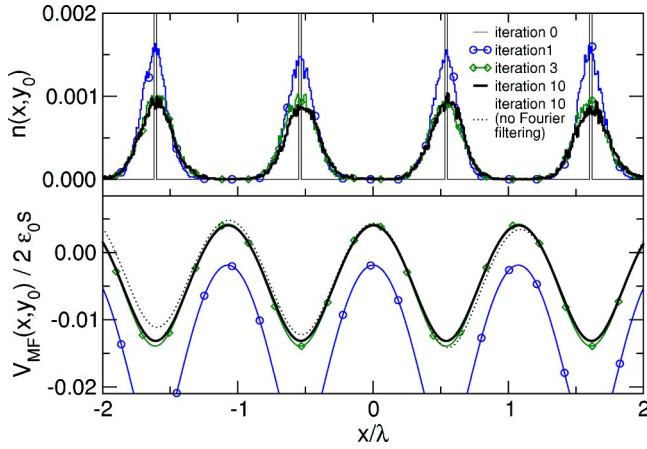


FIG. 4. Convergence to pancake lattice at $B=B_\lambda$ and $t=1/59 \approx 0.017$. Top: pancake histogram $n(x, y_0)$ taken along $y=y_0$. Bottom: substrate potential $V_{MF}(x, y_0)$. For each iteration, 200 000 Langevin time steps were used to compute the new pancake histogram and subsequently the new substrate potential.

$B=B_\lambda$ and at a temperature $t=1/59 \approx 0.017$. The top plot shows a one-dimensional slice of the 2D pancake histogram $n(x, y_0)$ taken along x at $y=y_0$. The histogram relates to the pancake density via $n(x, y) = \rho(x, y) \Delta x \Delta y$, where Δx and Δy are the spacings of the grid used to create the histogram. For the zeroth substrate iteration we set the histogram to have narrow and high peaks at the pancake equilibrium positions corresponding to δ peaks in a zero-temperature pancake density $\rho(\mathbf{r})$. Based on this initial pancake distribution, we compute the substrate potential, $V_{MF}(\mathbf{r})$, for the first substrate iteration, of which a one-dimensional slice at $y=y_0$ is shown in the lower part of Fig. 4. Using this substrate potential, we run the Langevin dynamics simulation for 200 000 steps. We compute the pancake histogram for iteration 1 (see upper plot of Fig. 4) based on the pancake positions in these 200 000 Langevin time steps. From the pancake histogram of iteration 1, we compute the substrate potential for iteration 2. This cycle of calculating a new pancake histogram in the presence of a substrate potential, and subsequently computing a new substrate, is repeated until the

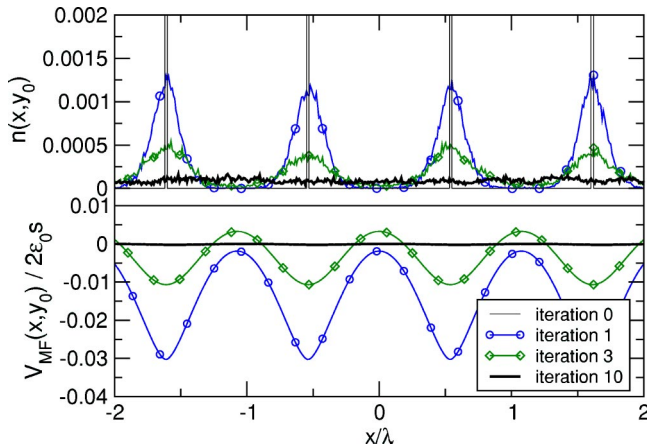


FIG. 5. Convergence to pancake liquid. As in Fig. 4 but at $t=1/50=0.02$ above the melting temperature at $B=B_\lambda$.

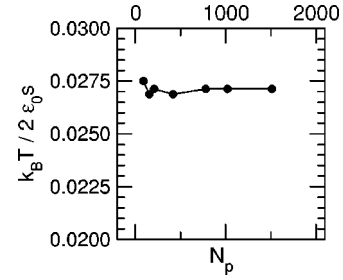


FIG. 6. Finite-size investigation of instability temperature at $B=0.4B_\lambda$.

substrate potential has reached a steady state (after typically less than 10 of these substrate iterations). Figure 4 demonstrates that the system converges quickly to a pancake solid at this temperature below melting.

The dotted line in the lower part of Fig. 4 shows a comparison substrate potential for iteration 10 computed using the full method. While the amplitude and width of the wells (and thus the resulting force) are virtually identical to the Fourier-filtered data, the magnitude of the substrate from the full method varies slightly. This is due to (long-wavelength) density fluctuations in the histogram data and reduces further if one uses more time steps for each substrate iteration.

Figure 5 shows data for $B=B_\lambda$ and a higher temperature $t=1/50=0.02$ that is above the melting temperature. Here, the pancake distribution broadens and consequently the substrate potential flattens quickly within the first few substrate iterations. Eventually, the system has become a disordered liquid with a local pancake density that is approximately constant and the substrate is virtually flat, as shown for iteration 10. We conclude that for this temperature and magnetic field the pancake lattice is unstable against melting into a pancake liquid.

For Figs. 4 and 5 we have used 200 000 Langevin time steps for each substrate iteration in order to be able to compare the full and the Fourier-filtered methods, but it would be sufficient to use much less time steps per substrate iteration for the Fourier-filtered methods. For production purposes, we use the small-harmonics Fourier-filtered method and update the substrate every 500 time steps (Sec. II C 2). Although more substrate iterations than with the full method are required before the system reaches a steady state, the small-harmonics Fourier-filtered approach is more efficient. The full method and both Fourier-filtered methods find that at $B=B_\lambda$ the pancake lattice becomes unstable for $0.017 \leq t \leq 0.018$.

Figure 6 shows how the instability temperature varies as a function of system size. For small numbers of vortices, N_p , the temperature oscillates slightly and for larger systems it becomes constant. Most importantly, there is no general trend visible although the data range from $N_p=90$ to $N_p=1512$. This insensitivity to the system size demonstrates the local nature of the melting transition at this field.

B. Hysteresis loop

Rather than starting from a hexagonal crystal for every temperature, a better approach to determine the instability

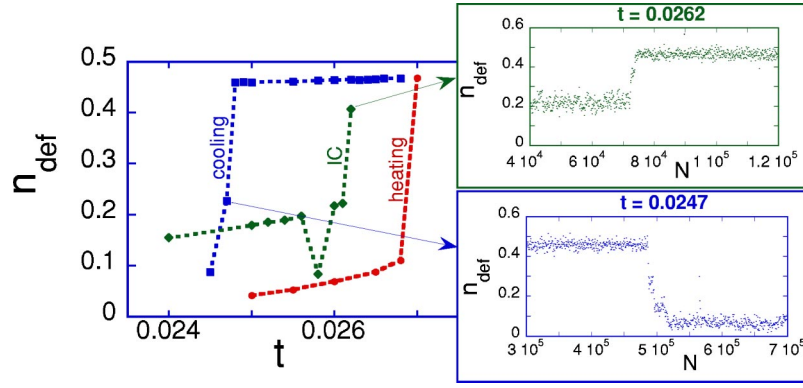


FIG. 7. Example of a hysteresis loop obtained by heating a crystal (circles) and cooling a liquid (squares) for $B = 0.4B_\lambda$. Each point was equilibrated for 9×10^5 Monte Carlo steps. The crystal melts at $t = 0.027$, while the liquid freezes at $t = 0.0247$. Rhombs represent results obtained by simulations starting from the same intermediate defective configuration with $n_{\text{def}} \approx 0.2$. This configuration melts at $t = 0.0262$, which we take as an estimate for the thermodynamic melting temperature. The insets show dependencies of the defect concentration on the Monte Carlo step at the temperatures where the intermediate configuration melts and the liquid configuration freezes. See also Ref. 32.

temperature is to subsequently increase t until the system melts. We also find that by starting from a liquid configuration and lowering t , the system jumps into the crystal state at a certain freezing temperature. We expect the true melting temperature to lie within the instability and the freezing temperature. Such a hysteretic run is shown in Fig. 7.

In order to estimate the thermodynamic melting temperature at which the free energy of the solid and the liquid phase cross, we proceed as follows. First, we store a vortex configuration taken from 2D melting-transition simulations. We chose a configuration from a time-step or sweep where the system was previously a solid but just starts melting, i.e., the defect density starts increasing up and the structure factor peaks start decaying. This vortex configuration is “intermediate” between a solid and a liquid. Second, we start the computation from this intermediate configuration (IC) for every temperature. The results for the IC simulations are shown in Fig. 7 (rhombs). We use the temperature at which the IC melts as a best approximation to the melting temperature of the physical 3D pancake-vortex lattice.

C. Temperature dependence of substrate curvature and pancake fluctuation width

We can quantify the strength of the substrate potential with

$$\alpha_s = \frac{1}{N_p} \sum_j \frac{\partial^2}{\partial x^2} V_{\text{MF}}(\mathbf{R}_j - \mathbf{R}_j^0). \quad (35)$$

This is the curvature of the potential evaluated at deviations $\mathbf{R}_j - \mathbf{R}_j^0$ from the equilibrium lattice positions \mathbf{R}_j^0 and averaged over pancake positions \mathbf{R}_j . The second derivative can be taken analytically from Eq. (32).

Figure 8 shows in the upper plot how α_s varies with temperature. The solid line is an analytical prediction from treating the substrate softening due to thermal fluctuations within the self-consistent harmonic approximation (SCHA).²³ All other data are simulation results from the Fourier-filtered method. The stars show α_s computed using the small-

harmonics Fourier-filtered method with \mathbf{Q}_μ vectors up to 20th order. For low temperatures the data nearly coincide with the SCHA-solution. Close to melting the SCHA α_s is larger than the numerical result. Therefore the simulations give a softer substrate and the lattice has larger thermal displacements. This difference could be due to the inadequacies of the SCHA, which does not include the thermal softening of the 2D lattice. The more complex two-vertex self-consistent harmonic approximation (2VSCHA) does include these effects.²³ The circles in Fig. 8 show results using \mathbf{Q}_μ vectors up to third order, as shown in Fig. 3. Close to the transition from solid to liquid around $t \approx 0.0175$ these data

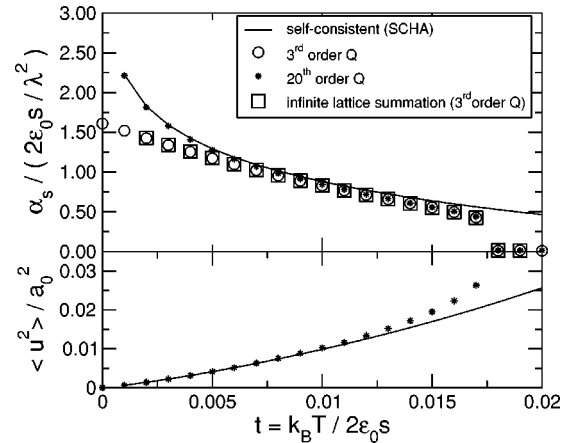


FIG. 8. Top: The substrate curvature α_s at $B = B_\lambda$. Shown is our numerical solution of the Fourier-filtered method using Q vectors up to 20th order (stars), and using up to third order (circles). We have also shown results for $Q_{\text{max}} = 3Q_0$ using an infinite lattice summation for the in-layer interaction (squares). Bottom: The pancake fluctuation width $\langle u^2 \rangle$ (stars). For comparison, we also show the results of a simple SCHA calculation (Ref. 23) (solid line) of the softening of the substrate potential (not including thermal softening effects of the 2D lattice). The results are close to our numerical data at low temperatures, but as the melting point is approached there are extra (anharmonic) fluctuations in the simulation data for $\langle u^2 \rangle$, resulting in an even softer substrate potential α_s .

agree perfectly with the higher-order data. At lower temperatures the third-order results deviate from the 20th order because $\langle u^2 \rangle$ becomes smaller in the Debye-Waller factor (34). However, as long as we are interested in temperatures close to the transition, the third-order approach is sufficient.

The square boxes are computed using the third-order approach, but instead of smoothly reducing the pancake interaction³¹ at a distance of $\approx 7a_0$, we use an infinite lattice summation technique for the logarithmic interaction.³³ This demonstrates that it is sufficient to use a (smooth) cutoff for the in-layer pancake interactions.

We compute the average pancake fluctuation width $\langle u^2 \rangle$ by fitting to a distribution where each pancake is normally smeared around its equilibrium position \mathbf{R}_j^0 :

$$\rho(\mathbf{r}) = \frac{1}{2\pi\sigma^2} \sum_j \exp\left(-\frac{|\mathbf{r} - \mathbf{R}_j^0|^2}{2\sigma^2}\right). \quad (36)$$

The Fourier transform of $\rho(\mathbf{r})$ is

$$\begin{aligned} \rho(\mathbf{q}) &= \exp\left(\frac{-\sigma^2 q^2}{2}\right) \sum_j \exp(-i\mathbf{q} \cdot \mathbf{R}_j^0) \\ &= (2\pi)^2 n_0 \exp\left(\frac{-\sigma^2 q^2}{2}\right) \sum_\mu \delta^2(\mathbf{q} - \mathbf{Q}_\mu). \end{aligned}$$

The Fourier components $\rho(\mathbf{Q}_\mu)$ have the Debye-Waller factor as an envelope, and by fitting a Gaussian to it, we can determine $\langle u^2 \rangle = 2\sigma^2$.

The lower part of Fig. 8 shows computed values for $\langle u^2 \rangle$. We express $\langle u^2 \rangle$ in units of a_0^2 and it increases from 0 at zero temperature towards 0.028 close to the transition, which corresponds to a Lindemann number of ≈ 0.168 at $B = B_\lambda$. In agreement with an overestimation of α_s by the SCHA, $\langle u^2 \rangle$ is underestimated in comparison with numerical results close to the melting transition.

D. Phase diagram

As demonstrated in Sec. III A, we can determine for each parameter pair (B, T) whether the pancake system remains a 3D pancake lattice, or whether it is unstable towards the liquid phase that consists of decoupled 2D liquids. (This is sometimes called a pancake gas, even though there are still very strong in-plane correlations in the decoupled layers. In the absence of Josephson coupling, a linelike liquid regime is expected only at extremely small magnetic fields.³⁴)

We probe parameter space in the B - T plane as described in Sec. III A and compute an instability line for the phase diagram of the system, which is shown in Fig. 9 (circles on dashed line). We also show an estimate of the instability line that has been computed using the two-vertex self-consistent harmonic approximation (2VSCHA, dash-dotted line) for the substrate model.^{23,35} Since in this work we explicitly compute the pancake positions without using approximations (within the substrate model), we expect our result to be more accurate than the 2VSCHA. It can be seen that the 2VSCHA slightly overestimates the temperature for the instability line.

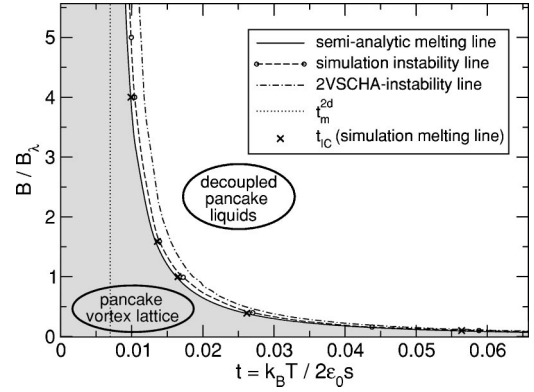


FIG. 9. Phase diagram of the electromagnetically coupled 3D pancake system. Numerically computed instability line (black circles on dashed line) in comparison with the instability line from the 2VSCHA (dash-dotted line). Also shown is a semi-analytical estimate for the melting line from Ref. 23, to be compared with our numerical estimate t_{IC} (crosses). We have shaded the solid phase underneath the melting in gray. The melting temperature $t_m^{2d} = 0.007$ of a 2D system is shown by a dotted line.

Our numerical estimates for the melting points at certain fields (see Sec. III B) are shown as the crosses in Fig. 9. Also shown as the solid line is the melting line calculated semi-analytically in Ref. 23. In this work the melting temperature was estimated by comparing approximate free energies $F = U - TS$ for the solid and liquid phases. The solid free energy was calculated from the SCHA, which gives a variational upper bound on the free energy. The liquid free energy was taken from earlier simulations of a single layer,²⁵ i.e., it was assumed that the layers are completely uncoupled in the liquid state. Remarkably, our melting points from simulations lie on top of the semi-analytic line (to within our error bars).

For increasing fields B , the substrate becomes weaker and weaker and the melting temperature drops. In the limit of $B \rightarrow \infty$ we recover a 2D system with logarithmic interactions for which melting has been estimated²⁴⁻²⁶ to occur at $\Gamma_m^{2d} \approx 140 \pm 10 \Leftrightarrow t_m^{2d} \approx 0.007$, which is consistent with our results.

At low fields the pancake stacks are widely separated and interact only weakly with each other. In this limit the system melts below the evaporation transition of an isolated stack of pancakes^{4,36} at $\Gamma = 4 \Leftrightarrow t = 0.25$. In agreement with this, we find that the instability line approaches $t \approx 0.25$ for $B \rightarrow 0$ (see Fig. 2).

E. Latent heat and jump in entropy

We compute the latent heat per pancake, L_p , by taking the difference of internal energy between the solid and the liquid phase at the melting temperature T_m ,

$$L_p = \frac{1}{N_p} (U_{\text{liquid}} - U_{\text{solid}}) \quad (37)$$

$$= \frac{1}{N_p} [E_{\text{liquid}}^{\text{in}} - (E_{\text{solid}}^{\text{in}} + E^{\text{coup}})]. \quad (38)$$

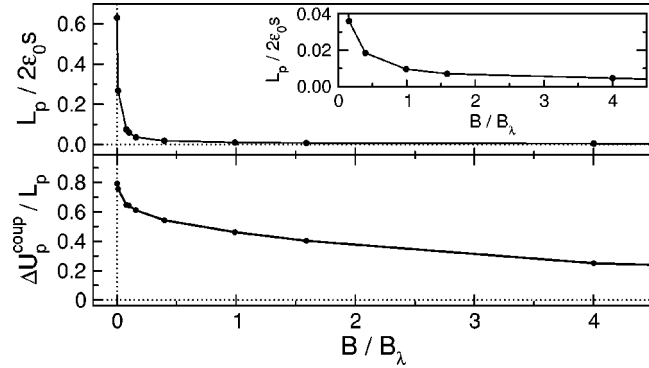


FIG. 10. Top: Latent heat L_p per pancake across the melting transition as a function of field. The inset shows the same data on a reduced scale. Bottom: Jump in interlayer coupling energy, $\Delta U^{\text{coup}} = E^{\text{coup}}$, normalized by latent heat.

The internal energy U of one layer in the solid phase consists of the in-plane energy E^{in} [Eq. (23)] and the interlayer coupling energy E^{coup} , whereas $E^{\text{coup}} = 0$ in the liquid phase in our model. In order to compute E^{coup} for the solid phase, we use Eq. (12), where the second sum vanishes due to the definition of $\delta\rho_n$:

$$E^{\text{coup}} = \frac{1}{2} \int d^2r V_{\text{MF}}(\mathbf{r}) [\rho(\mathbf{r}) - \bar{\rho}], \quad (39)$$

where $\bar{\rho} = \Phi_0/B$ is the mean density. For the Fourier-filtered methods,

$$E^{\text{coup}} = \frac{1}{2} \int d^2r \sum_{\mu \neq 0} \mathcal{U}(\mathbf{Q}_\mu) \rho_{\mathbf{Q}_\mu} \exp(i\mathbf{Q}_\mu \cdot \mathbf{r}) \rho(\mathbf{r})$$

via Eq. (31)

$$= \frac{1}{2} L_x L_y \sum_{\mu \neq 0} \mathcal{U}(\mathbf{Q}_\mu) |\rho_{\mathbf{Q}_\mu}|^2 \quad \text{via Eq. (27)}. \quad (40)$$

For the full method, we have $\rho(\mathbf{r})$ as a histogram, and we can integrate Eq. (39) numerically.

The top plot of Fig. 10 shows how the latent heat varies as a function of field. We have shown the jump in interlayer coupling energy normalized by the latent heat in the bottom part to demonstrate the contribution of the interlayer coupling to the latent heat. This plot shows that the substrate contribution to the latent heat dominates at low fields and becomes less and less important towards high fields.

Figure 11 shows the entropy jump across the transition, $\Delta S_p \equiv L_p/T_m$, as a function of field. We find that ΔS_p monotonically decreases with increasing field as the system approaches the 2D regime. An important issue is the crossover to the 2D melting regime at very large B . Two melting scenarios are possible in two dimensions: a usual first-order melting and continuous dislocation-mediated melting via the intermediate hexatic phase.^{37,38} In the first case ΔS_p has to approach a finite value at $B \rightarrow \infty$ and in the second case it should vanish. Early simulations for a relatively small number (≈ 500) of logarithmically interacting particles suggested

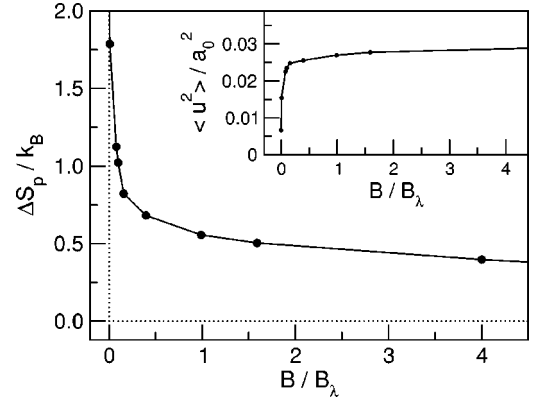


FIG. 11. Entropy jump ΔS_p per pancake as a function of field B . Inset: Pancake fluctuation width $\langle u^2 \rangle$ normalized to a_0^2 at the melting point (the Lindemann number squared).

a first-order phase transition.^{25,24,26} However, it is known that to resolve a continuous melting transition in two dimensions, very large systems are required (see, e.g., Ref. 39). Therefore, the nature of the melting transition of 2D particles with logarithmic interactions is an open issue. Resolving this issue is beyond the scope of this paper.

At low fields the entropy weakly diverges for $B \rightarrow 0$. We understand this as follows: the possible configurations scale as $\sim A/\xi^2$, where ξ^2 is the size of a pancake, and A is the space it can occupy. For the solid state close to the transition the reduced configuration space is $\sim \langle u^2 \rangle / \xi^2$, because the pancake is confined to an area $A \sim \langle u^2 \rangle$. In the liquid the reduced configuration space grows to $\sim a_0^2 / \xi^2$, where a_0 is the average spacing between pancakes. We thus get an entropy difference $\Delta S_p \sim \ln(a_0^2 / \langle u^2 \rangle)$. Since $\langle u^2 \rangle$ approaches at low fields a finite field-independent value of the order of λ^2 , this explains the observed divergence of ΔS_p at $B \rightarrow 0$. For a precise comparison with the experimentally extracted latent heat of vortex-lattice melting⁶ one should be careful to include the temperature dependence of λ , which was shown in Ref. 40 to give extra terms in the observable entropy jump.

IV. CONCLUSIONS

In this work we have applied a numerical simulation to the suggestion of Ref. 23 to treat the 3D layered pancake system with a mean-field approach for the interlayer interactions, turning the problem into a 2D system in the presence of a self-consistent substrate potential. In Ref. 23 this substrate model was studied with semianalytic methods. The two-vertex self-consistent harmonic approximation was used to estimate the instability line of the pancake lattice. Also the melting line was estimated by comparing the elastic free energy of the lattice within the substrate model to the free energy of independent 2D liquids, taken from numerical simulations.

Here we have presented results from full numerical simulations of the substrate model. We have directly calculated both the instability and melting lines. Our result for the instability line has a very similar field dependence to that of

the 2VSCHA, which we find to slightly overestimate the instability temperature. The melting line from the semianalytic approach agrees within error bars to our simulation results for the melting of an intermediate configuration. We have computed the pancake fluctuation width $\langle u^2 \rangle$ and showed how it varies as a function of temperature: the variation is significantly nonlinear below the melting transition, as predicted in Ref. 23. We also calculate the entropy jump across the melting transition, which diverges weakly towards small fields and large melting temperatures.

With such good agreement between the melting line found here and that calculated with the semianalytic method in Ref. 23, it is worth pointing out the value of the simulations we have presented. In the semianalytic method, the estimate for the melting temperature was based on a comparison of the free energies of the liquid and crystal. Both free energies were approximate and the accuracy of each approximation was not strictly checked. The free energy of the crystal was calculated using only the first fluctuation correction in which the magnetic tilt stiffness was evaluated self-consistently. The liquid free energy was obtained by integration of the numerically calculated energy down from the very high temperature at which an analytical result is known (the accuracy therefore relies on a small error in the numerically calculated energy compared to the strength of its temperature dependence). The fact that we find good agreement in the melting line between semianalytic results and simulations suggests that both free-energy approximations in Ref. 23 do not have large inaccuracies, which was not obvious by itself. Finally, we note that even though there is agreement in position of the melting line, simulations suggest that the analytical approach for the crystal always underestimates the strength of fluctuations at the melting point (see Fig. 8).

While we have found a satisfying agreement between our results here and the earlier approximate work of Ref. 23, the true motivation of this project is the possible extensions that can be studied. There is now the exciting prospect to study this pancake vortex system in the presence of pinning disorder. This has been a controversial topic in recent years⁴¹⁻⁴⁴ to which our method should bring some clarity. The effect of random pointlike pinning can be simulated by adding (along with the substrate potential) a random pinning potential to the simulation. If the Fourier-filtered method is used to compute the substrate potential, then the substrate will reflect disorder-induced random vortex displacements, which weaken it while still being periodic. This should adequately represent uncorrelated random pinning in a many-layer system because, due to the long-range nature of magnetic coupling, the disorder-induced layer-to-layer density variations will average out in the magnetic interaction. Note that it is the same averaging that we take advantage of for thermal density fluctuations in the present work.

Our results cannot be directly compared with experiments in available layered superconductors because even in the most anisotropic BSCCO the Josephson coupling energy is not negligible. However, the position of the melting line without the Josephson coupling provides a convenient reference, allowing one to understand the role of the Josephson coupling. In particular, it seems that even a very small Joseph-

son coupling such as in BSCCO gives a large upward shift to the melting field because the extra coupling increases the stability of the 3D pancake lattice. For example, the melting field in the optimally doped BSCCO at $T=50$ K is found to be around 300 G (see, e.g., Ref. 6). If we assume $\lambda(50\text{ K}) \approx 210$ nm, then our calculations suggest that the melting field without Josephson coupling would be at this temperature around 95 G, i.e., about three times smaller. In addition, the liquid phase may have considerable correlations from layer to layer that are not present in the absence of Josephson coupling. We also note that it is possible to suppress the effective Josephson coupling by applying a strong in-plane field, as was done in Ref. 45, bringing the melting line closer to the ‘‘Josephson-free’’ location. By extending the model to use several layers, it is possible to include Josephson coupling between them, which would realistically describe an anisotropic layered high-temperature superconductor.

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APPENDIX: CORRELATION CORRECTION TO FREE ENERGY

The correlation correction to the pancake energy, neglected within the mean-field approach, is given by

$$\delta\mathcal{E} = \frac{1}{2} \sum_{n \neq n'} \int d^2r d^2r' U(\mathbf{r} - \mathbf{r}', n - n') \delta\rho_n(\mathbf{r}) \delta\rho_{n'}(\mathbf{r}'). \quad (\text{A1})$$

The correction to the free energy due to this term is given up to second order by

$$\delta\mathcal{F} \approx \langle \delta\mathcal{E} \rangle_0 - \frac{\langle \delta\mathcal{E}^2 \rangle_0 - \langle \delta\mathcal{E} \rangle_0^2}{2T},$$

where $\langle \dots \rangle_0$ implies mean-field averaging. Substituting Eq. (A1) in the last equation and noting that $\langle \delta\mathcal{E} \rangle_0 = 0$, we derive for the free-energy correction per pancake, $\delta f \equiv a^2 \delta\mathcal{F}/(L_x L_y N)$,

$$\delta f \approx - \frac{a^2}{4TL_x L_y} \sum_{n \neq 0} \int d^2r d^2r' \int d^2r_1 d^2r'_1 U(\mathbf{r} - \mathbf{r}', n) \\ \times U(\mathbf{r}_1 - \mathbf{r}'_1, n) \langle \delta\rho(\mathbf{r}) \delta\rho(\mathbf{r}_1) \rangle_0 \langle \delta\rho(\mathbf{r}') \delta\rho(\mathbf{r}'_1) \rangle_0,$$

where N is the total number of layers, $L_x L_y$ is the layer area, and $\langle \delta\rho(\mathbf{r}) \delta\rho(\mathbf{r}_1) \rangle_0 \equiv \langle \delta\rho_n(\mathbf{r}) \delta\rho_n(\mathbf{r}_1) \rangle_0$ is the density correlation function inside one layer. In the next step we introduce notation for the sum

$$W(\mathbf{r}, \mathbf{r}_1) \equiv \sum_{n \neq 0} U(\mathbf{r}, n) U(\mathbf{r}_1, n),$$

for which, using the mixed representation for the interlayer magnetic interaction,

$$U(k_{\perp}, n) = -\frac{2\pi s^2 \varepsilon_0 \exp(-ns\sqrt{\lambda^{-2} + k_{\perp}^2})}{\lambda^2 k_{\perp}^2 \sqrt{\lambda^{-2} + k_{\perp}^2}},$$

we obtain the formula

$W(\mathbf{r}, \mathbf{r}_1)$

$$\begin{aligned} &= \frac{s^3 \varepsilon_0^2}{2\pi^2 \lambda} \int d^2 k d^2 k_1 \\ &\times \frac{\exp(i\mathbf{k} \cdot \mathbf{r} + i\mathbf{k}_1 \cdot \mathbf{r}_1)}{k^2 k_1^2 \sqrt{1 + \lambda^2 k^2} \sqrt{1 + \lambda^2 k_1^2} (\sqrt{1 + \lambda^2 k^2} + \sqrt{1 + \lambda^2 k_1^2})}, \end{aligned}$$

allowing us to represent $W(\mathbf{r}, \mathbf{r}_1)$ in a scaling form

$$W(\mathbf{r}, \mathbf{r}_1) = \frac{s^3 \varepsilon_0^2}{\lambda} w(\mathbf{r}/\lambda, \mathbf{r}_1/\lambda).$$

By also using a scaling representation for the in-plane density correlation function, $\langle \delta\rho(\mathbf{r}) \delta\rho(\mathbf{r}_1) \rangle_0 = (1/a^4)h(\mathbf{r}/a, \mathbf{r}_1/a)$, we derive the scaling representation for the free-energy correction

$$\delta f = -s\varepsilon_0 \frac{s\varepsilon_0}{4T} \frac{s}{\lambda} \mathcal{G}\left[\frac{a}{\lambda}, \frac{2s\varepsilon_0}{T}\right] \quad (\text{A2})$$

where

$$\begin{aligned} \mathcal{G}\left[\frac{a}{\lambda}, \frac{2s\varepsilon_0}{T}\right] &\equiv \frac{1}{L_x L_y a^6} \int d^2 r d^2 r' d^2 r_1 d^2 r'_1 w\left(\frac{\mathbf{r} - \mathbf{r}'}{\lambda}, \frac{\mathbf{r}_1 - \mathbf{r}'_1}{\lambda}\right) \\ &\times h(\mathbf{r}/a, \mathbf{r}_1/a) h(\mathbf{r}'/a, \mathbf{r}'_1/a) \end{aligned} \quad (\text{A3})$$

is a dimensionless function of the order unity. The mean-field free energy per pancake, f_{MF} , has the scaling property

$$f_{\text{MF}} = s\varepsilon_0 \mathcal{G}\left[\frac{a}{\lambda}, \frac{2s\varepsilon_0}{T}\right].$$

Therefore the free-energy correction due to interlayer correlations (A2) is smaller than the main term by the factor s/λ . In particular, the correlation correction shifts the melting temperature up as

$$\delta T_m = \frac{\delta f_{\text{liq}} - \delta f_{\text{cr}}}{\Delta S}, \quad (\text{A4})$$

where δf_{liq} (δf_{cr}) is the correlation correction to the liquid (crystal) free energy at the melting point and ΔS is the melting entropy jump. In principle, the mean-field simulations allow us to compute the correlation correction and the corresponding shift of the melting temperature. However, this computation includes the numerical evaluation of a quite cumbersome integral in Eq. (A3).

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