# Adsorbed states of NH<sub>3</sub> and C<sub>6</sub>H<sub>6</sub> on the Si(111)( $\sqrt{3}\times\sqrt{3}$ )R 30°-B surface: Thermal-desorption and electron-energy-loss-spectroscopy studies

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Adsorbed states of NH<sub>3</sub> and C<sub>6</sub>H<sub>6</sub> on a Si(111)( $\sqrt{3} \times \sqrt{3}$ )R 30°-B surface have been studied by the use of thermal desorption and high-resolution electron-energy-loss spectroscopy. At 90 K, a bulklike  $NH<sub>3</sub>$ multilayer,  $NH_3$  hydrogen bonded to a chemisorbed one, and molecularly chemisorbed  $NH_3$  exist on the  $(\sqrt{3}\times\sqrt{3})R30^\circ$ -B surface. These desorb at ~115, 140, and 170 K, respectively. Chemisorbed NH<sub>3</sub> has a very low NH stretching energy of  $\sim$  370 meV, and its N-H bonds are significantly weakened. A 2.0-eV loss, which corresponds to the electronic transition from an occupied backbond state to an empty dangling-bond state of the Si adatom on the  $(\sqrt{3}\times\sqrt{3})R30^\circ$ -B surface, is removed upon NH<sub>3</sub> adsorption. It is proposed that chemisorbed NH<sub>3</sub> is coordinated to the Si adatom on the surface.  $C_6H_6$  does not chemisorb on the  $(\sqrt{3} \times \sqrt{3})R30^\circ$ -B surface even at 90 K, but only physisorbs at <160 K. The adsorbed states of NH<sub>3</sub> and C<sub>6</sub>H<sub>6</sub> on the  $(\sqrt{3}X\sqrt{3})R30^{\circ}$ -B surface are quite different from those on clean Si surfaces, and the reactivity criterion of the surface is discussed. It is also found that the  $(\sqrt{3} \times \sqrt{3})$ R30<sup>o</sup>-B surface has few defects when a large number of B atoms ( $\sim 7 \times 10^{19}$  cm<sup>-3</sup>) exists in the subsurface, which was evaluated by analyzing the surface-plasmon loss.

### I. INTRODUCTION

Recent measurements have demonstrated that a  $(\sqrt{3} \times \sqrt{3})$ R 30°-B superstructure is formed on Si(111) by the high-temperature heat treatment of a heavily 8-doped Si wafer,<sup>1</sup> or by an adsorption of decaborane  $(DB)$ ,  $B_{10}H_{14}$ , followed by the thermal decomposition.<sup>2</sup> It has been found that the Si(111)( $\sqrt{3} \times \sqrt{3}$ )R 30°-B surface has Si adatoms at the topmost layer ( $T_4$  sites) and B atoms at the subsurface substitutional  $(S_5)$  sites directly beneath the adatoms, and that the Si adatom has an empty dangling-bond level due to a charge transfer from the adatom to the B atom beneath the adatom. $2^{-4}$  The chemical reactivity of Si modified by 8 atoms is of considerable interest, and the interactions of the  $(\sqrt{3} \times \sqrt{3})$ R 30°-B surface with hydrogen atom,<sup>5</sup> oxygen,<sup>6</sup> and ammonia<sup>2,7</sup> have been studied. It has been found that the  $(\sqrt{3} \times \sqrt{3})R30^\circ$ -B surface has chemical reactivities towards gas molecules quite different from those of clean Si surfaces.

We have investigated gas adsorption on clean Si surfaces for several years. For example, ammonia chemisorbs dissociatively on both  $Si(111)(7\times7)$  and  $Si(100)(2\times1)$  surfaces by forming the SiNH<sub>2</sub> and SiH surface species at 300  $K$ .<sup>8,9</sup> On the other hand, the adsorbed state of benzene shows a distinct crystal-face specificity, and we have proposed that molecularly chemisorbed benzene is  $\pi$  bonded to the Si(111)(7×7) surface at  $\leq 300 \text{ K},^{10}$  and di- $\sigma$  bonded to two adjacent Si atoms<br>on the Si(100)(2×1) surface at  $\lt 500 \text{ K}$ .<sup>11</sup> In this on the Si(100)( $2 \times 1$ ) surface at <500 K.<sup>11</sup> In this work, we have studied the interactions of the  $Si(111)(\sqrt{3} \times \sqrt{3})R30^\circ$ -B surface with NH<sub>3</sub> and C<sub>6</sub>H<sub>6</sub> at 90-300 K by using thermal-desorption spectroscopy (TDS) and high-resolution electron-energy-loss spectroscopy (EELS). The  $(\sqrt{3} \times \sqrt{3})R30^\circ$ -B surface was formed by heating  $Si(111)$  preexposed to DB.

By the use of photoemission spectroscopy, Avouris et  $al.$ <sup>2</sup> have proposed that there exists ammonia coordinated to the Si adatom on the Si(111)( $\sqrt{3} \times \sqrt{3}$ )R 30°-B surface at 90 K. A preliminary EELS study was carried out by Yates<sup>7</sup> for the NH<sub>3</sub>-exposed  $(\sqrt{3} \times \sqrt{3})R30^\circ$ -B surface. However, the discussion was not given in detail and an evidence for  $NH_3$  coordinated to the surface was not shown in the latter study. We have found vibrational and electronic transitions, which indicate the existence of NH<sub>3</sub> coordinated to the Si adatom on the  $(\sqrt{3} \times \sqrt{3})R30^\circ$ -B surface at 90 K. C<sub>6</sub>H<sub>6</sub> does not chemisorb on the  $(\sqrt{3} \times \sqrt{3})R30^\circ$ -B surface, but only physisorbs on the surface at  $\langle 160 \text{ K}$ . By comparing the interactions of the  $(\sqrt{3} \times \sqrt{3})R30^\circ$ -B surface with NH<sub>3</sub> and  $C_6H_6$ , the reactivity criterion of the surface has been proposed by both absolute electronegativity<sup>12</sup> of the molecule and the nature of the orbitals that interact with the surface.

In this work, we also examined the  $(\sqrt{3} \times \sqrt{3})R30^\circ$ -B surface by EELS. The EEL spectra of the surface have been analyzed with the aid of the dipole scatterin theory,  $13-15$  and the information on free carriers, which were introduced in the near-surface region by the hightemperature heat treatment in preparing the surface, was obtained: The concentration and Drude relaxation time of carriers in the near-surface region were estimated by the analysis of the surface-plasmon-loss profile, and the total number per unit area by that of the quasielastic peak.

#### II. EXPERIMENTAL

The experiments were performed by the in situ combined techniques of TDS, EELS, Auger electron spectroscopy (AES), and low-energy electron difFraction (LEED), by using an ultrahigh vacuum system with the base pressure of  $1 \times 10^{-8}$  Pa. The experimental apparatus used in the present study has been described elsewhere.<sup>11</sup> For the present study has been described elsewhere.<sup>11</sup> For specular-mode EELS measurements, an electron incidence angle  $\theta_i$ , of 65° with respect to the surface normal was used. The off-specular measurements were performed by rotating the monochromator around an axis, which was perpendicular to the incidence plane of electron beam. Electrons were scattered along the  $\lceil 2\overline{1} \overline{1} \rceil$  azimuth. All EEL spectra were measured at 90 K. The heating rate of 2.5 K/s was used for TDS measurements. The sample used  $(7 \times 8 \times 0.5 \text{ mm}^3)$  was a Si(111) wafer of p-type, B doped, and 3000  $\Omega$  cm (B concentration: <10<sup>13</sup> cm<sup>-3</sup>). The clean Si(111) surface, showing a sharp (7 $\times$ 7) LEED pattern, was prepared by several cycles of  $Ar^+$ -ion bombardment (450 eV,  $10-20 \mu A \text{ cm}^{-2}$ , 30 min) and annealing (1100K for 2 min). No impurities were observed on the surface thus prepared within the detection limit of AES and EELS. Research-grade  $NH<sub>3</sub>$  (99.99% purity) and  $C_6H_6$  (99.0 mol % purity) were used. DB (99% purity, SIGMA Chemical Company, U.S.A.) was used as a source of B, and the sample was exposed to DB after vaporization. DB and  $C_6H_6$  were purified by freeze-pumpthaw cycles and were checked for the purities with a mass spectrometer. The DB,  $NH_3$ , and  $C_6H_6$  gases were introduced into the vacuum chamber via an 8-mm-diam gas doser located 10 mm from the sample surface. The gas pressure was monitored by the use of a nude-type Bayard-Alpert ion gauge, and calibrated by the ion-gauge sensitivity factor relative to  $N_2$ : 0.6 for NH<sub>3</sub>, 5.2 for  $C_6H_6$ , and 7.2 for DB estimated by referring to that of  $C_8H_{16}$ , which has the same number of electrons in one molecule. Exposures are given below in langmuir (1  $L = 1.33 \times 10^{-4}$  Pa s).

#### III. RESULTS AND DISCUSSION

# A. Si $(111)(\sqrt{3}\times\sqrt{3})R30^{\circ}$ -B

We have formed the  $Si(111)(\sqrt{3}\times\sqrt{3})R30^\circ$ -B surface by exposing Si(111) to DB at 300 K and subsequently heating the sample to 1270 K for <sup>1</sup> min [heating (and cooling) rate:  $5 K/s$ ] by referring to the scanning tunneling microscope (STM) study.<sup>2</sup> The Auger-peak-height ratio of  $B(KLL)/Si(LVV)$  for  $Si(111)$  after heating to 1270 K was nearly saturated for the surface exposed to  $>0.3$ -L DB, and the value was 0.014 (for the primary energy of 2 keV) at the saturation in good agreement with the previous work.<sup>16</sup> For the Si(111) surface exposed to  $> 0.2$ -L DB and heated to 1270 K, a well established  $(\sqrt{3} \times \sqrt{3})$ R 30° LEED pattern was observed. According to these results, the  $(\sqrt{3} \times \sqrt{3})R^{\circ}$ -B superstructure is apparently completed on Si(111) with the DB exposure of  $>0.3$  L. In the previous studies,<sup>2,7</sup> however, a small amount of defects was found on the  $(\sqrt{3} \times \sqrt{3})R30^\circ$ -B surface prepared by the thermal decomposition of DB on

 $Si(111)$ . According to the STM study,<sup>2</sup> the defect characteristic of the  $(\sqrt{3} \times \sqrt{3})R30^{\circ}$ -B surface is a B deficiency at the  $S_5$  site, where the B atom is replaced by Si. Indeed we obtained the TD and EEL spectra, which indicate that the defects exist on the  $(\sqrt{3} \times \sqrt{3})R30^{\circ}$ -B surface formed with the DB exposure of  $< 0.5$  L. However, for the  $(\sqrt{3} \times \sqrt{3})$ R 30°-B surface formed with the DB exposure of  $\geq 0.5$  L, no experimental evidence indicative of the defects existing on the surface was found in both TDS and EELS measurements. Thus, it is considered that the  $(\sqrt{3} \times \sqrt{3})$ R 30°-B surface is formed with few defects by exposing Si(111) to  $\geq$  0.5-L DB and subsequent heating in our preparation procedure. In the following gas adsorption experiments, the  $(\sqrt{3} \times \sqrt{3})R30^\circ$ -B surface was prepared with 0.5-L DB. Since the high-temperature heat treatment is needed in order to bring B atoms from the surface of Si(111) to the  $S_5$  substitutional sites by breaking Si-Si bonds, it is unavoidable for a part of B atoms to diffuse deeper into the substrate. This is the reason why excess DB has to be supplied.

Figure 1(a) shows an EEL spectrum in the specular mode of the  $Si(111)(7\times7)$  surface exposed to 1.5-L DB at 90 K and subsequently heated to 300 K to remove physisorbed DB (recorded at 90 K). Losses are observed at 98, 116, 189, 261, 322, and 462 meV. Essentially, little change was observed in the vibrational spectra for different exposures except for the intensity variation. The loss energies agree with the vibrational energies of solid DB (Ref. 17) (Table I), except for the 261 and 462



FIG. 1. (a) EEL spectrum in the specular mode of  $Si(111)$  exposed to 1.5-L  $B_{10}H_{14}$  (DB) at 90 K and subsequently heated to 300 K. Primary energy  $E_p = 4.5$  eV. (b) and (c) EEL spectra in the specular mode of  $Si(111)$  preexposed to (b) 0.05- and (c) 0.5-L DB at 300 K and subsequently heated to 1270 K.  $E_p = 7.5$ eV. All the measurements were made at 90 K. Open circles represent the calculated spectrum (see text for more explanation).



Energy (meV)	Solid $B_{10}H_{14}^a$ Mode	$B_{10}H_{14}$ on $Si(111)(7\times7)$ Energy (meV)	
88 89 93	B-H bending parallel to $C_2$ axis		
101		98	
102	<b>B-B</b> skeletal		
106	stretching		
113			
114			
116		116	
192		189	
195	$B-H \cdots B$ bridge		
197			
232			
314			
315			
318	<b>B-H</b> stretching		
320		322	
324			
325			

<sup>a</sup>Only vibrational modes showing relatively strong intensities in the infrared spectrum are listed (see Ref. 17).

meV. Thus, it is concluded that DB is adsorbed nondissociatively on Si(111) at 300 K. This is consistent with the previous STM study:<sup>2</sup> a found disk image was observed for the DB adsorption on Si(111) at 300 K. The intensities of the 261 and 462-meV losses were observed to increase with time, and the losses are attributed to the  $SiH$  and OH stretching modes, respectively, of  $H<sub>2</sub>O$  from the background.<sup>18</sup>

Figures 1(b) and 1(c) show EEL spectra in the specular mode of the  $Si(111)$  surface exposed to 0.05 and 0.5-L DB at 300 K, subsequently heated to 1270 K and cooled to 90 K, respectively. For the Si(111) surface preexposed to 0.05-L DB and heated to 1270 K, the Auger-peak-height ratio of B(KLL)/Si(LVV) was 0.005 ( $\frac{1}{3}$  of saturation For this surface, losses are observed at 68- and 100 meV [Fig. 1(b)]. Loss features associated with the SiH and  $BH<sub>r</sub>$  species, e.g., whose energies of stretching modes are expected to be  $\sim$  260 and 320 meV, respectively, are not observed. Hydrogen on the surface is desorbed by heating. The 68- and 100-meV losses are attributed to surface vibrations related to the structure having a Si adatom at the  $T_4$  site on Si(111) along with a Si (68 meV) or B atom (100 meV) at the  $S_5$  site. According to Northrup,<sup>19</sup> it is expected that two surface-phonon modes at  $\overline{\Gamma}$  point are observed for the structure, which has adatoms at the  $T<sub>4</sub>$ sites on Si(111). One is a mode having the displacement eigenvector localized mainly on the two atoms located directly beneath the adatom, and the other a mode involving primarily the adatom and the surface Si atoms bonded to the adatom. The losses corresponding to the former were observed at 71 meV for the  $Si(111)(7\times7)$ 

surface<sup>20</sup> and at 69 meV for Si(111)( $\sqrt{3} \times \sqrt{3}$ )R 30°-Al.<sup>21</sup> It is interpreted that the phonon energy is shifted for the structure having B at the  $S_5$  site due to a lighter mass of the B atom compared with Si. Thus, it is concluded that the structure having the Si adatom at the  $T_4$  site and B at the  $S_5$  substitutional site is formed partially on Si(111), together with that having Si atoms both at the  $T_4$  adatom and  $S_5$  substitutional sites. The vibrational energy of the latter mode is expected to be  $\sim$  30 meV,  $20 - 22$  but we could not detect the mode.

As the surface B concentration is nearly saturated (0.5-L DB), an EEL spectrum of the surface is varied [Fig. 1(c)]. A broad and intense loss feature around  $\approx$  100 meV and a loss peak at 99 meV are observed. The 99-meV loss is the surface phonon as described above. The broad loss is not attributed to surface vibrations by referring to Fig. 1(b), but to a surface-plasmon excitation of free carriers in the near-surface region. The amount of the free carriers (and B atoms) can be estimated by analyzing the observed plasmon loss with the use of the dipole scattering theory.<sup>13</sup>

It is noted that the probing depth  $l$  of EELS depends on a given loss energy  $\hbar \omega$ :<sup>13</sup>  $l \sim (2E_p / \hbar \omega) /k$  with  $E_p = \hbar^2 \vec{k}^2 / 2m$  being a primary energy. For  $E_p = 7.5$  eV and  $\hbar \omega$  ~ 100 meV, l ~ 110 Å. We will assume that the carrier concentration  $n<sub>b</sub>$  in the near-surface region, which is probed by electrons suffering  $\sim$  100-meV energy loss, can be taken as constant. The contribution from the free carriers to the dielectric function  $\epsilon(\omega)$  is given by

$$
\epsilon(\omega) = \epsilon_0 - \frac{4\pi n_b e^2}{m^* \omega(\omega + i/\tau)} \tag{1}
$$

where  $\epsilon_0$  is a bulk dielectric constant (Si:11.7),  $m^*$  an effective mass, and  $\tau$  the Drude relaxation time. When the  $(\sqrt{3} \times \sqrt{3})R30^\circ$ -B superstructure is perfectly stoichiometric and ordered, surface (and defect) states do not exist in the bulk band gap. In this case, the Fermileve1 pinning does not occur, and the Hat-band condition exists near the surface.<sup>23</sup> The contribution from the surface optical phonon will be ignored here.

Under the above conditions, the electron-energy-loss probability  $P(\mathbf{k}, \mathbf{k}') d \Omega_{\mathbf{k}} d(\hbar \omega)$ , that an incident electron is scattered into the range of energy losses between  $\hbar \omega$ and  $\hbar(\omega+d\omega)$  and into the solid angle  $d\Omega_{\mathbf{k}}$  around the direction of  $k'$ , is given by  $13$ 

$$
P(\mathbf{k}, \mathbf{k}') = A(\mathbf{k}, \mathbf{k}') (n_{\omega} + 1) \text{Im} \left[ \frac{-1}{1 + \epsilon(\omega)} \right], \quad (2)
$$

where k and k' are the wave vectors of an incident and scattered electron;  $\hbar^2 (k^2 - k'^2)/2m = \hbar \omega$ ,  $n_\omega$  the Bose-Einstein factor, and  $A(k, k')$  a kinematic factor defined by k and  $k'.^{14}$  Actually, the measured loss intensity is proportional to  $P(k, k')$  integrated over the solid angle subtended by the entrance slit of the analyzer. The analyzer of our EEL spectrometer consists of a 127° cylindrical deflector with an entrance slit shaped rectangularly. In the following calculation, the kinematic factor A was integrated over  $|\alpha| \le 0.5^{\circ}$  and  $|\beta| \le 3^{\circ}$ , where the in-plane angle  $\alpha$  is the difference between the incidence angle  $\theta_i$  and the angle  $\theta_s$ , which is the angle between the projection of k' onto the scattering plane and the surface normal, and the out-of-plane angle  $\beta$  the angle between  $\mathbf{k}'$ and its projection onto the scattering plane.<sup>24</sup> These angles were determined consistently with the experiments.

The fitting parameters are the carrier concentration  $n<sub>b</sub>$ of the *substrate* and the Drude relaxation time  $\tau$  assuming the hole effective mass  $m^*$  of 0.35m.<sup>25</sup> Open circles in Fig. 1(c) show an EEL spectrum calculated by using Eq. (2) with  $n_b$  of  $7.0 \times 10^{19}$  cm<sup>-3</sup> and  $\tau$  of  $5.5 \times 10^{-15}$  s. The calculated spectrum agrees well with the observed one in spite of the assumptions made in the calculation. In the course of fitting, we have found that similar results are obtained for the in-plane angular apertures ranging from  $|\alpha| \leq 0.4$ ° to  $|\alpha| \leq 3$ °, and that both  $n_b$  and  $\tau$  are decreased at most 10% with expanding the aperture size.

The evaluated value of  $n_b$  is much larger than the intrinsic carrier concentration of  $5 \times 10^{18}$  cm<sup>-3</sup> at 1270 K. Thus, the diffusion of B atoms into the substrate is the extrinsic one.<sup>26</sup> The diffusion length is  $1.2 \times 10^3$  Å assuming the constant surface concentration of  $7 \times 10^{19}$  cm<sup>-3</sup>, the heating temperature of 1270 K, and the heating time of 60 s. $^{26}$  The length is the lowest limit of the actual one. Therefore, the first assumption made in the calculation of the plasmon profile, that  $n_b$  is constant in the nearsurface region, gives a good approximation. Further, the large value of  $n_b$  implies that freeze-out of carriers does not occur and that the concentration can be regarded as that of diffused B atoms. It is noted that  $n_b$  evaluated for our sample is 3.5 times as large as that for the  $(\sqrt{3} \times \sqrt{3})R30^{\circ}$ -B surface prepared by Chen, Colaianni, and Yates.<sup>16</sup> It is considered that the larger concentration suppresses the production of the B deficiency in the  $(\sqrt{3} \times \sqrt{3})R30^{\circ}$ -B layer. Thus, our sample is expected to have fewer defects than that prepared by them. With  $\tau$  of  $5.5 \times 10^{-15}$  s and  $m^*$  of 0.35m, the mobility in the nearsurface region is estimated to be 28 cm<sup>2</sup> V<sup>-1</sup> s<sup>-1</sup>. This is  $\sim \frac{1}{2}$  of the bulk value corresponding to the same impurity  $\frac{1}{2}$  of the bulk value corresponding to the same impurity concentration ( $\sim$  50 cm<sup>2</sup> V<sup>-1</sup> s<sup>-1</sup> at 300 K).<sup>26</sup> It is considered that this is related to the short heating time in the preparation and that a large number of imperfections may exist in the near-surface region.

We have found that the quasielastic-peak width of the EEL spectra of the Si $(111)(\sqrt{3} \times \sqrt{3})R30^\circ$ -B surface depends on the sample temperature. The full width at half maximum (FWHM) of the quasielastic peak of the EEL spectra of the  $(\sqrt{3} \times \sqrt{3})R30^{\circ}$ -B surface formed with 0.5-L DB is shown as a function of temperature by open circles in Fig. 2. The quasielastic-peak width is increased almost linearly with temperature in the measured range from 90 to 370 K. It is possible to obtain the total number of free carriers in the near-surface region by analyzing the quasielastic peak. Losses involved in the quasielastic peak are a few meV, and thus  $l > 10^3$  A. The length is so large that all free carriers introduced by the short-time heating adopted in our preparation procedure may be included.

For the quasielastic-peak profile, Persson and co-'workers<sup>14,15</sup> have derived a simple formula, taking account of multiple scattering, which becomes important



FIG. 2. Temperature dependence of the width (FWHM) of the quasielastic peak for the Si(111)( $\sqrt{3} \times \sqrt{3}$ )R30°-B surface. The experimental reseults are shown by open circles and the calculated results (for  $h = 0.052$ ) by the solid curve.  $E_p = 7.5$  eV. The inset shows the experimental (open circles) and calculated (solid curve) profiles of the quasielastic peak at 366 K.

for very small loss energies, i.e., below a few tens meV. According to Persson and co-workers,<sup>14,15</sup> the quasielastic-peak profile  $P(\omega)$  is given by

$$
P(\omega) \simeq \frac{1}{\pi} \left[ \frac{\Gamma/2}{\omega^2 + (\Gamma/2)^2} \cos \phi + \frac{\omega}{\omega^2 + (\Gamma/2)^2} \sin \phi \right], \quad (3)
$$

where  $\Gamma = 4C h k_B T$ ,  $\phi = C h$ ,

$$
C = \frac{4}{\pi} \frac{1}{\cos^2 \theta_i} \frac{1}{ka_0} \frac{1}{\epsilon_0 + 1} \tag{4}
$$

$$
h(1/\beta\tau,\theta_i) = \int_0^\infty dx \, f(x,\theta_i) \frac{1/\beta\tau}{(1/\beta\tau)^2 + x^2(1 - 1/\epsilon_0)^2} ,
$$
\n(5)

$$
\beta = \frac{4\pi n e^2}{m^*(\epsilon_0 + 1)} \frac{1}{v} \tag{6}
$$

and

$$
f(x, \theta_i) = \frac{1}{x} \int_0^{2\pi} d\phi \left[ 1 + \left[ \frac{1/x}{\cos \theta_i} - \tan \theta_i \cos \phi \right]^2 \right]^{-2} .
$$
\n(7)

In these expressions,  $\theta_i$  is an electron incidence angle with respect to the surface normal,  $k_B$  the Boltzmann constant,  $a_0$  the Bohr radius, n a carrier concentration per unit area, and  $v$  a velocity of an incident electron. Equation (3) holds if (i) almost all inelastically scattered electrons pass into the analyzer, which is met for very small loss energies in the specular-mode measurements, (ii)  $\hbar \beta \gg \hbar \omega$ , and (iii) Ch / $\pi \ll 1$ . C = 0.76 for  $E_p = 7.5$ eV and  $\theta_i = 65^\circ$ . The last two conditions are satisfied in the present case, which will be found shortly. It is noted that the broadening of the quasielastic peak for the  $(\sqrt{3} \times \sqrt{3})R30^\circ$ -B surface is a direct consequence of the Drude damping from the condition (ii), contrary to that for the  $Si(111)(7\times7)$  surface caused by multiple excitation of low-energy two-dimensional plasma oscillation.<sup>14</sup>

The temperature dependence of the quasielastic-peak width is calculated using Eq. (3) and compared with the experiment (Fig. 2). It is obvious from Eq. (3) that there is only a single fitting parameter  $h$ , which is a function of  $1/\beta\tau$ . The solid line in Fig. 2 shows the calculation carried out for  $h = 0.052$  ( $1/\beta\tau = 0.033$ ) independent of temperature. The spectrometer window is assumed to be a Gaussian of 7.7-meV FWHM. The agreement between the calculation and experiment is good. The reason why <sup>h</sup> is independent of the temperature is that the population of free carriers is concentrated in the high 8 concentration region near the surface and that both carrier concentration and mobility are almost independent of temperature for very high B concentration, say  $\geq 10^{19}$  $cm^{-3}$ .<sup>27</sup> The inset of Fig. 2 shows the experimental (open circles) and calculated (solid curve) profile of the quasielastic peaks at 366 K. By using  $1/\beta\tau$  of 0.033 and  $\tau$  of 5.5  $\times$  10<sup>-15</sup> s,  $\hbar$ B = 3.6 eV, and the total number of free carriers introduced near surface region is estimated to be  $1.2 \times 10^{15}$  cm<sup>-2</sup>.

# **B.** Si(111)( $\sqrt{3} \times \sqrt{3}$ )R 30°-B/NH<sub>3</sub>

TDS measurements after exposing the  $Si(111)(\sqrt{3}\times\sqrt{3})R30^{\circ}$ -B surface to NH<sub>3</sub> at 95 K showed that  $NH_3$  was the only desorption product and  $H_2$  was not detected. This suggests that  $NH<sub>3</sub>$  is adsorbed nondissociatively and is not decomposed thermally on the surface. TD spectra for the  $(\sqrt{3} \times \sqrt{3})R30^{\circ}$ -B surface exposed to an increasing amount of  $NH_3$  at 95 K are shown in Fig. 3 (heating rate was  $2.5 \text{ K/s}$ ). For small exposures



FIG. 3. TD spectra of NH, (mass 17) from the  $Si(111)(\sqrt{3}\times\sqrt{3})R30^\circ$ -B surface exposed to various amounts of  $NH<sub>3</sub>$  at 95 K. The heating rate was 2.5 K/s.

 $(< 1$  L), three desorption peaks are observed at  $\sim$  115, 140, and 170 K. No peak was observed at higher temperatures. The  $\sim$  115-K peak intensity is increased with increasing exposure. The other peaks are merged into the tail of this peak, which results from the slow pumping of  $NH<sub>3</sub>$  in the neighborhood of the ionization chamber of the mass spectrometer. The rising edges of the  $\sim$  115-K peak are common in the TD spectra of Fig. 3, and the peak is shifted to higher temperature with increasing exposure. These indicate the zeroth-order desorption, and the  $\sim$  115-K peak is attributed to the desorption from a bulklike multilayer. Similar desorption spectra, showing three intact desorption peaks, were observed for  $NH_3$  adsorbed on metals. For example, Gland, Sexton, and Mitchell<sup>28</sup> observed the NH<sub>3</sub> desorption peaks at 109, 122, and 180 K for the Ag(110)-NH<sub>3</sub> system, which were assigned to the  $NH_3$  multilayer,  $NH_3$  hydrogen bonded to a chemisorbed  $NH<sub>3</sub>$ , and molecularly chemisorbed  $NH<sub>3</sub>$ , respectively. EELS measurements indicate that chemisorbed NH<sub>2</sub> exists on the  $(\sqrt{3} \times \sqrt{3})R30^{\circ}$ -B surface at low temperatures, as will be described later. The heats of desorption for the 140 and 170-K peaks are estimated to be 8 and 10 kcal/mol, respectively, by the use of the Redhead formula<sup>29</sup> assuming a first-order desorption and a preexponential factor of  $10^{13}$  s<sup>-1</sup>. Thus, the 140 and 170-K peaks seen in Fig. 3 are attributed to  $NH_3$  hydrogen bonded to a chemisorbed  $NH_3$  and molecularly chemisorbed NH<sub>3</sub> on the  $(\sqrt{3} \times \sqrt{3})R30^{\circ}$ -B surface, respectively.

Figures 4(a) and 4(b) show EEL spectra measured in the specular and off-specular ( $\Delta\theta = 3^{\circ}$ ; see the inset of Fig. 4 for the definition of the off-specular angle  $\Delta\theta$ ) modes for the Si(111)( $\sqrt{3} \times \sqrt{3}$ )R 30°-B surface exposed to 5-L NH<sub>3</sub> at 90 K, respectively. For smaller exposures ( $\leq 1$ )



FIG. 4. EEL spectra of the Si(111)( $\sqrt{3} \times \sqrt{3}$ )R30°-B surface exposed to 5-L NH<sub>3</sub> at 90 K [(a)  $\Delta\theta$ =0°, (b)  $\Delta\theta$ =3°], and (c) subsequently heated to 125 K (heating rate was 2.<sup>5</sup> K/s) and cooled to 90 K ( $\Delta\theta$ =0°).  $E_p$  = 7.5 eV. The off-specular angle  $\Delta\theta$  is shown in the inset.

L), the losses associated with the  $(\sqrt{3} \times \sqrt{3})R30^{\circ}$ -B surface were found to predominate and those associated with adsorbed  $NH_3$  were difficult to be resolved. Losses associated with adsorbed NH<sub>3</sub> on the  $(\sqrt{3} \times \sqrt{3})R 30^{\circ}$ -B surface are resolved at 43,  $\sim$ 370, and 424 meV in the specular-mode spectrum [Fig. 4(a)], and at 43, 132, 163, 207, 362, 372, and 424 meV in the off-specular-mode spectrum [Fig. 4(b)]. By comparing Figs. 4(a) with 4(b), it is considered that the 43-meV loss is excited mainly by the dipole scattering mechanism and that the  $\sim$ 370 and 424-meV losses mainly by the impact mechanism.<sup>13</sup> As NH<sub>3</sub> is adsorbed on the  $(\sqrt{3} \times \sqrt{3})R30^\circ$ -B surface, the intensity of the plasmon loss relative to that of the elastic peak is decreased, and the spectral profile around 85—150 meV is changed [Figs. 1(c) and 4(a)]. One reason for the decreased intensity of the plasmon loss is the screening by the overlayer of  $NH<sub>3</sub>$  adsorbed on the surface. Since the surface-phonon loss, which is clearly resolved in Fig. 1(c), is merged into the loss feature around 85-150 meV in Fig. 4(a) and losses associated with adsorbed  $NH_3$  are observed in this loss energy region, the loss feature around 85-150 meV has some contribution from losses of adsorbed  $NH_3$ . The spectral profile of the plasmon loss may be varied by some change in the surface electronic states induced by the  $NH<sub>3</sub>$  adsorption. This has prevented us from deciding how to subtract the background of the plasmon loss for the other losses and by what mechanism the other losses are excited. Thus, we will not make any discussion about the observed losses by the excitation mechanisms.

The 43-, 132-, 207-, and 424-meV losses are close to vibrational energies of gas- or solid-phase  $NH_3$  (Refs. 30) and 31) (Table II), and are attributed to the hindered rotation, symmetric deformation, degenerate deformation, and (symmetric and) degenerate stretching modes of physisorbed  $NH<sub>3</sub>$ , respectively. It is considered that these losses also have contributions from the vibrational modes of  $NH<sub>3</sub>$  hydrogen bonded to the chemisorbed molecule, as will be described later.

The energies of the 163- and  $\sim$ 370-meV losses are markedly difFerent from any vibrational energies of free NH<sub>3</sub>, and the losses are attributed to the vibrational modes of chemisorbed NH<sub>3</sub> on the  $(\sqrt{3} \times \sqrt{3})R$  30°-B surface. Since the SiH stretching mode is not observed,  $NH<sub>3</sub>$ is chemisorbed nondissociatively on the surface. This is consistent with the TDS results. The 163-, 362-, and 372-meV losses are attributed to the symmetric deformation, symmetric stretching, and degenerate stretching modes, respectively. According to the infrared and Raman spectroscopy studies of metal ammine complexes,<sup>32</sup>  $NH<sub>3</sub>$  in the complexes has the vibrational energies of the symmetric deformation, symmetric stretching, and degenerate stretching modes at 135-170, 378-404, and 391—416 meV, respectively (Table II). For example,  $[Pt(NH<sub>3</sub>)<sub>6</sub>]Cl<sub>4</sub>$  has the corresponding energies at 170, 378, and 391 meV, respectively.<sup>32</sup> Thus, it is concluded that chemisorbed NH<sub>3</sub> is coordinated to the  $(\sqrt{3} \times \sqrt{3})R30^\circ$ -B surface. Because the Si adatom is charged positively owing to the charge transfer to the B atom,<sup>2-4</sup> it is considered that the chemisorbed state similar to the coordinate bonding to metal ions is realized on the  $(\sqrt{3} \times \sqrt{3})$ R 30°-B surface. The same conclusion has been reached by Avouris et  $al.$ <sup>2</sup>

One might attribute the  $NH_3$  stretching modes of  $\sim$ 370 meV, which are significantly lowered from the values of free  $NH<sub>3</sub>$ , to the hydrogen-bonding interaction with adsorbed  $NH<sub>3</sub>$  molecules or that with the surface Si atom(s). $^{13}$  The former case can be readily excluded by comparing the energies of gas- and solid-phase  $NH<sub>3</sub>$  listed in Table II. With respect to the latter case, the  $\sim$ 370meV-loss band has a peak width similar to that of the 424-meV loss, and the  $\sim$ 370- and 424-meV losses are clearly resolved. Further, the corresponding significant lowering of the CH stretching modes were not observed for the Si(111)( $\sqrt{3} \times \sqrt{3}$ )R 30°-B/C<sub>6</sub>H<sub>6</sub> system (Sec. III C). Thus, the possibility can be excluded.

The losses related to the other vibrational modes of chemisorbed  $NH<sub>3</sub>$  are not resolved, possibly due to the vibrational energies, which are close to those of physisorbed  $NH_3$  or surface phonon, and/or due to the small dynamic dipole moments of the modes.

Figure 4(c) shows an EEL spectrum in the specular mode of the  $(\sqrt{3} \times \sqrt{3})R30^\circ$ -B surface exposed to 5-L  $NH<sub>3</sub>$  at 90 K, and subsequently heated to 125 K (and cooled to 90 K}. Changes in relative intensities among losses, especially between the  $\sim$ 370- and 424-meV-loss

	NH,	NH <sub>3</sub>	$MetaI-NH3$	On Si(111)( $\sqrt{3} \times \sqrt{3}$ )R 30°-B	
Mode	gas (Ref. 30)	solid (Ref. 31)	complexes (Ref. 32)	Physisorbed NH,	Chemisorbed NH,
$l(NH_3)$		45		43	
$\rho_r(NH_3)$			$73 - 118$		
$\delta_{\rm s}(\mathbf{NH}_3)$	118	131	$135 - 170$	132	163
$\delta_d(NH_3)$	202	204	$194 - 207$	207	
$v_s(NH_3)$	414	400	$378 - 404$		362
$v_d(NH_3)$	423	419	$391 - 416$	424	372
$v(X-NH_3)$			$30 - 71$		

TABLE II. Vibrational energies of NH<sub>3</sub> (meV).  $l =$ hindered rotation,  $\rho_r =$ rocking,  $\delta =$ deformation,  $v$ =stretching, s =symmetric, d =degenerate, and X =substrate.

bands, are observed [Figs.  $4(a)$  and  $4(c)$ ]. By referring to the TDS results (Fig. 3), the  $NH_3$  multilayer is removed by the heating to 125 K, and there exist on the surface NH<sub>3</sub> hydrogen bonded to chemisorbed one and chemisorbed  $NH<sub>3</sub>$  molecule on the surface, corresponding to the 140- and 170-K desorption peaks, respectively. The 424-meV loss is decreased in intensity but still observed by heating to 125 K. This indicates that the loss is related to the (symmetric and) degenerate stretching mode not only of  $NH_3$  in the multilayer but also of  $NH_3$  hydrogen bonded to the chemisorbed molecule. Similarly, the 43-, 132-, 163-, and 207-meV losses are considered to have contribution from the vibrational modes of  $NH<sub>3</sub>$  hydrogen bonded to chemisorbed one. Further, the 424-meV loss has the intensity almost equal to that of the  $\sim$ 370meV loss. This may be related to the TDS result that the desorption peaks at 140 and 170 K have similar area intensities (Fig. 3). Vibrational losses associated with adsorbed  $NH_3$  were difficult to observe after heating to higher temperatures, because the population of adsorbed NH<sub>3</sub> was decreased and the losses related to the  $(\sqrt{3} \times \sqrt{3})R$  30°-B surface again predominated. The clean-surface spectrum was obtained after heating to  $> 200 K$ .

Figures 5(a) and 5(b) show EEL spectra in the specular mode with a higher primary energy of 12 eV of the  $(\sqrt{3} \times \sqrt{3})R30^{\circ}$ -B surface before and after 5-L NH<sub>3</sub> exposure at 90 K, respectively. Losses are observed at 2.0 eV for the  $(\sqrt{3} \times \sqrt{3})R30^{\circ}$ -B surface [Fig. 5(a)], and at 0.43 and 0.86 eV for the  $(\sqrt{3} \times \sqrt{3})R30^{\circ}$ -B surface exposed to  $NH_3$  [Fig. 5(b)]. By comparison with Fig. 4, the 0.43- and 0.86-eV losses are attributed to the stretching mode of  $NH_3$  in the multilayer and  $NH_3$  hydrogen bonded to chemisorbed one and the overtone (and multiple loss), respectively. The losses associated with the  $\sim$  370-



FIG. 5. EEL spectra in the specular mode of (a) the clean  $Si(111)(\sqrt{3}\times\sqrt{3})R30^\circ-B$  surface at 90 K; (b) and (c), the  $Si(111)(\sqrt{3}\times\sqrt{3})R30^{\circ}$ -B surface exposed to 5-L NH<sub>3</sub> and 5-L  $C_6H_6$  at 90 K, respectively.  $E_p = 12$  eV.

meV loss observed in Fig. 4 are not observed due to the weaker intensity compared with the 424-meV-loss intensity and also to the degraded resolution of the spectrometer (80-meV FWHM). Referring to the surface band dispersion of the  $(\sqrt{3} \times \sqrt{3})R30^\circ$ -B surface,<sup>4</sup> the 2.0-eV loss is attributed to a one-electron transition from an occupied backbond state to an empty dangling-bond state. Similar electronic transition was observed for the  $Si(111)(\sqrt{3}\times\sqrt{3})R30^{\circ}$ -Al surface.<sup>33</sup> It is noted that the 2.0-eV loss disappears upon  $NH<sub>3</sub>$  adsorption. This indicates that the initially empty dangling-bond orbital is involved in the  $NH_3$  chemisorption. The corresponding loss is still observed for  $\sim$  3 ML of C<sub>6</sub>H<sub>6</sub> physisorbed on the  $(\sqrt{3} \times \sqrt{3})R30^{\circ}$ -B surface (assuming that the saturation of the  $\alpha_1$  layer, which will be described later, corresponds to 1 ML) [Fig.  $5(c)$ ]. It is interpreted that the interaction of the dangling-bond orbital of the Si adatom with the lone-pair  $3a_1$  orbital of NH<sub>3</sub> lifts up an unoccupied antibonding orbital having mainly the danglingbond character, and the 2.0-eV loss disappears. The large Coulomb repulsion is expected when an electron is excited into the antibonding orbital having large overlap in space with the bonding orbital, which already accommodates two electrons. The loss corresponding to the electronic transition observed for the  $(\sqrt{3} \times \sqrt{3})R30^{\circ}$ -B surface at 2.0 eV may be merged into the substrate transitions or lifted to the energy higher than the loss-energy region measured.

Since  $NH_3$  is chemisorbed by donating lone-pair electrons to the Si adatom, it is considered that charge transferred to the 8 atom does not turn back to the Si adatom, different from the interaction of the Si(111)(7×7) surface with  $NH_3$ .<sup>34</sup>

The chemisorbed state of NH<sub>3</sub> on the  $(\sqrt{3} \times \sqrt{3})R30^\circ$ -8 surface is quite different from those on the  $Si(111)(7\times7)$  (Ref. 8) and  $Si(100)(2\times1)$  (Ref. 9) surfaces, where  $NH_3$  is dissociated to form  $SiNH_2$  and  $SiH$  species. The markedly decreased vibrational energies of the stretching modes of  $NH<sub>3</sub>$  chemisorbed on the  $(\sqrt{3} \times \sqrt{3})$ R 30°-B surface indicates that the N-H bonds are significantly weakened. If there were another active site appropriate for extracting H atom of  $NH<sub>3</sub>$  near the  $NH<sub>3</sub>$ -chemisorbed Si adatom,  $NH<sub>3</sub>$  would be dissociated with the molecularly chemisorbed state as a precursor state of the reaction. However, there is only a single kind of the Si adatom on the  $(\sqrt{3} \times \sqrt{3})R30^\circ$ -B surface, which is electrophile as mentioned above. Further, the distance between the neighboring Si adatoms is 6.7 A, which is larger than those between active sites on clean Si surfaces. These are the reasons why  $NH<sub>3</sub>$  is not dissociated on the  $(\sqrt{3} \times \sqrt{3})R30^\circ$ -B surface.

# C. Si(111)( $\sqrt{3} \times \sqrt{3}$ )R 30°-B/C<sub>6</sub>H<sub>6</sub>

TDS measurements after exposing the  $Si(111)(\sqrt{3}\times\sqrt{3})R30^\circ$ -B surface to C<sub>6</sub>H<sub>6</sub> at 95 K showed that  $C_6H_6$  was the only desorption product. Figure 6 shows TD spectra for the  $(\sqrt{3} \times \sqrt{3})R30^{\circ}$ -B surface exposed to an increasing amount of  $C_6H_6$  at 95 K (heating rate was 2.5 K/s). Three desorption peaks are observed



FIG. 6. TD spectra of  $C_6H_6$  (mass 78) from the  $Si(111)(\sqrt{3}\times\sqrt{3})R30^{\circ}$ -B surface exposed to various amounts of  $C_6H_6$  at 95 K. The heating rate was 2.5 K/s.

at 143, 150, and 158 K. No desorption peaks were observe '. at  $>$  200 K. The characteristic behavior of these peaks with increasing exposure is similar to that for benzene physisorbed on  $Si(111)(7\times7)$  (Ref. 10) and on some transition metals, e.g., on  $Ru(001).^{35}$  Thus, these peaks are attributed to desorption of  $C_6H_6$  physisorbed on the surface. The fact that only desorption of physisorbed  $C_6H_6$  is observed suggests that  $C_6H_6$  does not chemisorb on the  $(\sqrt{3} \times \sqrt{3})R30^{\circ}$ -B surface. The same conclusion has been obtained from EELS measurements as will be described later.

Following the labels used by Polta and Thiel,  $36$  the three peaks seen in Fig. 6 are associated with the  $\alpha_1$ (highest-temperature peak),  $\alpha_2$  (lowest temperature), and  $\alpha_3$  (intermediate temperature) layers. It is interesting to see that the desorption peak associated with the  $\alpha_1$  layer<br>is \_ observed for physisorbed  $C_6H_6$  on the physisorbed  $C_6H_6$  on the  $(\sqrt{3} \times \sqrt{3})R30^{\circ}$ -B surface, though the chemisorbed layer does not exist on the surface. The  $\alpha_1$  layer formed on  $Si(111)(7\times7)$  (Ref. 10) or transition metal<sup>35</sup> is the first physisorbed layer on top of the chemisorbed layer. Benzene in the  $\alpha_1$  layer on those surfaces interacts electrostatically with the chemisorbed molecule polarized by being  $\pi$  bonded to those surfaces, and desorbs at the temperature higher than those from the other two layers. It is interpreted that the first physisorbed  $C_6H_6$  on the  $(\sqrt{3} \times \sqrt{3})$ R 30°-B surface interacts electrostatically with the positively charged Si adatom in place of the polarized chemisorbed molecule.

Figure 7 shows EEL spectra in the specular mode of the  $(\sqrt{3} \times \sqrt{3})R\,30^\circ$ -B surface exposed to 1.5, 2.5, and 4.5-L  $C_6H_6$  at 90 K, respectively. Losses associated with adsorbed C<sub>6</sub>H<sub>6</sub> on the  $(\sqrt{3} \times \sqrt{3})$ R 30°-B surface are observed. The loss energies are close to those of gas-, liquid-, or solid-phase  $C_6H_6$ ,  $^{30}$  and are attributed to vibrational modes of physisorbed  $C_6H_6$  on the  $(\sqrt{3} \times \sqrt{3})R30^{\circ}$ -B surface. The assignments are summarized in Table III. The numbering system of Herzberg<sup>37</sup>



FIG. 7. EEL spectra in the specular mode of the  $Si(111)(\sqrt{3}\times\sqrt{3})R30^\circ$ -B surface exposed to various amounts of  $C_6H_6$  at 90 K: (a) 1.5 L, (b) 2.5 L, and (c) 4.5 L exposures.  $E_p = 7.5$  eV.

is used in the table. A loss associated with the SiH stretching mode and losses that can be ascribed to the vibrational modes of chemisorbed  $C_6H_6$  were not observed. By referring to the TDS results (Fig. 6), EEL spectra of Figs. 7(a), 7(b), and 7(c) are related to the  $\alpha_1$ ,  $\alpha_1 + \alpha_2$ , and  $\alpha_1+\alpha_3$  layers, respectively.

As the  $C_6H_6$  exposure is increased, the 86-meV-loss in-

TABLE III. Vibrational energies (meV) and their assignments of C<sub>6</sub>H<sub>6</sub> physisorbed on the Si(111)( $\sqrt{3}$ X $\sqrt{3}$ )R 30°-B surface at 90 K together with those of gas-, liquid-, or solid-phase  $C_6H_6$  (Ref. 30). The mode number and representation ( $D_{6h}$ ) symmetry) are included.

				Vibrational energies (meV)		
Mode number	Repr.	Mode	Gas-, liquid-, or solid-phase $C_6H_6$	$C_6H_6$ on $Si(111)$ $(\sqrt{3}\times\sqrt{3})R30^\circ\text{-B}$		
$v_{20}$	$(E_{2u})$	$\gamma_{\rm CC}$	51	57		
$v_4$	$(A_{2u})$	$\gamma$ <sub>CH</sub>	83	86		
$v_{11}$	$(E_{1g})$	$\gamma$ <sub>CH</sub>	105	108		
$v_2$	$(A_{1g})$	$\delta_{\rm CC}$	123			
$v_7$	$(B_{2g})$	$\gamma$ CH	123			
$v_{14}$	$(E_{1u})$	$\delta_{CH}$	129	128		
$v_{10}$	$(B_{2u})$	$\delta_{\rm CH}$	143			
$v_{17}$	$(E_{2g})$	$\delta_{\rm CH}$	146	146		
$v_3$	$(A_{2g})$	$\delta_{CH}$	164	170		
$v_{16}$	$(E_{2g})$	$v_{\rm CC}$	198	201		
$v_{15}$	$(E_{2g})$	$v_{\rm CH}$	378			
$v_1$	$(A_{1g})$	$v_{\rm CH}$	380			
$v_{12}$	$(E_{1u})$	$v_{\rm CH}$	380	383		
v <sub>5</sub>	$(B_{1u})$	$v_{CH}$	380			

 $\gamma$  = out-of-plane bending,  $\delta$  = in-plane bending, and  $\nu$  = stretching.

tensity is clearly increased in comparison with the intensities of the other losses of physisorbed  $C_6H_6$ . The offspecular measurements indicate that the 86-meV loss is predominantly excited by the dipole scattering. As shown in Table III, the loss is attributed to the out-ofplane CH bending mode, which has the dynamic dipole moment perpendicular to the carbon ring. Thus, it is considered that the mean angle with respect to the surface of the molecular plane of  $C_6H_6$  physisorbed on the  $(\sqrt{3} \times \sqrt{3})$ R 30°-B surface is largest in the  $\alpha_1$  layer and decreases with increasing exposure. The fact that the mean tilt angle for the  $\alpha_2$  layer is smaller than that for the  $\alpha_1$  layer is consistent with the TDS results: Contrary to the TD spectra observed for physisorbed benzene on  $Si(111)(7\times7)$  (Ref. 10) or transition metal,<sup>35</sup> the maximum area intensity of the  $\alpha_2$  desorption peak observed in Fig. 6 is smaller than that of the  $\alpha_1$  peak. Benzene physisorbed on Ru(001) (Ref. 35) [and possibly on Si(111)(7×7)] is oriented parallel to the surface in the  $\alpha_1$ layer, and tilted with a high mean angle in the  $\alpha_2$  layer. The orientation of  $C_6H_6$  physisorbed on the  $(\sqrt{3} \times \sqrt{3})$ R 30°-B surface shows an opposite trend. It is considered that the high mean tilt angle for  $C_6H_6$  in the  $\alpha_1$  layer is associated with the size of Si adatom, which is smaller than that of chemisorbed benzene.  $C_6H_6$  may be tilted (possibly standing nearly vertically) in order to interact electrostatically with the Si adatom in the most effective way.

Figure 5(c) shows an EEL spectrum in the specular mode of the  $(\sqrt{3} \times \sqrt{3})R30^\circ$ -B surface exposed to 5-L  $C_6H_6$  at 90 K. By referring to the TDS results, this exposure corresponds to the formation of  $\sim$ 3 ML of physisorbed  $C_6H_6$ . Losses are observed at 0.38, 0.76, 1.14, and 1.8 eV. The narrow losses at 0.38, 0.76, and 1.14 eV are attributed to the CH stretching mode and its overtones (and multiple losses) by comparing to Fig. 7. The loss feature rising sharply from  $\sim$  4 eV is attributed to an onset of the intramolecule  $\pi \rightarrow \pi^*$  electronic transition of  $C_6H_6^{38}$  The 2.0-eV loss observed for the The 2.0-eV loss observed for the  $(\sqrt{3} \times \sqrt{3})R30^\circ$ -B surface is lowered by 0.2 eV and is decreased in intensity upon  $C_6H_6$  adsorption. Possibly, the loss-energy shift of the electronic transition from the backbond to dangling-bond state may be owing to the electrostatic interaction of Si adatom with  $C_6H_6$  in the  $\alpha_1$ layer, and the decrease in intensity may be due to the adsorbed layer formed on  $(\sqrt{3} \times \sqrt{3})R30^\circ$ -B.

Benzene chemisorbs on the  $Si(111)(7\times7)$  (Ref. 10) and  $Si(100)(2\times1)$  (Ref. 11) surfaces, but does not on the  $Si(111)(\sqrt{3}\times\sqrt{3})R30^\circ$ -B surface. This is considered to be associated with the delocalization of  $\pi$  electrons involved in the reactions. As described in Sec. IIIB, the  $(\sqrt{3} \times \sqrt{3})$ R 30°-B surface is not suitable for the dissociative reaction of a gas molecule, and only the nondissociative adsorption is expected. If benzene were to bond to the Si adatom on the surface, the molecule would lose a part of the resonance energy  $(36 \text{ kcal/mol}^{39})$ . It is considered that the loss in the resonance energy is compensated by the saturation of the dangling bond(s) on  $Si(111)(7\times7)$  and  $Si(100)(2\times1)$  surfaces and benzene does chemisorb on these surfaces. On the other hand, the

dangling-bond level of the Si adatom on the  $(\sqrt{3} \times \sqrt{3})$ R 30°-B surface is initially empty. The bond formation between benzene with the adatom would compensate a loss in the resonance energy only insufficiently, and thus, benzene does not chemisorb on the  $(\sqrt{3} \times \sqrt{3})$ R 30°-B surface.

It is worth noting that the reactivity criterion of the  $(\sqrt{3} \times \sqrt{3})$ R 30°-B surface may be given by considering both absolute electronegativity and nature of orbitals of a molecule, which interacts with the surface.<sup>12</sup> Since the dangling-bond level of the Si adatom on the surface is empty, an electron donation from a molecule (or atom) to the adatom, i.e., an action as a Lewis base, should be required for the chemisorption on the  $(\sqrt{3} \times \sqrt{3})$ R 30°-B surface. It is possible to tell whether a molecule donates electron(s) to the adatom or not, by examining a quantity  $\chi$  called the absolute electronegativity. The definition of  $\chi$  is

$$
\chi = (I + A)/2 = -\mu , \qquad (8)
$$

where  $I$  is the ionization potential,  $A$  the electron affinity, and  $\mu$  the chemical potential.<sup>12</sup> If a molecule has a value of  $\chi$  smaller than that of the Si adatom, the molecule will donate the electron(s). Though the van der Waals interaction with the  $(\sqrt{3} \times \sqrt{3})R30^\circ$ -B surface may modify the value of  $\gamma$  of a molecule from that of the free one, the value of a free molecule gives a crude estimation. We assume tentatively  $\chi$  = 4.6 for the Si adatom, which is equal to the work function of the  $(\sqrt{3} \times \sqrt{3})R30^{\circ}$ -B surface.<sup>40</sup> Including the results obtained in this work, it has been found up to now that ammonia and potassium atom<sup>23,41</sup> chemisorb on the  $(\sqrt{3} \times \sqrt{3})R 30^{\circ}$ -B surface while atomic hydrogen,<sup>5</sup> oxygen,<sup>6</sup> and benzene do not. The values of  $\chi$ are 2.6, 2.4, 7.2, 6.3, and 4.1 for  $NH_3$ , K, H, O<sub>2</sub>, and  $C_6H_6$ , respectively.<sup>12</sup> The prediction based on the value of  $\chi$  for the chemisorption on the  $(\sqrt{3}\times\sqrt{3})R30^{\circ}$ -B surface is correct, except for  $C_6H_6$ . Benzene is a  $\pi$  base while the Si adatom has the accepting orbital of  $\sigma$  type. Thus, the reaction between them is considered to be unfavorable. According to the above discussion, for example, it is expected that H<sub>2</sub>O ( $\chi$ =3.1)<sup>12</sup> chemisorbs on the  $(\sqrt{3} \times \sqrt{3})R30^{\circ}$ -B surface, while CO ( $\pi$  acid and  $\gamma$  = 6. 1)<sup>12</sup> does not. For reactions on clean Si surfaces, it is unfortunately impossible to apply this simple criterion. Since the dangling-bond levels of the clean surfaces are partially filled, electron(s) may fiow both from a molecule to the surface Si atom and in the opposite direction upon chemisorption.

## IV. SUMMARY

We have studied the adsorbed states of  $NH_3$  and  $C_6H_6$ on the  $Si(111)(\sqrt{3} \times \sqrt{3})R30^\circ-B$  surface by using thermal-desorption spectroscopy and both vibrational and electronic electron-energy-loss spectroscopy.

(1) There exist  $NH_3$  in the multilayer,  $NH_3$  hydrogen bonded to a chemisorbed  $NH_3$ , and chemisorbed  $NH_3$  on the  $(\sqrt{3} \times \sqrt{3})$ R 30°-B surface at 90 K. These desorb at  $\sim$  115, 140, and 170 K, respectively. The N-H bonds of chemisorbed  $NH<sub>3</sub>$  are significantly weakened compared with those of free  $NH_3$ . Upon  $NH_3$  adsorption, the 2.0eV electronic transition from the backbond state to the dangling-bond state of the Si adatom on the  $(\sqrt{3} \times \sqrt{3})R30^\circ$ -B surface is removed. It is proposed that chemisorbed  $NH<sub>3</sub>$  is coordinated to the Si adatom on the surface by the interaction of the lone-pair orbital with originally empty dangling-bond orbital of the Si adatom. The chemisorbed state of NH<sub>3</sub> on the  $(\sqrt{3} \times \sqrt{3})R30^\circ$ -B surface is significantly different from those on clean Si surfaces, which have been studied, and the reason is discussed.

(2)  $C_6H_6$  physisorbs on the  $(\sqrt{3}\times\sqrt{3})R30^\circ$ -B surface at  $<160$  K. The physisorbed layer consists of three different layers.  $C_6H_6$  in the  $\alpha_1$  layer first physisorbed on the surface has the highest mean tilt angle and interacts with the Si adatom electrostatically. Contrary to the interactions with clean Si surfaces,  $C_6H_6$  does not chemisorb on the  $(\sqrt{3} \times \sqrt{3})R30^\circ$ -B surface, and the reason is discussed. We have shown that the concept of the absolute electronegativity can be applied to the interactions of

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the  $(\sqrt{3} \times \sqrt{3})R$  30°-B surface with molecules or atoms.

(3) We have also analyzed the electron-energy-loss spectra of the  $(\sqrt{3} \times \sqrt{3})R30^\circ$ -B surface by the use of the dipole scattering theory, and the information on the free carriers, which was introduced by the high-temperature heating in the preparation of the surface, was obtained. For the  $(\sqrt{3} \times \sqrt{3})$ R 30°-B surface having few defects, the carrier density of  $7 \times 10^{19}$  cm<sup>-3</sup> and Drude relaxation time of  $5.5 \times 10^{-15}$  s in the vicinity of the surface were evaluated by the analysis of the plasmon loss, and the total number of  $1 \times 10^{15}$  cm<sup>-2</sup> was estimated by that of the quasielastic peak.

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