# **Magnetic moments and interactions near the metal-insulator transition in amorphous magnetic semiconductors**

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We report magnetization, magnetic susceptibility, and specific-heat measurements of amorphous  $Gd_xY_ySi_{1-x-y}$  ternary alloy thin films near the metal-insulator (MI) transition as a function of temperature and applied magnetic field. Samples of the same magnetic moment concentration *x* but varying conduction-electron concentration  $x + y$  were measured to test the effect of the MI transition on the magnetic properties. The effective moment in the paramagnetic state (per Gd atom) shows a strong dependence on composition, with a peak at the MI transition, independent of the Gd/Y ratio. Addition of Y weakens the Gd-Gd magnetic interactions, consistent with an indirect RKKY-like exchange interaction, despite the localized or nearly localized nature of the conduction electrons. Specific-heat measurements show further evidence of weakened RKKY-like interactions in the  $y \neq 0$  sample via a spin-glass peak that is shifted to lower temperature, narrowed, and more field dependent.

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## **I. INTRODUCTION**

Previous magnetization measurements on amorphous  $Gd_xSi_{1-x}$  (*a*-Gd-Si) for compositions *x* near the metalinsulator (MI) transition show strong but balanced ferromagnetic and antiferromagnetic exchange interactions which suppress the magnetization well below the noninteracting Brillouin function and lead to a spin-glass freezing at temperatures below 10 K. This freezing shows many features of a classic spin glass including a split between zero field cooled and field cooled dc susceptibility and a weak but nonzero frequency dependence of the freezing temperature  $T_f$  ( $\Delta T_f/T_f$ = 0.04 per log decade change in frequency).<sup>1,2</sup>

In these alloys the susceptibility  $\chi$  in the paramagnetic state above the spin-glass freezing is well described by a Curie-Weiss law:

$$
\chi = A/(T - \theta) \tag{1}
$$

with small  $\theta$  ( $\leq$  2.5 K and significantly below  $T_f$  for all *x*). The effective moment  $p$  in the paramagnetic state (from  $A$  $= n_{\text{Gd}} p^2 \mu_B^2 / 3k_B$ ) shows a striking composition dependence, with a large peak at the MI transition and a significant suppression away from this composition. Related MI induced effects have been seen in magnetic measurements on crystalline P:Si, but with no spin glass-freezing, and the temperature dependence of susceptibility is not a Curie law. $3-6$ 

Gd is a  $4f^75d^16s^2$  atom, virtually always trivalent. The Gd<sup>3+</sup> ion has a large moment,  $J = S = 7/2$  and  $L = 0$ , due to the half-filled *f* shell, hence single-ion anisotropy is negligible. The effective moment  $p^2 = g^2 J(J+1)$  in the paramagnetic state is 7.9  $\mu_B$ . Because of the strongly local character of this large spin-only moment, Gd is not significantly affected by its environment, and exhibits values close to this ground-state moment  $(7\mu_B)$  and this effective moment  $(7.9\mu_B)$  in all known metallic and insulating alloys and compounds, with a possible exception observed by electron-spin resonance in Gd-doped  $\text{SmB}_6$ .<sup>7</sup> We have to date been unable to determine the saturation magnetization in *a*-Gd-Si, as *M* is not saturated even in fields as high as 25 T and still exhibits substantial susceptibility, indicating antiferromagnetic interactions of strength greater than  $25 T (an energy scale equiva$ lent to 120 K for  $J = 7/2$ ).<sup>8</sup>

The effective moment in the paramagnetic state in  $a$ -Gd-Si is however unambiguously not 7.9 $\mu$ <sub>B</sub>, instead dropping to as low as 5.5  $\mu_B$ , and is dependent on composition (up to  $8\mu_B$  at the MI transition). We have speculated that two effects are influencing this effective moment: local moments associated with Si dangling bonds surrounding each Gd ion could be antiferromagnetically coupled to the Gd, acting to reduce the effective moment (and possibly contribute to the high-field susceptibility), and a paramagnetic (positive) contribution from conduction electrons in singly occupied localized states, which peaks at the MI transition as in P:Si. Note that the latter should not be visualized as localized on a single atomic site, but instead in a state with a finite localization length which will diverge at the MI transition and exceeds the interatomic distances even well into the insulating state.

Specific-heat measurements on  $a - Gd_xSi_{1-x}$  show large magnetic contributions at temperatures below 70 K, with a broad peak centered at  $\approx$  1.8  $T<sub>f</sub>$ , which is a common feature of spin glasses. $9-11$  Near the MI transition these measurements also show magnetic entropies above the  $S_{\text{mag}}$  $=$ *R* ln(2*J*<sub>Gd</sub>+1)=*R* ln 8 expected from Gd moments, suggesting that carriers localizing at the MI transition contribute local magnetic moments to the spin system. Theoretical studies of the amorphous rare-earth silicon alloys to date have focused on lattice polarons and the strong effect of the spin disorder on the carrier wave functions via a local momentcarrier exchange interaction<sup>12,13</sup> and have yet to explain many of the observed phenomena.

In metals such as crystalline Gd and  $GdSi<sub>2</sub>$ , magnetism is mediated by an indirect Ruderman-Kittel-Kasuya-Yosida, (RKKY) exchange interaction between the Gd ions, which is ferromagnetic for Gd (Curie temperature  $T_c = 293 \text{ K}$ ) and antiferromagnetic for GdSi<sub>2</sub> (Ne<sup>el</sup> temperature  $T_N$ = 27 K).

Direct exchange between Gd ions is expected to be negligible due to the local nature of the  $4f$  electrons. Disorder alone does not fundamentally seem to change the nature of the interactions:  $a - Gd_xGe_{1-x}$  is ferromagnetic with  $T_c$  $>150$  K for  $x > 0.5$ .<sup>14</sup> In the *a*-Gd-Si alloys, the electron concentration at the MI transition is not precisely known, but lies between  $10^{20}$  cm<sup>-3</sup> (from IR absorption measurements) and  $10^{22}$  cm<sup>-3</sup> (assuming trivalent Gd). The nature of the RKKY interaction in a strongly disordered high electron concentration system has not been theoretically studied, but near the MI transition in amorphous alloys, even on the insulating side, the localization length exceeds the inter-Gd distance and the electron concentration is high, hence an indirect conduction-electron-mediated RKKY-like exchange is the likely magnetic exchange mechanism. The effects of weak disorder have been studied;<sup>15-17</sup> early reports that the RKKY interaction would be exponentially damped with the mean free path were shown to be incorrect, and the primary effect of disorder was shown to be randomization of the phase of interaction. We have proposed that this randomization of phase might be the source of the nearly perfect balancing of ferromagnetic and antiferromagnetic interactions, which leads to the observed Curie law  $\lceil \theta = 0 \rceil$  in Eq. (1) despite the strength of the interactions. $<sup>2</sup>$ </sup>

The purpose of the present work is to study the effect of adding conduction electrons to *a*-Gd-Si on the spin-glass behavior, specifically the magnetic interactions, effective moment, and magnetic entropy. Y is a  $4d<sup>15</sup> s<sup>2</sup>$  atom, also virtually always trivalent and with the same size ionic radius as Gd but with no local moment. Addition of Y while holding constant the Gd concentration therefore has the effect of adding conduction electrons at constant local moment. Effects on the RKKY interaction of changing distance between moments have been frequently studied in metallic systems, but to our knowledge, the systematic study of the effects of changing electron concentration have been reported in only one other system. Varying carrier concentration causes dramatic changes in the magnetic properties of crystalline IV-VI semiconductors such as  $Pb_1 - x - y Sn_y Mn_x Te.$ <sup>18–20</sup> In these materials, as carrier concentration increases, the RKKY interaction at a fixed Mn-Mn distance changes sign, causing a change from ferromagnetism to spin-glass behavior. Though this system and our *a*-Gd-Y-Si have similarly large carrier concentration, the *a*-Gd-Y-Si adds the effects of strong disorder to the problem.

#### **II. EXPERIMENT**

Samples were made by electron-beam coevaporation under UHV conditions onto amorphous Si-N coated Si substrates and Si-N membrane based microcalorimeters which are held near room temperature during the deposition. We have previously described the design and use of these microcalorimeters in detail. $2^{1,22}$  Structural characterization techniques including x-ray diffraction, transmission electron microscopy, and x-ray-absorption fine structure confirm that the samples are amorphous and show no measurable clustering of Gd or Y.

The Si-N coated Si substrates are used for magnetization

measurements. The magnetization and specific-heat samples are therefore grown in the same depositions, allowing direct comparison. The microcalorimeters were fabricated from the same wafer, and have  $\approx$  2000-Å thick Al thermal conduction layers which were grown in the same thermal evaporation. Additional microcalorimeters from this batch were left with no samples deposited in order to measure the specific heat of addenda, *Cadd* . This gives the lowest possible deviation in the thermal properties of the microcalorimeters, which aids in comparing the two magnetic samples. The compositions, Gd concentration  $(Gd~atoms/cm<sup>2</sup>)$ , and areal density (total  $atoms/cm<sup>2</sup>$ ) were determined by Rutherford backscattering (RBS). The thicknesses of the samples were found using a Dektak profilometer. For magnetization measurements, the lateral area of the films on the Si-N substrates was measured by creating a high-resolution image of the sample with a digital scanner. This image is then analyzed using photomanipulation software, and the area of the substrate covered by the film measured to within several percent. The area of the films deposited on the microcalorimeters is determined by the size of the micromachined shadow mask used to define the deposition area.

The samples chosen for this study are an  $a$ -Gd<sub>14</sub>Si<sub>86</sub> film and an  $a$ -Gd<sub>14</sub>Y<sub>7</sub>Si<sub>79</sub> film. The Gd-Si sample is very near the MI transition, with conductivity  $\sigma < 10(\Omega \text{ cm})^{-1}$  at 4 K and a large effective moment.<sup>2</sup> The conductivity of the ternary film with 14% Gd and 7% Y is  $>500(\Omega \text{ cm})^{-1}$  at 4 K, placing it it well on the metallic side of the transition.<sup>23</sup>

Magnetization measurements were made using two Quantum Design superconducting quantum interference device magnetometers, one with a high-field superconducting magnet and a similar system optimized for low-field ac susceptometry. In both high-field dc and ac measurements, the temperature-independent background was determined from a measurement at 300 K. At this temperature the sample's  $\chi$  is very small,  $\ll \chi$  at low T.

In order to measure specific heat, the microcalorimeter is mounted in a sample-in-vacuum He<sup>4</sup> cryostat. This cryostat is compact and can be inserted into either a LHe storage dewar or into the bore of an 8-T superconducting solenoid. In this cryostat the sample may be cooled to  $\approx$  2 K by pumping condensed LHe in a small 1-K pot. The microcalorimeters use  $a$ -Nb<sub>x</sub>Si<sub>1-x</sub> resistive thermometers for measurements at low temperatures. Because small variations in the composition of these thermometers can cause their resistances to be large (several megaohms) at low temperatures, the devices can be difficult to measure at the lowest temperatures. The microcalorimeters for this study were measured from  $\approx$  3.7-90 K. Measurements were taken using the small- $\Delta T$  relaxation method,<sup>24</sup> with  $\Delta T \cong 1\%$ . This method requires three separate measurements. The thermometers must be calibrated, meaning resistance vs *T* must be measured at each temperature. A measurement of the resistance change  $\Delta R$  when a known amount of power *P* is applied to the sample is converted to  $\Delta T$  using the measured calibration. The thermal conductance of the device is given by  $\kappa$  $= P/\Delta T$ . Finally the sample's temperature is recorded as it relaxes back to the block temperature,  $T_0$  from  $T_0 + \Delta T$ , and this single exponential decay is fit to give  $\tau$ . The specific



FIG. 1. Magnetization vs applied magnetic field at 4 K for both samples. Neither sample approaches saturation even at 50 000 Oe. They are also well below the Brillouin function, indicating strong antiferromagnetic interactions.

heat is then  $C = \kappa \tau$ . Note that the measurement of  $\kappa$  requires an accurate calibration of *R* vs *T*, while the  $\tau$  measurement is independent of this calibration.

## **III. RESULTS**

Figure 1 shows  $M(H)$  at 4 K for the  $x=14, y=0$  and x  $=14, y=7$  samples. Expected saturation magnetization values, assuming  $Gd^{3+}$  ions with  $J = S = 7/2$ , are  $M_{sat}$ =451 emu/cm<sup>-3</sup> for *a*-Gd-Si and  $M_{sat}$ =456 emu/cm<sup>-3</sup> for the *a*-Gd-Y-Si. No hysteresis is seen above 1000 Oe for any sample at any temperature. All samples have magnetizations significantly below saturation and below the Brillouin function, with large high-field susceptibility, indicating strong antiferromagnetic interactions (since single-ion anisotropy and other contributions such as Van Vleck susceptibility should be negligible). With increasing temperature,  $M(H)$  decreases for all samples, but  $M(H,T)$  does not scale for any sample with *H*/*T*, as found previously for pure *a*-Gd-Si, also an indication of the importance of interactions. The expanded scale inset in Fig. 1 shows that the initial susceptibility is significantly larger for the pure *a*-Gd-Si sample ( $y=0$ , very near the MI transition) than for the  $y=7$  sample (metallic), and that the data cross at about 10 000 Oe, above which the pure *a*-Gd-Si sample has lower magnetization and lower differential susceptibility  $\partial M/\partial H$  than the ternary *a*-Gd-Y-Si sample. These two samples have virtually identical Gd concentrations (Gd atoms/cm<sup>2</sup> and total atoms/cm<sup>2</sup>), as measured directly by RBS, and thicknesses  $(Gd atoms/cm<sup>3</sup>)$ .

Figure 2 shows the ac susceptibility  $\chi(T)$  measured in zero dc field with 406-Hz ac field amplitude of 4 Oe and 5 Oe. Data taken in both fields collapse together, indicating that the low-field magnetization is linear in field. The *a*-Gd-Si sample shows the same classic spin freezing as previously measured, with the 406-Hz peak value in  $\chi$  indicating a freezing temperature of  $T_f$ =5.9 K. The ternary sample shows a similar but smaller peak in  $\chi$  at a lower temperature  $T_f$ =5 K. For both samples above  $T_f$ , in the paramagnetic state,  $\chi$  is independent of frequency and has been fit with a Curie-Weiss law  $[Eq. (1)]$  shown as a solid line. It is clear even with no analysis that  $\chi$  is drastically affected by the



FIG. 2.  $\chi_{ac}$  for *a*-Gd-Si and *a*-Gd-Y-Si samples as a function of temperature from 2 to 50 K. The cusps occur at the spin freezing temperatures  $T_f$ . Solid lines are Curie-Weiss fits to the data above  $T_f$ . Inset: The region near  $T_f$ .

addition of Y; the susceptibility at all temperatures for *y*  $=0$  is twice that of the  $y=7$  sample. The Curie-Weiss law fits of the paramagnetic state above  $T_f$  show that *A* indeed is nearly doubled, indicative of an increase in the effective moment *p* by  $\sqrt{2}$ , as well as an increase in  $\theta$  from approximately zero for  $y=7$  (metallic) to 2.5 for  $y=0$  (MI transition).

Figure 3 shows *p* vs overall composition  $x + y$  for both samples in this paper, as well as the data from the earlier work on pure  $a$ -Gd<sub>x</sub>Si<sub>1-x</sub> ( $y=0$ ) and  $a$ -Tb<sub>x</sub>Si<sub>1-x</sub>.<sup>2,25</sup> *p* is determined from the constant  $A = n_{\text{Gd}} p^2 \mu_B^2 / 3k_B$  in the Curie-Weiss fit shown in Fig. 2. Here  $n_{\text{Gd}}$  is the number of Gd atoms/cm<sup>3</sup>. *p* is therefore measured in  $\mu_B$  per Gd atom. The MI transition (based on electrical conductivity measurements<sup>1,26,27</sup>) is shown as a line at  $x=14$ . The effective moment  $p$  is largest at the MI transition and is suppressed on either side, independent of whether the shift in overall composition is due to changes in Gd or Y concentration.

The suppression in  $p$  from the expected values of 7.9  $\mu_B$ for Gd<sup>3+</sup> and 9.7  $\mu_B$  for Tb<sup>3+</sup> far from the MI transition is not understood at present, but we suggest that it is associated with a local polarization of antiferromagnetically coupled Si



FIG. 3. Effective moment *p* as a function of metal ion content  $x + y$  for several Rare-Earth–Silicon alloys. The peak occurs near the MI transition. The Gd-Si sample measured for this study has 13.5 at.% Gd. Lines are a guide to the eye.



FIG. 4. The freezing temperature  $T_f$  and Curie-Weiss  $\theta$  vs Gd concentration. The peak in  $\theta$  and the near zero value found for the strongly metallic ternary sample (large solid circle) indicates the importance of the MI transition at 14 at.% Gd. The downwardpointing arrow indicates that 1.5 K is an upper limit on the freezing temperature of the 4 at.% Gd sample. Lines are a guide to the eye.

dangling-bond states surrounding each rare-earth ion; electron-spin-resonance measurements are underway to clarify this.

Figure 4 shows the freezing temperature  $T_f$  and  $\theta$  versus Gd concentration.  $T_f$  is not significantly affected by the MI transition (it passes smoothly through, with a possible change in slope) but does increase with increasing Gd concentration, as is expected.  $T_f$  also decreases on adding Y for the same Gd concentration.  $\hat{\theta}$  shows the same peak at the MI transition as *p*.

Figure 5 shows the specific heat of the *a*-Gd-Si and *a*-Gd-Y-Si samples from  $\approx$  3 K up to 40 K, as well as an  $a$ -Y<sub>21</sub>Si<sub>79</sub> sample. The units of specific heat in this plot are J/mol K, where a mole counts all atoms in the sample. We calculate the total number of atoms in each sample from the areal density, which is a direct result of the RBS measurements. Below  $T_f$  the specific heats of the two magnetic samples  $(x\neq 0)$  are nearly identical. As the temperature increases, *a*-Gd-Y-Si initially has a slightly larger specific heat, then falls well below that of *a*-Gd-Si. Both magnetic samples





FIG. 6. Magnetic contribution  $C_{mag}$  vs *T* for  $x=14$ ,  $y=0$  and  $x=14$ , $y=7$  samples. Both show broad peaks above  $T_f$  commonly seen in spin glasses. The addition of Y causes a reduction of  $T_f$ causing the peak to shift to lower temperature. Inset: a  $C_{mag}$  vs  $T/T_f$  for the same samples shows that the peak occurs at  $\approx 1.8T_f$ and that in the  $y=7$  sample  $C_{mag}$  is smaller than the  $y=0$  sample  $C_{mag}$  for  $T/T_f$ <8.

have specific heats much larger than  $a - Y_{21}Si_{79}$  at low temperatures. The inset in Fig. 5 shows the same samples up to 90 K. The specific heats of *a*-Y-Si and *a*-Gd-Si converge at the highest temperatures, while the ternary alloy has a  $C_p$ that is slightly larger than the other two samples, and larger than the *a*-Gd-Si sample above 40 K. Because the magnetic measurements described above and many aspects of the specific heat which will be detailed suggest that both the magnetic interactions and the effective moment are smaller in the *a*-Gd-Y-Si sample, it is unlikely that this additional specific heat is from an increased magnetic contribution, particularly since it is seen at 80–90 K. More likely, this larger  $C_p$  is a result of a small change in the phonon spectrum caused by the addition of the Y ions.

Figure 6 shows the estimated magnetic contribution to the specific heat C*mag* for both magnetic samples in units of J/mol[Gd] K, meaning that the mole counts only Gd atoms in the samples. To determine  $C_{mag}$  we treat  $C_p$  as a sum of magnetic, electronic, and phonon contributions,  $C_p = C_{mag}$  $+C_{el}+C_{phon}$ . Because yttrium and gadolinium have similar ionic radii and valences, we assume that  $C_{el}$  and  $C_{phon}$  are the same for *a*-Y-Si and *a*-Gd-Si samples of similar metal content. This is justified by their similar high-temperature  $C_p$ and is equivalent to assuming that *a*-Y-Si's specific heat has negligible  $C_{mag}$ . We estimate  $C_{mag}$  for the  $y=0$  sample by subtracting a high-order polynomial fit to the  $a - Y_{21} \text{Si}_{79}$   $C_p$ from the measured  $C_p$ . Because the specific heats of  $a-Y_{14}Si_{86}$  and  $a-Y_{21}Si_{79}$  are very similar at all but rather low temperatures (<15 K) where both  $a$ -Y-Si films'  $C_p$  are much smaller than the magnetic samples, we simplify the analysis by using  $a - Y_{21} \text{Si}_{79}$  for both magnetic alloys.<sup>28</sup>

FIG. 5. The specific heat of  $a$ -Gd<sub>14</sub>Si<sub>86</sub>,  $a$ -Gd<sub>14</sub>Y<sub>7</sub>Si<sub>21</sub> and  $a-Y_{21}Si_{79}$  samples in zero field. Inset: The same plot showing temperatures up to 90 K.

Adding atoms to a material's unit cell commonly results in additional optical-phonon bands appearing in the dispersion relation. To estimate  $C_{mag}$  for the  $y=7$  sample we must account for the slightly larger  $C_{phon}$  seen at higher temperatures in Fig. 5. To do this we have added to the  $a - Y_{21} \text{Si}_7$ data a contribution from an Einstein mode with  $\theta_F$  $=150$  K, and  $m=0.018$  of the form

$$
C_{Einstein} = 3mR(\theta_E/T)^2 \frac{e^{(\theta_E/T)}}{(e^{(\theta_E/T)} - 1)^2}.
$$
 (2)

This term approximates the contribution to the specific heat from a soft optical mode. The values of  $\theta_F$  and *m* were chosen so that the high-*T* data match in value and slope. We note that this  $C_{Einstein}$  will be very small at low *T*, where the obvious contributions to  $C_{mag}$  dominate, and only becomes significant when the ternary sample's  $C_p$  begins to exceed that of pure *a*-Gd-Si.

The resulting  $C_{mag}$  shown for both samples in Fig. 6 is similar to that seen in typical spin-glass materials, with a broad peak centered at  $\approx 1.8T_f$ . This spin-glass peak is shifted to lower temperature by addition of Y, reflecting a shift in  $T_f$ , as expected from the magnetization results. The *Cmag* peak is also at a slightly lower value in *a*-Gd-Y-Si, and narrower than that seen in  $a$ -Gd-Si. Also note that  $C_{mag}$  for both samples converge at higher temperature  $(>60 \text{ K})$ .

These points are reinforced by the  $C_{mag}$  vs  $T/T_f$  plot shown in the inset of Fig. 6. Here the temperature axis has been scaled by the value of  $T_f$  determined by the peak in the ac susceptibility  $(Fig. 2)$ . This makes clear that the spin-glass peak occurs at  $\approx 1.8T_f$  in both samples, and that the peak in the ternary alloy is narrower and smaller than the *a*-Gd-Si sample.

Figure 7 shows  $C_{mag}$  vs *T* from  $\approx$  3 K to 60 K for both samples in applied magnetic fields up to 8 T. These samples show behavior similar to both the previously measured *a*-Gd-Si samples and well-known spin glasses such as CuMn, with the specific heat reduced at low temperatures and increased at higher temperatures by applied magnetic field.10,29 A unique feature of the *a*-Gd-Si samples is that applied fields cause shifts in  $C_p$  up to quite high temperatures, a result also apparent in Fig. 7. For both samples the 8 T data are measurably larger than those taken in zero field even at 60 K, which is at least 10  $T_f$ .

Though the behavior of these two samples in applied field is qualitatively similar, the quantitative effects of applied field are rather different. Figure 7a shows that in an 8 T magnetic field, the *a*-Gd-Si's spin-glass peak is only slightly lowered and is broadened, but mostly retains its shape. The effect is that of the spin-glass peak shifting to higher temperatures. This is in contrast to the situation in *a*-Gd-Y-Si shown in Fig.  $7(b)$ , where at 8 T, the peak is significantly lower, and more broadened on the high-T side. This sample shows substantial field dependence at 4 T, and easily measurable shifts at 2 Tesla, though this data is omitted for clarity.

Figure 8 shows the field induced change in specific heat,  $\Delta C = C(H=8 T) - C(H=0 T)$  divided by  $C(H=0 T)$  as a function of scaled temperature  $T/T_f$ . The result is the relative change in specific heat at this field. Because we have previously confirmed that the thermal conductance  $\kappa$  of our



FIG. 7. Magnetic contribution to specific heat,  $C_{mag}$ , vs  $T$  for  $y=0$  and  $y=7$  samples in applied magnetic fields up to 8 Tesla. Applied field causes larger shifts in  $C_{mag}$  in the  $y=7$  sample.

microcalorimeter shows no field dependence<sup>22</sup> all the field dependence in our measurement appears in  $\tau$ . Note therefore that

$$
\frac{\Delta C}{C} = \frac{C_H - C_0}{C_0} = \frac{\kappa \tau_H - \kappa \tau_0}{\kappa \tau_0} = \frac{\Delta \tau}{\tau_0}.
$$
 (3)

This means that we obtain  $\Delta C/C$  using only measurements of  $\tau$  in high and zero field, with no possible influence from addenda or nonmagnetic analog subtractions or from fitting errors in the thermometer calibration required to calculate  $\kappa$ .



FIG. 8. Relative change in  $C_p$  at 8 T for both samples

Sample	$S_{mag}$ (4 < T < 90 K)	$S_{excess}$ $[S_{mag} - R \ln(2J_{Gd} + 1)]$	n (localized $e^-$ /Gd ion)	$\mathcal{D}_n$ $R \ln(2J_{eff}+1)$
$x=14, y=0$	20.0	2.7	3.9	17.1
$x=14, y=7$	18.5	1.2		15.7

TABLE I. Magnetic entropies and  $n$  [from Eq. 5] for the two samples. All entropies are in units of  $J/mol[Gd]$  K.

This greatly simplifies the reduction of the data and makes this quantity the most accurate measure of field induced changes in  $C_p$ .

There are several interesting features apparent in Fig. 8. The simplest is that the data elegantly confirm the greater influence of magnetic field on the  $y=7$  sample.  $\Delta C/C$  is larger (either more negative or more positive) for the ternary sample for all  $T/T_f$ . Apart from the difference in magnitude, the two data sets have remarkably similar shapes. Both datasets have an apparent maximum negative value very near the freezing temperature  $(T/T_f \approx 1.2)$ , both cross zero at  $T/T_f \approx 2.7$ , and both have a maximum positive value at  $T/T_f \approx 4$  with tails extending out to at least  $T/T_f = 10$ . The inset shows the negative maximum near  $T_f$ . This negative maximum indicates that as temperature increases from the lowest measured temperatures to  $T_f$ ,  $C_p$  is increasingly field dependent. At or slightly above  $T_f$  the field dependence begins to drop roughly linearly and at a common value of  $T/T_f \approx 2.7$  for both these samples,  $\Delta C/C = 0$ , meaning that the specific heat at 0 and 8 T is the same. Then  $\Delta C/C$  increases roughly linearly and peaks at  $T/T_f \approx 4$  for both samples.

That any feature is seen at  $T_f$  is somewhat surprising, as none of the other high-field experiments, including magnetoresistance<sup>1,26,27</sup> and magnetization<sup>2</sup> have seen any feature near  $T_f$  in this material. Though this feature has thus far not been reported for other spin glasses, it is confirmed by data on other *a*-Gd-Si films.<sup>10</sup>

Table I shows measured and expected values of magnetic entropy for both  $y=0$  and  $y=7$  samples. The magnetic entropy  $S_{mag}$  is given by

$$
S_{mag} = \int_{T=0}^{T_{max}} \frac{C_{mag}}{T} dT,
$$
\n(4)

hence the area under the curve in a  $C_{mag}/T$  vs  $T$  plot gives the measured  $S_{mag}$  for a particular temperature range. The *Smag* values that appear in Table I are determined by plotting  $C_{mag}/T$  vs  $T$  in zero field for each sample and numerically integrating over the temperature range  $3.99 \text{ K} < T < 88 \text{ K}$ . *Smag* is significantly larger for the pure *a*-Gd-Si sample, and both samples have larger entropy than is expected from Gd ions alone  $(S_{Gd} = R \ln(2J+1)) = R \ln 8 = 17.3 \text{ J/mol}$ [Gd] K. Because the lower limit of these measurements is 3.99 K, these  $S_{mag}$  values in fact underestimate the magnetic entropy because  $C_{mag}$  continues to lower temperatures. The entropies *Sexcess* are the results of subtracting the Gd ion entropy from the measured entropy. Note that the  $y=0$  sample's  $S_{excess}$  is roughly twice that of the  $y=7$  sample. We have previously suggested that  $S_{excess}$  is the contribution from localized electrons in singly occupied states.<sup>9</sup> We therefore write the maximum magnetic entropy of a collection of Gd moments  $(J_{\text{Gd}})$  $=7/2$ ) and *n* localized electron moments per Gd ion ( $s_e$ )  $= 1/2$ ) as

$$
S_{mag} = R[\ln(2J_{\text{Gd}} + 1) + n \ln(2s_e + 1)].
$$
 (5)

*n*, the number of localized electrons per Gd ion, may be calculated and the results appear in Table I. Note that *n* is also roughly a factor of 2 larger in the  $y=0$  sample.

A slightly different approach to the magnetic entropy is to first calculate the effective spin  $J_{eff}$  from the values of  $p$ shown in Fig. 3 from  $p^2 = g^2 J_{eff}(\tilde{J}_{eff} + 1)$ , where *g* is the Lande *g*-factor and is assumed to still be  $g=2$ . Then values of  $S_p$  are then calculated from  $S_p = R \ln(2J_{eff}+1)$ . Note that  $S_{mag} - S_p \approx 3$  J/mol[Gd] K for both  $y = 0$  and  $y = 7$  samples.

#### **IV. DISCUSSION**

The effect of increasing electron concentration on the magnetic interactions is very clearly seen in the changes in both  $M(H)$  for the  $x=14$  pair shown in Fig. 1 and in the magnetic susceptibility  $\chi(T)$  shown in Fig. 2. The increased effective moment of the  $y=0$  sample is visible in the lowerfield  $M(H)$  data, where  $y=0$  has larger values (Fig. 1, inset) than  $y=7$ , and is dramatically seen in the factor of 2 difference in  $\chi$ . At higher fields, the crossing of the data directly reflects the reduced interaction strength for the  $y=7$  sample caused by increasing electron concentration. Since the Brillouin function represents *M*(*H*,*T*) for noninteracting moments, suppression below this means antiferromagnetic interactions; the increased values of *M* at high fields for the ternary sample  $(y=7)$  therefore reflect a *reduction* in the strength of the antiferromagnetic interactions, consistent with the reduced freezing temperature  $T_f$  and  $\theta$  in the  $y=7$ sample relative to the  $y=0$  sample, shown in Fig. 4. We note again that these two samples have been shown by RBS to possess identical Gd concentrations and inter-Gd distances (within the accuracy of RBS, which is  $\approx$  1%). Because the magnitude of the RKKY interaction goes as  $(1/k<sub>F</sub>R)^3$ , a reduction in interaction strength with increasing electron concentration is consistent with an RKKY-like conductionelectron-mediated exchange, assuming a constant nearestneighbor Gd-Gd distance *R*. This is a slightly different result than observed in the  $Pb_{1-x-y}Sn_vMn_xTe$ , where the carrier induced ferromagnet to spin-glass transition is caused by a change in the sign of the RKKY interaction due to increased  $k_F$ . It is likely that the presence of strong disorder and resulting randomization of the phase of the interaction prevents such a well-defined transition from occurring in our system.

In the presence of strong frustrated interactions such as are found here, it is not clear that the Curie-Weiss mean-field approach  $[Eq. (1)]$  is the correct physics for the paramagnetic state above  $T_f$ . We note however that the peaks in p and  $\theta$  at the MI transition are both associated with an increased  $\chi$ , which is the essential physics. Also,  $\theta$  is quite small [relative to both the freezing temperature and the strength of the interactions as manifested by the high-field  $M(H)$  suppression below the Brillouin function and the high-field susceptibility]. We note that in metallic Gd glasses,  $\theta$  usually greatly exceeds the freezing temperature (typical  $\theta$ >25 K for  $T_f$  $\approx$  5 K). In the present samples, the near zero value of  $\theta$ , despite the clear presence of strong antiferromagnetic interactions, shows again the nearly perfect balancing of ferromagnetic and antiferromagnetic interactions, which we suggest is a consequence of the disorder-induced phase randomization of the RKKY interaction.<sup>15–17</sup> We suggest that the key elements for this near perfect balancing of strong interactions are strong disorder and lower electron concentration than the usual RKKY glasses (hence longer interaction range).

Several features of the magnetic contribution to the specific heat,  $C_{mag}(T)$ , also point to strong, balanced, longrange RKKY-like interactions, with weaker interactions in the  $y=7$  sample.  $C_{mag}$  provides information on the density of magnetic states in these materials, and thereby the distribution of interaction strengths present in these spin glasses. The large peaks in  $C_{mag}$  for both samples at  $\approx 1.8T_f$  indicate the large number of nearly degenerate magnetic states with energies close to  $T<sub>f</sub>$ . The fact that adding conduction electrons to the system causes the spin-glass peak in  $C_{mag}$  to be lower in value and narrower means that there are fewer magnetic states near  $T_f$ , with a narrower distribution in energy. This result is of course also clear from the  $S_{mag}$  values which are much lower for  $y=7$ , and is expected from an RKKYlike interaction with an increased electron concentration.

The difference between the magnetic specific heats for the two samples is less dramatic than that observed in the susceptibility. This follows a pattern also observed in crystalline phosphorus-doped silicon (P:Si), where measurements of  $\chi$ showed large differences across the MI transition as electrons localized into singly occupied states, while  $C_p$  showed more subtle effects.<sup>3,4,30</sup> This is expected, as  $\chi$  measures the fluctuations of magnetic moments in response to an applied field and is therefore sensitive to the total number of localized moments *n*.  $C_{mag}(T)$  depends on the density of magnetic states, and therefore *dn*/*dT*. Integrating to determine  $S_{mag}$  returns the emphasis to the total number of magnetic states, and in our measurements the difference between  $S_{excess} = S_{mag} - R \ln 8$  for the two samples (and the resulting difference in *n*) is approximately a factor of two, similar to the change seen in  $\chi$ . The difference in the field dependence of *Cmag* for these 2 samples is similarly dramatic, which again indicates weaker interactions and recalls the different field dependences of  $M(H)$  for the two samples.

Another interesting result is the fact that for both these samples  $C_{mag}$  and  $\Delta C/C$  persist to temperatures as high as  $T/T_f \approx 13$  ( $T \approx 80$  K) and are of similar value at these temperatures. This suggests that the strongest interactions in the distribution are less affected by moving away from the MI transition. This temperature range is also the same as that over which the *a*-Gd-Si demonstrates measurable magnetoresistance.<sup>1</sup> Such a link between the specific heat and electron transport is unique and suggests a need for further study.

## **V. CONCLUSIONS**

In summary, we have shown that the susceptibility of amorphous Gd-Si alloys has a strong peak (a factor of 2) at the MI transition, independent of whether the MI transition is caused by increasing magnetic Gd content or nonmagnetic Y content. This peak in susceptibility can be described in terms of an increased effective moment. The magnetic contribution to specific heat shows spin-glass peaks near  $T/T_f \approx 1.8$ , with the metallic  $(y=7)$  sample showing a smaller, more fielddependent  $C_{mag}$ . Both samples have measurable  $C_{mag}$  as high as  $T/Tf \approx 13$ ,  $T \approx 80$  K, the temperature at which negative magnetoresistance becomes measurable in these samples.

We have also shown through measurements of both magnetic susceptibility and specific heat that the magnetic interactions in amorphous Gd-Si alloys are strong, of mixed sign (nearly perfectly balanced ferromagnetic and antiferromagnetic) and are reduced in absolute magnitude by the addition of Y (while maintaining the Gd-Gd distance and concentration constant). We suggest that the latter observation is consistent with an RKKY-like indirect exchange mediated by conduction electrons, despite the conduction electrons nearly being localized, in that increasing electron concentration reduces the strength of the interaction at a given distance.

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