## **Experimental determination of**  $\Gamma$ **-X intervalley transfer mechanisms in GaAs/AlAs heterostructures**

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Zone-center–zone-boundary ( $\Gamma$ -*X*) intervalley transfer mechanisms in AlAs/GaAs heterostructures are deduced in an unambiguous way from transport and electroluminescence studies of single-AlAs-barrier diodes. We demonstrate that the tunneling is strongly sequential and depends on the nature of the *X* state involved. For  $X<sub>z</sub>$  the transfer is mainly elastic, whereas momentum conserving phonon assistance is dominant for  $X<sub>xy</sub>$ . Quantitative values for the scattering rates for each transfer mechanism are obtained. The conduction-band offset from the  $\Gamma$  minimum in GaAs to the *X* minimum in AlAs is shown to be only  $120 \pm 6$  meV.  $[$ S0163-1829(96)52436-7 $]$ 

The exact mechanisms governing  $\Gamma$ -*X* intervalley transfer in AlAs/GaAs heterostructures are still subject to controversy. In bulk materials the main intervalley scattering process is zone-edge phonon scattering. By contrast, in a heterostructure the potential modulation along the growth axis induces mixing of the  $\Gamma$  and  $X$  electronic states and direct transitions can take place. The importance of such mixing has been demonstrated from studies of optical transition lifetimes in type-II superlattices<sup>1,2</sup> (SL's) and from observations of  $\Gamma$ -*X* anticrossing.<sup>3</sup> Real-space electron transfer in such SL's, where electrons are created in the GaAs layers and transfer to *X* in the AlAs, has been studied by time-resolved photoluminescence  $(PL)$ . Feldman *et al.*<sup>4</sup> proposed that it is either due to LO-phonon scattering or to state mixing, depending upon the width of the layers. However, Deveaud *et al.*<sup>5</sup> found transfer rates incompatible with state mixing but in good agreement with phonon-assisted transfer.  $\Gamma$ -*X* transfer also plays a significant role in perpendicular transport through single or double AlAs tunnel barriers,<sup>6,7</sup> tunneling through excited  $X$  states being demonstrated.<sup>8,9</sup> The results were explained by a state mixing theory,  $10$  but only imperfectly.

These strong uncertainties are due to the intrinsically poor spectroscopic resolution of the experimental techniques, which access the integrated transfer and cannot resolve individual mechanisms. In this paper we propose an original continuous wave technique achieving very high spectroscopic resolution while still providing results of high time resolution. It permits the nature of the different transfer channels [elastic or inelastic, involving longitudinal  $(X<sub>z</sub>)$  or trans-

verse  $(X_{xy})$ ]<sup>1–3,11,12</sup> to be precisely evaluated in indirect-gap single-barrier GaAs/AlAs/GaAs tunneling structures. In addition, it enables us to determine generalized  $\Gamma$ -*X* transfer rates which are in good agreement with earlier time-resolved studies. $4,5$ 

We study a series of single-barrier diodes, designed to have simple two-dimensional  $(2D)$  emitter states Fig. 1(a). The structures are then very similar to type-II SL's, with continuously variable  $\Gamma$ -*X* separation. With increasing bias, the opening of a new transport channel, when the emitter Fermi energy is resonant with an excited *X* state, produces a sudden increase of the differential conductance  $(\sigma)$ . The number, position, and amplitude of these thresholds in  $\sigma$ versus bias allows the  $\Gamma \rightarrow X$  transport mechanisms and their strengths to be identified. Complementary electroluminescence (EL) measurements are employed to determine the distribution of the *X* point electrons in the AlAs layer. Recombination of hot electrons which are ejected out of the  $barrier^{13}$  is also observed and permits the nature of the  $X \rightarrow \Gamma$  transfer to be determined.

The samples studied are *p*-*i*-*n* GaAs/AlAs/GaAs heterodiodes, grown by molecular-beam epitaxy, with three different AlAs layer widths of 60, 80, and 100 Å. They were processed into  $200$ - $\mu$ m-diameter mesas and consist of the following layers:  $0.5-\mu m$   $n=2\times10^{18}$  cm<sup>-3</sup> GaAs buffer, 500-Å  $n=1\times10^{17}$  cm<sup>-3</sup> GaAs, 500-Å  $n=3\times10^{16}$  cm<sup>-3</sup> GaAs emitter, 50-Å undoped GaAs spacer, undoped AlAs barrier, 50-Å undoped GaAs spacer, 0.5-mm *p*  $=1.10^{17}$  cm<sup>-3</sup> GaAs collector, and 0.5- $\mu$ m *p*  $=1.10^{18}$  cm<sup>-3</sup> GaAs top contact. The current versus voltage  $(I-V)$  and  $\sigma$ -*V* curves at 2 K are presented in Figs. 2 and 3



FIG. 1. (a) Schematic band diagram of the structure under forward bias. (b) Schematic dispersion of the 2D emitter and barrier states, showing the elastic and phonon-assisted transfers.

for the three samples. A number of steplike features are observed, each one arising from the opening of a new  $\Gamma$ -*X* transfer channel. Modeling of the conduction-band profile and of the electronic states is required to identify these features.

The first step in the modeling is a self-consistent solution of Poisson and Schrödinger equations, within the envelope function approximation, as a function of charge density  $(n<sub>s</sub>)$  in the 2D electron and hole accumulation layers (EAL, HAL). There was assumed to be no charge accumulation in the AlAs layer. We performed current versus magnetic field  $(B||I)$  measurements (not presented here) for various biases. The  $1/B$  periodicity of the Shubnikov–de Haas–like oscillations in  $I - B$  provides a precise determination of  $n<sub>s</sub>$  for each



FIG. 2. Measured *I*-*V* characteristics for the three samples (dots) and theoretical current (solid lines) from the model parameters described in the text.

*V*. In addition, the linear variation observed between *n<sub>s</sub>* and *V* confirms the 2D nature of the emitter states.

The anisotropic conduction-band structure of the AlAs layers was included in the modeling. $1-3,11,12$  The opposing effects of quantum confinement and residual uniaxial strain in the AlAs layer $12$  lead to the ground electron state being  $X_{xy}$ -like for layers thicker than 70 Å, and  $X_{z}$ -like for thinner layers. The  $X<sub>z</sub>$  and  $X<sub>xy</sub>$  electronic levels and envelope functions were calculated<sup>14</sup> as a function of applied bias. In electric field the overlap of the *X* wave functions with collector states will be much greater than with emitter states. Thus, transfer out of the barrier is expected to be much more efficient than transfer from the emitter into the barrier.  $\Gamma$  to *X* transfer into the AlAs is then the current limiting process, and will govern the shape of the *I*-*V* characteristic.

We now consider the transfer mechanisms which can occur. Unlike bulk materials, elastic transitions are possible since  $\Gamma$  and  $X$ <sub>z</sub> states are mixed by the interface potential, and to a lesser extent  $\Gamma$  and  $X_{xy}$  are mixed by in-plane potential fluctuations. From PL results on type-II SL's, $^{2,12}$  three  $X$  point phonons are expected to be involved in the  $\Gamma$ - $X$ transfer: AlAs transverse acoustic ( $\hbar\Omega_{TA}$ =12 meV), GaAs longitudinal optic ( $\hbar\Omega_{\text{LO,Ga}}$ =29 meV), and AlAs longitudinal optic ( $\hbar\Omega_{\text{LO},\text{Al}}$ =48 meV). Phonon assistance will lead to transfer without conservation of in-plane wave vector  $(k_{xy})$ as will elastic transfer to  $X_{xy}$ , since its conduction-band minimum is at large  $k_{xy}$ . For  $\Gamma$ - $X_z$  elastic transfer, the heterostructure potential only mixes states with the same  $k_{xy}$ . However, interface roughness is likely to induce mixing between states of different  $k_{xy}$  [Fig. 1(b)].

Therefore we assume that  $k_{xy}$  is not conserved in the transfer, and that  $\Gamma$ -*X* transfer is governed only by energy conservation [Fig. 1(b)]. The current flowing through level  $X_i$  is then given by

$$
J_i = eN_s \langle \Gamma | X_i \rangle |^2 P_0, \qquad (1)
$$

where  $N<sub>s</sub>$  is the sheet density of emitter electrons with energy greater than  $X_i$  and  $P_0$  is an intrinsic transfer rate which only depends on the nature of the transfer process. The squared overlap of the envelope functions accounts for the spatial separation of the  $\Gamma$  and  $\overline{X}$  2D states.<sup>4</sup> The total current is the sum of  $J_i$  for all  $X_i$  levels.<sup>15</sup> Since  $N_s$  varies linearly with bias and the overlap is slowly varying, we can approximate  $\sigma_i$  by  $\sigma_i \propto |\langle \Gamma | X_i \rangle|^2 P_0$ . When a new transfer channel is opened,  $\sigma$  is thus increased by an approximately constant value, proportional to the strength of the transfer process corrected for the overlap. In the following, the exact derivative of  $(1)$  was used. Phonon-assisted transitions are included simply by changing the threshold energy  $X_i$  to  $(X_i)$  $+\hbar\Omega_{\text{ph}}$ ). For each value of *V*,  $N_s$  and the overlaps were then calculated. The calculation of  $J_i$  and  $\sigma_i$  then depends only on the  $P_0$  parameters. There are two elastic transfer rates  $P_{0z}$  and  $P_{0xy}$  and three phonon-assisted rates to  $X_{xy}$ :  $P_{0xy}(TA)$ ,  $P_{0xy}(LO_{Ga})$ , and  $P_{0xy}(LO_{Al})$ . The  $P_{0z}$ (ph) rates are given by the  $X_{xy}$  values corrected for the different final density of states  $[P_{0z}(ph) \approx 0.25 P_{0xy}(ph)].$ 

We first present the results for  $\Gamma$ - $X$ <sub>z</sub> elastic transfer  $(dashed lines in Fig. 3)$ , assumed in much previous work to be dominant. $8-10$  It is seen that only some of the experimental features can be explained. However, when phonon-



FIG. 3. Experimental  $\sigma$ -*V* data (dots) for the three samples and the theoretical conductance (solid lines) from the model described in the text. Their comparison allows the assignment of each experimental feature to a given intervalley transfer process. The dashed lines show the theoretical conductance including only elastic transfer to the  $X<sub>z</sub>$  states.

assisted transfer is included the calculation describes all the  $\sigma$ -*V* features very well for all samples (the full lines on Fig. 3) and allows their clear identification.

To adjust the  $X<sub>z</sub>$  resonances to fit the data we varied the conduction-band discontinuity between the  $\Gamma$  band edge in GaAs and the *X* band edge in AlAs  $(\Delta_{\Gamma} \cdot X)$ , which determines the biases at which resonances occur, and  $P_{0z}$  which determines their amplitude. It is important to note that the relative amplitude of the transfers to successive  $X<sub>z</sub>$  levels is very well described by the model. This depends only on the overlaps and is not adjustable. The best fit is obtained for  $P_{0z} = 7 \pm 2$  ps<sup>-1</sup> for all samples. The  $\Delta_{\Gamma_z X}$  values employed are 114 meV for 60 Å, 126 meV for 80 Å, and 123 meV for the 100-Å sample. They are in good agreement with that suggested by Mendez *et al.*<sup>6</sup> and are much less than the commonly used value of  $\sim$ 180 meV.<sup>16</sup>

The fitting to the  $X_{xy}$  resonance was obtained using the same  $\Delta_{\Gamma}$  values. The best fits are obtained for  $P_{0xy}$ (TA)=0.5±0.2 ps<sup>-1</sup>,  $P_{0xy}$ (LO<sub>Ga</sub>)=0.9±0.3 ps<sup>-1</sup>, and  $P_{0xy}$ (LO<sub>Al</sub>)=3±1 ps<sup>-1</sup> for all samples. For the 80- and 100-Å samples where  $X_{xy}$  is lowest the amplitude of the first conductance threshold is underestimated using the above value of  $P_{0z}$ . If we include an elastic transfer rate



FIG. 4. EL spectra for the 60- and 80-Å samples showing the type-II luminescence and the typical signal from a highly monoenergetic hot electron population  $E<sub>b</sub>$ . Inset: Evolution of the type-II *X*-*H* transition energy with bias. The solid lines are the fits discussed in the text.

 $P_{0xy}$ =0.15±0.05 ps<sup>-1</sup> to  $X_{xy}$ 1 better agreement is obtained. For the 100-Å sample the same value of  $P_{0xy}$  accounts well for the feature at 1.80 V ascribed to elastic transitions to  $X_{xy}$ 2. The relative values of the  $P_{0xy}$  and  $P_{0xy}$ (ph) parameters are in good agreement with those observed in  $X_{xy}$ -like type-II luminescence, $2,12$  and with the relative intensities of the  $\gamma_i$  lines observed in EL, described below. The prominence of the  $X_{xy}$  resonances relative to  $X_z$  is notable. This is well described by the model and arises because of the much greater overlap of the  $X_{xy}$  states with the  $\Gamma$  emitter states in finite electric field. The strong contribution from  $X_{xy}$  was completely unsuspected in previous studies of transport through AlAs barriers, which were based primarily on comparison with type-II PL results at zero electric field.

The above analysis has permitted a precise determination of the transfer rates for elastic and phonon-assisted intervalley scattering. These are *intrinsic* values independent of the details of the structure. Using the relevant envelope function overlap, the intervalley transfer rates can then be calculated for other heterostructures. We applied these results to SL's of Ref. 5 and found very good agreement with the measured transfer times. For example, for sample 1 of Ref. 5, our calculation gives 0.17 ps compared to 0.14 ps measured, and for sample 8 of Ref. 5, 1.7 ps compared to 2.6 ps measured.

These  $\Gamma \rightarrow X$  intervalley transfer mechanisms, which depend strongly on the nature  $(X_z \text{ or } X_{xy})$  of the *X* state, are confirmed by the EL measurements, which probe the complementary  $X \rightarrow \Gamma$  transfer. Figure 4 presents typical EL spectra for the 60- and 80-Å samples. For the 60-Å sample  $X<sub>z</sub>1$  is the lowest state. We observe a strong no-phonon

type-II line  $X<sub>z</sub>1$ -*H*, arising from recombination of  $X<sub>z</sub>1$  electrons with HAL (see Fig. 1). In addition, a hot electron population which relaxes by emitting a cascade of zone-center GaAs LO phonons ( $\hbar\Omega_{\text{LO},\Gamma}$ =36 meV) is observed from its recombination with neutral acceptors (series of lines  $E_b$ -*n*LO,  $n=0, \ldots, 3$  in Fig. 4). As shown in Ref. 13 for  $\text{Al}_x\text{Ga}_{1-x}$ As barrier samples, the acceptor levels are shifted from the confined level in the HAL by about 13 meV. We observe the same shift between  $X<sub>z</sub>1$ -*H* and  $E<sub>b</sub>$ . This indicates that electrons are injected into the collector at the  $X_z$ 1 energy. Thus we conclude that  $X_z$  to  $\Gamma$  transfer out of the barrier is elastic, in agreement with the conclusion for the  $\Gamma$ -*X<sub>z</sub>* transfer.

For 80 Å (and in a similar way for 100 Å),  $X_{xy}$ 1 is the lowest state and the type-II EL exhibits a weak no-phonon line  $X_{xy}$ 1-*H* and three phonon replicas  $\gamma_1$ ,  $\gamma_2$ , and  $\gamma_3$  arising from pnonon-assisted recombination with the HAL, involving AlAs, TA, GaAs LO, and AlAs LO zone-boundary phonons, respectively. A zone-center GaAs LO-phonon cascade is again observed, the highest-energy line  $(E_b)$  being observed 13 meV below the  $\gamma_3$  replica. We conclude in this case that the hot electrons are injected at an energy  $\hbar\Omega_{\text{LO},\text{Al}}$  below  $X_{xy}$ 1, thus showing that the dominant  $X_{xy}$ - $\Gamma$  tunneling out mechanism is AlAs *X* point LO-phonon scattering, again in agreement with the conclusion for  $\Gamma$ - $X_{xy}$  transfer.

No EL is observed at higher energy than  $X<sub>z</sub>1$ -*H* or  $X_{xy}$ 1-*H*, even for biases corresponding to the higher resonances in the conductance. This proves that intravalley relaxation in the AlAs is much more efficient than  $X$  to  $\Gamma$  tunneling out, which only occurs for electrons which have relaxed to the lowest *X* state. Thus tunneling through such indirect gap barriers is strongly sequential. Coherent tunneling induced by  $X_z$ - $\Gamma$  mixing discussed in much previous work<sup>8–10</sup> appears to play very little role in the transport.

We also calculated the *X*-*H* transition energies using the self-consistent band potential. Good agreement with experiment is obtained using  $\Delta_{\Gamma}$ *x* of 116 meV for 60 Å, 115 meV for 80 Å, and 122 meV for 100 Å (inset to Fig. 4). This strongly confirms the value  $\Delta_{\Gamma_X}=120\pm 6$  meV. The type-II energy *X*-*H* is very sensitive to charge accumulation in the AlAs. The good agreement indicates that such charge buildup is small  $(<10^{11} \text{ cm}^{-2})$ , justifying the assumption made in the modeling of  $\sigma$ -*V*.

In conclusion, we have shown that  $\Gamma$ -*X* and *X*- $\Gamma$  intervalley transfer in AlAs/GaAs 2D systems depends strongly on the nature  $(X_z \text{ or } X_{xy})$  of the *X* states involved, but with comparable strength in the two cases. Transfer to or from  $X<sub>z</sub>$  is mainly elastic but not coherent, whereas transfer to or from  $X_{xy}$  is dominated by zone-boundary phonon scattering. In both cases  $\Gamma$ -*X*- $\Gamma$  transport through indirect-gap AlAs barriers is sequential.

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