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Convergence of Reciprocal-Lattice Expansions and Self-Consistent Energy Bands of NaF[†]

N. E. Brener

Department of Physics and Astronomy, Louisiana State University, Baton Rouge, Louisiana 70803

and

J. L. Fry Department of Physics, University of Texas, Arlington, Texas 76010

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A convergence study is made of the reciprocal-lattice expansion of linear-combination-ofatomic-orbitals integrals in NaF. Converged integrals are obtained by direct summation and by extrapolation after different numbers of terms. It is found that extrapolation procedures give simple accurate results and substantially reduce labor involved. Comparison is made with other methods for obtaining convergence. Converged integrals are used as a basis for a self-consistent Hartree-Fock-Slater energy-band calculation for NaF using methods previously developed. Some transition energies are presented and correlated with available optical data. Self-consistent charge densities in NaF differ only slightly from a linear combination of ionic charge densities, confirming the ionic nature of NaF.

I. INTRODUCTION

The linear-combination-of-atomic-orbitals (LCAO) method which was first proposed¹ in 1928 has only recently been developed to its full power. $^{2-11}$ While there are still some theoretical questions of its applicability to quantitative solidstate calculations, practical experience indicates that it not only is as accurate as other methods when done correctly, but also has great advantage in self-consistent calculations⁵ or problems involving Brillouin-zone sampling or computation of matrix elements using Bloch wave functions. However, when reading the literature of LCAO or tight-binding calculations one must be careful to distinguish between approximate LCAO calculations and ones which are performed with new techniques. Neglect or approximation of multicenter integrals and inclusion of only limited numbers of neighbors usually means that the energy bands are qualitative at best. Discussions of the problems encountered in accurate LCAO procedures are given in Refs. 6, 11, and 12.

In this paper special attention is paid to convergence properties of the reciprocal-lattice series expansion of LCAO integrals. Some of the methods used to facilitate summation of these series are discussed in Sec. II. In Sec. III, well-converged integrals for NaF are used to perform a self-consistent Hartree-Fock-Slater energy-band calculation.

II. CONVERGENCE OF LCAO INTEGRALS

The LCAO method employed here has been discussed elsewhere^{2,3,7} and will not be presented again. Accuracy in LCAO energy bands requires a critical analysis of all approximations made. In this section attention is focused upon convergence of reciprocal-lattice expansions of various LCAO integrals.

Reciprocal-lattice sums are a result of expressing the crystal potential as a Fourier series in order to circumvent calculation of three-center integrals. Approximation or complete neglect of the difficult three-center integrals and failure to sum enough direct-lattice neighbors invalidated early efforts to use the LCAO method in first-principles calculations. The Fourier-series technique permits accurate evaluation of LCAO integrals and produces accurate energy bands.

Introduction of Gaussian orbitals instead of Slater (exponential) orbitals greatly simplifies and accelerates the numerical work. Use of the Fourier series for the crystal potential and use of Gaussian basis functions for the atomic orbitals yield analytic expression for all matrix elements needed in evaluation of the LCAO secular determinant. However, the rate of convergence of the reciprocal-lattice sums for some LCAO integrals may be very slow, making it difficult to estimate the error after a given number of terms. In some instances convergence is so slow that it is essential to find ways of accelerating it in order to obtain the necessary accuracy in reasonable computer times. Examples of the sort of errors which may occur with inadequate convergence are given in Ref. 12.

A typical LCAO integral is expressed as

$$\int \phi_{i}(\vec{\mathbf{r}}) V(\vec{\mathbf{r}}) \phi_{j}(\vec{\mathbf{r}} - \vec{\mathbf{R}}_{l}) d^{3}r$$

$$= \sum_{K_{n}} V(\vec{\mathbf{K}}_{n}) \int \phi_{i}(\vec{\mathbf{r}}) e^{i\vec{\mathbf{K}}_{n}\cdot\vec{\mathbf{r}}} \phi_{j}(\vec{\mathbf{r}} - \vec{\mathbf{R}}_{l}) d^{3}r$$

$$\equiv \sum_{K_{n}'} V(\vec{\mathbf{K}}_{n}) E_{ij}(\vec{\mathbf{K}}_{n}, \vec{\mathbf{R}}_{l}) ,$$

where ϕ_i and ϕ_j are orbitals of given symmetries centered at the origin and at site \overline{R}_i , respectively.

The Gaussian atomic orbitals used for Na and F are given in Table I. $V(\vec{r})$ is the periodic crystalline potential and $V(\vec{K}_n)$ is its Fourier transform. The starting (first-iteration) crystal potential was constructed as a linear superposition of ionic potentials as discussed in previous publications.^{7,12}

Convergence of the reciprocal-lattice sum is governed by both $V(\vec{K}_n)$ and the integral $E_{ij}(\vec{K}_n, \vec{R}_i)$. The latter will depend upon the symmetry of orbitals involved, values of the orbitals exponents, separation of the orbitals, \vec{R}_i , and the value of \vec{K}_n . At large \vec{K}_n the dominant contribution to $V(\vec{K}_n)$ comes from the nuclear point charge

$$V(\vec{K}_n) \rightarrow -4\pi Z e^2/K_n^2$$

For large \vec{K}_n the integral $E_{ij}(\vec{K}_n, \vec{R}_i)$ is more complicated, depending upon types of orbitals involved, but all types include a factor $\exp[-\vec{K}_n^2/4(\alpha_1 + \alpha_2)]$, where α_1 and α_2 are exponents in the Gaussian orbitals ϕ_i and ϕ_j . For large $\alpha_1 + \alpha_2$ the \vec{K}_n dependence of $E_{ij}(\vec{K}_n, \vec{R}_i)$ is weak, and the rate of convergence is determined primarily by $V(\vec{K}_n)$ until $\vec{K}_n^2 > 4(\alpha_1 + \alpha_2)$. $V(\vec{K}_n)$ gives very slow convergence, and it is mainly this term which must be overcome. For some combinations of symmetry and

TABLE I. Gaussian orbitals for sodium and fluorine. The s and p orbitals were obtained from Huzinaga and Sakai,^a and d orbitals were generated by a Hartree-Fock calculation.

	Sodium wave functions						Fluorine wave functions				
					Orbital expo	nents	of Gaussian	basis set			
	S			Þ			d		S		Þ
1s	36631.	L	2p	148.928	;	3 <i>d</i>	6.00000	1s	23342.2	2 <i>p</i>	65.6593
1s	5385.0	07	2p	34.514	9	3d	1.45200	1s	3431.25	2p	15.2187
1s	1216.2	20	2p	10.604	5	3d	0.102959	1s	757.667	$\hat{2p}$	4.78819
1s	339.	529	2p	3.671	.06	3d	0.027170	1s	209.192	2p	1.72755
1s	109.5	563	2p	1.284	90	3d	0.009342	1s	66.7261	2p	0.648123
1s	38.7	7834	2p	0.430	941			15	23.370 5	2p	0.244965
1s	14.9	5790	2p	0.166	490			1s	8.62372	2p	0.091 537
1s	5.2	27901	2p	0.051	793			1s	2,700 01	Ĩ	
1s	1.8	32902	2p	0.018	420			1s	1.087 50		
1s	0.6	320 535						1s	0.396536		
1s	0.0	058065						1s	0.172324		
1s	0.0	024 617									
				Orbita	l expansion c	oeffi	cients of Gau	ssian basis s	set		
	1s	2_S		3 <i>s</i>	2p		3 <i>p</i>	3d	1_{S}	2s	2p
0	.000 38	-0.00009	0	.00001	0.005147	_	0.000 511	0.000166	0.00040	-0.00009	0.00882
0	.00305	-0.00074	0	.00011	0.037 920		0.003805	0.002705	0.00327	-0.00075	0.05778
0	.01599	-0.00396	0	.000 60	0.153743		0.015 526	0.069251	0.01754	-0.00409	0.19341
0	.064 27	-0.01613	0	.00241	0.352821	-	0.036470	0.451313	0.07082	-0.016 99	0.35671
0	.194 81	-0.05310	0	.00822	0.457887		0.047405	0.630163	0.21290	-0.05476	0.39296
0	.400 99	-0.12776	0	.01915	0.239507		0.040177		0.42213	-0.13195	0.22990
0	.39267	-0.19952	0	.03311	-0.000105		0.070060		0.37480	-0.18473	0.03143
0	.08282	0.03427	- 0	.01052	0.006753		0.494 273		0.05866	0.08175	
- 0	.00265	0.59803	-0	.098 98	-0.002405		0.551837		-0.00944	0.55115	
0	.00177	0.48579	- 0	.20686					0.00468	0.44502	
- 0	.000 31	0.01663	0	.59948					-0.00136	0.06182	
0	.00017	-0.00762	0	.48401							

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FIG. 1. Convergence behavior of reciprocal-lattice sums for various Na-Na first-neighbor integrals. The integrals were computed in rydbergs, but arbitrary constant shifts have been made to display them on the same graph. 3x is a 3p function transforming like x, while $3x^2$ and $3z^2$ are 3d functions transforming like x^2-y^2 and $3z^2-r^2$, respectively.

 α_1 and α_2 the \vec{K}_n dependence of $E_{ij}(\vec{K}_n, \vec{R}_i)$ is strong enough to produce rapid convergence in spite of $V(\vec{K}_n)$, but this is only occasionally the case, e.g., *d-d* integrals encountered in this paper. While a simple analytic expression for the *n*th term in the reciprocal-lattice sum may be written down for a single *s*-like Gaussian orbital, the atomic orbitals, which are linear combinations of Gaussian orbitals, are more easily understood graphically.

Figure 1 shows the convergence behavior of different symmetry-type integrals as a function of the number of stars of reciprocal-lattice vectors summed. The integrals have been shifted by arbitrary constant amounts to be displayed on the same graph. While all have converged to three significant figures, some have converged to considerably more. Integrals shown are Na-Na firstneighbor values, but similar results are obtained for F-F- and NaF-type integrals. The problems occur with 3s-3p and 3s-3d symmetries. Corecore integrals are even more slowly convergent, but the only important ones are central-cell (twocenter) integrals which may be handled without difficulty.

Figure 2 shows the behavior as a function of \vec{R}_i . Once again the integrals have been shifted by constant amounts. Note the significant figures indicating different degrees of convergence for different integrals. As separation of orbital sites becomes greater the rate of convergence increases, but only slightly in most cases. Convergence to fewer significant figures than indicated may be obtained more readily for larger \vec{R}_{1} .

The degree of localization of the orbitals also determines the rate of convergence. For Gaussian integrals this is indicated by the orbital exponents, larger parameters corresponding to greater localization. If atomic wave functions are used, as here, localization is roughly indicated by the principle quantum numbers of the atomic orbitals. Figure 3 shows these trends for some of the integrals which are slowly convergent.

The slowness of convergence for some of the integrals shown in Figs. 1–3 can be appreciated fully by reference to Table II, where selected integrals are given at different stages of summation. In order to gain understanding of these convergence properties the authors attempted to perform an analysis similar to the methods used by Euwema and Stukel to study orthogonalized-plane-wave (OPW) convergence in tetrahedral semiconductors. ¹³ Their work is of great pedagogical as well as practical value and is recommended reading for anyone interested in band theory.

Attempts which were made here to perform a similar study met with immediate failure. In the LCAO method the Fourier-series expansion is made for the potential (rapidly varying near the origin), not the wave function (well-behaved near the origin). Also, the variational theorem is not working on the coefficients of expansion of the potential as it is in the case of the OPW expansion of the wave function. As a result it is necessary to sum terms to much larger K_{max} , corresponding to much smaller R_m , as defined in Ref. 13. The dominant potential term is the Coulomb potential of the nucleus, which is singular at the origin. Thus, convergence of LCAO Fourier-series expansion is more similar to the classic Ewald problem, as has already been pointed out.³ The large number of terms necessary for convergence as indicated in Table II makes it clear that methods of accelerating convergence must be found if computer times are to remain reasonable.

Three different techniques have been used to obtain convergence: (i) an Ewald-type method, ³ (ii) integration, ¹⁰ and (iii) extrapolation. The Ewald method is described in Ref. 3. It uses an arbitrary auxilary potential function which falls rapidly to zero over a distance comparable to the lattice constant, but behaves near the nucleus like a point negative charge equal in magnitude to the nuclear charge. The Fourier transform of this function cancels the ill-behaved term in the Fourier transform of the crystal potential for large K_n . On the other hand, the short-range nature of the auxilary function permits approximation of its integral by one- or two-center terms, so that the part added to the Fourier series is subtracted out afterwards by direct integration. This method works only if the errors made in the two-center approximation

to the auxilary potential integrals is valid for the particular orbitals and auxilary function used. It may be quite accurate if care is taken with the direct integral.³

The technique of integration consists of summing the Fourier series to some maximum \vec{K}_n and approximating the sum of the remaining terms by



FIG. 2. Convergence behavior for different neighbors. Note the difference in scale indicating differences in convergence. The integrals have been shifted by arbitrary amounts.



FIG. 3. Convergence behavior of various orbital types. The integrals have been shifted by arbitrary amounts.

converting the sum to an integral. Errors arise in this method from both the integration approximation and the numerical integration technique itself, but accurate answers may be obtained.

The technique of extrapolation can be the simplest of the three, depending upon the extrapolation procedure used. It should be possible to develop



a Padé approximate to the series, but it would be different for each type of integral, and hence a substantial effort. A simple extrapolation method is obtained by observing that the sum after N stars is approximately a linear function of 1/N, and an approximation to the converged value may be obtained by straight-line extrapolation to 1/N=0 or $N=\infty$.

Figure 4 shows the asymptotic behavior of some of the integrals as a function of 1/N. Figure 4(a) displays straight-line behavior in the region 1000 to $10\,000$ stars, while 4(b) shows the same integrals in the region to 47000 stars. The slope is constant to more figures in Fig. 4(b), reflected by more significant figures in the extrapolated result. The values of various integrals obtained by extrapolating to 1/N = 0 after different numbers of reciprocal-lattice vectors (RLV) are shown in Table III. Even in the worst cases, extrapolation after summing 1000 RLV appears to be adequate for the accuracy needed in present-day band calculations. This represents a substantial savings in computer time, and further reductions would be realized with a nonlinear extrapolation procedure which would permit a smaller K_{max} .

III. SELF-CONSISTENT ENERGY BANDS

Self-consistent energy bands for NaF have been obtained using converged LCAO integrals as a basis. The self-consistent procedure used here has been discussed thoroughly elsewhere, 5,12 so it will not be elaborated again. Its simplicity rests upon the fact that it is not necessary to recompute (and

TABLE II. Convergence properties of some reciprocallattice sums for LCAO integrals. Integrals shown are all first-neighbor (110) Na-Na integrals. The last row shows estimated convergence after 47 000 stars. The last digit of the converged values is not expected to change upon further summation. $3z^2$ is a 3d function transforming like $3z^2-r^2$.

Stars (10 ³)	2s-2s (10 ⁻³ Ry)	3s-3s (Ry)	3s-3p _x (Ry)	$\frac{2s-3z^2}{(\mathrm{Ry})}$	3 <i>s</i> -3z ² (Ry)
1	-0.588304	-0.125768	0.110 359	0.160 352	0.040 444
5	-0.575256	-0.127074	0.111 401	0.156095	0.041093
10	-0.573571	-0.127 241	0.111 534	0.155 546	0.041176
20	-0.572718	-0.127 324	0.111 601	0.155 268	0.041 217
30	-0.572439	-0.127352	0.111623	0.155178	0.041 231
40	-0.572302	-0.127 365	0.111634	0.155134	0.041 238
	-0.572	-0.1273	0.1116	0.155	0.0412

reconverge) LCAO integrals at each stage of iteration, but iterated matrix elements are expressed in terms of initial matrix elements. To simplify the self-consistent calculation, only the following

TABLE III. Extrapolated first-neighbor Na-Na integrals. Column 2 shows values obtained by extrapolating with slope determined by values after 1000 and 2000 stars, column 3 for 2000-5000, and column 4 for 5000-10000stars. The last column shows convergence estimated by extrapolating from 47000 stars. All energies are in rydbergs.

Integral	1-2	2-5	5-10	Converged
$1s - 3z^2$	0.417783	0.418666	0.418813	0.4188
$2s - 3z^2$	0.155124	0.155000	0.154997	0.154999
$3s - 3z^2$	0.412419	0.412594	0.412583	0.412406
1s - 2x	-0,364335	-0.364148	-0.364093	-0.36409
2s - 2x	0.308 940	0.308 998	0.308 994	0,309002
3s - 2x	0.282079	0,282133	0.282130	0.282137
1s - 3x	0.670 637	0.672053	0.672289	0.6723
2s - 3x	0.251 493	0.251294	0.251289	0,251 29
3s - 3x	0.111661	0.111641	0.111 669	0.111 666
1s - 1s	-0.510354	-0.511 438	-0.511614	-0.5116
2s - 1s	-0.649373	-0.650742	-0.650 972	-0.6510
3s – 1s	-0.421083	-0.421972	-0.422120	-0.4222
2s - 2s	-0.572287	-0.571 897	-0.571885	-0.571 89
3s – 2s	-0.160 260	-0.160136	-0.160132	-0.160134
3s – 3s	-0.127374	-0.127409	-0.127407	-0.127406



FIG. 4. Extrapolation of reciprocallattice sums. (a) shows extrapolation from the region of 1000-10000 stars of RLV while (b) shows the region 30000 to 47000 stars of RLV. The scale to the right-hand side indicates the number of significant figures and the left-hand scale shows extrapolated values. Agreement between extrapolated values in (a) and (b) is good, one more significant figure being available from (b) (energy in rydbergs).



FIG. 5. Non-self-consistent NaF energy bands.

orbitals were included in the basis set:

Na: 1s, 2s, 3s, 2p, 3p,

F: 1s, 2s, 2p.

These orbitals lead to 14×14 Hamiltonian and overlap matrices. During self-consistent interations the Na 3s and 3p components played only a very minor role in determining the self-consistent potential. *d*-like orbitals are expected to play an even smaller role and were not included in the set. Other calculations¹⁴ indicate that d-electron levels do lie fairly low in the conduction bands of alkali halides and must be included for an accurate description of optical transitions much above the fundamental absorption edge. For this purpose dfunctions may be added after the self-consistent potential has been obtained. Since they have not been included in this calculation only the lowest conduction band may be considered reliable at this point.

Energy bands obtained from the starting crystal potential are shown in Fig. 5. The lattice constant used was 4.62 Å. Three values of the X_{α} exchange parameter are shown: $\alpha = 1$ Slater exchange, $\alpha = 1.06$, and $\alpha = 0.764$. The immediate conclusion to be drawn from these bands is that it was not possible to find an exchange parameter in the range $\frac{2}{3} \le \alpha \le 1$ in a non-self-consistent band calculation to yield agreement with the measured

 $gap^{15,16}$ of 0.85 Ry. While this was not the experience with LiF, similar behavior occured for other alkali halides in the work of Page and Hygh.¹⁴

An integration grid consisting of 89 points in $\frac{1}{48}$ of the fcc Brillouin zone was used to obtain selfconsistent charge densities and energy bands. Rearrangement of charge during the self-consistent iterations produced wider energy gaps for fixed α : Figure 6 shows self-consistent energy bands for α = 1.06. The computed gap 0.84 Ry is in good agreement with the observed gap of about 0.85 Ry.¹⁷ The valence bandwidth is 0.7 eV, which is comparable with the bandwidth obtained in LiF calculations, ^{12, 14, 18, 19} but smaller than the value quoted by Gout et al.²⁰ of 3.44 eV. This latter value should not be regarded as quantitatively meaningful because of approximations made in constructing the crystal potential and also approximations in application of the tight-binding method. These approximations probably account for the off-center maximum of the valence band in Ref. 20. The minimum energy gaps obtained in all band calculations reported here were direct and occurred at the Γ point of the Brillouin zone.

Experimental data which may be compared with the band structures obtained here consist of the fundamental absorption edge, ^{15,16} photoemission measurements, ²¹ and extreme ultraviolet absorption. ^{22,23} Ultraviolet transmission or reflection



FIG. 6. Self-consistent NaF bands without polarization corrections.

experiments²⁴ above 12 eV are limited by the transmission of LiF windows conventionally used on low-temperature vacuum sample holders, so only a small portion of the optical spectrum of NaF above the fundamental absorption edge has been measured. While the usual exciton is clearly evident in the data, location of the fundamental absorption edge is somewhat uncertain because it falls in, or near, the region of absorption by the LiF excitons, and absorption by the LiF windows becomes large. The fundamental absorption edge is estimated to be about 11.5 eV for NaF. The photoemission edge has been found to be about 10-12 eV in the work of Duckett and Metzger.²¹

Extreme ultraviolet spectra has been obtained²² using synchrotron-radiation sources, and an initial attempt to analyze such data has been made by Brown *et al.*²³ Transitions arising from the 2plevels of the sodium ion have been tabulated by Haensel *et al.*²² and may be correlated with the fundamental absorption. Assuming the first two peaks in the NaF spectra (labeled A and B in Ref. 22) correspond to excitons, the absorption edge may be estimated. A numerical value for the absorption edge is not given in Ref. 22, but inspection of the spectrum suggests that it is approximately 34 eV for the transition Na(2p) – Γ_1 .

The following comparisons may be made between theory and experiment: (i) The fundamental absorption edge could be fit; (ii) the computed absorption edge for the Na(2p) transitions was 33.3 eV compared to 34.0 eV from experiment; and (iii) the photoemission edge was computed to be 14.7 eV compared to 10-12 eV.

The position of the Na(2p) absorption edge depends strongly upon the exchange potential, even more than the fundamental edge, and better agreement with this edge could be obtained when α was not constrained to give exactly the fundamental edge. A similar result was obtained in LiF: For α = 0.87 an exact fit occurred to the fundamental gap, but the Li(1s) edge was missed by 5 eV. A slightly larger α gave reasonable agreement for both edges. Failure to gain exact agreement for both gaps when fitting one of them probably reflects inadequacy of the local-exchange-potential approximation. This conclusion need not be drawn at this time, however, as polarization corrections to the bands have not been computed nor is the experimental data, particularly assignment of edge positions, all that certain.

In Table IV some of the calculated transitions $(\alpha = 1.06)$ are compared with the data of Ref. 22. There appears to be some correlation between theoretical values of transition energies from the top valence band $(L'_3 \text{ and } X'_5)$ to the conduction band and the extreme ultraviolet absorption peaks given by Haensel *et al.*²² Since the top valence band is very

narrow, the peak structure due to these transitions should resemble the spectra observed from the narrow Na(2p) bands. However, it is not possible to make definite assignments without a complete calculation of $\epsilon_2(\omega)$. While it is tempting to assign peaks to symmetry points in the Brillouin zone, it has been found by detailed calculation²⁵ that this is not reliable, since peak structures usually arise from oscillator contributions throughout the zone and, e.g., transitions which are dipole forbidden at symmetry points may appear as a result. Extreme ultraviolet data is especially hard to analyze because it represents many possible interband transitions from different sets of bands with the same energy separations. Possible contributions in the 34-eV range for NaF are Na(2p) to low conduction bands and F(2s) to low conduction bands as well as transitions from the upper valence band to higher conduction bands. A further complication in identifying the structure is the assignment of exciton peaks. The peaks labeled A and B in the data of Haensel et al. fall at the same position in all the sodium halides and are tentatively identified by them as exciton peaks. Brown et al.²³ question whether any of the extreme ultraviolet structure may be identified as exciton peaks with any certainty, but Fong and Cohen²⁵ assign peaks A, B, and C to excitons in their $\epsilon_2(\omega)$ studies for NaCl.

IV. CONCLUSION

In Sec. III it was demonstrated that the converged LCAO integrals provided an adequate basis for self-consistent Hartree-Fock-Slater energy-band calculations. Using the same techniques as in the LiF calculation it was possible to obtain reasonable agreement with several pieces of experimental data. A complete analysis of the extreme ultraviolet absorption data for NaF requires the inclu-

TABLE IV. Comparison of computed fundamental-absorption energies with measured extreme ultraviolet absorption. Column 1 lables the fundamental transitions and column 4 shows tentative assignments of terminal levels of the extreme ultraviolet transitions from the Na(2p) band. All energies are in rydbergs. The labels of column 7 were given in Ref. 22. The peak labeled C may be an exciton, as concluded in Ref. 25.

F(2p)	Calculated Energy	ΔE	Exper Energy	iment: N ΔE	la(2 <i>p</i>) Label
$\Gamma_{15} \rightarrow \Gamma_1$	11.4	0	$34.0 \\ 35.5$	0.0	
$L'_3 \rightarrow L'_2$	17.0	5.6	39.2	5.2	D
$L'_3 \rightarrow L_1$	17.2	5.8			
$X'_5 \rightarrow X'_4$	18.1	6.7	41.6	7.6	Е
$\underline{X_5' \to X_1}$	20.0	8.6	46.3	12.3	F



FIG. 7. Change in charge density along the [100] axis produced by selfconsistency. Similar changes occur along other axes.

sion of *d*-like conduction bands and a complete calculation of $\epsilon_2(\omega)$. This will be the subject of a future work.

Before concluding, a few observations about the self-consistent charge distribution of NaF should be made. The starting potential was constructed from ionic charge distributions for Na⁺ and F⁻. Charge redistribution which occurred in the selfconsistent calculation for NaF is shown in Fig. 7. A net movement of electrons from the fluorine ion to the sodium ion has occurred. An approximate value for the amount of charge moved was obtained by performing a volume integral from the ions out to the points where the charge density went through zero. About 0.06 electron was removed from the fluorine ion and 0.09 was placed upon the sodium ion. The rest of the negative charge placed on the sodium ion came from the interstitial regions. The total amount of valence charge redistributed was about 0.09 electron, so that the change in charge

density of the valence electrons from the ionic to the self-consistent configuration was less than 10%, and therefore NaF is at least 90% ionic. This is in agreement with the LiF results of Drost and Fry.¹² Simple binding-energy calculations, ²⁶ based upon

an ionicity of one, work well for most alkali halides, including LiF and NaF. The covalent-bond theory of Phillips and Van Vechten also predicts NaF should be highly ionic, possessing a fractional ionicity of 0. 946, similar to other alkali halides.²⁷ On the other hand, there has been some indication from experiment that NaF may be different from other alkali halides,²¹ and electron-diffraction investigations²⁸ have led to the conclusion that NaF is partially covalent. Analysis²⁹ of Compton scattering data is not conclusive, but seems to favor an ionic potential over a neutral-atom potential. These data need to be further examined to test the validity of approximations used²⁸ to describe the Compton scattering event.

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PHYSICAL REVIEW B

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Effects of Domain Shapes on Second-Harmonic Scattering in Triglycine Sulfate

G. Dolino

Laboratoire de Spectrométrie Physique, Cédex 53 (38) Grenoble-Gare, France (Received 30 March 1972)

A general description is given of the second-harmonic scattering of light by domains in a ferroelectric crystal. In addition to the usual second-harmonic beam, one observes second-harmonic light in new directions. The angular pattern of scattered light depends upon domain shapes. Plane parallel or cylindrical domains give very different diffraction diagrams. An experimental observation of this phenomenon in triglycine sulfate is given, where the variation of the second-harmonic intensity is correlated to changes in the domain structure. The anisotropy of the domains's cross sections leads to an anisotropic diffraction pattern.

I. INTRODUCTION

The phenomenon of second-harmonic generation (SHG) of light is observed when an intense light beam interacts with a noncentrosymmetric material.¹ If this material is inhomogeneous, second-harmonic scattering (SHS) is produced. This was observed for the first time in NH₄Cl by Freund² and then in triglycine sulfate (TGS).^{3,4} The origin of this effect can be easily understood by considering the spatial modulation of nonlinear polarization by the domain structure. The effect of domains on the intensity of SHG was first observed by Miller on ferroelectric crystals of BaTiO₃.⁵

We have previously described the angular pattern of SHS which is observed on TGS crystals.⁴ Here we give new theoretical and experimental results of the effects of domain shapes on SHS. The discussion is centered on ferroelectric materials, but some results can also be applied to the case of twins.

Section II is a general theoretical discussion of SHS where the effects of surface and domains are separated. Perfect correlation in one direction (cylindrical domains) or in two directions (plane

parallel domains) leads to the characteristic scattering pattern. Particular attention is paid to the asymptotic behavior of the intensity for large scattering angles. In Sec. III new experimental results obtained at room temperature on TGS are presented. After recalling the main nonlinear optical properties of TGS we study the variation of intensity as a function of the scattering vector $\Delta \mathbf{k}$, the effect of the anisotropy of domain cross sections, and give some information on the effects of thermal treatments on domain structure and on SHS.

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II. THEORY

A. Ferroelectricity and SHG

When a light beam with an electric field \vec{E}_{ω} of frequency ω is propagating in a crystal, it produces a linear polarization $\vec{P}_{\omega} = \vec{\epsilon} \cdot \vec{E}_{\omega}$ and also a nonlinear polarization $\vec{P}_{2\omega} = (\vec{d} \cdot \vec{E}_{\omega}) \cdot \vec{E}_{\omega}$ which is observable if the light beam is intense enough.¹ $\vec{\epsilon}$ is the linear susceptibility tensor and \vec{d} the nonlinear susceptibility tensor. SHG can occur only in noncentrosymmetric crystals, which is the case inferroelectrics.

In the phenomenological theory of ferroelectricity