

Superconductivity at 15 K in the metastable $\text{ScNi}_2\text{B}_2\text{C}$ compound

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A superconducting transition around 15–16 K is reported for metastable, as-melted $\text{ScNi}_2\text{B}_2\text{C}$ samples. Zero resistivity $T_c(\text{zero})$ was observed at 14.6 K with $T_c(\text{onset})$ around 16.5 K. A consistent diamagnetic T_c signal was observed around 14.8–15.6 K. The new superconducting phase was identified as isostructural to the quaternary $\text{LuNi}_2\text{B}_2\text{C}$ -type phase with tetragonal lattice parameters $a = 3.34 \text{ \AA}$ and $c = 10.2 \text{ \AA}$. This superconducting phase is unstable due to the extremely small Sc^{3+} ionic size, where the Sc atom is located in the cavity formed by the B-Ni-B-C network and no superconducting transition could be observed down to 5 K after 1050 °C annealing.

Superconducting intermetallic compounds with relatively high transition temperatures up to 23 K have been reported in $R\text{-T-B-C}$ systems, where $R = \text{Y}$ or rare earths and $T = \text{Ni}$, Pd , or Pt .^{1–5} In the nickel-based system, the superconducting phase has been identified as a quaternary $\text{LuNi}_2\text{B}_2\text{C}$ -type tetragonal structure with space group $I4/mmm$. The structure is a three-dimensionally connected framework with LuC NaCl -type layers alternated with Ni_2B_2 layers where nickel is tetrahedrally coordinated by four boron atoms.⁴ The phase formation was reported in $\text{RNi}_2\text{B}_2\text{C}$ ($R = \text{Y}$, La , Ce , Sm , Tb , Dy , Ho , Er , Tm , or Lu) with a maximum T_c of 16.6 K for nonmagnetic $\text{LuNi}_2\text{B}_2\text{C}$, and a next-highest T_c of 15.6 K for nonmagnetic $\text{YNi}_2\text{B}_2\text{C}$.^{1,3,4} For magnetic rare-earth compounds, a lower superconducting transition was observed for $R = \text{Ho}$, Er , and Tm , due to the magnetic pair-breaking effect.³ Here we report the discovery of a high- T_c metastable compound $\text{ScNi}_2\text{B}_2\text{C}$, which is isostructural to $\text{LuNi}_2\text{B}_2\text{C}$.

All $\text{RNi}_2\text{B}_2\text{C}$ ($R = \text{Sc}$ or Lu) samples were prepared by arc melting the high-purity elements (Sc and Lu : 99.9%; Ni : 99.999%; B : 99.9995%; and C : 99.999%) under an argon atmosphere in a Zr-gettered arc furnace. A two-stage procedure ($\text{RNi}_2\text{B}_2 + \text{C}$) was utilized with negligible weight loss (less than 2%). Crystallographic data were obtained with a Rigaku Rotaflex rotating

anode powder x-ray diffractometer using $\text{Cu } K\alpha$ radiation with a scanning rate of 1° in 2θ per minute. A Lazy Pulverix-PC program was employed for phase identification and lattice parameter calculation. Low-field magnetic measurements were made with either a Quantum Design MPMS superconducting quantum interference device (SQUID) magnetometer ($\geq 20 \text{ G}$) or a MPMS2 ($\leq 20 \text{ G}$) SQUID magnetometer down to 5 K. ac electrical resistivity measurements (16 Hz) were carried out by the standard four-probe method in a RMC-Cryosystems closed-cycle refrigerator down to 7 K.

Powder x-ray diffraction data for all as-melted $\text{ScNi}_2\text{B}_2\text{C}$ samples show multiphase patterns. However, a pattern search indicates that the major phase can be indexed as the $\text{LuNi}_2\text{B}_2\text{C}$ -type tetragonal structure (space group $I4/mmm$), with lattice parameters $a = 3.34(1) \text{ \AA}$, $c = 10.2(1) \text{ \AA}$ and unit-cell volume $V = 114(1) \text{ \AA}^3$. These values are smaller than $a = 3.464 \text{ \AA}$, $c = 10.63 \text{ \AA}$ and $V = 127.6 \text{ \AA}^3$ for isostructural $\text{LuNi}_2\text{B}_2\text{C}$ due to the smaller Sc^{3+} ionic radius of 0.732 \AA as compared with 0.85 \AA for the larger Lu^{3+} ion.⁴

Two samples (A and B) with better x-ray diffraction patterns were used for transport and magnetic measurements. The temperature dependence of the electrical resistivity $\rho(T)$ for the as-melted $\text{ScNi}_2\text{B}_2\text{C}$ sample A is shown in Fig. 1. Zero resistivity $T_c(\text{zero})$ was observed at 14.6 K, with the superconducting onset (1%) $T_c(\text{onset})$

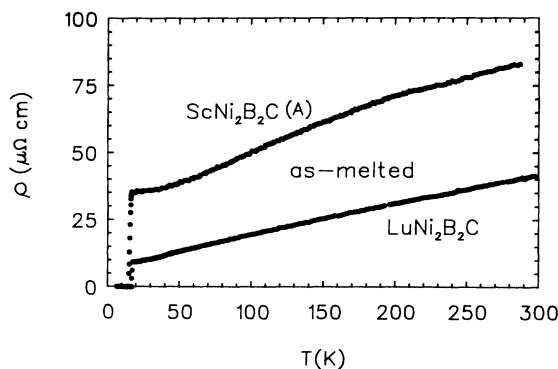


FIG. 1. Temperature dependence of electrical resistivity $\rho(T)$ for as-melted $\text{ScNi}_2\text{B}_2\text{C}$ (sample A). Data for stable, isostructural 16.6-K superconductor $\text{LuNi}_2\text{B}_2\text{C}$ are also included for reference.

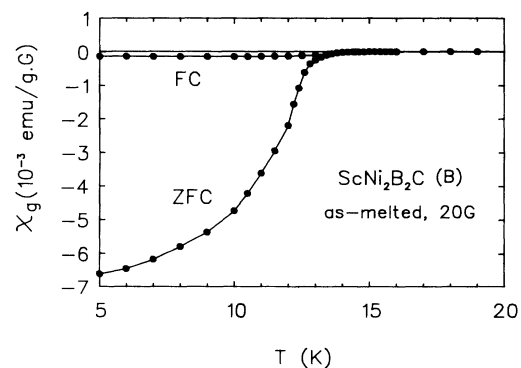


FIG. 2. Low-temperature mass magnetic susceptibility $\chi_g(T)$ for as-melted $\text{ScNi}_2\text{B}_2\text{C}$ (sample B).

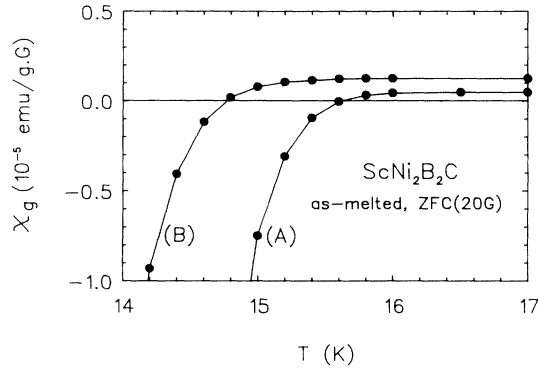


FIG. 3. Superconducting transition onset from diamagnetic measurements for as-melted $\text{ScNi}_2\text{B}_2\text{C}$ (samples *A* and *B*).

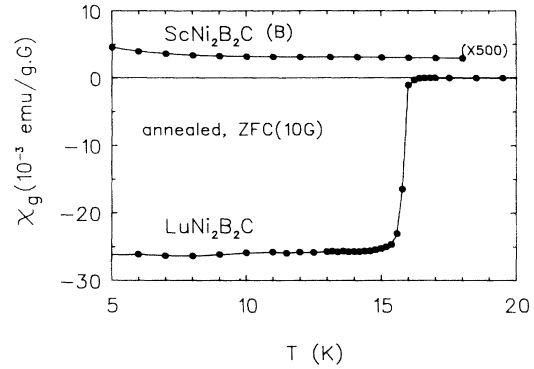


FIG. 4. Low-temperature mass magnetic susceptibility $\chi_g(T)$ for 1050°C annealed $\text{ScNi}_2\text{B}_2\text{C}$ (sample *B*) and $\text{LuNi}_2\text{B}_2\text{C}$.

around 16.5 K. The room-temperature resistivity is $82.8 \mu\Omega \text{ cm}$, with a resistivity ratio $\rho(RT)/\rho(20 \text{ K})$ of 2.33. Data for isostructural superconductor $\text{LuNi}_2\text{B}_2\text{C}$ are also included with $T_c(\text{zero})$ at 16.7 K and $T_c(\text{onset})$ around 17.6 K. The room-temperature resistivity is $41.2 \mu\Omega \text{ cm}$, with a resistivity ratio $\rho(RT)/\rho(20 \text{ K})$ of 4.43.

The low-temperature mass magnetic susceptibility $\chi_g(T)$ of the second as-melted $\text{ScNi}_2\text{B}_2\text{C}$ sample *B* is shown in Fig. 2 for both zero-field-cooled (ZFC) and field-cooled (FC) measurements in a low applied field of 20 G. A diamagnetic superconducting transition signal was observed around 14.8 K for this sample. A large ZFC shielding signal of $-6.7 \times 10^{-3} \text{ emu}/(\text{g G})$ at 5 K for the bulk sample indicates that the bulk superconductivity effect (using the calculated x-ray density of $5.42 \text{ g}/\text{cm}^3$). For the powdered sample (200 mesh), the diamagnetic shielding signal decreases to $-1.7 \times 10^{-3} \text{ emu}/(\text{g G})$ due to the multiphase nature of the as-melted sample.

The diamagnetic T_c transition is field independent in a low applied field of 5–50 G, but slightly sample dependent due to different arc-melting conditions. The highest diamagnetic T_c signal around 15.6 K for sample *A* was observed, as compared with 14.8 K for sample *B* (Fig. 3).

Contrary to the stable superconducting $\text{LuNi}_2\text{B}_2\text{C}$ phase, superconductivity in as-melted $\text{ScNi}_2\text{B}_2\text{C}$ samples is metastable. After annealing the samples at 1050°C in a sealed quartz tube (wrapped in Ta foil and then sealed under argon) for 24 h and water quenching, no superconducting transition can be observed down to 5 K from the low-temperature mass magnetic susceptibility $\chi_g(T)$ data (Fig. 4). The x-ray powder diffraction patterns for annealed samples show no trace of the tetragonal $\text{LuNi}_2\text{B}_2\text{C}$ -type phase, indicating that this superconducting tetragonal phase is entropy stabilized during the arc-melting process. Contrary to the weak, temperature-dependent, Pauli-like paramagnetic behavior of the annealed $\text{ScNi}_2\text{B}_2\text{C}$ sample [$+9.1 \times 10^{-6} \text{ emu}/(\text{g G})$ at 5 K], the annealed $\text{LuNi}_2\text{B}_2\text{C}$ bulk samples exhibits an excellent T_c transition of 16.7 K with a large and almost temperature-independent ZFC shielding signal of $-2.62 \times 10^{-2} \text{ emu}/(\text{g G})$ (x-ray density $8.49 \text{ g}/\text{cm}^3$).

The entropy-stabilized or metastable tetragonal phase

in $\text{ScNi}_2\text{B}_2\text{C}$ is apparently closely related to the size effect. Since the $\text{LuNi}_2\text{B}_2\text{C}$ -type phase is a variant on the ThCr_2Si_2 -type structure with additional carbon in the Lu plane, the Lu atom is located in the cavity formed by the B-Ni-B-C network.⁴ After the substitution of small Lu ions by even smaller Sc, the structure is thus highly unstable and the phase exists only in the high-temperature, high-entropy region of the phase diagram, with an exceptionally small tetragonal c lattice parameter. Similar metastability due to the size effect was observed in many other superconducting systems. For example, in the RRu_4B_4 system, the metastable phase was observed for the 7.2-K ScRu_4B_4 superconductor, as compared with the stable 2.1-K LuRu_4B_4 superconductor.⁶ In the superconducting $\text{R}(\text{Ir}_2\text{B}_4)$ systems ($R = \text{Ho}, \text{Er}, \text{or Tm}$), all superconducting ErRh_4B_4 -type phases are metastable.⁷ The unstable 23-K Y-Pd-B-C phase² is probably due to a similar origin, where the high- T_c $\text{YPd}_2\text{B}_2\text{C}$ -type phase can be stabilized only with the off-stoichiometric composition of $\text{YPd}_5\text{B}_3\text{C}_{1-x}$.

The T_c of 15–16 K for the present metastable $\text{ScNi}_2\text{B}_2\text{C}$ compound is only slightly lower than 16.7 K for the $\text{LuNi}_2\text{B}_2\text{C}$ compound, and almost equal to 15.6 K for the $\text{YNi}_2\text{B}_2\text{C}$ compound (with a larger Y^{3+} ionic radius of 0.893 Å), which indicates that the superconducting properties in this system may be irrelevant to the relative size of nonmagnetic R^{3+} ions ($R = \text{Sc}, \text{Y}, \text{or Lu}$). Detailed studies of metastability, lattice parameter variation, and superconducting properties of the pseudo-quaternary $(\text{Lu}_{1-x}\text{Sc}_x)\text{Ni}_2\text{B}_2\text{C}$ system are in progress. Preliminary results indicate that the unstable boundary is near the Sc-rich region ($x > 0.5$), with a stable T_c of 15.9 K for the $(\text{Lu}_{0.5}\text{Sc}_{0.5})\text{Ni}_2\text{B}_2\text{C}$ compound. For the $\text{LuNi}_2\text{B}_2\text{C}$ compound ($x = 1$), a lower critical field $H_{c1}(15 \text{ K})$ of 70 G and an upper critical field $H_{c2}(15 \text{ K})$ of 7 kG were observed from magnetization $M(H)$ measurements for this type-II superconductor.

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