

Onset of superconductivity in the antiferromagnetically ordered state of single-crystal $\text{DyNi}_2\text{B}_2\text{C}$

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Temperature-dependent static magnetization, ac magnetic susceptibility, and electrical-resistivity measurements of single-crystal $\text{DyNi}_2\text{B}_2\text{C}$ reveal bulk superconductivity below $T_c = (6.2 \pm 0.1)$ K. This T_c is well below the Néel temperature $T_N = 10.3$ K. $\text{DyNi}_2\text{B}_2\text{C}$ is the first $R\text{Ni}_2\text{B}_2\text{C}$ compound with $T_c < T_N$. The upper critical magnetic field $H_{c2}(T)$ increases approximately linearly from zero at 6.2 K to ~ 5 kG at 2 K.

The interplay between superconductivity and local-moment magnetism has been vigorously studied since the late 1950s.¹ In the 1970s the discovery of two families of magnetic superconductors, $R\text{Mo}_6(\text{S},\text{Se})_8$ and $RRh_4\text{B}_4$ (R = rare earth), led to a detailed study of the interaction between the magnetic sublattice and the superconducting electrons.¹ Recently the $R\text{Ni}_2\text{B}_2\text{C}$ family of magnetic superconductors was discovered.² For $R = \text{Lu}, \text{Y}, \text{Tm}, \text{Er},$ and Ho the superconducting transition temperatures for single-crystal samples are $T_c = 16.0, 15.0, 10.8, 10.5,$ and 8.5 K, respectively.³⁻⁷ Superconductivity coexists with antiferromagnetic (AF) order for $R = \text{Tm}, \text{Er},$ and Ho for temperatures below the Néel temperatures $T_N = 1.5, 5.85,$ and 6.0 K, respectively.^{4-7,9,10} Similar values of T_c and T_N have been found for polycrystalline samples.^{2,8} In virtually all known magnetic superconductors,¹¹ when there is a coexistence of AF ordering and superconductivity, $T_c > T_N$. The only local-moment systems that have $T_c < T_N$ are solid solutions such as the $R(\text{Ir}_x\text{Rh}_{1-x})_4\text{B}_4$ system where $T_N = 2.7$ K and $T_c = 1.4$ K for $R = \text{Ho}$ and $x = 0.7$.^{1,12} This reversal of T_c and T_N only occurs for $0.6 < x < 0.8$ in this alloy. Compounds exhibiting $T_c < T_N$ are expected to be rare since the stronger the conduction electron-local moment coupling (such as would be required for a higher T_N), the greater the anticipated suppression of T_c through magnetic pair breaking.¹ In this paper we report the discovery of superconductivity in single-crystal $\text{DyNi}_2\text{B}_2\text{C}$ below a $T_c = (6.2 \pm 0.1)$ K that is well below the $T_N = 10.3$ K. This is the first such member of the $R\text{Ni}_2\text{B}_2\text{C}$ family with $T_c < T_N$.

Single crystals of $\text{DyNi}_2\text{B}_2\text{C}$ were grown from Ni_2B flux³ using high-purity elements: B (99.9%), Ni (99.99%), C (99.99%), and Ames Lab Dy (99.99%). The crystals grow in the form of plates with the crystallographic c axis perpendicular to the largest plate surface. The static magnetization M was measured using a Quantum Design (SQUID) magnetometer and the ac susceptibility χ_{ac} was measured using a Lakeshore ac magnetometer. The single crystal used for the M and χ_{ac} measurements had approximate dimensions of $2.5 \times 2.5 \times 0.7$ mm³. The four-lead electrical resistivity ρ_{ab} was measured on a platelike crystal with the current flowing in the basal ab plane using a Linear Research, LR 400, ac resistance bridge operating at 15.9 Hz.

Figure 1 shows the powder x-ray-diffraction pattern of a crushed single crystal of $\text{DyNi}_2\text{B}_2\text{C}$. The diffraction

peaks index well to the tetragonal unit cell reported² for $\text{DyNi}_2\text{B}_2\text{C}$ with lattice parameters $a = 3.534$ Å and $c = 10.484$ Å. The only peak that is not indexed to $\text{DyNi}_2\text{B}_2\text{C}$ is the weak peak at $2\theta = 45.85^\circ$, attributed to the [211] peak of the Ni_2B flux.¹³

Figure 2 displays the temperature-dependent magnetic susceptibility $\chi(T)$ of $\text{DyNi}_2\text{B}_2\text{C}$ with a magnetic field $H = 1$ kG applied parallel ($\mathbf{H} \parallel c$) and perpendicular ($\mathbf{H} \perp c$) to the c axis. Magnetic neutron diffraction measurements¹⁴ indicate that the sharp feature at $T_N = 10.3$ K in Fig. 2 should be attributed to the onset of AF order. The large anisotropy seen between $\mathbf{H} \parallel c$ and $\mathbf{H} \perp c$ at low T is a feature common to the $R\text{Ni}_2\text{B}_2\text{C}$ materials for $R = \text{Er}, \text{Ho},$ and Tb .^{6,7,15} The inset to Fig. 2 shows $\chi^{-1}(T)$ for both directions of applied field as well as for the polycrystalline average of the $\chi_{\parallel c}$ and $\chi_{\perp c}$ data: $\chi_{\text{poly}} = \chi_{\parallel c}/3 + 2\chi_{\perp c}/3$. The $\chi_{\text{poly}}(T)$ data can be fitted by a Curie-Weiss law $\chi = C/(T - \theta)$ for $20 < T < 350$ K, giving an effective moment $\mu_{\text{eff}} = 9.85\mu_B$ and $\theta = (1 \pm 2)$ K. This value of μ_{eff} is slightly lower than the theoretical value of $10.63\mu_B$ for the $J = 15/2$ Hund's rule ground state of Dy^{3+} . The anisotropic $\chi(T)$ data can also be fitted by a Curie-Weiss form for $\mathbf{H} \parallel c$ for $230 \text{ K} < T < 375 \text{ K}$ giving $\mu_{\text{eff}} = 10.4\mu_B$ and $\theta = -82$ K and for $\mathbf{H} \perp c$ for $200 \text{ K} < T < 325 \text{ K}$ giving $\mu_{\text{eff}} = 9.8\mu_B$ and $\theta = 25$ K.

Figure 3(a) shows $\rho_{ab}(T)$ of a $\text{DyNi}_2\text{B}_2\text{C}$ crystal. There is a sharp loss of scattering associated with the AF transition at

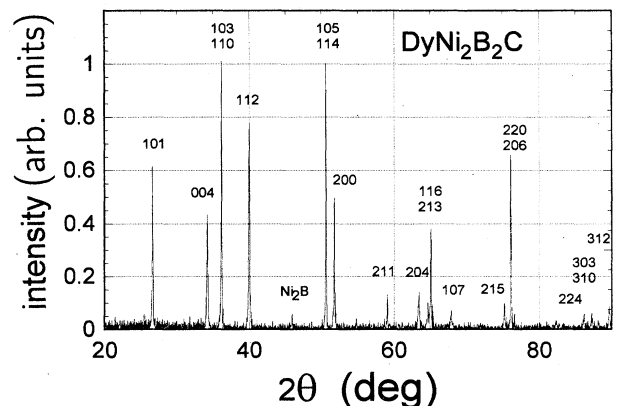


FIG. 1. Powder x-ray-diffraction pattern of a crushed $\text{DyNi}_2\text{B}_2\text{C}$ single crystal.

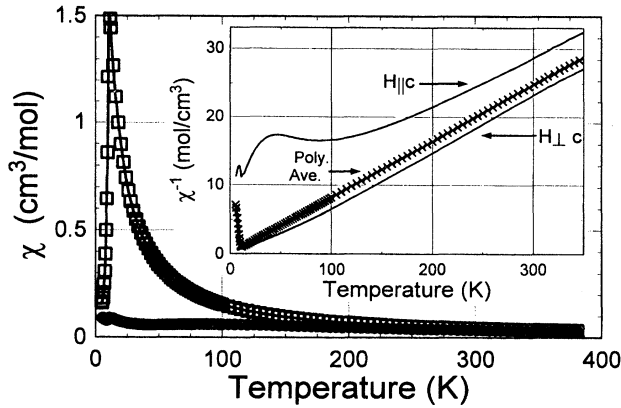


FIG. 2. Anisotropic magnetic susceptibility χ vs temperature T for $\text{DyNi}_2\text{B}_2\text{C}$, with $\mathbf{H}\perp c$ (squares) and $\mathbf{H}\parallel c$ (circles) and $H=1$ kG. Inset: Anisotropic (and polycrystalline average) χ^{-1} vs T .

$T_N=10.3$ K, followed by a superconducting transition with an onset at 6.4 K and zero resistivity at 6.0 K. We find $\rho_{ab}(300\text{ K}) = 55\ \mu\Omega\text{ cm}$. The residual resistivity ratio is $\rho_{ab}(300\text{ K})/\rho_{ab}(7\text{ K}) = 27$ indicating that the crystal has a high degree of perfection. Figure 3(b) shows $\rho_{ab}(H, T)$ for $2\text{ K} < T < 7\text{ K}$ and $\mathbf{H}\perp c$. For these applied fields ($H \leq 5$ kG), T_N is only weakly field dependent, decreasing to $T_N=10.0$ K for $H=5$ kG (not shown). As can be seen from Fig. 3(b), T_c is suppressed and the width of the superconducting transition is increased with increasing H .

Figure 4 shows the temperature-dependent upper critical magnetic field $H_{c2}(T)$ derived from the $\rho_{ab}(H, T)$ data in

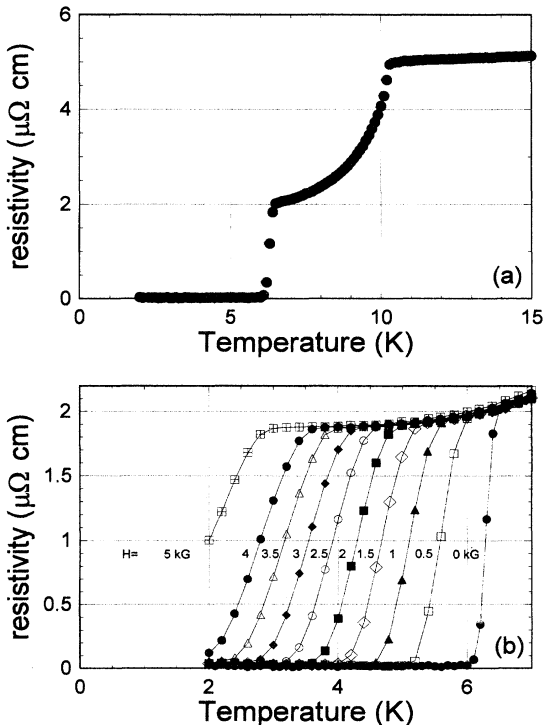


FIG. 3. Electrical resistivity of $\text{DyNi}_2\text{B}_2\text{C}$ in the ab plane vs temperature T : (a) $H=0$ and $T < 15$ K and (b) $\mathbf{H}\perp c$ and $T < 7$ K.

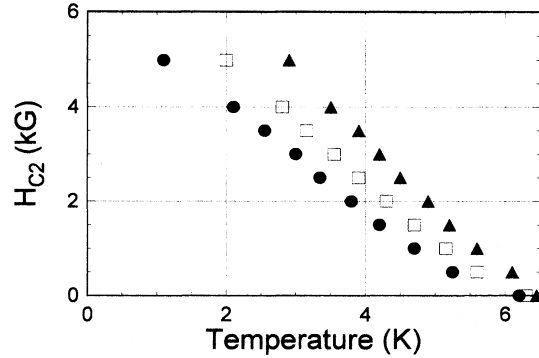


FIG. 4. Upper critical magnetic field H_{c2} vs temperature T for $\text{DyNi}_2\text{B}_2\text{C}$ with $\mathbf{H}\parallel c$. The circles, squares, and triangles show $H_{c2}(T)$ data determined from Fig. 3(b) by using the zero-resistivity, midpoint, and resistive onset, respectively, as the criterion for determining $T_c(H)$.

Fig. 3(b). $H_{c2}(T)$ increases nearly linearly with decreasing T from 10 G [just below $T_c(H=0)$] to ~ 5 kG near 2 K. No local minimum or other structure is seen in $H_{c2}(T)$, which is consistent with $T_c < T_N$. This is in contrast to the local extrema in $H_{c2}(T)$ seen for $\text{TmNi}_2\text{B}_2\text{C}$,^{5,8} $\text{ErNi}_2\text{B}_2\text{C}$,^{6,8} and $\text{HoNi}_2\text{B}_2\text{C}$.^{7,8} The value of $\|dH_{c2}/dT\|_{T_c} = (1.2 \pm 0.2)$ kG/K is less than the values of (2.8 ± 0.2) and (2.6 ± 0.2) kG/K for $\text{TmNi}_2\text{B}_2\text{C}$ (Ref. 5) and $\text{ErNi}_2\text{B}_2\text{C}$ (Ref. 6), respectively.

Figure 5(a) shows the low-temperature static volume magnetization M of $\text{DyNi}_2\text{B}_2\text{C}$ for $\mathbf{H}\parallel c$. From Fig. 2, for this field direction there is only a weak paramagnetic contribution from the Dy sublattice. In Fig. 5(a), the onset of a superconducting magnetization is seen at 6.1 K which becomes nearly independent of T below 4 K. At 2 K the flux expulsion magnetization is 10% of the ideal value of $H/4\pi$ and the shielding fraction is almost 300%. If the crystal is fully superconducting, the latter value indicates a demagnetization factor of 0.66, consistent with that calculated (0.68) for an ellipsoid of revolution with the sample dimensions.

To further confirm that the superconductivity in $\text{DyNi}_2\text{B}_2\text{C}$ is a bulk rather than a surface effect, the $M(T)$ of a powder sample made by crushing a single crystal of $\text{DyNi}_2\text{B}_2\text{C}$ was measured [Fig. 5(b)]. T_N is now seen at ≈ 10.2 K. In addition, a clear onset of a superconducting transition is seen at 5.9 K (see inset). Due to the contribution from the paramagnetic Dy sublattice, the measured magnetization does not become diamagnetic until somewhat lower temperatures. At 2 K the diamagnetic M/H for the field-cooled measurement is 60% of $1/4\pi$ and the zero-field-cooled shielding fraction is 140% which is close to the value (150%) anticipated from the powder average demagnetization factor. The polycrystalline data in Fig. 5(b) and the single-crystal data in Fig. 5(a) show a markedly different temperature dependence of the diamagnetism. For the polycrystalline sample there is an onset of superconductivity at 5.9 K followed by a shallow increase of the diamagnetic magnetization on cooling to 4 K, below which there is a rapid increase of diamagnetism. For the single-crystal sample there is a much more uniform and rapid increase of diamagnetism on cooling below 6.1 K which is nearly complete by 4 K.

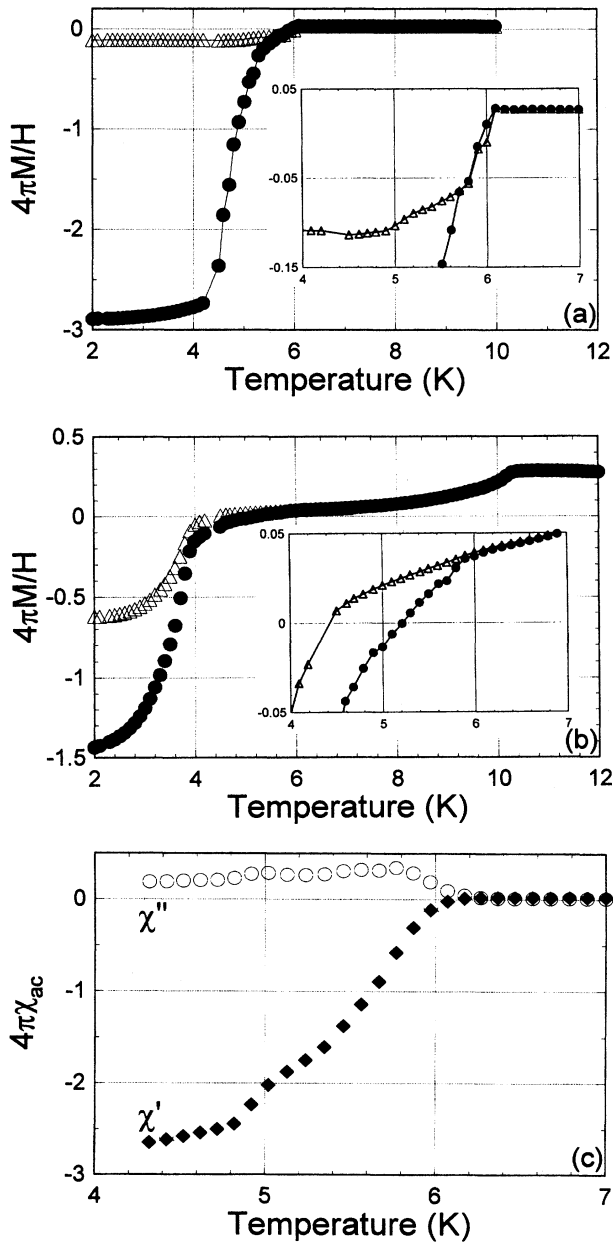


FIG. 5. Temperature-dependent magnetization M of $\text{DyNi}_2\text{B}_2\text{C}$: (a) static M with $\mathbf{H}\parallel c = 10$ G: zero-field-cooled (ZFC) data (circles) and field-cooled/warming (FCW) data (triangles); (b) static M of a powdered single crystal with $H = 10$ G: ZFC data (circles) and FCW data (triangles); and (c) real (χ' , diamonds) and imaginary (χ'' , circles) parts of the ac susceptibility χ_{ac} of the same single crystal used in (a) with $H_{ac} = 0.125$ G at a frequency of 125 Hz.

Figure 5(c) shows the ac susceptibility χ_{ac} for $\mathbf{H}_{ac}\parallel c$ taken on the same single-crystal sample that was used to provide the data shown in Fig. 5(a). The real part of χ_{ac} shows a clear onset of diamagnetism below 6.3 K. In addition, below 6.3 K there is an increase in the imaginary part. Both of these features are consistent with a bulk T_c of 6.3 K.

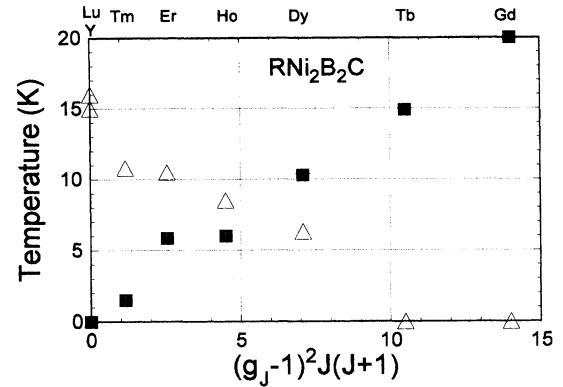


FIG. 6. Néel temperature T_N (squares) and superconducting transition temperature T_c (triangles) vs de Gennes factor $(g_J - 1)^2 J(J + 1)$ for $\text{DyNi}_2\text{B}_2\text{C}$ and other $\text{RNi}_2\text{B}_2\text{C}$ ($R = \text{Gd, Tb, Ho, Er, Tm, Lu, and Y}$) (Refs. 3–7,9,10,14,15,17) single crystals.

Figures 2–5 clearly indicate the existence of a bulk anti-ferromagnetic transition at $T_N = 10.3$ K and a bulk superconducting transition at $T_c = (6.2 \pm 0.1)$ K in single-crystal $\text{DyNi}_2\text{B}_2\text{C}$. The latter result is in conflict with an earlier report⁸ on a polycrystalline sample of $\text{DyNi}_2\text{B}_2\text{C}$ that showed no superconductivity above 2 K. One possible explanation for this difference is that there may be some residual strain in polycrystalline samples leading to an extrinsic suppression of T_c . As shown in Figs. 5(a) and 5(b), a broadening and suppression of the majority of the superconducting transition occurred in our powdered single-crystal sample. Another conspicuous difference between the polycrystalline and single-crystal samples is the value of the residual resistivity ρ_0 at $T > T_c$: for our single-crystal sample $\rho_0(7 \text{ K}) = 2.2 \mu\Omega \text{ cm}$ (Fig. 3), while for the polycrystalline sample⁸ $\rho_0(7 \text{ K}) > 20 \mu\Omega \text{ cm}$. This difference in ρ_0 may indicate that $\text{DyNi}_2\text{B}_2\text{C}$ shows a variability in composition and/or in the degree of crystallographic ordering; such variabilities could strongly affect T_c as in $A-15$ compounds such as Nb_3Ge .¹⁶

With $T_c < T_N$, the question of whether the T_c of $\text{DyNi}_2\text{B}_2\text{C}$ follows de Gennes scaling is a salient one. Figure 6 shows the T_N and T_c values for single crystals of $\text{RNi}_2\text{B}_2\text{C}$ ($R = \text{Gd-Tb, Lu, and Y}$)^{3–7,10,11,14,15,17} vs the de Gennes factor $(g_J - 1)^2 J(J + 1)$, where g_J is the Landé factor and J is the total angular momentum of the R^{3+} Hund's rule ground state. Good overall de Gennes scaling is seen for the whole heavy rare-earth series for both T_c and T_N . This indicates that both T_N and the suppression of T_c originate from the same conduction electron-local moment exchange interaction. In particular, our T_c value for single-crystal $\text{DyNi}_2\text{B}_2\text{C}$ is on the order of that expected from the variation of T_c vs de Gennes factor for the other superconducting members.

In summary, temperature-dependent electrical resistivity, static magnetization, and ac susceptibility measurements have revealed the onset of bulk superconductivity in single-crystal $\text{DyNi}_2\text{B}_2\text{C}$ at $T_c = (6.2 \pm 0.1)$ K, which is significantly lower than the antiferromagnetic ordering (Néel) temperature at $T_N = 10.3$ K. $\text{DyNi}_2\text{B}_2\text{C}$ is the first member of the $\text{RNi}_2\text{B}_2\text{C}$ series to exhibit $T_c < T_N$ and also appears to be a

crystallographically ordered compound outside of the heavy-fermion family to show this order of transition temperatures. While there is good overall de Gennes scaling of T_c and T_N across the $R\text{Ni}_2\text{B}_2\text{C}$ series, it is still an open question as to how well de Gennes scaling will work for a series of materials where T_c is lowered through T_N in a more continuous manner. Since $T_c < T_N$ for $\text{DyNi}_2\text{B}_2\text{C}$ and $T_c > T_N$ for $(\text{Ho,Er,Tm})\text{Ni}_2\text{B}_2\text{C}$, a study of the crossover of T_c and T_N in,

e.g., $(\text{Ho}_{1-x}\text{Dy}_x)\text{Ni}_2\text{B}_2\text{C}$ solid solutions should be very interesting.

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