

## Low-temperature structural behavior of $\text{Sr}_2\text{RuO}_4$

T. Vogt

*Department of Physics, Brookhaven National Laboratory, Upton, New York 11973*

D. J. Buttrey

*Department of Chemical Engineering, University of Delaware, Newark, Delaware 19716*

(Received 23 June 1995)

Using high-resolution neutron powder diffraction we have shown that in  $\text{Sr}_2\text{RuO}_4$  the onset of superconductivity is not accompanied by long-range structural distortions. This is in marked contrast to the isostructural copper-containing layered perovskites and points toward a more conventional superconducting mechanism. We have observed a structural response to the metal-to-insulator phase transition near 100 K.

The recent discovery of superconductivity in  $\text{Sr}_2\text{RuO}_4$  ( $T_c = 0.93$  K) (Ref. 1) provides clear evidence of superconducting behavior in a noncuprate oxide with the  $\text{K}_2\text{NiF}_4$  structure. Understanding the relationship between this ruthenate and the high- $T_c$  family of cuprates might point toward the peculiarities of the latter. It is of particular interest that in  $\text{Sr}_2\text{RuO}_4$  a second-row transition metal is in the  $B$  site and that the spin state is  $S = 1$  (low spin,  $4d^4$ ), as opposed to the  $S = 1/2$  ( $3d^9$ ) in the cuprates. Examining these differences in the electronic structure of isostructural compounds might address the question why copper is so special with regard to superconductivity.

Across the first-row transition metal series, initially the high-spin ions show a decrease in ionic radius for a given valence, which is due to changes in the crystal-field stabilization and the increased nuclear charge  $Z$ . As the high energy  $e_g$  orbitals are being populated, their antibonding character leads to a reversal of this trend and longer metal-to-oxygen distances are the result. The second- and third-row transition metals are generally low spin ions and no interruption of the ionic radius contraction is observed.

For further comparison, the first-row  $S = 1$   $\text{Ni}^{2+}$  (high spin,  $3d^8$ ) layered perovskites appear not to be superconducting, nor does the second-row  $S = 1/2$  rhodium analog (low-spin,  $4d^5$ )  $\text{Sr}_2\text{RhO}_4$ .<sup>1</sup> Another interesting difference between the first- and second-row transition metal containing compounds is the defect accommodation. Oxides of the type  $\text{Ln}_2\text{MO}_4$  with the  $\text{K}_2\text{NiF}_4$  structure containing first-row transition metals such as  $M = \text{Co}^{2+}$ ,  $\text{Ni}^{2+}$ ,  $\text{Cu}^{2+}$  are readily hole doped via intercalation of excess oxygen or by heterovalent substitution on the  $A$  site (e.g., by  $\text{Sr}^{2+}$  or  $\text{Ba}^{2+}$  for  $\text{Ln}^{3+}$ ). In the case of  $\text{Ru}^{4+}$  the electroneutrality of the  $\text{Sr}_2\text{O}_2$  rock-salt layer does not encourage the intercalation of excess oxygen. Thus the cuprates are insulating unless they are hole doped, whereas the ruthenate is intrinsically a poor metal or superconductor. In the cuprates, electronic states at the Fermi level are induced by hole doping; in the ruthenates, these states arise as a result of decreased electron correlations. Unfortunately, in the case of the ruthenate it seems that  $T_c$  is in the vicinity of 1 K, 20–50 times lower than the corresponding doped  $\text{K}_2\text{NiF}_4$ -type cuprates.

In the case of  $\text{La}_{2-x}\text{Ba}_x\text{CuO}_4$  the resistivity is strongly influenced by low-temperature structural phase transitions.

According to Maeno *et al.*,<sup>1</sup> x-ray investigations of the low-temperature structure of  $\text{Sr}_2\text{RuO}_4$  down to 5 K did not show any evidence of structural changes. Detailed structural data near  $T_c$  ( $\sim 1$  K) are therefore highly desirable to establish

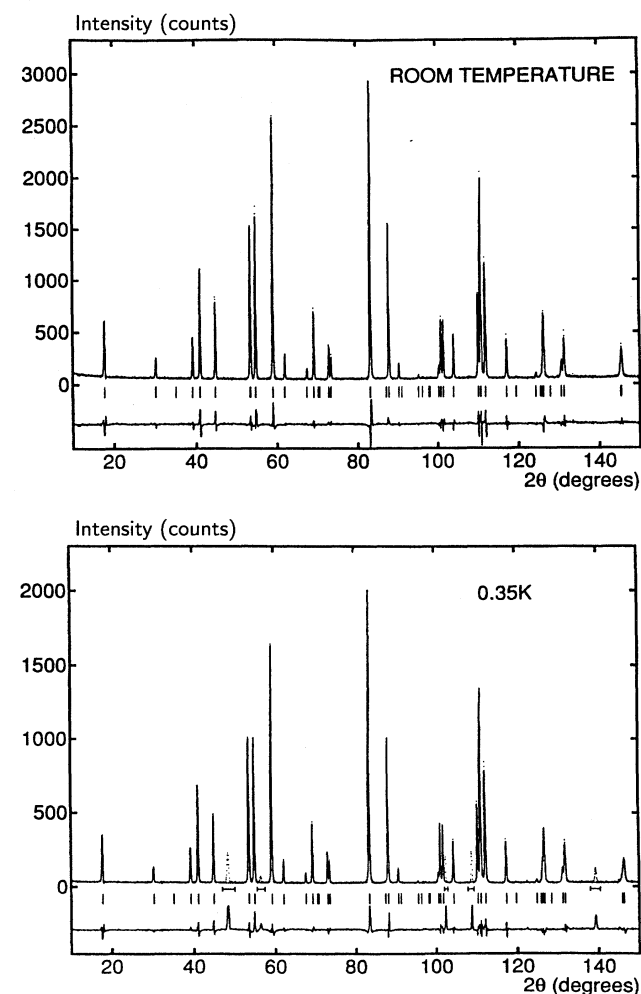


FIG. 1. Rietveld refinement plots of  $\text{Sr}_2\text{RuO}_4$  at room temperature and 0.35 K. For details see text.

TABLE I. Crystallographic parameters of  $\text{Sr}_2\text{RuO}_4$  at various temperatures refined in the space group  $I4/mmm$  (No. 139, International Tables for Crystallography).

Temperature	a			c			Temperature	a			c			
295 K	3.87100(9)			12.7397(4)			12 K	3.86122(9)			12.7214(4)			
	x	y	z	B(1,1)	B(2,2)	B(3,3)								
	Sr	0	0	0.3526(3)	0.6(1)	0.6(1)	0.5(2)	Sr	0	0	0.3526(4)	0.3(1)	0.3(1)	0.6(2)
	Ru	0	0	0	0.2(1)	0.2(1)	0.9(3)	Ru	0	0	0	0.0(1)	0.0(1)	0.6(3)
	O(1)	1/2	0	0	0.5(2)	0.4(2)	1.0(2)	O(1)	1/2	0	0	0.0(2)	0.5(2)	0.1(2)
	O(2)	0	0	0.1620(4)	0.7(1)	0.7(1)	0.4(3)	O(2)	0	0	0.1633(4)	0.4(1)	0.4(1)	0.1(3)
200 K	3.86656(8)			12.7323(3)			8 K	3.86132(8)			12.7218(4)			
	Sr	0	0	0.3525(3)	0.5(1)	0.5(1)	0.2(2)	Sr	0	0	0.3529(4)	0.3(1)	0.3(1)	0.6(2)
	Ru	0	0	0	0.1(1)	0.1(1)	0.3(2)	Ru	0	0	0	0.0(1)	0.0(1)	0.6(3)
	O(1)	1/2	0	0	0.1(2)	0.0(2)	0.8(2)	O(1)	1/2	0	0	0.0(2)	0.8(2)	0.0(2)
	O(2)	0	0	0.1615(4)	0.6(1)	0.6(1)	0.3(2)	O(2)	0	0	0.1634(4)	0.5(1)	0.5(1)	0.1(3)
150 K	3.8644(1)			12.7256(5)			6 K	3.86145(7)			12.7220(3)			
	Sr	0	0	0.3521(4)	0.4(1)	0.4(1)	0.4(3)	Sr	0	0	0.3526(3)	0.3(1)	0.3(1)	0.6(2)
	Ru	0	0	0	0.3(2)	0.3(2)	0.0(3)	Ru	0	0	0	0.0(1)	0.0(1)	0.5(3)
	O(1)	1/2	0	0	0.0(2)	0.7(2)	0.1(2)	O(1)	1/2	0	0	0.0(2)	0.7(2)	0.1(2)
	O(2)	0	0	0.1633(4)	0.4(1)	0.4(1)	0.1(3)	O(2)	0	0	0.1633(4)	0.4(1)	0.4(1)	0.1(3)
100 K	3.8630(1)			12.7240(4)			4.4 K	3.86135(7)			12.7216(3)			
	Sr	0	0	0.3524(3)	0.4(1)	0.4(1)	0.7(2)	Sr	0	0	0.3524(3)	0.4(1)	0.4(1)	0.7(2)
	Ru	0	0	0	0.1(1)	0.1(1)	0.7(2)	Ru	0	0	0	0.1(1)	0.1(1)	0.7(2)
	O(1)	1/2	0	0	0.0(2)	0.6(2)	0.1(2)	O(1)	1/2	0	0	0.0(2)	0.6(2)	0.1(2)
	O(2)	0	0	0.1635(3)	0.5(1)	0.5(1)	0.2(3)	O(2)	0	0	0.1635(3)	0.5(1)	0.5(1)	0.2(3)
50 K	3.86229(9)			12.7229(4)			0.35 K	3.86111(7)			12.7217(3)			
	Sr	0	0	0.3527(3)	0.3(1)	0.3(1)	0.4(2)	Sr	0	0	0.3533(3)	0.1(1)	0.1(1)	0.5(2)
	Ru	0	0	0	0.2(1)	0.2(1)	0.3(3)	Ru	0	0	0	0.0(1)	0.0(1)	0.4(2)
	O(1)	1/2	0	0	0.2(2)	0.2(2)	0.4(2)	O(1)	1/2	0	0	0.0(2)	0.7(2)	0.2(2)
	O(2)	0	0	0.1619(4)	0.4(1)	0.4(1)	0.3(3)	O(2)	0	0	0.1628(3)	0.4(1)	0.4(1)	0.0(2)

similarities or differences with copper-containing layered perovskites.  $\text{Sr}_2\text{RuO}_4$  shows a pronounced anisotropy of the normal-state resistivity: The resistivity in the  $\text{RuO}_2$  plane  $\rho_{ab}$  is  $\sim 10^{-4} \Omega \text{ cm}$  perpendicular to it  $\rho_c \sim 4 \times 10^{-2} \Omega \text{ cm}$ . The electronic anisotropy characterized by the ratio of  $\rho_c/\rho_{ab}$  is  $\sim 220$  at room temperature and increases to  $\sim 850$  at 2 K. Near  $T \sim 130 \text{ K}$   $\rho_c$  exhibits a crossover from nonmetallic to metallic behavior.<sup>1</sup> A careful search for a small structural variation associated with the metal-to-insulator transition would provide valuable insights into its origin which is thought to be purely electronic.

We performed low-temperature neutron powder diffraction measurements below and above  $T_c$  using the high-resolution neutron powder diffractometer at the High Flux Beam Reactor at Brookhaven National Laboratory. The instrument offers a resolution of  $\Delta d/d \sim 5 \times 10^{-4}$  at a wavelength of 1.8857 Å. The unique design of the vertically focusing monochromator is based on using Ge(115) wafer

stacks<sup>2,3</sup> where each wafer was deformed to obtain the desired anisotropic mosaic and then reassembled to 24 stacks. These were independently focused onto the sample position to provide a symmetrical peak profile at an optimized resolution and flux. The detector array is made out of 64  $^3\text{He}$  detectors separated by  $2.5^\circ$  in  $2\theta$ .

The data were refined using PROFIL, a Rietveld refinement program written by J. K. Cockcroft (University of Birbeck, U.K.). Refined crystallographic parameters are provided in Table I. The room temperature parameters agree well with those of a single crystal x-ray study published by Walz and Lichtenberg<sup>4</sup> as well as with the results of Rietveld refinements of neutron powder diffraction pattern by Huang *et al.*<sup>5</sup> and Neumeier *et al.*<sup>6</sup> The low-temperature structural data are also similar to the data at 13 and 10 K presented by Huang *et al.*<sup>5</sup> and Neumeier *et al.*<sup>6</sup> respectively.

On cooling no change in the symmetry of the unit cell could be detected. Two neutron powder diffraction patterns

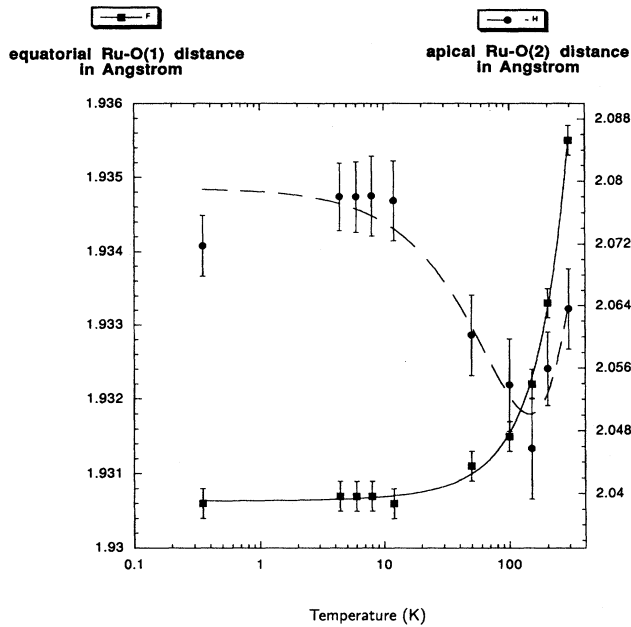


FIG. 2. The variation of the equatorial and axial Ru-O distances in the  $\text{RuO}_6$  octahedra of  $\text{Sr}_2\text{RuO}_4$  as a function of temperature. The ratio of these two distances is a measure of the  $D_{4h}$  distortion.

are presented in Fig. 1, one at room temperature and the second at  $T=0.35$  K. The extra peaks present in the 0.35 K are due to the aluminum heat shields in the  $^3\text{He}/^4\text{He}$  cryostat. The tick marks indicate the positions of the allowed Bragg reflections in space group  $I4/mmm$ ; the lower curve is the difference between observed and refined intensities.

It is apparent from the results presented in Table I that the  $a$  lattice constant contracts by roughly twice as much as  $c$  when cooling down to 0.35 K. This leads to an anisotropic squashing of the  $\text{RuO}_6$  octahedra: the four equatorial Ru-O(1) are shortened as the apical Ru-O(2) are elongated by the same amount as the shortest Sr-O(2) distance is reduced. The net result is a tighter octahedral packing and an increased tetragonal distortion of the  $\text{RuO}_6$  octahedra. This  $D_{4h}$  distortion of the octahedra can be accompanied by tilting at low temperatures, as is the case in other layered perovskites structures such as  $\text{La}_{2-x}\text{Sr}_x\text{CuO}_4$ . It has been shown that some phase transitions due to octahedral tilts are continuous so that the resulting low-temperature structure can be determined by irreducible representations of the high-temperature space group. The librational phonon modes involved transform accordingly and correspond in many cases to soft modes that condense at the phase transition. In the case of  $\text{Sr}_2\text{RuO}_4$ , the  $D_{4h}$  distortion is enhanced when cooling and no first-order structural phase transition is observed. There is a clear enhancement of the  $D_{4h}$  distortion around 100 K. Figure 2 shows that while the equatorial Ru-O(1) distance continues to decrease and then levels off near 10 K, the apical Ru-O(2) distance increases after the transition into the metallic state. In a simple angular overlap model the  $D_{4h}$  distortion of the  $\text{RuO}_6$  octahedra causes the low-lying

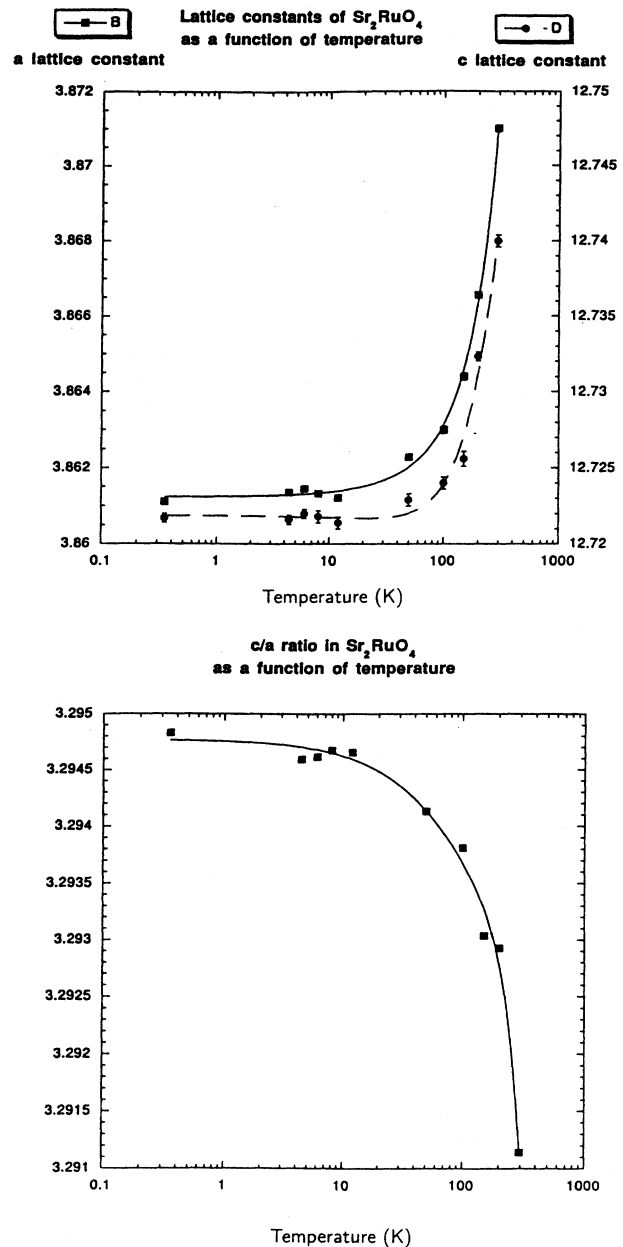


FIG. 3. The variation of the lattice constants and the  $c/a$  ratio of  $\text{Sr}_2\text{RuO}_4$  as a function of temperature.

$t_{2g}$  orbitals to be split into two groups. Increasing the tetragonal distortion when cooling by elongating the apical Ru-O(2) distance will perturb the  $d_{xz}$  and  $d_{yz}$  orbital more than the  $d_{xy}$  orbital. The ratio of the two ruthenium-to-oxygen distances in the  $\text{RuO}_6$  octahedra is a crude approximation of the energetic difference of the two  $t_{2g}$  levels created by the  $D_{4h}$  distortion. This can lead to an effective charge transfer from the  $d_{xz}$  and  $d_{yz}$  levels into the  $d_{xy}$  level. Cava *et al.*<sup>7</sup> have investigated the insulator-to-metal transition in  $\text{Sr}_2\text{Ir}_{1-x}\text{Ru}_x\text{O}_4$  solid solutions and found that the ratio of the lattice parameters  $c/a$  increases with increasing Ir concentra-

tion. At an Ir concentration of roughly 0.3 ( $x=0.7$ ) there is a noticeable change in slope. This critical concentration coincides with a maximum in the Curie moment per formula unit. Above 70% Ru concentration the local moment falls off dramatically and in pure  $\text{Sr}_2\text{RuO}_4$  virtually no localized moments are left. Thus the increase of the  $c/a$  ratio in

$\text{Sr}_2\text{RuO}_4$  (Fig. 3) on cooling appears to mimic this behavior and can be associated with a transition from a localized to itinerant electronic state near 100 K.

This research was supported by the Division of Materials Sciences, U.S. Department of Energy, under Contract No. DE-AC02-76CH00016.

---

<sup>1</sup>Y. Maeno, H. Hashimoto, K. Yoshida, S. Nishizaki, T. Fujita, J. G. Bednorz, and F. Lichtenberg, *Nature (London)* **372**, 532 (1994).

<sup>2</sup>T. Vogt, L. Passell, S. Cheung, and J. D. Axe, *Nucl. Instrum. Methods A* **338**, 71 (1994).

<sup>3</sup>J. D. Axe, S. Cheung, D. E. Cox, L. Passell, and T. Vogt, *Neutron News* **2** (3), 85 (1994).

<sup>4</sup>L. Walz and F. Lichtenberg, *Acta Crystallogr. Sect. C* **49**, 1268 (1993).

<sup>5</sup>Q. Huang, J. L. Soubeyroux, O. Chmaissem, I. Natali Sora, A.

Santoro, R. J. Cava, J. J. Krajewski, and W. F. Peck, Jr., *J. Solid State Chem.* **112**, 355 (1994).

<sup>6</sup>J. J. Neumeier, M. F. Hundley, M. G. Smith, J. D. Thompson, C. Allgeier, H. Xie, W. Yelon, and J. S. Kim, *Phys. Rev. B* **50**, 17 910 (1994).

<sup>7</sup>R. J. Cava, B. Batlogg, K. Kiyono, H. Takagi, J. J. Krajewski, W. F. Peck, Jr., L. W. Rupp, Jr., and C. H. Chen, *Phys. Rev. B* **49**, 11 890 (1994).