Impurity-induced antiferromagnetic order in the Haldane-gap compound $PbNi_{2-x}Mg_xV_2O_8$ (x = 0.24)

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An investigation of the effect of nonmagnetic Mg doping in the Haldane gap system $PbNi_2V_2O_8$ is presented. Specifically, the magnetic properties of the quasi-one-dimensional compound $PbNi_{2-x}Mg_xV_2O_8$ (x = 0.24), located in the heavily doped region of the T_N , x phase diagram, have been examined using neutron powder diffraction, muon-spin relaxation (μ^+SR) and magnetic susceptibility. Neutron-diffraction measurements show that antiferromagnetic long-range order (propagation vector $\kappa = [000]$) is established below $T_N = 3.2$ K. The spins in the Ni sublattice lie either parallel or antiparallel to the c axis, which is the magnetic easy axis of the system. The spin configuration is antiferromagnetic both along the NiO₆ chains, i.e., the c axis of the crystal structure, and within the ab-plane for the nearest-neighbor Ni atoms. The refined ordered moment at the Ni²⁺-site is $\mu = 0.9(1)\mu_B$. To our knowledge, this is the first diffraction evidence of the antiferromagnetic ordering in this system. The static magnetic correlations of the doping-induced magnetic moments along the Ni chains were probed by zero-field μ^+SR at $T < T_N$. The large number of detected spontaneous muon-precession frequencies ($0.5 < \nu_{\mu} < 3$ MHz) implies a nonuniform distribution in the magnetic behavior below T_N , consistent with the picture of an antiferromagnet with magnetic anisotropy and competing exchange interactions.

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I. INTRODUCTION

In recent years, the magnetic properties of lowdimensional quantum Heisenberg antiferromagnets (HAF's) have attracted a lot of interest both from theoretical and experimental points of view.^{1,2} The knowledge acquired by studying these relatively simple systems can be implemented in understanding physical phenomena in the more complex three-dimensional solids. For instance, in the quest for unravelling the mechanism of high- T_c superconductivity, questions regarding the relevance of the pseudogap in twodimensional cuprates demanded consideration. To help unravel this interrelationship, a vigorous research effort has been directed toward analogous low-dimensional spin systems, especially those with a spin-gap ground state. To this extent, various quasi-one-dimensional (1D) materials have been explored,² including the spin-Peierls $CuGeO_3$,³ the Haldane compound Y₂BaNiO₅, (Ref. 4) and spin-ladder systems like the two-leg ladder SrCu₂O₃.⁵ Such systems can also serve as a bridge between 1D and complex 2D systems.

The pioneering work of Haldane⁶ led to a breakthrough in our understanding of low-dimensional magnetism. Haldane

established that the actual ground states are different for integer and half-integer Heisenberg spin systems. Thus, while half-integer systems can show quasi-long-range order,⁷ integer spin systems have an energy gap in the spin excitation spectrum and exhibit a singlet, nonmagnetic ground state.⁶ A rigorous description of the Haldane conjecture is presented by the valence-bond-solid (VBS) model,⁸ which allows for a free $S = \frac{1}{2}$ state to be associated with each end of an open chain. Electron-spin-resonance experiments on the Cu (Ref. 9) and Zn (Ref. 10) doped S = 1 linear-chain Heisenberg antiferromagnet (AF) [Ni(C₂H₈N₂)₂(NO₂)]ClO₄ have provided strong experimental evidence in support of the VBS ground state.

Among the various factors that can suppress the Haldane gap and change the singlet ground state to a long-rangeordered (LRO) one, are single-ion anisotropy and 3D interchain coupling, which have been studied extensively.¹¹ Another possible factor predicted to result in the emergence of a LRO magnetic state is the creation of spin vacancies in quantum chains.¹² This largely unexplored issue of the suppression of a Haldane gap and the concurrent appearance of LRO demands new experimental investigations on appropriate *model* compounds. Recent experiments of this type have shown contrasting behavior, and hence are the source of motivation for further work in this direction. As such dilutioninduced free moments (on substituting Ni sites by nonmagnetic cations, e.g., $A = Zn^{2+}$ or Mg^{2+}) and quantum fluctuations in chain compounds with a spin-singlet ground state drive Y₂BaNiO₅ to a ground state, which resembles that of the VBS,¹³ in contrast to PbNi₂V₂O₈, which undergoes a transition to long-range ordering.¹⁴

In the Mg-doped Haldane-gap PbNi₂V₂O₈ compound, neutron powder diffraction data have unambiguously shown the transformation of the disordered ground state to an antiferromagnetic one.¹⁵ In this paper, we present a detailed account of variable temperature neutron powder diffraction (NPD), zero-field muon-spin relaxation (μ^+ SR), and magnetic susceptibility results in the $PbNi_{2-x}Mg_xV_2O_8$ (x =0.24) quasi-one-dimensional solid. We describe the spin configuration of the induced antiferromagnetic ground state. Zero-field μ^+ SR reveals that the transition to the LRO AF state is characterized by a magnetic-field distribution with large inhomogeneities. The magnitude of the staggered moment, generated by the Mg dopants, is derived by a Rietveld refinement of the neutron data. We show that the size of the moment and the presence of multiple muon-precession frequencies in the ZF- μ^+ SR spectra of PbNi_{1.76}Mg_{0.24}V₂O₈ imply a nonuniform ordered moment size along the helical NiO₆-chains. We discuss microscopic features of the AF LRO state based on doping-induced ordering phenomena in quasi-1D compounds. Extended dc and ac magneticsusceptibility measurements also show a low-temperature metamagnetic behavior for this composition.

II. EXPERIMENTAL DETAILS

Polycrystalline PbNi_{2-x}Mg_xV₂O₈ (x=0 and 0.24) powders were prepared by solid-state reaction of stoichiometric quantities of PbO, NiC₂O₄·2H₂O, MgO, and V₂O₅ in air. The mixture of the starting reagents (purities >99.9%) was first heated at 550 °C for 24 h to eliminate carbon-containing species. Successive sintering cycles at elevated temperatures, ranging from 750° to 780 °C, were carried out after repeated grinding and pelletizing of the powders. These hightemperature reactions (followed by slow cooling to room temperature) took place over a period of ten days with three intermediate grindings. The phase purity was established by x-ray powder diffraction using a Rigaku D/MAX-2000H rotating anode (12 kW) diffractometer. Data were collected at room temperature during routine scans between $10 \le 2\theta$ $\leq 80^{\circ}$ with a step size of 0.02° and a count time of 3 s/step, using Cu- $K\alpha$ radiation with a secondary graphite monochromator. For the indexing of these x-ray-diffraction patterns, longer count times of 8 s/step were used. No deviations of the crystal structure were found from that reported in the literature.¹⁶ Possibly, an impurity phase limited below the 2% level of the sensitivity limit of the diffractometer is present. dc magnetization and ac susceptibility measurements were performed with an Oxford Instruments MagLab EXA system in dc magnetic fields up to 7 T, in the temperature range 1.8 to 300 K. Powder neutron diffraction profiles for PbNi_{1.76}Mg_{0.24}V₂O₈ were measured in zero applied magnetic field at temperatures between 1.5 and 5 K on the DMC high-intensity diffractometer at the Swiss Spallation Neutron Source (SINQ), using neutrons with a wavelength of 2.567 Å. The 4-g powder sample was contained in a cylindrical Vcan ($\phi 5 \times 50 \text{ mm}^3$). An oscillating radial collimator suppressed Bragg peaks from the sample environment. For the purpose of chemical and magnetic structure determination, long-statistics scans were collected in the range $10 \le 2\theta$ $\leq 90^{\circ}$ (in steps of 0.2°) over a 24-h period at 2 and 4 K. In addition, shorter scans of 4 h each were measured on warming in order to determine the temperature evolution of the magnetic state. Rietveld refinements of the powder neutron diffraction patterns were performed with the FULLPROF (Ref. 17) suite of powder diffraction software, using the internal tables of neutron scattering lengths and magnetic form factors.

The μ^+ SR data in zero-field were collected on the GPS spectrometer of the μ^+ SR-dedicated π M3 beamline, on the 600-MeV proton accelerator at the Paul Scherrer Institute, Villigen, Switzerland. The instrument was equipped with a He-flow cryostat to access the temperature range down to 1.8 K. Pressed pellets of the powdered samples were attached with low-temperature varnish on an Ag sample-holder placed on the sample stick of the cryostat. When 100% spinpolarized positive muons are implanted in the solid sample, they come to rest at an interstitial lattice site, and then act as highly sensitive microscopic local magnetic probes. In the presence of a nonzero, static internal magnetic field the muon spin undergoes spontaneous Larmor precession with frequency $\nu_{\mu} = (\gamma_{\mu}/2\pi) H_{\text{int}}$, where $\gamma_{\mu}/2\pi = 13.553 \text{ kHz/G}$ is the muon gyromagnetic ratio. The μ SR technique in its variants (zero- or longitudinal-field) has proven extremely powerful in the field of small-moment magnetism, and in all cases, when magnetic order is of random, very short range, spatially inhomogeneous or incommensurate nature.¹⁸ For this reason we employed it in studying the low-temperature transition in the magnetically dilute PbNi1.76Mg0.24V2O8 compound.

III. RESULTS AND DISCUSSION

The present oxides are isomorphous to the alkaline-earth vanadate $\text{SrNi}_2\text{V}_2\text{O}_8$,¹⁶ and adopt a tetragonal crystal structure (a = b = 12.24 Å, c = 8.35 Å, space group $I4_1cd$) with arrays of NiO₆ octahedra forming screw chains along the c axis. The chains are separated by VO₄ tetrahedra and Pb²⁺ ions, resulting in an inherent quasi-1D arrangement of Ni²⁺ spins (S = 1) with an average intrachain Ni-spin separation of ~2.8 Å and the shortest interchain one of ~5.9 Å. These compounds are good *model* systems to examine the effect of spin-vacancies on inducing LRO magnetism in low-dimensional solids.

A. Macroscopic magnetic properties

The temperature dependence of dc magnetic susceptibility, $\chi_{dc}(T)$, measured at 0.1 T, for PbNi_{2-x}Mg_xV₂O₈ (x =0,0.24) is shown in Fig. 1. It is worth noting the features in the low-temperature behavior $(1.8 \le T \le 40 \text{ K})$ of the parent compound (x=0). The susceptibility could be fitted well to the expression for spin-gap compounds^{4,19}

$$\chi(T) = \chi(0) + \frac{C}{T} + \frac{a}{\sqrt{T}} \exp\left(-\frac{\Delta_{\text{eff}}}{kT}\right).$$
(1)

In addition to a *T*-independent term, $\chi(0) \approx 1.71$ $\times 10^{-3}$ emu/mol, $\chi_{\rm dc}(T)$ exhibits a minimum at ~8 K, while at lower temperatures it follows a Curie law with C= $3.66(2) \times 10^{-3}$ emu K/mol. This behavior has been observed for all samples studied to date, and has been attributed to the presence of unidentified paramagnetic impurities. The Curie moment corresponds to $\mu_{\rm eff} \approx 0.17 \mu_B$ per formula unit. We propose that the upturn in the low-T susceptibility is due to the presence of free $S = \frac{1}{2}$ spins at the ends of the chains terminating on the surface of the grains. Above 8 K, the susceptibility increases exponentially and exhibits a broad maximum around 120 K. The third term in Eq. (1) describes the bulk susceptibility due to the quasi-1D topology of the Ni²⁺ spins in the compound, where $a \approx 6.77$ $\times 10^{-2}$ emu K^{1/2}/mol is a constant factor and Δ_{eff} is a finite effective spin gap.^{4,19} A simple fitting of the susceptibility data between 1.8 and 40 K to Eq. (1) yields an estimate of the thermal activation energy $\Delta_{\rm eff}/k=45.3(1)$ K (~3.9 meV). This is roughly consistent with the two gaps for noninteracting chains, as derived from inelastic neutronscattering (INS) studies,²⁰ the transverse gap $\Delta_{\perp} = \langle \Delta \rangle$ -0.57D = 4.0(3) meV and the longitudinal one $\Delta_{\parallel} = \langle \Delta \rangle$ +1.41D=3.1(3) meV, where $\langle \Delta \rangle = 0.41J$ is the Haldane gap in the isotropic chain²¹ and D is the easy-axis (D < 0) anisotropy, estimated to be -0.45 meV for PbNi₂V₂O₈. Furthermore, the high-temperature part ($120 \le T \le 300$ K) of the measured $\chi_{dc}(T)$ curve was fitted to a high-temperature series expansion derived for a S=1 1D HAF.²² Having subtracted the Curie term from the data, the best fit was obtained for a nearest-neighbor intrachain exchange energy of J/k= -91.9(1) K (\sim 7.9 meV) and a Landé factor $g \approx 2.82$. The estimated in-chain coupling constant J differs by about 12% from that determined by INS measurements: J=9 meV.²⁰ From the susceptibility results on $PbNi_2V_2O_8$, we calculate a ratio $\Delta_{\rm eff}/|J| \sim 0.5$, with theory predicting that for isotropic S=1 HAF chains the ratio would reach an ideal value of 0.41,²¹ while deviations would imply important interactions between chains.

The susceptibility data of the x=0.24 Mg-doped sample are in sharp contrast to those of the parent compound. Between 100 and 300 K, $\chi_{dc}(T)$ obeys the Curie-Weiss law $\chi(T) = C/(T-\theta)$. A linear fit to the χ^{-1} vs *T* curve results in a Curie constant C=3.41(1) emu K/mol and a large Curie temperature $\theta = -313.2(1)$ K, indicating strong antiferromagnetic spin correlations. In this high-temperature range, we estimate $\mu_{eff} \approx 2.98 \mu_B / \text{Ni}^{2+}$, which is close to the expected value for an S=1 paramagnetic Ni complex. At low temperatures, a pronounced "peak" in $\chi_{dc}(T)$ appears at ~ 3.2 K, indicative of a magnetic phase transition. Both the peak position and the associated magnitude of the measured susceptibility, $\chi_{dc}(3.2 \text{ K}) \approx 48.1 \times 10^{-3} \text{ emu/mol}$, are in



FIG. 1. (a) Temperature dependence of the zero-field-cooled (ZFC) dc susceptibility $\chi(T)$ for PbNi₂V₂O₈ at H=0.1 T. (b) The ZFC evolution of $\chi(T)$ for PbNi_{1.76}Mg_{0.24}V₂O₈ under several magnetic fields (H=0.1, 1.1, and 4.0 T). Inset: χ_{dc} vs *T*, demonstrating marginal differences between ZFC and FC conditions. The solid lines for limited temperature ranges display the fit to the data with the functions described in Sec. III A.

quantitative agreement with previous results.14 Uchiyama and co-workers¹⁴ reported a phase diagram for $PbNi_{2-x}Mg_xV_2O_8$ in the T_N , x phase space, where the transition temperature rises rapidly at small x, then appears to display a maximum of ~ 3.5 K at a composition around x =0.16, while with further increase in x it decreases at a slow rate. These earlier dc susceptibility results indicate that while the "amplitude" of the peak scales with the concentration of the nonmagnetic Mg ions, its position depends on the Mgimpurity content in a nonmonotonic way. This enhancement may be attributed to the presence of free $S = \frac{1}{2}$ spins at the ends of the chains, broken by the substitutional Mg cations. In Secs. III B and III C, we discuss the nature of this phase transition with the aid of neutron diffraction and μ^+ SR experiments. It is found to be a paramagnetic-toantiferromagnetic phase transformation (vide infra).

The dc susceptibility of PbNi1.76Mg0.24V2O8 exhibits a considerable field dependence, and is shown in Fig. 1(b) at representative fields. We note that very little deviation is observed at $T < T_N$ between zero-field-cooled (ZFC) and fieldcooled measurements at relatively low-fields [inset of Fig. 1(b)]. At higher fields (e.g., $H \ge 1.1$ T), there are no differences at all and the characteristic AF maximum is suppressed while a response suggesting the presence of ferromagnetic correlations is observed. The field dependence of $\chi_{dc}(T)$ for $PbNi_{1.76}Mg_{0.24}V_2O_8$ points to a picture of an antiferromagnet with large magnetic anisotropy, which as demonstrated by the INS experiments of Zheludev et al.²⁰ is already present in the parent PbNi₂V₂O₈ compound in the form of a sizeable easy-axis single-ion anisotropy (D = -0.45 meV). Since D is mainly controlled by the spin-orbit interaction on the magnetic Ni sites, low dilution of the Ni sublattice by nonmagnetic Mg cations will not result in a substantial diminution of D. In systems of this type, when a field is applied parallel to the preferred axis of spin alignment (i.e., the c-axis, as seen by NPD), it tends to compete with internal exchange inter-



FIG. 2. Magnetization *M* as a function of the applied magnetic field *H* for various temperatures above and below T_N (3.2 K) in PbNi_{1.76}Mg_{0.24}V₂O₈. The "arrow" marks the *M*(*H*) "inflection" point corresponding to the critical field $[H_c(T)]$ for the transition to the metamagnetic phase. Inset: full hysteresis loop at T = 1.8 K.

actions. The result is that instead of causing the moments to flop perpendicular to the field (*spin-flop* phase²³), the antiferromagnet undergoes a first-order transition to a phase with a net magnetic moment. In the case of $PbNi_{1.76}Mg_{0.24}V_2O_8$, this is demonstrated in Fig. 2. The magnetization M as a function of the applied field H is shown for temperatures below and above the AF ordering temperature $T_N = 3.2$ K (under ZFC conditions). Well above T_N , the dc magnetization is linear with the applied magnetic field. For $T < T_N$, and at a critical field $H_c(T)$, one observes a significant deviation from linearity, as marked by an "inflection point" (Fig. 2). However, this is not abrupt, as it would be expected for a first-order transition, probably due to the polycrystalline form of our sample. In addition, the averaging out of the magnetization components, along the preferred axis of spin alignment and perpendicular to that, due to the powder nature of the measured samples results in a nonprominent hysteresis effect at low temperature (inset of Fig. 2, for 1.8 K); as the temperature increases this hysteresis becomes smaller and is absent for T=2.8 K. The results support a picture where, upon the application of a magnetic field at low temperatures, the material undergoes a transition from a state of low magnetization to one with relatively high magnetization. This field-induced transition is a common feature in other chemical systems, like FeCl2, exhibiting metamagnetic behavior.24

Additionally, the dc applied magnetic field dependence of the real component of the ac susceptibility (h_{ac} =1 Oe, f= 1 kHz) for the PbNi_{1.76}Mg_{0.24}V₂O₈ compound, at T< T_N , shows a characteristic "maximum" (Fig. 3). Its strong field dependence is a signature for the metamagnetic phase transition, while the position of the maximum indicates the critical field H_c for the transformation from the AF state to a ferromagnetic like state at each temperature. The obtained phase diagram is given as inset in Fig. 3. The *T* evolution of H_c demonstrates that the metamagnetic transition exists only below T_N . The transition field H_c , necessary for the reversal of the local spin directions along the chains, decreases with



FIG. 3. dc magnetic field (*H*) dependence of the real part of the ac susceptibility $[\chi'(H), \text{ at } h_{ac}=1 \text{ Oe}, f=1 \text{ kHz}]$ for PbNi_{1.76}Mg_{0.24}V₂O₈ at $T < T_N$. Inset: phase diagram of the same compound in the H-T plane; H_c is the critical field for the transition from the antiferromagnetic (AF) to the metamagnetic (MM) state. The line over the points is a phenomenological fit for extrapolation to the 0-K value of H_c .

increasing temperature and disappears at $T = T_N$. The critical line defined this way (inset of Fig. 3) can be extrapolated to T=0 K, reaching an upper value of $H_c = 1.7(1)$ T. The latter was extracted by a phenomenological power-law fit of the type $H_c(T) \sim H_c(0) [1 - (T/T_N)]^n$, with n = 0.68(5).

B. Powder neutron-diffraction measurements

The neutron powder diffraction pattern measured at 2 K is shown in Fig. 4(a). The chemical structure (space group $I4_1cd$) is described well by the crystallographic model published for $SrNi_2V_2O_8$.¹⁶ A few very weak extra peaks (e.g., at $2\theta \approx 16.2^{\circ}$, 36.6° , and 51.1°), which are not indexed in the given space group, may originate from little amount of an impurity phase or/and from a small deviation from the space group. However, they are temperature independent, and thus could not be associated with the magnetic transition identified by $\chi_{dc}(T)$. Important changes in the diffraction pattern were observed below $T_N = 3.2$ K, where the magnetic susceptibility $\chi(T)$ shows a prominent peak. The thorough investigation of the difference plot [Fig. 4(b)], between the 2and 4-K neutron-diffraction patterns, showed marked intensity changes. All the peaks in the difference plot can be indexed on the basis of the tetragonal chemical unit cell [a=b=12.2448(7) Å and c=8.3592(7) Å at 2 K], implying that the chemical and magnetic unit cells coincide (magnetic propagation vector $\kappa = \lceil 000 \rceil$). There is an increase in the intensity of reflections associated with nuclear Bragg peak positions [e.g., (220)] and the appearance of entirely new reflections, indexed as (011) and (031), which are extinct for the space group symmetry $(I4_1cd)$ describing the chemical structure. Assuming that the spins are located at the 16 Ni 16b sites, we analyzed all spin configurations and their orientation with respect to the crystal axes by Rietveld refinements of the difference pattern. The absence of certain magnetic Bragg reflections, like (100)/(010), (110), and (001), produces strong constraints on the possible magnetic



FIG. 4. Rietveld-refined neutron powder diffraction data ($\lambda = 2.567 \text{ Å}$) for PbNi_{1.76}Mg_{0.24}V₂O₈. Observed (points), calculated (full curve), and difference (bottom) profiles are shown. The upper (*N*) and lower (*M*) vertical lines mark the reflections due to chemical and the magnetic structures, respectively. (a) Full profile two-phase Rietveld refinement at 2 K. Inset: a schematic view of the unusual connectivity of the distorted NiO₆ octahedral units along one quasi-1D chain of the unit cell. (b) Purely magnetic neutron diffraction pattern obtained by subtraction of the 2-K profile from that at 4 K.

structures. For example, simple ferromagnetic spirals, antiferromagnetically coupled to each other would give large intensities at the (100) or (110) Bragg peak positions ($2\theta \approx 12.0^{\circ}, 17.0^{\circ}$). We have found that the spin configuration shown in Fig. 5(a) is the only one that describes the neutron data well. For this model, the calculated profile with the magnetic form factor for Ni²⁺ moment and the resulting difference curve from the observed data ($\chi^2 = 1.55$) are shown in Fig. 4(b). The 16 Ni atoms are arranged along four fourfold screw axes running along the *c* direction of the unit cell, resulting in an antiferromagnetic structure, with the spins lying only parallel or antiparallel to the *c* axis. That is, spins on the same chain (intrachain) and those at first nearestneighbor chains along the *a* and *b* axes (interchain) in each (00*z*)-plane are coupled antiferromagnetically, whereas



FIG. 5. The zero-field magnetic structure of PbNi_{1.76}Mg_{0.24}V₂O₈: (a) One chemical unit cell with the spin configuration of the Ni²⁺ ions at 2 K. (b) The temperature evolution of the antiferromagnetic (031)_M Bragg peak integrated intensity below $T_N = 3.2$ K. The line over the data points is a guide to the eye. Inset: the temperature dependence of the integrated neutron intensity for the (200)_N and (211)_N nuclear Bragg peaks.

those along the diagonal of the plane are coupled ferromagnetically.

Figure 4(a) shows the results of the Rietveld refinement of the NPD profile of $PbNi_{1.76}Mg_{0.24}V_2O_8$ at 2 K (R_{wp} = 5.4%, R_{exp} = 0.6%) with a two-phase model to account for both the chemical (space group: $I4_1cd$) and magnetic (space group: P1) structures. Table I presents a compilation of the refined crystallographic parameters ($R_{\text{Bragg}} = 4.1\%$) together with selected bond distances for $PbNi_{1.76}Mg_{0.24}V_2O_8$. In the inset of Fig. 4(a), the deformed rhombic connectivity of apical [i.e., short Ni-O(2) and Ni-O(3)] and equatorial Ni-O bonds (Table II) is shown for the distorted NiO_6 octahedra along the c axis of the unit cell. The medium resolution of the neutron data sets a limit for detailed refinement of the atom fractional coordinates and their site occupancies. We kept the latter constant to the values corresponding to the nominal stoichiometry of the compound. The atomic positions for Ni, Mg, and oxygen were refined, while those for V were not because of its small scattering length.

Diffraction patterns with inferior experimental statistics (4 h/temperature) were measured in the range $1.6 \le T \le 5$ K. The temperature evolution of the integrated intensity of the most intense (031) antiferromagnetic Bragg peak [Fig. 5(b)]

TABLE I. Structural parameters derived from Rietveld refinements of the 2-K PbNi_{1.76}Mg_{0.24}V₂O₈ neutron profile. *Note*: Vanadium fractional coordinates were not refined but kept constant to the position determined by earlier signal-crystal X-ray work on the isostructural SrNi₂V₂O₈ (Ref. 16). Space group: $I4_1cd$. a=b= 12.2448(7) Å and c=8.3592(7) Å.

Atom	Wyckoff position	x/a	y/b	z/c	Occupancy
Pb	8 <i>a</i>	0	0	0	0.5
Ni	16b	0.3262(9)	0.3303(7)	0.1714(14)	0.88
Mg	16b	0.3262(9)	0.3303(7)	0.1714(14)	0.12
V	16b	0.2596	0.0802	0.1015	1.0
O(1)	16b	0.155(1)	0.495(2)	-0.097(2)	1.0
O(2)	16b	0.338(2)	0.655(1)	0.434(2)	1.0
O(3)	16b	0.161(2)	0.689(1)	0.671(2)	1.0
O(4)	16b	0.330(1)	0.499(1)	0.199(2)	1.0

confirms that a LRO magnetic state starts to grow below a characteristic temperature $T_N = 3.2$ K.

In many systems, however, the large anisotropy in the interactions may promote canting, which leads to a weak ferromagnetic (FM) ground state. Therefore, we examined whether our NPD data could allow for any "rotation" of the local spin directions away from the "easy-axis" (c axis) spin configuration shown in Fig. 5(a). In the simplest case, if there is a FM component which is added to the main AF configuration, effectively leading to canting, it should give rise to some intensity contribution to the nuclear Bragg peaks. The inset of Fig. 5(b) shows the temperature dependence of the integrated neutron intensity for selected, (200) and (211), nuclear reflections. The lack of any temperature variation in their intensities, compared to that of the (031) AF reflection, and a Rietveld fit of the (2-4 K) difference pattern to the canted AF configuration, suggests that spin canting, if present, will account for less than $\sim 20\%$ of the AF moment.

From the best Rietveld fit to the 2-K data ($R_{\text{Mag}}=20\%$), we extracted the average size of the magnetic moment per Ni ion, $\langle \mu \rangle = 0.9(1)\mu_B$, in the AF spin configuration. It is worth noting that the ordered moment in PbNi_{1.76}Mg_{0.24}V₂O₈ displays a marked reduction from the spin-only ($\mu = gS$) value of 2.0 μ_B expected for S = 1 Ni²⁺

in antiferromagnetic or ferromagnetic substances. The unusual reduction of the experimentally deduced ordered moment poses certain questions about the microscopic mechanism responsible, which warrant further examination. Although low-dimensional "zero-point" spin fluctuations or nickel-oxygen covalency effects^{25,26} cannot be excluded as being responsible for moment reduction in this compound, we also need to examine this effect within the framework of dilution-induced ordering phenomena in quasi-1D HAF's.¹²

C. Muon-spin relaxation measurements

One may wonder how the magnetic structure shown in Fig. 5(a) can represent the real physical picture when we have drawn spins at all the Ni sites, whereas for the x=0.24 doping level we expect two out of the 16 Ni atoms in the unit cell to have been randomly replaced by Mg cations. Neither diffuse scattering nor peak broadening that would directly support spatial inhomogeneity in the moment size was evident in the neutron-diffraction experiments. The spatial inhomogeneity of ordered moments can be examined by local probe techniques such as muon-spin-relaxation $(\mu^+ SR)$ spectroscopy. $\mu^+ SR$ has a superb sensitivity to dilute and/or small magnetic moments,¹⁸ and for this reason we employed it in its zero-field (ZF) form to study PbNi_{1 76}Mg_{0 24}V₂O₈. In Fig. 6, we present the Fourier transform of the ZF- μ^+ SR time spectrum at 1.82 K, and compare it with that in the parent material. The spectral distribution in the two compounds displays important differences related to the nature of their associated ground states. The presence of "peak structure" (i.e., at $\nu_{\mu}{\sim}0.5,\,1.5,\,\text{and}~2.6$ MHz) in the Fourier transform of the doped compound suggests that the ZF- μ^+ SR signal is the result of the *convolution* of more than one *static* internal magnetic fields (H_{int}) at which the muon spin performs Larmor precession. No such features are evident in the Fourier spectrum of PbNi₂V₂O₈, in support of the disordered character of its nonmagnetic, dynamically modulated, spin-liquid ground state.

Earlier ZF- μ^+ SR work on the doping effects on the linear-chain spin-Peierls (Cu_{1-x}Zn_x)(Ge_{1-y}Si_y)O₃ compound²⁷ demonstrated that induced staggered moments around the Zn or Si defects lead to antiferromagnetic order with large spatial inhomogeneity in the order moment size. In lightly doped (x or y ≤ 0.02) spin-Peierls CuGeO₃ single crystals, the field distribution and muon-spin precession

TABLE II. Selected bond lengths (Å) and bond angles (deg) in $PbNi_{1.76}Mg_{0.24}V_2O_8$ at 2 K. The longer Ni-O(2) and Ni-O(3) bonds correspond to *apical* oxygen atoms along the NiO₆ chains, and the shorter bonds together with those of Ni-O(1) and Ni-O(4) to *equatorial* oxygens. The subscripts in the angles are associated with *lower* (*L*) or *higher* (*H*) *z*-coordinate Ni atoms, respectively.

$d_{ m Pb-O}/{ m \AA}$		$d_{ m Ni-O}/{ m \AA}$		Bond angles/deg	
Pb-O(1)	2.288(25)×2	Ni-O(1)	2.081(16)	Ni-O(2)-Ni	88.1(6)
	$3.464(19) \times 2$	Ni-O(2)	1.997(21)	Ni-O(3)-Ni	86.5(7)
Pb-O(2)	2.796(18)×2		2.080(19)	$O(3)-Ni_L-O(2)$	90.7(1.2)
Pb-O(3)	$3.107(18) \times 2$	Ni-O(3)	2.042(24)	$O(3)-Ni_{H}-O(2)$	94.7(1.4)
Pb-O(4)	2.661(14)×2		2.096(22)		
	3.267(14)×2	Ni-O(4)	2.089(21)		



FIG. 6. (a) The time evolution of the zerofield μ^+ -spin polarization $P_{\mu}(t)$ for PbNi_{2-x}Mg_xV₂O₈ (x=0 and 0.24) at 1.82 K. The lines over the data points are guides to the eye. (b) The Fourier transform of the corresponding μ^+ SR time spectra.

in zero field arises from a single muon site, associated with a sizable internal field of $H_{int} \sim 500$ G. On the other hand, in the ZF- μ^+ SR experiments of polycrystalline $PbNi_{1.76}Mg_{0.24}V_2O_8$, the Fourier analysis resolves a number of intrinsically broad peaks, reflecting a nonhomogeneous internal field distribution, which is composed of a number of static local fields: $H_{int} \sim 36-192$ G. We note that the muon local field is proportional to the size of the nearby moments; $H_{int}(z) \sim \mu(z)$. Therefore, multiple local fields in PbNi_{1.76}Mg_{0.24}V₂O₈ indicate spatial inhomogeneity in the order parameter (i.e., the staggered moment) of the antiferromagnetic ground state, that is a nonuniform ordered moment size along the NiO₆ helical chains. Crystallographically nonequivalent muon sites, associated with the unusual topology of the Ni²⁺ ions in the 3D lattice, can give rise to such a complex internal field distribution.

IV. CONCLUSIONS

In the quasi-one-dimensional PbNi₂V₂O₈ system, substitution of the magnetic Ni²⁺ (*S*=1) sites by nonmagnetic Mg²⁺ (*S*=0) cations leads to a transition from a magnetically disordered spin-gap ground state to essentially threedimensional Néel order. We have investigated this effect for one Mg-substituted material, namely, PbNi_{1.76}Mg_{0.24}V₂O₈, located in the "heavily doped" region of the T_N , *x* phase diagram. Neutron powder diffraction and muon-spin relaxation measurements were employed in the low-temperature range where the ac and dc susceptibility data illustrated that a magnetic transition takes place. Neutron powder diffraction in PbNi_{1.76}Mg_{0.24}V₂O₈ probes an average size of the ordered

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moment of $0.9(1)\mu_B$ per Ni site at 2 K. The spins in the Ni sublattice lie either parallel or antiparallel to the c axis, which is the magnetic easy axis of the system. The spin-spin interaction is *antiferromagnetic* (below $T_N = 3.2$ K) both in the NiO₆-chains running along the c axis (*intrachain*) and within the ab plane for nearest-neighbor Ni atoms (interchain). Zero-field muon-spin relaxation below T_N confirms the static character of the magnetic interactions arising from the doping-induced, end-of-chain local moments. Also, it provides evidence of a nonuniform moment size over the severed chain segments. It manifests as a convolution of *multiple* spontaneous muon-precession *frequencies* that give rise to an internal magnetic field distribution with spatial inhomogeneities. The magnetic susceptibility shows that the compound exhibits metamagnetic behavior at $T < T_N$, supporting a picture of an antiferromagnet with significant magnetic anisotropy and competing intrachain and interchain exchange interactions.

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