## **Electronic phase diagram and phase separation in Cr-doped manganites**

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Magnetic properties are systematically investigated for Cr-doped manganites  $R_{1/2}$ Ca<sub>1/2</sub>(Mn,Cr)O<sub>3</sub> (R=La, Nd, Sm, and Eu) with variation of the averaged ionic radius  $r_R$  of the rare-earth ion, or the one-electron bandwidth *W*. Cr-doping procedure on the perovskite *B* site significantly unstabilizes the charge-ordered (CO) state, and induces the ferromagnetic metallic (FM) state in the large- $r_R \ge 1.29$  Å) region. We have found coexistence of the CO domains in the FM phase, or the electronic phase separation, near the FM-CO phase boundary. [S0163-1829(99)08737-8]

Recent extensive study on the perovskite-type doped manganites, which show "colossal" magnetoresistance,<sup>1</sup> begins to reveal the unusual microscopical structures of spin-, charge-, and orbital-ordered state.<sup>2-4</sup> The most prototypical example is  $La_{1/2}Ca_{1/2}MnO_3$ , which is a ferromagnet  $[T_{CO}]$  $(\sim 190 \text{ K}) \leq T \leq T_{\text{C}} (= 230 \text{ K})$  but is transferred into an antiferromagnetic charge-ordered (CO) insulator below  $T_{\rm C}$ .<sup>5</sup> Electron-diffraction measurements<sup>4</sup> on La<sub>1/2</sub>Ca<sub>1/2</sub>MnO<sub>3</sub> have revealed coexistence of the ferromagnetic metallic (FM) and CO microdomains of order of  $\sim 100\,$  nm. In addition,  $^{139}$ La and <sup>55</sup>Mn NMR measurements<sup>3</sup> on La<sub>1/2</sub>Ca<sub>1/2</sub>MnO<sub>3</sub> have confirmed that both the FM and antiferromagnetic CO phases coexist below  $T_{\rm CO}$ . Such a coexistence of the FM and CO phases is an interesting aspect of the doped manganites. In the material point of view, La<sub>1/2</sub>Ca<sub>1/2</sub>MnO<sub>3</sub> locates near the phase boundary between the FM and CO states;  $La_{1/2}Sr_{1/2}MnO_3$  with larger one-electron bandwidth W is ferromagnet ( $T_{\rm C}$ =360 K, Ref. 6), while Pr<sub>1/2</sub>Ca<sub>1/2</sub>MnO<sub>3</sub> with smaller-W shows the charge-ordering transition at  $T_{CO}$ = 240 K.<sup>7</sup> Then it is plausible that subtle balance of the FM and CO states causes the noble electronic phase separation (PS) in the vicinity of the FM-CO phase boundary.

The generic behavior of paramagnetic-to-ferromagnetic transition of doped manganites is understood within the framework of double-exchange theory,<sup>8–10</sup> which includes only the transfer integral *t* of the  $e_g$  electrons and the on-site exchange interaction (Hund's-rule coupling;  $J_H$ ) between the itinerant  $e_g$  electrons and localized  $t_{2g}$  spins (S=3/2). We need to include additional effects, e.g., electron-phonon coupling, however, to understand the insulating behavior in the paramagnetic state.<sup>11</sup> It is well established that the CO state is amenable to the external perturbations, such as, magnetic field,<sup>7,12</sup> external pressure,<sup>13</sup> chemical pressure.<sup>13</sup> In particular, Raveau, Maignan, and Martin<sup>14</sup> have found that substitution of the perovskite *B* site with other transition metals,

e.g., Cr and Co, significantly suppresses the CO phase. Therefore the *B*-site substitution is the fourth perturbation to control the stability of the CO phase.

In this paper we have investigated the electronic and magnetic properties for Cr-doped manganites  $R_{1/2}$ Ca<sub>1/2</sub>(Mn,Cr)O<sub>3</sub> (R = La, Nd, Sm, and Eu) with changing the averaged ionic radius  $r_R$  of the rare-earth ion. Here, we can reduce the one-electron bandwidth W by decreasing  $r_R$  (chemical pressure).<sup>15</sup> Without Cr doping, the ground states of these compounds would be the CO insulating state. We have derived an electronic phase diagram for Cr-doped manganites against  $r_R$  (or W), which is dominated by the CO phase in the small- $r_R$  region and the FM phase in the large $r_R$  region. The magnetization curve in the FM phase is found to be suppressed near the FM-CO phase boundary (1.24  $\leq r_R \leq 1.27$  Å). Dark-field image obtained by electron microscope from a charge-ordering superlattice indicates that the suppression is due to the electronic PS between the CO and FM states.

Ceramics compounds  $R_{1/2}Ca_{1/2}(Mn,Cr)O_3$  (R =La,  $La_{0.25}Nd_{0.75}$ ,  $La_{0.5}Nd_{0.5}$ ,  $La_{0.75}Nd_{0.25}$ , Nd, Nd<sub>0.5</sub>Sm<sub>0.5</sub>, Sm, and Eu) were synthesized by solid-state reaction in air atmosphere. Stoichiometric mixture of commercial La<sub>2</sub>O<sub>3</sub>, Nd<sub>2</sub>O<sub>3</sub>, Sm<sub>2</sub>O<sub>3</sub>, Eu<sub>2</sub>O<sub>3</sub>, CaCO<sub>3</sub>, and  $Mn_3O_4$ powder was well ground and calcined two times at 1250 °C for 24 h. Then the resulting powder was pressed into a disk with a size of 20 mm $\phi \times 4$  mm and sintered at 1250 °C for 24 h. Powder x-ray-diffraction measurements at room temperature and Rietveld analysis<sup>16</sup> indicate that both the Cr-undoped and Cr-doped compounds were single phase without detectable impurity. The crystal symmetry is orthorhombic (*Pbnm*; Z=4). We show in Fig. 1 prototypical examples of the x-ray powder patterns together with the best-fitted Rietveld results; upper and lower for  $Nd_{1/2}Ca_{1/2}MnO_3$  ( $R_{wp}=11.16$ ) panels are and

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FIG. 1. Powder x-ray pattern (filled circles) for  $Nd_{1/2}Ca_{1/2}MnO_3$  (upper panel) and  $Nd_{1/2}Ca_{1/2}(Mn_{0.97}Cr_{0.03})O_3$  (lower panel). Solid curve is result of the Rietveld analysis.

 $Nd_{1/2}Ca_{1/2}(Mn_{0.97}Cr_{0.03})O_3$  ( $R_{wp}=11.13$ ), respectively. As seen in the x-ray powder patterns, the quality of the Crdoped compound is as good as the Cr-undoped one. Lattice constants are determined to be a = 5.4178(3) Å, b =5.3945(3) Å, and c=7.6150(4) Å for the Cr-undoped compound. These values slightly increase with Cr-doping to a = 5.4216(4) Å, b = 5.3993(3) Å, and c = 7.6200(4) Å, perhaps reflecting larger ionic radius of  $Cr^{3+}$  (=0.61 Å) as compared with  $Mn^{3+}$  (=0.59 Å). The lattice structure is nearly cubic;  $a \approx b \approx c/\sqrt{2}$ . Magnetization M was measured under a field of  $\mu_0 H = 0.5$  T after cooling down to 5 K in the zero-field cooled (ZFC), using a superconducting quantum interference device (SQUID) magnetometer. For fourprobe resistivity measurements, the crystal was cut into a rectangular shape, typically of  $3 \times 2 \times 1$  mm<sup>3</sup>, and electrical contacts were made with a heat-treatment-type silver paint.

Figure 2 shows Cr concentration dependence of mangetization for (a)  $Sm_{1/2}Ca_{1/2}MnO_3$  and (b)  $Nd_{1/2}Ca_{1/2}MnO_3$ . A definite cusp structure, which corresponds to the antiferromagnetic CO transition, is observed at ~280 K (~250 K) for Cr-undoped  $Sm_{1/2}Ca_{1/2}MnO_3$  ( $Nd_{1/2}Ca_{1/2}MnO_3$ ). The cusp structure gradually blurred with Cr doping and the *M* value at lower temperature increases [see Fig. 2(a)]. A drastic effect of the Cr doping is observed in the larger-*W* system, i.e.,  $Na_{1/2}Ca_{1/2}MnO_3$  [Fig. 2(b)]. With increases of Cr concentration beyond 2%, the *M* value steeply rises below ~160 K, showing a ferromagnetic behavior. Such a significant effect has been ascribed to the random effect of the *B*-site substitution.<sup>17</sup> It is well known that the CO state of



FIG. 2. Cr concentration dependence of magnetization M for (a)  $Sm_{1/2}Ca_{1/2}MnO_3$  and (b)  $Nd_{1/2}Ca_{1/2}MnO_3$ . M was measured after cooling down to 5 K in the zero field (ZFC). Thick curves stand for the Cr-undoped compounds.

doped manganites is accompanied by the alternation of the doubly degenerated  $e_g$  orbitals. The *B*-site randomness destroys the coherence of the orbital and/or charge alternation, and hence makes the ordering short ranged. Katsufuji *et al.*<sup>17</sup> have investigated the microscopic structure of Cr-doped La<sub>0.375</sub>Ca<sub>0.625</sub>MnO<sub>3</sub> by means of electron microscope, and have found that CO state is of short range with a correlation length of less than 10 nm. (Note that the carrier-doping procedure is qualitatively different from the *B*-site substitution, because the doped carrier can move and does not destroy the coherence.) On the other hand, the FM state remains long ranged even with the *B*-site randomness, and has less entropy as compared with the short-ranged CO state. Accordingly, with decrease of temperature, the free energy of the FM state becomes lower than that of the short-ranged CO state.

Now, let us survey the *W* dependence of the magnetization curve at a fixed Cr-doping level. We show in Fig. 3 temperature dependence of magnetization *M* of the 3% Crdoped manganites  $R_{1/2}Ca_{1/2}(Mn_{0.97}Cr_{0.03})O_3$ . Without the Cr doping, the ground states of these compounds would be antiferromagnetic CO phase with so-called CE-type<sup>18</sup> spin ordering. The *M*-*T* curve for La<sub>1/2</sub>Ca<sub>1/2</sub>(Mn<sub>0.97</sub>Cr<sub>0.03</sub>)O<sub>3</sub> steeply rises below  $T_C = 240$  K (indicated by downward arrow). In the case of R = Sm and Eu, no trace of the ferromagnetic transition is observed in the *M*-*T* curve.  $T_C$  was determined from the inflection point of the *M*-*T* curves. Inset shows the magnetization curve measured at 30 K. The *M*-*H* curve of  $R = Nd_{0.75}La_{0.25}$  rapidly increases with magnetic field, and reaches  $\sim 3\mu_B$  near the ideal value



FIG. 3. Temperature dependence of magnetization  $M(\mu_B H = 0.5 \text{ T})$  for 3% Cr-doped manganites  $R_{1/2}\text{Ca}_{1/2}(\text{Mn}_{0.97}\text{Cr}_{0.03})\text{O}_3$ , with variation of the averaged ionic radius  $r_R$  of the rare-earth ion. M was measured after cooling down to 5 K in the zero field (ZFC). Arrows indicate the Curie temperatures  $T_C$  determined from the infection point of the M-T curves. Inset shows magnetization curves for the Cr-doped manganites.

 $(=3.5\mu_{\rm B})$  at ~1 T. A similar feature of the *M*-*T* curve is observed for R = La, Nd<sub>0.25</sub>La<sub>0.75</sub>, and Nd<sub>0.5</sub>La<sub>0.5</sub>, except for the shift of  $T_{\rm C}$  toward the higher temperature side. The curve for R = Sm increases linearly with the applied magnetic field, reflecting the antiferromagnetic CO state.

In the intermediate  $r_R$  region, that is, R = Nd and Nd<sub>0.5</sub>Sm<sub>0.5</sub>, a curious magnetic behavior is observed. As seen in Fig. 3, the *M*-*T* curve for R = Nd (Nd<sub>0.5</sub>Sm<sub>0.5</sub>) rises ~140 K (~110 K), suggesting a paramagneticat ferromagnetic transition. The magnetization curve shown in the inset, however, increases rather gradually with applied field; M reaches at  $\sim 1.6\mu_{\rm B}$  (0.8 $\mu_{\rm B}$ ) at  $\mu_0 H = 1$  T, and continues to increase up to the highest field ( $\mu_0 H = 5$  T). Such an M-H behavior is considered to be due to coexistence of antiferromagentic CO domains in the FM state. To confirm this hypothesis, we show in Fig. 4 a dark-field image of  $Nd_{1/2}Ca_{1/2}(Mn_{0.97}Cr_{0.03})O_3$  at 120 K. The image is taken using a charge-ordering superlattice reflection (see inset). The CO microdomains with bright contrast are clearly observed in the FM domains with dark contrast.<sup>4</sup> Typical size of the CO microdomains is  $\sim 20-50$  nm.

Here, let us argue the  $e_g$ -electron density (1-x) of the CO and FM microdomains. The wave vector of the charge ordering is found to be ~0.445, indicating that the electron density of the CO microdomain is ~0.445 (x~0.555). With use of the nominal electron density (=0.485) of the 3% Cr-doped Nd<sub>1/2</sub>Ca<sub>1/2</sub>MnO<sub>3</sub>, the doping level of the FM microdomain is estimated to be ~0.525.<sup>19</sup> In other words, the electron density is slightly dense in the FM microdomain as



FIG. 4. Dark-field image at 120 K ( $\leq T_C$ ) for Nd<sub>1/2</sub>Ca<sub>1/2</sub>(Mn<sub>0.97</sub>Cr<sub>0.03</sub>)O<sub>3</sub> obtained from a charge-ordering superlattice reflection (indicated by an arrow in the inset). The bright and dark contrast correspond to the CO and FM microdomains, respectively.

compared with the CO microdomain. Such a *spontaneous* fluctuation of the electron density, or the electronic phase separation (PS), has been theoretically predicted in the doped mangnaites.<sup>2,20</sup> In the actual material, however, the electrostatic energy loss should supress such a density fluctuation. Nevertheless, in the present case, (i) subtle competition between the FM and CO state controlled by chemical pressure as well as (ii) the short-ranged CO state induced by Cr doping seems to stabilize the electronic PS state.

To summarize, we plot in Fig. 5  $T_{\rm C}$  (filled circles) and the charge-ordering temperature  $T_{\rm CO}$  (filled squares) for the Crdoped  $R_{1/2}{\rm Ca}_{1/2}({\rm Mn}_{0.97}{\rm Cr}_{0.03}){\rm O}_3$  against  $r_R$ .  $T_{\rm CO}$  was determined from the maximum point of the activation energy  $E_{\rm ac}$  [= $d \ln(\rho)/d(1/T)$ , where  $\rho$  is resistivity]. The open



FIG. 5. Electronic phase diagram for 3% Cr-doped manganites  $R_{1/2}Ca_{1/2}(Mn_{0.97}Cr_{0.03})O_3$  against averaged ionic radius  $r_R$  of the rare-earth ion. Closed circles and squares are Curie temperatures and critical temperatures for the charge-ordering transition, respectively. CO, FM, and CO stand for the charge-ordered, ferromagnetic metallic, and phase separation states, respectively. Open symbols represent the data for the Cr-undoped compounds.

symbols stand for the temperatures for the Cr-undoped compounds. With decrease of  $r_R$ ,  $T_C$  gradually decreases from  $T_C \approx 220$  K for R = La, and eventually the transition vanishes at  $r_R = 1.24$  Å (R = Sm). Accordingly,  $T_{CO}$  appears below  $r_R = 1.27$  Å(R = Nd), and then gradually increases up to  $T_{CO} \approx 260$  K for R = Eu. The electronic PS, which causes the suppressed M-H curve, appears near the FM-CO phase boundary. Thus the electronic PS effect is enhanced

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in the vicinity of the phase boundary, where two adjacent phases strongly compete.

The authors are grateful to T. Katsufuji for fruitful discussion. This work was supported by a Grant-In-Aid for Scientific Research from the Ministry of Education, Science, Sports and Culture and from Precursory Research for Embryonic Science and Technology (PRESTO), Japan Science and Technology Corporation (JST).

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