## Large finite-size effect of giant magnetoresistance in magnetic granular thin films

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We have observed a finite-size effect of giant magnetoresistance (GMR) and extraordinary Hall effect (EHE) in  $Co_{20}Ag_{80}$  and  $Fe_{20}Ag_{80}$  granular thin films of varying thickness. The GMR approaches zero as the thickness approaches zero. The thickness dependence of GMR can be explained by a dimensional-confinement effect coupled with a large surface-induced spin-Gipping cross section. The EHE exhibits a strong correlation with GMR, which is likely due to the surface-induced spin-orbit interaction.

Since the discovery of giant magnetoresistance (GMR) in layered structures, we have witnessed a series of developments in the understanding of the magnetotransport in heterogeneous systems.<sup> $1-5$ </sup> The recent finding that GMR also exists in granular systems<sup>5</sup> has broadened the scope of this research area. A prevailing physical picture of GMR is that there exist two independent currents made of spin-up  $(+)$  and spin-down  $(-)$  electrons. They are scattered by the magnetic components with different scattering rates  $1/\tau_+$ ,  $1/\tau_-$  dependent on the relative orientation between the electron spin and the magnetic moment vector of a scatterer. As one increases the magnetic field  $(H)$ , the magnetic moments in different layers or granules are gradually aligned. This subdues the scattering suffered by one of the spin currents, which in turn reduces the resistance of a sample due to a short-circuit-like effect. Despite the formidable structural complexities, the "two-current" model in its numerous variations<sup>3,4</sup> seems to account for GMR rather well in multilayers. However, there are important issues yet to be addressed in the GMR of granular systems. We do not understand what role the spin-Gipping process plays in granular systems where spin-orbit interaction may be strong. The spin-Gipping effect causes mixing of the two currents and is destructive to GMR. We also lack the knowledge of dimensional confinement in magnetotransport. It is believed that the essence of GMR resides in the asymmetry of two electron mean free paths (MFP),  $\lambda_+$  and  $\lambda_-$ . Dimensional confinement allows one to probe these two important length scales and to examine their effect on GMR. Currently, there are no viable models addressing these issues in granular systems, and the models developed for multilayers are of limited utility due to the particularity of granular microstructures.

In this paper, we present measurement of thickness  $(t)$ dependence of GMR, extraordinary Hall effect (EHE), and. magnetic properties for Co-Ag and Fe-Ag granular films. We have taken advantage of the fact that strong spin-orbit interaction is often dominant near the surfaces of a thin film.<sup>6,7</sup> By varying t we have observed a large finite-size effect of both GMR and EHE, while keeping the structure and the magnetic properties unchanged. The GMR results have been explained by a model that includes both the efFect of spin-Gipping and dimensional confinement. The EHE, an inseparable component of the magnetotransport, exhibits an intimate correlation with GMR, but defies conventional explanation commonly used for homogeneous alloys.

We have chosen  $Co<sub>20</sub>Ag<sub>80</sub>$  and  $Fe<sub>20</sub>Ag<sub>80</sub>$  (in volume fraction) granular systems for this study. They are immiscible alloys and exhibit GMR in bulk forms.  $\stackrel{\text{8}-10}\{2}$  Both samples attain a maximum GMR in their alloy series. Their bulk structural and magnetic properties have been investigated previously.<sup>9,10</sup> Our samples were fabricated by using a high vacuum magnetron-sputtering system. To achieve a high degree of consistency, films with variable t from 40 to 3200 Å were deposited during a single run on a  $2 \times 5$  cm<sup>2</sup> SiO<sub>2</sub> substrate. A computerized stepping shutter swept across the substrate in ten steps according designed time intervals for each step. The total exposure time for each step determined the t of that step. All the samples used in this study were protected by a layer of photoresist immediately after deposition to avoid oxidation. The samples were not thermally annealed. The step-wedged samples were subsequently patterned using photolithography (see the inset in Fig. 1). Samples at each t consist of Hall bars for magnetoresistance (MR) and Hall effect measurement and rectangles for magnetic measurement on a SQUID magnetometer. <sup>A</sup> representative MR curve is shown in the inset in Fig. l.

Figure 1 shows the t dependence of  $\Delta \rho = \rho(0) - \rho(8)$ T), the saturated  $H$ -induced change in resistivity. In all samples,  $\rho(H)$  is nearly saturated at  $H = 8$  T and  $T = 4.2$  K. For both series, two sets of samples were made and measured, covering approximately the same  $t$  range. The data for each set, very consistent among each other, are represented by different symbols. From Fig. 1, we observe two interesting common features shared by both series: First, GMR decreases with the reducing  $t$  starting at a large  $t$  scale. Even at  $t$  as large as 1000 Å, GMR is still not fully saturated at the bulk values. Second, GMR vanishes as t approaches zero. Hence GMR cannot be sustained in the two-dimensional limit.

Before addressing the t dependence, we need to exam-



FIG. 1. Thickness dependence of the magnetically induced resistivity  $\Delta \rho = \rho(0) - \rho(8 \text{ T})$  at  $T = 4.2 \text{ K}$  for the  $Co<sub>20</sub>Ag<sub>80</sub>$  (upper panel) and  $Fe<sub>20</sub>Ag<sub>80</sub>$  (lower panel) series. The two symbols represent data from two sets of samples for each series. The solid lines are from the model fitting (see text). The upper inset is the magnetoresistance of a  $Co<sub>20</sub>Ag<sub>80</sub>$ sample with  $t = 2030 \text{ Å}$ . The lower inset shows the geometry of the step-wedged samples with varying  $t$ .

ine some relevant transport and magnetic parameters of our samples. The resistivities at  $T = 4.2$  K are 27 and 31  $\mu\Omega$  cm for bulk Co<sub>20</sub>Ag<sub>80</sub> and Fe<sub>20</sub>Ag<sub>80</sub>, respectively. The resistivity ratio  $\rho$  (300 K)/ $\rho$  (4.2 K) is about 1.1. From the measured resistivities and Hall coefficients, we calculated the effective MFP  $\lambda_{\text{eff}}$  for the unmagnetized samples (at  $H = 0$ ), shown in Fig. 2 as a function of t. It is noted that, at zero field when samples are unmagnetized,  $\lambda_{\text{eff}}$  for spin-up and -down electrons is the same because of the random distribution of magnetization vectors. The value of  $\lambda_{\text{eff}}$  remains at approximately a constant 35 Å for  $t > 200$  Å. As t decreases toward zero,  $\lambda_{\text{eff}}$  decreases slightly because of increasing surface scatterings. The short  $\lambda_{\text{eff}}$  is mainly caused by the strong scatterings at the particle boundaries. As we will show later, the magnetic particle size is about 30  $\AA$  and so is the average interparticle distance at 20% volume fraction. For a continuous film with t larger than  $\lambda_{\text{eff}}$ , its resistivity consists of two contributions,  $^{11}$  one is the bulk resistivity,  $\rho_B$ , and the other from the surface,  $\frac{3}{8}\rho_B \lambda_{\text{eff}}^B/t$ ,



FIG. 2. Thickness dependence of the effective electron mean free path  $\lambda_{\text{eff}}$ , and the resistivity-thickness product  $\rho \cdot t$ for the Co20Agso and Fe20Agso series.

$$
\rho = \rho_B + \frac{3}{8} \rho_B \lambda_{\text{eff}}^B / t = \frac{m^*}{2ne^2} \left( \frac{1}{\tau_B} + \frac{3}{8} \frac{v_F}{t} \right). \tag{1}
$$

This expression assumes diffusive scatterings at the surfaces (the total carrier density is  $2n$ ). In Fig. 2, we plot  $\rho t$  vs  $t$  for both series. As expected from relation (1) a straight line fits the data very well. The intercepts along the vertical axis provides  $\frac{3}{8}\rho_B\lambda_{\text{eff}}^B$ . Since  $\lambda_{\text{eff}}^B$  is known, we can estimate  $\rho_B$  and the results are consistent with the bulk values. The confirmation of relation (1) reveals that our thin films are continuous in the thinnest limit and  $t$  is rather uniform. It is noted that a completely different t dependence of  $\rho$  is expected for discontinuous or uneven thin films.

The magnetic properties of a granular film are of paramount importance to GMR.<sup>10</sup> One naturally raises the question whether a possible variation in the underlying magnetic state could lead to the observed  $t$  dependence of GMR. Our magnetic measurement on different samples rules out this scenario. Figure  $3(a)$  shows the t dependence of the spontaneous magnetization  $M_S$  at  $T =$ 5 K and 300 K for the  $Co<sub>20</sub>Ag<sub>80</sub>$  series. At low T, the samples are ferromagnetic.  $M_S$  was obtained directly from the measured  $M(H)$  curves  $(0 \le H \le 5.5 \text{ T})$ . At from the measured  $M(H)$  curves  $(0 \le H \le 5.5 \text{ T})$ . At  $T = 300 \text{ K}$ , the samples become superparamagnetic.  $M_S$ was obtained by fitting  $M(H)$  to a Langevin equation.<sup>10</sup> Simultaneously, the particle size was also obtained from such a fitting; the data are shown in Fig. 3(b). Clearly, both  $M<sub>S</sub>$  and the particle size remain approximately constant over the whole  $t$  range. Therefore as  $t$  varies, the magnetic state and the microstructure are not changed appreciably and they should have little, if any, bearing on the t dependence of GMR.

To understand the t dependence of GMR, we need to focus on two major features of a thin film: the existence of surfaces and the geometrical confinement to transport. The surfaces are important spin-orbit (SO)



FIG. 3. Thickness dependence of (a) the spontaneous magnetization,  $M_s$ , and (b) the magnetic particle size for the  $Co<sub>20</sub>Ag<sub>80</sub>$  series.

scattering centers, which not only affect electron mobility, but also randomize electron spin orientation. Often a phenomenological parameter  $\epsilon$  is used to describe the  $\text{probability of spin flipping in elastic SO scatterings}, ^{7}1/\epsilon$ being the average number of scattering events needed to cause spin randomization. For a thin film the spinflipping rate,  $1/\tau_{\uparrow\downarrow}$ , can be separated into a bulk and a surface contribution:<sup>7</sup>

$$
\frac{1}{\tau_{\uparrow\downarrow}} = \frac{\epsilon_B}{\tau_B} + \frac{\epsilon_S}{\tau_S} = \frac{\epsilon_B}{\tau_B} + \frac{3v_F \epsilon_S}{8t}.
$$
 (2)

In writing relation (2), we have used  $1/\tau_s = 3v_F/8t$ from relation (1). Relation (2) has been confirmed in weak-localization experiments,  $6,7$  an efficient technique to measure  $1/\tau_{\uparrow\downarrow}$ . Lindelof and Wang<sup>7</sup> have found that on the surfaces of a thin film,  $\epsilon_S$  is enhanced over  $\epsilon_B$  by 1—<sup>2</sup> orders of magnitude. In other words, spin Bipping is much stronger at the surfaces than in the bulk.

Another important effect of a thin film is its geometrical confinement, which sets a physical limit to the eleccal confinement, which sets a physical limit to the electron MFP. The asymmetry in MFP  $(\lambda_+ > \lambda_-)$  is the origin of GMR. Because  $\lambda_+$  is longer than  $\lambda_-$ , the geometrical confinement affects  $\lambda_+$  preferentially. Therefore, it tends to reduce the asymmetry between  $\lambda_+$  and  $\lambda_-$ . creating a negative impact on GMR. It should be pointed out that  $\lambda_+$  and  $\lambda_-$  are related to magnetic scatterings only. Ordinary surfaces have little effect on them. However, for surfaces with large SO interactions, the strong spin-Bipping effect does set a physical limit on MFP.

The MR of a magnetic system with spin-mixing effect can be generally described by a "two-current" model developed by Fert.<sup>12</sup> This model has explained the magnetotransport of some dilute and concentrated magnetic alloys.<sup>12-14</sup> According to this model, the *H*-induced resistivity is given by

$$
\Delta \rho(H) = \rho_0 - \frac{\rho_{\uparrow} \rho_{\downarrow} + \rho_{\uparrow \downarrow} (\rho_{\uparrow} + \rho_{\downarrow})}{\rho_{\uparrow} + \rho_{\downarrow} + 4\rho_{\uparrow \downarrow}}, \tag{3}
$$

where  $\rho_0 = m^*/2ne^2\tau_0$  is the zero-H resistivity,  $\rho_{\uparrow} =$  $m^*/ne^2\tau_+$  are  $\rho_{\downarrow}=m^*/ne^2\tau_-$  are the resistivities of the two spin channels, and  $\rho_{\uparrow\downarrow} = m^*/ne^2\tau_{\uparrow\downarrow}$  is due to the spin-mixing effect. We will extend this phenomenological model to account for the finite-size effect of GMR by including both the spin-mixing effect and the geometrical confinement. To do so we need to find three scattering rates,  $1/\tau_{\pm}$  and  $1/\tau_{\uparrow\downarrow}$ . The t dependence of  $1/\tau_{\uparrow\downarrow}$  has already been given by relation (2). As for  $1/\tau_{\pm}$ , their t dependence should include the geometrical confinement effect. In bulk samples,  $1/\tau_{+}^{B} = (1/\tau_{0})(1 \mp \Gamma)$ , where  $\Gamma$ is a parameter characterizing the asymmetry of the two spin channels. The ratio of MFP's for bulk samples is  $\lambda_+/\lambda_- = (1+\Gamma)/(1-\Gamma)$ . For a thin film,  $1/\tau_{\pm}$  also contains a contribution due to the geometrical confinement, i.e.,

$$
\frac{1}{\tau_{\pm}} = \frac{1}{\tau_{\pm}B} + \frac{1}{\tau_{\pm}S} = \frac{1}{\tau_{\pm}B} + \frac{3v_F \epsilon_S}{8t}.
$$
 (4)

The factor  $\epsilon_S$  ensures that only the spin-flipping events on the surfaces can set a physical limit to  $\lambda_+$  and  $\lambda_-$ .

According to relation (3), the t dependence of  $\Delta \rho$  is caused by the finite-size effect on  $1/\tau_{\pm}$  and  $1/\tau_{\uparrow\downarrow}$ . There are only two unknown parameters,  $\rho_0$  and  $\epsilon_S$ . All other parameters,  $n, v_F$ ,  $\Gamma$ , and  $t$ , can be obtained from independent experiments. We have calculated the Fermi velocity,  $v_F \approx 1.3 \times 10^8$  cm/s for  $\rm Co_{20}Ag_{80}$  and  $1.2 \times 10^8$ cm/s for  $Fe_{20}Ag_{80}$ , from the carrier density n obtained from the Hall effect measurement. The parameter  $\Gamma$ was obtained by fitting MR data in bulk samples using a model developed for granular solids:<sup>10</sup>  $\Gamma = 0.60$ for  $Co_{20}Ag_{80}$  and 0.47 for  $Fe_{20}Ag_{80}$ . Based on  $\Gamma$  and the zero-H  $\lambda_{\text{eff}}$  (see Fig. 2), we have calculated the spindependent MFP:  $\lambda_{+}$  = 90 Å and  $\lambda_{-}$  = 23 Å for bulk  $Co_{20}Ag_{80}$ ;  $\lambda_+ = 68$  Å and  $\lambda_- = 25$  Å for bulk Fe<sub>20</sub>Ag<sub>80</sub>. With these parameters, we have used relations (3) and (4) to fit the data of the t dependence of MR shown in Fig. 1, where the best fits are represented by the solid lines. In  $Co<sub>20</sub>Ag<sub>80</sub>$  system, the fitting quality is excellent, whereas in  $Fe_{20}Ag_{80}$  system, it is satisfactory. The deviation in the latter system may be due to the fact that even at H=8 T the MR is not completely saturated in the  $Fe_{20}Ag_{80}$  system.

The two parameters obtained from the fitting are  $\rho_0 = 22.7 \ \mu\Omega \ \text{cm}, \ \epsilon_S = 0.72 \ \text{for} \ \text{Co}_{20}\text{Ag}_{80} \ \text{and} \ \rho_0 = 26.4$  $\mu\Omega$  cm,  $\epsilon_S = 0.54$  for Fe<sub>20</sub>Ag<sub>80</sub>. The difference between the measured  $\rho_0$  (27 and 31  $\mu\Omega$  cm, respectively) and the fitted  $\rho_0$  is the nonmagnetic disorder resistivity. For both systems,  $\epsilon_S$  is of the order of 60%. It is known that for bulk 3D solids the ratio between the spin-fiipping and non-spin-flipping cross section is a few percent.<sup>15</sup> Therefore in granular  $Co<sub>20</sub>Ag<sub>80</sub>$  and  $Fe<sub>20</sub>Ag<sub>80</sub>$  the surface-



induced spin-flipping rate is enhanced by about 1 order of magnitude, similar to weak-localization results of thin films.

Our fitting results have revealed that the direct spinmixing effect  $(\rho_{\uparrow\downarrow})$  can only account for about 10% of the reduction in  $\Delta \rho$  as t is reduced, the predominant cause  $(> 90\%)$  of the finite-size effect is the geometrical confinement in the context of strong surface-enhanced spin flippings. To date there has been no study on the effect of spin fiipping on GMR in granular systems. In studies involving multilayers the role of spin-Gipping is receiving increasing attention.<sup>15,16</sup> In the CIP (current in plane) geometry, it is believed that the efFect of spin fIipping is not important.<sup>17</sup> However, in the CPP (current perpendicular to plane) geometry, the important length scale is the spin-diffusion length,  $^{15,16}$   $\ell_{\rm sf}$ . GMR is caused by a spin-dependent interface resistance due to an imbalance of the number of + and - electrons within  $\ell_{sf}$  of an interface. Our study here has shown that the role of spin flipping in granular systems is different from that in multilayers. The direct effect of spin flipping, in terms of mixing spin channels, is too small to be important. However, because of the enhanced spin flipping near the surfaces, the spin-dependent MFP's  $(\lambda_+ \text{ and } \lambda_-)$  become geometrically limited, and the confinement is preferentially more significant on  $\lambda_+$  than on  $\lambda_-$ . The finite-size effect of GMR is, therefore, the result of a gradual equalization of  $\lambda_+$  and  $\lambda_-$ . The comparison between granular thin films and multilayers illustrates the significant disparity in the basic physics between the two systems.

Finally, we present another anomalous magnetotransport property: the finite-size effect of the extraordinary Hall effect (EHE). Figure  $4(a)$  shows the t dependence of the spontaneous Hall resistivity  $\rho_{xy}^s$  due to EHE for the Co<sub>20</sub>Ag<sub>80</sub> series measured at  $T = 4.2$  K. The total Hall resistivity  $\rho_{xy}$  vs H is presented in the inset in Fig. 4(a) for two representative  $Co<sub>20</sub>Ag<sub>80</sub>$  samples in the

FIG. 4. (a) Thickness dependence of the spontaneous extraordinary Hall resistivity  $\rho_{xy}^{s}$  for the Co<sub>20</sub>Ag<sub>80</sub> series. Also shown is the field dependence of the Hall resistivity  $\rho_{xy}$  in the thick- (2030 Å) and thin- (110 Å) film limit. (b) The linear correlation between  $[\Delta \rho/\rho]^2$  and  $\rho^s_{xy}(t)$ .

thick- and thin-film limits. The value of  $\rho_{xy}^s$  is obtained by extrapolating the  $\rho_{xy}$  (H) curve in the high H region to  $H = 0$ . Quite interestingly,  $\rho_{xy}^s$  is negative in the bulk limit, whereas it gradually reduces its magnitude and eventually turns positive as  $t$  approaches the thinfilm limit. This shows that  $\rho_{xy}^s$  in thin films also includes a bulk and a surface contribution differing in sign. It is well known that a SO interaction induces EHE through two mechanisms: $18$  skew scatterings or side jumps. In homogeneous systems, there exists a scaling relation<sup>18</sup> between  $\rho_{xy}^s$  and  $\rho$ , i.e.,  $\rho_{xy}^s \propto \rho^n$ ,  $n = 1$  for skew scatterings and  $n = 2$  for side jumps. A recent theory on  $EHE$  by  $Zhang<sup>19</sup>$  has shown the invalidity of the above scaling relation applied to heterogeneous systems. In the current  $Co<sub>20</sub>Ag<sub>80</sub>$  series, the conventional theory of EHE fails. One observes in Fig. 4(a) that  $\rho_{xy}^s$ , similar to MR, exhibits a large finite-size efFect and even changes sign at a certain  $t$ . However,  $\rho$  itself changes little over a large  $t$ range due to the short  $\lambda_{\text{eff}}$  (see Fig. 2). A careful examination has shown that there is a new scaling relation between the GMR quantity  $\Delta \rho / \rho$  and  $\rho_{xy}^s$ . In Fig. 4(b), we plot  $[\Delta \rho(t)/\rho]^2$  vs  $\rho_{xy}^s(t)$ . The observed linear correlation between  $[\Delta \rho / \rho]^2$  and  $\rho_{xx}^s(t)$  is highly unconventional. Although the exact mechanism is not clear at the moment, we believe that the surface-induced SO interaction is the common cause of the finite-size effect of both GMR and EHE. The new correlation implies convincingly that the traditional treatment of EHE is invalid in heterogeneous systems. It is imperative in theory to treat both GMR and EHE simultaneously and self-consistently.

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