

Antiferromagnetic metallic state in doped manganites

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Electronic and magnetic properties are systematically investigated for doped manganites $R_{1-x}\text{Sr}_x\text{MnO}_3$ ($R=\text{La}_{1-z}\text{Nd}_z$) by changing the nominal hole concentration x and the averaged ionic radius of the perovskite A site (z), or the one-electron bandwidth W . In the heavily doped region ($x \geq 0.50$), we observed a layered-type (A -type) antiferromagnetic state with metallic conductivity ($\rho \approx 4 \times 10^{-4} \Omega \text{ cm}$ at 5 K for $\text{La}_{0.46}\text{Sr}_{0.54}\text{MnO}_3$). This observation suggests that the antiferromagnetic metallic (AFM) state is the alternative ground state for doped manganites, besides the ferromagnetic metallic (FM) state. [S0163-1829(98)51610-4]

The recent observation of colossal magnetoresistance¹ (the CMR effect) in the doped manganites has sparked a great amount of effort to understand the unusual electronic and magnetic properties of these materials. The most commonly studied manganites $R_{1-x}A_x\text{MnO}_3$, where R and A are the trivalent rare-earth and divalent alkaline-earth ions, respectively, have the distorted perovskite structure with three-dimensional networks of the MnO_6 octahedra. Its generic behavior of paramagnetic-to-ferromagnetic transition is understood within the framework of the double-exchange (DE) theory,²⁻⁴ which includes only the transfer integral t of the e_g electrons and the on-site exchange interaction (Hund's-rule coupling; J_H) between the itinerant e_g electrons and localized t_{2g} spins ($S=3/2$). To explain the colossal value of the MR for the system having small W , e.g., $\text{La}_{1-x}\text{Ca}_x\text{MnO}_3$,⁵ however, we need additional mechanisms for carrier localization above T_C as well as magnetic field release of the localization. Perhaps the most probable mechanism to supplement the DE model is the polaron formation originating from the Jahn-Teller (JT) instability of the MnO_6 octahedra.⁶

The doped manganites show much more multifarious properties in the heavily doped region ($x \geq 0.5$). In the case of $\text{La}_{1-x}\text{Ca}_x\text{MnO}_3$,^{5,7} the ground state changes with hole doping beyond $x \sim 0.5$ from the ferromagnetic metallic (FM) state to an antiferromagnetic insulating (AFI) state. Ramirez *et al.*⁷ insisted that the FM state and the charge-ordered state located above the AFI state are the essential cooperative states governed by the Hund's-rule coupling (J) and the JT energies. Recently, Kawano *et al.*⁸ have performed the neutron-diffraction measurements on $\text{Pr}_{1-x}\text{Sr}_x\text{MnO}_3$ ($x=0.50$) and $\text{Nd}_{1-x}\text{Sr}_x\text{MnO}_3$ ($x=0.55$), and found that these compounds exhibit a layered-type (A -type) antiferromagnetic spin-ordering, similar to the parent LaMnO_3 .

In this paper, we have investigated the electronic and magnetic properties for $(\text{La}_{1-z}\text{Nd}_z)_{1-x}\text{Sr}_x\text{MnO}_3$, especially for the heavily doped region, by changing the doping level x

and the averaged ionic radius (z). We can reduce the one-electron bandwidth W by partially substituting the smaller Nd^{3+} ions for the larger La^{3+} ions (increasing z ; *chemical pressure*).⁹ Even in the $\text{La}_{1-x}\text{Sr}_x\text{MnO}_3$ system with maximal W , further hole doping ($x \geq 0.54$) is found to alter the ground state from the FM state to the A -type antiferromagnetic state. Surprisingly, the compound remains metallic ($\rho \approx 4 \times 10^{-4} \Omega \text{ cm}$ at 5 K) even below $T_N (= 200 \text{ K})$, indicating that the ground state is an *antiferromagnetic metallic* (AFM) state. The AFM state is found to be distributed over the wide $x-z$ region adjacent to the FM state, indicating that the state is another ground state for the doped manganites. We discuss the AFM state in terms of the anisotropy of the e_g orbitals.

Crystals of $(\text{La}_{1-z}\text{Nd}_z)_{1-x}\text{Sr}_x\text{MnO}_3$ [$(x, z) = (0.40, 1.0), (0.44, 1.0), (0.46, 0.6), (0.46, 0.8), (0.48, 1.0), (0.50, 0.0), (0.50, 1.0), (0.52, 1.0), (0.54, 0.0), (0.54, 0.2), (0.54, 0.4), (0.54, 0.8), (0.54, 1.0), (0.56, 1.0), (0.57, 0.0), (0.60, 0.0)$ and $(0.60, 1.0)$] were grown by the floating-zone method at a feeding speed of 7–9 mm/h. A stoichiometric mixture of commercial La_2O_3 , Nd_2O_3 , SrCO_3 , and Mn_3O_4 powder was ground and calcined two times at 1350 °C for 24 h. Then, the resulting powder was pressed into a rod with a size of 5 mm $\phi \times 60$ mm and sintered at 1350 °C for 48 h. The ingredient could be melted congruently in a flow of O_2 . Large crystals, typically 4 mm in diameter and 20 mm in length, were obtained. Powder x-ray-diffraction measurements at room temperature and Reitveld analysis¹⁰ indicate that the crystals were almost single phase. A small amount of SrMnO_3 impurity phase, however, is inevitably introduced to the melt-grown crystal ingot, especially for the higher- x samples. Except for $\text{La}_{0.5}\text{Sr}_{0.5}\text{MnO}_3$ ($x=0.50, z=0.0$),⁹ the crystal symmetry is orthorhombic ($Pbnm$; $Z=4$).

We show in Fig. 1 temperature dependence of resistivity ρ and magnetization M for crystals of $\text{La}_{1-x}\text{Sr}_x\text{MnO}_3$ ($z=0.0$) at $x=0.50$ and 0.54. For four-probe resistivity measurements, the crystal was cut into a rectangular shape, typi-

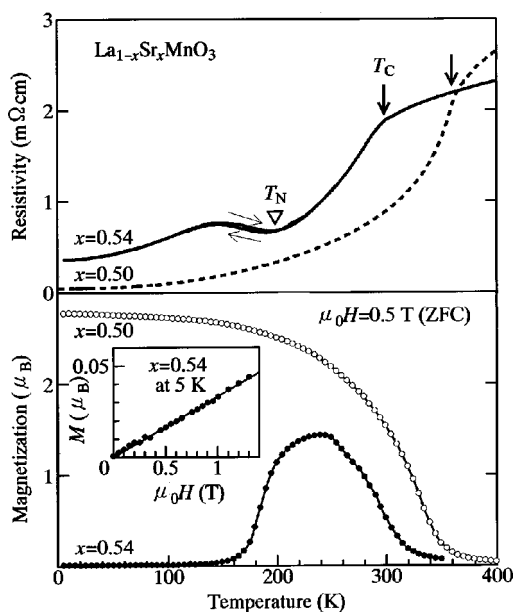


FIG. 1. Temperature dependence of resistivity (upper panel) and magnetization (lower panel) for crystals of $\text{La}_{1-x}\text{Sr}_x\text{MnO}_3$. Solid curves are for $x=0.54$ and broken curves for $x=0.50$. Arrows and triangles represent the Curie temperatures (T_C) and Néel temperatures (T_N), respectively. Inset shows magnetization curve at 5 K for $x=0.54$.

cally of $3 \times 1 \times 0.5 \text{ mm}^3$, and electrical contacts were made with a heat-treatment-type silver paint. M was measured under a field of $\mu_0 H = 0.5 \text{ T}$ after cooling down to 5 K in zero field (ZFC) using a superconducting quantum interference device (SQUID) magnetometer. Curie temperature T_C was determined from the inflection point of the $M-T$ curve. At $x=0.50$, the $\rho-T$ curve steeply decreases below T_C ($=360 \text{ K}$; downward arrow) down to $\sim 5 \times 10^{-5} \Omega \text{ cm}$ at 5 K. By contrast, the $x=0.54$ compound shows a ferromagnetic-to-antiferromagnetic transition at $T_N=200 \text{ K}$ (see the lower panel of Fig. 1), accompanying a slight increase of the ρ value. Nevertheless, the resistivity for $x=0.54$ remains low ($\sim 4 \times 10^{-4} \Omega \text{ cm}$ at 5 K) even in the antiferromagnetic phase, indicating that the hole-doping procedure beyond $x \sim 0.5$ alters the ground state for $\text{La}_{1-x}\text{Sr}_x\text{MnO}_3$ from the FM state to the AFM state. A similar AFM state is observed also for $x=0.57$ (not shown). For $x=0.60$, however, the ρ value steeply increases below $T_N=250 \text{ K}$.

To investigate the spin structure for the AFM state, neutron-diffraction measurements were performed with a high-efficiency powder diffractometer with a 150-detector system, HERMES, installed at the JRR-3M reactor in Japan Atomic Energy Research Institute, Tokai, Japan. Neutrons with wavelength 1.819 \AA were obtained by the (331) reflection of a Ge monochromator, and a combination of 12'-Blank-Sample-18' collimator. Melt-grown crystal ingots were crushed into fine powder and sealed in a vanadium capsule with helium gas, and mounted at the cold head of the closed-cycle He-gas refrigerator. The nuclear Bragg reflections in the paramagnetic state (330 K) are well reproduced with the orthorhombic structure ($Pbnm$; $Z=4$) with lattice constants of $a=5.4329(7) \text{ \AA}$, $b=5.4470(8) \text{ \AA}$, and $c=7.7556(9) \text{ \AA}$. Figure 2 shows the powder pattern for

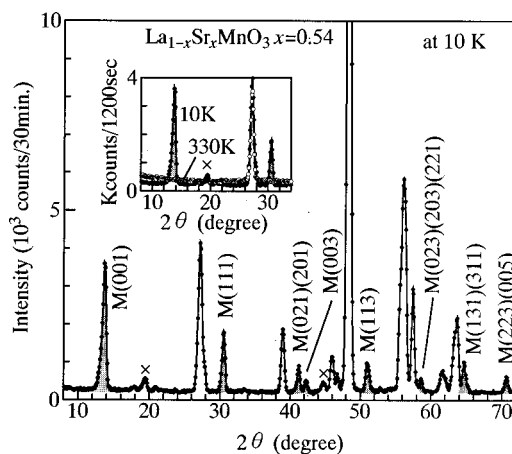


FIG. 2. Neutron powder profiles for $\text{La}_{0.46}\text{Sr}_{0.54}\text{MnO}_3$ ($x=0.54$) at 10 K. Shaded areas indicate magnetic superlattice reflections, which are indexed in the $Pbnm$ setting. Inset shows temperature dependence of the powder neutron profiles: open and closed circles are at 330 K ($>T_C$) and at 10 K ($<T_N$). Small impurity peaks (cross) are originated hexagonal SrMnO_3 .

$\text{La}_{0.46}\text{Sr}_{0.54}\text{MnO}_3$ ($x=0.54$, $z=0.0$) in the antiferromagnetic phase at 10 K. The observed magnetic superlattice reflections (indicated by hatching) can all be indexed by the A-type antiferromagnetic structure.¹¹ A similar spin ordering has been reported for $\text{Nd}_{0.45}\text{Sr}_{0.55}\text{MnO}_3$ ($x=0.55$, $z=1.0$),⁸ suggesting that the A-type structure is one of the most stable magnetic structures for the doped manganites.

Now, let us proceed to the lattice effect on the AFM state. Figure 3 shows the $\rho-T$ curves for $(\text{La}_{1-z}\text{Nd}_z)_{0.46}\text{Sr}_{0.54}\text{MnO}_3$ ($x=0.54$) with various z values. The $\rho-T$ curves were obtained in the cooling run. Downward arrows and open triangles represent T_C and T_N , respec-

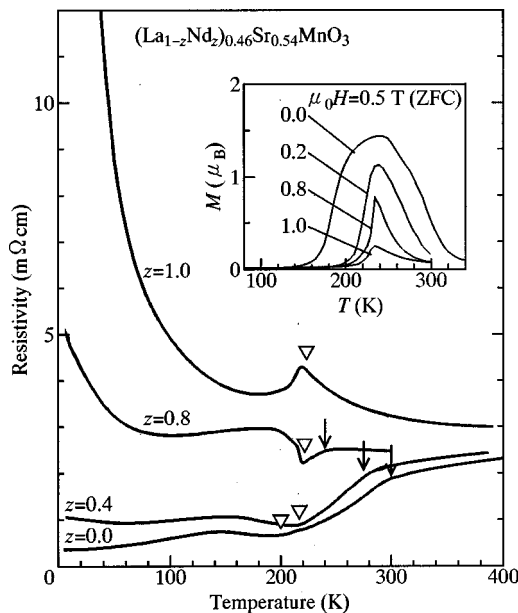


FIG. 3. Resistivity curves for crystals of $(\text{La}_{1-z}\text{Nd}_z)_{0.46}\text{Sr}_{0.54}\text{MnO}_3$ ($x=0.54$). Arrows and triangles represent the Curie temperatures (T_C) and Néel temperatures (T_N), respectively. Inset shows magnetization M under a field of $\mu_0 H = 0.5 \text{ T}$ after cooling down to 5 K in zero field (ZFC).

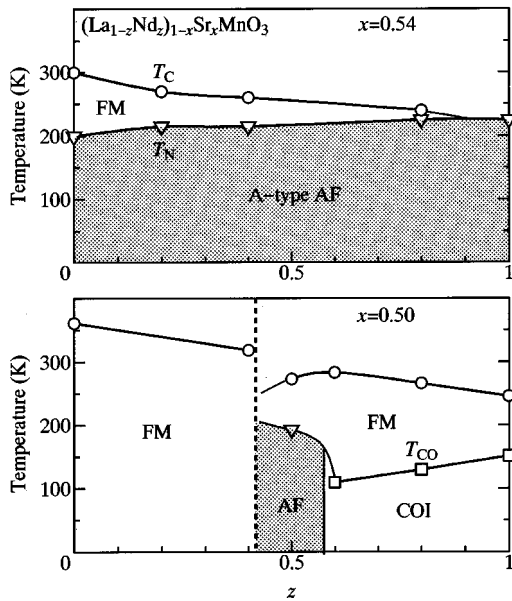


FIG. 4. Upper panel: electronic phase diagram for $(\text{La}_{1-z}\text{Nd}_z)_{0.46}\text{Sr}_{0.54}\text{MnO}_3$ ($x=0.54$). Lower panel: phase diagram for $x=0.50$ [$(\text{La}_{1-z}\text{Nd}_z)_{0.50}\text{Sr}_{0.50}\text{MnO}_3$] cited from Ref. 9. T_C , T_N , and T_{CO} represent Curie, Néel and charge-ordering temperatures, respectively. FM, AF, and COI means ferromagnetic metallic, antiferromagnetic, and charge-ordering insulating states, respectively. Vertical broken line in the lower panel is the structural phase boundary between the rhombohedral ($R\bar{3}c$; $z \leq 0.4$) and orthorhombic ($Pbmn$; $z > 0.4$) ones.

tively (see also the $M-T$ curves in the inset). With increasing z (decreasing W), the reduced carrier itineracy suppresses the DE interaction and hence T_C , and eventually the FM state disappears above $z=0.8$. The $\rho-T$ curve below $z=0.8$ shows a metallic behavior in the antiferromagnetic phase, except for the low-temperature region (≤ 60 K) where weak localization effect is observed. In the case for $z=1.0$, the curve shows a semiconducting behavior though the magnitude of ρ remains low ($\sim 10^{-1} \Omega \text{ cm}$ at 5 K).¹² The less-itinerant carriers, which cannot contribute to the macroscopic charge transport, can mediate the ferromagnetic DE interaction in a bond-percolative manner. (An insulating but ferromagnetic state exists near the insulator-to-metal phase boundary for $\text{La}_{1-x}\text{Sr}_x\text{MnO}_3$.¹³) A similar antiferromagnetic state is observed for $(\text{La}_{0.5}\text{Nd}_{0.5})_{0.5}\text{Sr}_{0.5}\text{MnO}_3$ ($x=0.5$ and $z=0.5$) (Ref. 9) and for $\text{Pr}_{0.5}\text{Sr}_{0.5}\text{MnO}_3$.^{8,14} Here we have found that the metallic conductivity remains down to the lowest temperature for the large- W compound ($z \leq 0.4$) even with the A-type spin structure. This suggests that the layered-type spin ordering is originated in the DE mechanism mediated by the mobile e_g electrons.

Thus obtained T_C and T_N for $x=0.54$ are plotted with circles and triangles in the upper panel of Fig. 4 as a function of z (or W). The apparent suppression of T_N with decreasing z (or increasing W) is due to the FM phase located at the high-temperature side, which suppresses the competing A-type AF state. In the whole z region, the ground state is the A-type AF state. Especially, the resistivity remains metallic ($d\rho/dT \geq 0$) down to the lowest temperature in the large- W region ($z \leq 0.4$). A similar highly-conductive AF state is observed for $x=0.50$ (lower panel; cited from Ref.

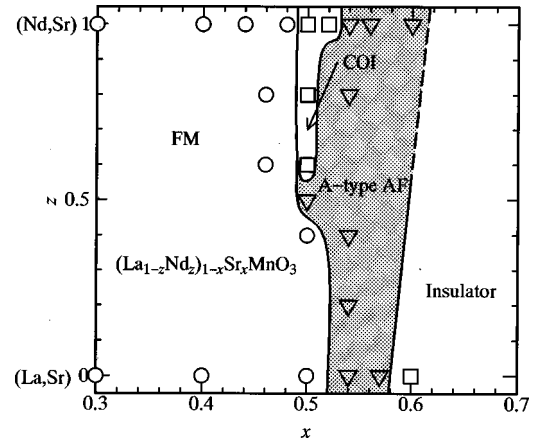


FIG. 5. Ground states for $(\text{La}_{1-z}\text{Nd}_z)_{1-x}\text{Sr}_x\text{MnO}_3$ as a function of the doping level x and the one-electron bandwidth W (or z). $(x,z)=(0.30, 0.0)$ and $(0.40, 0.0)$ are cited from Ref. 12. Circles, triangles, and squares represent the ferromagnetic metallic (FM), antiferromagnetic (AF), and insulating ground states, respectively. COI means charge-ordering insulating state.

9), even though the diagram is complicated partly due to the structural transition from rhombohedral ($R\bar{3}c$; $z < 0.4$) to orthorhombic ($Pbmn$; $z \geq 0.4$) ones. With further decreasing W , the AF state around $x=0.5$ is replaced by the charge-ordered insulating (COI) state. The spin structure in the COI state is a rather complicated one, with the $4 \times 4 \times 2$ unit cell in the cubic perovskite setting (CE type).¹⁵ In Fig. 5 thus obtained ground states for the doped manganites in the $x-z$ space are summarized. The most important message of the phase diagram is that the ground state changes from the FM state to the A-type AF state with hole doping beyond $x \sim 0.5$ nearly independent of z (or W). Near $x \approx 1/2$ and the small- W region, however, the COI state appears, reflecting the enhanced charge-ordering instability due to commensuration of the nominal hole concentration with the lattice periodicity. Here we should emphasize that the ground-state spin structure is of the layered type, and hence the state can be viewed as a two-dimensional (2D) FM state, while the usual FM state is viewed as a 3D FM state. With further increasing x , an insulating state appears, as exemplified in $\text{La}_{0.4}\text{Sr}_{0.6}\text{MnO}_3$ ($x=0.6$, $z=0.0$), which may correspond to the COI state observed in $\text{La}_{1-x}\text{Ca}_x\text{MnO}_3$.^{5,7}

The key factor that realizes the AFM state is considered to be the anisotropy of the two e_g orbitals, i.e., the two-dimensional $d_{x^2-y^2}$ state and the one-dimensional $d_{3z^2-r^2}$ state. Reflecting the anisotropy of the e_g orbitals, the electron transfer integral t_{ij} between the i and j sites strongly depends not only on the orbital species but also on the relative direction $\vec{r}_j - \vec{r}_i$. For example, t_{ij} between the $d_{x^2-y^2}$ states at $\vec{r}_j = \vec{r}_i \pm \hat{x}$ (\hat{y}, \hat{z}) is $3/4t_0$ ($3/4t_0, 0$), while t_{ij} between the $d_{3z^2-r^2}$ states at $\vec{r}_j = \vec{r}_i \pm \hat{x}$ (\hat{y}) is $1/4t_0$ ($1/4t_0$), where t_0 is defined as the transfer integral between the $d_{3z^2-r^2}$ states at $\vec{r}_j = \vec{r}_i \pm \hat{z}$. Therefore, the e_g -electron system gains the maximum kinetic energy when the $d_{x^2-y^2}$ orbitals form a pseudo-2D band. If such a pseudo-2D band was realized, the in-plane exchange interaction would be ferromagnetic mediated by the itinerant $d_{x^2-y^2}$ electrons (DE interaction), while the antiferromagnetic superexchange interaction should

dominate along the z direction because of negligible t_{ij} value. This can explain not only the layered-type antiferromagnetic spin structure but also the metallic conductivity. In fact, for $(\text{La}_{0.5}\text{Nd}_{0.5})_{0.5}\text{Sr}_{0.5}\text{MnO}_3$,⁹ elongation of lattice constants a and b as well as shrinkage of c is observed below T_N , which is advantageous to the formation of the pseudo-2D band. With decreasing x (see Fig. 5), or with increasing temperature (see Fig. 4), the orbital-ordered ground state changes into the 3D FM state. The most probable origin that destroys the 2D $d_{x^2-y^2}$ band is the JT instability inherent to the Mn^{3+} (d_4) ions, which lifts the degeneracy of the e_g orbitals and stabilizes the one-dimensional $d_{3z^2-r^2}$ state.

The A-type AF state, or the AFM state, is the fundamental ground state for the DE model with degenerated e_g orbital. Quite recently, Maezono *et al.*¹⁶ have calculated a global phase diagram for the DE model explicitly taking account of the degeneracy of the e_g orbital and the anisotropy of the transfer integral in the mean-field approximation. They observed a crossover of the ground state at around $x \sim 0.3$ from the ferromagnetic spin state to A-type AF spin state with the $d_{x^2-y^2}$ orbital structure, which shows excellent agreement with the experimentally derived phase diagram. The stabilization of the ferromagnetic state in the low x has been attributed to the superexchange interaction between the degenerated e_g orbitals. In an actual system, however, there exists the JT instability of the MnO_6 octahedra, which also favors the 3D FM state. The novel A-type state seems to be a fun-

damental ground state even for the doped manganites with the layered structure. Battle *et al.*¹⁷ have reported a layered-type AF state for $\text{NdSr}_2\text{Mn}_2\text{O}_7$ ($x=0.50$) and $\text{Nd}_{1.1}\text{Sr}_{1.9}\text{Mn}_2\text{O}_7$ ($x=0.45$). We have measured the resistivity for the melt-grown single crystals of these antiferromagnetic manganites, and found that the AF state is highly conductive. Study along this trend is now in progress and will be published in a separate paper.

In conclusion, we have investigated the electronic and magnetic properties for $R_{1-x}\text{Sr}_x\text{MnO}_3$ ($R=\text{La}_{1-z}\text{Nd}_z$) by changing the nominal hole concentration x and the one-electron bandwidth W (or z). We have observed transformation of the ground state from the ferromagnetic metallic (FM) to the antiferromagnetic metallic (AFM) state with x beyond ~ 0.5 . We propose that the AFM state is observed due to the formation of a pseudo-2D $d_{x^2-y^2}$ band. Our observation indicates that the e_g -orbital degree of freedom plays a crucial role in the ground state electronic structure for doped manganites.

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¹For example, S. Jin, T. H. Tiefel, M. McCormack, R. Fastnacht, R. Ramesh, and L. H. Chen, *Science* **264**, 13 (1994).

²P. W. Anderson and H. Hasagawa, *Phys. Rev.* **100**, 675 (1955).

³P.-G. de Gennes, *Phys. Rev.* **118**, 141 (1960).

⁴N. Furukawa, *J. Phys. Soc. Jpn.* **63**, 3214 (1994); **64**, 2734 (1995); **64**, 2754 (1995); **64**, 3164 (1995).

⁵P. Schiffer, A. P. Ramirez, W. Bao, and S.-W. Cheong, *Phys. Rev. Lett.* **75**, 3336 (1995).

⁶A. J. Millis, P. B. Littlewood, and B. I. Shraiman, *Phys. Rev. Lett.* **74**, 5144 (1994); **77**, 175 (1996).

⁷A. P. Ramirez, P. Schiffer, S.-W. Cheong, C. H. Chen, W. Bao, T. M. Palstra, P. L. Gammel, D. J. Bishop, and B. Zegarski, *Phys. Rev. Lett.* **76**, 3188 (1996).

⁸H. Kawano, R. Kajimoto, H. Yoshizawa, Y. Tomioka, H. Kuwahara, and Y. Tokura, *Phys. Rev. Lett.* **78**, 4253 (1997).

⁹Y. Moritomo, H. Kuwahara, Y. Tomioka, and Y. Tokura, *Phys. Rev. B* **55**, 7549 (1997).

¹⁰F. Izumi, in *The Rietveld Method*, edited by R. A. Young (Oxford University Press, Oxford, 1993), Chap. 13; Y.-I. Kim and F. Izumi, *J. Ceram. Soc. Jpn.* **102**, 401 (1994).

¹¹Outer side of the crystal ingot of $\text{La}_{0.46}\text{Sr}_{0.54}\text{MnO}_3$ is ferromagnetic possibly due to the reduced Sr concentration, which makes the Rietveld refinement of the powder pattern difficult.

¹²Low-temperature resistivity measurement indicates that $\text{Nd}_{0.45}\text{Sr}_{0.55}\text{MnO}_3$ is an anisotropic metal with finite conductivity at zero temperature [H. Kuwahara *et al.* (unpublished)].

¹³A. Urushibara, Y. Moritomo, T. Arima, A. Asamitsu, G. Kido, and Y. Tokura, *Phys. Rev. B* **51**, 14 103 (1995).

¹⁴Y. Tomioka, A. Asamitsu, Y. Moritomo, H. Kuwahara, and Y. Tokura, *Phys. Rev. Lett.* **74**, 5108 (1995).

¹⁵E. O. Wollan and W. C. Koehler, *Phys. Rev.* **100**, 545 (1955).

¹⁶R. Maezono, S. Ishihara, and N. Nagaosa (unpublished).

¹⁷P. D. Battle, M. A. Green, N. S. Laskey, J. E. Millburn, P. G. Radaelli, M. J. Rosseinsky, S. P. Sullivan, and J. F. Vente, *Phys. Rev. B* **54**, 15 967 (1996).