Evidence that all high-temperature superconductors are *p***-type**

Howard A. Blackstead

Department of Physics, University of Notre Dame, Notre Dame, Indiana 46556

John D. Dow

Department of Physics, Arizona State University, Tempe, Arizona 85287-1504

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The ground charge state of Ce substituting for Nd^{+3} in Nd_2CuO_4 is usually assumed to be Ce⁺⁴, indicating that Ce is a donor which causes $Nd_{2-z}Ce_zCuO_4$ to be an *n*-type superconductor. Using the self-consistent bond/charge method, we show that the Ce ground state is Ce^{+3} , not Ce^{+4} , in *ideal* Nd₂CuO₄, implying that monatomic Ce dopants are isoelectronic centers, not *n*-type donors. The observed Ce doping of $Nd_{2-z}Ce_zCuO_4$ can be understood if an interstitial oxygen binds to the Ce, elevating its charge state to Ce^{+4} . This is a paired defect of Ce and oxygen, which dopes the host *p*-type, not *n*-type, because of oxygen's 2*p* holes. The condition that the interstitial oxygen does dope the host *p*-type translates into an approximate steric constraint that the ionic radius of the rare-earth ion R^{+3} must exceed that of Tb⁺³ for R_{2} _zCe_zCuO₄ to superconduct. A similar situation occurs for the infinite-layer material $Sr_{1-n}Nd_vCuO₂$, where the Nd dopant is isoelectronic Nd⁺², not donor Nd⁺³ as assumed, unless an extra oxygen is adjacent to the Nd. The data are reviewed in light of these results, and show no convincing evidence either of particle-hole symmetry in doping, or of *n*-type doping of any high-temperature superconductor. Instead they hint that the superconductivity is associated with oxygen, rather than with any particle-hole-symmetric cuprate-plane phenomenon. $[$ S0163-1829(97)01209-5 $]$

I. INTRODUCTION

An important issue concerning high-temperature superconductivity is whether all such superconductors are *p*-type or not. The models that rely on oxygen as the generator of superconductivity, whether cuprate-plane $oxygen^{1,2}$ or charge-reservoir $oxygen$,³ assume that the current carriers are the 2*p* holes of oxygen. Such models would require major modification to be compatible with *n*-type superconductivity by electrons: there is no particle-hole symmetry in an oxygen model of superconductivity, because O^{-1} and O^{-2} naturally form in ceramics much more easily than O^{+1} or O^{+2} . The more traditional models based on cuprate planes⁴ generally do not provide a clear and specific picture of the assumed remote source of carriers, but merely treat the source as a parameter to be specified independently of the superconductivity, allowing the carriers to be either electrons or holes. The models themselves typically have particle-hole symmetry, with the *theoretical* cuprate planes that generate the superconductivity being equally compatible with *n*-type and *p*-type carriers. To our knowledge, no single compound has exhibited both *n*-type and *p*-type high-temperature superconductivity with different dopants, although this doping behavior is expected of theories with particle-hole symmetry. Moreover, all assignments of *n*-type superconductivity are controversial, including that for the most studied candidate, $Nd_{2-z}Ce_zCuO_4$. The evidence is such that one can credibly argue that *no* high-temperature superconductor exhibits the particle-hole symmetry common to so many theories, both because none of the main candidates for *n*-type superconductivity also exhibits *p*-type behavior with another dopant and also because all the *n*-type candidates themselves are controversial.

The main candidates for *n*-type superconductivity are R_{2-z} Ce_zCuO₄, for the lighter, magnetic rare earths *R*=Pr, Nd, Sm, and Eu, 5^{-8} and the infinite-layer compounds $Sr_{1-n}R$ ⁿCuO₂ for $R=$ La, Pr, Nd, Sm, Gd, and Y.^{9–12} However there is substantial evidence that defects play a central role in infinite-layer superconductivity, 13 and the case for *n*-type behavior is rather weak. For example, although $Sr_{1-v}Nd_{v}CuO_{2}$'s chemical formula suggests *n*-type doping,¹² Smith *et al.* found the Hall coefficient of $Sr_{1-n}Nd_{n}CuO_{2}$ to be *positive*, normally interpreted as strong evidence of *p*-type superconductivity. Therefore, except to make a few comments near the end of this paper about the infinite-layer materials, we shall restrict our attention to materials which assume the T' crystal structure, namely homologues of Nd_2CuO_4 (Fig. 1).

In addition to the Ce-doped compounds R_{2} _zCe_zCuO₄ with $R=Pr$, Nd, Sm, and Eu, three Th-doped homologues also superconduct, with Th⁺⁴ assuming the role of Ce^{+4} . \Pr_{2-z} Th_zCuO₄, Nd₂_{$-z$}Th_zCuO₄, and Sm₂_{$-z$}Th_zCuO₄ (with $T_c \approx 2$ K).¹⁴ A closely related material is Ca-doped $\text{Tr}_{2-z}\text{Ca}_{z}\text{CuO}_{4}$,⁹ which exhibits the *T'* crystal structure of $Nd_{2-z}Ce_{z}CuO_{4}$, is *p*-type, and superconducts.¹⁵ In all of these superconductors, the optimal doping composition *z* is typically in the range $0.13-0.17$. Except for $\text{Im}_{2-z}\text{Ca}_{z}\text{CuO}_{4}$, all of these compounds that superconduct are thought to be *n*-type electron superconductors for the same reasons that $Nd_{2-z}Ce_zCuO_4$ is. Six closely related materials, which have been fabricated in the T' crystal structure but do not superconduct, are Ce-doped $Gd_{2-z}Ce_zCuO_4$ (Refs. 5–8 and 14) and $\text{LaGd}_{1-z}\text{Ce}_{z}\text{CuO}_{4}$, ¹⁶ Th-doped $\text{Eu}_{2-z}\text{Th}_{z}\text{CuO}_{4}$ and $Gd_{2-z}Th_zCuO_4$, ¹⁴ and Ca-doped $Gd_{2-z}Ca_zCuO_4$ and $Dy_2-z^2Ca_2CuO_4$.⁹ A number of compounds, including $Nd_{2-z-a}Ce_{z}R_{a}CuO_{4}$, for $R=Gd$, Y, and La, have also been

FIG. 1. Crystal structure of (ideal) $Nd_{1.85}Ce_{0.15}CuO₄$ in the T' structure. The approximate possible locations of any (extra) interstitial oxygen ion are denoted ''*X*''. When Ce substitutes for Nd, atoms move slightly from their ideal sites.

studied, 8 with Gd generally being unfavorable to superconductivity.

The main purpose of this paper is to contribute to the discussion of whether the prototypical *n*-type superconductor, $Nd_{2-z}Ce_zCuO_4$, is actually *n*-type or if it is *p*-type.

II. IS $Nd_{2-z}Ce_zCuO_4n$ **-TYPE?**

The *n*-type characterization of Nd_{2} _zCe_zCuO₄ has been controversial: (i) Mangelschots $et al.¹⁷$ observed positive normal-state thermopowers in Nd_{2} _zCe_zCuO₄ samples which superconducted at lower temperatures. The sign of the thermopower is an indicator of the sign of the charge carrier (at least for a single-band free-electron gas undergoing energy-independent scattering¹⁸). (ii) Jiang *et al.*¹⁹ concluded that $Nd_{2-z}Ce_zCuO_4$'s transport has multiband character, with contributions from both electrons and holes, and Mao *et al.*²⁰ observed a virtually symmetric current-voltage characteristic for a heterostructure composed of indisputably *p*-type $YBa_2Cu_3O_7$ with purportedly *n*-type $Nd_{2}Z-Ce_zCuO_4$. (One might have expected some evidence of a *p*-*n* junction to remain even though the doping levels of these materials are high by semiconductor standards.)

A. Charge state of Ce

Completely absent from previous discussions of the *n*-type or *p*-type character of a dopant such as Ce is the question, "In $Nd_{2-z}Ce_zCuO_4$, is Ce^{+3} or Ce^{+4} the ground state of the ion?'' Obviously, if the ground state of Ce in Nd_2CuO_4 is Ce^{+3} , it does not dope its host *n*-type; Ce^{+3} merely replaces Nd^{+3} , and is an isoelectronic center rather than the expected *n*-type donor. An unexpected doping character of an impurity is common in semiconductors, where, for example, oxygen replacing P in GaP is not a donor (as expected), but instead is neither a donor nor an acceptor, tending to make the GaP semi-insulating by virtue of a deep electron and hole trap oxygen produces in the band gap.²¹ It is well known in semiconductor physics that the doping character of an impurity depends on its host. For example, in PbTe, anion-site In is a donor, but in SnTe the same defect is an acceptor. 22 A similar situation occurs in superconductors, where one cannot simply guess that an impurity is a donor because it sometimes assumes a particular charge state, as in the case of Ce^{+4} . Indeed, we shall present evidence that Ce doping of ideal $Nd_{2-z}Ce_zCuO_4$ leads to the formation of the isoelectronic defect Ce^{+3} , not the donor Ce^{+4} —namely that *Nd-site monatomic Ce does not liberate its fourth electron and dope (ideal)* Nd_2CuO_4 *n*-*type*, as is widely assumed.

B. The actual dopant is interstitial oxygen paired with Ce

We propose that Ce doping normally results in formation of a ~*Ce, interstitial-oxygen*! *pair* consisting of substitutional Ce on a Nd site (actually slightly distorted from the ideal Nd site) and an adjacent interstitial oxygen (Fig. 1). Such interstitials have been reported in various experiments, $17,23-26$ and imply inhomogeneous superconductivity.^{27,28} For the Madelung potential at a Ce site to reach the ionization potential for Ce^{F4} , there must be an interstitial oxygen adjacent to the Ce and bound to it. Thus the dopant is not actually monatomic Ce, but *Ce plus oxygen*, and this paired defect ideally²⁹ liberates a net of one hole: one electron as Ce undergoes the transition Ce⁺³→Ce⁺⁴ plus two holes as oxygen becomes O^{-2} . Thus the ideal pair is a *net p-type dopant, liberating one hole, not one electron*.

C. The self-consistent bond/charge approach

In attempting to determine the ground state of monatomic Ce in Nd_2CuO_4 , we first refer to an early self-consistent-field local-density-approximation calculation for the electronic structure of *T*-crystal-structure $\text{La}_{2-\beta}\text{Sr}_{\beta}\text{CuO}_4$ with $\beta=0.0$, 0.5, and 1.0, implemented using the spin-polarized pseudofunction method. 30 That calculation produced the following significant results: (i) the holes of the doped material were predicted to lie in the La_{1-(β /2)}Sr_(β /2)O layers, not in the cuprate planes; (ii) the predicted hole location of that theory is in excellent agreement with both the recently developed selfconsistent bond/charge model and with recent measurements;^{31,32} (iii) the doping ($\beta \neq 0$) does not alter the band structure much, but does alter the Fermi energy somewhat—an amount of the same order as is assumed in a rigid-band approximation; (iv) although no comparable calculations for Nd_{2} _z- Ce_7CuO_4 were executed using the pseudofunction method, other local-density calculations³³ reported that the electronic structures of Nd_2CuO_4 and La_2CuO_4 near the Fermi surface are extremely similar; and (v) the calculated magnetic moment on the Cu in La_2CuO_4 was localized and in excellent agreement with the measured value.³⁰ The result that the holes lie in the (La, Sr) -O layers is the opposite of what was once widely assumed: that the holes transferred from the (La, Sr)-O charge-reservoir layers and doped the cuprate planes *p*-type. However, the cuprate planes in $La_{2-\beta}Sr_{\beta}CuO_4$ are now known to be *n*-type,^{31,32} and the *p*-type hole-doped regions of the crystal structure are the (La, Sr) -O layers. Only these *p*-type (La, Sr) -O regions (and not the *n*-type cuprate planes) can produce p -type superconductivity.

TABLE I. Computed ionic charges (in units of $|e|$), magnitudes of Madelung potentials (in V), and ionization potentials IP (in V) (Ref. 37) for ideal Nd_{2z} , Ce_zCuO_4 (with $z=0.15$), and for the ideal infinitelayer material $Sr_{1-n}Nd_nCuO_2$ (with $v=0.08$). The numbers in parentheses with the ionization potentials are the charges of the final states of ionization. The ionization potentials (in V) to the R^{+3} charge state are, for *R*=La, Nd, Pr, Sm, Gd, and Y, 19.2, 22.1, 21.6, 23.4, 20.6, and 20.5, respectively. When (Ce, interstitial oxygen) pairs are introduced into Nd_{2} _zCe_zCuO₄, the Madelung potential at the Ce site becomes sufficient to ionize Ce to Ce⁺⁴ (>36.8 V). The charge on the interstitial oxygen is -1.16 |e|, both from the volume of the oxygen cage containing it, and from the bond-valence sum method $(Refs. 39–42)$, provided the Cu ion which has the interstitial at its apex is displaced away from the interstitial by the experimentally required distance of between 0.5 and 1.0 Å (Ref. 24). A displacement of \sim 0.5 Å meets both the steric and charge constraints. The Ce also relaxes ~ 0.2 Å away from its adjacent cuprate plane (Ref. 25).

Ion	$Nd_{2-z}Ce_{z}CuO_{4}$			$Sr_{1-n}Nd_nCuO_2$	
	Charge	Potential	IP	Charge	Potential
Nd	$+3.09$	-29.1	$22.1(+3)$, $40.4(+4)$	$+2.14$	-16.6
Ce	$+3.23$	-29.1	$20.2(+3)$, $36.8(+4)$		
Cu	$+1.84$	-19.3	$20.3(+2)$, $36.3(+3)$	$+1.81$	-18.7
Sr			$11.0(+2)$, $42.9(+3)$	$+2.14$	-16.7
O (cuprate plane)	-1.82	$+23.3$		-1.98	$+25.2$
O (O ₂ layer)	-2.21	$+23.7$			

The success of the pseudofunction electronic-structure calculations for $La_{2-\beta}Sr_{\beta}CuO_4$, combined with various successes of the easier-to-implement self-consistent bond/charge theory and the similarity of Nd_2CuO_4 to La_2CuO_4 , provides justification for applying the self-consistent bond/charge approach to the problem of Ce doping of Nd_2CuO_4 : (i) The effective charge Q_{σ} on the σ th ion is computed from its bond lengths $\left|\mathbf{r}_{\sigma',g}\right|$ using the phenomenological bond-valencesum method.^{34,35} With bond lengths $|\mathbf{r}_{\sigma',\sigma}|$, the relationship between the bond lengths and charges is

$$
|Q_{\sigma}| = |e| \sum_{\sigma'} w_{\sigma'} \exp\{[R_{\sigma} - |r_{\sigma',\sigma}|]/B\},\
$$

where we have $B=0.37$ Å, $|e|$ is the proton's charge, the sum is over all ions σ' directly bonded to the σ th ion,³⁶ *w*_{σ'} is the probability that the site is occupied by a particular atom of type σ' , and $|\mathbf{r}_{\sigma',\sigma}|$ is the bond length between the atom on site σ and an atom of type σ' at one of its neighboring sites. The parameters R_{σ} are extracted from known bond lengths of many chemical compounds and are tabulated in Ref. 34. (ii) With these charges, the Madelung potential $V(\mathbf{r}_{\sigma})$ at each site \mathbf{r}_{σ} is computed by summing over all ions v:

$$
V(\mathbf{r}_{\sigma}) = \sum_{\nu} Q_{\nu} / |\mathbf{r}_{\sigma} - \mathbf{r}_{\nu}|.
$$

(iii) The Madelung potential is compared with the ionization potentials of the ion on that site, 37 to determine if the potential is strong enough to ionize the ion to the expected charge state.³⁸ If the magnitude of the Madelung potential is weaker than the ionization potential by more than \sim 1 V, then in all likelihood the ion on that site is not ionized to the higher charge state. Only if the various ionic charges obtained from the bond lengths are nearly equal to the charges deduced from the ionization potentials do we accept the computed charge states. Otherwise we start over, selecting different integral ionic charges as input for the bond-length analyses, until we find a self-consistent solution.

1. Application to $Nd_{2-z}Ce_zCuO_4$

We applied this method to $Nd_{2-z}Ce_zCuO_4$, and found that the charge state of Ce is $Ce^{+3.2\overline{3}}$, and that the magnitude of the Madelung potential at the Ce site is \sim 29.1 V, compared with 36.8 V required to ionize Ce to Ce^{+4} . See Table I; (Refs. 37, 39–42). This \sim 7.7 V difference is so large that it is unnecessary to do a more sophisticated calculation, especially since both the computed charge $(+3.23|e|)$ and the potential $(-29 V)$ self-consistently indicated that, at the Ce site, Ce is Ce^{+3} , is an isoelectronic center, and is not a dopant of any type. In particular, Ce is not an *n*-type dopant. In *ideal* Nd_2CuO_4 , Ce replacing Nd is in the charge-state Ce⁺³, and does not account for the observed superconductivity.

Experimentally, Ce in Nd_2CuO_4 yields superconductivity, each Ce donates about one electron to a Cu-related band while leaving the typical Cu ion in the Cu^{+2} ionic state, 43 and Ce is nonmagnetic¹⁴ (i.e., closed-shell Ce^{+4}). The only way that the electrostatic point-ion potential at a Ce site can be made large enough to ionize Ce to Ce^{+4} is if an additional anion is placed close to the site—an oxygen interstitial. There is space available in the unrelaxed lattice to accommodate hypocharged oxygen (not fully charged to O^{-2}) at or near the ideal, unrelaxed interstitial site, which is at the facecenter of the Nd plane and \approx 2.3 Å distant from the nearest Ce site. $39-41$ This oxygen could meet the steric constraints even without local lattice relaxation if its charge $-Z₁|e|$ were such that Z_1 <1.16,⁴⁴ making the interstitial oxygen small enough to fit in the cage of surrounding oxygen ions. This interstitial charge would contribute an additional potential of magnitude $(6.5 \text{ } Z_I)$ V at the Ce site—compared with the 7.7 V needed to achieve Ce^{+4} . Thus, for reasonable interstitial charges $-Z_I|e|$ (and with reasonable lattice relaxation), 24.25 the potential at the Ce site is likely to be strong enough to ionize Ce to Ce^{+4} . Likewise, the charge at the Ce site will increase by $\sim |e|$ due to the inclusion of the bond between the interstitial and the Ce, while the Cu charge will remain near $\pm 2|e|$. Thus the Ce charge assumes a value around $+4|e|$, making the (Ce, interstitial-oxygen) defect a dopant that self-consistently satisfies the conditions of the bond/charge method, with Ce^{+4} . We executed calculations for Nd_{2} _z $Ce_{z}CuO_{4}$ containing interstitial oxygen, and found that this is indeed the case.⁴⁵

Thus the self-consistent bond/charge method offers two possible solutions for $Nd_{2-z}Ce_zCuO_4$: (i) Ce is isolated, in the charge-state Ce^{+3} , and electrically inactive, neither doping n -type nor causing superconductivity; and (ii) Ce is accompanied by interstitial oxygen, contributes its fourth electron to doping, and is Ce^{+4} . The Ce^{+4} solution is the one relevant to superconductivity in $Nd_{2-z}Ce_zCuO_4$, but leaves the material p -type, because interstitial (initially neutral) oxygen ideally has two $2p$ holes which it can release.²⁹ In the ideal weak defect-host coupling limit, the dopant is a (Ce, O) pair: ideally $Ce^{+4} - Nd^{+3} + e^{-} + O^{-2} + 2h^{+}$. Consequently the (Ce, interstitial-oxygen) defect complex can be a net *acceptor*, doping Nd₂CuO₄ p-type. Previous work *as*sumed that the Ce^{+4} observed was associated with an *isolated* defect, and thus that Ce's ionized electron doped the cuprate planes *n*-type.^{23,43,46} The self-consistent bond/charge scheme shows that the relevant defect is not isolated Ce, which is an isoelectronic center and a neutral defect (Ce^{+3}) rather than a donor, as assumed. It also indicates that the defect complex $(Ce,$ interstitial oxygen) produces p -type superconductivity in the vicinity of the interstitial and in the nearby O_2 layer, as discussed in Ref. 31.

III. IMPLICATIONS

Having concluded that Ce in $Nd_{2-z}Ce_zCuO_4$ forms a (Ce, interstitial-O) complex that is a p -type dopant, it is straightforward to draw similar conclusions for both the Ce-doped and the Th-doped homologues.

A. Oxygen solubility and interstitial cage size

A major factor governing the occurrence of superconductivity in the R_2CuO_4 homologues is the Ce-induced doping by interstitial oxygen. Therefore the solubility of interstitial oxygen in the cage of surrounding oxygen ions becomes an issue, as does the question of whether the cage is large enough to accommodate an oxygen ion with the required charge. The highest values of the critical temperature are obtained when the superconducting R_{2-z} Ce_zCuO₄ compounds are fabricated under high pressure, $4^{\frac{3}{7}}$ consistent with the expectation that the solubility of oxygen in the interstitial cage can be increased by pressure.⁴⁸

The size of the interstitial cage determines a condition for *p*-type doping and superconductivity. The cage of oxygen ions surrounding the interstitial oxygen must be large enough to accommodate a superconductivity-sensitizing interstitial oxygen ion that has acquired a charge $-Z_I|e|$ with $Z_I>1$, namely an oxygen that has yielded more than one hole to the host (to both compensate the electron from dopant Ce and dope the host p -type). An examination of the crystal structure of $Nd_{2-z}Ce_zCuO_4$ reveals that the sphere inscribed in the interstitial cage^{39,40,49} is sterically incapable of containing an oxygen ion of charge $-Z_l|e|$ if $Z_l > 1.16$, assuming an oxygen ionic radius linearly interpolated between those of O^0 and Q^{-2} : $(0.65+0.375 Z_I)$ Å.⁴⁴ Consequently the range of interstitial oxygen charges compatible with *p*-type doping

and superconductivity $(1 \leq Z_I \leq 1.16)$ is small even in $Nd_{2-z}Ce_zCuQ_4$ and smaller yet for the heavier rare-earth homologues.^{50,51} Since the radius of the inscribed sphere decreases approximately linearly with the decreasing ionic radius⁵² of the trivalent rare-earth ion of the host, the R_{2-z} Ce_zCuO₄ homologues cease to superconduct once the rare-earth ionic size becomes too small for the interstitial cage to accommodate the required Z_I $>$ 1 oxygen ion. From an examination of the lattice constants,⁴⁹ the $Z_I = 1$ condition is met for a rare-earth ionic radius near Tb^{+3} 's, namely, slightly smaller than Gd^{+3} 's. This trend with size is visible in the data: T_c decreases from \approx 32 K in pressure-fabricated Nd_{2} _z- $Ce_{z}CuO_{4}$ (Ref. 47) (21 K for normal fabrication⁵⁰) to ≈ 7.5 K for $Eu_{2-z}Ce_zCuO_4$,⁵⁰ as the oxygen cage size decreases, and vanishes for Gd_{2} _zCe_zCuO₄ and R_{2} _zCe_zCuO₄ with smaller trivalent rare-earth ions *R*, although the failure of Gd_{2} _z $Ce_{z}CuO_{4}$ to superconduct is probably not due to a size effect, but to its total angular momentum $L=0.53-56$ For doping by Th⁺⁴, a slightly larger dopant than Ce^{+4} , the onset of R_{2-z} Th_zCuO₄ superconductivity corresponds to a slightly larger trivalent rare-earth radius, between those of $Eu⁺³$ and $\rm{Sm}^{+3.14}$

B. Particle-hole symmetry

If particle-hole symmetry is valid, and if interstitial oxygen is involved in the doping as we propose, then the Ce^{+4} -doped and Th⁺⁴-doped *R*₂CuO₄ compounds must not *cease* superconducting for R^{+3} smaller than Gd⁺³ (as they do), but must switch to n -type superconductivity, as the interstitial oxygen's charge becomes $Z_I < 1$.

C. Ca-doped Tm_2CuO_4

The Ca-doped Tm_{2} _zCa_zCuO₄ appears to be another manifestation of the *p*-type R_2CuO_4 compounds¹⁵—except that the Ca doping is by an isolated defect (Ca) on a Tm site), rather than by a defect pair (Ce on a rare-earth site, paired with interstitial oxygen). This significant difference between the Ca-doped $Tm_{2}ZZ}Ca_ZCuQ_4$ and the Ce-doped $Nd_{2}Z_{z}Ce_{z}CuO_{4}$ implies that the *p*-type doping, not the interstitial oxygen itself, is critical for superconductivity in the R_2CuO_4 compounds.¹⁵ The Ce-doped compound is expected to have roughly one interstitial oxygen ion per Ce, while the interstitial oxygen concentration in the Ca-doped material is likely to be much smaller because there is no strongly positive Ce^{+4} ion attracting the negative oxygen. Thus, we expect that T_c of Tm_{2} _zCa_zCuO₄ should not be depressed by a lack of greater interstitial-oxygen solubility; this appears to be the case for this pressure-fabricated superconductor.⁹ T_c =30 K, compared with 32 K for Nd_{2} _zCe_zCuO₄. Moreover the effect of replacing Cu by Ni and Zn in Ca-doped Tm_2CuO_4 and Ce-doped Nd_2CuO_4 will be rather different: In $Nd_{2-z}Ce_zCuO_4$, less than 1% Ni destroys the superconductivity, while about six times as much Zn is required to accomplish the same effect.⁵⁷ This is because the interstitial oxygen is apical to the impurity site and directly bonded to it, and the supercurrent in $Nd_{2-z}Ce_zCuO_4$ passes through this interstitial site. Therefore the superconductivity of Nd_{2} _z- Ce _zCuO₄ senses the adjacent magnetic moment of Ni, and T_c is depressed rapidly by the Cooper pair-breaking effect of that magnetic moment. In contrast, we expect very little interstitial (apical) oxygen in $Tm_{2}Zm_{2}CuO_{4}$, and hence any superconducting condensate should primarily occupy the O_2 planes, where the strongly electropositive Ca will insert the holes. 31 This means that, to a good approximation, the supercurrent, which does not involve interstitial oxygen, will not experience exchange scattering and Cooper pair breaking by Ni on a Cu site (because the distance from the impurity to the nearest hole-carrying oxygen exceeds the range of the exchange scattering)—and hence that Ni and Zn should have similar effects on T_c due to their similar longranged scattering potentials. Studies of T_c *versus* Ni and Zn dopant concentrations *u* for the two materials should exhibit this dramatic difference: $T_c(u)$ for Ni and Zn should be about the same in superconducting $Tm_{2-z}Ca_{z}CuO_{4}$, but $T_c(u)$ for Ni is smaller than for Zn in $Nd_{2-z}Ce_zCuO_4$ and vanishes for a considerably smaller value of \vec{u} .⁵⁷

IV. CONCLUSIONS

Our main conclusion is that isolated Nd-site Ce in ideal Nd_{2} _z-Ce_zCuO₄ is Ce⁺³ and does not provide *n*-type doping, as widely assumed. Rather, most Ce ions facilitate the formation of an adjacent interstitial oxygen, and the (Ce, interstitial-oxygen) pair causes Ce to ionize to Ce^{+4} , while the oxygen dopes Nd_2CuO_4 net *p*-type. A dramatic test of this picture would be the demonstration that $T_c(u)$, as a function of Ni or Zn dopant concentration *u*, would be the same for Ni and Zn in superconducting $Tm_{2}Zc_{2}CuO_{4}$, which is not the case for Ni- and Zn-doped $\text{Nd}_{2-z}\text{Ce}_{z}\text{CuO}_{4}$.⁵⁷

A similar story holds for the infinite-layer compounds that are thought to be *n*-type. As Table I shows, the predicted ground charge state of Nd on a Sr site in $Sr_{1-v}Nd_vCuO_2$ is

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 Nd^{+2} , not Nd^{+3} . Thus Nd does not dope $Sr_{1-y}Nd_vCuO_2$ at all, much less *n*-type, and off-site oxygen or a cation vacancy must be involved in its superconductivity, which must be *p*-type.

In summary, the analyses presented here indicate that $Nd_{2-z}Ce_{z}CuO_{4}$ and its *T*'-crystal-structure homologues are all *p*-type. They also suggest that the $Sr_{1-v}R_vCuO_2$ infinitelayer compounds are not doped *n*-type by isolated rare-earth ions *R* either, because those ions appear unlikely to achieve the R^{+3} charge state in isolation.

Thus one can propose that there are *no n*-type hightemperature superconductors, which, if true, points toward superconductivity by hypocharged $oxygen₅₈$ rather than particle-hole-symmetric cuprate-plane superconductivity, as the general cause of high-temperature superconductivity. This, of course, means that Cu-less $Ba_{1-a}K_aPb_{1-b}Bi_bO_3$ can be reclassified as a high-temperature superconductor,⁵⁹ although discovered by Sleight, Gillson, and Bierstedt⁶⁰ long before the discovery of high-temperature superconductivity by Bednorz and Müller.⁶¹

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- 38These superconductors are low-density metals, but we do not include screening, which requires a different calculation for each charge density, because it does not have a major effect on our results, and because it is zero in the insulating and low-doping limits.
- ³⁹The lattice constants of $\text{Nd}_{2-z}\text{Ce}_z\text{CuO}_4$ are $a=3.94$ Å and $c=12.1$ Å. The positions of the atoms in the unit cell are Cu= $(0, 0)$ 0, 0) or $(0.5, 0.5, 0.5)$; cuprate-plane oxygen= $(0, 0.5, 0)$ or $(0, 0.5)$ 0.5, 0.5), O₂-layer oxygen= $(0, 0.5, 0.25)$, and Nd= $(0, 0, 0.5)$ 0.3523) or (0.5, 0.5, 0.1477). [See T. Kamiyama, F. Izumi, H. Takahashi, J. D. Jorgensen, B. Dabrowski, R. L. Hitterman, D. G. Hinks, H. Shaked, T. O. Mason, and M. Seabaugh, Physica C **229**, 377 (1994).
- 40We assume that the interstitial oxygen will reside approximately at the center of the cage of neighboring oxygen ions, about midway between the O_{2} - and cuprate-plane layers, at $(0.5, 0.5,$ 0.375 .
- ⁴¹ Steric considerations alone dictate that the Ce⁺⁴ ion be at least 0.1 Å closer to the interstitial oxygen than the Nd^{+3} it replaced was (because of the difference in ionic radii). The Coulomb force between the interstitial oxygen and the Ce^{+4} causes the Ce to relax even more toward the interstitial, but x-ray data (Ref. 25), indicate that this accounts for less than another 0.1 Å. Therefore we estimate the location of the Ce to be at $(0.5, 0.5, 0.16)$.
- 42We estimate that the interstitial oxygen pushes the Cu away by \sim 0.5 to \sim 1.0 Å in order to meet steric constraints, while allowing the Cu to remain Cu^{+2} and maintaining a Cu-interstitialoxygen bond length that approximates the sum of ionic radii.
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