

**Manganese nanoclusters and nanowires on GaAs surfaces**

Mogus Mochena

*Department of Physics, Florida A & M University, Tallahassee, Florida 32307*

P. J. Lin-Chung

*Naval Research Laboratory, Washington, DC 20375*

(Received 23 August 2002; revised manuscript received 23 December 2002; published 31 March 2003)

We have computed the ground-state local magnetic moments of manganese and neighboring arsenic for various cluster configurations on the (001) surface of a GaAs bulk crystal using a cluster of 512 atoms. For manganese we obtained a substantial local magnetic moment of  $(3.66 \pm 0.01)\mu_B$  for all cases considered. The induced magnetic moment of arsenic is less than that of manganese by two orders of magnitude, and falls off drastically beyond the nearest-neighbor distance. A small amount of charge is transferred from the manganese atom to the arsenic atom. The possibility of a spin-polarized wire channel on the arsenic layer below the surface is suggested.

DOI: 10.1103/PhysRevB.67.125317

PACS number(s): 73.20.-r, 75.50.Pp, 75.75.+a

**I. INTRODUCTION**

There is currently a great deal of interest in introducing spin-dependent functionality such as memory and storage capabilities into nonmagnetic semiconductors that have been widely used in technology such as silicon, germanium, and gallium arsenide.<sup>1,2</sup> This trend is part of an emerging field that is commonly referred to as “spintronics,” the study of the possibility of utilizing the spin degree of freedom of an electron for device purposes. Semiconductor physics based on the manipulation of the electronic charge has been well established over the past four or five decades. Introducing such a new dimension to devices is a particularly interesting extension in light of current interest in quantum computing and quantum information theory.<sup>3</sup>

Most of the current research towards realizing spin dependent semiconductor devices is focused in two areas: injecting spin-polarized electrons from ferromagnets<sup>4-7</sup> or diluted magnetic semiconductors<sup>8-10</sup> into a semiconductor, or doping a semiconductor with a magnetic impurity.<sup>11-13</sup> Both efforts involve dealing with the physics of the bulk semiconductor. The behavior of the surfaces of these systems has received relatively little effort to date, except for the interface problem of spin injection.

Manganese clusters consisting of two to five atoms in free space have been shown in theoretical studies to retain their atomic magnetic moments, in sharp contrast to bulk manganese behavior.<sup>14</sup> Low-dimensional systems such as surfaces and multilayers are well known to enhance magnetization.<sup>15</sup> The present work explores whether claims of large magnetic moments for clusters in Ref. 14 could be realized on a surface of a semiconductor. We have studied the effect of substituting manganese atoms at the sites of Ga on the (001) surface of GaAs. A tight-binding model is used to take into account the interaction between the bulk semiconductor and the magnetic impurities. The choice of manganese and GaAs is based on their widespread use for study of diluted magnetic semiconductors.

Experimentally the controlled deposition of atoms on semiconductor surface is realized with tunneling micro-

scopy.<sup>16</sup> We are not aware of any experiments involving such a deposition of a few atoms of manganese on the surface of a semiconductor. The results of the present work, therefore, could be helpful to experimental investigations of induced magnetization on semiconductor surfaces.

The remainder of the paper is organized as follows. In Sec. II, we briefly discuss the computational method that has been used to studying local effects, especially of large systems at  $T=0$  K. In Sec. III, we present our results for a dimer, a rectangular cluster, a wire, and a ladder consisting of two wires.

**II. COMPUTATIONAL METHOD**

We use a continuous fraction method that has been developed to study a local perturbation resulting from a defect, adatoms on surface, or local changes in a ferromagnetic or paramagnetic material.<sup>17,18</sup> The advantages of this method relative to the pseudopotential method, the small-cluster augmented method, the scattering theoretical method and the self-consistent Green’s-function technique have been discussed elsewhere.<sup>18</sup> The continued fraction method is based on the real-space Green’s function which can be expanded as

$$G_{\alpha\alpha}(E) = \langle \alpha R_l | [E - H]^{-1} | \alpha R_l \rangle$$

$$= \frac{1}{E - a_0 - \frac{b_1^2}{E - a_1 - \frac{b_2^2}{E - a_2 - \frac{b_3^2}{E - \dots}}}}, \quad (1)$$

where  $H$  is a tight binding Hamiltonian acting on localized orbital  $\alpha$  at site  $R_l$ ,  $|\alpha R_l\rangle$ . The coefficients  $a_i$  and  $b_i$  in the Green function are elements of a tridiagonal matrix representation of a tight binding Hamiltonian resulting from a unitary transformation on the local orbital basis  $|\alpha R_l\rangle$  such that

TABLE I. Slater-Koster parameters in Rydberg units. The superscripts stand for neighboring distances.

	Ga-As	As-As <sup>1</sup>	As-As <sup>2</sup>	Ga-Ga	Mn-Mn <sup>1</sup>	Mn-Mn <sup>2</sup>	As-Mn	Ga-Mn
$ss\sigma$	-0.1311	-0.107	-0.0011	-0.0036	-0.068	-0.032	-0.088	-0.0178
$sp\sigma$	0.1478	0.064	0.0016	0.0043	0.149	0.054	0.107	0.0291
$pp\sigma$	0.3665	0.191	0.0492	0.0629	0.179	0.052	0.185	0.0578
$pp\pi$	-0.1833	-.325	-0.0248	-0.0315	0.049	0.0012	0.0	-0.0151
$sd\sigma$					-0.021	-0.014	-0.010	-0.007
$pd\sigma$					-0.05	-0.025	-0.025	0.012
$pd\pi$					0.023	0.0047	0.0115	0.0024
$dd\sigma$					-0.051	-0.006		
$dd\pi$					0.042	0.0017		
$dd\delta$					-0.0072	-0.00002		

$$\langle U_m | H | U_n \rangle = \begin{cases} a_m, & n = m \\ b_{m+1}, & n = m + 1 \\ b_m^*, & n = m - 1 \\ 0, & \text{otherwise,} \end{cases} \quad (2)$$

where

$$|U_0\rangle \equiv |\alpha R_l\rangle, \quad (3a)$$

$$b_n |U_n\rangle = (H - a_{n-1}) |U_{n-1}\rangle - b_{n-1} |U_{n-2}\rangle, \quad (3b)$$

$$\langle U_n | U_n \rangle = 1. \quad (3c)$$

A tridiagonal Hamiltonian representation has also been used to study elementary excitations at surfaces and interfaces.<sup>19</sup>

As seen from the Green's-function expansion above, a complete knowledge of the coefficients is equivalent to determination of the Green's function. The expansion in Eq. (1) is terminated when the convergence of results for a given size of cluster is achieved. Following Ref. 18 for the bulk GaAs case, we terminated the expansion at twenty five terms in this computation. The local density of states,  $N_{\alpha l}$ , and the integrated density of states,  $n_{\alpha l}$ , are then given by

$$N_{\alpha l}(E) = -\pi^{-1} \text{Im} G_{\alpha l \alpha l}(E + i\varepsilon), \quad (4a)$$

$$n_{\alpha l} = \int_{-\infty}^{E_F} N_{\alpha l}(E) dE. \quad (4b)$$

From the integrated DOS, we obtain the magnetic moments and the amount of charge transfer between Mn and its arsenic neighbors.

In practice the coefficients are obtained by first determining the Slater-Koster (SK) parameters through an interpolative scheme from band structure calculations.<sup>18,20</sup> In this work we used SK two-center parameters from earlier work<sup>20-22</sup> to construct a complete set of SK parameters in Table I. The parameters were scaled according to the prescription given by Ref. 23 to account for variations in lattice constants, and according to Ref. 24 to account for coordination numbers. The energy contribution from the magnetic interaction is incorporated into the diagonal on-site energies as in the Hubbard model.<sup>25</sup> For Mn atoms, the  $3d$ ,  $4s$ , and  $4p$  orbitals are used to study the mixing of the orbitals and

the deviation of the magnetic moment from the free-atom value. For the GaAs bulk crystal, which is represented as a large cluster of 512 atoms, only  $s$  and  $p$  orbitals are included. Currently there are no reliable SK parameters available for ferromagnetic Mn. Therefore, we used the paramagnetic values of Mn from Ref. 21, and modified their site energies according to the energy splitting for ferromagnetic iron as an approximation. The site energies are given in Table II.

### III. RESULTS AND DISCUSSION

We first present the results of substituting a single Mn at Ga site on the (001) surface of the crystal to gain insight into local effects. In this work we do not take into account surface reconstruction; our goal is to obtain an understanding of the effect of the dangling bond on the magnetic interaction at an ideal surface. In Fig. 1 we plot the orbital DOS for Mn substituted at the center of the surface for spin-up and down states and similarly for a site at the center of the bulk for comparison. The  $d$  orbitals peak at their site energy of  $-1.73$  eV for spin-up states and  $0.52$  eV for spin-down states both for the surface and bulk states, and their structure remains more or less intact, in agreement with their localized nature. The  $s$  DOS is less broad and has a higher peak at the surface than at the center. The surface  $p$  DOS changes more from the center site both structurally and for the location of their peaks since they constitute higher-energy levels than the other two orbitals. The overlap of  $s$  and  $p$  DOSs suggests there is more mixing between the  $s$  and  $p$  orbitals with more interaction taking place at the center of the bulk than at the surface. The surface states are more localized because of the reduced degree of freedom and the dangling bonds. There is very little overlap between either the  $s$  or  $p$  orbitals with the

TABLE II. Site energies of Mn in Rydberg at the surface, and at the center of GaAs crystal. The site energies for Ga and As are given in Ref. 20.

	Surface up	Surface down	Bulk up	Bulk down
$3d$	-0.1269	0.0381	-0.1329	0.0281
$4s$	0.0896	0.1976	0.1616	0.1256
$4p$	0.6426	0.7086	0.6161	0.7411

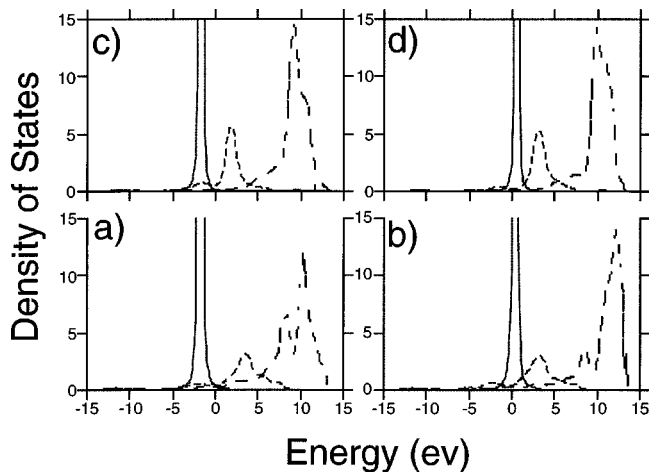


FIG. 1. Density of states for orbitals of Mn at the center and surface of a 512-atoms GaAs cluster. (a) and (b) are for spin-up and down states at the center, and (c) and (d) are for a surface site. The solid lines are  $d$  orbitals, the dotted lines are  $p$  orbitals, and the dashed lines are  $s$  orbitals.

$d$  orbitals with a little bit more overlap for spin up  $s$  surface states.

Local magnetic moments are obtained from the difference of the integrated DOS for spin-up and down states at the Fermi energy which is determined by conservation of the total number of electrons locally. For a single Mn at a surface site, we fixed the Fermi energy by considering the total number of valence electrons on Mn and on the two nearest neighbor As sites. We obtained a magnetic moment of  $3.65\mu_B$  for Mn and  $-0.013$  for As. This result seems reasonable in comparison to some of recent results for similar systems. The experimental magnetic moment of MnAs crystal is around  $3.4\mu_B$ , and Sanvito and Hill reported a saturation magnetic moment of about  $4\mu_B$  in their calculations<sup>26,27</sup> by taking a pair of Mn and As sites. Our As moment is an order of magnitude smaller than that of Ref. 26. The amount of charge transfer from Mn to As is  $0.096e$ , indicating some interaction between the neighbors. This flow of charge from Mn to As and a certain amount of mixing between the  $s$  and  $d$  orbitals at the Mn site could be responsible for the reduction of the Mn moment from the free atom value of  $5\mu_B$ . The 0.093-eV spin splitting of As bands gives an effective exchange coupling constant between the As and the Mn moments of  $J = -1.884$  eV. This value compares reasonably well with similar values deduced from experiments for II-VI ( $-1.1$  eV) (Ref. 28) and for GaMnAs ( $-1.2$  eV).<sup>29</sup> Three higher values of 2.5, 2.8, and 3.3 eV have also been reported.<sup>30-33</sup> The effective coupling constant between manganese at second nearest neighbor distance is equal to a tenth of a meV in our computation. It is much weaker than that between As and Mn. This agrees with kinetic exchange model where the exchange coupling between manganese and the hole only is taken into account.

Next we consider various cases to see the effect of substituting more Mn at Ga sites. Magnetic moments of thin films on semiconductor substrates are found to be negligible due to antiferromagnetic coupling between Mn.<sup>33</sup> Such cou-

pling, at least in the bulk case, according to theoretical arguments<sup>34</sup> depends on distance between the Mn sites; as distance decreases between the Mn, the coupling changes from ferromagnetic to antiferromagnetic. In diluted magnetic semiconductors, a ferromagnetic transition is observed at an average Mn-Mn distance of  $6 \text{ \AA}$ .<sup>32,35</sup> According to recent studies of  $\text{Mn}_x\text{Ge}_{1-x}$ , a ferromagnetic transition occurs at a distance of  $10 \text{ \AA}$ .<sup>13</sup> Recently it has been reported that in small GaMnN clusters in free space Mn couples ferromagnetically with other Mn atoms, but antiferromagnetically with the nitrogen. The latter coupling in turn enhances the ferromagnetic coupling among the Mn atoms.<sup>36</sup> Therefore, the remaining important issue is whether clusters of a few Mn atoms or wires of Mn on a semiconductor surface would also couple ferromagnetically.

We studied a dimer of Mn at the center of the surface, with Mn's at the next-nearest-neighbor distance from each other. The next-nearest-neighbor distance is the shortest distance between Mn's in the substitutional case. Mn could also be placed at closer distances of interstitial or arsenic sites. In doping experiments, Mn "prefers" to go into substitutional (Ga) and interstitial sites, but it rarely goes into an As site.<sup>37</sup> However, we defer the study of Mn at interstitial and arsenic sites that are closer than the next-nearest-neighbor distance to future work. The Fermi energy of the dimer is determined by the conservation of the total number of valence electrons among the dimer, and three of the nearest As neighbors. One of the As's ( $\text{As}_1$ ) is bonded to both of the Mn's of the dimer while the other two ( $\text{As}_2$ ) are bonded singly to the Mn on either side. The moments are found to be  $3.667\mu_B$  for Mn,  $-0.0211$  for  $\text{As}_1$  and  $-0.0151$  for  $\text{As}_2$ . The Mn moment is the same as that of a single Mn impurity, indicating the exchange interaction between Mn's at a next-nearest-neighbor distance is not strong enough to affect the moment. The spin polarization at the  $\text{As}_1$  site is a factor of 1.6 greater than that for a single substitution Mn case, whereas that for  $\text{As}_2$  deviates only slightly from the single Mn case. The spin polarization drops off drastically to  $-0.00028\mu_B$  at the As site that is a neighbor to  $\text{As}_2$  on the same layer but on the opposite side of  $\text{As}_1$ .

Next we considered a rectangular cluster consisting of two dimers. The magnetic moment of Mn remained the same as that of a single dimer, confirming the short-ranged nature of the exchange interaction. The spin polarization of As is same as that of a dimer, and is limited to the nearest-neighbor distance from Mn as before.

We also investigated a wire of Mn atoms on the (001) GaAs surface layer. Then we considered two parallel wires forming a ladder on the surface. The magnetic moments of the Mn were the same in both cases as that of a single substituting impurity, confirming once more the short-ranged nature of the exchange interaction. We note that the magnetic order in the wire of Mn and the ladder does not violate the Mermin-Wagner theorem for one- and two-dimensional systems. The manganese atoms are also coupled antiferromagnetically with arsenic atoms that are on a different plane along the [001] direction. The spin polarizations of As in the two-dimer rectangle case and the ladder case are similar to that of the single-dimer case. Thus the ladder configuration

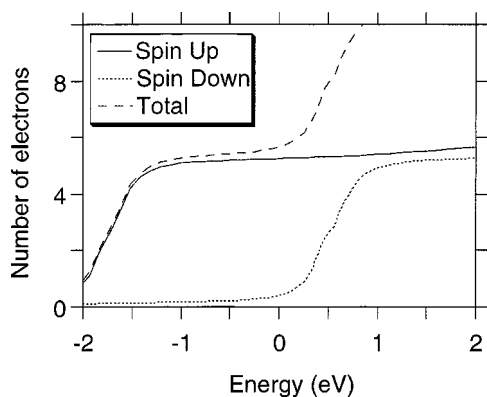


FIG. 2. Integrated density of states for a Mn dimer on GaAs surface.

may provide a spin-polarized wire channel for the transport of holes on the layer beneath the surface.

Our results indicate that a substitutional monolayer of Mn on the surface of GaAs may also retain a ferromagnetic behavior. This indication is consistent with the recent experimental observations of a high ferromagnetic transition temperature in a Mn  $\delta$ -doped GaAs heterostructure.<sup>38</sup>

The sizable local moments retained at Mn sites can be explained in terms of integrated DOS. In Fig. 2, a typical relationship between the integrated local DOS and the energy is plotted. At high energies both spin-up and -down states are equally likely at a site, so that the net magnetic moment

is zero. At low energies, the sites will be occupied by spin-up states only, which is a case of total polarization. In the intermediate range where the spin-down curve is rising in Fig. 2, and where the Fermi energy lies in our calculation, both spin-up and -down states are probable. Since the spin-up curve rises faster, and also earlier, it leads to a net moment.

In conclusion, our ground-state results clearly indicate that locally magnetized nanostructures are possible when magnetic impurities such as manganese are substituted into a tetrahedral structure on a semiconductors surface. Ferromagnetic coupling is possible at a second-nearest-neighbor distance between Mn in GaAs, and is perhaps enhanced by the much smaller antiferromagnetic coupling to arsenic. Here we point out that whether such a large magnetic moment can hold at finite transition temperatures is an open question beyond the capability of our present work. While the moment of Mn is due to the localized  $d$  orbitals, the less localized but spin-polarized orbitals of arsenic below a surface layer of Mn atoms could be used to create spin currents.

#### ACKNOWLEDGMENTS

The authors are grateful to Dr. A. K. Rajagopal for suggesting the problem of nanoclusters on a surface of a semiconductor and many helpful discussions. M.M. is thankful to ASEE for support during the course of this work at NRL. P.J.L.C was supported in part by the Office of Naval Research.

- <sup>1</sup>D. D. Awschalom and R. K. Kawakami, *Nature (London)* **408**, 923 (2000).
- <sup>2</sup>H. Ohno, *Science* **291**, 840 (2001).
- <sup>3</sup>D. Loss and D. P. Divincenzo, *Phys. Rev. B* **57**, 120 (1998).
- <sup>4</sup>Y. Ohno, D. K. Young, B. Beschoten, F. Matsubara, H. Ohno, and D. D. Awschalom, *Nature (London)* **402**, 790 (1999).
- <sup>5</sup>P. R. Hammer, B. R. Bennett, M. J. Yang, and M. Johnson, *Phys. Rev. Lett.* **83**, 203 (1999).
- <sup>6</sup>H. J. Zhu, M. Ramsteiner, H. Kostial, M. Wassermeier, H. P. Schoneherr, and K. H. Ploog, *Phys. Rev. Lett.* **87**, 016601 (2001).
- <sup>7</sup>A. T. Hanbicki, B. T. Jonker, G. Itskos, G. Kioseoglou, and A. Petrou, *Appl. Phys. Lett.* **80**, 1240 (2002).
- <sup>8</sup>R. Fitzgerald, *Phys. Today* **53** (4), 21 (2000).
- <sup>9</sup>B. T. Jonker, Y. D. Park, B. R. Bennett, H. D. Cheong, G. Kioseoglou, and A. Petrou, *Phys. Rev. B* **62**, 8180 (2000).
- <sup>10</sup>I. Malajovich, J. J. Berry, N. Samarth, and D. D. Awschalom, *Nature (London)* **411**, 770 (2001).
- <sup>11</sup>H. Ohno, H. Munekata, T. Penney, S. von Molnar, and L. L. Ching, *Phys. Rev. Lett.* **68**, 2664 (1992).
- <sup>12</sup>H. Ohno, A. Shen, F. Matsubara, A. Oiwa, A. Endo, S. Katsumoto, and Y. Iye, *Appl. Phys. Lett.* **69**, 363 (1996).
- <sup>13</sup>Y. D. Park, A. T. Hanbicki, S. C. Erwin, C. S. Hellberg, J. M. Sullivan, J. E. Mattson, T. F. Ambrose, A. Wilson, G. Spanos, and B. T. Jonker, *Science* **295**, 651 (2002).
- <sup>14</sup>S. K. Nayak and P. Jena, *Chem. Phys. Lett.* **289**, 473 (1998).
- <sup>15</sup>F. J. Himpsel, J. E. Ortega, G. J. Mankey, and R. F. Willis, *Adv. Phys.* **47**, 511 (1998).
- <sup>16</sup>I. W. Lyo and Ph. Avouris, *Science* **253**, 173 (1992).
- <sup>17</sup>R. Haydock, V. Heine, and M. J. Kelly, *J. Phys.* **8**, 2591 (1975); and in *Solid State Physics*, edited by F. Seitz, D. Turnbull, and H. Ehrenreich (Academic, New York, 1980), Vol. 35.
- <sup>18</sup>See for example, P. J. Lin-Chung and A. J. Holden, *Phys. Rev. B* **23**, 3414 (1981); P. J. Lin-Chung and T. L. Reinecke, *ibid.* **27**, 1101 (1983).
- <sup>19</sup>M. Mostoller and A. K. Rajagopal, *Phys. Rev. B* **25**, 6168 (1982); B. X. Xu, M. Mostoller, and A. K. Rajagopal, *ibid.* **31**, 7413 (1985).
- <sup>20</sup>Y. Li and P. J. Lin-Chung, *Phys. Rev. B* **27**, 3465 (1983); *J. Phys. Chem. Solids* **46**, 241 (1985).
- <sup>21</sup>M. J. Mel and D. A. Papaconstantopoulos, *Europhys. Lett.* **31**, 537 (1995); also from <http://cst-www.nrl.navy.mil/bind/mn.html>.
- <sup>22</sup>D. W. Bullett, *Solid State Commun.* **17**, 965 (1975).
- <sup>23</sup>D. A. Papaconstantopoulos, *Handbook of The Band Structure of Elemental Solids* (Plenum, New York, 1986).
- <sup>24</sup>H. J. Gotsis, D. A. Papaconstantopoulos, and M. J. Mehl, *Phys. Rev. B* **64**, 134101 (2002).
- <sup>25</sup>J. Hubbard, *Phys. Rev. B* **19**, 2626 (1979); **20**, 4584 (1979).
- <sup>26</sup>S. Sanvito and N. Hill, *Phys. Rev. B* **62**, 15553 (2000).
- <sup>27</sup>S. Sanvito, P. Ordejon, and N. A. Hill, *Phys. Rev. B* **63**, 165206 (2001).
- <sup>28</sup>T. Mizokawa and A. Fujimori, *Phys. Rev. B* **56**, 6669 (1997).

- <sup>29</sup>J. Okabayashi, A. Kimura, O. Rader, T. Mizokawa, A. Fujimori, T. Hayashi, and M. Tanaka, *Phys. Rev. B* **58**, R4211 (1998).
- <sup>30</sup>J. Szczytko, W. Mac, A. Stachow, A. Twardowski, P. Becla, and J. Tworzydło, *Solid State Commun.* **99**, 927 (1996).
- <sup>31</sup>V. I. Litvinov and V. K. Dugaev, *Phys. Rev. Lett.* **86**, 5593 (2001).
- <sup>32</sup>F. Matsukura, H. Ohno, A. Shen, and Y. Sugawara, *Phys. Rev. B* **57**, R2037 (1998).
- <sup>33</sup>S. M. Porkes and O. J. Glembocki (private communication).
- <sup>34</sup>C. Herring in *Magnetism*, edited by G. T. Rado and H. Suhl (Academic, New York, 1966) Vol Iib, p. 77.
- <sup>35</sup>H. Ohno and F. Matsukara, *Solid State Commun.* **117**, 179 (2001).
- <sup>36</sup>B. K. Rao and P. Jena (unpublished).
- <sup>37</sup>B. T. Jonker (private communication).
- <sup>38</sup>A. M. Nazmul, S. Sugahara, and M. Tanaka, e-print/cond-mat/0208299 (unpublished).