

Comparison of bulk $R_{2-z}Ce_zCuO_4$ with superlattice $R_{2-z}Ce_zCuO_4/SrO/NbO_2/SrO/CuO_2$

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Bulk $R_{2-z}Ce_zCuO_4$ compounds superconduct for trivalent ions $R = Pr, Nd, Sm, \text{ and } Eu$, but not for Gd or for Cm (with Th replacing Ce). $R_{2-z}Ce_zSr_2Cu_2NbO_{10}$ is a natural superlattice of $R_{2-z}Ce_zCuO_4$ and the layers $SrO/NbO_2/SrO/CuO_2$; it exhibits bulk superconductivity for $R = Nd, Sm, Eu, \text{ and } Gd$, but not for $R = Pr$. These differences imply that the superconducting regions in the bulk and in the superlattice must be different, and not both cuprate planes. The primary superconductivity is assigned to interstitial oxygen and $Nd-O$ layers in the bulk and to $Sr-O$ layers in the superlattice. [S0163-1829(99)05741-0]

I. RARE-EARTH COMPOUNDS

Comparative studies of the properties of different but similar high-temperature superconductors offer opportunities to learn details of the mechanisms of superconductivity that might otherwise be obscured. Classes of materials that have proven especially illuminating are the superconducting homologues containing different rare-earth ions, such as $R_{2-z}Ce_zCuO_4$ ($R21-4$) ($R = \text{rare earth}$) (Refs. 1–6) or $RBa_2Cu_3O_7$ ($R123-7$) (Refs. 7–17). The trivalent rare-earth ions all have about the same radius, $\sim 1 \text{ \AA}$, ranging from 0.85 \AA for Lu^{3+} to 1.14 \AA for La^{3+} ,¹⁸ and exhibit the famous but small lanthanide contraction as they become heavier. Chemically the trivalent ions are all similar, having Xe-like outer shells within which lie $4f$ electronic shells of radii $\sim 0.4 \text{ \AA}$. These $4f$ shells are responsible for the ions' magnetic moments, and they can be thought of as point magnets with the different rare-earth ions having different magnetic moments. Therefore, changing the rare earths changes the magnetism of the ions but only slightly changes the size; otherwise, to an excellent approximation, there is no change.

II. NATURAL SUPERLATTICE COMPOUNDS

A relatively unexplored area of investigation involves the "natural superlattices," materials such as $R_{2-z}Ce_zSr_2Cu_2NbO_{10}$ ($R222Nb-10$),^{19–24} which are superlattices of $R21-4$ and the oxide layers $SrO/NbO_2/SrO/CuO_2$ (see Fig. 1). Although there are slight differences in bond lengths, the oxide layers appear not to greatly influence the superconductivity: For example, with $R = Nd$, the $Nd21-4$ layers in $Nd222Nb-10$ appear to be otherwise structurally identical to those of the bulk superconductor $Nd21-4$; the main difference seems to be the continuously variable Ce -doping level z , which is optimal for $z \approx 0.15$ in $Nd21-4$ and $z \approx 0.5$ in $Nd222Nb-10$, and which might conceivably be different in order to maintain a specific hole density at some location. Thus one is led to believe that the superconducting layers in both materials are the same: a subset of the layers in $Nd21-4$, perhaps even the cuprate planes.

A. Different superconducting units

However, if the superconducting layers are *the same* in both $Nd21-4$ and the natural superlattice material $Nd222Nb-10$, then *replacement of Nd by another rare-earth ion (such as Gd) must render either both or neither $Gd21-4$ and $Gd222Nb-10$ superconducting—with similar results for Pr replacing Nd . In fact this does not happen for either Gd or Pr (although it does happen for Sm and Eu): $Gd21-4$ does not superconduct,²⁵ while $Gd222Nb-10$ does,^{21,22} and $Pr21-4$ does superconduct,² while $Pr222Nb-10$ does not exhibit bulk superconductivity.^{21,22,26}*

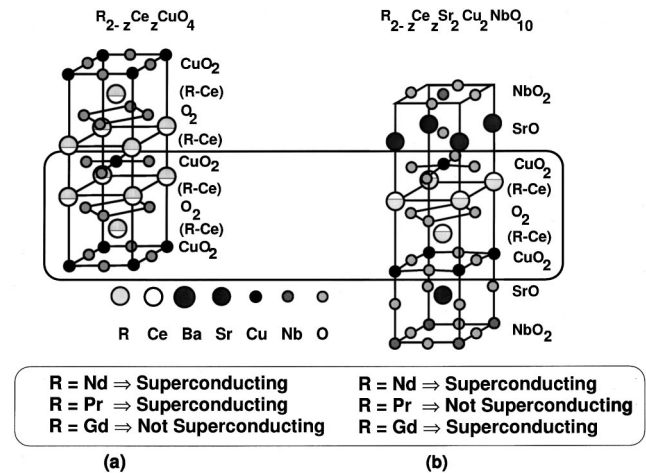


FIG. 1. Crystal structures of (a) $R_{2-z}Ce_zCuO_4$ (with z typically ≈ 0.15), where R is a rare-earth ion, and (b) $R_{2-z}Ce_zSr_2Cu_2NbO_{10}$ (with z typically ≈ 0.5), which is a natural superlattice of $R_{2-z}Ce_zCuO_4$ and the oxide layers $SrO/NbO_2/SrO/CuO_2$. The boxed parts of the structures are common to both. For the trivalent rare-earth ions $Nd, Pr, \text{ and } Gd$, structure (a) superconducts for Nd and Pr , but not Gd , while structure (b) exhibits bulk superconductivity for Nd and Gd , but not Pr . Recently, we have shown that $Pr222Nb-10$ exhibits *granular* superconductivity, as discussed in the text. The fact that both (a) and (b) do not have the same properties for $R = Nd, Pr, \text{ and } Gd$ indicates that the superconducting structural region is necessarily different in the two sets of homologues.

This is strong evidence that the communal geometrical unit, namely either the cuprate planes, the Nd21-4 unit cell, or whatever universal structural entity is supposedly hosting the primary superconducting condensate²⁷ in Nd21-4 is *different* from the corresponding superconducting entity in Nd222Nb-10. It also is an important clue concerning the origin of high-temperature superconductivity. If the unifying structural superconducting entity in both compounds is the same when the rare earth is Nd, then making the same change to both compounds Nd21-4 and Nd222Nb-10, Nd→Gd or Nd→Pr, must produce compounds that either both superconduct or both do not superconduct—or else the unifying structural superconducting entities were not the same to begin with. Clearly, whatever is superconducting in Nd21-4 is different from whatever is superconducting in Nd222Nb-10. In particular, the failures of Gd21-4 and Pr222Nb-10 to exhibit bulk superconductivity must be either due to defects, or else the cuprate planes cannot be the primary superconducting layers in both the R_{21-4} and the $R_{222}Nb-10$ compounds.

B. Superconducting Sr-O planes

We propose that the superconducting entity involves the R-O layers and interstitial oxygen (near the face center of the R plane) in R_{21-4} ,²⁸ but is associated with the Sr-O layers of $R_{222}Nb-10$. The reasons for these assignments are: (i) These are the locations of the holes or the hypocharged oxygen (O^{-Z} with $Z < 2$, as determined by the bond-valence-sum method²⁹), and we expect the primary superconducting condensate to be where the holes are; and (ii) these are the layers that either do not contain pair-breaking magnetic moments (as in $R_{222}Nb-10$), or that contain only moments that are rendered impotent for pair breaking by the crystal-field splitting (as for R_{21-4} , with $R = Pr, Nd, Sm,$ and Eu).^{28,30} It is certainly true in conventional superconductors that the supercurrent is where the carriers are and resides in regions lacking pair-breaking magnetic moments. We think that these principles carry over to high-temperature superconductors much more easily than some of the hypotheses that have been proposed assigning the supercurrent to cuprate planes.^{31,32} Evidence supporting this notion includes (i) $Ba_{1-a}K_aBiO_3$ is a 32-K superconductor with most of the properties of the superconducting cuprates,³³ except no cuprate planes; (ii) YSr_2RuO_6 doped with Cu is also reported to be an 80-K superconductor,³⁴ despite having no cuprate planes; and (iii) Cu-less WO_3 doped with Na is a 90-K superconductor.³⁵ Moreover, $LaBa_2Cu_2NbO_8$ does have cuprate planes but does not superconduct.³⁶ Therefore, cuprate planes are neither necessary nor sufficient for high-temperature superconductivity, and so there is no rational basis for assigning them a central role in the theory.

C. Additional evidence

Further evidence that the superconductivity of the R_{21-4} and $R_{222}Nb-10$ homologues originates in different parts of the crystal structures is provided by different crystal-field and size effects.

(i) The introduction of an $L=0$ ion (Gd^{3+} or Cm^{3+}) to a rare-earth site quenches the superconductivity in the R_{21-4} homologues,^{1,37,38} but not in the $R_{222}Nb-10$ structures.^{21,22}

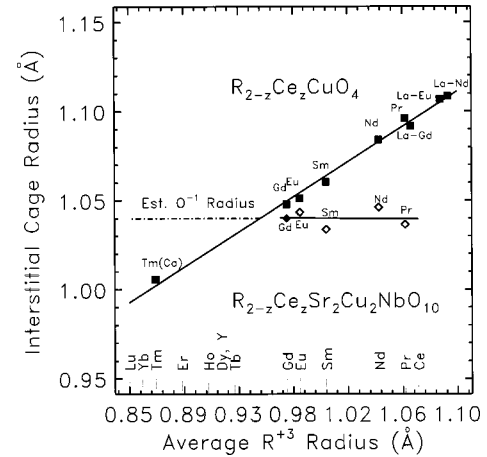


FIG. 2. Inscribed radius of the cage of oxygen ions (in a hard-sphere approximation) surrounding the interstitial oxygen site near the face center of the rare-earth plane in $R_{2-z}Ce_zCuO_4$ (slanted line, open squares) and $R_{2-z}Ce_zSr_zCu_2NbO_{10}$ (nearly horizontal line, diamonds) versus trivalent rare-earth radius (in Å). The $R = Tm$ datum (Ref. 41) corresponds to Ca doping, not Ce, and so is not constrained by a cage-size effect, as when the dopant is (Ce, interstitial oxygen). The solid diamond for Gd indicates that neutron-diffraction data are not available, but the material superconducts, and so we have assigned the material the average inscribed radius for the class. The data are for $R = La-Eu$ (Ref. 42), $La-Nd$ (Ref. 1), and $La-Gd$ alloys (Ref. 1), and for $Gd, Eu, Sm, Nd,$ and Pr (Ref. 6). The chained line is our estimate of the radius of an O^{-1} ion and lies within 0.01 Å of the radius of O^{-} interpolated between O^0 and O^{2-} . Note the size effect: none of the $R_{2-z}Ce_zCuO_4$ materials forms with Ce doping or superconducts for trivalent R ions smaller than Gd.

This difference is attributed to Cooper pair breaking by the rare-earth magnetic moments, which is allowed for $L=0$ but forbidden by crystal-field splitting of $L \neq 0$ ions in the R_{21-4} structures.^{30,39} The importance of crystal-field splitting in R_{21-4} homologues but not in $R_{222}Nb-10$ structures (although the fields near the rare-earth sites should be virtually the same) implies that the superconducting condensate is close to the rare-earth in R_{21-4} but remote in $R_{222}Nb-10$.

(ii) The mechanism of superconductivity in R_{21-4} homologues is controversial, with the only self-consistent proposal involving co-doping of Ce by interstitial oxygen.^{3,4} In this picture, there is a size effect:^{4,28,40} when the cage of ions surrounding the interstitial site becomes too small to contain O^{-1} , the superconductivity must vanish. This size effect, which forbids superconductivity for smaller rare-earth ions than Gd^{+3} , appears to be present in R_{21-4} materials, but not in $R_{222}Nb-10$ [Fig. 2 (Refs. 41 and 42)], again indicating that whatever entity is superconducting in R_{21-4} is different in $R_{222}Nb-10$.

(iii) The failure of $Pr_{222}Nb-10$ to exhibit bulk superconductivity²⁶ is assigned to pair breaking by Pr_{Sr} , analogous to the case of Pr_{123-7} (Ref. 43), which superconducts only after Pr_{Ba} is minimized.^{10,11,26} Again, no comparable effect is observed in Pr_{21-4} .

III. EXPERIMENTAL STATUS

Based on these ideas, (i) Gd_{21-4} intrinsically cannot superconduct, and (ii) $Pr_{222}Nb-10$ will produce bulk supercon-

ductivity only if the Pr_{Sr} defects are minimized.

Accordingly, we initiated searches for *granular* superconductivity in both compounds, and found none in Gd21-4 (as expected).⁴⁴ In Pr222Nb-10 we found evidence of *granular* superconductivity, together with neutron diffraction data indicating the presence of heavy concentrations of the pair-breaking defect Pr_{Sr} .²⁶ These observations are fully consistent with the ideas presented here: Pr_{Sr} is a Cooper-pair breaker, but rare-earth site Pr is not—implying that the primary superconducting condensate²⁷ cannot be in the cuprate planes, adjacent to the Pr site.

IV. $PrBa_2Cu_3O_7$ (Pr123-7)

Thus the facts for R222Nb-10 and R21-4 buttress the facts that Pr123-7 superconducts in the ideal crystal structure at a critical temperature approaching ≈ 90 K, but does not superconduct when significant amounts of Pr occupy Ba sites.^{10-17,43,45-47} The facts for Pr123-7 are contrary to the predictions of cuprate-plane theories, but *agree* with the predictions of a theory that places the primary superconductivity in the charge-reservoir layers, not in the cuprate planes.^{43,48-50} The reason that the Pr123-7 data are inconsistent with cuprate-plane superconductivity is that they arise from a fundamental asymmetry: Pr on a rare-earth site has no adverse effect on ≈ 90 -K superconductivity, but Pr on a Ba site destroys the superconductivity. If the superconducting condensate lies primarily in the cuprate planes, then a rare-earth site and a Ba site are located almost symmetrically with respect to the cuprate plane in between them (at almost the same distance from the plane). Since Pr has a magnetic moment, it is expected to break Cooper pairs in an adjacent superconducting condensate (Fig. 3). But the magnetic environments on both sides of a cuprate plane are essentially the same, and so the observation of asymmetric pair-breaking behavior, namely, Ba site Pr breaks pairs but rare-earth site Pr does not, is inconsistent with a symmetric magnetic environment—and can be explained only if the primary condensate is not in the cuprate plane, but in the charge-reservoir layer adjacent to the Ba site. Thus one is *forced* to adopt charge-reservoir superconductivity, not only by the many predictions of cuprate-plane theories that failed to be realized, but also by symmetry.

Many other experiments now require *p*-type charge-reservoir superconductivity as well. We mention only four: (i) In $Pb_2Sr_2Ca_{0.5}R_{0.5}Cu_3O_8$ no superconductivity occurs for $R=Ce$ or Am , which are tetravalent and so should dope the material *n* type, while trivalent (*p* type) $R=Pr$ and Tb superconduct—implying that the superconductivity lacks the expected particle-hole-doping symmetry implicit in most cuprate-plane theories. (ii) The chemical trends, both in the critical temperatures T_c and in the amount of Ni or Zn, u_c , required to quench superconductivity in the common high-temperature superconductors, as the cuprate planes become more distant from the charge-reservoir layers,⁵¹⁻⁵³ indicate

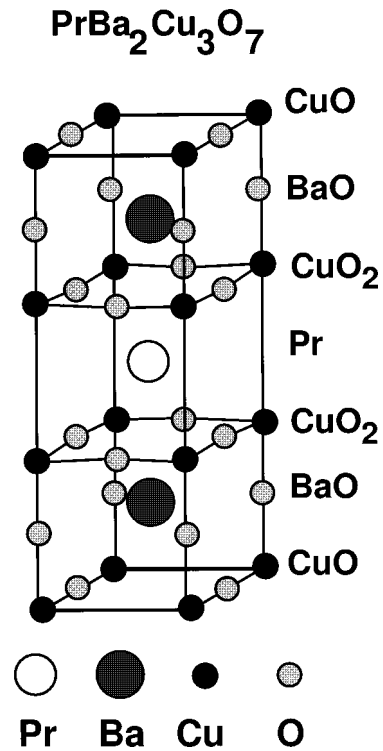


FIG. 3. Crystal structure of $PrBa_2Cu_3O_7$.

very clearly that the primary superconducting condensates lie in the charge-reservoir layers, not in the cuprate planes, (iii) The sizes of the (interstitial) dopant oxygen ions in superconductors such as $Tl_2Ba_2CaCu_2O_8$ indicate that they still have most of their holes attached, indicating that the holes have not left the doping layer for the cuprate planes.⁵⁴ (iv) By starting with the inconsistency that Ce in isolation cannot account for the doping in Nd21-4 and its homologues, one is naturally led to a model of doping by (Ce, interstitial O) pairs, *p*-type superconductivity, a lack of crystal-field splitting for the Gd and Cm homologues, and a size effect on superconductivity for rare-earth ions smaller than Gd^{+3} (Refs. 28 and 40)—all of which are observed.

V. SUMMARY

The evidence speaks for itself: the structural units responsible for superconductivity in $R_{2-z}Ce_zCuO_4$ and $R_{2-z}Ce_zSr_2Cu_2NbO_{10}$ are different. The widely accepted conventional cuprate-plane picture of high-temperature superconductivity requires revision.

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- ¹Y. K. Tao, M. Bonvalot, Y. Y. Sun, R. L. Meng, P. H. Hor, and C. W. Chu, *Physica C* **165**, 13 (1990).
- ²M. Brinkmann, T. Rex, M. Stief, H. Bach, and K. Westerhalt, *Physica C* **269**, 76 (1996).
- ³H. Okada, M. Takano, and Y. Takeda, *Phys. Rev. B* **42**, 6813 (1990).
- ⁴Not only do the $R_2\text{CuO}_4$ compounds fail to superconduct for trivalent rare-earth ions smaller than Gd^{3+} , but also significant Ce doping of these small-radius rare-earth compounds apparently has not been achieved [Y. Takeda (private communication)]. This is likely because $Z \geq 1$ is required for (Ce, O) pair formation. For $R = \text{Lu}$ and Yb , these compounds have not been fabricated yet. See H. Okada, M. Takano, and Y. Takeda, *Physica C* **166**, 111 (1990). We are unaware of fabrication of Tb_2CuO_4 , but $\text{LaTb}_{0.85}\text{Ce}_{0.15}\text{CuO}_4$ has been fabricated and does not superconduct: see, Y. Y. Xue, P. H. Hor, R. L. Meng, Y. K. Tao, Y. Y. Sun, Z. J. Huang, L. Gao, and C. W. Chu, *ibid.* **165**, 357 (1990); Y. K. Tao, Y. Y. Sun, J. Paredes, P. H. Hor, and C. W. Chu, *J. Solid State Chem.* **82**, 176 (1989).
- ⁵G. H. Hwang, J. H. Shieh, and H. C. Ku, *Physica C* **185-189**, 1163 (1991).
- ⁶J. T. Markert, J. Beille, J. J. Neumeier, E. A. Early, C. L. Seaman, T. Moran, and M. B. Maple, *Phys. Rev. Lett.* **64**, 80 (1990).
- ⁷Z. Fisk, J. D. Thompson, E. Zirngiebl, J. L. Smith, and S. W. Cheong, *Solid State Commun.* **62**, 743 (1987).
- ⁸P. Hor, R. L. Meng, Y.-Q. Wang, L. Gao, Z. J. Huang, J. Bechtold, K. Forster, and C. W. Chu, *Phys. Rev. Lett.* **58**, 1891 (1987).
- ⁹H. A. Blackstead, D. B. Chrisey, J. D. Dow, J. S. Horwitz, A. E. Klunzinger, and D. B. Pulling, *Physica C* **235-240**, 1539 (1994); *Phys. Lett. A* **207**, 109 (1995); H. A. Blackstead, J. D. Dow, D. B. Chrisey, J. S. Horwitz, P. J. McGinn, M. A. Black, A. E. Klunzinger, and D. B. Pulling, *Phys. Rev. B* **54**, 6122 (1996).
- ¹⁰A. I. Romanenko and L. P. Kozeeva, *Phys. Lett. A* **223**, 132 (1996).
- ¹¹Zhigang Zou, Kunihiko Oka, Toshimitsu Ito, and Yoshikazu Nishihara, *Jpn. J. Appl. Phys., Part 2* **36**, L18 (1997); Z. Zou, J. Ye, K. Oka, and Y. Nishihara, *Phys. Rev. Lett.* **80**, 1074 (1998).
- ¹²M. Łuszczek, W. Sadowski, and J. Olchowik, *The Fifth International Conference on Materials and Mechanisms of Superconductivity*, Abstract Book (Chinese Academy of Science, Beijing, 1997); see also M. Łuszczek, W. Sadowski, T. Kilmczuk, J. Olchowik, B. Susła, and R. Czajka, *Physica C* **322**, 57 (1999); W. Sadowski, M. Łuszczek, J. Olchowik, B. Susła, and R. Czajka, *Mol. Phys. Rep.* **20**, 213 (1997).
- ¹³K. Kadowaki (unpublished).
- ¹⁴T. Usagawa, Y. Ishimaru, J. Wen, T. Utagawa, S. Koyama, and Y. Enomoto, *Jpn. J. Appl. Phys., Part 2* **36**, L1583 (1997).
- ¹⁵T. Usagawa, Y. Ishimaru, J. Wen, T. Utagawa, S. Koyama, and Y. Enomoto, *Appl. Phys. Lett.* **72**, 1772 (1998) find that thin (0,0,1) films do not superconduct although (1,1,0) films do superconduct, an indicator of the importance of Pr_{Ba} defects and O(5) oxygen.
- ¹⁶J. C. Cooley, W. E. Hults, E. J. Peterson, and J. L. Smith, *Int. J. Mod. Phys. B* **12**, 3254 (1998).
- ¹⁷A. Shukla, B. Barbiellini, A. Erb, A. Manuel, T. Buslaps, V. Honkimäki, and P. Suortti, *Phys. Rev. B* **59**, 12 127 (1999).
- ¹⁸F. S. Galasso, *Structure, Properties, and Preparation of Perovskite-Type Compounds* (Pergamon, Oxford, 1969), p. 52.
- ¹⁹R. J. Cava, J. J. Krajewski, H. Takagi, H. W. Zandbergen, R. B. Van Dover, W. F. Peck, Jr., and V. Hsien, *Physica C* **191**, 237 (1992).
- ²⁰Rukang Li, Yingjie Zhu, Yitai Qian, and Zuyao Chen, *Physica C* **176**, 19 (1991); R.-k. Li, Y.-j. Zhu, C. Xu, Z.-y. Chen, Y.-t. Qian, and C.-g. Fan, *J. Solid State Chem.* **94**, 206 (1991).
- ²¹T. J. Goodwin, H. B. Radousky, and R. N. Shelton, *Physica C* **204**, 212 (1992). These authors have interpreted their results in terms of cuprate-plane superconductivity, contrary to our present work.
- ²²See also Shiwei Wang, Yitai Qian, Rukang Li, Zuyao Chen, Nanlin Wang, Liezhao Cao, Guien Zhou, and Yuheng Zhang, *Physica C* **210**, 463 (1993); I. Felner, U. Yaron, U. Asaf, T. Kröner, and V. Breit, *Phys. Rev. B* **49**, 6903 (1994); A. J. Wright, R. A. Jennings, and C. Greaves, *Supercond. Sci. Technol.* **6**, 514 (1993).
- ²³H. W. Zandbergen, R. J. Cava, J. J. Krajewski, and W. F. Peck, Jr., *Physica C* **196**, 252 (1992).
- ²⁴H. A. Blackstead, J. D. Dow, and D. B. Pulling, *J. Low Temp. Phys.* **105**, 705 (1996).
- ²⁵H. Ishii, T. Koshizawa, T. Hanyu, and S. Yamaguchi, *Jpn. J. Appl. Phys., Part 1* **32**, 1070 (1993).
- ²⁶Recently we have presented strong experimental evidence that $\text{Pr}_{22}\text{Nb}_{10}$ both exhibits *granular* superconductivity and contains a significant number of the expected pair-breaking defects P_{Sr} : See H. A. Blackstead, J. D. Dow, I. Felner, H.-h. Luo, D. B. Pulling, and W. B. Yelon, *Int. J. Mod. Phys. B* **12**, 3072 (1998).
- ²⁷B. V. Kresin, H. Morawitz, and S. A. Wolf, *Mechanisms of Conventional and High T_c Superconductivity* (Oxford, New York, 1993), pp. 103–127.
- ²⁸H. A. Blackstead and J. D. Dow, *Phys. Rev. B* **55**, 6605 (1997).
- ²⁹I. D. Brown and D. Altermatt, *Acta Crystallogr., Sect. B: Struct. Sci.* **B41**, 244 (1985); D. Altermatt and I. D. Brown, *ibid.* **B41**, 241 (1985); I. D. Brown, *J. Solid State Chem.* **82**, 122 (1989); I. D. Brown and K. K. Wu, *Acta Crystallogr., Sect. B: Struct. Crystallogr. Cryst. Chem.* **B32**, 1957 (1976); I. D. Brown, *Structure and Bonding in Crystals*, edited by M. O'Keefe and A. Navrotsky (Academic, New York, 1980), Vol. II, pp. 1–20.
- ³⁰H. A. Blackstead and J. D. Dow, *Physica C* **282-287**, 1005 (1997).
- ³¹R. J. Cava, A. W. Hewat, E. A. Hewat, B. Batlogg, M. Marezio, K. M. Rabe, J. J. Krajewski, W. F. Peck, Jr., and L. W. Rupp, Jr., *Physica C* **165**, 419 (1990).
- ³²Y. Tokura, J. B. Torrance, T. C. Huang, and A. I. Nazzari, *Phys. Rev. B* **38**, 7156 (1988).
- ³³The material is p type, has a Sommerfeld specific-heat characteristic of an oxide superconductor, does not superconduct unless the valence balance indicates hypocharged oxygen, and appears to undergo superconductor \leftrightarrow insulator transitions.
- ³⁴M. K. Wu, D. Y. Chen, F. Z. Chien, S. R. Sheen, D. C. Ling, C. Y. Tai, G. Y. Tseng, D. H. Chen, and F. C. Zhang, *Z. Phys. B: Condens. Matter* **102**, 37 (1997); and D. Y. Chen, F. Z. Chien, D. C. Ling, J. L. Tseng, S. R. Sheen, M. J. Wang, and M. K. Wu, *Physica C* **282-287**, 73 (1997) report evidence of superconductivity in $\text{YSr}_2\text{Ru}_{1-x}\text{Cu}_x\text{O}_6$ at ≈ 60 and ≈ 80 K, respectively.
- ³⁵S. Reich and Y. Tsabba, *Eur. Phys. J. B* **9**, 1 (1999).
- ³⁶M. Bennahmias, J. C. O'Brien, H. B. Radousky, T. J. Goodwin, P. Klavins, J. M. Link, C. A. Smith, and R. N. Shelton, *Phys. Rev. B* **46**, 11 986 (1992).
- ³⁷C. Lin, G. Lu, Z.-X. Liu, and Y. F. Zhang, *Physica C* **194**, 66 (1992).

- ³⁸L. Soderholm, C. W. Williams, and U. Welp, *Physica C* **179**, 440 (1991).
- ³⁹O. Peña and M. Sargent, *Prog. Solid State Chem.* **19**, 165 (1989).
- ⁴⁰H. A. Blackstead and J. D. Dow, *Phys. Lett. A* **26**, 97 (1997).
- ⁴¹W. J. Zhu, Y. S. Yao, X. J. Zhou, B. Yen, C. Dong, Y. Z. Huang, and Z. X. Zhao, *Physica C* **230**, 385 (1994).
- ⁴²H. Itoh, M. Kusuhashi, and M. Taniwaki, *Jpn. J. Appl. Phys., Part 2* **29**, L1604 (1990).
- ⁴³H. A. Blackstead and J. D. Dow, *Phys. Rev. B* **51**, 11 830 (1995).
- ⁴⁴H. A. Blackstead, J. D. Dow, D. D. Jackson, and D. B. Pulling (unpublished); measurement of surface resistance.
- ⁴⁵Reference 11 reports $T_c \approx 80$ K; Ref. 13 reports 85 K. Probably both are somewhat reduced from the typical $\text{YBa}_2\text{Cu}_3\text{O}_7$ homologue temperature of ≈ 93 K, due to pair breaking by a limited amount of Ba-site Pr.
- ⁴⁶Z. Zou, K. Oka, T. Ito, and Y. Nishihara, *Physica C* **282-287**, 481 (1997).
- ⁴⁷K. Oka, Z. Zou, and J. Ye, *Physica C* **300**, 200 (1998).
- ⁴⁸H. A. Blackstead and J. D. Dow, *Superlattices Microstruct.* **14**, 231 (1993).
- ⁴⁹H. A. Blackstead and J. D. Dow, *Proc. SPIE* **2696**, 120 (1996).
- ⁵⁰L. Soderholm, S. Skanthakumar, U. Staub, M. R. Antonio, and C. W. Williams, *J. Alloys Compd.* **250**, 623 (1997). The $\text{Pb}_2\text{Sr}_2\text{Pr}_{1-x}\text{Ca}_x\text{Cu}_3\text{O}_8$ structure for $x \rightarrow 0$ is the same as Pr123-7 in the Pr, cuprate-plane, and SrO layers, except that Sr replaces Ba. The Cu-O chains of Pr123-7 are replaced by two PbO layers and one monovalent Cu in $\text{Pb}_2\text{Sr}_2\text{Pr}_{1-x}\text{Ca}_x\text{Cu}_3\text{O}_8$ for $x \rightarrow 0$, which superconducts, invalidating models of nonsuperconductivity of Pr123-7, due to the Pr and the cuprate planes hybridizing.
- ⁵¹H. A. Blackstead and J. D. Dow, *Philos. Mag. B* **73**, 223 (1996).
- ⁵²H. A. Blackstead and J. D. Dow, *Philos. Mag. B* **74**, 681 (1996).
- ⁵³H. A. Blackstead and J. D. Dow, *Europhys. Lett.* **41**, 659 (1998).
- ⁵⁴H. A. Blackstead and J. D. Dow, *Solid State Commun.* **95**, 613 (1995).