Phonons in a nanoparticle mechanically coupled to a substrate

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The discrete nature of the vibrational modes of an isolated nanometer-scale solid dramatically modifies its low-energy electron and phonon dynamics from that of a bulk crystal. However, nanocrystals are usually coupled—even if only weakly so—to an environment consisting of other nanocrystals, a support matrix, or a solid substrate, and this environmental interaction will modify the vibrational properties at low frequencies. In this paper we investigate the modification of the vibrational modes of a spherical insulating nanoparticle caused by a weak mechanical coupling to a semi-infinite substrate. The phonons of the bulk substrate act as a bath of harmonic oscillators, and the coupling to this reservoir shifts and broadens the nanoparticle's modes. The vibrational density of states in the nanoparticle is obtained by solving the Dyson equation for the phonon propagator, and we show that environmental interaction is especially important at low frequencies. As a probe of the modified phonon spectrum, we consider nonradiative energy relaxation of a localized electronic impurity state in the nanoparticle, for which good agreement with a recent experiment is found.

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I. INTRODUCTION

There is currently great interest in properties of nanometer-scale mechanical systems, such as cantilevers, nanoparticles, and resonators.¹ Because of the extremely small size and volume-to-surface ratio of these systems, interactions with their surroundings can dramatically alter their properties. In particular, it is well known that the vibrational spectrum of an isolated nanometer-scale crystal, being discrete, is qualitatively different than that of the same bulk solid, leading to important changes in any property dependent on the phonon density of states (DOS). The differences between the vibrational DOS in a nanoparticle and a bulk solid are most evident at low frequencies: A spherical nanoparticle with diameter *d* and characteristic bulk sound velocity *v* cannot support a mode with frequency less than about $\pi v/d$. Thus an acoustic energy "gap" in the low-energy phonon spectrum is present in contrast with that of the bulk, which has a continuous spectrum down to zero energy. However, mechanical interaction with the environment will modify the vibrational modes.

In an interesting experiment by Yang and co-workers, $²$ the</sup> phonon DOS deep inside this gap was measured in insulating Y_2O_3 nanoparticles. The experiment used nanoparticles whose sizes ranged from 7 to 23 nm in diameter and was performed by measuring the nonradiative lifetimes of an excited electronic state of a Eu^{3+} dopant. The lowest vibrational mode, referred to as the Lamb mode, for a nanoparticle with the mean size³ of 13 nm has a frequency of approximately 9 cm⁻¹. At 3 cm⁻¹ the DOS measured was more than 100 times smaller than that of bulk Y_2O_3 at 3 cm^{-1} .

In this paper we propose and investigate a mechanism that could be responsible for the observed broadening of a nanoparticle's phonon modes. Several broadening mechanisms could be responsible for the observed effect. For example, anharmonicity leads to broadening and therefore to a lowenergy DOS, but anharmonicity is ineffective at low energy and was found to be too small to account for the

experiment.⁴ Another possibility could be adsorbed "dirt" on the outside of the nanoparticle. This might lower the quality factor *Q* of the nanoparticle, regarding it as a resonator, reflecting a broadening of the vibrational modes. A third, and in our opinion more likely mechanism, follows from the realization that these nanoparticles are not isolated, but instead were prepared in a powered form. Thus each nanoparticle is in weak contact with a cluster of other nanoparticles. Because the cluster is relatively large, including at least several hundred nanoparticles, each nanoparticle is mechanically coupled to a reservoir that has a continuous vibrational DOS at low energy. This interaction broadens the modes and allows phonons in the nanoparticle to escape and be absorbed into the cluster. We will investigate the effect this mechanical environmental interaction has on the nanoparticle's phonon spectrum.

Because we are only interested in determining the correct origin of the observed broadening,⁵ and do not hope to be able to exactly reproduce the experimental results of Ref. 2, we propose the following simplified model: The cluster of nanoparticles is replaced by a semi-infinite elastic substrate, and one nanoparticle is placed in weak mechanical contact with it. The weak contact is imagined to be a few atomic bonds or small neck of elastic material, which we model by a harmonic spring. For simplicity, we take the substrate and the nanoparticle to be made out of the same isotropic elastic material. Because we are interested in the low-energy regime, continuum elasticity theory will be used to describe the dynamics of the nanoparticle and substrate. After defining and analyzing our simplified model, in Secs. II–V, we explain in Sec. VI how the model can be adapted to address the experiment of Ref. 2, and good agreement is obtained.

The simple model we study is related to, but different than, models used to study energy relaxation by molecules adsorbed on surfaces.⁶ However, in surface science the interest is usually in the relaxation of rigid translational motion, rotational motion, or simple internal vibrations of adsorbates. In contrast, we investigate the broadening of complex internal vibrational modes of much larger objects (which are

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crystalline). Our work also has much in common with that of Gurevich and Schober, 7 where many of the same considerations and modeling were used to study the Lamb-mode decay rate of nanoparticles caused by both anharmonicity and coupling to an enviroment of other nanoparticles.

II. NANOPARTICLE AND SUBSTRATE MODEL

As discussed in the introduction, the model we study is that of a single nanoparticle in weak mechanical contact with a semi-infinite substrate. Linear elasticity theory will be used to describe the phonons of this system. We assume the nanoparticle and substrate to be made of an isotropic nonpolar dielectric. Because we take the nanoparticle and substrate to be made of the same material, we will use the same density ρ and Lame´ coefficients λ and μ for both. The Lagrangian for the entire system is

$$
L = \int_{V} d^{3}r \left[\frac{1}{2}\rho(\partial_{t}\mathbf{u})^{2} - \frac{1}{2}\lambda u_{ii}^{2} - \mu u_{ij}^{2}\right],
$$
 (1)

where $\mathbf{u}(\mathbf{r},t)$ is the displacement field, and

$$
u_{ij} \equiv (\partial_i u_j + \partial_j u_i)/2 \tag{2}
$$

is the strain tensor. V is the combined volume of the nanoparticle, substrate, and connecting material, as shown in Fig. 1.

Because the Lagrangian density is local, the integration volume in Eq. (1) can be split into three independent parts: the nanoparticle, the substrate, and the connecting region. In the limit of weak coupling (diameter d_c of connecting region much smaller than *d*), the surface area on the nanoparticle and substrate over which the actual boundary conditions differ from stress-free conditions are negligible, and the Hamiltonian can be written as

$$
H = H_{\text{nano}} + H_{\text{sub}} + \delta H,\tag{3}
$$

where H_{nano} is the Hamiltonian for an isolated nanoparticle (with stress-free boundaries), H_{sub} is that for an isolated substrate, and δH is the interaction between the two. The connecting region is taken to be a few atomic bonds or small neck of material, as shown in Fig. 1, and is discussed further below.

Our analysis will require the vibrational normal modes and spectra of the isolated nanoparticle and semi-infinite substrate, calculated with stress-free boundary conditions. The long-wavelength modes of interest here may be obtained from elasticity theory, to which we now turn.

A. Isolated nanoparticle

Here we derive the normal modes of an isolated elastic sphere. The method we shall use is different than (but equivalent to) that used in the classic paper by $Lamb⁸$ but is better suited for our purposes. The equation of motion given by Eq. (1) is

 $\partial_t^2 \mathbf{u} - v_1^2 \nabla (\nabla \cdot \mathbf{u}) + v_1^2 \nabla \times \nabla \times \mathbf{u} = 0,$ (4)

FIG. 1. (a) Model of nanoparticle, substrate, and connecting region. d is the diameter of the nanoparticle. (b) Expanded view of cylindrical connecting region with dimensions d_c and l_c . In our numerical study we assume $d=10$ nm, $l_c=2.5$ nm, and d_c $=0.5$ nm.

where $v_1 = \sqrt{\left(\frac{\lambda + 2\mu}{\rho}\right)}$ is the bulk longitudinal sound velocity and $v_t = \sqrt{\mu/\rho}$ is the transverse velocity. To solve Eq. (4) the displacement field can be decomposed into longitudinal and transverse parts,

$$
\mathbf{u} = \mathbf{u}_1 + \mathbf{u}_t, \tag{5}
$$

where

and

$$
\nabla \times \mathbf{u}_l = 0 \tag{6}
$$

$$
\nabla \cdot \mathbf{u}_t = 0. \tag{7}
$$

With harmonic time dependence, the equation of motion (4) then separates into two vector Helmholtz equations for the longitudinal and transverse parts,

$$
(\nabla^2 + p^2)\mathbf{u}_1 = 0, \quad p \equiv \omega/v_1 \tag{8}
$$

and

$$
(\nabla^2 + q^2)\mathbf{u}_t = 0, \quad q \equiv \omega/v_t. \tag{9}
$$

The longitudinal equation (8) can be solved by introducing a scalar potential

$$
\mathbf{u}_{\mathbf{l}} = \nabla \phi^{(p)},\tag{10}
$$

where $\phi^{(p)}$ is a solution of the scalar Helmholtz equation $(\nabla^2 + p^2) \phi^{(p)} = 0$. The transverse equation (9) has two linearly independent solutions, $\mathbf{u}_t = \mathbf{M}$ and \mathbf{N} , where

$$
\mathbf{M} = \nabla \phi^{(q)} \times \mathbf{r}
$$
 (11)

and

$$
\mathbf{N} = \frac{1}{q} \nabla \times \mathbf{M}.
$$
 (12)

Here $\phi^{(q)}$ is a solution of $(\nabla^2 + q^2)\phi^{(q)} = 0$. The prefactor 1/*q* is included for dimensional convenience. The scalar Helmholtz equations are separable in spherical coordinates and the solutions can be written as

$$
\phi_{lm}(\mathbf{r}) \equiv j_l(kr) Y_{lm}(\theta, \varphi), \quad k = p, q \tag{13}
$$

where

$$
j_l(x) \equiv \sqrt{\frac{\pi}{2x}} J_{l+\frac{1}{2}}(x)
$$
 (14)

is a spherical Bessel function of the first kind (regular at origin) and

$$
Y_{lm}(\theta,\varphi) \equiv (-1)^m \sqrt{\frac{2l+1}{4\pi} \frac{(l-m)!}{(l+m)!}} P_{lm}(\cos\theta) e^{im\varphi}.
$$
\n(15)

Here

$$
P_{lm}(x) = (1 - x^2)^{\frac{m}{2}} \frac{\partial^m}{\partial x^m} P_l(x),
$$
 (16)

where the $P_l(x)$ are Legendre polynomials.

Now we use the ϕ_{lm} to construct three linearly independent solutions of Eq. (4) ,

$$
\mathbf{L}_{lm} = \frac{1}{p} \nabla \phi_{lm}(pr), \qquad (17)
$$

$$
\mathbf{M}_{lm} \equiv \nabla \phi_{lm}(qr) \times \mathbf{r},\tag{18}
$$

$$
\mathbf{N}_{lm} \equiv \frac{1}{q} \mathbf{\nabla} \times \mathbf{M}_{lm} \,. \tag{19}
$$

The general solution is a linear combination of \mathbf{L}_{lm} , \mathbf{M}_{lm} , and N_{lm} ,

$$
\mathbf{u}(\mathbf{r}) = \sum_{lm} \left[a_{lm} \mathbf{L}_{lm} + b_{lm} \mathbf{M}_{lm} + c_{lm} \mathbf{N}_{lm} \right]. \tag{20}
$$

Although they are linearly independent, the vector fields \mathbf{L}_{lm} , \mathbf{M}_{lm} , and \mathbf{N}_{lm} are not orthogonal in space. However, they can be rewritten in terms of orthogonal vector spherical harmonics P_{lm} , B_{lm} , and C_{lm} , defined as

$$
\mathbf{B}_{lm}(\Omega) \equiv \frac{1}{\sqrt{l(l+1)}} \left(\partial_{\theta} Y_{lm}(\Omega) \mathbf{e}_{\theta} + \frac{im Y_{lm}(\Omega)}{\sin \theta} \mathbf{e}_{\varphi} \right), \tag{22}
$$

$$
\mathbf{C}_{lm}(\Omega) \equiv \frac{1}{\sqrt{l(l+1)}} \left(\frac{im Y_{lm}(\Omega)}{\sin \theta} \mathbf{e}_{\theta} - \partial_{\theta} Y_{lm}(\Omega) \mathbf{e}_{\varphi} \right), \tag{23}
$$

with the following properties:

$$
\int d\Omega \mathbf{X}_{lm}^* \cdot \mathbf{X}_{l'm'} = \delta_{ll'} \delta_{mm'} \tag{24}
$$

for $X \in B$, C , P and

$$
\int d\Omega \mathbf{X}_{lm}^* \cdot \mathbf{X}_{l'm'}' = 0 \tag{25}
$$

for $X \neq X'$. Expressed in terms of orthogonal vector spherical harmonics, \mathbf{L}_{lm} , \mathbf{M}_{lm} , and \mathbf{N}_{lm} are given by

$$
\mathbf{L}_{lm} = j'_l(pr)\mathbf{P}_{lm}(\Omega) + \frac{\sqrt{l(l+1)}}{pr}j_l(pr)\mathbf{B}_{lm}(\Omega), \quad (26)
$$

$$
\mathbf{M}_{lm} = \sqrt{l(l+1)} j_l(qr) \mathbf{C}_{lm}(\Omega),\tag{27}
$$

and

$$
\mathbf{N}_{lm} = \frac{l(l+1)}{qr} j_l(qr) \mathbf{P}_{lm}(\Omega) + \frac{\sqrt{l(l+1)}}{qr}
$$

$$
\times [j_l(qr) + qr j_l'(qr)] \mathbf{B}_{lm}(\Omega), \qquad (28)
$$

where prime denotes differentiation with respect to the argument.

Next we impose stress-free boundary conditions

$$
\sigma_{ij} n_j = 0 \tag{29}
$$

at the surface $r = R$ of the nanoparticle. Here **n** is an outward pointing normal vector and σ_{ij} is the strain tensor. In an isotropic elastic continuum,

$$
\sigma_{ij} = \lambda (\nabla \cdot \mathbf{u}) \, \delta_{ij} + 2 \, \mu u_{ij} \,. \tag{30}
$$

In spherical coordinates (29) implies

$$
\sigma_{rr} = \sigma_{\theta r} = \sigma_{\varphi r} = 0. \tag{31}
$$

The three conditions (31) require that

$$
\lambda(\nabla \cdot \mathbf{u}) + 2\mu u_{rr} = 0,\tag{32}
$$

$$
u_{\theta r} = 0,\t\t(33)
$$

and

$$
u_{\varphi r} = 0.\t\t(34)
$$

In terms of the displacement field,

$$
u_{rr} = \partial_r u_r, \qquad (35)
$$

$$
u_{\theta r} = \frac{1}{2} \left(\partial_r u_{\theta} - \frac{1}{r} u_{\theta} + \frac{1}{r} \partial_{\theta} u_r \right),
$$
 (36)

$$
u_{\varphi r} = \frac{1}{2} \left(\frac{1}{r \sin \theta} \partial_{\varphi} u_r + \partial_r u_{\varphi} - \frac{1}{r} u_{\varphi} \right). \tag{37}
$$

The boundary-condition equations (31) then become

$$
a_{lm}[-\lambda p j_l(pR)Y_{lm} + 2\mu p j_l''(pR)Y_{lm}]
$$

+ $c_{lm}2\mu l(l+1)EY_{lm} = 0,$ (38)

$$
a_{lm} 2\mathsf{D}\partial_{\theta} Y_{lm} + b_{lm} im\mathsf{E} Y_{lm} \csc \theta + c_{lm} \mathsf{F} \partial_{\theta} Y_{lm} = 0, \quad (39)
$$

and

$$
a_{lm} 2im \mathsf{D} Y_{lm} \csc \theta + b_{lm} \mathsf{E} \partial_{\theta} Y_{lm} + c_{lm} im \mathsf{F} Y_{lm} \csc \theta = 0,
$$
\n(40)

where

$$
D \equiv \frac{j'_l(pR)}{R} - \frac{j_l(pR)}{pR^2},\tag{41}
$$

$$
E = \frac{j'_{l}(qR)}{R} - \frac{j_{l}(qR)}{qR^{2}},
$$
\n(42)

$$
\mathsf{F} \equiv qj''_l(qR) + \frac{l(l+1)}{qR^2}j_l(qR) - \frac{2j_l(qR)}{qR^2}.
$$
 (43)

Finally, we rewrite Eqs. $(38)–(40)$ in matrix form as

$$
\begin{pmatrix}\n-\lambda p j_l (pR) Y_{lm} + 2 \mu p j_l'' (pR) Y_{lm} & 0 & 2 \mu l(l+1) E Y_{lm} \\
2 D \partial_{\theta} Y_{lm} & im E Y_{lm} \csc \theta & F \partial_{\theta} Y_{lm} \\
2 i m D Y_{lm} \csc \theta & E \partial_{\theta} Y_{lm} & im F Y_{lm} \csc \theta\n\end{pmatrix}\n\begin{pmatrix}\na_{lm} \\
b_{lm} \\
c_{lm}\n\end{pmatrix} = 0.
$$
\n(44)

Г

For a nontrivial solution of Eqs. $(38)–(40)$ to exist, the determinant of the above matrix must vanish. Taking the determinant and simplifying we obtain

$$
[-\lambda p j_l(pR) + 2\mu p j_l''(pR)] \text{EF} - 4\mu l(l+1) \text{E}^2 \text{D} = 0. \tag{45}
$$

This implies that either

$$
E = \frac{j'_l(qR)}{R} - \frac{j_l(qR)}{qR^2} = 0
$$
 (46)

or

$$
[-\lambda p j_l(pR) + 2\mu p j_l''(pR)]F - 4\mu l(l+1)ED = 0.
$$
\n(47)

If condition (46) is met, then this imposes certain constraints on a_{lm} , b_{lm} , and c_{lm} , which require $a_{lm} = c_{lm} = 0$. This can easily be seen in the matrix of Eq. (44) by setting $E=0$. If Eq. (47) is met, b_{lm} has to be zero.

In conclusion, we have two branches of vibrational modes: The branch in which Eq. (46) is satisfied,

$$
\mathbf{u}(\mathbf{r}) = b_{lmn} \mathbf{M}_{lmn}(\mathbf{r}),\tag{48}
$$

is referred to as the *torsional* branch, where *n* specifies the radial quantum number $[nth$ solution of Eq. (46) . These modes have no radial component [see Eq. (27)] and are *not* broadened by the mechanism considered in this paper. The other branch is found when Eq. (47) is satisfied,

$$
\mathbf{u}(\mathbf{r}) = a_{lmn} \mathbf{L}_{lmn}(\mathbf{r}) + c_{lmn} \mathbf{N}_{lmn}(\mathbf{r}),
$$
(49)

which is called the *spheroidal* branch.

We define the Lamb mode to be the vibrational eigenfunction associated with the lowest frequency solution of Eq. (45) . For the cases considered in this paper, the Lamb mode is a fivefold degenerate $l=2$ torsional mode with angular frequency

$$
\omega_{\rm L} = \xi_0 \frac{v_{\rm t}}{R},\tag{50}
$$

where, according to Eq. (46), $\xi_0 \approx 2.50$ is the smallest positive solution of

$$
j_2'(\xi) - \frac{1}{\xi} j_2(\xi) = 0.
$$
 (51)

Thus the Lamb frequency is simply

$$
\omega_{\text{L}} \approx 1.59 \times \frac{\pi v_{\text{t}}}{d}.\tag{52}
$$

We emphasize, however, that the Lamb mode has no radial displacement component and is not broadened in our model.

To quantize the vibrational modes we write the displacement field as⁹

$$
\mathbf{u}_{\text{nano}}(\mathbf{r}) = \sum_{J} \sqrt{\frac{\hbar}{2\rho \omega_{J}}} [a_{J} \Psi_{J}(\mathbf{r}) + a_{J}^{\dagger} \Psi_{J}^{*}(\mathbf{r})], \quad (53)
$$

where

$$
J = [S \text{ or } T, n, l, m] \tag{54}
$$

is a label uniquely specifying a nanoparticle eigenmode. The first entry S or T specifies whether the mode is in the spheroidal or torsional branch, respectively. *n* is the radial quantum number and *l* and *m* are the usual angular-momentum quantum numbers. *a* and a^{\dagger} are phonon annihilation and creation operators which satisfy the Bose commutation relation

$$
[a_J, a_{J'}^{\dagger}] = \delta_{JJ'} . \tag{55}
$$

The Ψ_J are vibrational eigenvectors normalized such that

$$
\int_{V} |\Psi_{J}(\mathbf{r})|^{2} d^{3} r = 1,
$$
\n(56)

where V is the volume of the nanoparticle. Assuming (without proof) that the modes Ψ _{*I*} form a complete set,

$$
\sum_{J} \Psi_{J}^{i*}(\mathbf{r}) \Psi_{J}^{j}(\mathbf{r}') = \delta^{ij} \delta(\mathbf{r} - \mathbf{r}'), \tag{57}
$$

it can easily be shown that **u** satisfies the correct equal-time canonical commutation relation with $\pi \equiv \rho \partial_t \mathbf{u}$, namely

$$
[u^{i}(\mathbf{r}), \pi^{j}(\mathbf{r}')] = i\hbar \,\delta^{ij}\delta(\mathbf{r} - \mathbf{r}'). \tag{58}
$$

B. Isolated substrate

The vibrational modes for a semi-infinite isotropic elastic substrate, with a free surface at the *xy* plane and extending to infinity in the negative *z* direction, were quantized previously by Ezawa.¹⁰ Therefore the details will be left out here. The displacement field can be written $as⁹$

$$
\mathbf{u}_{\text{sub}}(\mathbf{r}) = \sum_{I} \sqrt{\frac{\hbar}{2\rho \omega_{I}}} [b_{I} \mathbf{f}_{I}(\mathbf{r}) + b_{I}^{\dagger} \mathbf{f}_{I}^{*}(\mathbf{r})], \qquad (59)
$$

where *b* and b^{\dagger} are the annihilation and creation operators for the substrate phonons. The index *I*, like the index *J* for the nanoparticle, uniquely specifies a phonon mode for the substrate. f_I are eigenfunctions of Eq. (4) subject to stressfree boundary conditions at the $z=0$ plane.

In what follows we will need the spectral density of the isolated substrate, which is defined as

$$
N_{\text{sub}}(\mathbf{r}, \omega) \equiv -\frac{1}{\pi} \text{Im} \, D_{\text{sub}}^{zz}(\mathbf{r}, \mathbf{r}, \omega), \tag{60}
$$

where $D_{sub}^{ij}(\mathbf{r}, \mathbf{r}', \omega)$ is the Fourier transform of the retarded phonon Green's function

$$
D_{\text{sub}}^{ij}(\mathbf{r}, \mathbf{r}', t) \equiv -i \theta(t) \langle [u_{\text{sub}}^{i}(\mathbf{r}, t), u_{\text{sub}}^{j}(\mathbf{r}', 0)] \rangle \qquad (61)
$$

of the substrate. The spectral density at the free surface of silicon, regarding it as an isotropic elastic continuum, was calculated in Appendix B of Ref. 11. There we obtained

$$
N_{sub}(\omega) = C_{Si}\omega
$$
, $C_{Si} \approx 1.4 \times 10^{-46} \text{ cm}^2 \text{ s}^2$. (62)

It turns out that Eq. (62) is quite close to that resulting from the simpler Debye model for three-dimensional bulk Si.

C. Nanoparticle-substrate interaction

As discussed above, the connection between the nanoparticle and substrate is assumed to be weak, allowing perturbation theory to be applied. The connection consists of a small cylindrical neck of elastic material, with diameter d_c and length l_c . In addition, we assume that $d_c \ll l_c$, in which case the dominant interaction between the nanoparticle and the substrate is mediated by the longitudinal compression/ extension of this cylinder. Thus the interaction between the nanoparticle and substrate is that of a harmonic spring

$$
\delta H = \frac{1}{2} K \left[u_{\text{nano}}^z(\mathbf{r}_0) - u_{\text{sub}}^z(\mathbf{r}_0) \right]^2; \tag{63}
$$

where *K* is an effective spring constant, and u_{mano}^z and u_{sub}^z are the *z* components of the displacement field of the nanoparticle and substrate at the point of contact \mathbf{r}_0 . We take the *z* direction to be along the upward pointing normal to the substrate surface. The Hamiltonians we will introduce below [see Eq. (75)] for H_{nano} and H_{sub} are normal-ordered; it is therefore necessary to normal order δH as well. This operation is denoted by the colons in Eq. (63) . The force constant *K* given by

$$
K = \frac{\pi d_c^2}{4l_c} Y,\tag{64}
$$

where $Y \approx 1.3 \times 10^{12}$ dyn cm⁻² is the Young modulus for Si. Assuming neck dimensions of l_c =2.5 nm and d_c =0.5 nm, we obtain

$$
K = 1.0 \times 10^4 \text{ erg cm}^{-2}.
$$
 (65)

Of course, Eq. (65) is just an estimate of the actual interaction strength and should not be taken seriously beyond the order-of-magnitude level.

III. GOLDEN-RULE LIFETIMES

The relaxation rate or inverse lifetime of the perturbed eigenmodes of the nanoparticle can be calculated using Fermi's golden rule (setting $\hbar=1$),

$$
\tau_J^{-1} = 2 \pi \sum_{\mathbf{f}} |\langle \mathbf{f} | \delta H | \mathbf{i} \rangle|^2 \delta(\omega_{\mathbf{i}} - \omega_{\mathbf{f}}), \tag{66}
$$

where the initial and final states are

$$
|i\rangle = a_J^{\dagger}|0\rangle \text{ and } |f\rangle = b_I^{\dagger}|0\rangle. \tag{67}
$$

Using Eqs. (53) , (59) , and (63) , leads to⁹

$$
\tau_J^{-1} = \frac{\pi K^2}{2\rho^2} \frac{|\Psi_J^z(\mathbf{r}_0)|^2}{\omega_J} \sum_I \frac{|f_I^z(\mathbf{r}_0)|^2}{\omega_J} \delta(\omega_J - \omega_I). \quad (68)
$$

Noting that

$$
\sum_{I} |f_{I}^{z}(\mathbf{r}_{0})|^{2} \delta(\omega_{J} - \omega_{I}) = 2 \rho \omega N_{\text{sub}}(\omega), \qquad (69)
$$

we obtain (reinstating factors of \hbar)

$$
\tau_J^{-1} = \frac{\pi K^2}{\hbar \rho} \frac{N_{\text{sub}}(\omega_J)}{\omega_J} |\Psi_J^z(\mathbf{r}_0)|^2.
$$
 (70)

TABLE I. A few representative relaxation rates and quality factors. Only the $m=0$ spheroidal modes are broadened by the mechanism considered in this paper.

(S, l, m, n)	ω (rad s ⁻¹)	$\tau^{-1}(s^{-1})$	O factor
(S, 2, 0, 1)	3.1×10^{12}	2.5×10^{-4}	2.2×10^{15}
(S,1,0,1)	3.4×10^{12}	1.0×10^{-8}	7.8×10^{19}
(S,0,0,1)	1.0×10^{13}	3.6×10^{-4}	4.9×10^{15}
(S,0,0,2)	1.6×10^{13}	1.1×10^{-3}	2.5×10^{15}

Using $K=1.0\times10^4$ erg cm⁻² and $\rho=2.3$ g cm⁻³, which are appropriate (see above) for our model, the resulting relaxation rates and quality factors are given in Table I for some low-lying modes. The quality factor *Q* is defined here as the lifetime τ divided by the period T ,

$$
Q = \frac{\tau}{T} = \frac{\hbar \,\omega}{2\,\pi\,\gamma},\tag{71}
$$

where $\gamma = \hbar \tau^{-1}$ is an energy width.

The values of the *Q* factors we obtain for the low lying modes are incredibly large, reflecting the fact that the reservoir (substrate) is extremely ineffective at absorbing energy at these low frequencies. As we will discuss below in Sec. VI, the lifetimes (and Q factors) for the model considered here cannot be directly compared with the experiment of Ref. 2 without accounting for the difference in sound speeds between a solid Si substrate and a weakly bound nanoparticle cluster (as well as some other less important modifications). There we shall show that the coefficient C in Eq. (62) should be enhanced by a factor of about $10³$ before making such a comparison, which decreases the *Q* factors by this same factor. However, the *Q* factors corrected in this way are still huge, and the good agreement with the observed lowfrequency DOS (see below) suggests that the Q factors of the nanoparticles studied experimentally in Ref. 2 are also very large.

In Ref. 12 we used the golden-rule result (70) to estimate the phonon DOS at low energies. This is achieved by replacing, in accordance with Fermi's golden rule, each discrete mode in the isolated nanoparticle by a Lorentzian with a width given by Eq. (70) . (More precisely, this amounts to approximating the energy-dependent phonon self-energy for each mode *J* with its value at $\omega = \omega_J$, a procedure often called the quasiparticle-pole approximation.) However, this procedure is unreliable at low energies because the actual line shapes of the broadened modes are non-Lorentzian in the tails. Nevertheless, we obtained a DOS at 3 cm^{-1} that was only 20 times smaller than that observed.¹³

IV. MANY-BODY THEORY OF THE DOS

A. Local DOS

To leading order in the electron-phonon interaction strength, the electronic population relaxation rate due to phonon emission (for example, as measured in Ref. 2) is given by Fermi's golden rule,⁵ which states that the rate (for a deformation potential electron-phonon interaction) is proportional to the square of the electron-phonon coupling strength times the phonon DOS. In a translationally invariant system the DOS does not have any position dependence, but in a nanoparticle one must distinguish between the *global* DOS (the DOS relevant for thermodynamics) and the *local* eigenfunction-weighted DOS, which is the one that determines phonon emission rate. We will call this positiondependent DOS the local DOS and denote it by $g(\mathbf{r},\omega)$. The precise definition of $g(\mathbf{r},\omega)$ will be given below in Eq. (77).

From a theoretical point of view, the quantity describing the local vibrational dynamics in the nanoparticle is the (retarded) phonon Green's function

$$
D^{ij}_{\rm R}(\mathbf{r}, \mathbf{r}', t) \equiv -i \theta(t) \langle [u^{i}(\mathbf{r}, t), u^{j}(\mathbf{r}', 0)] \rangle_{H}, \qquad (72)
$$

where

$$
\langle \cdot \rangle_H \equiv \frac{\text{Tr}(e^{-\beta H} \cdot)}{\text{Tr}e^{-\beta H}},\tag{73}
$$

with the Hamiltonian given by

$$
H = H_0 + \delta H. \tag{74}
$$

Here H_0 is the Hamiltonian of the isolated nanoparticle and substrate,

$$
H_0 = \sum_J \omega_J a_J^{\dagger} a_J + \sum_I \omega_I b_I^{\dagger} b_I, \qquad (75)
$$

and, as mentioned in Sec. II, δH is a harmonic spring potential given in Eq. (63) .

In this section the phonon Green's function *D* always refers to the nanoparticle, and the label ''nano'' will be suppressed. The imaginary part of the Fourier transform of $D_{\text{R}}^{ij}(\mathbf{r}, \mathbf{r}, t)$ defines the nanoparticle's phonon spectral density

$$
N^{ij}(\mathbf{r}, \omega) \equiv -\frac{1}{\pi} \text{Im} \, D_{\text{R}}^{ij}(\mathbf{r}, \mathbf{r}, \omega). \tag{76}
$$

For an electron system (or any system of particles), the spectral density in Eq. (76) is precisely the local DOS. However, because the elasticity theory equation of motion (4) is second order in time, the spectral density and DOS (both local and global) differ by a factor of $2\rho\omega$. In addition, the vibrational spectral density (76) is a tensor, whereas the phonon emission rate probes some coupling-constant-weighted sum of tensor elements. Because we are ascribing the observed reduction in phonon emission (in going from bulk to nanoparticle) to a reduction in the local DOS, our results are not sensitive to the precise way in which a scalar quantity is constructed from the tensor, as long as the same measure is used in both the nanoparticle and bulk. It will be most convenient to investigate the trace of the local DOS tensor. Therefore the quantity we calculate in this paper is

$$
g(\mathbf{r}, \omega) \equiv 2 \rho \omega \sum_{i=1}^{3} N^{ii}(\mathbf{r}, \omega), \qquad (77)
$$

which we shall refer to as the local DOS. $g(\mathbf{r},\omega)$ characterizes the number of states per unit energy per unit volume near position **r**. In a bulk material with Debye spectrum, Eq. (77) reduces at low frequency to

$$
g(\mathbf{r}, \omega) = \frac{\omega^2}{2\pi^2} \left(\frac{1}{v_1^3} + \frac{2}{v_t^3} \right),\tag{78}
$$

independent of r . Equation (78) is the well-known Debye formula for the vibrational DOS of a three-dimensional bulk crystal.

The local DOS $g(\mathbf{r},\omega)$ controls the phonon emission rate for an impurity atom sitting at position **r**. Although the impurity locations in a real nanoparticle are assumed to be random, dopants near the surface are known to be optically inactive. Hence most optical experiments (including that of Ref. 2) do not probe impurities near the nanoparticle's surface. Therefore we introduce a particular volume-averaged DOS

$$
\bar{g}(b,\omega) \equiv \frac{\int_{r \leq b} d^3 r g(\mathbf{r}, \omega)}{\frac{4}{3} \pi b^3},\tag{79}
$$

which characterizes the average $g(\mathbf{r},\omega)$ within a sphere of radius *b*. In the limit $b \rightarrow R$, in which case the local DOS is averaged over the full nanoparticle volume, we obtain the global (or thermodynamic) DOS, which for an *isolated* nanoparticle would be simply

$$
\overline{g}(R,\omega) = \frac{1}{V} \sum_{J} \delta(\omega - \omega_{J}).
$$
 (80)

Physically, we expect *b* to be somewhere between *R*/2 and *R*.

B. Perturbative calculation of the local DOS

The retarded Green's function (72) for the nanoparticle can be obtained by calculating the Euclidean time-ordered (or imaginary time) Green's function defined by

$$
D^{ij}(\mathbf{r}, \mathbf{r}', \tau) = -\langle T u^i(\mathbf{r}, \tau) u^j(\mathbf{r}', 0) \rangle_H. \tag{81}
$$

In the interaction representation,

$$
D^{ij}(\mathbf{r}, \mathbf{r}', \tau) = -\frac{\left\langle T u^i(\mathbf{r}, \tau) u^j(\mathbf{r}', 0) e^{-\int_0^\beta \delta H(\tau') d\tau'} \right\rangle_{H_0}}{\left\langle e^{-\int_0^\beta \delta H(\tau') d\tau'} \right\rangle_{H_0}},
$$
\n(82)

where the expectation values are with respect to H_0 . By expanding the exponentials to leading order in the perturbation (63) and Fourier transforming, Eq. (82) can be written as

$$
D^{ij}(\mathbf{r}, \mathbf{r}', \omega) = D_0^{ij}(\mathbf{r}, \mathbf{r}', \omega) + \sum_{kl} \int D_0^{ik}(\mathbf{r}, \mathbf{r}'', \omega)
$$

$$
\times \Pi^{kl}(\mathbf{r}'', \mathbf{r}''', \omega) D_0^{lj}(\mathbf{r}''', \mathbf{r}', \omega) d^3 r'' d^3 r''',
$$
(83)

$$
D_0^{ij}(\mathbf{r}, \mathbf{r}', \omega) = \sum_J \frac{\Psi_J^i(\mathbf{r}) \Psi_J^{j*}(\mathbf{r}')}{2\rho \omega_J} \left[\frac{1}{i\omega - \omega_J} - \frac{1}{i\omega + \omega_J} \right] \tag{84}
$$

is the free propagator and Π^{ij} is the leading-order phonon self-energy, given at zero temperature by⁹

$$
\Pi^{ij}(\mathbf{r}, \mathbf{r}', \omega) = \frac{K^2}{2\rho} \sum_{\mathbf{I}} \frac{|f_{\mathbf{I}}^z(\mathbf{r}_0)|^2}{\omega_{\mathbf{I}}} \left[\frac{1}{i\omega - \omega_{\mathbf{I}}} - \frac{1}{i\omega + \omega_{\mathbf{I}}} \right] \times \delta^{iz} \delta^{iz} \delta(\mathbf{r} - \mathbf{r}_0) \delta(\mathbf{r}' - \mathbf{r}_0). \tag{85}
$$

The $f_I(r)$ are substrate eigenfunctions discussed in Sec. II, and \mathbf{r}_0 is the point at which the nanoparticle is connected to the substrate. The self-energy (85) is correct away from the centers of the vibrational modes, but ignores small *elastic* scattering contributions that shift the modes without broadening them. Retarded quantities are obtained by analytically continuing $i\omega \rightarrow \omega + i0^+$.

To calculate the local DOS (77) we need to solve the Dyson equation for the nanoparticle Green's function, written symbolically as

$$
D = D_0 + D_0 \Pi D. \tag{86}
$$

The solution to Eq. (86) can be obtained by introducing matrix representations for *D*, *D*₀, and Π , in which case

$$
D = (D_0^{-1} - \Pi)^{-1}.
$$
 (87)

The matrix representation we use is defined by

$$
O(J,J') = \sum_{ij} \int_V d^3r d^3r' \Psi_j^{i*}(\mathbf{r}) O^{ij}(\mathbf{r}, \mathbf{r}') \Psi_{J'}^j(\mathbf{r}'),
$$
\n(88)

where $O = D$, D_0 , or Π . In Eq. (88) the integration is over the volume *V* of the nanoparticle, and the $\Psi_J(\mathbf{r})$ are the nanoparticle eigenfunctions. The inverse transformation is

$$
O^{ij}(\mathbf{r}, \mathbf{r}') = \sum_{JJ'} \Psi^i_J(\mathbf{r}) O(J, J') \Psi^{j*}_{J'}(\mathbf{r}'). \tag{89}
$$

A nanoparticle with a diameter of 10 nm has approximately 8000 atoms in it. Thus, there are roughly 24000 acoustic vibrational modes. By knowing the total number of modes, a Debye energy can be defined: The Debye energy is the energy at which there are 24000 elasticity-theory modes that lie below in energy. For our nanoparticle, the Debye energy is about 320 cm^{-1} . The Debye energy cutoff truncates the Hilbert space, which leads to finite-size matrices.¹⁴ This enables every mode *J* of the nanoparticle to be included in the calculation of the Green's function (87) .

V. RESULTS

In this section we present our results for the phonon DOS in a 10-nm Si nanoparticle, obtained by solving the Dyson equation (86) for the phonon Green's function, as explained above. As we have discussed, the DOS $g(\mathbf{r},\omega)$, defined in Eq. (77) , is a local quantity that varies with position within the nanoparticle, and, as mentioned in the previous section,

where

FIG. 2. The phonon DOS, given in states per wave number per $cm³$, of a 10-nm Si nanoparticle, weakly coupled to a semi-infinite Si substrate.

the quantity we are interested in is $\overline{g}(b,\omega)$, which is $g(\mathbf{r},\omega)$ averaged over a sphere of radius *b* positioned at the center of the nanoparticle. Because we have found no significant dependence of $\overline{g}(b,\omega)$ on *b*, for the physically relevant values of *b* between $R/2$ and *R*, we plot simply $\overline{g}(R,\omega)$. As stated above in Sec. IV A, $\overline{g}(R,\omega)$ is just the *global* phonon DOS in the nanoparticle.

For simplicity we assume both the nanoparticle and the substrate to be made of Si; this allows us to use the surface spectral density (62) calculated in Appendix B of Ref. 11, where Si is treated as an isotropic elastic continuum with longitudinal and transverse sound velocities

$$
v_1 = 8.5 \times 10^5 \text{ cm s}^{-1},
$$

$$
v_t = 5.9 \times 10^5 \text{ cm s}^{-1},
$$
 (90)

and mass density ρ =2.3 g cm⁻³. In the final section of this paper, where we compare our results to the experiment of Ref. 2, we will introduce an important correction to account for the differences between a solid Si substrate and a nanoparticle cluster.

In Fig. 2 the global DOS $g(R,\omega)$ of a 10-nm diameter nanoparticle is given up to 100 cm^{-1} . The (unbroadened) Lamb mode has a frequency [see Eq. (52)] of 15.7 cm⁻¹, but is obscured by the nearby $(l=2, m=0, n=1)$ spheroidal mode at 16.5 cm⁻¹. The modes above 100 cm⁻¹ were included in the calculation, but the long-wavelength approximation of elasticity theory becomes invalid at high energy. Thus only the lower part of the spectrum is shown. Figure 3 shows the low-energy phonon DOS up to about 23 cm^{-1} . An expanded view of the low-energy DOS is given in Fig. 4. The phonon DOS at 3 cm⁻¹ is approximately 4.5×10^{10} states per wave number per cm³.

FIG. 3. Vibrational DOS at low energies.

VI. COMPARISON WITH EXPERIMENT

In this section we compare our results to the experiment of Ref. 2, where the one-phonon emission rate (and therefore the phonon DOS at 3 cm⁻¹) in a cluster of Y_2O_3 nanoparticles was observed to be 8.2×10^{-3} times that in bulk Y_2O_3 . In particular, the excited ${}^5D_1(II)$ state of Eu³⁺ in the nanoparticles had a phonon-emission lifetime of $27 \mu s$, compared with a bulk value of 221 ns. In order to make a comparison of our results to that of the experiment, two modifications of our calculation have to be performed.

(i) In our model, the cluster of nanoparticles has been replaced by a solid substrate. However, the spectral density (62) of the substrate, which at long wavelengths is determined by the sound speeds and mass density of Si, is much smaller than that of the nanoparticle cluster. Treating the long-wavelength modes of the cluster with elasticity theory (or, even simpler, approximating the random cluster by an ordered cubic lattice), shows that the spectral density (62)

FIG. 4. Expanded view of the low-energy DOS. Note that the DOS vanishes at zero energy, as expected.

$$
N_{\rm cl}(\omega) = C_{\rm cl}\omega,\tag{91}
$$

where¹⁵

$$
C_{\rm cl} = \frac{v_{\rm Si}^3 \rho_{\rm Si}}{v_{\rm cl}^3 \rho_{\rm cl}} C_{\rm Si}.
$$
 (92)

Here $v_{\rm cl}$ is a characteristic sound speed in the cluster and $\rho_{\rm cl}$ is its mass density. The $1/v_{\text{cl}}^3$ dependence in Eq. (92) comes from the well-known velocity dependence of the Debye DOS, and the $1/\rho_{\rm cl}$ factor comes from the definition of spectral density [see the discussion following Eq. (76)]. Approximating the cluster by an ordered cubic array with lattice constant d (the nanoparticle diameter) yields

$$
v_{\rm cl} \approx \sqrt{\frac{K}{M}} d \tag{93}
$$

and

$$
\rho_{\rm cl} \approx \frac{\pi}{6} \rho_{\rm Si},\tag{94}
$$

where *K* is the effective spring constant connecting the nanoparticles, given in Eq. (65) , and

$$
M = \frac{4}{3} \pi \rho_{\text{Si}} \left(\frac{d}{2}\right)^3 \tag{95}
$$

is the mass of one nanoparticle. Using $d=13$ nm (the mean nanoparticle diameter in Ref. 2), we obtain an enhancement factor of

$$
\frac{C_{\rm cl}}{C_{\rm Si}} = 9.8 \times 10^2. \tag{96}
$$

This factor increases the nanoparticle DOS (at frequencies below the Lamb mode) by nearly three orders of magnitude. We emphasize that this correction originates from the increased spectral density of the environment—the nanoparticle cluster—compared with that of a solid substrate.

There are several other marginally important corrections, most of which will be ignored, and one that will be included for completeness.

(ii) In the model analyzed above, the nanoparticle was connected to its surroundings by only a single contact point, whereas a nanoparticle in a cluster most likely has more than one connection. As the number of contacts increases, this simply scales the DOS (away from the peaks) linearly with the number of contact points. Conservatively, we expect that the multiple contact points present in the real system will increase the decay rate of the nanoparticles' modes, and hence the phonon DOS well below the Lamb mode, by a factor of 2.

The following additional corrections have also been considered and were found not to be significant (and are not included in our final results):

(iii) The actual experiment of Ref. 2 was done on an ensemble of nanoparticles with mean diameter³ 13 nm and standard deviation of 5 nm. To understand the effects of this size distribution, we have calculated the DOS at 3 cm^{-1} averaged over a Gaussian distribution of diameters centered at 10 nm. Even for very wide distributions (standard deviations up to 8 nm), the ensemble averaged DOS at 3 cm^{-1} is increased by no more than a factor of $2¹⁶$ In addition, the correction for re-centering the size distribution from 10 nm to the experimentally observed 13-nm average size leads to corrections only of order unity.

(iv) Our calculations were done for a nanoparticle and substrate made of Si, while the experiment was done on Y_2O_3 nanoparticles, which, of course, have different mass density and sound velocities. The differences in sound speeds and mass only shift the modes of the nanoparticle by a small amount.¹⁷ This has the same effect as a small change in the diameter of the nanoparticles, which we have found to be negligible. As for the substrate (or more precisely the replacement of the substrate with a cubic lattice of nanoparticles), the change in mass density does effect the spectral function (62) by changing the velocity (93) and mass density (94) , but this change is only of order unity.

 (v) The nanoparticles of Ref. 2 were immersed in He, either liquid (for T <4.21 K) or gas (T >4.21 K). However, the results were found not to change through the liquid-gas transition, presumably because of the large sound-speed mismatch between superfluid He and Y_2O_3 . Therefore we have ignored the presence of He in our theory.

(vi) The experiment of Ref. 2 was done at temperatures between 1.5 and 10 K (excluding the interval $2.17-4.21$ K), whereas our calculations assume zero temperature. The effect of finite temperature is to stimulate phonon emission into the bath (substrate or cluster). However, this is not important until the Bose distribution function of the bath at the Lamb mode frequency (approximately 9 cm⁻¹) becomes of order unity, which does not occur until the temperature exceeds about 13 K.

Taking into account the first two modifications, and ignoring the others, we obtain a 3 -cm⁻¹ DOS given by

$$
\overline{g}(R,3cm^{-1}) = 4.5 \times 10^{10} \frac{\text{states}}{\text{wave number cm}^3} \times 980 \times 2
$$

$$
= 8.8 \times 10^{13} \frac{\text{states}}{\text{wave number cm}^3}.
$$
 (97)

The DOS of bulk Si is given by

$$
\sum_{\lambda} \frac{\epsilon^2}{2\pi^2 h^3 v_{\lambda}^3} = 3.9 \times 10^{15} E^2 \frac{\text{states}}{\text{wave number cm}^3}, \quad (98)
$$

where E is the energy in wave numbers. Thus the theoretical ratio of the DOS of the nanoparticle to that of the bulk material at 3 cm⁻¹ is 2.5×10^{-3} . As stated above, the experimental ratio of nanoparticle to bulk DOS was found to be approximately 8.2×10^{-3} . The agreement between our theory and the experiment of Ref. 2 is excellent considering the simplicity and robustness of our model. We conclude that the low-energy phonon DOS observed in Ref. 2 is consistent with our enviromental broadening mechanism.

VII. CONCLUSIONS

Motivated by an interesting experiment² measuring the low-energy phonon DOS in a insulating nanoparticle, we have thoroughly investigated a simplified model of a single nanoparticle weakly coupled to its environment, a semiinfinite substrate. The environmental interactions were found to significantly affect the DOS at energies below the Lamb mode.

Additionally, we have used the results of our model to predict the effect of environmental interaction in a cluster of nanoparticles like that studied in Ref. 2. Although it is necessary to estimate the value of several quantities appearing in the model, we believe that we can do this accurately enough to obtain a final result that is correct at the order-ofmagnitude level, with no free parameters. Because our results for the 3 -cm⁻¹ DOS is only about three times smaller than that observed in Ref. 2, our broadening mechanism and the resulting phonon spectrum is clearly consistent with that experiment.

Our theory makes several predictions that could be investigated in future experiments: The vibrational DOS as a function of energy at energies below the Lamb mode could be measured by observing the one-phonon emission rate from a magnetic impurity, whose energy splitting can be changed

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- 3 The average nanoparticle size was stated incorrectly in Ref. 2. The correct value is 13 nm. See H. S. Yang, K. S. Hong, S. P. Feofilov, B. M. Tissue, R. S. Meltzer, and W. M. Dennis, J. Lumin. **83**, 139 (1999).
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- 5The experiment of Ref. 2 did not *directly* probe the phonon spectrum of the nanoparticle. However, both the observed exponential decay of the excited electronic state and our estimates of the electron-phonon interaction strength suggest that the nanoparticle is in the relaxational regime where the electron's lifetime is determined by the phonon DOS through Fermi's golden rule. See M. R. Geller, W. M. Dennis, V. A. Markel, K. R. Patton, D. T. Simon, and H. S. Yang, Physica B 316, 430 (2002) for a brief discussion of this question.
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- ⁹Throughout this work we use *I* to label the vibrational modes of

away from 3 cm^{-1} by varying an external magnetic field. Also, the predicted sensitivity of the low-energy DOS to the sound velocity (and hence spectral density) of the environment could be probed by modifying the mechanical properties of the nanoparticle cluster, say, by applying external pressure and measuring the cluster sound velocity along with the phonon emission rate. The lifetime of the broadened vibrational modes could be directly measured as well (by neutron scattering, for example), but splitting of the (degenerate) vibrational modes caused by deviations from spherical symmetry would then be important. We hope that detailed investigations such as these will be carried out in the future.

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- the substrate and *J* for the nanoparticle.
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- 11 K. R. Patton and M. R. Geller, Phys. Rev. B 64 , 155320 (2001) .
- 12 K. R. Patton and M. R. Geller, J. Lumin. **94**, 747 (2001) . The vibrational eigenmodes in this preliminary report have frequencies slightly different than the present work because of a numerical error brought to our attention by Dan Murray.
- ¹³This agreement is fortuitous, however, because we did not account for the modified sound velocity of the bath. After this modification is made, the golden-rule estimate for the lowfrequency DOS is too *large* (by a factor of 50 at 3 cm⁻¹), as expected.
- ¹⁴Each such double-precision complex matrix requires about 9 GB of memory.
- ¹⁵For Si we use the branch-averaged sound velocity

$$
v_{\text{Si}} = \left(\frac{1}{3} \sum_{\lambda} \frac{1}{v_{\lambda}^3}\right)^{-1/3} = 6.4 \times 10^5 \text{ cm s}^{-1}.
$$

- ¹⁶We note, however, that when the Lamb mode frequency becomes resonant with the electronic two-level system, the electronic relaxation rate is not simply related to the phonon DOS.
- ¹⁷For example, the Lamb mode frequency for a 13-nm Y_2O_3 nanoparticle is about 9 cm⁻¹, and for one made of Si is 12 cm⁻¹. These estimates follow from Eq. (52) using the Y₂O₃ transverse sound speed of $v_t = 4.3 \times 10^5$ cm s⁻¹ as measured by C. Proust, Y. Vaills, and L. E. Husson, Solid State Commun. **93**, 729 $(1995).$