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## Superconductivity and Madelung potential of $YBa_2Cu_3O_{6+x}$ ordered superstructures

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The Madelung potential of each possible ordered superlattice of YBa<sub>2</sub>Cu<sub>3</sub>O<sub>6+x</sub> is calculated as the oxygen content x varies from zero to 1.  $\Delta V_A$ , the difference in Madelung site potential between apex and in-plane (CuO<sub>2</sub>) oxygen atoms, is presented as a function of the composition x. The correlation of  $T_c$  and x is discussed in terms of Ohta's empirical curve of  $T_c$  vs  $\Delta V_A$ . The theoretical results suggest that the superconductivity depends on the superlattice structure.

The physical properties, resistivity, magnetic susceptibility, and the temperatures of the structural phase transition and of the superconducting transition of the high- $T_c$ oxide superconductors are quite sensitive to oxygen stoichometry, and specifically to the positions of the oxygen atoms.<sup>1,2</sup> In YBa<sub>2</sub>Cu<sub>3</sub>O<sub>6+x</sub>, T<sub>c</sub> depends on x. Figure 1 shows the change of  $T_c$  with the oxygen content. On the other hand, corresponding structures that differ in the arrangement of oxygen atoms in the  $CuO_x$  planes have been reported to exist for different values of x at low temperatures.<sup>3</sup> The reported superstructures were the tetragonal phase for x = 0, the orthorhombic phase for x = 1,<sup>4</sup> and the orthorhombic with a double unit cell for  $x = \frac{1}{2}$  5.6 More complex superstructures were observed for  $x = \frac{7}{8}, \frac{7.8}{7}$  x near  $\frac{3}{5}, \frac{9}{7}$  and x near  $\frac{2}{3}$ .<sup>10</sup> A recently observed structure is the tetragonal phase for  $x = \frac{3}{8}$ .<sup>11</sup> For  $x = \frac{1}{4}$  and  $\frac{3}{4}$ , a set of superstructures have been proposed.<sup>12</sup> To establish a relationship between the various observed types of oxygen ordering in the Cu-O basal plane and their superconductivities, a variable  $\Delta V_A$  defined within an ionic model<sup>13</sup> is used.  $\Delta V_A$  equals the difference between  $V_{O(A)}$  and  $V_{O(P)}$ , where  $V_i$  [i=O(A), O(P)] is the Madelung site potential for a given charge distribution of corresponding superstructure. O(A) and O(P) denote the apex and in-plane (CuO<sub>2</sub>) oxygen atoms, respectively.

Several experimentally observed and theoretically possible superstructures at low temperature in YBa<sub>2</sub>Cu<sub>3</sub>O<sub>6+x</sub> systems with different x are shown in Fig. 2, where the distributions of oxygen atoms and oxygen vacancies in the Cu-O basal plane is exhibited. The valence of Cu in the plane is assigned in terms of the following rules:<sup>14</sup>

O-Cu-O, fourfold coordination Cu with a valence of +2,

 $\Box$ -Cu- $\Box$ , twofold coordination Cu with a valence of +1,

O-Cu- $\Box$ , threefold coordination Cu with a valence of +2.

Therefore, the valence of all ions of each superstructure in the  $YBa_2Cu_3O_{6+x}$  system is given for different x with the condition of compound charge neutrality (see Fig. 2):

(1) 
$$YBa_2Cu_3O_7 Y^{3+} + 2Ba^{2+}O^{2-} + 2Cu^{2+}O^{1.75-} + Cu^{2+} + O^{2-}$$
  
(2)  $YBa_2Cu_3O_{6.875}$  (a)  $8Y^{3+} + 16Ba^{2+}O^{2-} + 16Cu^{2+}O^{1.78125-} + 7Cu^{2+} + Cu^{1+} + 7O^{2-}$   
(b)  $8Y^{3+} + 16Ba^{2+}O^{2-} + 16Cu^{2+}O^{1.75-} + 6Cu^{2+} + 2Cu^{1+} + 7O^{2-}$   
(c)  $YBa_2Cu_3O_{6.75}$  (a)  $4Y^{3+} + 8Ba^{2+}O^{2-} + 8Cu^{2+}O^{1.8125-} + 3Cu^{2+} + Cu^{1+} + 3O^{2-}$   
(d)  $YBa_2Cu_3O_{6.67} 3Y^{3+} + 6Ba^{2+}O^{2-} + 6Cu^{2+}O^{1.83334-} + 2Cu^{2+} + Cu^{1+} + 2O^{2-}$   
(f)  $YBa_2Cu_3O_{6.6} 5Y^{3+} + 10Ba^{2+}O^{2-} + 10Cu^{2+}O^{1.85-} + 3Cu^{2+} + 2Cu^{1+} + 3O^{2-}$   
(g)  $YBa_2Cu_3O_{6.5} 2Y^{3+} + 4Ba^{2+}O^{2-} + 4Cu^{2+}O^{1.875-} + Cu^{2+} + Cu^{1+} + O^{2-}$   
(h)  $8Y^{3+} + 16Ba^{2+}O^{2-} + 16Cu^{2+}O^{1.90625-} + 3Cu^{2+} + 5Cu^{1+} + 3O^{2-}$   
(k)  $8Y^{3+} + 16Ba^{2+}O^{2-} + 16Cu^{2+}O^{2-} + 6Cu^{2+} + 2Cu^{1+} + 3O^{2-}$   
(k)  $8Y^{3+} + 16Ba^{2+}O^{2-} + 8Cu^{2+}O^{2-} + 3Cu^{2+} + 2Cu^{1+} + 3O^{2-}$   
(k)  $8Y^{3+} + 16Ba^{2+}O^{2-} + 8Cu^{2+}O^{2-} + 6Cu^{2+} + 2Cu^{1+} + O^{2-}$   
(g)  $YBa_2Cu_3O_{6.25}$  (a)  $4Y^{3+} + 8Ba^{2+}O^{2-} + 8Cu^{2+}O^{2-} + 2Cu^{2+} + 2Cu^{1+} + O^{2-}$   
(h)  $4Y^{3+} + 8Ba^{2+}O^{2-} + 8Cu^{2+}O^{2-} + 2Cu^{2+} + 2Cu^{1+} + O^{2-}$   
(g)  $YBa_2Cu_3O_6 Y^{3+} + 2Ba^{2+}O^{2-} + 2Cu^{2+}O^{2-} + Cu^{2+}$ 

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FIG. 1. Correlation of superconducting transition temperature  $T_c$  and composition x in YBa<sub>2</sub>Cu<sub>3</sub>O<sub>6+x</sub>, derived from Ref. 1.

The structural type and hole densities are listed in Table I.

We assume the charge on each ion is distributed as a point charge, i.e., ionic model. Such an ionic approximation has been shown<sup>15</sup> to be complementary to normal band-structure calculations. More accurate calculations will improve and go beyond this model, but the essential physics will remain.<sup>16</sup> Because the compounds under study are ionic crystals and the most important factors to determine the hole distribution are the Madelung energy, the ionization energy of cations, and the electron affinity of anions, the ionic model is appropriate for calculating the site potential of this compound, although covalency may also play an important role in some cases.<sup>17</sup> For each superstructure and corresponding valence distribution the Madelung potentials of each ion in the unit cell are calculated by use of the standard Ewald method. The Madelung potentials are listed in Table II. For the case when all the ordered states of  $YBa_2Cu_3O_{6+x}$  (0 < x < 1) are the orthorhombic superstructures, the curve of  $\Delta V_A$ versus concentration x is sketched in Fig. 3. Using Ohta's empirical curve<sup>13</sup> showing  $T_c = 0$  when  $\Delta V_A < -2$  (eV), the curve of  $T_c$  vs x is obtained and shown in Fig. 4. As a result, it is predicted that the Y system is superconducting for the whole range of 0 < x < 1. For the case when the ordered states of YBa2Cu3O6.75, YBa2Cu3O6.375, and YBa<sub>2</sub>Cu<sub>3</sub>O<sub>6.25</sub> are taken as tetragonal superstructures, and the others as still orthorhombic, the curve of  $\Delta V_A$  vs x in Fig. 5 is obtained. The function of  $T_c$  vs x that results is shown in Fig. 6 by use of the same empirical curve of  $T_c$ vs  $\Delta V_A$ .<sup>13</sup> From Fig. 6 it is clear that  $T_c$  decreases as x decreases and finally approaches zero at about x = 0.4, while the corresponding hole density in the plane is zero

FIG. 2. Several possible superstructures at low temperature in the YBa<sub>2</sub>Cu<sub>3</sub>O<sub>6+x</sub> system: •, copper;  $\bigcirc$ , oxygen;  $\square$ , oxygen vacancy; the lines show the superstructural unit cell. (1) YBa<sub>2</sub>Cu<sub>3</sub>O<sub>7</sub>; (2) YBa<sub>2</sub>Cu<sub>3</sub>O<sub>6.875</sub>; (3) YBa<sub>2</sub>Cu<sub>3</sub>O<sub>6.75</sub>; (4) YBa<sub>2</sub>-Cu<sub>3</sub>O<sub>6.67</sub>; (5) YBa<sub>2</sub>Cu<sub>3</sub>O<sub>6.6</sub>; (6) YBa<sub>2</sub>Cu<sub>3</sub>O<sub>6.5</sub>; (7) YBa<sub>2</sub>Cu<sub>3</sub>-O<sub>6.375</sub>; (8) YBa<sub>2</sub>Cu<sub>3</sub>O<sub>6.25</sub>; (9) YBa<sub>2</sub>Cu<sub>3</sub>O<sub>6</sub>.



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FIG. 3. Curve of  $\Delta V_A$  vs concentration x when all ordered states are orthorhombic.



FIG. 4. Curve of  $T_c$  vs concentration x for all orthorhombic states.



FIG. 5. Curve of  $\Delta V_A$  vs concentration x when part of the superstructures are tetragonal and others are still orthorhombic.



FIG. 6. Curve of  $T_c$  vs concentration x for the second case.

TABLE I. Supe	erstructure ty	pe and car	rier densit	y in	$YBa_2Cu_3O_{6+x}$	systems.
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Compound	Superstructure type	Hole density in CuO <sub>2</sub>	Electron density in basal plane
YBa <sup>2</sup> Cu <sub>3</sub> O <sub>7</sub>	1×1 orthorhombic	1.0	0.0
YBa <sup>2</sup> Cu <sup>3</sup> O <sub>6 875</sub>	(a) 8×1 orthorhombic	0.875	0.125
	(b) $2\sqrt{2} \times 2\sqrt{2}$ tetragonal	1.0	0.25
YBa <sup>2</sup> Cu <sup>3</sup> O <sub>6</sub> 75	(a) 4×1 orthorhombic	0.75	0.25
	(b) 2×2 tetragonal	0.5	0.0
YBa <sub>2</sub> Cu <sub>3</sub> O <sub>6.67</sub>	$3 \times 1$ orthorhombic	0.67	0.33
YBa <sub>2</sub> Cu <sub>3</sub> O <sub>6.6</sub>	5×1 orthorhombic	0.6	0.4
YBa <sub>2</sub> Cu <sub>3</sub> O <sub>6.5</sub>	2×1 orthorhombic	0.5	0.5
YBa <sub>2</sub> Cu <sub>3</sub> O <sub>6,375</sub>	(a) 8×1 orthorhombic	0.375	0.625
	(b) $2\sqrt{2} \times 2\sqrt{2}$ tetragonal	0	0.25
YBayCu3O6 25	(a) 4×1 orthorhombic	0.25	0.75
	(b) $2 \times 2$ tetragonal	0	0.5
YBa <sub>2</sub> Cu <sub>3</sub> O <sub>6</sub>	1×1 tetragonal	0	1.0

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Compounds	Structural type	V <sub>O(A)</sub> <sup>a</sup> (a.u.)	V <sub>O(P)</sub> <sup>a</sup> (a.u.)	$\Delta V_A = V_{O(A)}^{a} - V_{O(P)}^{a} \text{ (eV)}$
YBa <sub>2</sub> Cu <sub>3</sub> O <sub>7</sub>	Orthorhombic	-0.612	-0.897	7.70
<b>YB</b> a <sub>2</sub> Cu <sub>3</sub> O <sub>6.875</sub>	Orthorhombic	-0.630	-0.872	6.28
	Tetragonal	-0.476 <sup>b</sup>	-0.883	
<b>YBa</b> 2Cu3O6.75	Orthorhombic	-0.658	-0.850	5.18
	Tetragonal	-0.712	-0.802	2.43
<b>YBa</b> <sub>2</sub> Cu <sub>3</sub> O <sub>6.67</sub>	Orthorhombic	-0.687	-0.834	3.94
YBa <sub>2</sub> Cu <sub>3</sub> O <sub>6.6</sub>	Orthorhombic	-0.683	-0.818	3.65
YBa <sub>2</sub> Cu <sub>3</sub> O <sub>6.5</sub>	Orthorhombic	-0.694	-0.801	2.91
YBa <sub>2</sub> Cu <sub>3</sub> O <sub>6,375</sub>	Orthorhombic	-0.720	-0.780	1.62
	Tetragonal	-0.762	-0.535	-6.13
<b>YBa</b> <sub>2</sub> Cu <sub>3</sub> O <sub>6.25</sub>	Orthorhombic	-0.728	-0.753	0.68
	Tetragonal	-0.974	-0.735	-6.45
Y BayCu3O6	Tetragonal	-0.775	-0.706	-1.86

TABLE II. Madelung potentials of apex and in-plane oxygen atoms.

<sup>a</sup>Average value over different sites in the superstructural unit cell.

<sup>b</sup>This value is more than -0.5 (a.u.), the critical value of the Madelung potential for a stable  $O^{2^-}$  ion, leading to unstable structure. Therefore, the corresponding  $\Delta V_A$  is excluded in this model.

but the in-plane electron density is not zero as given in the Table I. This is consistent with  $YBa_2Cu_3O_{6+x}$  being a hole-type superconductor. Comparing Figs. 4 and 6, we can conclude that the theoretical prediction in Fig. 6 is in much better agreement with the experimental results<sup>1</sup> in Fig. 1. It is further concluded that the structures of  $YBa_2Cu_3O_{6.375}$  and  $YBa_2Cu_3O_{6.25}$  systems are tetragonal and those systems have valence distributions that lead to zero hole density in the CuO<sub>2</sub> planes.

Ohta, Tohyama, and Maekawa<sup>13</sup> thought that the energy splitting  $\Delta E_z$  between the in-plane (CuO<sub>2</sub>) and out-ofplane 2p-3d transition correlates with  $T_c$ , in particular, with the one-body energy level splittings  $\Delta \epsilon_A$  [for the  $P_z$ orbital of O(A) relative to  $P_\sigma$  orbital of O(P)] and  $\Delta \epsilon_d$ (for the  $3z^2 - r^2$  orbital of Cu 3d relative to its  $x^2 - y^2$  orbital). Madelung potentials are screened due to core polarization to give, e.g., the energy levels  $\Delta \epsilon_A = \Delta V_A / \epsilon(\infty)$ , where  $\epsilon(\infty)$  is the dielectric constant at optical frequencies. Crystal-field theory suggests that  $\Delta \epsilon_d$  may be linked with  $\Delta V_A$ . Thus, a larger  $\Delta V_A$  yields a larger  $\Delta \epsilon_A$  and  $\Delta \epsilon_d$ , leading to the larger  $\Delta E_z$ . Covalency between the  $P_z$ and  $3z^{2}-r^{2}$  orbitals may be suppressed due to the splitting  $\Delta \epsilon_A$ , which also enhances  $\Delta E_z$ . It seems clear that there is a strong correlation between the  $\Delta V_A$  and the superconducting temperature; in other words, any parameters or factors changing  $\Delta V_A$  will influence  $T_c$ . We have shown that the relationship between  $T_c$  and concentration x in the YBa<sub>2</sub>Cu<sub>3</sub>O<sub>6+x</sub> system can be interpreted by using a empirical curve of  $\Delta V_A$  and  $T_c$ . Recently, Ledbetter and Lei<sup>18</sup> presented a direct relationship between  $\Delta V_A$  and  $T_c$  is worth studying further theoretically and experimentally.

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