

Electronic structure and optical properties of ThPd₃ and UPd₃

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We present calculations of the electronic and optical properties of the actinide compounds ThPd₃ and UPd₃ using the state-of-the-art full-potential linearized augmented-plane-wave method as implemented in the WIEN97 code. Both compounds crystallize in the complex TiNi₃ structure with 16 atoms per unit cell. For comparison with earlier work, we also studied these compounds in the AuCu₃ structure. However, we find that the results in this hypothetical structure do not compare that well with experiments as in the actual TiNi₃ structure. We have calculated the density of states (DOS), the coefficient of the electronic specific heat, and the frequency-dependent optical conductivity. These quantities are compared with experiments where good agreement is obtained. The DOS is dominated by Pd *d* states below the Fermi energy and by actinide *f* states near and above the Fermi energy. Transitions between these states dominate the optical conductivity and we find that the optical matrix elements do not play an important role. This finding is supported by experiments on a comparison of x-ray photoemission spectroscopy (XPS) data with optical conductivity.

I. INTRODUCTION

The existence of *5f* electrons in actinide compounds has given rise to the seemingly perpetual question of whether these states are localized, intermediate-valent, or itinerant. These compounds are particularly interesting as they bridge the gap between the itinerant *3d* electrons in transition metals and the localized *4f* electrons in rare earths. Much work has already been done in order to understand the behavior of *5f* electrons in actinide compounds.¹ Hence there exists a general interest in *5f* systems from the experimental as well as the theoretical side. UPd₃ with the electronic configuration of *5f*³*6d*¹*7s*² has been a subject of several experimental studies using different techniques. Some of these are resistivity studies,² electron-spin-resonance (ESR) measurements,³ neutron scattering,⁴ susceptibility measurements,^{5,6} low-temperature specific-heat measurements,^{6,7} photoemission,⁸ optical measurements,⁹ and Fermi-surface measurements.¹⁰ Most of these studies agree on an *f*² configuration of a U⁴⁺ ion with complicated phase transitions at very low temperatures.^{11,12} Hence, the literature on UPd₃ contains a wealth of experimental data. However, many fewer theoretical studies have been published. Earlier theoretical studies on UPd₃ mainly consist of electronic-structure calculations by Norman *et al.*¹³ and by Eriksson *et al.*¹⁴ Norman *et al.*¹³ have performed calculations for the experimentally observed TiNi₃ (DO₂₄) structure of UPd₃ using the linear muffin-tin-orbital (LMTO) method in the atomic-sphere approximation (ASA), where the *5f*² electrons have been treated as core states. They have calculated the DOS and the Fermi surface. Their DOS compares well with experiments except for the peak position corresponding to the localized *f* level. They have presented the DOS in arbitrary units and did

not deduce the electronic specific-heat coefficient. Their Fermi surface of UPd₃ shows reasonable agreement with experimental data but has some additional undesirable features. They find that it is necessary to shift bands in order to get quantitative agreement with experiment. Other attempts to improve the calculations such as inclusion of *l*=3 basis functions, use of an exchange-only functional, and addition of combined correction terms had little effect. Naturally, the discrepancy is thus suspected to be due to inadequacies of the local-density approximation (LDA) in correctly placing the *d* bands with respect to the *s-p* bands or, more likely, due to the ASA. These authors have also studied UPd₃ earlier¹⁵ in the AuCu₃ structure as part of an investigation of the electronic structure and properties of UX₃ (*X*=Ru, Rh, Ir, Pt) compounds. The other work¹⁴ has focused on the electronic structure of ARH₃ (*A*=actinide) systems and the *5f* localization in UPd₃ using both the scalar-relativistic as well as the fully relativistic LMTO-ASA method. In order to understand the effect of ligand states on the nature of *5f* electrons in U, they calculated the electronic structure for UM₃ (*M*=Mo to Ag) compounds. All of these materials, except UPd₃, crystallize in AuCu₃ structure, hence UPd₃ was also studied in this space group to facilitate that kind of comparison. They¹⁴ obtained a much smaller theoretical equilibrium volume for UPd₃ than the experimental value and found UPd₃ to be stable in a ferromagnetic state even at highly compressed volumes. Experimentally, magnetic order has been found below 4.5 K,^{11,12} where the magnetic moment was reported to be extremely small.

In this work, we present results on electronic structure and optical properties of ThPd₃ and UPd₃, two actinide-transition-metal intermetallic compounds. We have used the state-of-the-art full-potential linearized augmented-plane-

TABLE I. Atomic positions in units of lattice parameter a in internal coordinates for XPd_3 compounds ($X=U, Th$) in $TiNi_3$ structure.

Atom	Wyckoff position	Position in the unit cell
X1	$2a$	0.0 0.0 0.0
X2	$2d$	$\frac{1}{3}$ $\frac{2}{3}$ $\frac{1}{4}$
Pd1	$6g$	$\frac{1}{2}$ 0.0 0.0
Pd2	$6h$	$\frac{5}{6}$ $\frac{2}{3}$ $\frac{1}{4}$

wave (FP-LAPW) method¹⁶ including spin-orbit interaction with the generalized gradient approximation (GGA) and thus avoid shortcomings as present in the previous ASA-based LMTO calculations. Our results are compared to experimental data on specific heat,^{6,7} XPS,^{8,17} as well as optical conductivity.^{9,18}

II. METHOD AND CRYSTAL STRUCTURES

The calculations are performed using the FP-LAPW method as is implemented in the WIEN97 code,¹⁹ including local orbitals for the high-lying ‘‘semicore’’ states. Exchange and correlation are treated in the generalized gradient approximation (GGA) within density-functional theory²⁰ using the parametrization of Perdew *et al.*²¹ Core states are treated fully relativistically,²² while for the valence states, spin-orbit interaction is added in a second variational step using the scalar-relativistic orbitals as a basis.²³ The DOS has been calculated using the modified tetrahedron method of Bloechl *et al.*²⁴ The frequency-dependent optical properties are obtained using the joint density of states (JDOS) weighted by the dipole matrix elements²⁵ of the optical transitions. We also include intraband transitions using a lifetime broadening of 0.01 eV. The $TiNi_3$ structure is hexagonal and thus the optical conductivity has two components, σ_{xx} and σ_{zz} . Since the experimental data for these compounds have been measured on polycrystalline samples, we present also an average over the two components, $(2\sigma_{xx} + \sigma_{zz})/3$. UPd_3 and $ThPd_3$ crystallize in hexagonal $TiNi_3$ (DO_{24}) structure with 16 atoms per unit cell.²⁶ We use the experimental lattice constants²⁶ $a = 11.065$ a.u. and $c = 18.568$ a.u. for $ThPd_3$ and $a = 10.879$ a.u. and $c = 18.181$ a.u. for UPd_3 . These lattice constants are almost the same as the latest.²⁷ In the unit cell of XPd_3 , we have two nonequivalent X atoms and two kinds of Pd sites (see Table I).

In order to study the influence of the crystalline environment, and for comparison with earlier work,^{14,15} we also performed calculations for $ThPd_3$ and UPd_3 in the hypothetical $AuCu_3$ crystal structure. In these calculations, we adopted the same volume per formula unit resulting in lattice constants of 7.895 a.u. for $ThPd_3$ and 7.752 a.u. for UPd_3 . These calculations have been converged with 84 k points in the irreducible ($\frac{1}{48}$ th) Brillouin zone, whereas we used 76 k points in the irreducible ($\frac{1}{24}$ th) Brillouin zone of the $TiNi_3$ structure. In the $AuCu_3$ structure, each atom X ($X=Th, U$) has 12 equivalent nearest Pd neighbors, while in the $TiNi_3$ structure only six Pd atoms surround the X atom and there are two different kinds of X and Pd atoms. Therefore, we can expect different properties of the compounds in the two crystal structures.

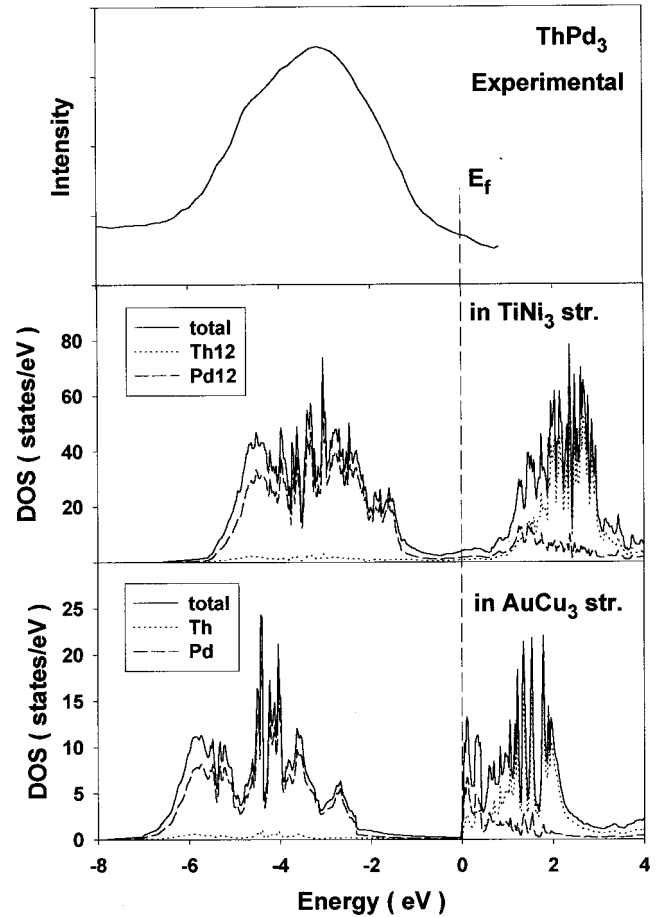


FIG. 1. The XPS data for $ThPd_3$ (top) in arbitrary units, calculated total and partial DOS for $ThPd_3$ in $TiNi_3$ structure (middle), and in $AuCu_3$ structure (bottom). The E_f is set to 0.0 eV.

III. RESULTS AND DISCUSSION

A. $ThPd_3$ in the $TiNi_3$ structure

The total DOS and partial DOS for $ThPd_3$ are displayed in Fig. 1 along with the measured XPS spectrum.¹⁷ The DOS of $ThPd_3$ is dominated by contributions from Thf and $Pd d$ states, and only these contributions are displayed together with the total DOS in Fig. 1. The DOS for the Pd1 and Pd2 sites is not much different and thus only the sum is shown. These states dominate from 5.5 to 1.5 eV below the Fermi energy (E_f) but there are also some Pd d contributions to the unoccupied DOS up to 2 eV above E_f , which has also significant Th d contributions. The main peak at about 3 eV above E_f comes of course from Thf states, which are completely empty as expected from the electronic configuration of Th. The two Th sites have slightly different charges and thus the main $5f$ peaks between 2 and 3 eV above E_f have more Th1 contributions at higher energies. In the experimental XPS data,¹⁷ we find a structure at 2 eV, the main peak at 3 eV, and a shoulder around 4.5 eV below the Fermi level. These features may correspond to peaks around 1.8, 3, and 4.5 eV, respectively, below E_f in the theoretical DOS in reasonably good agreement with the experimental data. The total DOS at the E_f is 3.95 states/eV cell. The corresponding coefficient of the electronic specific heat is 2.33 mJ/mol K^2 compared to the experimental value⁶ of 1.5 mJ/mol K^2 . The

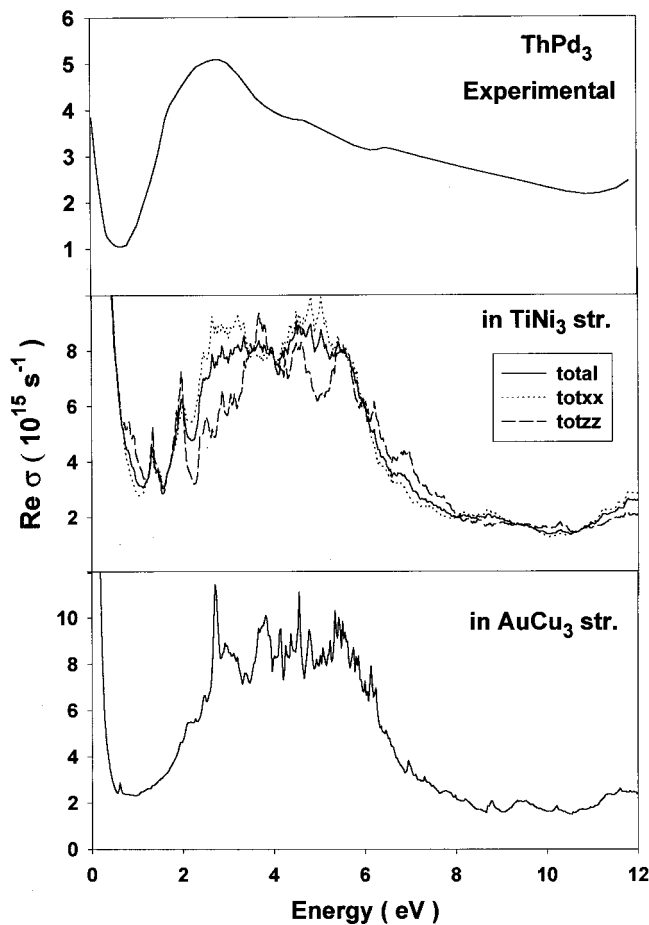


FIG. 2. Real part of the frequency-dependent optical conductivity for ThPd_3 : experimental curve (top), calculated [in TiNi_3 structure (middle) and in AuCu_3 structure (bottom)].

theoretical value is larger than the experimental one and this seems to suggest that localization effects play a significant role here.

The total DOS concentrates in two main regions: one between 1.5 and 5.5 eV below E_f due to Pd d states, and another one between 1 and 3 eV above E_f due to Th f states. Transitions between these regions yield the main contribution to the frequency-dependent optical conductivity. The real part of the calculated frequency-dependent optical conductivity, σ , for ThPd_3 is displayed in Fig. 2 along with the experimental curve.¹⁸ Our curve is too peaky as compared to the experimental curve as we have included very small lifetime broadening. The theoretical curve starts with a sharp drop and a dip at about 1 eV followed by a steep rise and a structure at about 2 eV culminating into the main peaks, which are broad and centered around 3 and 5 eV. Then it falls sharply with a shoulder at about 6.6 eV. After this there is a steady decrease and a dip at about 10 eV and a rise thereafter. The optical conductivity has a noticeable anisotropy and σ_{xx} shows a more pronounced double-peak structure (at 3 and 5 eV), while σ_{zz} has its main peak around 3.8 eV. The general features of the theoretical curve are similar to the experimental spectrum,¹⁸ but our main feature is broader. The dip at 10 eV and the upturn for higher energies is present in both. To study the effect of the dipole matrix elements for the optical conductivity, we also calculated the

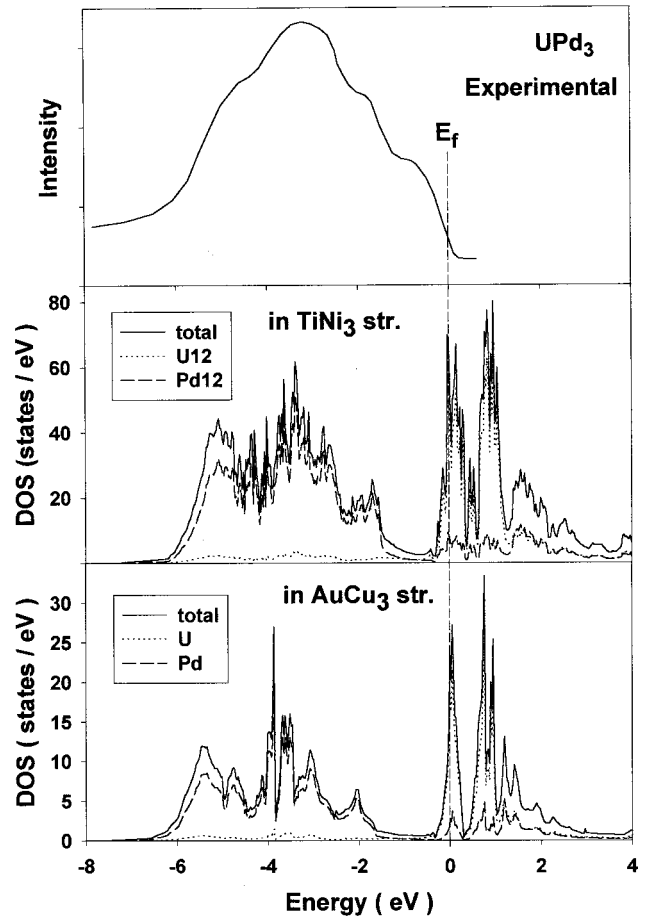


FIG. 3. The XPS data for UPd_3 (top) in arbitrary units, calculated total and partial DOS for UPd_3 in TiNi_3 structure (middle), and in AuCu_3 structure (bottom). The E_f is set to 0.0 eV.

JDOS. Interestingly, the JDOS is quite similar to the optical conductivity (hence we choose not to show the JDOS), indicating that for this compound the selection rules do not play an important role. This is in agreement with the observation of Schoenes and Andres from a comparison of their optical conductivity data^{9,18} with the XPS data^{8,17} that the matrix elements for the transitions play a minor role in determining the spectra of these compounds. Hence there is an overall fairly good qualitative as well as quantitative agreement with the experimental data for the calculated DOS, coefficient of electronic specific heat, and optical conductivity of ThPd_3 .

B. UPd_3 in the TiNi_3 structure

Next we discuss our results for UPd_3 . The calculated total DOS and the partial DOS together with the experimental XPS data⁸ are illustrated in Fig. 3. As in ThPd_3 , the main contributions below E_f come from Pd d states, whereas the DOS at and above E_f is dominated by U f contributions. Both Pd1 and Pd2 DOS are similar and overlap, hence only the sum is displayed in the figure. For UPd_3 , the f DOS for the U1 and U2 sites are quite similar; however, in contrast to ThPd_3 , these exhibit a much more pronounced splitting into U $f_{5/2}$ and $f_{7/2}$ states of the main peak above E_f observed in the total DOS. The DOS between 1.5 and 6 eV below E_f has mainly Pd d character, whereas the large peaks around E_f

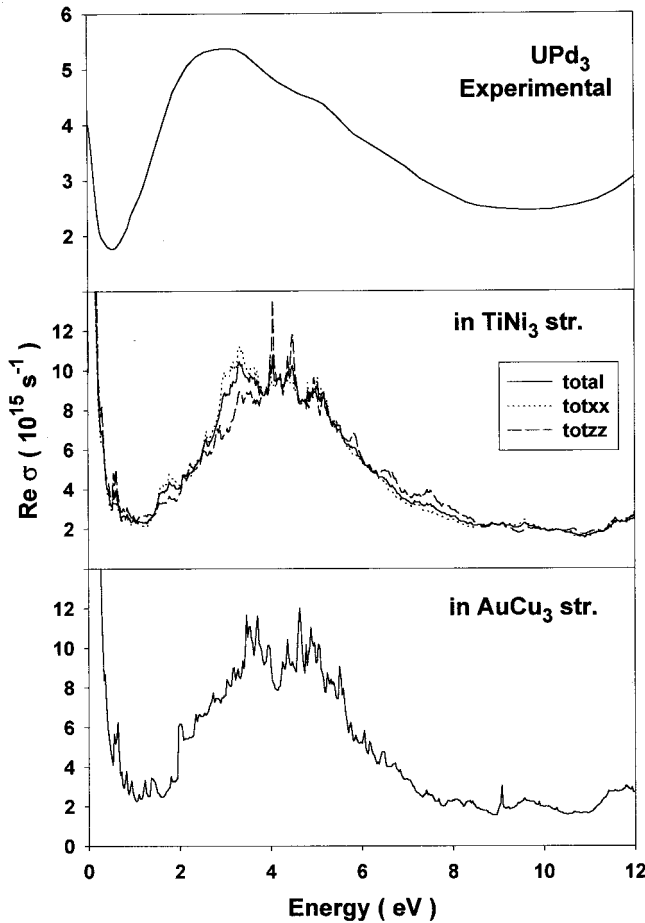


FIG. 4. Real part of the frequency-dependent optical conductivity for UPd₃: experimental curve (top), calculated [in TiNi₃ structure (middle) and in AuCu₃ structure (bottom)].

originate mainly from Uf states, indicating partially filled Uf states. The smaller peaks beyond 1.6 eV above E_f are mainly from Ud states with a strong admixture of $Pd d$. Comparing the total DOS with the experimental XPS spectrum,⁸ we find that the calculations can reproduce most of the experimental features showing structures at 1.0, 2.0, 2.9, 4.6, and 5.0 eV below E_f . The only exception is that the theoretical DOS (Fig. 3) has a strong peak at E_f but no feature around 1 eV below E_f , which is also absent in the work of Norman *et al.*¹³ They obtained a localized Uf peak at 2.5 eV below E_f , which does not show up in the experimental data. As expected, for density-functional theory (DFT) calculations we find all the Uf states around E_f . The total DOS at E_f is 68.2 states/eV cell (with a large uncertainty due to the spiky DOS structure and the limited k mesh in the DOS calculation), which yields a γ value of 40.3 mJ/mol K² for UPd₃. The specific heat for UPd₃ has been measured by Andres *et al.*⁶ and Zaplinski *et al.*⁷ Although it is difficult to separate the linear part from the total specific heat because of phase transitions near 7 K, the γ was estimated to be between 4 and 10 mJ/mol K². Our value is, as in ThPd₃, much larger than this experimental estimate and indicates that the $5f$ states cannot be described properly by an itinerant (delocalized) picture.

The real part of the theoretical σ of UPd₃ in TiNi₃ structure is presented in Fig. 4 together with the experimental⁹

curve. In the calculated σ , we find structures starting at around 2 eV, a broad main peak ranging from 3 to 5 eV, followed by a shoulder above 7 eV and a broad dip ranging from 8 to 11 eV. The general shape of the experimental spectrum (top of Fig. 4) agrees quite well with the theoretical curve. The major difference is that the theoretical main peaks are broader and shifted to slightly higher energies compared to experiment, and that the experimental feature at 5 eV is much weaker than in theory. The experimental XPS spectra⁸ and the optical conductivity⁹ of UPd₃ have a remarkable similarity, which suggests that the final states of the optical transition must be very close to E_f and also that the dipole matrix elements for transition are not playing a significant role. These observations are in accordance with our findings that the JDOS is quite similar to the optical conductivity (which includes these matrix elements). Similar to ThPd₃, we conclude that the transitions from $Pd d$ states to Uf states govern the optical conductivity of UPd₃.

Comparing our results for ThPd₃ with those for UPd₃, we find that the $5f$ states are unoccupied in ThPd₃ and are well above E_f . This results in a much smaller DOS at E_f in agreement with experiment. For UPd₃, the $U 5f$ states are right at E_f , resulting in a large γ value. The occupied $Pd d$ DOS is slightly different for the two compounds and in particular the unoccupied actinide $d/Pd d$ band is below the dominant $5f$ peak in ThPd₃ while it is above it in UPd₃.

C. ThPd₃ and UPd₃ in the hypothetical AuCu₃ structure

Since most of the UM_3 compounds crystallize in the AuCu₃ structure, we have also investigated ThPd₃ and UPd₃ in this structure. Total-energy calculations verify, however, that the TiNi₃ structure is more stable by 4.7 (14.6) mRy per formula unit for ThPd₃ (UPd₃) in agreement with experiment. Nevertheless, we discuss here the calculations in the hypothetical AuCu₃ structure to check the effect of the crystal structure on the DOS and optical conductivity. The DOS for ThPd₃ and UPd₃ in AuCu₃ structure is shown in the lower panel of Figs. 1 and 3, respectively. A comparison with the calculated DOS in TiNi₃ structure shows that there are some differences in details of the DOS curves, but of course the general features of occupied $Pd d$ states and (partly) unoccupied f states remain the same. The $Pd d$ DOS below E_f is quite similar for both compounds in the AuCu₃ structure but is different from the corresponding DOS in TiNi₃ structure, implying that the $Pd d$ states are affected differently in the two crystal structures. In addition, in AuCu₃ structure, they are shifted down in energy by about 1 eV, in disagreement with experimental XPS data. The largest difference is in the position of the Fermi level in ThPd₃, which is right at the beginning of the large peaks above E_f in AuCu₃ structure while it is 1 eV below those peaks in TiNi₃ structure. Comparing our DOS for UPd₃ in AuCu₃ structure with that from Eriksson *et al.*,¹⁴ we find that there is a good qualitative agreement in the shape of the two DOS curves, but their DOS is spread over a larger energy region resulting in peak positions that are shifted away from E_f , both above and below E_f . We find the total DOS at E_f in ThPd₃ and UPd₃ in the AuCu₃ structure as 2.10 and 21.2 states/eV cell, respectively. (Note that here the unit cell contains only one formula unit compared to four in the case of TiNi₃ structure.) The

corresponding γ values are 4.96 and 49.9 mJ/mol K², respectively, which are even larger than those in TiNi₃ structure.

The frequency-dependent optical conductivity of ThPd₃ and UPd₃ in the hypothetical AuCu₃ structure is shown in the lower panels of the Figs. 2 and 4, respectively. The calculated σ of ThPd₃ and UPd₃ is not very sensitive to the crystal structure, but the main peaks are even broader. In summary, we conclude that the hypothetical AuCu₃ structure does not represent faithfully the properties of these two compounds compared to the actual TiNi₃ structure. As expected, also in this structure the transitions from the Pd *d* states to the actinide *f* states dominate the optical conductivity, and σ with and without matrix elements turn out to be similar.

IV. CONCLUSIONS

In this work, the compounds ThPd₃ and UPd₃ have been studied in their experimentally verified TiNi₃ structure. The FP-LAPW method as implemented in the WIEN97 code has been used for these calculations. The ground state has been studied and very encouraging agreement with the experimental data for both compounds has been obtained for the DOS and the frequency-dependent optical conductivity. Our calculations show that the 5*f* bands in UPd₃ are partly occupied while those in ThPd₃ are completely empty. The DOS beyond 2 eV below E_f essentially arises due to Pd *d* states and is similar in ThPd₃ and UPd₃. The same is also reflected in the optical conductivity curves as the σ curves for the two compounds are similar in parallel with experiments. Most of the Pd *d* DOS is confined to the region below E_f and supports the conclusion of Schoenes and Andres⁹ that the Pd *d* states are filled when Pd is alloyed with Th or U.

The optical conductivity shows only little changes when matrix elements are excluded. The same interesting feature is

observed by comparing XPS data and experimental optical conductivity of these compounds.

On the other hand, we note that the experimental sigma of UPd₃ is broader than that of ThPd₃, but this is reversed in the calculation. Also, the coefficients of the electronic specific heat γ are much too large in these calculations compared to the experimental data, indicating that the 5*f* electrons are more likely localized and an itinerant description is not very suitable. This may be compared to a previous study²⁸ of the optical conductivity of the uranium monochalcogenides, of which UTe has nearly localized 5*f* electrons, where a number of features in the conductivity are explained but not all, substantiating the more localized nature of the 5*f* electrons.

We also studied these two compounds in the hypothetical AuCu₃ structure, as has been done for UPd₃ in earlier work,^{14,15} to ascertain if the properties are affected significantly by crystal structure. Our results for UPd₃ in the AuCu₃ structure agree with the previous calculations. Although we do find an appreciable difference with the results from the actual TiNi₃ structure, we notice that this hypothetical structure does not yield as good an agreement for the DOS and frequency-dependent σ with the experimental data as the experimentally established TiNi₃ structure does. Hence the hypothetical simple AuCu₃ structure is not suitable for studying the properties of these compounds.

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