# Magnetic properties of palladium-graphite multilayers

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dc magnetization and ac magnetic susceptibility of Pd graphite based on natural graphite have been measured using a superconducting quantum interference device magnetometer. In Pd-*G*, Pd multilayers are sandwiched between adjacent graphite layers. This sandwich structure is periodically stacked along the *c* axis. Pd layers are formed of Pd nanoparticles in the form of either a one-dimensional chain or two-dimensional platelets. There are two kinds of magnetic behavior depending on sample. Pd-*G*(1) is an itinerant spin system with an antiferromagnetic ordered phase below  $T_N$ (=3.8 K). No spin-glass-like behavior occurs around  $T_N$ . Pd-*G*(2) is a quasi-two-dimensional *XY*-like ferromagnet with very weak antiferromagnetic interplanar interaction. It undergoes two magnetic phase transitions at  $T_{cu}$ (=14.2±0.2 K) and  $T_{cl}$ (=11 K). Between  $T_{cu}$  and  $T_{cl}$ , a two-dimensional ferromagnetic spin order is established within nanoparticles in the Pd layers. Below  $T_{cl}$ 

### I. INTRODUCTION

Metal graphites have received considerable attention partly because of physical and chemical properties related to the low dimensional nature of metal atoms encapsulated in the interlamellar space between graphite layers.<sup>1–7</sup> A metal graphite has a unique layered structure. Ideally, metal atoms (such as V, Cr, Mn, Fe, Co, Ni, Cu, Mo, Pd, Ta, and W) form a metal monolayer sandwiched between adjacent graphite layers. Such sandwich structures are periodically stacked along the c axis perpendicular to the graphite layer. In reality, the observed bodies are likely to consist of metal multilayers, but not a single layer, since sample preparation of the metal graphite is carried out at high temperatures where surface diffusion becomes significant. Each metal layer is formed of nanoparticles in the form of either a onedimensional (1D) chain or two-dimensional (2D) platelets.<sup>4,5</sup> Unlike graphite intercalation compounds (GIC's) no charge transfer occurs between metal monolayers and graphite layers. The interplanar interaction between metal and graphite layers in metal graphite is considered to be of the van der Waals type and is much weaker than that in GIC's because of the absence of attractive Coulomb interactions.

Pristine Pd metal has a face-centered-cubic (fcc) structure with a lattice constant (a = 3.8898 Å). It has an almost filled narrow d band and a partly filled s band, which can be designated formally as  $(4d)^{10-\xi}(5s)^{\xi}$  ( $\xi \approx 0$ ). Pd has the largest Pauli susceptibility with a Stoner factor (s = 9.37).<sup>8</sup> This enhancement brings Pd metal close to a ferromagnetic instability. However, Pd is still paramagnetic because the Stoner criterion for the occurrence of ferromagnetism is not satisfied. This means that the Fermi energy of Pd is not situated at an energy where the density of states has a peak. Such a situation may change in Pd graphite (Pd-G) where Pd atoms form multilayers in the limited space. Theoretically Bouarab *et al.*<sup>9</sup> have studied how the positional relation between the peak energy in the density of states and the Fermi energy changes with the number of Pd metal layers. They have predicted that the system is ferromagnetic for two to five layers, and is paramagnetic for systems with a monolayer or more than five layers. Thus it is expected that Pd-*G* may exhibit a quasi-2D ferromagnetic behavior.

In this paper we report our experimental dc and ac magnetic susceptibility results of Pd-G based on natural graphite (NG) using a superconducting quantum interference device (SQUID) magnetometer. The static and dynamic properties of spin fluctuations are extensively studied. We show that Pd-G shows magnetic phase transitions around either 3.8 K [for convenience this sample is denoted as Pd-G(1)] or 14 K [Pd-G(2)], depending on the sample. An irreversible effect of magnetization is observed, suggesting the frustrated nature of granular systems. The magnetic properties of Pd-G(2) will be discussed in comparison with those of stage-2 NiCl<sub>2</sub> GIC,<sup>10</sup> where the ferromagnetic layers are antiferromagnetically stacked along the c axis. Here we note that the SQUID dc magnetization of Pd-G(2) was already measured by Mendoza et al.<sup>1</sup> The zero-field-cooled magnetization  $(M_{ZFC})$  at an external magnetic field H (= 10 Oe)shows a small peak around 14 K in addition to a broad maximum above 50 K which is typically observed in pristine Pd metal.

## II. BACKGROUND: STRUCTURE OF Pd-G

PdCl<sub>2</sub> GIC belongs to an acceptor type GIC, where charge transfer occurs from graphite layer to intercalate layer.<sup>11</sup> In the Raman spectra of PdCl<sub>2</sub> GIC the graphite signal is shifted up in wave number relative to that of pristine graphite.<sup>6</sup> The synthesis of Pd-*G* can be made from the reduction of PdCl<sub>2</sub> GIC as a precursor material. The reduction can be carried out (i) in liquid Li diphenlide at room temperature<sup>6</sup> or (ii) in the gaseous phase by contact with hydrogen gas at high temperature (350–400 °C).<sup>5,7</sup> The structural properties of Pd-*G* are different for these two methods of reduction. In Pd-*G* pre-

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pared by method (i), there is charge transfer between C and Pd.<sup>6</sup> In fact, the Raman spectrum is similar but not identical to that of PdCl<sub>2</sub> GIC precursor. The Pd nanoparticles form mainly monolayers. The size of the Pd nanoparticles is relatively small. The in-plane structure of Pd nanoparticles is commensurate with the graphite lattice, forming a  $p(3 \times 3)$  superlattice.<sup>6</sup>

In Pd-*G* prepared by method (ii),<sup>7</sup> Pd nanoparticles form mainly multilayer systems with an average thickness of trilayers. The size of Pd nanoparticles is larger than that of Pd nanoparticles prepared by method (i). The Raman spectrum is similar to that of graphite, suggesting that no charge transfer occurs between C and Pd. The in-plane structure of nanoparticles is almost commensurate with that of the graphite host, forming either a  $p(2 \times 2)$  or a  $p(3 \times 3)$  within a mismatch of  $\approx 4\%$  in lattice constant.<sup>7</sup> Note that such a mismatch of  $\approx 4\%$  is quite often observed between bulk material and nanoparticles of the same material.

#### **III. EXPERIMENTAL PROCEDURE**

In the present work we used two kinds of Pd-G samples [denoted as Pd-G(1) and Pd-G(2)]. Both samples were prepared from hydrogen reduction of PdCl<sub>2</sub> GIC based on natural graphite flakes (grade: RFL 99.9 S) from Kropfmuhl, Germany. Pd-G(1) was reduced for 2 h at 350 °C and Pd-G(2) for 2 days at 400 °C. The structure was studied by transmission electron microscope (TEM)/selected area electron diffraction (SAED) for Pd-G(1) and Pd-G(2) and additionally by x-ray photoelectron spectroscopy (XPS) for Pd-G(2). Bright field transmission electron microscope (TEM) images showed encapsulated nanoparticles. The corresponding SAED pattern from areas occupied by particles and empty regions showed that the reduction was complete. No reflections from the PdCl<sub>2</sub> GIC precursor could be obtained. On the other hand all SAED patterns showed reflections from carbon and additional reflections from palladium, if nanoparticles were examined. These Pd particles showed hexagonal and hcp reflections<sup>6,7</sup> which are quite distinguishable from the common fcc Pd. Commensurate superstructures to graphite were observed.<sup>6,7</sup> A comparison of XPS core level spectra from PdCl<sub>2</sub> GIC precursor and Pd-G showed that the reduction was complete. Diffraction pattern as well as the XPS signal gave no hint for Pd-hydride formation. The majority of the Pd particles had oval shapes. Their sizes ranged (measured on the long axis of the oval shaped particles) from 20 to 2000 Å. The average size was  $530\pm340$  Å. Nanoparticles remain immobile because of their relatively large sizes and high masses. They are not able to diffuse from the carbon lattice. So far no determination has been made for the composition of Pd-G samples. The stoichiometry of stage-3 PdCl<sub>2</sub> GIC as a starting material is known as C14.8PdCl2.12 If only Cl atoms leave as HCl during the reduction, the composition of Pd-G is considered to be described by  $C_n Pd$  with n = 14.8. The dc magnetization and ac magnetic susceptibility of Pd-G were measured using a SQUID magnetometer (Quantum Design, MPMS XL-5) with an ultra-low-field capability option.

(i) *SQUID dc magnetization*. Before setting up a sample at 298 K, a remanent magnetic field in the superconducting magnet was reduced to less than 3 mOe using an ultra-low-



FIG. 1. *T* dependence of  $M_{ZFC}(\bullet)$  and  $M_{FC}(\bigcirc)$  for Pd-*G*(1). *H*=1 Oe.

field capability option. For convenience, hereafter this remanent field is noted as the state H=0. The sample was cooled from 298 to 1.9 K at H=0. After an external magnetic field (*H*) was applied at 1.9 K, the zero-field-cooled magnetization ( $M_{\rm ZFC}$ ) was measured with increasing *T* from 1.9 to 298 K. Subsequently, the field-cooled magnetization ( $M_{\rm FC}$ ) was measured with decreasing *T* from 298 to 1.9 K in the presence of the same *H*. After the measurement of  $M_{\rm ZFC}$  and  $M_{\rm FC}$  at *H* were completed, a series of measurements of  $M_{\rm FC}$ were carried out at higher *H* in the following way. The sample was heated from 1.9 to 298 K. The magnetic field *H* was changed at 298 K. Then  $M_{\rm FC}$  was measured with decreasing *T* from 298 to 1.9 K in the presence of *H*.

(ii) SQUID ac magnetic susceptibility  $(\chi = \chi' + i\chi'')$ . A sample was cooled from 298 to 1.9 K at H=0. Then both  $\chi'$  and  $\chi''$  were simultaneously measured with increasing T from 1.9 K to high temperature  $T_h$  in the absence and presence of H, where the frequency and amplitude of the ac magnetic field were f=0.07 Hz-1 kHz and h=1-2 Oe, respectively. After each T scan the magnetic field H was changed at high temperature  $T_h$ . The sample was cooled from  $T_h$  to 1.9 K. Then the measurement was repeated with increasing T from 1.9 K to  $T_h$  in the presence of H.

### **IV. RESULT**

#### A. Pd-G(1)

The sample used in the present work consists of many flakes whose *c* axis are assumed to be randomly aligned over all directions. Figure 1 shows the *T* dependence of  $M_{ZFC}$  and  $M_{FC}$  for Pd-G(1) at H=1 Oe. The deviation of  $M_{ZFC}$  from  $M_{FC}$  starts to appear below 298 K, showing a behavior reminiscent of spin glasses.  $M_{ZFC}$  shows a sharp peak around 3.8



FIG. 2. *T* dependence of  $\chi_{FC}(=M_{FC}/H)$  for Pd-*G*(1). *H* = 3 kOe. The inset shows the *T* dependence of the reciprocal susceptibility, where the straight line is denoted by Eq. (1) with parameters given in the text.

K, while  $M_{\rm FC}$  shows a peak around 3.9 K (see the inset). Figure 2 shows the *T* dependence of  $\chi_{\rm FC}(=M_{\rm FC}/H)$  for Pd-*G*(1) at H=3 kOe. The susceptibility  $\chi_{\rm FC}$  obeys a Curie-Weiss law over a wide temperature range ( $40 \le T \le 298$  K). The least-squares fit of the data to the Curie-Weiss form

$$\chi_g = \frac{C_g}{T - \Theta} + \chi_g^0 \tag{1}$$

yields  $C_g = (1.389 \pm 0.020) \times 10^{-2} \text{ emu K/g}, \quad \Theta = -679.2 \pm 6.0 \text{ K}, \text{ and } \chi_g^0 = (-1.189 \pm 0.012) \times 10^{-5} \text{ emu/g}, \text{ where } \Theta$ is the Curie-Weiss temperature,  $C_g$  is the Curie-Weiss constant, and  $\chi_g^0$  is the temperature-independent susceptibility. The inset of Fig. 2 shows the reciprocal susceptibility defined by  $(\chi_g - \chi_g^0)^{-1}$  as a function of *T*. The data are well fitted with a straight line described by Eq. (1). The negative sign of  $\Theta$  indicates that the interaction between Pd atoms is antiferromagnetic. The susceptibility  $\chi_g^0$  is negative, reflecting the nature of diamagnetic susceptibility arising from graphite layers. If the magnetic moment is assigned to every Pd atom and the composition is given by  $C_{14.8}$ Pd for Pd-G(1), the effective magnetic moment  $P_{\text{eff}}$  can be estimated as  $P_{\rm eff} = 5.62 \pm 0.04 \mu_B$ . For further discussion, in Sec. V A we define the effective magnetic moment per Pd atom,  $p_c$ , which is related to  $P_{\text{eff}}$  through the relation  $P_{\text{eff}}^2 = p_c(p_c$ +2):  $p_c = 4.71 \mu_B$ . What is the saturation magnetization  $M_s$ ? Since we have no data of M vs T at H above 3 kOe, the value of  $M_s$  cannot be exactly determined experimentally. In spite of that, we assume that  $M_s$  is roughly equal to the value of  $M_{\rm FC}$  at 2 K for H=3 kOe. Then we have  $M_s=4.32$  $\times 10^{-2}$  emu/g=12.29 emu/Pd mol if the composition C<sub>14.8</sub>Pd is assumed. The saturation magnetic moment per Pd atom



FIG. 3. T dependence of (a)  $\chi'$  and (b)  $\chi''$  for Pd-G(1) at various f. H=0 Oe. h=2 Oe.

 $p_s(=M_s/N_A\mu_B)$  can be estimated as  $p_s=2.2\times10^{-3}\mu_B$ . This value of  $p_s$  is much smaller than that of  $p_c$ . The ratio  $p_c/p_s$  is 2141. Note that this ratio is independent of the composition and mass used in the calculation.

Figure 3(a) shows the *T* dependence of  $\chi'$  for Pd-*G*(1) at various *f* in the absence of *H*. The amplitude of the ac field is h=2 Oe. The dispersion  $\chi'$  shows a sharp peak at  $T_N$  (=3.8 K). This peak temperature is independent of *f* for  $0.07 \le f \le 1000$  Hz, suggesting that no spin-glass-like behavior occurs around  $T_N$  in spite of the irreversible effect of magnetization below 298 K (Fig. 1). In contrast, the value of



FIG. 4. f dependence of (a)  $\chi'$  and (b)  $\chi''$  for Pd-G(1) at various T. h=2 Oe. H=0 Oe.

 $\chi'$  is strongly dependent on *f*, reflecting the nature of disordered spin system. In the presence of an external magnetic field *H*, the peak of  $\chi'$  shifts to the low-temperature side with increasing *H* (3.3 K at *H*=140 Oe), indicating the antiferromagnetic phase below  $T_N$ . Figure 3(b) shows the *T* dependence of  $\chi''$  for Pd-*G*(1). A drastic decrease of  $\chi''$  with increasing *T* is observed near 3.5 K just below  $T_N$ .

Figure 4(a) shows the f dependence of  $\chi'$  for Pd-G(1) at various T. The dispersion  $\chi'$  decreases with increasing f. Figure 4(b) shows the f dependence of  $\chi''$  for Pd-G(1) at various T. For any T between 2 and 8 K,  $\chi''$  is almost indepen-

dent of f below 10 Hz but drastically increases with increasing f above 10 Hz. Here we assume that  $\chi''$  is described by a Debye-type relaxation with a single characteristic relaxation time  $\tau$  given by

$$\chi'' \approx \frac{\omega \tau}{1 + (\omega \tau)^2},\tag{2}$$

which has a peak at  $\omega \tau = 1$ . Our data suggest that the peak may exist for f > 1000 Hz for any T between 2 and 8 K, implying that the relaxation time  $\tau$  is at least shorter than  $1.6 \times 10^{-4} \text{ sec.}$ 

### B. Pd-G(2)

Figures 5(a) and (b) show the *T* dependence of  $M_{ZFC}$  and  $M_{FC}$  for Pd-*G*(2) at H=1 Oe. The magnetization  $M_{ZFC}$  shows a broad peak around 52 K and a sharp peak at 14 K. The deviation of  $M_{ZFC}$  from  $M_{FC}$  starts to appear below 298 K, showing a behavior reminiscent of spin glasses. This behavior may be due partly to spin frustration effects arising from competing interactions among nanoparticles: spins may be partially ordered within nanoparticles even at 298 K. The magnetization  $M_{FC}$  dramatically increases with decreasing *T* below 14–15 K. The *T* dependence of  $M_{FC}$  can be well described by a power-law form given by

$$M_{\rm FC} = A[(T_0 - T)/T_0]^{\beta} + B, \qquad (3)$$

for  $9 \le T \le 14.4$  K, where  $T_0 = 14.81 \pm 0.05$  K,  $\beta = 0.367 \pm 0.009$ ,  $A = 0.0298 \pm 0.0002$  (emu/g), and  $B = 0.0791 \pm 0.0004$  (emu/g). The critical exponent  $\beta$  of magnetization thus obtained is close to that predicted for 3D spin systems:  $\beta = 0.31$  for Ising symmetry,  $\beta = 0.33$  for XY symmetry, and  $\beta = 0.35$  for Heisenberg symmetry.<sup>13</sup> This result suggests that a 3D long-range spin order may develop below  $T_0$ .

Figure 6 shows the T dependence of  $\chi_{FC}$  for Pd-G(2) at H=3 kOe. The measurement was done with increasing T in the presence of H. The susceptibility  $\chi_{\rm FC}$  obeys a Curie-Weiss behavior in the limited temperature range  $(100 \le T)$  $\leq$  270 K). The least-squares fit of these data  $\chi$  vs T to Eq. (1) yields  $C_g = (5.0041 \pm 0.098) \times 10^{-2} \text{ emu K/g}, \quad \Theta = 25.42 \pm 1.25 \text{ K}, \text{ and } \chi_g^0 = (-1.431 \pm 0.035) \times 10^{-4} \text{ emu/g}.$  The inset of Fig. 6 shows the reciprocal susceptibility defined by  $(\chi_g - \chi_g^0)^{-1}$  as a function of T. The data are well fitted with a straight line described by Eq. (1) within the limited temperature range. The positive sign of  $\Theta$  suggests that the intraplanar exchange interactions between Pd atoms may be ferromagnetic. Since  $T_0 = 14.81$  K and  $\Theta = 25.42$  K, the ratio  $T_0/\Theta$  is estimated as 0.58, which is much smaller than 1 as predicted by the molecular-field theory. This result suggests that the system magnetically behaves like a quasi-2D spin system. Here we assume that Pd nanoparticles act as localized spins and the effective magnetic moment is assigned to every Pd atom. If the composition of Pd-G(2) is given by  $C_{14.8}$ Pd, the effective magnetic moment  $P_{eff}$  can be estimated as  $P_{\rm eff} = 10.76 \pm 0.08 \mu_B$ . Correspondingly, another effective magnetic moment  $p_c$  per Pd atom is given by  $p_c = 9.81 \mu_B$ .

Figure 7 shows the *H* dependence of  $M_{\rm FC}$  for Pd-*G*(2) at various *T*. The magnetization drastically changes around a characteristic field  $H_a^0 \approx 200$  Oe, reflecting a transition from the antiferromagnetic state to the ferromagnetic state. Similar



FIG. 5. (a) and (b) *T* dependence of  $M_{ZFC}(\bullet)$  and  $M_{FC}(\bigcirc)$  for Pd-*G*(2). *H*=1 Oe.

behavior is also observed in stage-2 NiCl<sub>2</sub> GIC.<sup>10</sup> Here we note that Fig. 7 is derived from the data of  $M_{\rm FC}$  vs *T* for various *H*, but not the data obtained directly from the measurement of  $M_{\rm FC}$  vs *H* with increasing *H* for each *T*. What is the saturation magnetization  $M_s$  in Pd-*G*(2)? From Fig. 7 the saturation magnetization  $M_s$  is roughly estimated as  $M_s$ = 5 emu/g or 1421 emu/Pd mol at H=40 kOe when the composition of C<sub>14.8</sub>Pd is assumed. If the saturation magnetization is assigned to every Pd atoms, the saturation magnetiz moment per Pd atom  $p_s(=M_s/N_{A\mu B})$  can be estimated as  $p_s$ =0.25 $\mu_B$ . This value of  $p_s$  is very different from that of  $p_c$ . The ratio  $p_c/p_s$  is 39.2.

Figures 8(a) and (b) show the *T* dependence of  $\chi'$  and  $\chi''$ 



FIG. 6. *T* dependence of  $\chi_{FC}$  at H=3 kOe for Pd-G(2). The inset shows the *T* dependence of the reciprocal susceptibility, where the straight line is denoted by Eq. (1) with parameters given in the text.

for Pd-G(2) at various f, where h = 1 Oe. The dispersion  $\chi'$  exhibits a peak at 14.3 K and a small shoulder around 11 K. The *T* dependence of  $\chi'$  in Pd-G(2) is similar to that of stage-2 NiCl<sub>2</sub> GIC.<sup>17</sup> The peak temperature (=14.3 K) is



FIG. 7. *H* dependence of  $M_{\rm FC}$  at various *T* for Pd-*G*(2).



FIG. 8. T dependence of (a)  $\chi'$  and (b)  $\chi''$  for Pd-G(2) at various f. H=0 Oe. h=1 Oe.

independent of *f* for  $0.07 \le f \le 1000$  Hz. In contrast, the value of  $\chi'$  drastically decreases with increasing *f* at the same *T*, reflecting the nature of disordered spin systems. The absorption  $\chi''$  also shows a sharp peak at  $T_{cu}(=14.2 \text{ K})$  and a small shoulder around  $T_{cl}(=11 \text{ K})$ . The peak temperature of  $\chi''$  is almost independent of *f* for  $0.07 \le f \le 1000$  Hz, suggesting that no spin-glass-like behavior occurs around  $T_{cu}$  in spite of the irreversible effect of magnetization below 298 K [Fig. 5(a)].



FIG. 9. T dependence of (a)  $\chi'$  and (b)  $\chi''$  for Pd-G(2). H = 0 Oe. f = 100 Hz. h = 1 Oe.

Figure 9 shows the *T* dependence of  $\chi'$  and  $\chi''$  for Pd-*G*(2) in the absence of *H*, where f=100 Hz and h=1 Oe. The data are taken with increasing *T*. Both  $\chi'$  and  $\chi''$  exhibit a peak around 14 K. Above this temperature  $\chi''$  exhibits a very broad peak around 100 K. Note that  $M_{ZFC}$  at H=1 Oe for Pd-*G*(2) has a broad peak around 52 K [see the inset of Fig. 5(a)], while the dc magnetic susceptibility of pristine Pd has a very broad peak around 85 K.<sup>8</sup>

Figures 10(a) and (b) show the *T* dependence of  $\chi'$  and  $\chi''$  for Pd-*G*(2) in the presence of *H*, respectively, where *h* = 1 Oe and *f* = 100 Hz. The peak of  $\chi'$  slightly shifts to the higher temperature side with increasing *H*: 14.3 K at *H*=0 and 14.85 K at *H*=70 Oe. This peak disappears above 70 Oe. This implies that the in-plane ferromagnetic order is apparently enhanced by the application of *H*. In contrast, the peak of  $\chi''$  shifts to the low-temperature side with increasing *H*: 14.3 K at *H*=0 and 13.19 K at *H*=5 Oe. The peak disappears above 10 Oe, suggesting that the anisotropy field in the *c* plane,  $H_A^{in}$ , is on the order of 5 Oe.

Figure 11(a) shows the *f* dependence of  $\chi'$  for Pd-*G*(2) at various *T*. It is found that  $\chi'$  can be well described by a power-law form ( $\chi' \approx \omega^{-x}$ ) over the whole frequency range used in the present work. Figure 11(b) shows the *T* dependence of *x* for Pd-*G*(2), which is similar to the *T* dependence of  $\chi''$  at H=0 [see Fig. 8(b)]. The exponent *x* is positive and very close to zero, and exhibits a local maximum around  $T_{cu}$  as well as a broad shoulder around  $T_{cl}$ . Similar behavior is observed in stage-2 NiCl<sub>2</sub> GIC, <sup>10</sup> where the value of *x* takes a local maximum at 16.5 K just below  $T_{cl}$  (=17.2 K) and at 21.1 K just above  $T_{cu}$ (=20.5 K).

Figure 12 shows the *f* dependence of  $\chi''$  for Pd-*G*(2) at various *T*. The *f* dependence of  $\chi''$  for f < 20 Hz is rather different from that for  $f \ge 20$  Hz depending on *T*. Below 17 K the absorption  $\chi''$  decreases with increasing *f* for f < 20 Hz,



FIG. 10. T dependence of (a)  $\chi'$  and (b)  $\chi''$  for Pd-G(2) at various H. f = 100 Hz. h = 1 Oe.

while it increases with increasing f for f > 20 Hz. Note that such behavior is also observed in stage-2 NiCl<sub>2</sub> GIC.<sup>10</sup>

### **V. DISCUSSION**

#### A. Itinerant spin systems

Before discussion, we summarize the magnetic properties of Pd-G(1) and Pd-G(2). Pd-G(1) shows an antiferromagnetic phase transition at  $T_N = 3.8$  K. The Curie-Weiss temperature is given by  $\Theta = -679.2 \pm 6.0$  K. The ratio of the



FIG. 11. (a) *f* dependence of  $\chi'$  for Pd-*G*(2) at various *T*. *h* = 1 Oe. *T*=9 ( $\bullet$ ), 11 ( $\bigcirc$ ), 12 ( $\blacktriangle$ ), 13 ( $\triangle$ ), 13.5 ( $\blacksquare$ ), 14 ( $\square$ ), 14.5 ( $\bullet$ ), 15 ( $\diamond$ ), 15.5 ( $\triangledown$ ), 16 ( $\bigtriangledown$ ), and 16.5 K ( $\odot$ ). (b) *T* dependence of exponent *x* ( $\chi' \approx \omega^{-x}$ ) ( $\bullet$ ) and *Y*[=(2/ $\pi$ )( $\chi''/\chi'$ )] at *f* = 0.07 Hz ( $\bigcirc$ ) and 1 Hz ( $\triangle$ ).

magnetic moments  $p_c/p_s$  is about 2140. This ratio may be overestimated because the magnetization used in this calculation is not saturated at 3 kOe which is the highest field for the experiment on Pd-G(1). In contrast, Pd-G(2) undergoes two magnetic phase transitions at  $T_{cu}(=14.2 \text{ K})$  and  $T_{cl}(=11 \text{ K})$ . The Curie-Weiss temperature is  $\Theta = 25.42 \pm 1.25 \text{ K}$ . The intraplanar interaction is strongly ferromagnetic, while the interplanar interaction is weakly antiferromagnetic. The ratio of the magnetic moments  $p_c/p_s$  is 39. Such a large ratio of  $p_c/p_s$  is a feature common to itinerant



FIG. 12. *f* dependence of  $\chi''$  for Pd-*G*(2) at various *T*. *h* = 1 Oe. The same notations are used as in Fig. 11(a).

spin systems where the d electrons responsible for magnetism are mobile. According to Moriya,14 there is some correlation between  $p_c/p_s$  vs  $T_C$  for ferromagnetic metals, where  $T_C$  is the Curie temperature. In the local limit for well-defined localized moment, the ratio  $p_c/p_s$  is independent of  $T_{C}$  and is nearly equal to 1. In the opposite weakly ferromagnetic limit, the theory predicts the divergence of this ratio as  $p_s \rightarrow 0$  or  $T_C \rightarrow 0$ , since the Curie-Weiss constant in this case is independent of  $p_s$  or  $T_c$ . The ratio  $p_c/p_s$  is much larger than 1 for small  $T_C$  and the ratio is generally close to 1 for very large  $T_C$ . The Rhodes-Wohlfarth plot  $(p_c/p_s \text{ vs } T_c)$  (Ref. 15) for various spin systems clearly indicates that the values of  $p_c/p_s$  are distributed almost continuously between these two extremes. The spin density fluctuations show various properties ranging between the two opposite extremes: the local moment limit (spin fluctuations localized in real space) and the limit of weakly ferro- and antiferromagnetic metals (spin fluctuations localized in reciprocal space; extended in real space). It has been believed that the Curie-Weiss law is attributed exclusively to the local spin system. However, this law is also observed in the itinerant spin system. In conclusion, Pd-G(1) is an itinerant spin system with mainly antiferromagnetic interactions, while Pd-G(2) is an itinerant spin system with mainly ferromagnetic interactions. We note that the magnetic properties of Pd-G strongly depend on the reaction times required for the hydrogen gas reduction: ferromagnetic nature for the long reaction time [Pd-G(2)] and the antiferromagnetic nature for the short reaction time [Pd-G(1)]. This result suggests that the magnetic property of Pd-G is closely related to the structure of Pd nanoparticles.

## B. Effect of Pd nanoparticles on magnetic susceptibility in Pd-G(1)

We discuss the magnitude of the temperature-independent susceptibility  $\chi_g^0 = (-11.89 \pm 0.12) \times 10^{-6}$  emu/g for Pd-*G*(1). We assume that  $\chi_g^0$  may be described by

$$\chi_g^0 = [12.011n \,\chi_g^0(\text{C}) + 106.4\chi_g^0(\text{Pd})]/(12.011n + 106.4),$$
(4)

for Pd-*G* having the composition  $C_nPd$ , where  $\chi_g^0(C)$  and  $\chi_g^0(Pd)$  are the susceptibility per gram of pristine graphite and Pd, respectively. For n = 14.8,  $\chi_g^0$  can be rewritten as  $\chi_g^0 = 0.626 \chi_g^0(C) + 0.374 \chi_g^0(Pd)$ . The susceptibility  $\chi_g^0(C)$ corresponds to a powder-averaged susceptibility of graphite given by  $\chi_g^0(C) = (\frac{2}{3}) \chi_g^a + (\frac{1}{3}) \chi_g^c$ , where  $\chi_g^c$  along the *c* axis is diamagnetic and weakly dependent on *T*. The susceptibility  $\chi_g^a$  is positive and almost independent of *T*. Note that the magnitude of  $\chi_g^c$  is much larger than that of  $\chi_g^a$ 

It is interesting to estimate the value of  $\chi_g^0$  as a function of T using the data of pristine graphite (highly oriented pyrolytic graphite, HOPG) (Ref. 16) and pristine Pd. The data of pristine Pd is obtained by van Leeuwen et al.8 The calculated value of  $\chi_{q}^{0}$  is dependent on T. It decreases with decreasing  $T:-1.69 \times 10^{-6}$  emu/g at 290 K to  $-2.38 \times 10^{-6}$  emu/g at 100 K. These values are relatively larger than the experimental value of  $\chi_{o}^{0}(=-11.89\times10^{-6} \text{ emu/g})$  for Pd-G(1). What is the cause of this discrepancy? One possibility is that the diamagnetic susceptibility of graphite  $\chi_g^0(C)$  may be described by  $\chi_g^c$  because samples are formed of small flakes which may be rather different from ideal powders:  $\chi_g^0(C) = \chi_g^c$ . When the formula  $\chi_g^0 = 0.626 \chi_g^c + 0.374 \chi_g^0(Pd)$  is used, we find that the calculated value of  $\chi_g^0$  decreases with decreasing  $T:-9.734 \times 10^{-6}$  emu/g at 290 K to -13.420 $\times 10^{-6}$  emu/g at 100 K. These values are close to the experimental result. A second possibility is that the susceptibility from Pd may decrease with decreasing size of nanoparticles. In fact, in Pd clusters/colloids with diameters ranging from 25-150 Å, van Leeuwen *et al.*<sup>8</sup> have observed a dramatic reduction of the susceptibility with particle size.

#### C. Nature of spin ordering in Pd-G(2)

Pd-G(2) approximates a quasi-two-dimensional (2D) XYlike ferromagnet with an extremely weak antiferromagnetic interplanar exchange interaction, in spite of the character of itinerant spin systems. The effective spin Hamiltonian of Pd-G(2) may be described by

$$\mathcal{H} = -2J \sum_{\langle i,j \rangle} \mathbf{S}_i \cdot \mathbf{S}_i + D \sum_i (S_i^z)^2 - 2J' \sum_{\langle i,m \rangle} \mathbf{S}_i \cdot \mathbf{S}_m, \quad (5)$$

where  $\hbar S$  is the effective spin angular momentum, the *z* axis coincides with the *c* axis, *J* (>0) is the ferromagnetic intraplanar interaction, *D*(>0) is the single ion anisotropy term, and *J'*(<0) is the antiferromagnetic interplanar interaction. For convenience the equivalent interaction fields at 0 K are defined as  $H_A^{\text{out}} = DS/g_c\mu_B$ ,  $H_{E'} = 2z'|J'|S/g_a\mu_B$ ,  $H_E = 2zJS/g_a\mu_B$ , and  $H_{SF} = [2H_A^{in}H_{E'}]^{1/2}$  where  $g_a$  and  $g_c$  are the *g* values along the *c* plane and *c* axis ( $g_a = g_c = 2$  is

assumed).  $H_E$  is the intraplanar exchange field,  $H_{E'}$  the interplanar exchange field,  $H_{SF}$  the spin flop field,  $H_A^{in}$  the anisotropy field in the *c* plane,  $H_A^{out}$  the anisotropy field along the *c* axis, and *z* and *z'* the numbers of interacting neighbors in and between the *c* plane, respectively (z=z'=6). The value of *S* is estimated as 4.9 from the relation  $P_{eff}=g[S(S+1)]^{1/2}$  with  $P_{eff}=10.76\pm0.08\mu_B$ . The value of *J* is calculated to be  $J=0.22\pm0.01$  K from the relation  $\Theta=2zJS(S+1)/3k_B$ . The value of  $H_{E'}$  is estimated as  $H_{E'}=H_a^{0/2}$ =100 Oe (see Ref. 10 for this relation) using the value of  $H_a^0$ derived from Fig. 7. The value of  $H_{A'}^{in}$  is also estimated as  $H_A^{in}\approx 5$  Oe from Fig. 10(b), leading to  $H_{SF}\approx 30$  Oe. Then the value of |J'| is on the order of  $2.8 \times 10^{-4}$  K, which is much weaker than the intraplanar interaction:  $|J'|/J\approx 1.3 \times 10^{-3}$ .

What is the mechanism of two magnetic phase transitions at  $T_{cu}$  and  $T_{cl}$  in Pd-G(2)? The Pd layers of this compound are formed of Pd nanoparticles. This finite size of nanoparticles is a crucial element in the magnetic phase transition of this compound. The effective interplanar exchange interaction  $J'_{eff}$  between spins over the in-plane spin correlation length  $\xi$  in adjacent Pd layers is described by  $J'_{eff} = J'(\xi/a)^2$ , where J' is the interplanar exchange interaction and a is the in-plane lattice constant. The growth of  $\xi$  is limited by island size as T is decreased. Suppression of the increase in  $J'_{eff}$ leads to the realization of 2D spin ordering between  $T_{cu}$  and  $T_{cl}$ . Below  $T_{cl}$  there occurs a 3D spin ordering where the 2D ferromagnetic layers are antiferromagnetically stacked along the c axis. Note that similar behavior is observed in stage-2 NiCl<sub>2</sub> GIC with  $T_{cu} (=20.5 \pm 0.2 \text{ K})$  and  $T_{cl} (=17.2 \text{ K})$ .<sup>10</sup>

#### **D.** Origin of magnetic interaction

It is well known that pristine Pd is close to ferromagnetism, but not a ferromagnet. The enhanced susceptibility of pristine Pd is described by  $\chi_g(Pd) = s\chi_g^0(Pd)$  with  $\chi_g^0(Pd) = 2\mu_B^2 N_d(E_F)$ , where *s* is a Stoner factor (*s*=9.37) and  $N_d(E_F)$  is the density of states for *d*-band electrons per spin at the Fermi level  $E_F$ . The value of *s* is much larger than that of other metals. What is the origin of the intraplanar ferromagnetic interaction in Pd-*G*(2)? This ferromagnetic nature is considered to be related to the Stoner condition of ferromagnetism. When *U* is the Coulomb interaction between *d* electrons belonging to the same atom, the ferromagnetic state can exist at T=0 K only for  $U>U_{cr}$ , where  $U_{cr}$  becomes as small as possible. This means that Fermi energy should be located at a maximum of the density of states.

The origin of the interplanar antiferromagnetic exchange interaction may be similar to that of CoCl<sub>2</sub> GIC. According to Yeh *et al.*,<sup>17</sup> the dominant interplanar exchange interaction in stage-1 CoCl<sub>2</sub> GIC is the superexchange interaction, while both the dipole-dipole interaction and superexchange interaction are equally important in stage-2 CoCl<sub>2</sub> GIC. For higher stage CoCl<sub>2</sub> GIC, the dipole-dipole interaction dominates because of the rapid decrease of the superexchange interaction may be described by the dipole-dipole interaction between  $S_i$  and  $S_m$ ,

$$H_{d-d} = (g_a \mu_B)^2 \left( \frac{\mathbf{S}_i \cdot \mathbf{S}_m}{R_{im}^3} - \frac{3(\mathbf{S}_i \cdot \mathbf{R}_{im})(\mathbf{S}_m \cdot \mathbf{R}_{im})}{R_{im}^5} \right).$$
(6)

If we assume that the direction of spin  $\mathbf{S}_i$  is perpendicular to  $\mathbf{R}_{im}$  (or perpendicular to the *c* axis), then the Hamiltonian  $H_{d-d}$  can be approximated as a form of  $2J_{d-d}(\mathbf{S}_i \cdot \mathbf{S}_m)$  where  $J_{d-d} = (g_a \mu_B)^2 / 2(R_{im})^3$  is positive and favors the antiferromagnetic interplanar interaction. The value of  $J_{d-d}$  is calculated as  $1.246d^{-3}$  [K]= $3.0 \times 10^{-4}$  K, where  $g_a = 2$  is assumed and *d* is the *c* axis repeat distance in units of Å: d = 16.6 Å for stage-3 PdCl<sub>2</sub> GIC.<sup>11</sup> This value of  $J_{d-d}$  is in good agreement with the value of  $|J'| (= 2.8 \times 10^{-4} \text{ K})$  derived above.

### E. Nature of dynamic spin fluctuations in Pd-G(2)

We have shown that the power-law form  $(\chi' \approx \omega^{-x})$  is valid over the whole frequency range at any *T* in Pd-*G*(2). Then  $\chi''$  can be calculated as<sup>10,18</sup>

$$\chi''(\omega) = \frac{\pi x}{2} \chi'(\omega) \operatorname{sgn}(\omega), \tag{7}$$

using the Kramers-Kronig relation

$$\chi''(\omega) = -\frac{1}{\pi} P \int_{-\infty}^{\infty} d\omega' \frac{1}{\omega' - \omega} [\chi'(\omega') - \chi^{\infty}], \quad (8)$$

where the notation  $[\operatorname{sgn}(\omega)]$ , which is 1 for  $\omega > 0$  and -1 for  $\omega < 0$ , is included in Eq. (7) because  $\chi''$  should be an odd function of  $\omega$ , and  $\chi^{\infty}$  is the complex susceptibility at  $\omega = \infty$  and is assumed to be zero here. In Fig. 11(b) we also show the *T* dependence of  $Y[=(2/\pi)\chi''/\chi']$  at f=0.07 and 1 Hz. Although the value of *Y* is a little larger than that of *x* at each *T*, the *T* dependence of *Y* is very similar to that of *x*. These results are consistent with the prediction from Eq. (7). According to the fluctuation-dissipation theorem, the Fourier spectrum  $S(\omega)$  of the time-dependent magnetization fluctuation  $\langle M(0)M(t) \rangle$  is related to  $\chi''(\omega)$  by

$$S_{aa}(\omega) = \int_{-\infty}^{\infty} \langle M_a(0)M_a(t)\rangle e^{-i\omega t}dt = \frac{2k_B T}{\hbar\omega}\chi_{aa}''(\omega),$$
(9)

where *t* is time and M(t) is the time-dependent magnetization. It is expected from Eq. (7) that  $\chi''$  at low frequencies is described by the same power-law form as  $\chi': \chi'' \approx \omega^{-x}$ . Thus  $S(\omega)$  has the form  $\omega^{-(1+x)}$ , indicating that  $\langle M(0)M(t) \rangle$  varies with *t* as  $t^x$ . In the limit of  $x \rightarrow 0$ ,  $\langle M(0)M(t) \rangle$  has a logarithmic time dependence. As shown in Fig. 5(a) at temperatures below 298 K, the magnetization  $M_{\rm ZFC}$  is much smaller than the magnetization  $M_{\rm FC}$  corresponding to the magnetization in thermal equilibrium. The magnetization  $M_{\rm ZFC}$  may increase and reach  $M_{\rm FC}$  with *t* as  $\ln(t)$ .

As shown in Fig. 12,  $\chi''$  for Pd-G(2) has the characteristic f dependence:  $\chi''$  decreases with increasing f for  $f \leq 20$  Hz and then increases with f above 20 Hz. This may be explained in terms of the following model. Around  $T_{cu}$  and  $T_{cl}$ , the internanoparticle spin correlation is still random within each Pd layer. The magnetization of each Pd nanoparticle will fluctuate and change direction relative to other Pd nanoparticles on a certain characteristic time scale depending on

the nanoparticle size and internanoparticle interaction. There are two kinds of relaxation time associated with intrananoparticle fluctuation ( $\tau_{\rm in}$ ) and internanoparticle fluctuation ( $\tau_{\rm out}$ ). The relaxation time  $\tau_{\rm out}$  is assumed to be much longer than the relaxation time  $\tau_{\rm in}$ . Correspondingly the characteristic frequencies  $f_{\rm in}$  and  $f_{\rm out}$  are defined by  $(2\pi\tau_{\rm in})^{-1}$  and  $(2\pi\tau_{\rm out})^{-1}$ , respectively:  $f_{\rm in} \gg f_{\rm out}$ . When the relaxation of these fluctuations is of the Debye type,  $\chi''$  may be described by  $^{10,18}$ 

$$\chi'' = \chi_{\rm in}(\mathbf{Q} = 0) \frac{\omega \tau_{\rm in}}{1 + (\omega \tau_{\rm in})^2} + \chi_{\rm out}(\mathbf{Q} = 0) \frac{\omega \tau_{\rm out}}{1 + (\omega \tau_{\rm out})^2},$$
(10)

where  $\chi_{in}(\mathbf{Q})$  and  $\chi_{out}(\mathbf{Q})$  are the wave-vector-dependent susceptibilities related to the intrananoparticle and internanoparticle fluctuations, respectively. The absorption  $\chi''$  has two peaks at  $\omega \tau_{in} = 1$  (or  $f = f_{in}$ ) and  $\omega \tau_{out} = 1$  (or  $f = f_{out}$ ). Our results indicate that (i)  $f_{out}$  is lower than 0.07 Hz and that (ii)  $f_{in}$  is higher than 1 kHz at least below 17 K. No *T* dependence of  $f_{in}$  and  $f_{out}$  is observed.

#### VI. CONCLUSION

The structure of Pd-G is characterized by multilayered Pd nanoparticles. Different reduction conditions lead to different degrees of developed nanoparticles. The magnetic properties

of Pd-G are dependent on the reduction conditions: the antiferromagnetic nature for the short reaction time [Pd-G(1)]and ferromagnetic nature for the long reaction time [Pd-G(2)]. Pd-G(1) magnetically behaves like a antiferromagnetic itinerant spin system. It undergoes antiferromagnetic phase transition at  $T_N$  (=3.8 K). No spin-glass-like behavior occurs around  $T_N$ . Pd-G(2) magnetically behaves like a quasi-2D XY-like ferromagnet with very weak antiferromagnetic interplanar interaction. It undergoes magnetic phase transitions at  $T_{cu}$  (=14.2±0.2 K) and  $T_{cl}$  (=11 K). Between  $T_{cu}$  and  $T_{cl}$  a 2D ferromagnetic spin order is established within each Pd nanoparticle in the Pd layer. The ordered phase below  $T_{cl}$  is a 3D antiferromagnetic one. The existence of Pd nanoparticles in a reduced dimension is a key to understanding the uniqueness of spin ordering in Pd-G. To be more conclusive on the nature of nanoparticles, further studies are required using small-angle neutron scattering, x-ray scattering, and so on.

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